



# **Characterization and electrochromic properties of NiO thin flms prepared using a green aqueous solution by pulsed spray pyrolysis technique**

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**Nickel acetylacetonate of 0.2 M aqueous solution is used to synthesize (NiO) thin flms by pulsed spray pyrolysis method onto glass substrates. Structural and morphological properties of the deposited flms were examined using (XRD) and (FESEM) techniques. The results revealed that at low substrate temperatures and short spray time, the obtained flms were amorphous, whereas the flms deposited at higher temperatures and long spray time has a cubic phase. Cross-sectional FESEM is used to properly estimate flms thicknesses. The energy band gap is calculated to be 3.52–3.89 eV using optical spectrum measurements. Electrochromic properties of NiO flms deposited on FTO working electrode substrates are measured in a three-electrode cell containing KOH with Ag/AgCl reference electrode at a scan rate of 20 mV/s. The charge density increases with the thickness, and it is 98.0 mC/cm<sup>2</sup> at the thickness of 360 nm after 30-min deposition time at 400 °C and for most of the samples the flm stability increases after cycling.**

## **Introduction**

Spray pyrolysis technique is used for deposition n-type transparent conductive oxides such as zinc oxide (ZnO) and indium-tin oxide (ITO) which are widely used for solar cell electrodes, window coatings, etc. [\[1\]](#page-9-0). However, new p-type transparent conducting thin flms are prepared for useful versatile applications such as fabrication of dye-sensitized solar cells, chemical sensors, smart windows (ultraviolet, visible or (near) infrared radiation block) for saving energy consumption [[2–](#page-9-1)[9](#page-9-2)]. Electrochromic phenomenon means change in material transmission to nearly opaque by reverse the polarity of low applied potential 1.5 V. It is very useful in solar radiation spectrum control. Buildings and vehicle glass windows coated with an electrochromic flm reduce the entered solar radiation thereby the internal temperature will be minimized. Based upon the windows coated with heat mirror fabricated by spray pyrolysis technique will reduce the consumed energy for cooling appliances; nickel oxide is a prominent in electrochromic family material. Nickel oxide flms have various properties due to their high electrochromic efficiency, including cyclic reversibility (very stable), endurance, and color [\[10\]](#page-9-3). Nickel oxide (NiO) has excellent chemical stability and wide band gap of 3.0–4.3 eV is considered as one of the best examples of p-type semiconductor metal oxide thin films  $[11-15]$  $[11-15]$ . The conductivity in p-type semiconductor materials is usually related to the excess of oxygen atoms or due to the lack of metal atoms compound in their forming structure [\[16\]](#page-9-6). Moreover, due to NiO electronic band structure, robust durability, electrochromic efficiency, and cyclic reversibility all make it valuable and promising for diferent modern applications such as smart windows technology.

Many of chemical and physical deposition techniques have been used to synthesis NiO thin flms such as sol–gel, sputtering, pulsed spray pyrolysis, spin coating, and pulsed laser deposition [\[17](#page-9-7)–[21](#page-10-0)]. Among these techniques pulsed spray pyrolysis (PSP) has many advantageous such as low cost, easy to operation, no vacuum, produce multi-state flm (amorphous, crystalline, porous, compact, excellent adherence) with the ability for mass production of films with wide area  $[21]$  $[21]$ . The main advantage of this system is keeping the substrate temperature constant all over the spray process which is a main problem in pneumatic spray systems. The properties of the produced film depend mainly on



molarity of the precursors, substrate temperatures, substrate to nozzle distance, spray time, and gas carrier flow rate. The mostly used precursors to prepare NiO are nitrates and chlorides which need more precautions due to its hazards. Herein, nickel Acetylacetonate is used as safe and benign precursor for synthesis of an electrochromic NiO flm.

This manuscript is devoted to exploring at what extent the quality of the deposited NiO flm on bare and FTO-coated glass substrates by the LAB team homemade and assembled PSP system using green spray solution via its characterization. Structural, optical, and electrochromic properties are appointed by XRD, FESEM, optical transmission, and cyclic voltammetry. The obtained results not only support a simple, easy, and safety technique but also enrich the data related to electrochromic phenomenon of NiO films. The preparation and characterization of NiO flms could be considered valuable, since up to date no studies have been done using nickel acetylacetonate as a precursor in the form of aqueous solution  $[12]$  $[12]$ . The electrochromic (EC) properties of the prepared NiO flms from nickel acetylacetonate precursor at both diferent deposition temperatures and spray time will be investigated in the light of structural, morphological, and optical results of the synthesized flms to determine the optimum preparation conditions giving rise to produce an electrochromic flm with high performance of shorter response time and high reversibility from transparency to dim nature at lower applied voltage.

### **Results and discussion**

#### **XRD characterization**

The X-ray diffraction patterns of the deposited films at both diferent temperatures and spray times are shown in Fig. [1](#page-2-0)(a, b). The deposited films are polycrystalline; all the diffraction peaks (111), (200), (220), (311), (222) are indexed with JCPDS fles and it is matched with the data sheet No. (# 00-065-2901) corresponding to NiO with cubic phase and space group [Fm-3m (225)]. No other phases or crystalline impurities are detected, indicating the phase singularity of the synthesized NiO thin flms.

The lattice parameter  $a$  is calculated by  $a = d \times \sqrt{h^2 + k^2 + l^2} \ a = d \times \sqrt{h^2 + k^2 + l^2}$ , where (hkl) are the Miller indices and (*d*) is the interplanar spacing given by  $2d_{hkl}$  sin( $\theta_{hkl}$ ) =  $\lambda$ ,  $\theta_{hkl}$  is Bragg's angle,  $\lambda$  is the wavelength of X-ray. In Table [1,](#page-2-1) the crystallite size D and strain *ε* values for all the samples are calculated (from the slope and the cutting part from the *y*-axe of the plot) using Williamson–Hall Plot defned  $\beta_{hkl}$  cos  $\theta_{hkl} = (k\lambda/D) + 4\varepsilon \sin \theta_{hkl}$  [[22](#page-10-1)], since  $\beta_{hkl}$  is the broadening (FWHM) of the peak. The obtained results show that the

deposited flms are polycrystalline due to the existence of more than one plane of growth with a nanosized scale because of lower intensity of peaks. An observable increase in both lattice parameter and crystallite size is induced by increasing deposition temperature and spray time, whereas the strain decreases. This could be interpreted as some sort of relaxation resulting in lattice when both deposition time and temperature are increased which allows an increase and decrease in crystallite size and internal strain, respectively.

#### **SEM**

Field Emission Scanning Electron Microscope FESEM is used to investigate the surface morphological structure of the deposited NiO films. The obtained images for the examined sample NO2, NO5, NO7, NO8, and NO9 are displayed in Fig. [2\(](#page-3-0)a–e). Nearly individual round small particles are found in neighbor with some large cluster embrace, where some of them indicating some sort of surface roughness of NO2 sample marked by both deposition temperature 400 °C & time 30 min are observed in Fig.  $2(a)$  $2(a)$ . The protruding surface roughness may be due to that the growth process is fast inducing small crystallites conjugated with some clusters.

The limiting factors temperature and time for diffusing process are not sufficient to make complete diffusion of the composed microstructures allowing lower level of roughness. This is supported by the decrease in surface roughness level by the increase in temperature or time or both as observed in Fig. [2](#page-3-0)(a, d) of samples deposited at 400 & 500 °C. The improvement in surface homogeneity is found as the deposition temperature rises to 450 and 500 °C as shown in Fig. [2](#page-3-0)(b, c) at spray time of 30 min. The crystalline quality of the deposited films is increased which synergies with the decrease in broadening and increase in intensity of peaks recorded in XRD patterns. The surface morphologies of the flms NO7, NO8, and NO9 deposited at diferent spray times (10, 30, 50 min) and constant substrate temperature of 500 °C are shown in Fig.  $2(c-e)$  $2(c-e)$ .

For NO7 and NO8 samples, a well-closed surface structure with an interconnected grains and compact morphology is found, indicating good adhesion to the substrate surface without micro cracks or pinholes as shown in Fig. [2](#page-3-0)(c, d). In contrast as shown in Fig. [2](#page-3-0)(e), sample NO9 has formed agglomerates which may be attributed to the long spray time. EDX spectra revealed the presence of Ni and O atoms in the synthesized films. The thickness of the investigated samples is estimated from the cross-sectional FESEM images [[21\]](#page-10-0). The recorded thicknesses for NO2, NO5, NO7, NO8, and NO9 flms are 360, 327, 192, 233, and 403 nm, respectively.





<span id="page-2-0"></span>**Figure 1:** XRD charts of the spray deposited NiO flms, (a) XRD patterns of flms deposited for constant 30 min at 400 °C (NO2), 450 °C (NO5), and 500 °C (NO8), (b) XRD patterns of flms deposited for at constant 500 °C for 10 (NO7), 30 (NO8), and 50 (NO9) minutes.

<span id="page-2-1"></span>**TABLE 1:** The calculated Crystallographic information for the deposited flms.

Sample	Crystallographic information		
	Lattice param- eter $a$ , (Å)	Crystallite size $D$ , (nm)	Strain $(\varepsilon)$
NO2 (400 °C, 30 min)	4.15	3.18	0.006
NO5 (450 °C, 30 min)	4.17	3.68	0.005
NO7 (500 °C, 10 min)	4.18	4.22	0.002
NO8 (500 °C, 30 min)	4.19	4.49	0.002
NO9 (500 °C, 50 min)	4.20	5.50	0.001

### **Optical measurements**

Nickel oxide thin flms have a fascinating optical switching property. High transparency of NiO flm is reversibly converted

to nearly opaque when a lower potential (about 1 V) is applied. Such material is promising for a wide range of electrochromic applications especially solar spectrum modulation. To provide a comprehensive insight, the optical Transmittance (*T*) & Refectance (*R*) spectra are measured for NiO flms when subjected to lower potential in an electrolyte by a spectrophotometer provided with a special measuring cell.

The LAB facilities for performing these measurements are not available. Therefore, the typical transmission and reflection measurements are carried out by a normal double-beam spectrophotometer without immersing the samples in an electrolyte and potential application. Therefore, optical switching as an electrochromic phenomenon is measured by the well-known cyclic voltammetry technique. The typical measurements of T&R for the NiO flm samples (group A) prepared at constant substrate



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(a) Sample NO2 (400 $\degree$ C-30 minutes)



(b) Sample NO5 (450 $\degree$ C-30 minutes



#### (c) Sample NO8 (500°C-30 minutes) SEM, EDX and Cross section images.



(d) Sample NO7  $(500°C-10$  minutes)

(e) Sample NO9 (500 $\textdegree$ C-50 minutes)

**Figure 2:** FESEM images for NiO flms deposited at diferent temperatures and spray times.

temperature 500 °C and variable spray time 10, 30, and 50 min and those (group B) deposited at constant spray time 30 min and varied substrate temperatures 400, 450, and 500 °C are executed.

The spectrum curve for each sample is demonstrated in Fig. [3](#page-4-0). The most salient feature is all samples show high transmittance (nearly 90 to 70%) and round spectrum curves. Also, the sharp absorption edge is not observed which means that the energy gap is not empty, but it contains non-intentional impurity levels and non-stoichiometry level such as oxygen vacancies <span id="page-3-0"></span>or excess Nickel. It is clear that samples of group (A) exhibit reasonable decrease in transmission conjugated with the increase in deposition time.

This observation is mainly related to the increase in film thickness and the induced film compactness as supported by FESEM images and the measured values of film thickness. Also, thicker films have lower transmittance due to the existence of larger crystallite sizes, bigger clusters as well as higher surface roughness which increase light scattering. On the





<span id="page-4-0"></span>**Figure 3:** Transmittance & refectance spectrum of NiO flms deposited at diferent spray times (10, 30, & 50 min) and deposition temperatures (400, 450, & 500 °C).



<span id="page-4-1"></span>**Figure 4:** The (ahv)<sup>2</sup> vs. *hv* plot for NiO thin films at different substrate temperatures and spray times.

contrary, the samples group (B) show significant increase in transmission as the substrate temperature increases due to the decrease in film thickness which may be caused by the probable evaporation of non-intentional impurities. For precise estimation of the energy gap, Tauc's relation [\[23](#page-10-2)] is applied by plotting (*αh*ʋ) 2 versus (*h*ʋ). The energy gap *E*g is computed from the intersection point of the linear part of the curve with (*h*ʋ) axe as shown in Fig. [4](#page-4-1).

Where,  $\alpha$  is the absorption coefficient,  $\nu$  is the frequency of the incident photon, and *h* is the Planck constant. The computed band gap energy values  $E_{\rm g}$  for the measured NiO samples NO2, NO5, and NO8 are found to be 3.89, 3.66 and 3.65 eV, respectively. The decrease in  $E_g$  with the increase of the substrate temperatures could be attributed to crystallinity improvement of the deposited films. The estimated band gap for the samples deposited at diferent spray times of NO7, NO8, and NO9 are found to be 3.52, 3.65, and 3.77 eV. It is apparent that sample NO2 has the higher  $E_g$  value than other samples which may be

due to the amorphous phase as recorded in its XRD pattern. The obtained values  $(3.0-4.0 \text{ eV})$  are in good agreement with those reported [\[15\]](#page-9-5).

#### **Electrochromic properties**

Nickel oxide NiO flms are deposited at diferent temperatures on glass substrates coated with the prepared conductive tin oxide-doped fuorine (FTO) to be a working electrode in cyclic voltammetry measurements technique. Typical three-electrode cell provided with Ag/AgCl reference electrode, pure platinum sheet as counter electrode, and NiO flms on FTO as working electrode in 0.1 M KOH solution is used to measure cyclic voltammetry (CV) and chronoamperometry (CA) for the examined samples marked as NO2, NO5, NO7, NO8, & NO9. The effect of changing substrate temperature on (CV) measurements for samples NO2, NO5, & NO8 in a potential window of [− 0.8 to 1.2 V] versus Ag/AgCl reference electrode at a scan rate of 20 mV/s are shown in Fig. [5](#page-5-0)(a) and the photographs of the NiO thin flms in the bleached (at  $-$  0.4 V) and colored (at  $+$  0.7 V) states are shown in Fig.  $5(b)$  $5(b)$ .

It is noted that the voltammograms show signifcant differences, where the voltammograms of NO2 and NO5 show two well-resolved oxidation peaks at potentials 0.04 and 0.42 V for NO2 and at 0.05 and 0.75 V for NO5 related to  $Ni^{2+}/Ni^{3+}$ and  $Ni^{3+}/Ni^{4+}$  and only one reduction peak at potential 0.15 V [\[24](#page-10-3)]. However, NO8 voltammogram is a common shape with two well-defned reversible anodic and cathodic peaks for the redox couple of  $Ni^{2+}/Ni^{3+}$  which correlated to the Ni (OH)<sub>2</sub>/ NiOOH phases  $[25-27]$  $[25-27]$ . The sharp current increase showed after 1.0 V is usually associated with the oxygen evolution reaction  $(2H_2O= 4e^- + O_2 + 4H^+).$ 

The current decreases from NO2 to NO8 may be due to the decrease in the flm thickness at the FTO surface. It is merit to observe that during the CV scanning of the three samples the color changes from transparent to black and vice versa. The change in color may be due to the electron transfer processes happening at the grain boundaries and on grain surfaces [[28,](#page-10-6)  $29$ ]. The effect of changing the time of deposition at constant substrate temperature 500  $^{\circ}$ C is also studied. The voltammograms of the samples NO7, NO8, and NO9 in a potential window of [− 0.8 to 1.2 V] versus Ag/AgCl reference electrode, at a scan rate of 20 mV/s are shown in Fig.  $5(c)$  $5(c)$ .

One redox peak is observed through the three voltammograms which may be due to the increase in flm thickness with an increase in deposition time indicated by the increase in the current peak intensity. It was suggested that the mechanism of electrochromism through the nickel oxide thin flm in KOH solution is as shown in the following reaction [[30](#page-10-8)].

$$
\begin{array}{ccc} \mathrm{NiO} + \mathrm{OH}^- & \leftrightarrow & \mathrm{NiOOH} + e^- \\ \mathrm{Blacked} & & \mathrm{colored} \text{ (brownish gray)} \end{array}
$$





(a) Cyclic voltammograms of NiO films deposited at constant time 30 minutes and different substrate temperatures  $(400.450 \& 500^{\circ}$ C





(c) Cyclic voltammograms of NiO films deposited at constant substrate temperature 500°C and different spray time  $(10, 30 \& 50)$  minutes

(b). The photographs of the NiO thin films in the bleached (at  $-0.4$  V) and colored (at  $+0.7$  V) states

<span id="page-5-0"></span>**Figure 5:** (a) Cyclic voltammograms of NiO flms deposited at constant time 30 min and diferent substrate temperatures (400, 450, & 500 °C). (b) The photographs of the NiO thin flms in the bleached (at − 0.4 V) and colored (at +0.7 V) states. (c) Cyclic voltammograms of NiO flms deposited at constant substrate temperature 500 °C and diferent spray times (10, 30, & 50) min.

and out of the flm (*q*a & *q*c), Δ*q* ratio, and the charge density (*Q*). Name NO2 NO5 NO7 NO8 NO9 *q*a (mC) 5.97 2.61 0.04 0.22 0.36

*q*b (mC) 6.33 3.48 0.05 0.20 0.38 Δ*q* 1.06 1.30 1.25 0.91 1.05

) 98.0 58.56 0.77 4.04 7.0

<span id="page-5-1"></span>**TABLE 2:** Determination of the amount of the charge intercalated into

The amount of the charge intercalated out  $(q \text{ anodic})$  and into (*q* cathodic) of the film ( $q_a$  &  $q_c$ ) during the redox reaction is calculated by the integration of the anodic and cathodic area separately for the three samples. The obtained values are listed in Table [2](#page-5-1). The ratio  $\Delta q$  is calculated ( $\Delta q = q_c/q_a$ ) and listed in Table [2](#page-5-1) where the values near one indicate the reversibility of the redox process, which suggests that NiO flms have a good reversibility in 0.1 M KOH electrolyte [[24\]](#page-10-3). The samples NO2 and NO9 fulfill the reversibility of charge intercalation. The charge density  $(Q)$  of the NO2, NO5, NO7, NO8, and NO9 flms calculated from the CV curve at scan rate of 0.02 V/s according to the following relation [\[31\]](#page-10-9) is found to be 98.0, 58.56, 0.77, 4.04, and 7.0 mC/cm<sup>2</sup> respectively.

$$
Q = \frac{1}{\nu A} \int\limits_{V_i}^{V_f} I dV, \tag{1}
$$

where  $\nu$  is the scan rate,  $V_i$  is the initial scanned potential,  $V_f$  is the fnal scanned potential, *I* is the current, and *A* is the area of the working electrode surface. To investigate the redox stability of the flm in 0.1 M KOH, the cyclic voltammetry was repeated for 500 cycles at scan rate 0.1 V/s as shown in Fig.  $6(a-e)$ .

It is fascinating to show good redox stability after 500 oxidation–reduction cycles where the peak current reduced to nearly 50% of the peak current of the frst cycle for NO2 sample which may be due to the detachment of the flm from FTO surface [\[3,](#page-9-9) [32\]](#page-10-10) or may be due to the irreversible intercalation of the hydroxyl group to the oxide flm [\[32\]](#page-10-10). However, the peak current afer 100–500 cycles increased to more than twofold of the peak current of the frst cycle for samples NO5, NO7, NO8, and NO9 indicating the increase in the stability of the redox reaction with time which suggests modifcation in the nature and morphology of the flm during cycling [[32\]](#page-10-10).

This behavior is happened when a porous film is used and due to that not all the possible surface area was

 $Q$  (mC/cm<sup>2</sup>)



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The cyclic voltammetry measurements at different number of scanning cycles.

<span id="page-6-0"></span>**Figure 6:** (a–e) The cyclic voltammetry measurements at diferent number of scanning cycles.





Figure 7: (a-e) The recorded speed of the redox reaction of the deposited NiO thin films.

accessible to the electrolyte at the start of the experiment. But with cycling, all the sites become accessible where the intercalation of OH ions in the pores establish an electronic connectivity throughout the deposited film [[3](#page-9-9), [32\]](#page-10-10). In addition, a shift of the oxidation to higher values and the <span id="page-7-0"></span>reduction peaks to lower values result in the increase in the peak separation. To determine the speed of the redox reaction, chronoamperometry (CA) technique was performed for the samples in the potential range from − 0.7 to 0.7 V (Ag/ AgCl) as shown in Fig. [7\(](#page-7-0)a–e).



It was noted that the current decreases with time due to the increase in the chemical potential of the intercalated ions. The response time for the coloration  $(t_c)$  and discoloration  $(t_d)$ , which are the time needed for anodic and cathodic current to reach a steady state after applying certain potential, are very essential to study the performance of the electrochromic devices. Afer applying 0.7 V, the anodic current for NO2, NO5, NO7, NO8, and NO9 decreases to reach a steady state of about 5.0, 4.5, 3, 2.8, and  $4.7 \text{ mA/cm}^2$  within 4, 16, 18, 6, and 0.1 s, respectively, while on switching to the negative potential  $-0.7$  V, the current decreases to attain steady state at 0.03, − 0.17, − 15.4, 0.28, and − 0.39 mA/cm<sup>2</sup> within 1.5, 5, 19, 2, and 1.5 s, respectively. The values of  $t_c$  and  $t_d$  obtained for NO9F are better than the values obtained in different previous works  $[27, 32, 33]$  $[27, 32, 33]$  $[27, 32, 33]$  $[27, 32, 33]$  $[27, 32, 33]$  $[27, 32, 33]$  $[27, 32, 33]$ . The response time was dependent on several factors such as applied potential, thickness of the flms, and electrolyte conductivity [\[34\]](#page-10-12).

## **Conclusion**

At diferent deposition temperatures and spray times an electrochromic dopant-free NiO flm is prepared successfully on glass substrates by PSP system, and the following conclusions were reached from the fndings:

- At high deposition temperatures, NiO flms are polycrystalline with a cubic phase and nanosized scale, according to XRD analysis. At low deposition temperature and long spray time, FESEM micrographs revealed a rough flm surface, and the existence of Ni and O atoms was revealed by EDX spectra.
- The measured optical spectra curves stated clearly strong transmittance of 85% with a predicted band gap ranging from 3.52 to 3.89 eV.
- Electrochromic (EC) characteristics of NiO thin flms deposited on FTO substrates showed that the samples prepared at highly deposition temperatures and long spray times of 500 °C (30 and 50 min) fulfll the reversibility of charge intercalation condition during dim and bleaching electrochromic behavior, and the best recorded values for  $t_c$  and  $t<sub>d</sub>$  values were found in samples deposited at 500 °C and 50 min. Based on these obtained results, the prepared NiO thin flms could be considered as a promising candidate to reduce the consumed energy required for cooling appliances.

## **Materials and methods**

#### **Preparation of NiO thin flms**

Anhydrous nickel (II) acetylacetonate  $(C_{10}H_{14}NiO_4)$  with a concentration of 0.2 M dissolved in 95 ml of distilled water and 5 ml acetic acid stirred vigorously at 40 °C for 15 min are used as a precursor solution for deposition of nickel oxide thin flms. Pulsed spray pyrolysis (PSP) system is used to prepare the synthesized flms at diferent temperatures and varied deposition time [NO2 (400 °C, 30 min), NO5 (450 °C, 30 min), NO7 (500 °C, 10 min, NO8 (500 °C, 30 min), and NO9 (500 °C, 50 min)] on ultrasonically and chemically cleaned glass substrates. To keep the temperature, stable at the substrate surface, the aerosol solution is intermittent through interval of (1 s for ON and 4 s for Off mode).

All other spraying parameters such as gas (air) carrier flow rate of 25 cm<sup>3</sup>/min, atomizer to substrate distance maintained at 30 cm. Solution deposition interval and molarities are kept constant to obtain uniform, homogeneous, and well-adhered films on the surface of the glass substrate. The applied procedure to great extent is diferent by its safety and simplicity than those used in published papers.

#### **Characterization of NiO thin flms**

X-ray difractometer (XRD) Empyrean from PANalytical Netherland is used to record the XRD patterns for the investigated samples to elucidate the existed phases and crystal nature of the deposited films in the scan range 2θ from 10° to 80°. The surface morphology of the synthesized thin flms is examined using feld emission scanning electron microscope (FESEM-Quanta FEG 250 with feld emission gun, FEI company-Netherland) and the energy-dispersive X-ray spectroscopy (EDX) is also used to provide elemental identification of the prepared films. The thickness of the samples is determined from the cross-sectional FESEM images. Optical properties measurements are performed using UV–Vis–NIR double-beam spectrophotometer (Jasco-750 V) within a wavelength from 0.3 to 2.5  $\mu$ m. The optical band gap  $(E_{\alpha})$ of NiO flms is determined using Tauc's extrapolation plot [\[23](#page-10-2)].

#### **Electrochromic measurements**

Electrochromic investigations are carried out in a three-electrode conventional glass cell in 0.1 M KOH solution at 25 °C using Origalys OGS 200 potentiostat/galvanostat. The potential of the working electrode will be measured against Ag/AgCl reference electrode  $(E_0 = 0.203 \text{ V} \text{ vs. SHE})$  and pure Pt sheet will be used as counter electrode. The working electrode is prepared as a NiO thin flm on transparent conductive glass substrates coated with FTO. Cyclic voltammetry curves are used to characterize the electrochemical behavior of electrodes at 20 and 100 mV/s sweep rates covering a potential range between − 0.8 and 1.0 V (vs. Ag/AgCl) after 5 min of immersion in electrolyte (25 °C). The chronoamperometry (CA) measurements are evaluated at a potentials range of − 0.7 and +0.7 V.



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## **Data availability**

The authors confirm that the data supporting the findings of this study are available within the article and its supplementary materials.

# **Declarations**

**Conflict of interest** The authors declare that there are no confict of interest.

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