

Radiation Shielding Properties for NaO–CdO–Bi₂O₃–B₂O₃ Glasses Using XCOM, Phy-X/PSD and Srim Programs

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Received December 16, 2021; revised February 1, 2022; accepted February 8, 2022

Abstract—Achieving an excellent and efficient radiation shielding for various applications is impervious. A series of glass samples $20\text{Na}_2\text{O}\cdot 15\text{CdO}\cdot x\text{Bi}_2\text{O}_3\cdot (65-x)\text{B}_2\text{O}_3$ where ($x = 0, 10, 20$ and 30) were fabricated by the traditional melt-quench method. The structural, chemical stability and significant mechanical properties were explored. The obtained results revealed that the addition of Bi₂O₃ to the glass context enhances the density and reduce values of Poisson ratio (σ). Comprehensive shielding properties for gamma, neutron, proton and alpha ray were estimated by using XCOM, Phy-X/PSD and SRIM programs. MCNP (version 5) has been used to simulate the mass attenuation coefficient (μ/ρ) at different energies (0.1, 0.5, 1, 3, 5 and 15 MeV). A consistent behavior reported with XCOM and Phy-X. The addition of Bi₂O₃ showed significant enhancement in μ/ρ values. In contrast, the addition of Bi₂O₃ in the cost of B₂O₃ leads to a gradual reduction in the values of the projected range and Mass Stopping Power (MSP) for alpha and proton, and the same behavior can be noted for the effective removal cross-section of fast neutron (Σ_R) values. Finally, the new glass composition can be valuable for ionizing radiation shielding applications.

Keywords: bismuth borate glass, SRIM, radiation shielding

DOI: 10.1134/S1087659621070026

INTRODUCTION

Understandable and fast developments in technological and communications sectors in recent years, which led human beings to expose to various types of ionizing radiation such as gamma, neutron radiation and X-ray used in industries, medical and research for various purposes. The radiation shielding is one of the protective measures to reduce hazards of radiation exposure through avoiding the direct interaction with biological tissues. The hazard of radiation progressively increase by increasing radiation penetration power. The degree of penetration depends mainly on the type and energy of radiation involved, thus the studies made in different type of materials for shielding purposes. Radiation shielding materials for radiation attenuation should have high density, good radiation attenuation, high stability and low toxicity in manufacture process. The efficiency of shielding materials with better radiation shielding resistivity is depends on few factors such as cost, quality, thermal and optical properties, mechanical stability, toxicity and efficacy of attenuation [1–7].

For long time, lead is used as the best material for radiation shielding because of its high atomic number, economical evaluable, available in different form and cheap. However, lead is heavy in weight, easy to crack

and somewhere harmful to human health and the environment. Also, concrete shielding materials are becoming more widespread at the different fields of high-energy and nuclear radiations protection. Nevertheless, concretes are opaque and blocking visible light. Therefore, major challenges occur since it is hard to look through the concretes based shield during the control mechanism [8–10].

The glass is highly flexible for chemical and structural changes by adding modifiers to meet specific applications requirements. However, a typical glass can be defined as transparent, non-toxic and useable with advanced material characteristics for different aims. Hence, researchers desired use glasses as shielding materials in recent applications as they have good capability physically and chemically. Glass has good light transparency, ease of manufacture, non-toxic and high density. Furthermore, glasses can be prepared in the desired shape with high space homogeneity and low costing [11–14].

From the several glass formers, boron oxides (B₂O₃) are the most efficient and important commercial glass. These oxides have attractive properties such as high transmittance, low melting point, thermal and mechanical stability. Bi₂O₃ is highly unstable due to presence of ion pairs, which make these glasses to

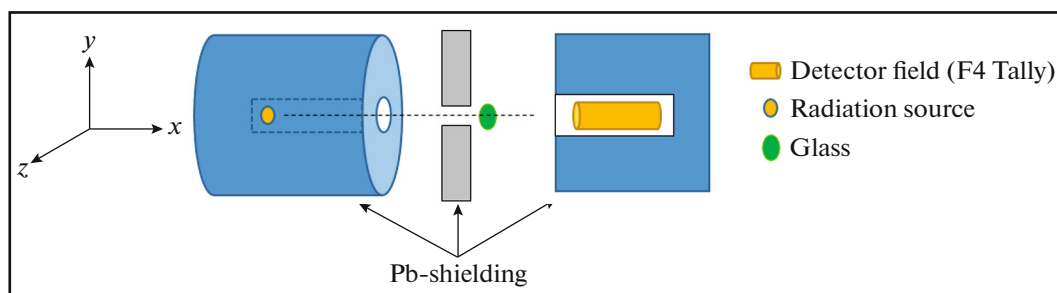


Fig. 1. Total simulation geometry using MCNP5.

exhibit non-linear optical effects. High third order non-linear optical susceptibility and long infrared cutoff wavelengths make them useful for ultra-fast optical switches and optoelectronic devices. Previous studies proved that the increment of Bi₂O₃ content increase the number of non-bridging oxygen decreased melting temperature to 980°C and decrease coordination number of boron [15–21]. The current study focuses mainly on providing new glasses for gamma ray and neutron shielding applications. It aims to explore the effect of increasing various amounts of Bi₂O₃ in altering structural and shielding properties of borate glass network.

EXPERIMENTAL

Glass Preparation

The current raw materials (Na₂O, CdO, Bi₂O₃ and B₂O₃) are obtained from Alfa Aesar with high purity (99.5% and above). The melting-quenching technique was performed to fabricate the new glasses. Various melting temperatures are performed depend on the Bi₂O₃ concentration (800 to 900°C). finally, the samples shifted to the annealing furnace (temperature 400°C) for 3 h then brought down the temperature to the room temperature during 24 h. The glass systems were prepared based on the stoichiometry:

- S1: 20Na₂O, 15CdO and 65B₂O₃ (mol %)
- S2: 20Na₂O, 15CdO, 10Bi₂O₃ and 55B₂O₃ (mol %)
- S3: 20Na₂O, 15CdO, 20Bi₂O₃ and 45B₂O₃ (mol %)
- S4: 20Na₂O, 15CdO, 30Bi₂O₃ and 35B₂O₃ (mol %).

Shielding Properties

In this study, the mass attenuation coefficients of the new samples were estimated by using different methods MCNP5, Phy-X and XCOM programs. A series of gamma shielding properties such as buildup factor (exposure and absorption), electron density (electron/m³), specific gamma ray constant factor (Rm²/Ci.h) and gamma dose rate (R/h) were determined by using Phy-X [22].

The shielding parameters of neutron beam were also identified and compared with the available shield-

ing materials by determining the fast removal cross-section (cm⁻¹), and compared with that measured by Phy-X.

In addition, the mass stopping power and projected range of proton and alpha particles were estimated by using SRIM code [23, 24]. This code is a Monte Carlo simulation base that used to estimate the stopping power of ions in materials. The interaction of heavy charged particles (proton/alpha) in a specific material determined based on the Beth-Bloch theorem [23].

MCNP5 Simulation and Interaction Code

Concerning the simulation part, the mass attenuation coefficient of the new samples were estimated by using MCNP5 [25–27]. Figure 1 exhibits the simulation conducted to estimate attenuation coefficients of the prepared glasses. In the x-axis direction, a simple sphere of glass was exposed with a mono-energetic beam emission, which supposed as a conventional point source. Mesh tally type F4 was activated to estimate the sum of all contributions in the proposed detecting area (NaI-detector). To eliminate effect of background radiation, all irradiation process starting from the source to the detector guided and collimated with Pb-shielding. The simulation was conducted by using the weight fractions obtained from the glass compositions. All required data card and commands such as energy (ERG), type of particle (PAR), position (POS) and beam direction (DIR) were defined accordingly. To reduce errors and variance of variance, the simulation was simplified, repeated three times for each glass sample, termination steps applied after high number of histories (NPS > 10⁸) and CUT-OFF option was used to exclude low radiation exposure (<1 keV).

RESULTS AND DISCUSSION

Amorphous Phase and Physical Measurements

Figure 2 shows the X-ray powder diffraction (XRD) of the prepared samples. No sharp diffraction peak reported between 20° and 80° theta angle in all tested samples. Instead, broad peaks at 28° and 47° were clearly shown with the new fabricated samples.

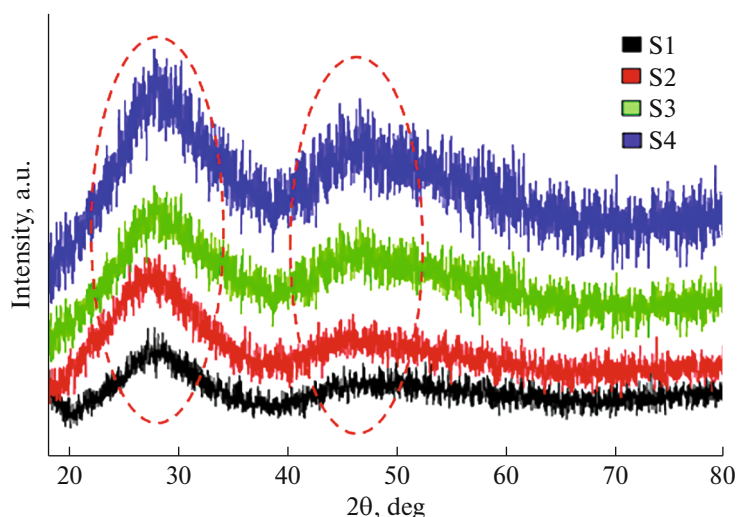


Fig. 2. XRD spectra of the prepared samples.

The spectra obtained from the measurements confirm the amorphous nature of the samples.

Glass Durability

The dissolution rate (Dr) of the fabricated glasses was estimated by using the formulae (1) [28]:

$$Dr = \Delta w / At,$$

where, Δw is the weight loss during the immersion, A is the surface area of the fabricated glass and t is the time period of dissolution. The behavior in Fig. 3 exhibited the increasing of durability with increasing of immersion time and decreased with increasing of Bi_2O_3 content. The increasing of Dr attributed to the hydration process resulted from the ion exchange in the glass network [29]. This hydration reduced noticeably by the increment of Bi_2O_3 , which indicates that ability of bismuth oxide to improve water-resistant of the prepared glasses.

Shielding Properties

Table 1 shows density data, molar volume, packing density and other properties of the studied glass systems.

Table 1. The density (ρ) (g/cm^3), molar volume (V_m), Packing density (V_t), Poisson's ratio (σ), number of bonds per unit volume (N_B), metallization (M), average boron–boron separation $\times 10^{-23}$ ($\langle d_{B-B} \rangle$) (Nm), molar polarizability $\times 10^{-24}$ (α_m) (cm^3), refractive index (n) and dielectric constant (ϵ)

Code	ρ	V_m	V_t	σ	N_B	$\langle d_{B-B} \rangle$	α_m	n	ϵ	M
S1	4.244	18.122	0.909	0.347	9.969	1.433	4.058	2.211	4.888	0.435
S2	4.987	23.367	0.728	0.309	9.019	1.437	0.219	2.317	5.368	0.407
S3	5.732	27.246	0.644	0.284	8.178	1.371	0.264	2.388	5.702	0.389
S4	6.624	29.561	0.611	0.273	7.945	1.259	0.288	2.402	5.769	0.386

In the current study, we calculated the mass attenuation coefficients (μ/ρ) by using MCNP5 simulation code, Phy-X and XCOM software. Table 2 illustrates the calculated μ/ρ values of the new fabricated samples were estimated at six different energies (0.15, 0.5, 1, 3, 5 and 15 MeV). A good agreement between the performed methods as exhibited in Fig. 4. The new fabricated glasses showed decreasing in the calculated μ/ρ with increasing the energy of incident photon. Basically, the photoelectric effect is dominant at the low energy level (up to 0.1 MeV) due to the high effect of atomic number at this level (Z^{4-5}) [30, 31]. Thus, the μ/ρ values of the fabricated glass take the largest values in low energies then gradually decreased (Compton effect dominant) and finally, be almost constant at the level of 3–15 MeV due to the dominance of pair production (Z^2) [32, 33].

The buildup factor obtained from the contribution of scattering photon through exposure and absorption interaction. In the current study, we use the G–P fitting parameters to estimate both of exposure buildup factor (EBF) and energy absorption buildup factor (EABF) at different depths (1, 10, 20, 30 and 40 mfp).

The EBF for the S3 was estimated for an incident photon in the energy levels of 0.01 to 15 MeV as shown in Figs. 5a, 5b. The EBF and EABF showed sharp peak at 0.9 MeV attributed to the K -absorption edge of

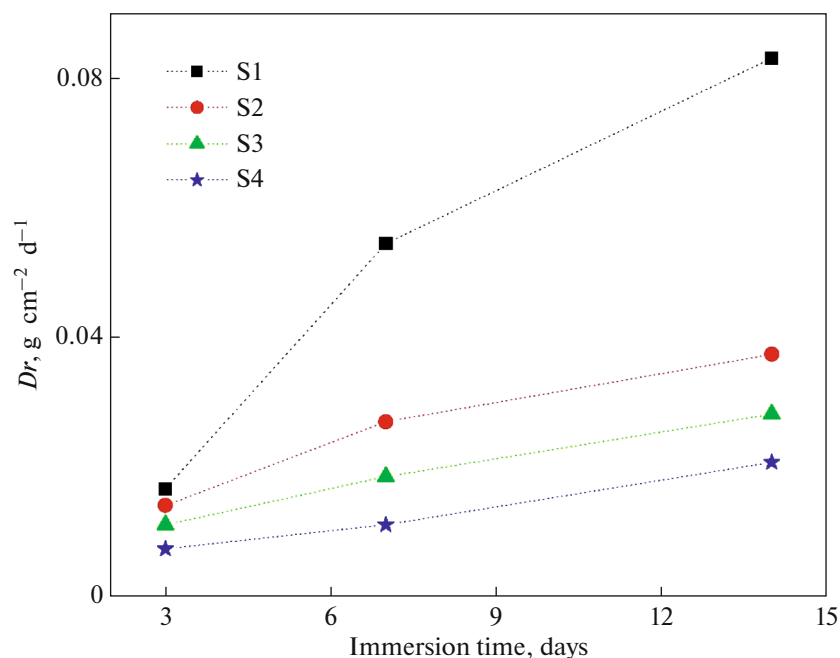


Fig. 3. Dissolution rates of the prepared glass for 14 days.

Bi [29]. Another small peak revealed at low energy level (0.3 MeV), which attributed to the *K*-shell absorption of Cd [34, 35]. The current spectra revealed that the values of exposure and absorption buildup factors are small at low levels (up to 0.1 MeV), then increased at the intermediate energy levels (0.1 up

to 1 MeV) and finally decreased at high energy levels. The heightening of buildup factors at the intermediate level is attributed to the high likelihood of Compton scattering. Eventually the values decreased in high energy level due to pair production. Then start to increase again due to pair production process varies as $\sim Z^2$. In addi-

Table 2. Mass attenuation coefficients at specific energies (Phy-X, MCNP5 and XCOM)

	S1					S2				
	XCOM	Phy-X	Δa	MCNP5	Δb	XCOM	Phy-X	Δa	MCNP5	Δb
0.15	0.199	0.206	0.034	0.211	0.058	0.833	0.894	0.071	0.901	0.078
0.50	0.086	0.087	0.011	0.093	0.078	0.093	0.088	0.055	0.103	0.102
1.00	0.061	0.061	0.00	0.072	0.165	0.063	0.065	0.031	0.072	0.133
3.00	0.035	0.035	0.00	0.040	0.133	0.036	0.038	0.054	0.044	0.200
5.00	0.028	0.029	0.035	0.032	0.133	0.030	0.034	0.125	0.036	0.182
15.0	0.021	0.031	0.385	0.030	0.353	0.025	0.032	0.246	0.033	0.276
	S3					S4				
	XCOM	Phy-X	Δa	MCNP5	Δb	XCOM	Phy-X	Δa	MCNP5	Δb
0.15	1.335	1.323	0.001	1.348	0.010	1.836	1.818	0.010	1.822	0.010
0.50	0.100	0.110	0.095	0.116	0.148	0.108	0.115	0.063	0.122	0.121
1.00	0.063	0.067	0.061	0.071	0.119	0.064	0.069	0.075	0.088	0.315
3.00	0.037	0.039	0.052	0.043	0.150	0.037	0.040	0.078	0.042	0.126
5.00	0.031	0.037	0.176	0.039	0.228	0.033	0.038	0.141	0.041	0.216
15.0	0.028	0.034	0.193	0.039	0.328	0.032	0.040	0.222	0.047	0.379

$$\Delta a = [(XCOM - Phy-X) / ((XCOM + Phy-X) / 2)] \times 100.$$

$$\Delta b = [(XCOM - MCNP5) / ((XCOM + MCNP5) / 2)] \times 100.$$

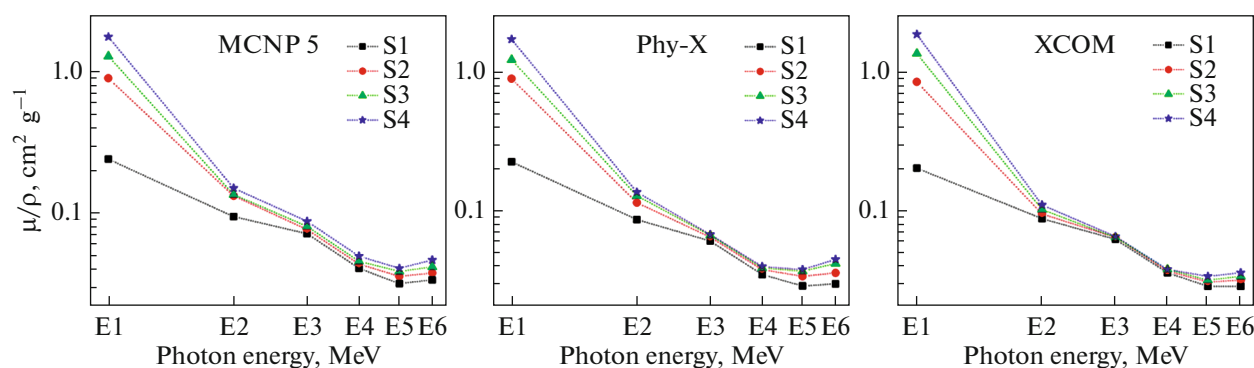


Fig. 4. The mass attenuation coefficients of the fabricated glasses using MCNP5, Phy-X and XCOM.

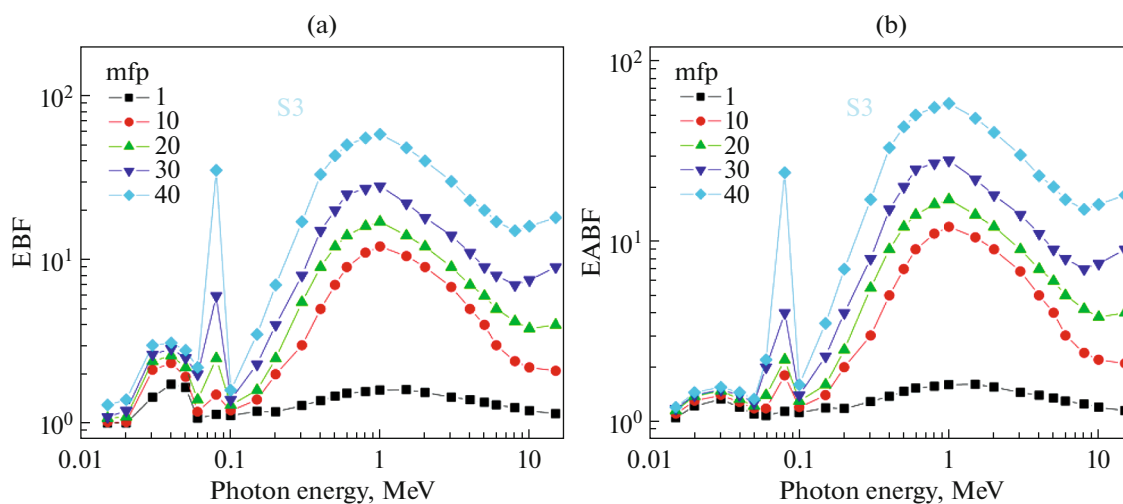


Fig. 5. The effect of energy on the (a) EBF and (b) EABF values for S3.

tion, the values of EBF and EABF clearly increased with increasing of penetration depths from one to 40 mfp.

Figure 6 shows the variation of the estimated electron densities for the prepared glasses at different energies (0.015, 0.05, 0.1, 0.5, 1, 5 and 15 MeV). The trend observe in the figure proved the depending of electron density on the incident of photon energy. For all glasses, the electron densities showed high values at low energy region (0.015, 0.05 and 0.1 MeV), and decreased at high energy region (0.5, 1, 5 and 15 MeV) [3]. In addition, it obvious the increasing of electron densities with the gradual increasing of Bi concentration.

The specific gamma-ray constant (aka gamma factor) is defined as the dose rate absorbed at a given material from a specific distance, and this factor used for radiation protection purposes to determine the dose collected at thickness or penetration depth in a specific material. Figure 7 illustrates the effect of different energy levels on the gamma factor for the new fabricated glasses. The values of specific gamma ray

constant increase with increasing of incident photon energy [3, 34].

Figure 8 shows the dose rate variations (R/h) obtained from gamma ray energy with different thickness in a specific medium. The fabricated glasses were used as target or absorber for the incident energy. The dose rate variation was estimated at different thickness (0.1, 0.5, 1 and 1.5 cm). Furthermore, for all glasses and at different thickness the dose rate reduced at low energy level (0.015 up to 0.03 MeV), and then starting to increase from 0.04 up to 15 MeV.

It has already proved that neutron reactions increase with light element and decreases with increasing of neutron energy. In addition, the likelihood of neutron scattering increases gradually with the increase of collision occurred in the glass context. Consequently, it is important to examine the ability of neutron-energy retention through new shielding materials. For the current glasses, the effective removal cross-section for fast neutron (FRCS: Σ_R) was estimated mathematically and by using Ph-X program as shown in Table 3. Despite the reduction of neutron

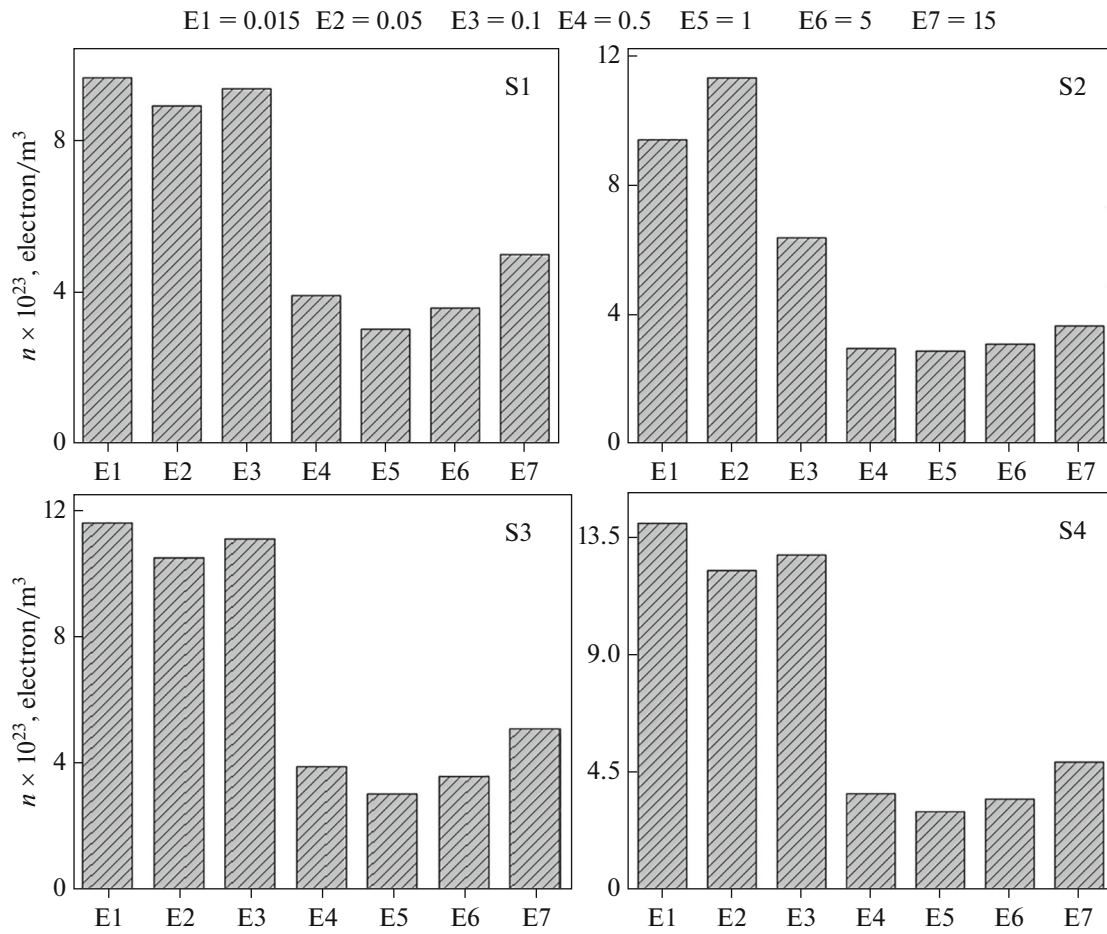


Fig. 6. The effective electron densities of the fabricated glasses at different energies.

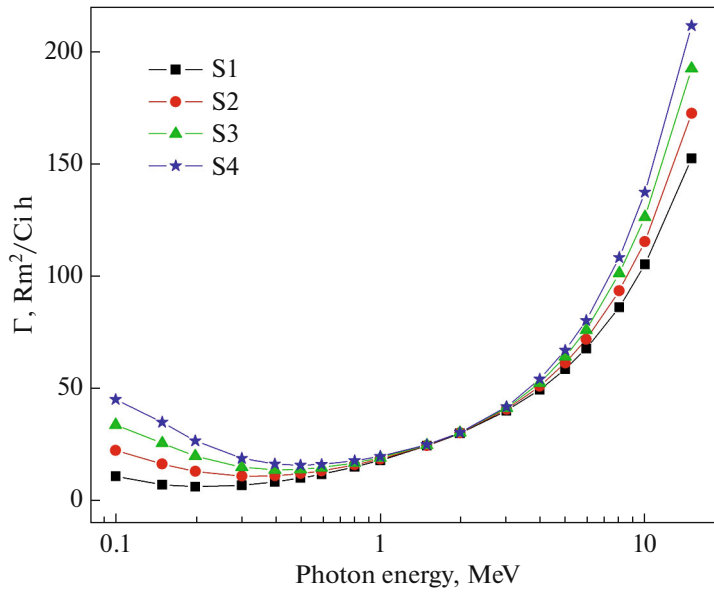


Fig. 7. The specific gamma ray constants for the fabricated glasses at different energy levels.

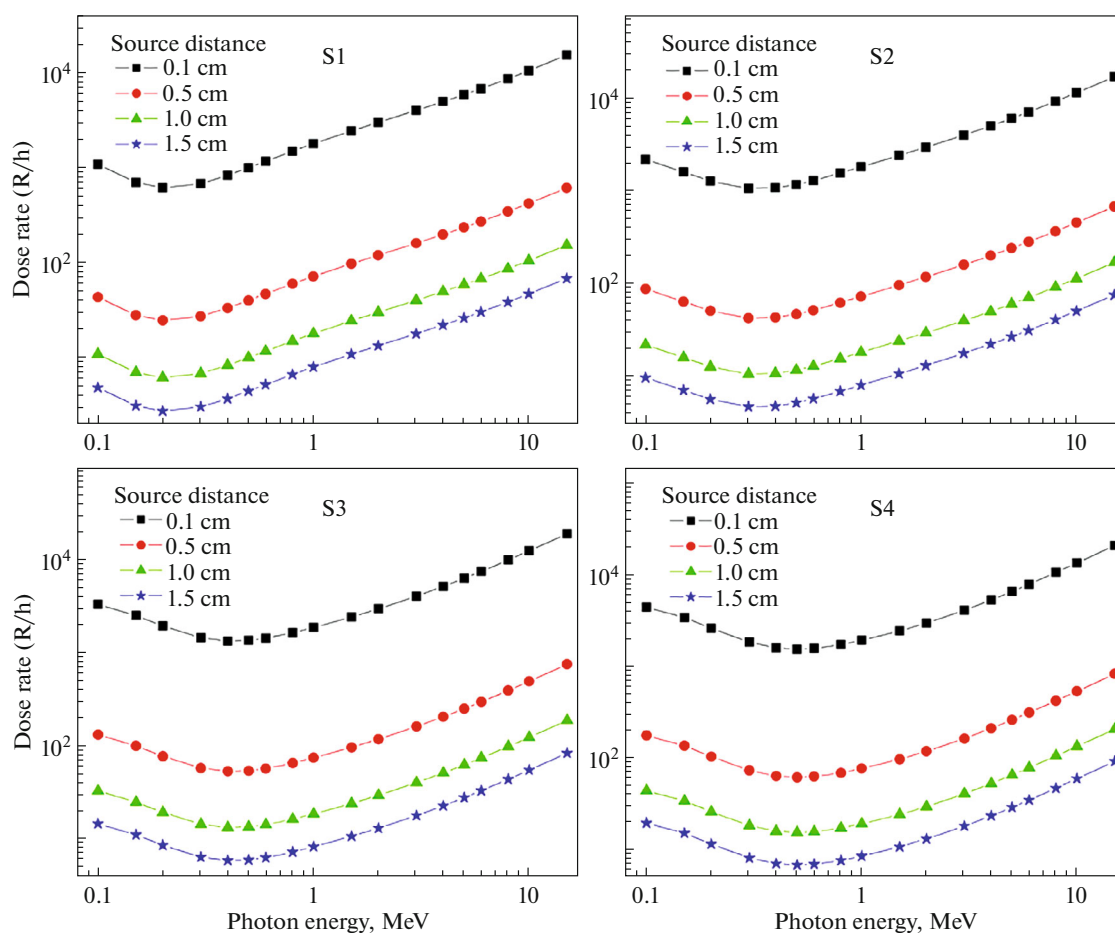


Fig. 8. The dose rate of gamma ray at different energies levels for the fabricated glasses.

Table 3. The removal cross-sections and mass removal cross-section for the fabricated glasses

Sample code	Element	Weight fraction	Partial density P , g cm^{-3}	Mass cross section Σ_R , $\text{cm}^2 \text{g}^{-1}$	Removal cross section Σ_R , cm^{-1}	
					manual calculation	using Phys-X
S1	B	0.1627	0.6906	0.0771	0.164	0.159
	O	0.4185	1.7759	0.0405		
	Na	0.1796	0.7621	0.0341		
	Cd	0.2392	1.0153	0.013		
S2	B	0.1010	0.5039	0.0771	0.141	0.137
	O	0.3057	1.5248	0.0405		
	Na	0.0719	0.3586	0.0341		
	Cd	0.1327	0.6617	0.013		
	Bi	0.3886	1.9381	0.0103		
S3	B	0.0621	0.3559	0.0771	0.132	0.129
	O	0.2316	1.3276	0.0405		
	Na	0.0569	0.3260	0.0341		
	Cd	0.1060	0.6074	0.013		
	Bi	0.5434	3.1150	0.0103		
S4	B	0.0386	0.2560	0.0771	0.131	0.128
	O	0.1879	1.2448	0.0405		
	Na	0.0470	0.3111	0.0341		
	Cd	0.0861	0.5704	0.013		
	Bi	0.6403	4.2417	0.0103		

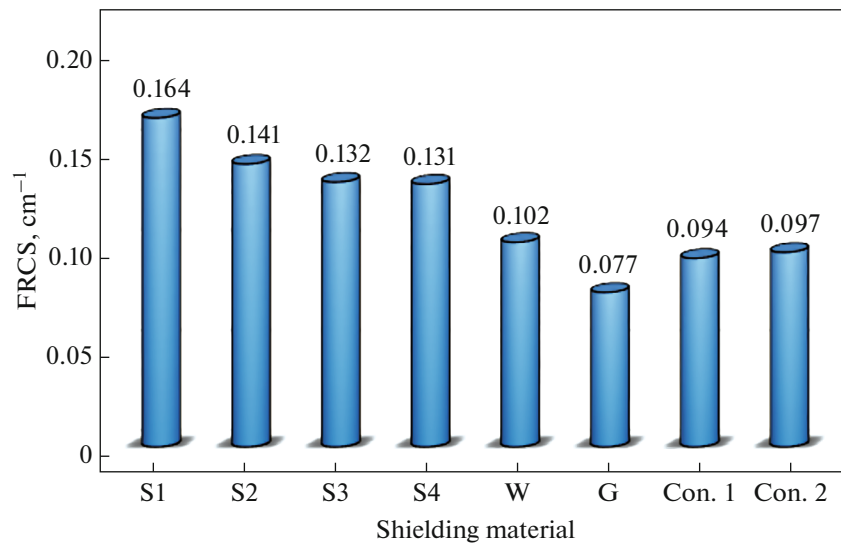


Fig. 9. The effective removal cross-section values of the fabricated glasses compared with other standard materials [W: water; G: graphite; Con. 1: ordinary concrete; Con. 2: serpentine concrete].

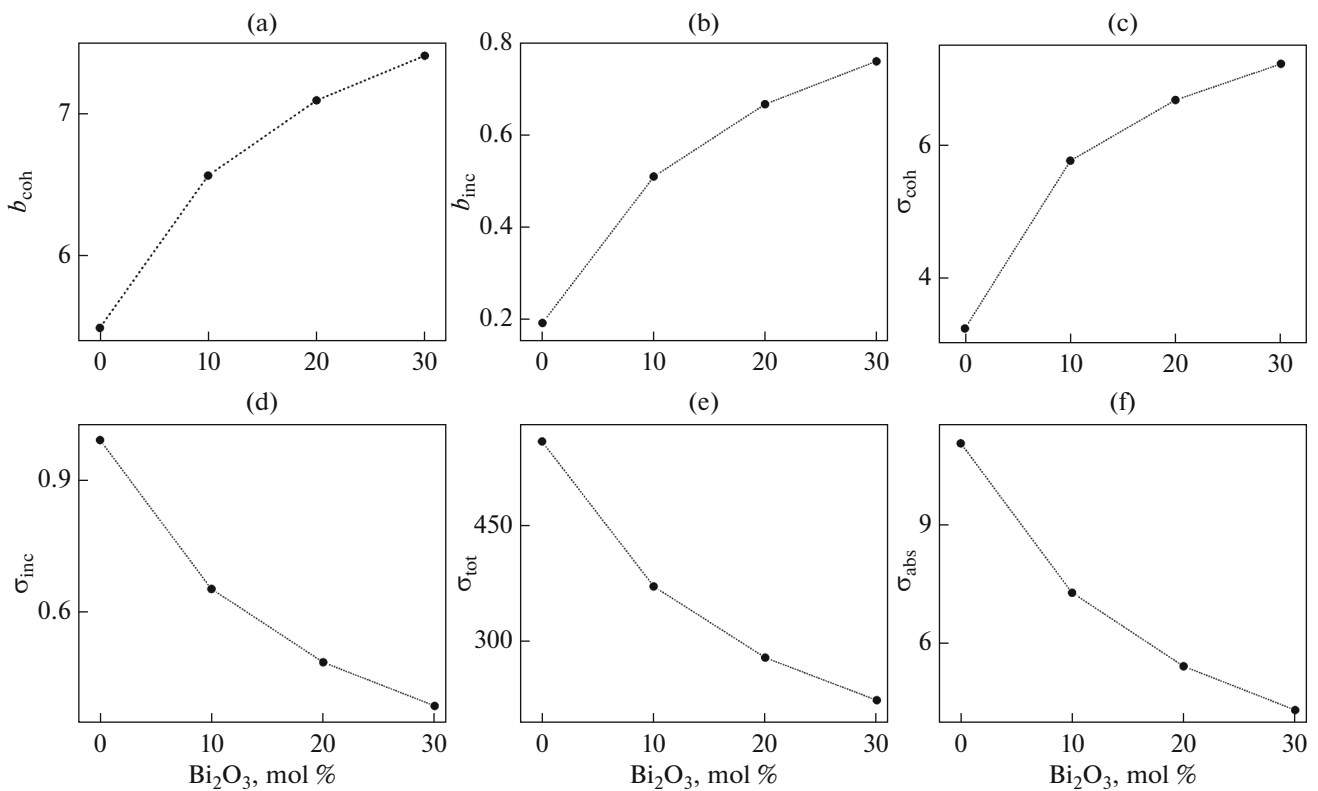


Fig. 10. The neutron scattering and absorption parameters for new prepared samples; (a) b_{coh} : the coherent neutron scattering length, (b) b_{inc} : the incoherent neutron scattering length, (c) σ_{coh} : the coherent neutron scattering cross section, (d) σ_{inc} : the incoherent neutron scattering cross section, (e) σ_{tot} : the total neutron scattering cross section, (f) σ_{abs} : the absorption neutron scattering cross section).

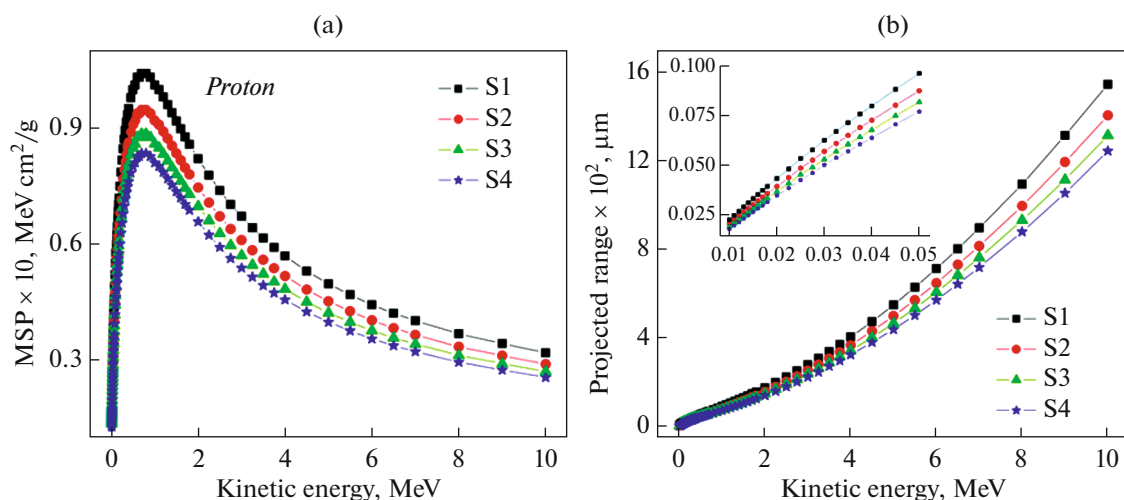


Fig. 11. Proton shielding properties for the prepared glasses as a function of kinetic energy for: a) mass stopping power variations; b) projected range variations.

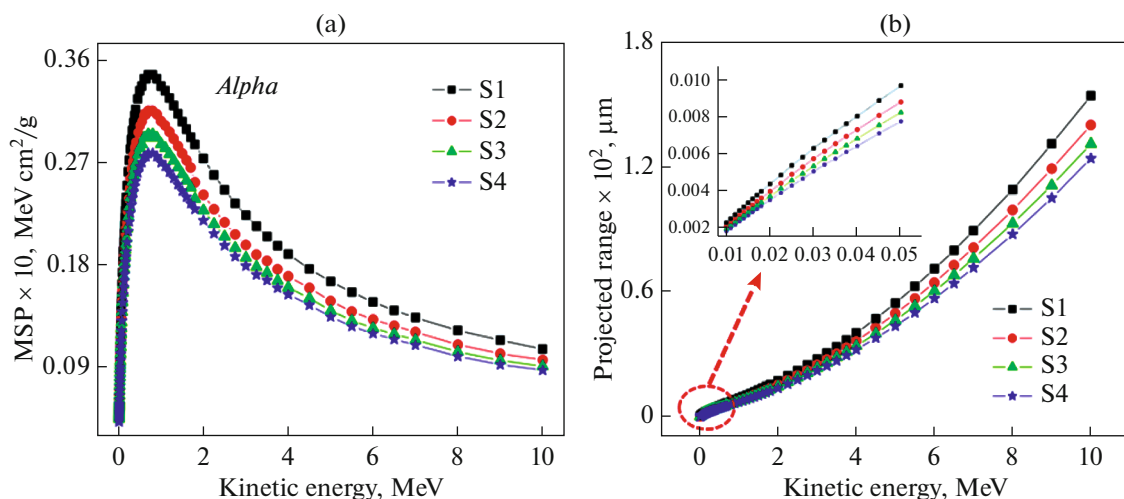


Fig. 12. Alpha shielding properties for the prepared glasses as a function of kinetic energy for: (a) mass stopping power variations; (b) projected range variations.

attenuation values reported with increasing Bi content, the Σ_R of all prepared glasses is better than some standard materials such as water (0.102 cm^{-1}), graphite (0.077 cm^{-1}) and concrete (ordinary = 0.094 cm^{-1} and serpentine = 0.097 cm^{-1}) as shown in Fig. 9 [36, 37].

Moreover, the neutron shielding parameters (NSP) for the new fabricated glasses were calculated concerning to the estimated mass fraction, neutron scattering lengths and cross sections of the elements of each single element in the glass composition [38]. Six neutron scattering parameters (coherent and incoherent neutron scattering lengths, coherent and incoherent neutron scattering cross-sections, total and absorption neutron scattering cross-sections). Figure 10 revealed the growth of coherent/incoherent neutron scattering

lengths and coherent neutron scattering cross-sections with increasing of Bi content. Conversely, the incoherent neutron scattering cross-sections, total and absorption neutron scattering cross-sections declined with increasing of Bi. The NSP attained by G4 are ($b_{\text{coh}} = 7.40 \text{ fm}$; $b_{\text{inc}} = 0.758 \text{ fm}$; $\sigma_{\text{coh}} = 7.207 \text{ b}$ and $\sigma_{\text{abs}} = 4.352 \text{ b}$).

Figures 11 and 12 show the values of proton and alpha MSP of the prepared glasses. As shown in the figures, both values of proton and alpha increased with increasing of kinetic energy (up 1 MeV). In the energy range of 1 to 2 MeV, the MSP values radically decreased and the reduction continue in the higher energies ($>2 \text{ MeV}$) but with low reduction rate. The sample with the highest Bi content (S4) revealed the lowest values of proton and alpha MSP.

The projected range is a concept that used to define the expected depth to which alpha or proton will penetrate up to rest in an absorber material [23]. In the current study, the projected ranges of proton and alpha were also determined by using SRIM code. As shown in Figs. 11 and 12, the prepared samples, especially S4, attained low projected range values.

CONCLUSION

Via the traditional melt-quench technique, four samples of 20Na₂O·15CdO·xBi₂O₃·(65 – x)B₂O₃ (0.00 ≤ x ≤ 0.20) were fabricated to assess their shielding features for different radiation types (photons, neutrons, alpha, and protons). The XRD instrument asserted the amorphous nature of the prepared samples. The replacement of Bi₂O₃ instead of B₂O₃ enhanced the density, durability, and μ/ρ values for the current samples, whereas the values of Σ_R, MSP and projected range for the alpha and protons display slightly decrement with addition Bi to the glass structure. Based on the obtained results, the current glass samples appear excellent shielding features for electromagnetic and particulate radiation nominate to utilize as structural shielding in different radiation facilities.

CONFLICT OF INTEREST

The authors declare that they have no conflict of interest.

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