Microwave-Hydrothermal Synthesis of Nb-Doped BaTiO₃ Nanoparticles under Various Conditions¹

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Abstract—Tetragonal BaTiO₃ powders were synthesized by the microwave-hydrothermal (MH) method. The effects of the reaction time and temperature on the MH synthesis were investigated. Typical experiments were performed with a solid state reaction for the system BaTiO₃ + Nb₂O₃ at a high temperature. In the MH method, at a shorter time and lower temperature, the rate of the formation of tetragonal BaTiO₃ and of the growth of particles went down substantially. The microwave heating can be very fast and uniform through a self-heating process that results from the direct absorption of the microwave energy in the reaction mixture. The synthesis of Nb-doped BaTiO₃ has been investigated under MH conditions in the temperature range of $120-180^{\circ}$ C for 1-3 hours using $C_{16}H_{36}O_4$ Ti, BaH₂O₂ · 8H₂O, and NbCl₅ as Ba, Ti and Nb sources, respectively. In the phase evolution studies, the X-ray diffraction measurements and Raman spectroscopy were employed. The Transmission Electron Microscope and the Field Emission Scanning Electron Microscopy images were taken for the detailed analysis of the grain size, surface, and morphology. The MH method was applied for the synthesis of Nb-doped BaTiO₃ via provided advantages of the fast crystallization and decreased crystallite size. The nucleation rate should have been high because of a high heating rate in microwave heating processes.

Keywords: ceramics, chemical synthesis, microwave-hydrothermal synthesis, crystal structure, perovskites **DOI**: 10.3103/S1068375515050063

INTRODUCTION

Perovskite-type oxides have the general formula of ABO₃ in which A is a rare earth or alkaline earth metal and B is a transition metal. Applications of this group of materials in electrochemistry [1, 2], oxygen separation membranes [3], chemical sensors for the detection of humidity [4], alcohol [5], gases such as oxygen [6] and hydrocarbon [7], and of nitric oxide [8] for over 30 years are the basis for understanding their mixed conductivity by both ion and electron migrations and their high nonstoichiometric composition. Since there is a strong dependence of ferroelectric properties on the grain size and compositional aspects, the microstructural control has become very important [9]. BaTiO₃-based positive temperature co-efficient (PTC) thermistors are widely used as over-current protectors, motor starters for refrigerators and air conditioners, temperature sensors, and so on [10-14]. The hydro-thermal method is attractive for synthesizing BaTiO₃ powder because the combined effects of the solvent, temperature, and pressure on the ionic reaction equilibrium can stabilize desirable products while inhibiting formation of undesirable compounds. Hydro-thermal synthesis also makes it possible to pre-

It has been reported [15, 16] that the nanostructural and dielectric features of nano-sized Nb-doped BaTiO₃ powders vary sensitively with their size and preparation conditions. To the best of our knowledge, there are few reports on the doped BaTiO₃ metal-semiconductor transition on Nb doping, in particular. Sintered bulk ceramics of Nb-doped BaTiO₃ show the semiconducting behavior when doped with a small amount of Nb, but when Nb concentration exceeds 0.4-0.5 mol %, the specimens revert to the insulator state. The main reason for the increase of resistivity when the donor concentration exceeds a certain value

pare BaTiO₃ powder in a single processing step and does not require elaborate apparatuses or expensive reagents However, just like in most convectional hydro-thermal processes, BaTiO₃ powders prepared by the microwave-hydrothermal (MH) synthesis at temperatures below 200°C were cubic or pseudo cubic, though the tetragonal structure is thermodynamic, stable at room temperature. To convert hydrothermal BaTiO₃ from the cubic to tetragonal shape, heating at temperatures over 1000°C is required, which always leads to the grain growth and particle aggregation. Therefore, a direct generation of tetragonal BaTiO₃ through the hydrothermal process is of a considerable interest, both in science and in industry [13–15].

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is the fact that the compensation mechanism changes from one that mainly involves electrons to one that primarily involves cation vacancies [15-18]. In the present paper, we analyze the effect of incorporating different amounts of Nb on the microstructural development of BaTiO₃. A particular emphasis is placed in the morphology and composition of secondary phases.

Compared to the traditional solid-state reaction and other wet-chemical routes (including sol-gel processing, oxalate route, and homogeneous precipitation), the hydrothermal method is low-cost and convenient to prepare BaTiO₃ nanoparticles without a following high temperature calcination process [16–21]. In order to produce powders of a desired size and shape, several researchers investigated a variety of hydrothermal conditions, such as the pH value, Ba/Ti ratio, reaction temperature as well as various precursors [21-25]. Though many efforts have been pulled to prepare BaTiO₃ particles, still there is insufficient knowledge about the effects of doping and solvent type on the size and shape under hydrothermal conditions. A number of authors [21-25] have synthesized BaTiO₃ by the MH method at T below 200°C but the MH processing of Nb-doped BaTiO₃ has not been reported so far with other authors. We succeeded in our initial efforts to synthesis of Nb doped BaTiO₃ in the temperature of 150°C for only 2 hours [26]. Therefore in this work, we have investigated the effect of incorporating Nb on the properties of BaTiO₃ via the MH conditions in the temperature range of 120–180°C for 1–3 hours.

EXPERIMENTAL

The MH reactions were performed to synthezise Nb-doped BaTiO₃. For more details, please, refer to our earlier manuscript [26]. The system was heated in the temperature range of 120–180°C for 1–3 hours. The obtained powders were characterized by the X-ray diffraction (XRD), with the Cu K_{α} radiation in the 20 range from 20° to 60° and Raman spectroscopy (spectral range 80–3500 cm⁻¹). Micro-structural characterization was performed by the field emission scanning electron microscopy (FE-SEM) and transmission electron microscopy (TEM).

RESULTS AND DISCUSSION

Figure 1 shows the XRD patterns for the MH Nb-doped BaTiO₃ powders prepared in the temperature range of $120-180^{\circ}$ C for 1-3 hours. From the XRD patterns it can be considered that the sample prepared at 120° C for 1 h (MH-120-1) forms a pattern similar to an amorphous product, but when the time and temperature rise, the BaTiO₃ powder becomes a structurally uniform crystalline with high crystallinity. Except the sample prepared at 120° C for 2 h (MH-120-2), the nanoparticles were pure perovskite Nb-BaTiO₃, without some intermediate carbonate phase that was observed at $2\theta 28^{\circ}$ and 34° in the sample MH-120-2 [16]. The presence of BaCO₃ can be accounted for either by an incomplete reaction or by the presence of carbonate in the Ba alkali source, or by the reaction of CO₂ in air [14].

It is clearly seen that with the enhancement of time and temperature, the maximum of the peak shifts towards higher 2 θ values and the intensity of the peaks grows. According to the investigations of some researchers [28, 29], the shift of peaks can be related to the displacements of barium. In the tetragonal phase, the optimization of the crystal geometry of BaTiO₃ after doping with Nb is attained by the impurity-outward displacements of barium. The movements are mainly in the *ab* plane while the motion along the *c* axis is practically negligible. In the sample synthesis at the highest time and temperature, we have the highest intensity.

When the size of BaTiO₃ powders is below 1 μ m, the full-width at half-maximum (FWHM) of the corresponding (110) XRD peaks becomes larger and such broadening leads to the overlapping of two peaks located close to each other [14, 23, 27]. The structural changes associated with the phase transition usually have a large effect on the Raman spectrum. The XRD studies produce data that are consistent with an increasingly cubic structure at smaller particle sizes, not distinguishing between an average and a local structure. Thus, the XRD patterns reveal the characteristics of cubic BaTiO₃ without certain peak splitting while the Raman results have supported the existence of a tetragonal symmetry at small dimensions, even though it was not discernible by XRD. The Raman spectra for the obtained Nb-BaTiO₃ ceramic powder samples are presented in Fig. 2. The peaks at 185, 305, 518 and 715 cm⁻¹ are characteristics of the tetragonal BaTiO₃. The peaks at 191 and 392 cm⁻¹ are assigned to $TiO_{2}[30-33].$

The dependence of the degree of the tetragonal fraction on the particle size has been extensively investigated for BaTiO₃ [23, 30]. A variety of explanatory models were proposed and researchers reported that the tetragonal distortion decreased when the particle size was below 1 μ m [23, 24]. Thus, a reasonable explanation for a larger tetragonality in the sample synthesized at 180°C is the fact that the particles grow faster under MH conditions and at higher temperatures.

The FE-SEM images of Nb-BaTiO₃ powders prepared at 150 and 180°C for 2 h are shown in Figs. 3 and 4, respectively. It is evident that the powder prepared at 150°C consists of ultrafine, well dispersed uniform particles with the average size of 45 nm, the particle size being distributed from 40 to 50 nm. Whereas for the powders synthesized at 180°C, the average particle size increases substantially to the average size of



Fig. 1. XRD patterns of Nb-BaTiO₃ nanoparticles at different time and temperature: (Up) Zoom in (110) and (200) planes (Down).

90 nm, with the particle size distribution being from 70 to 110 nm.

In the MH process, the microwave radiation can be absorbed by $BaTiO_3$ particles and smaller $BaTiO_3$ particles will dissolve more quickly. According to the dissolution/recrystallization mechanism, this will lead to larger particles growing faster with time and temperature [23, 32, 33].

The MH synthesis leads to the homogeneous nucleation of a large number of tiny stable Nb-BT particles, which further grow uniformly; the primary Nb-BT particles form agglomerates [24, 28]. The crystallite size of the particles was calculated with the XRD using the Scherrer equation as in [35]:

$$D_{\rm v} = \frac{K\lambda}{B_{\rm size}\cos\theta},\tag{1}$$

where D_v is the crystallite size, K is a constant whose value is 0.9, λ is the X-ray wavelength; the width of the peak, B_{size} , was determined as the full width at half-maximum. The crystallite size was 52 and 95 nm for the Nb-BaTiO₃ powder prepared at 150 and 180°C for 2 hours, respectively. These results are in good agreement with the FE-SEM data, too.



Fig. 2. Raman spectra of hydrothermal nanosized Nb-BaTiO₃ powders.

The TEM images taken to estimate the size of the particles and the morphology of the particles for powders prepared at 150 and 180°C for 1 h are shown in Figs. 5 and 6, respectively. The decreased crystal size in MH-150-2 might be related to a high heating rate in MH processes. At the microwave frequency of 2.45 GHz, the changes of plus/minus in the waveguide occur 2450 million times in 1 s. Depending on the characteristics of the dielectric, it tends to modify the alternating electric field through structural changes. When a water sample is placed in a reactor and subjected to an alternating electric field, the electric deviation (electric dipole) of the water molecule tends to follow the alternating electric field. However, since water consists of clusters of water molecules through hydrogen bonding, not all can follow the alternating electric field. The result is the dielectric heating of water by the irradiating microwaves. Moreover, as for



Fig. 3. Nb-BaTiO₃ powder prepared at 150°C for 2 h.



Fig. 4. Nb-BaTiO₃ powder prepared at 180° C for 2 h.



Fig. 5. Nb-BaTiO₃ powders prepared at 150°C for 1 h.



Fig. 6. Nb-BaTiO₃ powders prepared at 180°C for 1 h.

the case of ions in solution, the Joule heating takes place by space-charge polarization. When microwave heating, an electrolyte-water solution dielectric heating, and the Joule heating occur simultaneously, compared with pure water, then exothermic efficiency becomes remarkably high. Accordingly, dielectric heating by orientation polarization of water and resistance heating by the Joule process are enhanced in electrolyte—water media. This rapid homogeneous heating, not possible by convectional heating, provides a uniform nucleation [36].

CONCLUSIONS

The present investigation shows that nanosized particles of Nb-BaTiO₃ powder can be crystallized via a rapid and cost-effective MH process without the temperature gradient and uneven nucleation and growth of particles of various sizes. The MH can be very fast and uniform through a self-heating process that arises from the direct absorption of the microwave energy in the reaction mixture. When microwave heating an electrolyte-water solution dielectric heating and the Joule heating occur simultaneously compared with pure water, and thus exothermic efficiency becomes remarkably high. Accordingly, dielectric heating by orientation polarization of water and resistance heating by the Joule process are enhanced in electrolyte-water media. This rapid homogeneous heating, not possible by convectional heating, provides a uniform nucleation. Because of a homogeneous nucleation, uniformly-sized particles can be prepared. But according to the dissolution/recrystallization mechanism, both the extent of tetragonality and the particle size increase quickly with time and temperature. When the reaction temperature goes down, the formation of the tetragonal structure and growth of particles are slowed down substantially, showing a critical effect of the reaction temperature on the MH processing of tetragonal BaTiO₃.

REFERENCES

- Kharton, V.V., Yaremchenko, A.A., and Naumovich, E.N., Research on the electrochemistry of oxygen ion conductors in the Former Soviet Union, II: Perovskiterelated oxides, *J. Solid State Electrochem.*, 1999, vol. 3, no. 6, pp. 303–326.
- Bi, Z., Cheng, M., Dong, Y., Wu, H., She, Y., and Yi, B., Electrochemical evaluation of La_{0.6}Sr_{0.4}CoO₃-La_{0.45}Ce_{0.55}O₂ composite cathodes for anode-supported La_{0.45}Ce_{0.55}O₂-La_{0.9}Sr_{0.1}Ga_{0.8}Mg_{0.2}O_{2.85} bilayer electrolyte solid oxide fuel cells, *Solid State Ionics*, 2005, vol. 176, nos. 7–8, pp. 655–661.
- Takamura, H., Enomoto, K., Aizumi, Y., Kamegawa, A., and Okada, M., Preparation and oxygen permeability of Pr–Al-based perovskite-type oxides, *Solid State Ionics*, 2004, vol. 175, nos. 1–4, pp. 379–382.
- Holc, J., Slunečko, J., and Hrovat, M., Temperature characteristics of electrical properties of (Ba,Sr)TiO₃ thick film humidity sensors, *Sensors Actuators, Ser. B*, 1995, vol. 26, no. 1–3, pp. 99–102.
- 5. Kong, L.-B. and Shen, Y.-S., Gas-sensing property and mechanism of Ca_xLa_{1-x}FeO₃ ceramics, *Sensors Actuators, Ser. B*, 1996, vol. 30, no. 3, pp. 217–221.
- Lukaszewicz, J.P., Miura, N., and Yamazoe, N., A LaF₃-based oxygen sensor with perovskite-type oxide electrode operative at room temperature, *Sensors Actuators, Ser. B*, 1990, vol. 1, nos. 1–6, pp. 195–198.
- Brosha, E.L., Mukundan, R., Brown, D.R., Garzon, F.H., Visser, J.H., Zanini, M., Zhou, Z., and Logothetis, E.M., CO/HC sensors based on thin films of LaCoO₃ and La_{0.8}Sr_{0.2}CoO_{3-δ} metal oxides, *Sensors Actuators, Ser. B*, 2000, vol. 69, nos. 1–2, pp. 171–182.
- Traversa, E., Matsushima, S., Okada, G., Sadaoka, Y., and Sakai, Y., Watanabe: NO₂ sensitive LaFe₃ thin films prepared by r.f. sputtering, *Sensors Actuators, Ser. B*, 1995, vol. 25, no. 1, pp. 661–664.
- Vijatović, M.M., Bobić, J.D., and Stojanović, B.D., History and challenges of barium titanate, Part I, *Sci. Sintering*, 2008, vol. 40, pp. 155–165.
- Kim, H.T. and Han, Y.H., Sintering of nanocrystalline BaTiO₃, *Ceram. Int.*, 2004, vol. 30, no. 7, pp. 1719– 1723.
- 11. Bomlai, P., Effects of surrounding powder in sintering process on the properties of Sb and Mn-doped barium strontium titanate PTCR ceramics, *Songklanakarin J. Sci. Technol.*, 2006, vol. 28, no. 3, pp. 669–675.
- Yoo, H.-I. and Song, Ch.-R., Thermoelectricity of BaTiO₃₊₈, J. Electroceram., 2001, vol. 6, no. 1, pp. 61–74.
- Hotta, Y., Tsunekawa, K., Isobe, T., Sato, K., and Watari, K., Synthesis of BaTiO₃ powders by a ball milling-assisted hydrothermal reaction, *Mater. Sci. Eng.*, *Ser. A*, 2008, vol. 475, nos. 1–2, pp. 12–16.
- 14. Wei, X., Xu, G., Ren, Z., Wang, Y., Shen, G., and Han, G., Size-controlled synthesis of BaTiO₃ nanocrystals via a hydrothermal route, *Mater. Lett.*, 2008, vol. 62, no. 21, pp. 3666–3669.

- McCormick, M.A. and Slamovich, E.B., Microstructure development and dielectric properties of hydrothermal BaTiO₃ thin films, *J. Eur. Ceram. Soc.*, 2003, vol. 23, no. 12, pp. 2143–2152.
- 16. Yuan, Y., Zhang, S.R., Zhou, X.H., and Tang, B., Effects of Nb₂O₅ doping on the microstructure and the dielectric temperature characteristics of barium titanate ceramics, *J. Mater. Sci.*, 2009, vol. 44, pp. 3751– 3757.
- Liu, L., Guo, H., Lü, H., Dai, Sh., Cheng, B., and Chen, Zh., Effects of donor concentration on the electrical properties of Nb-doped BaTiO₃ thin films, *J. Appl. Phys.*, 2005, vol. 97, no. 5, p. 054102.
- Brzozowski, E., Castro, M.S., Foschini, C.R., and Stojanovic, B., Secondary phases in Nb-doped BaTiO₃ ceramics, *Ceram. Int.*, 2002, vol. 28, no. 7, pp. 773– 777.
- Shao, Y., Maunders, C., Rossouw, D., Kolodiazhnyi, T., and Botton, G.A., Quantification of the Ti oxidation state in BaTi_{1-x}Nb_xO₃ compounds, *Ultramicroscopy*, 2010, vol. 110, pp. 1014–1019.
- Noh, H.-J., Lee, S.-G., Nam, S.-P., and Lee, Y.-H., Effect of sintering temperature on structural and dielectric properties of (Ba_{0.54}Sr_{0.36}Ca_{0.10})TiO₃ thick films, *Trans. El. Electron. Mater.*, 2009, vol. 10, no. 2, pp. 49–52.
- Chen, C., Wei, Y., Jiao, X., and Chen, D., Hydrothermal synthesis of BaTiO₃: crystal phase and the Ba²⁺ ions leaching behavior in aqueous medium, *Mater. Chem. Phys.*, 2008, vol. 110, pp. 186–191.
- Xu, H. and Gao, L., Hydrothermal synthesis of highpurity BaTiO₃ powders: control of powder phase and size, sintering density, and dielectric properties, *Mater. Lett.*, 2004, vol. 58, pp. 1582–1586.
- Sun, W., Li, C., Li, J., and Liu, W., Microwave-hydrothermal synthesis of tetragonal BaTiO₃ under various conditions, *Mater. Chem. Phys.*, 2006, vol. 97, pp. 481– 487.
- 24. Liu, S.-F., Abothu, I.R., and Komarneni, S., Barium titanate ceramics prepared from conventional and microwave hydrothermal powders, *Mater. Lett.*, 1999, vol. 38, pp. 344–350.
- 25. Sun, W. and Li, J., Microwave-hydrothermal synthesis of tetragonal barium titanate, *Mater. Lett.*, 2006, vol. 60, pp. 1599–1602.
- Khanfekr, A., Tamizifar, M., and Naghizadeh, R., Micro-wave-hydrothermal synthesis and characterization of high-purity Nb doped BaTiO₃ nanocrystals, *J. Nanostruct.*, 2014, vol. 4, no. 1, pp. 31–36.

- Newalkar, B.L., Komarneni, S., and Katsuki, H., Microwave-hydrothermal synthesis and characterization of barium titanate powders, *Mater. Res. Bull.*, 2001, vol. 36, pp. 2347–2355.
- Liu, L., Guo, H., Lü, H., Dai, S., Cheng, B., and Chen, Z., Effects of donor concentration on the electrical properties of Nb-doped BaTiO₃ thin films, *J. Appl. Phys.*, 2005, vol. 97, p. 054102.
- Ávila, H.A., Ramajo, L.A., Reboredo, M.M., Castro, M.S., and Parra, R., Hydrothermal synthesis of BaTiO₃ from different Ti-precursors and microstructural and electrical properties of sintered samples with submicrometric grain size, *Ceram. Int.*, 2011, vol. 37, no. 7, pp. 2383–2390.
- Min, B., Moon, S.-M., and Cho, N.-H., Structural and dielectric features of Nb-doped nano-sized BaTiO₃ powders prepared by hydro-thermal synthesis methods, *Curr. Appl. Phys.*, 2011, vol. 11, no. 3, pp. S193–S196.
- Lazarević, Z., Romcević, N., Vijatović, M., Paunović, N., Romcević, M., Stojanović, B., and Dohcevic-Mitrović, Z., Characterization of barium titanate ceramic powders by Raman spectroscopy, *Proceedings of the Tenth Annual Conference of the Materials Research Society* of Serbia, YUCOMAT 2008, Herceg Novi, Montenegro, 2008, Acta Phys. Polonica, Ser. A, 2009, vol. 115, no. 4, pp. 808–810.
- Hai-Zhong, G., Li-Feng, L., Hui-Bin, L., and Yi-Yan, F., Structure, electrical, and optical properties of Nbdoped BaTiO₃ thin films grown by laser molecular beam epitaxy, *Chin. Phys. Lett.*, 2004, vol. 21, no. 2, pp. 396– 399.
- 33. Jhung, S.H., Lee, J.H., Yoon, J.W., Hwang, Y.K., Hwang, J.S., Park, S.E., and Chang, J.S., Effects of reaction conditions in microwave synthesis of nanocrystalline barium titanate, *Mater. Lett.*, 2004, vol. 58, no. 25, pp. 3161–3165.
- 34. Hotta, Y., Tsunekawa, K., Duran, C., Sato, K., Nagaoka, T., and Watari, K., Low-temperature sintering of BaTiO₃ powders prepared by a hydrothermal process with ball milling system, *Mater. Sci. Eng., Ser. A*, 2008, vol. 475, pp. 57–61.
- 35. Newalkar, B.L., Komarneni, S., and Katsuki, H., Microwave-hydrothermal synthesis and characterization of barium titanate powders, *Mater. Res. Bull.*, 2001, vol. 36, no. 13–14, pp. 2347–2355.
- Menéndez, J.A., Arenillas, A., Fidalgo, B., Fernández, Y., Zubizarreta, L., Calvo, E.G., and Bermúdez, J.M., Microwave heating processes involving carbon materials, *Fuel Process. Technol.*, 2010, vol. 91, no. 1, pp. 1–8.