Microwave-Hydrothermal Synthesis of Nb-Doped BaTiO₃ Nanoparticles under Various Conditions¹

A. Khanfekr*, M. Tamizifar, and R. Naghizadeh

*Department of Metallurgical and Materials Engineering, Iran University of Science and Technology (IUST), Narmak, Tehran, Iran *e-mail: khanfekr@iust.ac.ir*

Received August 13, 2014; in final form, November 24, 2014

Abstract—Tetragonal BaTiO₃ powders were synthesized by the microwave-hydrothermal (MH) method. The effects of the reaction time and temperature on the MH synthesis were investigated. Typical experiments were performed with a solid state reaction for the system $BaTiO₃ + Nb₂O₃$ at a high temperature. In the MH method, at a shorter time and lower temperature, the rate of the formation of tetragonal BaTiO₃ and of the growth of particles went down substantially. The microwave heating can be very fast and uniform through a self-heating process that results from the direct absorption of the microwave energy in the reaction mixture. The synthesis of Nb-doped BaTiO₃ has been investigated under MH conditions in the temperature range of 120–180°C for 1–3 hours using C_{16} H₃₆O₄Ti, BaH₂O₂ ⋅ 8H₂O, and NbCl₅ as Ba, Ti and Nb sources, respectively. In the phase evolution studies, the X-ray diffraction measurements and Raman spectroscopy were employed. The Transmission Electron Microscope and the Field Emission Scanning Electron Microscopy images were taken for the detailed analysis of the grain size, surface, and morphology. The MH method was applied for the synthesis of Nb-doped BaTiO₃ via provided advantages of the fast crystallization and decreased crystallite size. The nucleation rate should have been high because of a high heating rate in microwave heating processes.

Keywords: ceramics, chemical synthesis, microwave-hydrothermal synthesis, crystal structure, perovskites **DOI:** 10.3103/S1068375515050063

INTRODUCTION

Perovskite-type oxides have the general formula of $ABO₃$ in which A is a rare earth or alkaline earth metal and B is a transition metal. Applications of this group of materials in electrochemistry [1, 2], oxygen separa tion membranes [3], chemical sensors for the detec tion of humidity [4], alcohol [5], gases such as oxygen [6] and hydrocarbon [7], and of nitric oxide [8] for over 30 years are the basis for understanding their mixed conductivity by both ion and electron migra tions and their high nonstoichiometric composition. Since there is a strong dependence of ferroelectric properties on the grain size and compositional aspects, the microstructural control has become very impor $tant$ [9]. BaTiO₃-based positive temperature co-efficient (PTC) thermistors are widely used as over-cur rent protectors, motor starters for refrigerators and air conditioners, temperature sensors, and so on [10–14]. The hydro-thermal method is attractive for synthesiz ing $BaTiO₃$ powder because the combined effects of the solvent, temperature, and pressure on the ionic reac tion equilibrium can stabilize desirable products while inhibiting formation of undesirable compounds. Hydro-thermal synthesis also makes it possible to prepare $BaTiO₃$ powder in a single processing step and does not require elaborate apparatuses or expensive reagents However, just like in most convectional hydro-thermal processes, $BaTiO₃$ powders prepared by the microwave-hydrothermal (MH) synthesis at tem peratures below 200°C were cubic or pseudo cubic, though the tetragonal structure is thermodynamic, stable at room temperature. To convert hydrothermal $BaTiO₃$ from the cubic to tetragonal shape, heating at temperatures over 1000°C is required, which always leads to the grain growth and particle aggregation. Therefore, a direct generation of tetragonal Ba $TiO₃$ through the hydrothermal process is of a considerable interest, both in science and in industry [13–15].

It has been reported [15, 16] that the nanostruc tural and dielectric features of nano-sized Nb-doped $BaTiO₃$ powders vary sensitively with their size and preparation conditions. To the best of our knowledge, there are few reports on the doped BaTiO_3 metal-semiconductor transition on Nb doping, in particular. Sin tered bulk ceramics of Nb-doped BaTiO₃ show the semiconducting behavior when doped with a small amount of Nb, but when Nb concentration exceeds 0.4–0.5 mol %, the specimens revert to the insulator state. The main reason for the increase of resistivity when the donor concentration exceeds a certain value

 $¹$ The article is published in the original.</sup>

is the fact that the compensation mechanism changes from one that mainly involves electrons to one that primarily involves cation vacancies [15–18]. In the present paper, we analyze the effect of incorporating different amounts of Nb on the microstructural devel opment of $BaTiO₃$. A particular emphasis is placed in the morphology and composition of secondary phases.

Compared to the traditional solid-state reaction and other wet-chemical routes (including sol-gel pro cessing, oxalate route, and homogeneous precipita tion), the hydrothermal method is low-cost and con venient to prepare $BaTiO₃$ nanoparticles without a following high temperature calcination process [16–21]. In order to produce powders of a desired size and shape, several researchers investigated a variety of hydrothermal conditions, such as the pH value, Ba/Ti ratio, reaction temperature as well as various precur sors [21–25]. Though many efforts have been pulled to prepare $BaTiO₃$ particles, still there is insufficient knowledge about the effects of doping and solvent type on the size and shape under hydrothermal conditions. A number of authors $[21-25]$ have synthesized BaTiO₃ by the MH method at T below 200°C but the MH pro cessing of Nb-doped BaTiO₃ has not been reported so far with other authors. We succeeded in our initial efforts to synthesis of Nb doped BaTiO₃ in the temperature of 150°C for only 2 hours [26]. Therefore in this work, we have investigated the effect of incorporating Nb on the properties of BaTiO₃ via the MH conditions in the temperature range of 120–180°C for 1–3 hours.

EXPERIMENTAL

The MH reactions were performed to synthezise Nb-doped BaTiO₃. For more details, please, refer to our earlier manuscript [26]. The system was heated in the temperature range of $120-180^{\circ}$ C for $1-3$ hours. The obtained powders were characterized by the X-ray diffraction (XRD), with the Cu K_{α} radiation in the 2θ range from 20° to 60° and Raman spectroscopy (spectral range $80-3500$ cm⁻¹). Micro-structural characterization was performed by the field emission scanning electron microscopy (FE-SEM) and trans mission electron microscopy (TEM).

RESULTS AND DISCUSSION

Figure 1 shows the XRD patterns for the MH Nb -doped BaTiO₃ powders prepared in the temperature range of $120-180^{\circ}$ C for $1-3$ hours. From the XRD patterns it can be considered that the sample prepared at 120°C for 1 h (MH-120-1) forms a pattern similar to an amorphous product, but when the time and temperature rise, the $BaTiO₃$ powder becomes a structurally uniform crystalline with high crystallinity. Except the sample prepared at 120°C for 2 h (MH-120-2), the nanoparticles were pure perovskite $Nb-BaTiO₃$, without some intermediate carbonate

phase that was observed at 2θ 28° and 34° in the sam ple MH-120-2 [16]. The presence of $BaCO₃$ can be accounted for either by an incomplete reaction or by the presence of carbonate in the Ba alkali source, or by the reaction of $CO₂$ in air [14].

It is clearly seen that with the enhancement of time and temperature, the maximum of the peak shifts towards higher 2θ values and the intensity of the peaks grows. According to the investigations of some researchers [28, 29], the shift of peaks can be related to the displacements of barium. In the tetragonal phase, the optimization of the crystal geometry of $BaTiO₃$ after doping with Nb is attained by the impurity-out ward displacements of barium. The movements are mainly in the *ab* plane while the motion along the *c* axis is practically negligible. In the sample synthesis at the highest time and temperature, we have the highest intensity.

When the size of $BaTiO₃$ powders is below 1 µm, the full-width at half-maximum (FWHM) of the corre sponding (110) XRD peaks becomes larger and such broadening leads to the overlapping of two peaks located close to each other [14, 23, 27]. The structural changes associated with the phase transition usually have a large effect on the Raman spectrum. The XRD studies produce data that are consistent with an increasingly cubic structure at smaller particle sizes, not distinguishing between an average and a local structure. Thus, the XRD patterns reveal the charac teristics of cubic $BaTiO₃$ without certain peak splitting while the Raman results have supported the existence of a tetragonal symmetry at small dimensions, even though it was not discernible by XRD. The Raman spectra for the obtained $Nb-BaTiO₃$ ceramic powder samples are presented in Fig. 2. The peaks at 185, 305, 518 and 715 cm⁻¹ are characteristics of the tetragonal BaTiO₃. The peaks at 191 and 392 cm⁻¹ are assigned to $TiO₂ [30–33]$.

The dependence of the degree of the tetragonal fraction on the particle size has been extensively inves tigated for BaTiO₃ [23, 30]. A variety of explanatory models were proposed and researchers reported that the tetragonal distortion decreased when the particle size was below 1 μ m [23, 24]. Thus, a reasonable explanation for a larger tetragonality in the sample synthesized at 180°C is the fact that the particles grow faster under MH conditions and at higher tempera tures.

The FE-SEM images of $Nb-BaTiO₃$ powders prepared at 150 and 180°C for 2 h are shown in Figs. 3 and 4, respectively. It is evident that the powder prepared at 150°C consists of ultrafine, well dispersed uniform particles with the average size of 45 nm, the particle size being distributed from 40 to 50 nm. Whereas for the powders synthesized at 180°C, the average particle size increases substantially to the average size of

KHANFEKR et al.

Fig. 1. XRD patterns of Nb-BaTiO₃ nanoparticles at different time and temperature: (Up) Zoom in (110) and (200) planes (Down).

90 nm, with the particle size distribution being from 70 to 110 nm.

In the MH process, the microwave radiation can be absorbed by $BaTiO_3$ particles and smaller $BaTiO_3$ particles will dissolve more quickly. According to the dis solution/recrystallization mechanism, this will lead to larger particles growing faster with time and tempera ture [23, 32, 33].

The MH synthesis leads to the homogeneous nucleation of a large number of tiny stable Nb-BT par ticles, which further grow uniformly; the primary Nb-BT particles form agglomerates [24, 28].

The crystallite size of the particles was calculated with the XRD using the Scherrer equation as in [35]:

$$
D_{\rm v} = \frac{K\lambda}{B_{\rm size}\cos\theta},\tag{1}
$$

where $D_{\rm v}$ is the crystallite size, K is a constant whose value is 0.9 , λ is the X-ray wavelength; the width of the peak, B_{size} , was determined as the full width at half-
maximum. The crystallite size was 52 and 95 nm for the Nb-BaTiO₃ powder prepared at 150 and 180° C for 2 hours, respectively. These results are in good agree ment with the FE-SEM data, too.

Fig. 2. Raman spectra of hydrothermal nanosized Nb-BaTiO₃ powders.

The TEM images taken to estimate the size of the particles and the morphology of the particles for pow ders prepared at 150 and 180°C for 1 h are shown in Figs. 5 and 6, respectively. The decreased crystal size in MH-150-2 might be related to a high heating rate in MH processes. At the microwave frequency of 2.45 GHz, the changes of plus/minus in the waveguide occur 2450 million times in 1 s. Depending on the characteristics of the dielectric, it tends to modify the

alternating electric field through structural changes. When a water sample is placed in a reactor and sub jected to an alternating electric field, the electric devi ation (electric dipole) of the water molecule tends to follow the alternating electric field. However, since water consists of clusters of water molecules through hydrogen bonding, not all can follow the alternating electric field. The result is the dielectric heating of

Fig. 3. Nb-BaTiO₃ powder prepared at 150° C for 2 h.

Fig. 4. Nb-BaTiO₃ powder prepared at 180° C for 2 h.

Fig. 5. Nb-BaTiO₃ powders prepared at 150°C for 1 h.

Fig. 6. Nb-BaTiO₃ powders prepared at 180°C for 1 h.

the case of ions in solution, the Joule heating takes place by space-charge polarization. When microwave heating, an electrolyte-water solution dielectric heat ing, and the Joule heating occur simultaneously, com pared with pure water, then exothermic efficiency becomes remarkably high. Accordingly, dielectric heating by orientation polarization of water and resis tance heating by the Joule process are enhanced in electrolyte—water media. This rapid homogeneous heating, not possible by convectional heating, pro vides a uniform nucleation [36].

CONCLUSIONS

The present investigation shows that nanosized particles of $Nb- BaTiO₃$ powder can be crystallized via a rapid and cost-effective MH process without the temperature gradient and uneven nucleation and growth of particles of various sizes. The MH can be very fast and uniform through a self-heating process that arises from the direct absorption of the microwave energy in the reaction mixture. When microwave heat ing an electrolyte–water solution dielectric heating and the Joule heating occur simultaneously compared with pure water, and thus exothermic efficiency becomes remarkably high. Accordingly, dielectric heating by orientation polarization of water and resis tance heating by the Joule process are enhanced in electrolyte—water media. This rapid homogeneous heating, not possible by convectional heating, pro vides a uniform nucleation. Because of a homoge neous nucleation, uniformly-sized particles can be prepared. But according to the dissolution/recrystalli zation mechanism, both the extent of tetragonality and the particle size increase quickly with time and temperature. When the reaction temperature goes down, the formation of the tetragonal structure and growth of particles are slowed down substantially, showing a critical effect of the reaction temperature on the MH processing of tetragonal BaTiO₃.

REFERENCES

- 1. Kharton, V.V., Yaremchenko, A.A., and Naumovich, E.N., Research on the electrochemistry of oxygen ion con ductors in the Former Soviet Union, II: Perovskite related oxides, *J. Solid State Electrochem.*, 1999, vol. 3, no. 6, pp. 303–326.
- 2. Bi, Z., Cheng, M., Dong, Y., Wu, H., She, Y., and Yi, B., Electrochemical evaluation of $La_{0.6}Sr_{0.4}CoO_3 La_{0.45}Ce_{0.55}O₂$ composite cathodes for anode-supported $La_{0.45}Ce_{0.55}O_2 - La_{0.9}Sr_{0.1}Ga_{0.8}Mg_{0.2}O_{2.85}$ bilayer electrolyte solid oxide fuel cells, *Solid State Ionics*, 2005, vol. 176, nos. 7–8, pp. 655–661.
- 3. Takamura, H., Enomoto, K., Aizumi, Y., Kamegawa, A., and Okada, M., Preparation and oxygen permeability of Pr–Al-based perovskite-type oxides, *Solid State Ionics*, 2004, vol. 175, nos. 1–4, pp. 379–382.
- 4. Holc, J., Slunečko, J., and Hrovat, M., Temperature characteristics of electrical properties of $(Ba, Sr)TiO₃$ thick film humidity sensors, *Sensors Actuators, Ser. B*, 1995, vol. 26, no. 1–3, pp. 99–102.
- 5. Kong, L.-B. and Shen, Y.-S., Gas-sensing property and mechanism of Ca_xLa₁_{-x}FeO₃ ceramics, *Sensors Actuators, Ser. B*, 1996, vol. 30, no. 3, pp. 217–221.
- 6. Lukaszewicz, J.P., Miura, N., and Yamazoe, N., A LaF3-based oxygen sensor with perovskite-type oxide electrode operative at room temperature, *Sensors Actu ators, Ser. B*, 1990, vol. 1, nos. 1–6, pp. 195–198.
- 7. Brosha, E.L., Mukundan, R., Brown, D.R., Garzon, F.H., Visser, J.H., Zanini, M., Zhou, Z., and Logothetis, E.M., CO/HC sensors based on thin films of $LaCoO₃$ and $La_{0.8}Sr_{0.2}CoO_{3 - δ}$ metal oxides, *Sensors Actuators*, *Ser. B*, 2000, vol. 69, nos. 1–2, pp. 171–182.
- 8. Traversa, E., Matsushima, S., Okada, G., Sadaoka, Y., and Sakai, Y., Watanabe: $NO₂$ sensitive LaFe₃ thin films prepared by r.f. sputtering, *Sensors Actuators, Ser. B*, 1995, vol. 25, no. 1, pp. 661–664.
- 9. Vijatović, M.M., Bobić, J.D., and Stojanović, B.D., History and challenges of barium titanate, Part I, *Sci. Sintering*, 2008, vol. 40, pp. 155–165.
- 10. Kim, H.T. and Han, Y.H., Sintering of nanocrystalline BaTiO₃, *Ceram. Int.*, 2004, vol. 30, no. 7, pp. 1719– 1723.
- 11. Bomlai, P., Effects of surrounding powder in sintering process on the properties of Sb and Mn-doped barium strontium titanate PTCR ceramics, *Songklanakarin J. Sci. Technol.*, 2006, vol. 28, no. 3, pp. 669–675.
- 12. Yoo, H.-I. and Song, Ch.-R., Thermoelectricity of BaTiO_{3+ δ}, *J. Electroceram.*, 2001, vol. 6, no. 1, pp. 61–74.
- 13. Hotta, Y., Tsunekawa, K., Isobe, T., Sato, K., and Watari, K., Synthesis of BaTiO₃ powders by a ball milling-assisted hydrothermal reaction, *Mater. Sci. Eng., Ser. A*, 2008, vol. 475, nos. 1–2, pp. 12–16.
- 14. Wei, X., Xu, G., Ren, Z., Wang, Y., Shen, G., and Han, G., Size-controlled synthesis of BaTiO₃ nanocrystals via a hydrothermal route, *Mater. Lett.*, 2008, vol. 62, no. 21, pp. 3666–3669.
- 15. McCormick, M.A. and Slamovich, E.B., Microstruc ture development and dielectric properties of hydro thermal BaTiO₃ thin films, *J. Eur. Ceram. Soc.*, 2003, vol. 23, no. 12, pp. 2143–2152.
- 16. Yuan, Y., Zhang, S.R., Zhou, X.H., and Tang, B., Effects of $Nb₂O₅$ doping on the microstructure and the dielectric temperature characteristics of barium titan ate ceramics, *J. Mater. Sci.*, 2009, vol. 44, pp. 3751– 3757.
- 17. Liu, L., Guo, H., Lü, H., Dai, Sh., Cheng, B., and Chen, Zh., Effects of donor concentration on the elec trical properties of Nb -doped $BaTiO₃$ thin films, *J. Appl. Phys.*, 2005, vol. 97, no. 5, p. 054102.
- 18. Brzozowski, E., Castro, M.S., Foschini, C.R., and Sto janovic, B., Secondary phases in Nb-doped Ba $TiO₃$ ceramics, *Ceram. Int.*, 2002, vol. 28, no. 7, pp. 773– 777.
- 19. Shao, Y., Maunders, C., Rossouw, D., Kolodiazhnyi, T., and Botton, G.A., Quantification of the Ti oxidation state in $BaTi_{1-x}Nb_xO_3$ compounds, *Ultramicroscopy*, 2010, vol. 110, pp. 1014–1019.
- 20. Noh, H.-J., Lee, S.-G., Nam, S.-P., and Lee, Y.-H., Effect of sintering temperature on structural and dielectric properties of $(Ba_{0.54}Sr_{0.36}Ca_{0.10})TiO₃$ thick films, *Trans. El. Electron. Mater.*, 2009, vol. 10, no. 2, pp. 49–52.
- 21. Chen, C., Wei, Y., Jiao, X., and Chen, D., Hydrother mal synthesis of BaTiO₃: crystal phase and the Ba²⁺ ions leaching behavior in aqueous medium, *Mater. Chem. Phys.*, 2008, vol. 110, pp. 186–191.
- 22. Xu, H. and Gao, L., Hydrothermal synthesis of high purity $BaTiO₃$ powders: control of powder phase and size, sintering density, and dielectric properties, *Mater. Lett.*, 2004, vol. 58, pp. 1582–1586.
- 23. Sun, W., Li, C., Li, J., and Liu, W., Microwave-hydro thermal synthesis of tetragonal $BaTiO₃$ under various conditions, *Mater. Chem. Phys.*, 2006, vol. 97, pp. 481– 487.
- 24. Liu, S.-F., Abothu, I.R., and Komarneni, S., Barium titanate ceramics prepared from conventional and microwave hydrothermal powders, *Mater. Lett.*, 1999, vol. 38, pp. 344–350.
- 25. Sun, W. and Li, J., Microwave-hydrothermal synthesis of tetragonal barium titanate, *Mater. Lett.*, 2006, vol. 60, pp. 1599–1602.
- 26. Khanfekr, A., Tamizifar, M., and Naghizadeh, R., Micro-wave-hydrothermal synthesis and characteriza tion of high-purity Nb doped BaTiO₃ nanocrystals, *J. Nanostruct.,* 2014, vol. 4, no. 1, pp. 31–36.
- 27. Newalkar, B.L., Komarneni, S., and Katsuki, H., Microwave-hydrothermal synthesis and characteriza tion of barium titanate powders, *Mater. Res. Bull.*, 2001, vol. 36, pp. 2347–2355.
- 28. Liu, L., Guo, H., Lü, H., Dai, S., Cheng, B., and Chen, Z., Effects of donor concentration on the elec trical properties of Nb -doped $BaTiO₃$ thin films, *J. Appl. Phys.*, 2005, vol. 97, p. 054102.
- 29. Ávila, H.A., Ramajo, L.A., Reboredo, M.M., Castro, M.S., and Parra, R., Hydrothermal synthesis of $BaTiO₃$ from different Ti-precursors and microstructural and electrical properties of sintered samples with submicrometric grain size, *Ceram. Int.*, 2011, vol. 37, no. 7, pp. 2383–2390.
- 30. Min, B., Moon, S.-M., and Cho, N.-H., Structural and dielectric features of Nb-doped nano-sized $\operatorname{BaTiO_3}$ powders prepared by hydro-thermal synthesis methods, *Curr. Appl. Phys.*, 2011, vol. 11, no. 3, pp. S193–S196.
- 31. Lazarević, Z., Romcević, N., Vijatović, M., Paunović, N., Romcević, M., Stojanović, B., and Dohcevic-Mitrović, Z., Characterization of barium titanate ceramic powders by Raman spectroscopy, *Proceedings of the Tenth Annual Conference of the Materials Research Society of Serbia, YUCOMAT 2008, Herceg Novi, Montenegro, 2008, Acta Phys. Polonica, Ser. A,* 2009, vol. 115, no. 4, pp. 808–810.
- 32. Hai-Zhong, G., Li-Feng, L., Hui-Bin, L., and Yi-Yan, F., Structure, electrical, and optical properties of Nb doped $BaTiO₃$ thin films grown by laser molecular beam epitaxy, *Chin. Phys. Lett.*, 2004, vol. 21, no. 2, pp. 396– 399.
- 33. Jhung, S.H., Lee, J.H., Yoon, J.W., Hwang, Y.K., Hwang, J.S., Park, S.E., and Chang, J.S., Effects of reaction conditions in microwave synthesis of nanoc rystalline barium titanate, *Mater. Lett.*, 2004, vol. 58, no. 25, pp. 3161–3165.
- 34. Hotta, Y., Tsunekawa, K., Duran, C., Sato, K., Nagaoka, T., and Watari, K., Low-temperature sinter ing of $BaTiO₃$ powders prepared by a hydrothermal process with ball milling system, *Mater. Sci. Eng., Ser. A*, 2008, vol. 475, pp. 57–61.
- 35. Newalkar, B.L., Komarneni, S., and Katsuki, H., Microwave-hydrothermal synthesis and characteriza tion of barium titanate powders, *Mater. Res. Bull.*, 2001, vol. 36, no. 13–14, pp. 2347–2355.
- 36. Menéndez, J.A., Arenillas, A., Fidalgo, B., Fernández, Y., Zubizarreta, L., Calvo, E.G., and Bermúdez, J.M., Microwave heating processes involving carbon materials, *Fuel Process. Technol.*, 2010, vol. 91, no. 1, pp. 1–8.