GENERAL-PURPOSE MATERIALS

Advanced Materials for Separation and Purification of Gas Mixtures in Cyclic Adsorption Processes

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Abstract—We fabricate and study experimentally a promising nanostructured composite adsorbent based on crystalline zeolite NaX. The adsorbent exhibits improved physical mechanical and adsorption properties and is intended for use in pressure swing adsorption systems. Fluoro derivatives of ethylene are found to be the most promising for creating the matrix of block-shaped composite adsorbent materials, since these ensure a mechanical strength and good adhesion between the matrix and incorporated adsorbent. We investigate the effects that the type, ratio of starting materials, and conditions of forming have on the adsorption and mechanical parameters of prepared composite adsorbents. The choice of materials for the adsorbent filler and polymer matrix used in synthesis of block-shaped sorption-active composite materials is substantiated. Studies of the adsorption parameters of prepared materials suggest that the film constituting the polymer (fluoroplastic) matrix, a base of our block-shaped adsorbents, is not continuous, which allows adsorbate molecules to penetrate the bulk of adsorbent. A comparative analysis establishes that the dynamic activity, as measured using water vapor, was 20 to 40% higher in our samples than in conventional commercial adsorbents of the NaX-V-1G type.

Keywords: sorption-active composite materials, pressure swing adsorption, gas mixture, separation, polymer matrix, zeolite, ultimate strength, dynamic activity

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INTRODUCTION

Technologies widely adopted for separation of gas mixtures are currently based on cyclic adsorption processes in which alternating stages of gas adsorption and desorption ensure a continuous production of a target gas at the outlet of a gas separating unit $[1-15]$. Granulated sorbents based on mixtures of clay and zeolites in a form of powdered charge are conventionally used in units operating on the pressure swing adsorption (PSA) principle, their one drawback being wearing and dusting of the adsorbent material during long operation time, which results in gas transporting pores being blocked and a fall in their adsorption capacity. Moreover, a need arises for additional tightening in adsorbent modules.

For PSA systems operating under variable aerodynamic loads and at considerably high gas flow rates, it is more appropriate to use zeolite adsorbents prepared as sorption-active composite materials (SACMs) with a high mechanical strength. This type of materials consists of a pliable support material, i.e., a matrix, and fillers, which can be adsorbent materials of different types. The matrix functions as a binding component that defines the strength and pliability of the composite adsorbent as a whole when it is impacted by mechanical, aerodynamic, and other loads [16–20].

For SACMs, the parameters of their porous structure, adsorption properties, and durability can be tailored via a few groups of controlling factors: the nature, degree of dispersion, and fractions of initial adsorbent fillers and the binding material; the nature of bonding between adsorbent particles and the binding material; conditions for forming of the composite; and the procedure for removing the liquid phase and subsequent final thermal treatment of the formed composite adsorbent materials [22–25].

The aim of this work is to investigate experimentally variations in the strength and adsorption properties of block-shaped SACMs with the composition of the initial mixture and the conditions of forming. Additionally, we develop recommendations concerning industrial production of these materials.

Sample no.	$A: B$, wt $\%$	Dew point of dried air, $T_{\rm dp}$, °C	Duration of protective effect of SACM, t_{pr} , min	Dynamic capacity, a_{d} , mg/g [28]	
	75:25	-65.30	35	172	
2	80:20	-67.30	36	180	
3	83:17	-72.10	42	187	
$\overline{4}$	85:15	-72.30	45	193	
5	87:13	-72.30	46	199	
6	90:10	-70.10	40	196	
7	100:0	-63.20	31	141	

Table 1. Sorption parameters of prepared SACMs measured in dynamic conditions with forward flow of air

MATERIALS AND METHODS

In preparing SACMs, we used crystalline zeolite NaX (STO 05766575-2009), powdered fluoroplas-42 (type F-42V; GOST 25428-82), acetone (analytical grade; GOST 2603-79), and methyl ethyl ketone as a solvent.

Characterization methods included mechanical tests for our block-shaped SACMs, which were prepared as cuboids with the side ratio of 1 : 1 : 3. Tests were performed in the constant strain mode $(10 \mu m/s)$ using a multipurpose MTS 870 Landmark servohydraulic test system (MTS Systems, USA).

The sorption parameters of prepared SACMs were determined from dynamic measurements of air drying and from the dynamic activity of the prepared materials (using the dew point technique) under the following conditions: relative humidity of air stream, 40%; temperature of air stream, 20 ± 2 °C; and the specific volumetric flow rate, $0.2 \text{ m}^3 / (\text{m}^2 \text{ s})$.

The porous structure of prepared SACMs was studied (in terms of the morphology) by electron microscopy on a Neon instrument (Carl Zeiss) and by low-temperature nitrogen adsorption using a NOVA-1200e surface area and pore size analyzer (Quantachrome, USA). Measurements were carried out at liquid nitrogen temperature $(T = -195.8^{\circ}C)$ and relative pressures in the range of 0.0025 to 0.995 (relative pressure is defined as the ratio of actual pressure to saturation pressure). Adsorption isotherms were processed using the Dubinin–Radushkevich equations [26].

RESULTS AND DISCUSSION

In designing an SACM, we have to identify the type of host matrix for adsorbent particles. In fabricating adsorbent materials having a complex geometrical configuration, use of different clays or other mineral materials as host matrices encounters difficulties because of their small elastic strain. With polymer solutions used in syntheses of SACMs, the surface of an adsorbent filler becomes coated with a thin film (thickness of $10-50 \mu m$) that blocks access of adsorbate species to meso- and micropores of the adsorbent [27]. This issue can be circumvented by using, as a binder, organic polymers (e.g., fluoropolymers) that do not form films and can adjust to strains caused by, e.g., a drop in temperature or an aerodynamic load on adsorbent granules. This ensures that sorption materials used will not alter their shape during cyclic adsorption–desorption processes. We studied the dynamic activity of SACMs with different ratios of adsorbent (*A*) to binder (*B*) and the results are summarized in Table 1.

The analysis of these experimental data showed that a SAMC sample based on crystalline zeolite NaX incorporated into a fluoroplast host matrix exhibited faster kinetics of water vapor adsorption (by 15–20%) than its counterpart containing granulated zeolite NaX-V-1G (sample 7).

We investigated (Fig. 1) the dependences of ultimate strength σ and dynamic capacity a_d of the prepared block-shaped SACMs on amount *A* of adsorbent and the degree of dispersion *d* of the initial filler adsorbent (crystalline zeolite NaX).

The experimental dependences of Fig. 1a show that ultimate strength σ and dynamic capacity a_d diminish with increasing particle size of the initial adsorbent filler. This can be explained by the fact that, as the degree of dispersion of the adsorbent increases, its specific surface area decreases, which slows down the mass exchange and increases the diffusion resistance to the transfer of the adsorbate from gas phase to the solid phase in both the forward and backward directions (the effect is especially pronounced for degree of dispersion $d \ge 6$ µm). A decrease in the ultimate strength can be attributed to a fall in the number of point contacts per unit volume between adsorbent filler particles and the host matrix [27].

Increasing the degree of dispersion of the initial powdered zeolite beyond 6 μm leads to degradation of operational characteristics of the prepared adsorbing material: with numerous adsorption–desorption cycles, zeolite particles detach from the polymer matrix and undergo wearing, which results in higher aerodynamic resistance to gas flow.

Fig. 1. Variations of (*1*) ultimate strength σ and (*2*) dynamic capacity a_d with (a) the degree of dispersion *d* of the initial adsorbent filler (i.e., NaX crystallite) and (b) the content of adsorbent filler *A* in SACM samples.

The experimental dependences of Fig. 1b show that, as the adsorbent content in our SACMs grows, their ultimate strength σ and dynamic capacity a_d increase. We can hypothesize that the increase in a_d is related to a decrease in the thickness of polymer matrix film, a coating on the surface of adsorbent filler, and, accordingly, a decrease in diffusion resistance to adsorbate mass transfer. For $A \geq 85$ wt %, the dynamic capacity of prepared SACMs shows a dramatic fall. This is explained by densification of their structure, a decrease in the volume of transporting pores, and hindered mass transfer of the adsorbate into the bulk of the adsorbent material, as a consequence. In our best SACM samples, the adsorbent (*A*) : polymer matrix (*B*) ratio was in the range of $(80-88)$: $(20-$ 12) wt %.

The nature and quantity of a solvent used in preparing the initial suspension are two important factors affecting the quality of SACMs. The nature of the solvent affects the stability of the suspension, and good stability ensures compositional homogeneity of fabricated materials and lessens their defectiveness. In addition, the amount of solvent removed during the drying stage and the size of its molecules have a significant effect on the formation of the secondary porous structure of SACMs, which greatly defines their a_{d} , and influence the bonding strength between the adsorbent and binder.

Table 2 summarizes the results of experimental studies of our block-shaped SACMs with $d = 4-6 \,\mu$ m. In preparing these samples, dimethyl ketone (i.e., propanone-2 or acetone) and methyl ethyl ketone (butanone-2) were used as solvents, and the solvent was removed at temperatures T_S of 50 and 80 \degree C. The samples were also prepared with different values for the ratio of solvent (*S*) to polymer matrix (*M*).

The data of Table 2 tell us that substitution of methyl ethyl ketone for acetone (as solvents) had virtually no effect on the variations in a_d and σ. The optimal *S* : *M* ratio fell in the range of $15-30$ mg/g. At $S: M \le 15$ mg/g, a_d and σ diminish and the process of forming of SACM becomes more complicated owing to a high viscosity of the resulting suspension.

	Ratio of solvent	Dimethyl ketone		Methyl ethyl ketone	Solvent removal	
Sample no.	to polymer matrix ratio $S: M$, mg/g : mg/g	strength, σ , MPa	dynamic adsorption capacity, a_d , mg/g	strength, σ , MPa	dynamic adsorption capacity, a_d , mg/g	temperature, T_s , $\rm ^{\circ}C$
	10:1	0.73	187	0.75	186	50
$\overline{2}$	15:1	0.70	189	0.73	187	50
3	20:1	0.68	190	0.70	189	50
4	25:1	0.63	193	0.64	193	50
5	30:1	0.56	195	0.59	193	50
6	35:1	0.48	196	0.54	195	50
7	40:1	0.42	197	0.47	199	50
8	30:1	0.52	199	0.51	201	80

Table 2. Effect of solvent ratio on the mechanical and adsorption properties of prepared SACMs

Fig. 2. Variations of (*1*) mechanical strength σ and (2) dynamic adsorption capacity a_d for water vapor with solvent removal temperature T_S for prepared SACM samples.

This can be explained as being a result of a fall in the number of transporting pores that are produced during the solvent removal owing to the inhomogeneous structure of the resulting SACMs.

The interrelation between solvent removal temperature T_S , used at the stage of suspension drying, and dynamic adsorption capacity a_d and mechanical strength σ is shown in Fig. 2. In fabricating SACM samples, the solvent functioned as a plasticizing and a pore-forming agent.

The presented results suggest that raising T_S from 25 to 57°C produced an increase in dynamic capacity a_d of our SACM samples, while their strength σ showed a fall. A dramatic change in the properties of our SACM samples was observed at $T_s = 55-57$ °C owing to boiling of the acetone, which was accompanied by its intense evaporation and elimination from the suspension undergoing the forming process. At the same time, the number of transporting pores drastically increased. Parenthetically, the volume of produced SACM samples was larger by a factor of 2.5–3 than that of the initial suspension.

Experimental values for ultimate strength σ and Young's modulus *E* for samples with different *A* : *B* ratios and solvent removal temperature $T_s = 60^{\circ}$ C are listed in Table 3.

The data of Table 3 show that the maximum strength (σ = 0.65 MPa) is reached for the zeolite-tobinder ratio of 90 : 10, and the maximum elastic modulus $(E = 13 \text{ MPa})$ was reached when the ratio was 75 : 25. The best combination of strength and elastic properties was reached for the SACM sample with the ratio of 83 : 17 (indicated in this table).

Listed in Table 4 are the parameters characterizing the porous structure and true density ρ of our SACMs (samples 1–4), NaX crystallite (sample 5), and commercial granulated zeolite adsorbent NaX-V-1G (sample 6) manufactured by OAO Korporatsiya Roskhimzashchita according to TU 6-19-20-90. In fabricating SACM samples, acetone was used as a solvent in all cases.

The adsorption properties of investigated samples, as calculated using the Dubinin–Radushkevich equations, taking into account activity coefficient $\beta =$ 0.3918 [26], are summarized in Table 5.

The data of Table 5 show that the maximum limiting sorption volume $(0.308 \text{ cm}^3/\text{g})$ is observed for SACM sample 1, in which the zeolite content is 89.687 wt %.

Parameters	Zeolite-to-binder ratio, $A : B$, wt %							
	75:25	80:20	83:17	85:15	87:13	90:10		
Ultimate strength, σ , MPa	0.49	0.51	0.52	0.54	0.58	0.65		
Young's modulus, E , MPa	13.00	12.40	12.20	12.00	11.70	9.40		

Table 3. Mechanical parameters of our SACM samples

Table 4. Comparing the parameters of porous structure of different SACM samples

Sample no.	Sample composition	Ratio of zeolite to polymer matrix, temperature, $T_{\rm S}$, $A: M$, wt $\%$	$\rm ^{\circ}C$	Solvent removal Ratio of solvent to polymer matrix, S: M, mg/g: mg/g	Degree of dispersion, d , um	True density, ρ , kg/m ³
	SACM	90:10	50	20		2050
	SACM	85:15	50	20		2020
3	SACM	80:20	50	20		2030
4	SACM	75:25	50	20		2010
	NaX	100:0				1920
6	$NaX-V-1G$	87:13				2150

Sample no.	Sample composition	Limiting adsorption volume, W_0 , cm ³ /g	surface area, $S_{\rm SD}$, m^2/g	SACM specific Characteristic energy of adsorption, E_0 , kJ/mol	Zeolite aperture size, d_w Å	Mesopore diameter, d_{meso} , nm	A, wt $\%$
	SACM	0.308	844.291	16.750	8.950	4.333	89.687
2	SACM	0.267	809.460	16.180	9.256	3.504	85.796
3	SACM	0.258	736.486	16.670	9.388	2.787	81.178
4	SACM	0.244	707.752	16.950	9.306	3.143	75.317
5	$\rm NaX$	0.316	854.876	16.080	9.102	3.114	100.000
6	$NaX-V-1G$	0.243	733.846	16.110	7.938	1.911	86.986

Table 5. Absorption properties of different SACM samples

With its higher strength properties (provided no dusting occurs during use), this SACM sample is comparable in terms of parameter W_0 with granulated zeolite adsorbent NaX and surpasses zeolite NaX-V-1G by 27%.

CONCLUSIONS

Prepared block-shaped SACMs are shown to be promising materials for PSA-based purification and separation of gas mixtures.

The dynamic capacity, as measured by water vapor adsorption, of the SACM samples prepared using NaX crystallite with granule diameter of no more than 6 μm, which was incorporated into a fluoroplast matrix, was higher on average by 15–20% than that of commercial granulated zeolite NaX-V-1G.

In industrially produced SACMs, the mass ratio between adsorbent and polymer matrix must be 83 : 17 and the solvent removal temperature at the drying stage must be 60°C, with dimethyl ketone being the solvent of choice. This will ensure that no dusting occurs during use of SACMs and that the limiting sorption volume will be 27% greater than that of commercially produced zeolite adsorbent NaX-V-1G.

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