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POLYMERIZATION

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Abstract—Linear and branched copolylactides of various structures are synthesized and characterized. Poly styrene suspensions with various particle diameters are obtained via heterophase polymerization with the use of copolylactides as biodegradable surfactants. The influence of the surfactant structure on the polymer microsphere size and the type of size distribution is studied.

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INTRODUCTION

Heterophase polymerization is a highly efficient method of synthesis making it possible under certain conditions to obtain polymers with a narrow diame ter distribution. The production conditions of PS and PMMA dispersions with poorly soluble surfactants (such as diparatolylcarbalkoxyphenylcarbinol, oli goglycolmaleinatebanzoates, monoalkylphthalates, and organosilicon surfactants) as stabilizers were determined in $[1-5]$. Note that poor solubility in water is a key requirement imposed on the surfactants used in heterophase polymerization. In this case, polymer–monomer particles (PMPs) are formed via a single mechanism, namely, from microdrops of the monomer, and have a narrow size distribution. The characteristics of the obtained polymer particles depend on the chemical structures of the used surfac tants [6].

A substantial disadvantage of the industrial pro duction of emulsion polymers and latexes is the high consumption of emulsifiers; therefore, the problem of the wastewater purification has become highly impor tant and urgent. The prospects for application of bio degradable surfactants lie in the fact that they decom pose in the environment into biologically neutral oli gomers and lowcompounds, circumstance that substantially simplifies wastewater purification [7, 8]. Amphiphilic copolymers with high biocompatibility obtained on the basis of aliphatic copolyesters whose hydrophilic blocks are formed by polyethers and whose hydrophobic blocks are formed by polymers of hydroxyacids (glycolic, lactic, hydrox-

ybutyric, etc.) can be ranged among such biodegrad able surfactants. It is known that the hydrolysis of such amphiphilic macromolecules leads to decompo sition into environmentally harmless natural hydroxyacids and biocompatible oligomers [9].

It was important to evaluate the application effi ciency of the amphiphilic hydroxyacid copolymers as surfactants during heterophase polymerization of styrene. For this purpose, a series of new linear and branched copolylactides containing hydrophobic blocks with a molecular mass exceeding that of the hydrophilic blocks were synthesized. In the present study, the results of the heterophase polymerization of styrene with their participation are discussed.

EXPERIMENTAL

*L-*Lactide (99%, Aldrich); polyglycerol with $M_{\rm w}$ = 500 and 2000 (synthesized at Johannes Gutenberg University, Mainz, Germany); monomethylene esters of poly(ethylene glycol) with $M_w = 350, 550,$ and 750 (MPEG, Aldrich); and tin 2-ethylhexanoate $(Sn(Oct)₂, 95\%, Signa)$ were used without further purification. Chloroform, methanol, methyle nechloride (high-purity, Khimmed) were purified via standard techniques [10].

The linear block-copolymers were obtained through the interaction of *L-*lactide and the monom ethylester of poly(ethylene glycol).

$$
n
$$

\n n \n CH_3 \n \uparrow \n O \n CH_3 \n CH_3 \n CH_3 \n CH_3 \n CH_2CH_2O \n m \n CH_3 \n $^{CH-CH-CH-CH_2CH_2O}$ \n m \n CH_3 \n $^{CH-CH-CH-CH_2CH_2O}$ \n m \n CH_3

Branched copolymers were obtained via the interaction of *L-*lactide and polyglycerol.

The copolylactides were synthesized via the follow ing technique. A round-bottomed flask equipped with a magnetic stirrer was loaded with 0.1 mol *L-*lactide; 0.00505 mol catalyst, $Sn(Oct)_{2}$; and 0.001 mol PEG or polyglycerol. The synthesis was performed in melt at 125°С for 10 h; then, the melt was cooled to room temperature and the polymer was dissolved in chloro form and deposited in methanol. The deposited poly mer was filtered and dried in vacuum at 60°С for 48 h.

The IR spectra were registered for the copolymer solutions in $CHCl₃$ on a Specord M 80 IR spectrophotometer.

The NMR spectra were obtained for 10% solutions of copolymers in $CDCl₃$ on a Bruker spectrometer with operating frequencies of 600.22 MHz for ¹H NMR and 150.94 MHz for ¹³C NMR (the internal standard was tetramethylsilane) at the Laboratory of Nuclear Mag netic Resonance, Nesmeyanov Institute of Organoele ment Compounds, Russian Academy of Sciences.

Elemental analysis was performed via a gravimetric method by the Group of Special Organic Analysis at the Nesmeyanov Institute of Organoelement Com pounds, Russian Academy of Sciences.

The molecular-mass characteristics of the copoly mers were analyzed via GPC on a Waters 150 chro matograph equipped with a PL-GEL 5μ MIX C col umn (300 mm \times 7.5 mm) in THF at a flow rate of 1 mL/min at the Petrov Plastics Institute.

The styrene was polymerized under conditions usually used during synthesis of polymer suspensions for immunochemical investigations [1, 2]. The volume ratio of styrene to water was 1 : 9, the surfactant (copolymers 1–6) and initiator concentrations were 1 wt % per monomer, and the temperature was 80 ± 0.5 °C. The styrene polymerization rate was studied via a dilatometric method. The emulsion was formed via rotation of the mixing stirrer in a wide part

Fig. 1. Typical IR absorption spectra of a (co)polylactide.

of the dilatometer at a rate of 600 rpm. Monomer con version *Р* was calculated through the equation

$$
P = (\Delta h / \Delta h_{\text{max}}) \times 100\%
$$

Here, Δh is the running change in the level in the dilatometer capillary (cm) and Δh_{max} is the change in the level in the dilatometer capillary corresponding to 100% conversion:

$$
h_{\text{max}} = V_{\text{m}} \times (\rho_{\text{p}} - \rho_{\text{m}})/(S \times \rho_{\text{p}}),
$$

where $V_{\rm m}$ is the monomer volume (cm³), $\rho_{\rm m}$ and $\rho_{\rm p}$ are the respective densities of the monomer and the polymer $(g/cm³)$, and *S* is the capillary cross-sectional area $(cm^2) [1, 2].$

The polymer-suspension particle size was deter mined via the scanning electron microscopy on a Hitachi S-570 apparatus.

The weight-average and number-average diame ters of particles, D_w and D_n , were determined from the relationships

$$
D_{\rm n} = \Sigma D_i N_i / \Sigma N_i
$$

$$
D_{\rm n} = \Sigma D_i^2 N_i / \Sigma D_i N_i
$$

RESULTS AND DISCUSSION

The copolylactides were obtained as dry powders with a symmetric or anisotropic particle form that are easily soluble in acetone and chloroform. The IR spec tra of the synthesized copolymers show absorption bands at $3000-2800$ cm⁻¹, corresponding to the valence vibrations of the C–H bond of both blocks; in the region of 1768 cm–1, corresponding to the valence vibrations of the carbonyl groups of the ester bond; in the region of $1385-1300$ cm⁻¹, corresponding to the valence vibrations of $-CH_3$ and $-CH_2$ – groups; in the region of $1130-1060$ cm⁻¹, corresponding to the ether bond. The presence of the absorption bands at 3700– 3670 cm^{-1} confirms the presence of hydroxyl groups in copolymers (Fig. 1).

There are proton signals of $-CH$ and $-CH_3$ groups of the polylactide units at 5.04 and 1.45 ppm, respectively, in the ¹H NMR spectra of the synthesized copolymers. In addition, there is a signal corre sized copolymers. In addition, there is a signal corre-
sponding to the protons of $-CH_2-O$ — and $-O$ —
CH₃— groups of MPEG (3.50–3.58 ppm and
3.32 ppm) in the spectra of the linear copolymers
(Fig. 2, spectra 2–4). The sp $CH₃$ – groups of MPEG (3.50–3.58 ppm and 3.32 ppm) in the spectra of the linear copolymers (Fig. 2, spectra *2*–*4*). The spectra of the branched CH₃– groups of MPEG (3.50–3.58 ppm and
3.32 ppm) in the spectra of the linear copolymers
(Fig. 2, spectra 2–4). The spectra of the branched
copolymers have proton signals of $-CH_2-CH_3$
(0.99 num) and CU_L CU_L (1.27 n (0.88 ppm) and $-CH_2-CH_3$ (1.37 ppm) groups characteristic for polyglycerol as well as a wide multiplet signal at $3.00-4.20$ ppm, related to protons of $-CH_2-O$ and $-CH$ groups of polyglycerol (Fig. 2, signal at $3.00-4.20$ ppm, related to protons of $$ spectra *5*–*8*).

The compositions of the synthesized copolymers were determined from the data of NMR spectroscopy and elemental analysis according to the techniques given in [11]. As is seen from Table 1, all compounds were obtained with high yields, and the close values of the theoretical and experimentally determined molec ular masses and elemental compositions testify the correspondence of their structures to the assumed structures.

All dependences of styrene conversion on time, independently of the chemical nature of the copoly lactides, are S-shaped, a circumstance that is charac teristic for heterophase polymerization (Fig. 3). There is an induction period corresponding to the time of PMP formation (-15 min) on the kinetic curves; then,

Fig. 2. ¹H NMR spectra of (2–4) linear and (5–8) branched copolylactides with various contents of polylactide units.

Fig. 3. Dependence of conversion on time during styrene polymerization in the presence of (co)polylactides. Here and therein after the numbering of curves corresponds to the numbering of the samples in Table 1.

the polymerization proceeds at a high constant rate $((4.4 \times 10^{-7})$ – (5.5×10^{-7}) mol/(L s), Table 2) up to complete conversion of the monomer. The time to complete the conversion is 220–270 min.

The polystyrene suspensions synthesized in the presence of polylactide surfactants differ by high sta bility during polymerization, a circumstance that is indicated by the absence of coagulum. The polysty rene suspension particles are spherical and have a narrow size distribution in samples 3 and 5–8 (Fig. 4, Table 2).

The results can be explained in terms of two approaches to the determination of the role of the poorly soluble surfactant used for PMP stabilization.

The first approach $[1-5]$ is based on the formation of the interfacial layer on the particle surface of the surfactant incompatible with the obtained polymer. With an increase in the polymer concentration in the

* Calculated on the basis of monomer loading.

Fig. 4. Micrographs of polystyrene suspension particles and their size-distribution histograms obtained during styrene polymer ization in the presence of (co)polylactides.

PMPs, the surfactant is displaced from the volume to the interface. This results in the formation of particles with the core–shell structure, a circumstance that promotes the aggregative stability of the obtained dis persion. The immobilization of the poorly soluble sur factant at the interface makes it possible to lyophilize the surfaces of PMPs owing to the orientation toward the aqueous phase of the hydrophilic fragments of the surfactant molecules and creates the conditions for the Rehbinder structural–mechanical barrier.

The second approach is based on the geometrical construction of the surfactant molecules at the inter-

face promoting the minimum gradient of polarity dur ing the transition from the aqueous phase to the hydrocarbon. In this case, the results may be explained in terms of the concept of the critical parameters of packing. For example, the idealized diblock–copoly mer molecule can be represented as two contacting spheres formed by corresponding blocks; in this case, the sizes of these spheres are proportional to the molecular masses of the copolymer blocks. This mac romolecule can be characterized with the use of such parameters as volume *v*, length *l*, and block cross-sec tional area S_0 (Fig. 5a). The critical parameter of pack-

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Fig. 5. Schematic representation of diblock-copolylactides and main parameters describing their sizes according to theory of crit ical parameters of macromolecule packing.

ing is determined as the ratio of macromolecule vol ume *v* to the product of length *l* and hydrophilic-block cross-sectional area S₀.

For the macromolecules with similar sizes of the hydrophilic and hydrophobic blocks, such as those given in Fig. 5a, the critical parameter of packing is \approx 1. It is known [12, 13] that the aggregation in water of the macromolecules with a critical parameter of packing close to 1 leads to the formation of the lamellar phase. If the size of the hydrophilic block of the macromolecules exceeds the size of the hydro phobic block (Fig. 5b), the critical parameter of packing is less than 1, and an oil-in-water emulsion forms and stabilizes. If the hydrophilic block of the macromolecule is smaller than the hydrophobic block (Fig. 5c), the critical parameter of packing is more than 1 and a water-in-oil emulsion forms and stabilizes. However, in the latter case, dynamically stable oil-in-water emulsions, whose stability is pro vided by the mixing of the system, is possible.

Exactly copolylactides 2–4 can be among such macromolecules with a critical parameter of packing above 1. It was considered [12] that the change in the block ratio of such polymers affects the interface-sur face curvature in water-in-oil emulsions. At the same time, the change in the block ratio in copolymers 2– 4 may insignificantly affect the interface-surface cur-

vature in oil-in-water emulsions and, respectively, the PMP sizes during heterophase polymerization.

Such dynamically stable emulsions can be dis persed via two mechanisms. First, this is possible owing to the shear-stress gradient created by the mixing rate and the construction of the stirrer arms. In this case, drops of a large size of about $1-2 \mu m$ are formed. Second, the additional dispersion of the emulsion microdrops is possible because of the for mation of a microemuslion layer on their surfaces.

Table 2. Characteristics of polymer suspensions obtained with the use of (co) polylactides $1-8$ as surfactants

Polymer	$V_{\text{p}} \times 10^7$, $\frac{1}{2}$ mol/L s	d , μ m	$D_{\rm w}/D_{\rm n}$	$M_{\rm n} \times 10^{-3}$
1	3.5	0.53	1.03	530
2	4.4	1.09	1.50	595
3	4.5	0.64	1.04	310
$\overline{4}$	4.7	0.62	1.11	125
5	4.8	0.82	1.02	295
6	5.1	0.80	1.05	410
7	4.8	0.55	1.01	490
8	5.5	0.63	1.02	260

Fig. 6. Schematic representation of branched copolylactides: (a) polymer 5, (b) polymer 7, (c) polymer 8.

The microemulsification on the monomer micro drop surface is observed as a result of the mass trans fer of the components through the interface. Then, a finely dispersed emulsion of the monomer is formed. The presence of the monomer microdrops of various sizes in the initial emulsion is a cause of the forma tion of polymer suspensions with a bimodal size dis tribution (Fig. 4).

It is characteristic that, for the branched copolylac tides, the suspension-particle size depends on the molecular mass of the hydrophilic block, which, in this case, is polyglycerol: The increase in its fraction leads to a decrease in the suspension-particle size. In this case, the mass ratio of the hydrophobic and hydro philic blocks is comparable with the values obtained for linear copolylactides (Table 1).

The structure of the branched copolylactides can be presented as several spherical hydrophobic blocks connected by the same sphere of the hydrophobic block (Fig. 6). Comparatively slight changes in the hydrophobic-block volumes (in particular, of poly mers 6 and 7) cause substantial changes in surface cur vature. As a result, even a comparatively slight change in the molecular mass of the hydrophobic blocks leads to a noticeable decrease in suspension-particle size, a result that is explained by the monomer dispersion via one mechanism.

Polymers 7 and 8 have close molecular masses of the hydrophobic blocks, but the molecular masses of the hydrophilic blocks differ. Nevertheless, the particle sizes of the polystyrene suspension obtained in the presence of these polymers are close (Table 2). With an increase in the molecular mass of the polyglycerol block from 500 to 2000, its func tionality rises noticeably (from 8 hydroxyl groups to 28). In the amphiphilic block-copolymers, these hydroxyl groups are replaced with short polylactide blocks. For example, the molecular mass of the hydrophilic block of copolylactide 8 increases 4 times in comparison to that of polymer 7, and the

amount of hydrophobic blocks increases 3.5 times. As a result of such multidirectional effects, the par ticle size of the polystyrene latex of polymer 8 is slightly higher than that of polymer 7.

CONCLUSIONS

The conditions of the heterophase polymerization determined in the presence of the amphiphilic copol ymers of oxyacids with poly(ethylene glycol) or polyg lycerol have made it possible to obtain polystyrene microspheres with an average diameter from 0.55 to 0.8 µm that have a narrow size distribution. The inves tigated surfactants differ by high biodegradability, a phenomenon that is necessary for the development of green-chemistry principles in the field of synthesis of polymer dispersions of different uses.

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REFERENCES

- 1. I. G. Krasheninnikova, Extended Abstract of Candi date's Dissertation in Chemistry (RKhTU, Moscow, 2007).
- 2. I. A. Gritskova, V. S. Papkov, I. G. Krasheninnikova, and A. M. Evtushenko, Polym. Sci., Ser. A **49** (3), 235 (2007).
- 3. O. V. Chirikova, Extended Abstract of Candidate's Dis sertation in Chemistry (MITKhT, Moscow, 1994).
- 4. D. I. Shragin, I. A. Gritskova, V. V. Kopylov, E. V. Milushkova, L. A. Zlydneva, and S. M. Levachev, Silicon **7**, 217 (2015).
- 5. L. A. Zlydneva, Extended Abstract of Candidate's Dis sertation in Chemistry (MITKhT, Moscow, 2013).

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- 6. C. Chorng-Shyan, *Principles and Applications of Emul sion Polymerization* (Wiley, New York, 2008).
- 7. A. M. Fernandez, U. Held, A. Willing, and W. H. Breuer, Prog. Org. Coat. **53** (4), 246 (2005).
- 8. M. J. Scott and M. N. Jones, Biochim. Biophys. Acta **1508**, 235 (2000).
- 9. V. A. Vasnev, Polym. Sci., Ser. B **39** (11–12), 474 (1997).
- 10. W. L. E. Armarego and D. D. Perrin, *Purification of Laboratory Chemicals* (Buttleworth-Heinemann, Oxford, 1998).
- 11. V. V. Istratov, V. T. Tarasyuk, V. A. Vasnev, and N. A. Borisova, Polym. Sci., Ser. B **55** (3–4), 218 (2013).
- 12. K. Holmberg, B. Jönsson, B. Kronberg, and B. Lind man, *Surfactants and Polymers in Aqueous Solution* (Wiley, New York, 2002).
- 13. S. K. Mehta and Kaur. Gurpreet, "Microemulsions: Thermodynamic and Dynamic Properties," in *Ther modynamics*, Ed. by M. Tadashi (InTech, 2011).

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