Nanocomposites Based on Silicon Dioxide of Different Nature with Functional Titanium Dioxide Nanoparticles

L. N. Obolenskaya*^a* **, A. A. Gaynanova***^a* **, G. V. Kravchenko***^a* **, G. M. Kuz'micheva***^a* **, E. V. Savinkina***a***, E. N. Domoroshchina***^a* **, A. M. Tsybinsky***^b* **, and A. V. Podbelsky***^c*

^a Federal State Budget Educational Institution of Higher Education "Moscow Technological University", pr. Vernadskogo 86, Moscow, 119571 Russia

b Fedorovsky All-Russia Research Institute of Mineral Raw Materials, Staromonetnyi per. 31, Moscow, 119017 Russia

c National Research University Higher School of Economics, ul. Myasnitskaya 20, Moscow, 101000 Russia

e-mail: galina_kuzmicheva@list.ru

Received February 19, 2015; in final form, October 15, 2015

Abstract—Nanocomposites $TiO₂/SiO₂$ with photocatalytic and adsorptive properties were prepared by codispersing of η -modification and anatase (commercial Hombifine N) and SiO₂ (opal, granules, ultrafine) in ethanol (or ethanol–water mixture in the presence of chlorophylls or porphyrins) with ultrasonic treatments of the mixture (method 1) and an aqueous solution of KOH with a microwave treatment (method 2), as well as the introduction of $SiO₂$ in the reaction mixture during the synthesis of TiO₂ by brief hydrolysis of sulfate titanyl (method 3). It was found that the state of titania in the sample (X-ray amorphous or nanocrystalline) and its deposition on $SiO₂$ nanocomposites depend on the method and the conditions of obtaining. It was established that the photocatalytic activity of nanocomposite $TiO₂/SiO₂$ (granules) (method 1) photosensitized by coproporphyrin I in the visible range and the photocatalytic activity of nanocomposite $TiO₂/SiO₂$ (opal) (method 3) in the near UV range exceed activity of the commercial sample of $TiSiO₄$ by more than 20-fold and \sim 7-fold, respectively. It was shown that the nanocomposite TiO₂/SiO₂ (opal) significantly reduces the concentration of cations (in particular, Be, Ni, Bi) in the model water systems.

DOI: 10.1134/S1995078016010110

INTRODUCTION

Nanocomposites are multiphase solid materials consisting of several components, wherein at least one of them has an average crystallite (particle) size in the nanoscale range (up to 100 nm). This determination can be used, in particular, for designation of solid combinations of massive matrix (carrier) and nanosized phase(s) (functional nanoparticles, reinforcing phase), which differ in their properties because of the differences in the structure and physicochemical properties. Mechanical, electrical, thermal, optical, electrochemical, catalytic, and other properties of nanocomposites differ from the properties of its components. Limitation in the scale for these effects is estimated as follows: <8 nm for (photo)catalytic activity, ≤ 10 nm for adsorption capacity, ≤ 20 nm for transferring hard magnetic material into soft, ≤ 50 nm for changing the refractive index, and ≤ 100 nm in order to achieve superparamagnetism, mechanical strength, or limiting changes in the structure of the composite.

In the formation of nanocomposites, either the properties of the matrix (carrier) to improve a number of physicochemical properties of the introduced therein functional nanoparticles can be used, or manifestation of synergism can be used—summing effect of the interaction of the properties of matrix and nanoparticles significantly superior to the effect of each individual component in the form of their simple sum. In this regard, promising are $TiO₂/SiO₂$ nanocomposites, where the components are amorphous silica matrix (carrier) and nanocrystalline titanium dioxide (functional nanoparticles).

Amorphous silica is characterized by thermal stability and high surface area. The silicon atoms on the surface of $SiO₂$ nanoparticles of any nature differ in the properties (in particular, crystal-chemical and chemical) of the same atoms in bulk, where each silicon atom is connected to four oxygen atoms: surface silicon atoms contain free bonds, which are saturated by hydroxyl OH groups, which determines the hydrophility and sorption properties of $SiO₂$. Silanol groups (Si-OH) of silica in strength are comparable with acetic acid ($pK_a \sim 6$), and for some SiO₂ brands, they are even stronger [1]. They play a key role in modifying the silica, the role of which is performed in this case by titanium dioxide nanoparticles.

The manifestation of the properties of nanoparticles of titanium dioxide (photocatalyst, adsorption, bactericidal, and others) is caused by the formation of surface hydroxyl groups with high reactivity upon

Fig. 1. View of initial $SiO₂$ carriers: granules—GR (a), opal—OP (b), ultrafine (Aeroxide)—UF (c).

decreasing the crystallite size to 100 Å or less, moreover crucially, so that the amount of such crystallites in the sample is close to 100% (approaching a unimodal particle size distribution) [2, 3]. These groups are self-generated upon contact with electrolytes on semiconductors with a large bandgap (Δ*Е =* 3.1– 3.4 eV for nano-scaled titanium dioxide), resulting in high adsorbing capacity of the surface of titanium dioxide nanoparticles [2–8].

Irradiation of the semiconductor of titanium dioxide with light with excitation energy above the bandgap energy of the material results in the absorption of photons (hv) and transfer of electrons (e^-) from the valence band (VB) to the conduction band (CB), thus generating a positively charged hole (h^+) in the valence band. The electron-hole charge carriers $(h⁺_{VB} + e⁻_{CB}),$ in turn, can recombine and dissipate the excess energy by a radiationless mechanism: *semiconductor* + *h*v → h^+ _{VB} + e⁻_{CB}; e⁻_{CB} + h^+ _{VB} \rightarrow *energy*, which reduces the overall efficiency of the photoexcitation process. The charge carriers not subjected to annihilation can migrate to the catalyst surface and initiate secondary reactions with the surface of adsorbed materials. The photoexcited electrons in the conduction band (CB) can react with oxygen to form superoxide or hydroperoxide radicals $e^-_{CB} + O_2 \rightarrow O_2$, and these reactive forms of oxygen can be involved in the decomposition of organic contaminants. Positively charged holes (*h*+) in the valence band (VB) can generate hydroxyl radicals: $h_{VB}^+ + H_2O \rightarrow \cdot OH + H^+$, which, in turn, oxidize the organic pollutants; in addition, ·OH radicals are formed under UV exposure, which play active role in photocatalytic activity [9, 10].

Titanium dioxide is widely used as a coating that is applied to porous oxide matrices of $SiO₂$ to increase its specific surface area and mechanical and thermal strength and to increase the selectivity of catalysts obtained on its basis [11–13]. In the literature, $TiO₂/SiO₂$ composites are described, which may be used as photocatalysts [14], solar energy converters [15], and key components to create self-cleaning surfaces [16].

The purpose of this work is to obtain, study, and compare the properties of $TiO₂/SiO₂$ nanocomposites with nanocrystalline titanium dioxide on the basis of silica of different nature.

OBJECTS AND METHODS OF STUDY

The objects of study are $TiO₂/SiO₂$ nanocomposites, for which silica served as a carrier: granules—GR; opal—OP (sample provided by Prof. M.I. Samoilovich); and ultrafine $SiO₂$ (Aeroxide) — UF (Fig. 1).

Table 1. Preparation of titanium dioxide samples

| Sample | Initial reagent/mass, g | Volume of water, mL | Temperature, °C/duration, min | Coagulant/concentration, mol $L^{-1}/$ volume, mL |
|----------|--|------------------------|----------------------------------|--|
| $TiO2-1$ | \vert TiOSO ₄ · xH_2SO_4 · $yH_2O/7.35$ | 40 | 90/9 | HC1/8.8/54 |
| | TiO_{2} -2 $TiOSO_{4} \cdot xH_{2}SO_{4} \cdot yH_{2}O/7.35$ | 20 | 80/44 | HCl/8.8/54 |
| $TiO2-3$ | $TiOSO_4 \cdot xH_2SO_4 \cdot yH_2O/7.35$ | 40 | 89/11 | KCl/8.8/30 |
| $TiO2-4$ | $TiOSO4 \cdot 2H2O/4.32$ | 42 ¹ | 89/44 | $Cs_2SO_4/0.7/30^2$ |
| $TiO2-5$ | $TiOSO_4 \cdot xH_2SO_4 \cdot yH_2O/7.35$ | 40 | 75/8 | KCl/8.8/40 ³ |

¹4.7 M aqueous solution of H_2SO_4 was taken instead of water.

 2 After adding the coagulant, the reaction mixture was stirred at room temperature for 45 min.

 3 Before adding coagulant, the reaction mixture was aged at room temperature for 30 days.

¹ Addition of 100 mL of water, holding at 80 °C for 39 min.

² Addition of 54 mL 8.8 M hydrochloric acid.

 3 Addition of 34 mL 8.8 M hydrochloric acid.

Samples with titanium dioxide (Table 1) used for the production of composites and/or as a reference sample were obtained by hydrolysis of titanyl sulfate followed by separation of the product by vacuum filtration (overnight), washing with water and acetone, and drying at 90°C (starting reagents are TiOSO₄*x*H₂SO₄*y*H₂O (Aldrich, CAS no. 123334-00-9) or TiOSO₄ 2H₂O (Aldrich, CAS no. 13825-74-6)). In addition, commercial samples of Hombifine N and "TiSiO₄" (Aldrich, MDL no. MFCD00799909) were used.

Method 1. Composites were synthesized by joint dispersal of the samples with titanium dioxide and silica in ethanol or ethanol–water mixture in the presence of porphyrins.

The sample of $TiO₂-3/SiO₂(GR-2)$ was obtained by dispersal of 0.4 g of $TiO₂-3$ in ethanol followed by the introduction of coproporphyrin I tetraisopropyl ester (0.0058 g) using ultrasound (20 min), addition of 10 mL of 1 M NaOH, stirring (15 min), and introduction of 0.1 g of granular silica. The resulting solid phase a day later was separated by vacuum filtration, washed with water and acetone, and dried at 90°C.

The sample of $TiO₂-4/SiO₂$ (UF-Chl) was obtained by co-dispersion of 0.34 g of $SiO₂ (UF)$ and 0.06 g of TiO₂-4 in a water−ethanol mixture (2 : 1 by volume) in the presence of chlorophyll A (CAS no. 479-61-8) and B (CAS no. 519-62-01) (total weight is 50 mg) followed by sonication (5 min). The resulting solid phase a day later was separated by vacuum filtration, washed with water and acetone, and dried in air, and then ground in a porcelain mortar.

Method 2: Composites were obtained by co-dispersion of nano-TiO₂ and $SiO₂$ in an aqueous solution of KOH with microwave (MW) treatment.

Samples of TiO₂-5/SiO₂ (OP), TiO₂-5/SiO₂ (UF), Hombifine $N/SiO_2(UF)$, and Hombifine $N/SiO_2(OP)$ were prepared by dispersion of 0.3 g $TiO₂$ -5 or Hombifine N and 0.1 g of $SiO₂$ (OP, UF) in 40 mL of 0.1 M aqueous solution of KOH, followed by microwave (MW) treatment of the obtained dispersions for 30 min at room temperature. The resulting solid phase was separated by vacuum filtration, washed with water and acetone, and dried at 90°C.

Method 3: Composites were synthesized by introducing silica into the reaction mixture during the synthesis of titanium dioxide by brief hydrolysis of titanyl sulfate (Table 2). After further treatment (in some cases), the resulting solid phase a day later was separated by vacuum filtration, washed with water and acetone, and dried at 90°C. As reference samples in the study of the photocatalytic activity (PCA), $TiO₂-1$ and $TiO₂$ -2 samples not containing $SiO₂$ were obtained, so before introduction of $SiO₂$ to the reaction mixture $(TiO₂/H₂SO₄$ sol), part of the reaction mixture was separated and $TiO₂$ was precipitated from it in the absence of $SiO₂$ (Table 1).

X-ray survey of the samples with rotation was performed on an HZG-4 diffractometer (Ni-filter): Cu*K*_α on the diffracted beam in continuous $(1^{\circ}C/\text{min})$ or stepwise (dialing time of pulses of 10 s., step size of 0.02°, range of 20 angles from 2° to $50^\circ - 80^\circ$) modes. The crystallite size were calculated (excluding possible microstrains) according to the formula *D* = *К*λ/βcosθ, where $\lambda = 1.54051$ Å is the wavelength (Cu K_{α}), $2\theta \sim 25^{\circ}$ for TiO₂ and $2\theta \sim 22^{\circ}$ for SiO₂ (θ is the scattering angle), β is the integral width of the peak, and the particle shape factor (Scherrer ratio) *K* was taken to be 0.9. The standard deviation is $\pm 5\%$.

Study of the microstructure of $TiO₂/SiO₂$ nanocomposite (OP) was performed by scanning electron microscopy (SEM) using a JSM 7500F high resolution electron microscope. Chemical qualitative and quantitative composition was determined by X-ray microanalysis (XRMA) at three points, using an energy dispersive X-ray microanalyzer, established in the same high-vacuum chamber of the microscope. Quantitative analysis was performed using the INCA Penta FET mathematical program for all elements with atomic number >3. The limit of repeatability of the results was 0.1% [17]. For the analysis of photographs obtained by SEM, the authors developed a specialized program. Its purpose was automatization of the isolation process on the photograph of spherical contours of the images of particles and construction of the histogram of particle size distribution. The program was written in the C# language and oriented on Net Framework platform versions 3.5 and above. The initial data for the program are the image of micrographs and particle size range upon the construction of the histogram. Processing (counting) of the photograph begins with automatic recognition of the spherical contours of particles. To select the borders of images in the program, the Canny operator is used [18], supplemented by the reduction filter of the boundaries to a spherical shape. Owing to the fact that the automatic marking of the contours of particles for "bad" (noisy) photos gives unsatisfactory results, the program includes a visual check of the results of automatic selection of boundaries and manual dialogue adjustment of marking boundaries. In addition, the program allows one to exclude the stage of automatic construction of boundaries and highlight spherical boundaries of particles manually. The procedure of such dialogue markup is partially automated.

Study of the adsorption capacity of the samples was performed on synthetic mixtures (ICP-MS-68A-A standard from High-Purity Standards, USA; the solution contains 48 elements in the concentration of 10 mg/L of each); 10 mL of standard solution with the concentration of 2 ppm $(C_{initial})$ ions in 0.4% nitric acid was added to the weighed portion of the sample with the mass of 30 mg. The solutions were shaken on a shaker for 3 h and then filtered through a 0.45 μm membrane filter. The concentration of metals in the solutions after sorption (C_{final}) was measured by inductively coupled plasma mass spectrometry (ICP-MS) (Agilent 7500c, USA). To optimize the performance of the instrument with respect to the sensitivity, mass resolution, and calibration of masses of elements being determined, Agilent Tuning Solution containing Li, Y, Ce, Tl, and Co (concentration of $10 \mu g/L$) in 2% HNO₃ solution was used. The ratios of $140Ce^{16}O^{+}/140Ce^{+}$ ($137Ba^{2+}/137Ba^{+}$) were minimized to reduce the interferences from oxide ions (doubly charged ions). Such instrumental parameters as carrier gas flow, position of the burner, accelerating voltage, speed of the peristaltic pump, and integration parameters were optimized to achieve higher sensitivity and reduce the number of isobaric interferences. The maximum power delivered to the plasma was 1200 W. Matched argon flow rates were as follows: plasma forming—15 L/min, auxiliary—0.8 L/min, spray—1.4 L/min. For each mass (from 27 to 208 amu), 40 scans were used (50 ms) in triplicate. All results of

ICP-MS analysis (construction of calibration straight lines, setting of the device using special solutions, statistical data, and concentrations of elements in solutions) were obtained using the ICP-MSTopAgilent software package.

Electronic absorption spectra of the samples were taken with an Akvilon SF 103 spectrophotometer.

Photocatalytic activity was studied on the model decomposition reaction of methyl orange organic dye (MeO, CAS no. 547-58-0). Discoloration of methyl orange was the standard model reaction allowing one to compare the activity of different photocatalysts. To the aqueous dye solution (initial concentration of 0.05 mmol/L) of pH \sim 3–4, powder of either nanocomposite or its starting components were added (mass content of 0.05 g/L). The resulting slurry (volume of 50 mL) was irradiated with light in the UV range of 250–400 nm (DRT irradiator—UVO-V-4, DRT 125 mercury lamp, irradiator power—125 W; Cam—Camelion UV lamp, 26 W) or in the visible range (Hal—halogen lamp, irradiation at wavelengths of ≥450 nm, power of 250 W; Lum—energy saving reflector lamp with color temperature of 4200 K, which corresponds to the radiation maximum in the visible range of about 560–570 nm; power of 15 W (according to the amount of emitted light photons, it corresponds approximately to a 75 W halogen lamp) in an open container (the irradiation area of 0.07– 0.09 m^2) at constant stirring using a magnetic stirrer for 150–240 min. Aliquots (10 mL) were taken from the reaction mixture and, if necessary, the solid phase was separated by centrifugation (OPn-8UKhL42 centrifuge).

Change in the concentration of MeO during the photoreaction $(c/c_0$ value) was controlled by changing the intensity of the absorption band of the chromophore group at $\lambda = 450 - 470$ nm. The used model photoreaction refers to reactions of a pseudo first order, for which the rate constant (*k* value) is calculated as the slope of the dependence of the change in the substrate concentration c/c_0 on the irradiation time τ in rectifying semilogarithmic coordinates (the accuracy of the linear approximation of kinetic curves on the *R*-factor in all cases is not less than 0.95; the error in determining the reaction rate constant is 10%).

RESULTS AND DISCUSSION

Characterization of Initial Samples and Nanocomposites

Figure 2 shows the diffraction patterns of the silica of different nature used in the present work.

Samples with ultrafine (UF) (Fig. 2a) and granular (GR) (Fig. 2b) $SiO₂$, in addition to amorphous silica, contain the impurity of amorphous hydrated silica (HS) of probable composition of $SiO_{2-x}(OH)_{2x} \cdot yH_2O$ (characteristic diffraction reflection at $2\theta \sim 11^{\circ}-12^{\circ}$) in different amounts (according to the ratio of the

Fig. 2. Diffractograms of ultrafine $SiO₂$ (UF) (a), $SiO₂$ granules (GR) (b), and opal (OP) (c).

reflection intensity with $2\theta \sim 11^{\circ} - 12^{\circ}$, in GR more than compared to UF). Sample with opal is singlephase (Fig. 2c).

In addition to the composition, samples with $SiO₂$ differ in the characteristics of the microstructure (lowest specific surface, determined by the comparative method [17], and the pore sizes for OP, and the largest for GR). The average crystallite sizes of them are about the same, a little more for granular (GR) SiO₂ (Table 3). In this case, the average crystallite size correlates with the content of HS in the composition of the samples: with the increase in the intensity ratio of

Table 3. Characteristics of silica

¹ Coarse.

2 Crushed.

Fig. 3. Diffractogram of TiO₂-3 (a) and TiO₂-3/SiO₂ (GR) nanocomposite (b) (method 1).

diffraction reflections of $SiO₂$: HS, which is proportional to $SiO₂$ and HS contents in the sample (Table 3), the average crystallite size (D value) of SiO₂ tends to decrease. This may indicate the impurity nature of HS in the samples with $SiO₂$, instead of depositing on the surface of the crystallites, which would cause an inverse relationship.

Samples Obtained by Method 1

Figure 3 shows the diffraction patterns of $TiO₂-3$ and $TiO₂-3/SiO₂$ (GR) nanocomposite obtained by joint dispersion of $TiO₂ - 3$ and granular silica in the presence of coproporphyrin I tetraisopropyl ester. Initial $TiO₂-3$ contains η-TiO₂ (characteristic diffraction reflections at $2θ$ ~ $4^{\circ}-5^{\circ}$ and $\sim 33^{\circ}$ [2]) and a large amount of hydrated titanium dioxide $TiO_{2-x}(OH)_{2x} \cdot yH_2O$ (HD). Nanocomposite $TiO₂-3/SiO₂(GR)$ (Fig. 3b) also contains HD as does the source titanium dioxide (Fig. 3a), although its amount in the nanocomposite is much less (Table 4). The diffractogram of $TiO₂-3/SiO₂(GR)$ does not contain diffraction reflections belonging to titanium dioxide (Fig. 3b), indicating either its small amount in the nanocomposite (below the sensitivity of this X-ray experiment) or its X-ray amorphism, which

is more likely. The average crystallite size of $SiO₂$ for $TiO₂-3/SiO₂(GR)$ (Table 4, Fig. 3b) is greater than that for the initial $TiO₂-3$.

Comparing the diffraction patterns of the $TiO₂ - 4$ sample (Fig. 4a) and $TiO₂-4/SiO₂(UF-Chl)$ nanocomposite prepared by joint dispersion of $SiO₂$ (UF) and $TiO₂-4$ in the presence of chlorophylls (Fig. 4b) shows the opposite effect (Table 4). In turn, for $TiO₂/SiO₂(UF-Chl)$ (Table 4, Fig. 4b), the mean crystallite sizes were greater $(D = 47(2)$ Å) than for $SiO₂(UF-Chl)$ (composition: $SiO₂ + HS$ (1.5 : 1), $D =$ 41(2) Å) (Fig. 4c), which is synthesized similarly to $TiO₂/SiO₂(UF-Chl)$, but without the addition of titanium-containing components. Analyzing the data in Table 4, one can notice an inverse relationship between the size *D* and HD or HS content in the composition of the samples with η -TiO₂ or nanocomposites: the more HD or HS, the smaller the value *D*. This relationship indicates the presence of hydrated particles in the nanocomposites and nanoparticles of initial samples of $TiO₂$ on the surface, unlike the initial samples of $SiO₂$ of different nature (see above).

| Samples of $TiO2$ | | | | | |
|----------------------------|---|--------------------------------|--|--|--|
| Sample/method of synthesis | Composition according to X-ray phase analysis | Size of crystallites D , A | | | |
| $TiO2-3$ | η -TiO ₂ + HD (1 : 1) (Fig. 3a) | 37(2) | | | |
| $TiO2-4$ | η -TiO ₂ + HD (2 : 1) (Fig. 4a) | 50(3) | | | |
| $TiO2-5$ | η -TiO ₂ (Fig. 5a) | 38(2) | | | |
| Hombifine N | anatase (Fig. 6a) | 81(4) | | | |
| $TiO2-1$ | η -TiO ₂ + HD (1.5 : 1) | 47(2) | | | |
| $TiO2-2$ | η -TiO ₂ + HD (4 : 1) | 55(3) | | | |
| $TiO2/SiO2$ nanocomposites | | | | | |
| $TiO2-3/SiO2(GR-2) / 1$ | $SiO2 + HD (5:1) (Fig. 3b)$ | 44(2) | | | |
| $TiO2-4/SiO2(UF-Chl) / 1$ | $SiO2 + HS (3.5:1) (Fig.3c)$ | 47(2) | | | |
| $TiO2-5/SiO2(UF) / 2$ | η -TiO ₂ ¹ (Fig. 5b) | | | | |
| $TiO2-5/SiO2(OP) / 2$ | η -TiO ₂ ¹ (Fig. 5c) | | | | |
| Hombifine $N/SiO2(UF) / 2$ | anatase (Fig. 6b) | 77.2(4) | | | |
| Hombifine $N/SiO2(OP) / 2$ | anatase + HS $(8:1)$ (Fig. 6c) | 60.0(3) | | | |
| $TiO2/SiO2(UF) / 3$ | $SiO2$ (Fig. 7a) | 52(3) | | | |
| $TiO_2/SiO_2(UF-1)/3$ | $SiO2$ + anatase + HD (or HS) $(4:1)3$ | $40(2)/52(2)^2$ | | | |
| | (Fig. 7b) | | | | |
| $TiO2/SiO2(UF-2) / 3$ | Anatase (Fig. 7c) | $-$ /58(3) ² | | | |
| $TiO_2/SiO_2(GR) / 3$ | $SiO2$ (Fig. 7a) | 50(3) | | | |
| $TiO_2/SiO_2(GR-1)/3$ | $SiO2$ (Fig. 7b) | 55(3) | | | |
| $TiO_2/SiO_2(OP)/3$ | $SiO2 + HD (8:1)$ (Fig. 8a) | 38(3) | | | |
| $TiO2/SiO2(OP-1)/3$ | SiO_2 + anatase + HD $(7:1)^3$ (Fig. 8b) | | | | |

Table 4. Characteristics of the samples with $TiO₂$ and $TiO₂/SiO₂$ nanocomposites

¹ Strongly hydrated.

 2 D_{SiO2}/D_{TiO₂.}

³ In parentheses, the ratio of SiO_2 : HD or HS is given.

Samples Obtained by Method 2

Figure 5 shows the diffraction patterns of the initial sample of TiO₂-5 (Fig. 5a), containing η -TiO₂, and the $TiO₂-5/SiO₂(UF)$ (Fig. 5b) and $TiO₂-5/SiO₂(OP)$ (Fig. 5c) nanocomposites.

As can be seen from Figs. 5b and 5c, nanocomposites containing highly hydrated η -TiO₂ are obtained, as indicated by the blurred reflection in a large range of 20 angles of \sim 24°–35°, while the nature of the initial silica does not affect the composition of nanocomposites and average crystallite size.

On the diffractogram of the commercial sample of Hombifine N (Fig. 6a) and nanocomposites of Hombifine $N/SiO₂$ (UF) and Hombifine $N/SiO₂$ (OP) (Figs. 6b, 6c; Table 4), diffraction reflections are present (2 $\theta \sim 25^{\circ}$, ~38°, ~48°), belonging to anatase (JCPDS 89-4921), and in Hombifine $N/SiO₂$ (UF), HD/HS is found (Table 4, Fig. 6b) as in the original UF (Fig. 2a). Diffraction peaks belonging to $SiO₂$ are absent in the diffractograms.

NANOTECHNOLOGIES IN RUSSIA Vol. 11 Nos. 1–2 2016

The average size of crystallites of anatase in Hombifine $N/SiO₂$ is less than that in the initial Hombifine N (Table 4), while in this case the presence of HD or HS in the composition of the samples decreases the size of crystallites.

Thus, the dispersion medium of initial dioxides of silicon and titanium and the processing method of the reaction mixture have a significant effect on the composition of the resulting nanocomposites: upon using method 1, nanocomposites contain X-ray amorphous titanium dioxide, and method 2 promotes the formation of nanocrystalline titanium dioxide.

Samples Obtained by Method 3

Upon the preparation of nanocomposites based on ultrafine (UF) silica (Fig. 7), an increase in the duration of heating of the reaction mixture (Table 4) leads to an increase in the content of nano titanium dioxide in their composition: only $SiO₂$ is present in $TiO₂/SiO₂ (UF)$ (although an asymmetry of the peak

Fig. 4. Diffractogram of TiO₂-4 (a), TiO₂-4/SiO₂ (UF-Chl) nanocomposite (b) (method 1), and reference sample of SiO₂ (UF-Chl) (c).

Fig. 5. Diffractograms of TiO₂-5 precursor (a) and TiO₂-5/SiO₂ (UF) nanocomposites (b) and TiO₂-5/SiO₂ (OP) (c) (method 2).

Fig. 6. Diffractograms of the initial sample of Hombifine N (precursor) (a) and nanocomposites of Hombifine $N/SiO₂ (UF)$ (b) and Hombifine $N/SiO₂$ (OP) (c) (method 2).

Fig. 7. Diffractograms of TiO_2/SiO_2 (UF) nanocomposites (a), TiO_2/SiO_2 (UF-1) (b), TiO_2/SiO_2 (UF-2) (c) (method 3).

NANOTECHNOLOGIES IN RUSSIA Vol. 11 Nos. 1–2 2016

Fig. 8. Diffractogram of TiO_2/SiO_2 (GR) (a) and TiO_2/SiO_2 (GR-1) (b) nanocomposites (method 3).

is observed with $2\theta \sim 22^{\circ}$ to higher angles, i.e., toward $2\theta \sim 25^{\circ}$ with 100% diffraction reflection for anatase) (Fig. 7a); in $TiO₂/SiO₂$ (UF-2), anatase and an amorphous phase are found together with silica (Fig. 7b); and in $TiO₂/SiO₂$ (UF-2), only TiO₂ is observed with the structure of anatase (Fig. 7c).

Reflections are detected on diffraction patterns of $TiO₂/SiO₂$ (GR) (Fig. 8a) and $TiO₂/SiO₂$ (GR-1) (Fig. 8b) nanocomposites corresponding to only silica (Table 4, Fig. 8) (probably, titanium dioxide is in the X-ray amorphous state). Increasing the duration of synthesis upon the preparation of nanocomposites based on opal promotes the phase transition X-ray amorphous–nanocrystalline anatase; the latter is absent in the composition of $TiO₂/SiO₂ (OP)$ (Fig. 9a) and observed in the composition of $TiO₂/SiO₂ (OP-1)$ (Fig. 9b).

At the same time, there is the same tendency in the variation of the average crystallite size in nanocomposites: a greater content of HS in the samples leads to a decrease in the average crystallite sizes in nanocomposites of $TiO₂/SiO₂$ (Table 4), which are larger than the initial matrices of $SiO₂$ (Table 3). This does not preclude the deposition of nanoparticles of titanium dioxide on the surface of the matrix.

Electronic absorption spectra (EAS) of the obtained samples differ little from the electronic absorption spectrum of titanium dioxide. Figure 10 as an example presents EAS of a $TiO₂/SiO₂ (OP)$ sample.

Photocatalytic Properties

The samples obtained by method 1 are colored, so photocatalytic activity (PCA) in the visible region of the spectrum is expected for them. The samples obtained by methods 2 and 3 are colorless; i.e., they can display PCA only upon irradiation in the UV range.

Method 1

The samples of titanium dioxide used to obtain nanocomposites by method 1, TiO₂-3 and TiO₂-4, exhibit PCA in the UV region of the spectrum with $k =$ 0.024 and 0.022 min⁻¹ (the source of radiation is DRT), respectively. In the visible range of the spectrum, they are not active. PCA of the samples were compared with the photoactivity of the commercial reference material ("TiSiO₄" powder), on the diffractogram of which (Fig. 11) amorphous reflexes can be seen, likely from HS ($2\theta \sim 11^{\circ} - 12^{\circ}$) and SiO₂ ($2\theta \sim 22^{\circ}$),

Fig. 9. Diffractogram of TiO_2/SiO_2 (OP) (a) and TiO_2/SiO_2 (OP-1) (b) nanocomposites (method 3).

as well as characteristic diffraction reflections for anatase with crystallite sizes $(D = 160(3)$ Å) significantly higher than the similar dimensions of nano titanium dioxides presented in this paper (Table 4).

To improve the activity of $TiO₂$ in visible light, different ways are developed, and sensitization by dyes is

Fig. 10. EAS of $TiO₂/SiO₂$ (OP) sample suspended in water $(\Delta E_g = 3.61 \text{ eV})$.

NANOTECHNOLOGIES IN RUSSIA Vol. 11 Nos. 1–2 2016

considered a promising strategy for activating $TiO₂$ photocatalysis by visible light [9]. The process involves the absorption of light (mainly visible) by an organic dye (or colored contaminant), known as a sensitizer, followed by injection of electrons from the excited molecule of the sensitizer to the conduction band of the semiconductor material. The obtained cation radicals can lead to a number of oxidative processes and formation of active oxygen forms, which leads to degradation of various organic contaminants [19–21] or dyes, in particular, MeO (the mechanism of the process is virtually unknown, but likely cleavage of nitrogen–nitrogen bond occurs in the first stage, which leads to the disappearance of the color).

In the visible range (radiation source is Hal), the activity of the $TiO₂/SiO₂$ (GR-2) sample photosensitized by coproporphyrin I obtained by method 1 exceeds the activity of the commercial reference sample of "TiSiO₄" under the same conditions more than 20-fold (Fig. 12). This indicates high efficiency of the sensitization of the composite by coproporphyrin I molecules.

The degree of concentration reduction of MeO due to its photodegradation (irradiation with Lum lamp) in the presence of $SiO₂$ (UF-Chl) nanocomposite not

Fig. 11. Diffractogram of "TiSiO₄" (Aldrich; MDL no. MFCD00799909).

containing titanium dioxide is equal to the degree of reduction of its concentration in the absence of light quanta (Table 5); i.e., the sample of $SiO₂$ (UF-Chl) does not manifest appreciable photocatalytic activity. It should be noted that, upon using "TiSiO₄" as a reference sample, reduction of the concentration of methyl orange (MeO) within the accuracy of the spectrophotometric method (0.1%) is not observed.

The rate of decrease in the concentration of MeO in the presence of $TiO₂/SiO₂(UF-Chl)$ composite upon irradiation with visible light is \sim 15-fold more than in the presence of $SiO₂(UF-Chl)$ (Fig. 13). It significantly exceeds the rate of decrease in the concentration of MeO in the dark, which indicates relatively high photocatalytic activity of $TiO₂/SiO₂(UF-Chl)$. At the same time, the recovery rate of MeO by the sample of $TiO₂/SiO₂(UF-Chl)$ in the dark exceeds the corresponding value for the sample of $SiO₂(UF-Chl)$ \sim 3-fold (Table 5). This suggests that the introduction of the titanium oxide component to $SiO₂(UF-Chl)$ leads not only to the appearance of the photocatalytic activity for it but also to the increase in its sorptive capacity with respect to MeO.

Method 2

Upon the model photocatalytic decomposition reaction of methyl orange dye in the UV range (source

Fig. 12. Dependence of interpolated values of negative natural logarithm of the relative change in the concentration of MeO ($-\ln(c/c_0)$) on the duration (τ) of irradiation of photoreactivity mixture with light in the visible range (OSRAM halogen lamp, power of 250 W; temperature of 45−55°C) in the presence of the samples: (1) TiO₂/SiO₂ (GК-2) nanocomposite, sensitized by Coproporphyrin I (reaction rate constant $k = 0.0094$ min⁻¹); (2) reference sample of "TiSiO₄" (Aldrich) $(k = 0.0004 \text{ min}^{-1})$.

power is 26 W, irradiation time is 4 h), it was shown that the application of commercial titanium dioxide of Hombifine N grade on opal or ultrafine (UF) silica by method 2 results in a decrease in photocatalytic activity (Fig. 7, $k = 0.005$, 0.002, and 0.003 min⁻¹, respectively). Application of $TiO₂-5$ obtained in this paper on Opal (OP) and ultrafine (UF) silica gives the samples with even lower photocatalytic activity (Fig. 7). Thus, method 2 does not allow obtaining nanocomposites that are effective photocatalysts.

Method 3

Obtaining photocatalytic active nanocomposites based on ultrafine (UF) silica by method 3 is more promising than by method 2. Thus, the sample of $TiO₂/SiO₂$ (UF-1) is characterized by higher PCA than Hombifine N (Fig. 14).

It should be noted that increasing the synthesis time upon the preparation of nanocomposites based

Table 5. Decrease in the concentration of MeO (%) after 3 h of stirring of its solution with the samples modified by the mixture of chlorophyll A and B (Table 1)

| Irradiation conditions Sample | | In the absence of light quanta \vert Upon irradiation by fluorescent lamp (power of 15 W) |
|----------------------------------|-----|---|
| $SiO2(UF-Chl)$ | 0.4 | 0.4 |
| $TiO2/SiO2(UF-Chl)$ | 1.4 | 14.7 |

Fig. 13. Dependence of interpolated values of negative natural logarithm of the relative change in the concentration of MeO ($-\ln(c/c_0)$) on the duration (τ) of irradiation of photoreactivity mixture with light in the visible range (Camelion energy-saving lamp, power of 15 W; temperature of 20−30°C) in the presence of the samples: (1) SiO₂ (UF-Chl) (reaction rate constant $k = 0.00003 \text{ min}^{-1}$); (2) TiO₂/SiO₂ nanocomposite (UF-Chl) $(k=0.0007 \text{ min}^{-1})$; (3) reference sample of "TiSiO₄" (Aldrich) (MeO concentration remains constant).

on ultrafine (UF) of silica reduces the photocatalytic activity (Fig. 14, curves *1* and 2 for $TiO₂/SiO₂$ (UF-1) and $TiO₂/SiO₂(UF-2)$, respectively). It is possible that large differences in PCA of $TiO₂/SiO₂(UF-1)$ and $TiO₂/SiO₂(UF-2)$ nanocomposites (method (Fig. 14) can be explained by the combined action of Si-OH and Ti-OH on the surface of nanoparticles in the case of $TiO₂/SiO₂(UF-1)$ (Fig. 7).

With reduction of the synthesis time by method 3, the photocatalytic activity of the nanocomposite based on ultrafine (UF) silica exceeds the activity of Hombifine N to a greater degree (Fig. 15, curve *2*): $k = 0.017$ and 0.028 min⁻¹ for Hombifine N and $TiO₂/SiO₂$ (UF), respectively (radiation source is DRT). Thus, the synthesis time of such samples should be minimized.

Method 3 allows also obtaining more active photocatalysts as compared with method 2 if opal (OP) is used as a carrier. The sample of $TiO₂/SiO₂$ (OP) (Fig. 15, curve *1*) shows greater activity than Hombifine N, $TiSiO₄$, and titanium dioxide obtained under similar conditions without adding silica: $k = 0.057$, 0.017, 0.008, and 0.019 min-1, respectively (radiation source is DRT). However, with slightly increasing temperature and duration of the synthesis, PCA of the nanocomposite significantly decreases and becomes less than that for the reference samples (Fig. 14, curve *3* for $TiO₂/SiO₂ (OP-1)$).

Fig. 14. Dependence of interpolated values of negative natural logarithm of the relative change in the concentration of MeO ($-\ln(c/c_0)$) on the duration (τ) of irradiation of photoreactivity mixture with light in UV range (Camelion energy-saving lamp, power of 26 W; temperature of 30−40°C) in the presence of the samples: (*1*) $TiO₂/SiO₂(UF-1)$ ($k = 0.006$ min⁻¹); (2) $TiO₂/SiO₂(UF-2)$ $(k = 0.0008 \text{ min}^{-1});$ (3) TiO₂/SiO₂(OP-1) ($k =$ 0.001 min⁻¹); (4) TiO₂-5/SiO₂(UF) ($k = 0.0004$ min⁻¹); (5) TiO₂-5/SiO₂(OP) ($k = 0.0006$ min⁻¹); (6) Hombifine $N/SiO₂(UF)$ ($k = 0.003$ min⁻¹); (7) Hombifine $N/SiO_2(OP)$ ($k = 0.002 \text{ min}^{-1}$); (8) "TiSiO₄" reference sample (Aldrich) $(k = 0.001 \text{ min}^{-1})$.

The application of titanium dioxide by method 3 on granules (GR) of titanium dioxide gives a less efficient photocatalyst than upon using opal (OP). Although the obtained sample of $TiO₂/SiO₂$ (GR-1) exceeds the industrial samples of Hombifine N and $TiSiO₄$ in photocatalytic activity, but is inferior to the sample of titanium dioxide $TiO₂-2$ obtained under same conditions, but without addition of silica: $k =$ 0.019, 0.017, 0.008, and 0.031 min-1, respectively (radiation source is DRT) (Fig. 15).

When comparing photocatalytic properties of $TiO₂/SiO₂$ composites on all three carriers upon irradiation in the near UV range (irradiation power is 125 W, temperature is about 60° C, irradiation time is 1−1.5 h), it was found that the activity of the $TiO₂/SiO₂(OP)$ composite (Fig. 15, curve *1*) exceeds the activity (under same conditions) of the "TiSi O_4 " commercial sample (Fig. 15, curve *4*) ~7-fold, a factor of two higher than the activity of the $TiO₂/SiO₂(UF)$ composite (Fig. 15, curve *2*), which, in turn, is half as much as the activity of $TiO_2/SiO_2(GR-1)$ (Fig. 15,

Fig. 15. Dependence of interpolated values of negative natural logarithm of the relative change in the concentration of MeO ($-\ln(c/c_0)$) on the duration (τ) of irradiation of photoreactivity mixture with light in UV range (DRT-125 mercury lamp, power of 125 W; temperature of 60−70°C) in the presence of the samples: (1) TiO₂/SiO₂ nanocomposite (OP) $(k = 0.057 \text{ min}^{-1})$; (2) TiO₂/SiO₂ nanocomposite (UF) $(k = 0.028 \text{ min}^{-1})$; (3) TiO₂/SiO₂ nanocomposite (GR-1) $(k = 0.019 \text{ min}^{-1})$; (8) "TiSiO₄" reference sample (Aldrich) $(k = 0.008 \text{ min}^{-1})$. TiO₂-1; $k =$ 0.019 min⁻¹. TiO₂-2; $k = 0.031$ min⁻¹.

curve *3*). At lower lamp power and temperature of the photoreactivity mixture (irradiation power is 26 W, temperature is about 40° C), the most active is the nanocomposite with ultrafine (UF) $SiO₂$: PCA of $TiO₂/SiO₂(UF-1)$ sample (Fig. 14, curve *1*) is sixfold higher than the activity of both the $TiO₂/SiO₂(OP-1)$

Table 6. Degree of sorption $(R, \%)$ of TiO₂/SiO₂ (OP) and $TiO₂/SiO₂$ (UF) nanocomposites obtained by method 3 (Table 1)

| Element | $R, \%$ | | |
|----------------|---|---|--|
| | TiO ₂ /SiO ₂ (OP) | TiO ₂ /SiO ₂ (UF) | |
| Be | 33.03 | \ast | |
| Mg | 20.44 | 34.68 | |
| Co | 31.44 | \ast | |
| Ni | 34.78 | \ast | |
| In | 25.57 | \ast | |
| Ce | 25.36 | \ast | |
| P _b | 26.83 | \ast | |
| Bi | 36.56 | \ast | |
| \mathbf{A} | \ast | 20.78 | |
| Ca | \ast | 29.47 | |

*Sorption degree *R <* 20%.

sample (Fig. 14, curve *3*) and the reference sample of "TiSiO₄" (Fig. 14, curve δ).

On kinetic curves of the photocatalytic decomposition reaction of MeO in the presence of all obtained nanocomposites of $TiO₂(TiO₂+HD)/SiO₂$ (upon irradiation both in the UV and in the visible range), the induction period characteristic of kinetic curves of MeO photodecomposition in the presence of the samples with η-modification, including with HD impurity, is absent [17, 22–24]. This can be explained by a higher value of the specific surface area of silicate carriers compared with the samples composed of only η-modification+HD [24].

Furthermore, as was stated in the introduction, pK_a of silanol surface groups is close to pK_a of acetic acid. During studies conducted in parallel, we found that namely this acid is the best solvent for the application of esterified coproporphyrins on TiO₂ [25]. Probably, namely the acidity of the medium created by acetic acid (pH \sim 3) and (locally) by surface silanol groups, on one hand, is sufficient to hydrolyze ester groups of the dye and, on the other hand, is not so great as to completely protonate carboxylate groups (−COO−) of the dye after cleavage of isopropoxide groups from them. This promotes the binding of dye molecules to the surface of $TiO₂$ nanoparticles (deprotonated ligands form chelate complexes with titanium (IV) more actively than protonated ones). An indirect confirmation of this hypothesis is data on high photocatalytic activity of $TiO₂/SiO₂$ nanocomposites sensitized with zinc carboxyphenylporphyrinates [26].

Adsorption Properties

The sorption capacity of $TiO₂/SiO₂$ (OP) and $TiO₂/SiO₂ (UF) nanocomposites is small: the greatest$ degree of extraction of Be, Ni, and Bi ions ($R \sim 35\%$) and Mg ions $(R \sim 35\%)$ is found on TiO₂/SiO₂ (OP) and $TiO₂/SiO₂$ (UF) sorbents, respectively (Table 6). However, the degree of sorption of namely these ions on $TiO₂/SiO₂$ nanocomposites is greater than that on the samples with η-modification $(R \sim 10\%)$ doped with cations, for which the value of ion extraction of groups V–VI is >90% [27]. Probably, such sorption behavior of nanocomposites can be explained by the acidity of the medium (pH \sim 3–5), generated by locally surface silanol groups, which is greater (pH \sim 1–3) for nanoparticles with titanium dioxide.

Thus, as a result of the performed systematic work on obtaining $TiO₂/SiO₂$ nanocomposites in a variety of ways and radiographic characterization, $TiO₂/SiO₂$ (OP) nanocomposite is distinguished by its properties (PCA in the UV range, adsorption capacity), obtained by application of titanium dioxide to opal (CP) with the structure of η-modification with average crystallite size of $D \sim 50$ Å and spherical particles (associates)

Fig. 16. Photographs of the microstructure of $TiO₂/SiO₂$ composite (OP) (a) and size distribution of $TiO₂/SiO₂$ (OP) composite particles (b).

with a size of 280 nm in the range of 175–450 nm (Fig. 16).

Moreover, PCA of nanocomposites depends on the specific surface of the silica, and the adsorption activity does not depend on its composition, but is determined by it, especially the composition of the surface of the nanocomposite. It is found that the photocatalytic activity in the visible range of the $TiO₂/SiO₂$ nanocomposite (granules) photosensitized with coproporphyrin I (method 1) exceeds more than 20-fold the activity of the commercial reference sample of " $TiSiO₄$."

ACKNOWLEDGMENTS

This work was partially supported by Russian Foundation for Basic Research (project no. 15-03- 01289).

REFERENCES

- 1. G. Erlikh, *Small Objects—Great Ideas. A Broad View of Nanotechnology* (BINOM. Laboratoriya Znanii, 2012) [in Russian].
- 2. M. Dadachov, "Novel titanium dioxide, process of making and method of using same," US Patent No. 0171877 (2006).
- 3. M. Dadachov, "Novel adsorbents and process of making and using same," US Patent No. 0144793 (2006).
- 4. S. Singh, K. C. Barick, and D. Bahadur, "Functional oxide nanomaterials and nanocomposites for the removal of heavy metals and dyes," Nanomater. Nanotechnol. **3**, 20 (2013).
- 5. K. E. Engates and H. J. Shipley, "Adsorption of Pb, Cd, Cu, Zn and nitotitanium dioxide nanoparticles: effect of particle size, solid concentration, and exhaustion," Environ. Sci. Pollut., No. 18, 386–395 (2011).
- 6. P. A. Demina, A. A. Kuz'michev, A. M. Tsybinsky, L. N. Obolenskaya, G. M. Kuz'micheva, E. N. Domoro-

shchina, and E. V. Savinkina, "Synthesis, characterization and adsorption behavior of Mo(VI) and W(VI) ions on titanium dioxide nanoparticles containing anatase modification," Appl. Nanosci. **4**, 979–987 (2014).

- 7. P. A. Demina, A. M. Zybinskii, G. M. Kuz'micheva, L. N. Obolenskaya, E. V. Savinkina, and N. A. Prokudina, "Adsorption ability of samples with nanoscale anatase to extract $Nb(V)$ and $Ta(V)$ ions from aqueous media," Crystallogr. Rep. **59**, 430 (2014).
- 8. G. M. Kuz'micheva, E. N. Domoroshchina, E. V. Savinkina, and L. N. Obolenskaya, "Nanosized titania with anatase structure: synthesis, characterization, applications and environmental effects," in *Titanium Dioxide: Chemical Properties, Applications and Environmental Effects,* Chemical Engineering Methods and Technology (Nova Science, USA, 2014), Chap. 9, pp. 177–227.
- 9. S. Banerjee, S. C. Pillai, P. Falaras, K. E. O'Shea, J. A. Byme, and D. D. Dionysiou, "New insights into the mechanism of visible light photocatalysis," J. Phys. Chem. Lett. **5**, 2543–2514 (2014).
- 10. A. Fujishima, K. Hashimoto, and T. Watanabe, $TiO₂$ *Photocatalysis: Fundamentals and Applications* (BKC, Tokyo, 1999).
- 11. J. G. Yoon, H. K. Oh, and Y. J. Kwag, "Structural and optical properties of $TiO₂-SiO₂$ composite films prepared by aerosol-assisted chemical-vapor deposition, J. Korean Phys. Soc. **33**, 699–704 (1998).
- 12. K. Balachandaran, R. Venckatesh, and R. Sivaraj, "Synthesis of nano-TiO₂-SiO₂ composite using sol-gel method: effect on size, surface morphology and thermal stability," Int. J. Eng. Sci. Technol. **2**, 3695–3700 (2010).
- 13. A. Nilchi, S. Janitabar-Darzi, and S. Rasouli-Garmarodi, "Sol-gel preparation of nanoscale $TiO₂/SiO₂$ composite for eliminating of con red azo dye," Mater. Sci. Appl. **2**, 476–480 (2011).
- 14. K. Balachandran, R. Venckatesh, and R. Sivaraj, "Photocatalytic decomposition of isolan black by $TiO₂–SiO₂$

core shell nanocomposites," Int. J. Res. Eng. Technol. **2** (9), 46–51 (2013).

- 15. D. A. Kumar, J. M. Shyla, and F. P. Xavier, "Synthesis and characterization of $TiO₂/SiO₂$ nanocomposites for solar cell applications," Appl. Nanosci. **2**, 429–436 (2012).
- 16. L. Pinho, F. Elhaddad, D. S. Facio, and M. J. Mosquera, "A novel $TiO₂$ –SiO₂ nanocomposite converts a very friable stone into a self-cleaning building material," Appl. Surf. Sci. **275**, 389–396 (2013).
- 17. L. N. Obolenskaya, G. M. Kuz'micheva, E. V. Savinkina, N. V. Sadovskaya, A. V. Zhilkina, N. A. Prokudina, and V. V. Chernyshev, "Influence of the conditions of the sulfate method on the characteristics of nanosized anatase-type samples," Russ. Chem. Bull. **61**, 2049– 2055 (2012).
- 18. J. Canny, "A computational approach to edge detection," IEEE Trans. Pattern Anal. Machine Intelligence **8**, 679–698 (1986).
- 19. D. S. Tsoukleris, A. I. Kontos, P. Aloupogiannis, and P. Falaras, "Photocatalytic properties of screen-printed titania," Catal. Today **124**, 110–117 (2007).
- 20. P. Falaras, "Synergetic effect of carboxylic acid functional groups and fractal surface characteristics for efficient dye sensitization of titanium oxide," Sol. Energy Mater. Sol. Cells **53**, 163–175 (1998).
- 21. J. Zhao, C. Chen, and W. Ma, "Photocatalytic degradation of organic pollutants under visible light irradiation," Top. Catal. **35**, 269–278 (2005).
- 22. E. V. Savinkina, G. M. Kuzmicheva, and L. N. Obolenskaya, "Synthesis, characterization and photocatalytic properties of η-titania," Int. J. Energy Environ., No. 6, 268 (2012).
- 23. E. V. Savinkina, L. N. Obolenskaya, G. M. Kuz'micheva, A. V. Dorokhov, and A. Yu. Tsivadze, "A new η-titania-based photocatalyst," Dokl. Phys. Chem. **441**, 224 (2011).
- 24. E. V. Savinkina, L. N. Obolenskaya, G. M. Kuz'micheva, and A. V. Dorokhov, "Nanosized η-titania-based photocatalyst," Patent RF No. 24697.
- 25. I. A. Zamilatskov, E. V. Savinkina, A. N. Volov, M. S. Grigor'ev, I. S. Lonin, L. N. Obolenskaya, G. V. Ponomarev, O. I. Koifman, A. S. Kuzovlev, G. M. Kuz'micheva, and A. Yu. Tsivadze, "Synthesis, structures and photosensitizing properties of new porphyrins, Pt(II) and Pd(II)," Macroheterocycles **5**, 308 (2012).
- 26. J. H. Cai, Y. J. Ye, J. W. Huang, H. C. Yu, and L. N. Ji, "Synthesis, characterization and visible-light photocatalytic activity of $TiO₂–SiO₂$ composite modified with zinc porphyrins," J. Sol-Gel Sci. Technol. **62**, 432–440 (2012).
- 27. A. Gainanova, G. Kuz'micheva, E. Kabachkov, and A. Zybinskiy, "Preparation, characterization, photocatalytic and adsorption properties of samples with nano-η-TiO₂," in *Proceedings of the 12th International Conference on Crystal Chemistry of Intermetallic Compounds, Lviv, Ukraine, Sept. 22–26, 2013*, p. 183.

Translated by Sh. Galyaltdinov