

## New Formula Type of Phosphates with Langbeinite Mineral Structure

V. I. Pet'kov<sup>a,\*</sup>, A. A. Alekseev<sup>a</sup>, E. A. Asabina<sup>a</sup>, and A. M. Koval'skii<sup>b</sup>

<sup>a</sup> National Research Lobachevsky State University, Nizhny Novgorod, 603950 Russia

<sup>b</sup> Moscow Institute of Steel and Alloys (National University of Science and Technology), Moscow, 119049 Russia

\*e-mail: petkov@inbox.ru

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**Abstract**—Phosphates of the new formula type  $A_{5/3}MgE_{4/3}(PO_4)_3$  ( $A = K, Rb; E = Ti, Zr$ ) with a langbeinite structure have been synthesized via the sol-gel method followed by heat treatment and studied using X-ray diffraction, IR spectroscopy, and electron microscopy methods.

**Keywords:** phosphates, langbeinite structure, Rietveld method, thermal expansion

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Phosphates structurally analogous to natural mineral langbeinite  $K_2Mg_2(SO_4)_3$  [1, 2] are attractive due to the possibility to combine various chemical elements in a single lattice which can lead to interesting optical, luminescent, magnetic, and photoelectrical properties [3–12]. Owing to wide isomorphism of the cations in the substances with the langbeinite structure as well as their radiation and chemical stability, they have been considered as inert stable ceramic matrices for nuclear waste immobilization [13, 14].

The langbeinite  $M_2L_2(XO_4)_3$  crystal structure contains the  $\{[L_2(XO_4)_3]^{p-}\}_{3\infty}$  framework built of the vertex-linked  $LO_6$  octahedrons and  $XO_4$  tetrahedrons, each octahedron being linked to six tetrahedrons, and each tetrahedron being linked to four octahedrons. The framework determines the geometry of the interstitial closed voids. The crystallochemical formula of this type of phosphates is as follows:  $(M1)^{9l}(M2)^{[6-12]l}[(L1)^{6l}(L2)^{6l}(PO_4)_3]_{3\infty}$ . Position L of the framework usually contains the same or different small cations with +2, +3, or +4 oxidation state. There are two void positions (M1) and (M2) per a formula unit which can be occupied by two or (more seldom) less of the cations, being in most cases nonacoordinate with +1 or +2 oxidation state. Most of the compounds with the langbeinite structure are cubic with the  $P2_13$  space group or (less commonly) orthorhombic (for example, certain sulfates and vanadates) with the  $P2_12_12_1$  space group; some of them can undergo the  $P2_13 \rightarrow P2_12_12_1$  phase transitions [15–17].

Theoretically possible formula types (compositions) of phosphates with the langbeinite structure could be calculated from the conditions of the framework formation and the correspondence between its negative charge to the total positive charge of two populated void positions. General formula of the framework with two cations (identical or different) in positions L being in the  $m$  and  $n$  oxidation states is as follows:  $[L_r^{m+}L_s^{n+}(PO_4)_3]^{p-}$ . The  $r, s, m, n$ , and  $p$  variables are related as  $r + s = 2$  and  $9 - (m \cdot r + n \cdot s) = p$ . The latter expression follows from the electroneutrality principle. If two void positions are occupied, the  $p$  charges of the langbeinite framework can be of 2 to 4. In view of the mentioned limitations, we have earlier explored the following langbeinite-type phosphates:  $A_2^+[M^{2+}_{0.5}E^{4+}_{1.5}(PO_4)_3][Cs_2Mn_{0.5}Zr_{1.5}(PO_4)_3]$  [18] with  $r = 0.5, m = 2, s = 1.5, n = 4, p = 2$  and  $A^+M^{2+}[MgE^{4+}(PO_4)_3]$  ( $A = K, Rb, Cs; M = Sr, Pb, Ba; E = Ti, Zr$ ) [19] with  $r = 1, m = 2, s = 1, n = 4, p = 3$ . If positions L of the framework are populated with  $M^{2+}$  and  $E^{4+}$  cations (as in the latter case) yet introduce the compensating cations in oxidation states +1 and +4 in the void positions (M1 and M2) an earlier unknown formula type of the phosphates with should be crystallized in the considered structural type:  $A_{5/3}^+E_{1/3}^{4+}[M^{2+}E^{4+}(PO_4)_3] = A_{5/3}M^{2+}E_{4/3}^{4+}(PO_4)_3$ .

This study aimed to synthesize the  $A_{5/3}MgE_{4/3}(PO_4)_3$  ( $A = K, Rb; E = Ti, Zr$ ) phosphates with expected langbeinite structure, investigate their phase formation and thermal behavior, and refine their crystal structure.

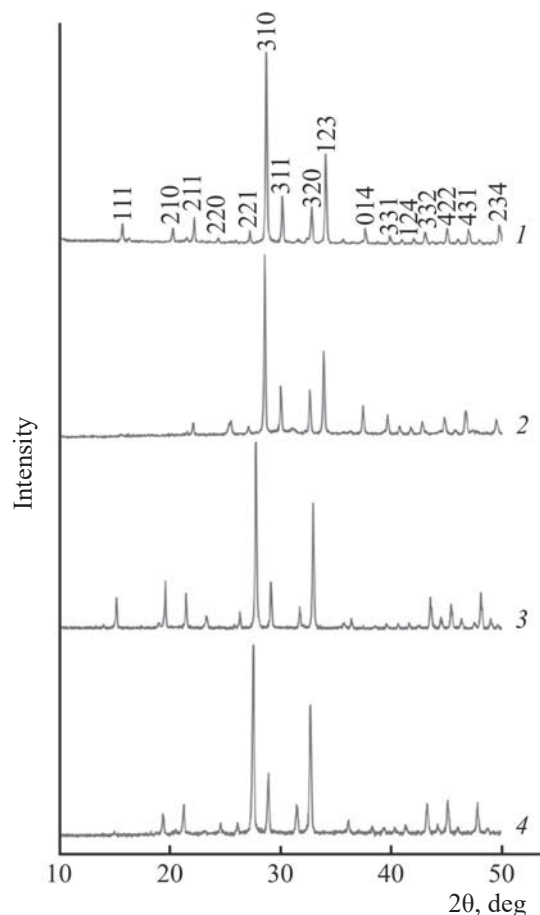
The X-ray diffraction analysis revealed the formation of single-phase  $A_{5/3}MgE_{4/3}(PO_4)_3$  ( $A = K, Rb; E = Ti, Zr$ ) phosphates with the langbeinite structure. The X-ray diffraction patterns of the samples annealed at 600–700°C contained strong reflections of the langbeinite-type phases as well as of  $TiP_2O_7$ ,  $K_3PO_4$ ,  $Rb_3PO_4$ ,  $ZrO_2$ , and  $Mg_3(PO_4)_2$ , which were completely transformed into the single-phase phosphates upon stepwise heating to 850°C (Fig. 1). Further heating improved the samples crystallinity. The unit cell parameters of the synthesized compounds were increased with the increase in radius of the alkali and(or) transition metal cation (Table 1). The compounds were crystallized in the  $P2_13$  space group, the formula units number being 4.

The data of electron microscopy and microprobe analysis revealed that the samples were homogeneous and contained of grains differing in shape. The grain size was of 1 to 5  $\mu m$ . The microanalysis data gave, for example, the following composition:  $Rb_{1.69(4)}Mg_{0.97(3)}Zr_{1.38(5)}P_{2.97(4)}O_{12}$ .

The data of IR spectroscopy which is sensitive to the local order of the atoms arrangement in the structure confirmed the orthophosphates formation. IR spectra of the obtained samples with the langbeinite structure were typical of the phosphates crystallizing in the  $P2_13$  space group. The position symmetry of the  $PO_4^{3-}$  ion was reduced to  $C_1$  in the langbeinite structure. The selection rules allowed the appearance of a single  $\nu_s$  band, two  $\delta_s$  bands, three  $\nu_{as}$  bands, and  $\delta_{as}$  bands in the IR spectra. Experimental spectra of the phosphates contained the allowed bands in the ranges of  $\nu_{as}$  (1170–1000  $cm^{-1}$ ) and  $\delta_{as}$  (670–500  $cm^{-1}$ ) vibrations, the band with maximum at  $\sim 920$   $cm^{-1}$  was assigned to  $\nu_s$  vibration, and the bands at 450 and 420  $cm^{-1}$  corresponded to  $\delta_s$  vibrations.

Crystal structure of  $Rb_{5/3}MgZr_{4/3}(PO_4)_3$  was refined using the powder X-ray diffraction data via the Rietveld method. The experiment conditions, unit cell parameters, and major results of the structure refinement are given in Table 2. Figure 2 displays the experimental, calculated, and difference X-ray diffraction patterns of  $Rb_{5/3}MgZr_{4/3}(PO_4)_3$ . Atomic coordinates and the isotropic heat parameters are collected in Table 3. Part of the  $Rb_{5/3}MgZr_{4/3}(PO_4)_3$  structure is shown in Fig. 3. The structure consisted in the  $(Mg/Zr)O_6$  octahedrons and  $PO_4$  tetrahedrons linked via the vertices and forming the three-dimensional framework. Rubidium and zirconium cations were located in large closed voids.

The  $Mg^{2+}$  and  $Zr^{4+}$  ions in the framework were disordered between two crystallographic independent

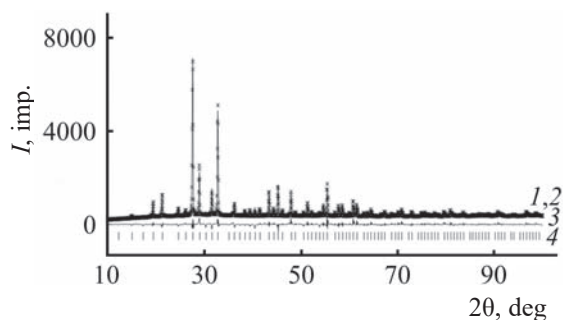


**Fig. 1.** X-ray diffraction patterns of the  $K_{5/3}MgTi_{4/3}(PO_4)_3$  (1),  $Rb_{5/3}MgTi_{4/3}(PO_4)_3$  (2),  $K_{5/3}MgZr_{4/3}(PO_4)_3$  (3), and  $Rb_{5/3}MgZr_{4/3}(PO_4)_3$  (4) phosphates.

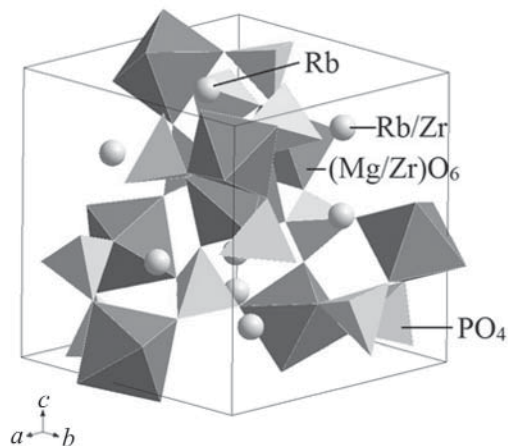
positions and were coordinated with six oxygen atoms. The populations of one of the positions were 55.7% (Mg) and 44.3% (Zr) and those for the other one were 44.3% (Mg) and 55.7% (Zr). The  $Rb^+$  and  $Zr^{4+}$  ions in the voids were also disordered between two crystallographic independent positions. One of the positions was populated by rubidium, and the populations for the other one were 66.7% (Rb) and 33.3% (Zr).

**Table 1.** Crystallographic parameters of the  $A_{5/3}MgE_{4/3}(PO_4)_3$  phosphates ( $A = K, Rb; E = Ti, Zr$ )

Phosphate	Space group (Z)	$a$ , Å	$V$ , Å <sup>3</sup>
$K_{5/3}MgTi_{4/3}(PO_4)_3$	$P2_13$ (4)	9.836(3)	951.6(1)
$Rb_{5/3}MgTi_{4/3}(PO_4)_3$	$P2_13$ (4)	9.924(4)	977.4(2)
$K_{5/3}MgZr_{4/3}(PO_4)_3$	$P2_13$ (4)	10.176(6)	1053.7(4)
$Rb_{5/3}MgZr_{4/3}(PO_4)_3$	$P2_13$ (4)	10.203(3)	1062.1(4)



**Fig. 2.** Experimental (1), calculated (2), and difference (3) X-ray diffraction patterns of  $\text{Rb}_{5/3}\text{MgZr}_{4/3}(\text{PO}_4)_3$ . The strokes (4) point at the positions of the Bragg's reflections.



**Fig. 3.** Part of the  $\text{Rb}_{5/3}\text{MgZr}_{4/3}(\text{PO}_4)_3$  structure.

Table 4 gives the calculated bond lengths in the polyhedrons; they were typical of the langbeinite structure. The void positions populated exclusively with rubidium formed the nonacoordinate oxygen polyhedrons, the bond lengths being 2.74 to 3.39 Å. The positions populated with rubidium and zirconium formed the polyhedrons with bond lengths 3.19 to 3.48 Å. Magnesium and zirconium cations in the unequally populated positions formed oxygen octahedrons, the (Mg/Zr)–O bond lengths in the framework octahedrons being 1.92 to 2.31 Å. The  $\text{PO}_4$  tetrahedrons were distorted,

**Table 2.** Experimental parameters and results of refinement for the  $\text{Rb}_{5/3}\text{MgZr}_{4/3}(\text{PO}_4)_3$  phosphate crystal structure

Parameter	Value
Space group	$P2_13$ (no. 198)
$Z$	4
$2\theta$ range, deg	10–100
$a$ , Å	10.2032(4)
$V$ , Å <sup>3</sup>	1062.2(1)
Number of reflections	112
Number of refined parameters <sup>a</sup>	28 + 18
Validity factors, %:	
$R_{\text{wp}}$ ; $R_{\text{p}}$	6.11; 4.26
$R_{\text{i}}$ ; $R_{\text{F}}$	4.15; 2.3
$S$	4.11

<sup>a</sup> The first number corresponds to background and profile parameters, scale factor, texture, and unit cell parameters; the second number corresponds to position and heat parameters of the atoms and the populations).

forming two equivalent P–O bonds (1.66–1.69 Å) and two shorter bonds (1.46 and 1.54 Å).

Our studies of the  $\text{A}_{5/3}\text{MgE}_{4/3}(\text{PO}_4)_3$  ( $\text{A} = \text{K}, \text{Rb}$ ;  $\text{E} = \text{Ti}, \text{Zr}$ ) compounds extended the knowledge of the phase formation of the known langbeinite-type phosphates. The refined structure revealed that the framework was populated with  $\text{E}^{4+}$  and magnesium cations, whereas the  $\text{A}^+$  and  $\text{E}^{4+}$  cations were located in voids. Such distribution of the cations between the framework and void positions allowed relaxation of the deformed structure, decrease in the inner strains, and, hence, increase in the thermal stability.

Thermal behavior of the obtained  $\text{K}_{5/3}\text{MgTi}_{4/3}(\text{PO}_4)_3$  and  $\text{K}_{5/3}\text{MgZr}_{4/3}(\text{PO}_4)_3$  phosphates was studied by means of high-temperature X-ray diffraction analysis. The parameter  $a$  and volume  $V$  of the cubic unit cells of the langbeinite structure grew linearly with temperature. Those materials exhibited isotropic expansion upon heating, showing positive local thermal expansion coefficients. Thermal expansion directly affected the mean kinetic energy of the vibrating particles in the body and average distance between the lattice sites and was related to asymmetry (anharmonicity) of heat vibrations of the atoms, leading to the changes in the interatomic distances with temperature. Thermal expansion coefficients calculated over 25–800°C showed that  $\text{K}_{5/3}\text{MgZr}_{4/3}(\text{PO}_4)_3$  was an intermediate-expanding substance ( $\alpha_a = 5 \times 10^{-6} \text{ }^\circ\text{C}^{-1}$ ), whereas  $\text{K}_{5/3}\text{MgTi}_{4/3}(\text{PO}_4)_3$  was a strongly expanding substance ( $\alpha_a = 11 \times 10^{-6} \text{ }^\circ\text{C}^{-1}$ ). Thermal expansion coefficient is related to the strength of the chemical bonds in the structure: stronger expansion corresponds to weaker bonding of the atoms [20, 21].

**Table 3.** Coordinates and isotropic heat parameters of atoms in the  $\text{Rb}_{5/3}\text{MgZr}_{4/3}(\text{PO}_4)_3$  structure

Atom	Position	<i>x</i>	<i>y</i>	<i>z</i>	<i>B</i> , Å <sup>2</sup>
A <sup>1a</sup>	4 <i>a</i>	0.2923(4)	0.2923(4)	0.2923(4)	2.2(4)
A <sup>2a</sup>	4 <i>a</i>	0.0941(7)	0.0941(7)	0.0941(7)	3.1(7)
M <sup>1a</sup>	4 <i>a</i>	0.8536(9)	0.8536(9)	0.8536(9)	0.74(3)
M <sup>2a</sup>	4 <i>a</i>	0.5686(6)	0.5686(6)	0.5686(6)	0.91(6)
P	12 <i>b</i>	0.2614(13)	-0.3758(16)	0.4655(15)	1.2(5)
O <sup>1</sup>	12 <i>b</i>	-0.0047(24)	0.0791(23)	0.3428(22)	0.65(7)
O <sup>2</sup>	12 <i>b</i>	0.2612(23)	-0.3966(22)	0.3043(26)	0.65(7)
O <sup>3</sup>	12 <i>b</i>	0.4915(28)	0.0389(29)	0.3017(26)	0.65(7)
O <sup>4</sup>	12 <i>b</i>	0.1841(25)	-0.2569(29)	0.4807(29)	0.65(7)

<sup>a</sup> Population: A<sup>1</sup> [0.667Rb<sup>+</sup> + 0.333Zr<sup>4+</sup>], A<sup>2</sup> [Rb<sup>+</sup>], M<sup>1</sup> [0.557Zr<sup>4+</sup> + 0.443Mg<sup>2+</sup>], M<sup>2</sup> [0.443Zr<sup>4+</sup> + 0.557Mg<sup>2+</sup>].

Evidently, the P–O bonds in the tetrahedrons of the isostructural and isoformula phosphates were stronger than the L–O bonds in the LO<sub>6</sub> polyhedron (L = Zr, Ti). Zirconium and titanium formed the bonds differing in the covalent contribution. Larger size of the Zr<sup>4+</sup> cation determined higher strength (covalent contribution) of the Zr–O bonds in comparison with the Ti–O ones. As a result, heat-induced deformation of the  $\text{K}_{5/3}\text{MgZr}_{4/3}(\text{PO}_4)_3$  structure was less than for  $\text{K}_{5/3}\text{MgTi}_{4/3}(\text{PO}_4)_3$ . Since expansion of the substances was isotropic ( $\alpha_a = \alpha_b = \alpha_c$ ), their volumetric thermal expansion coefficient ( $\alpha_V$ ) was three times the linear thermal expansion coefficient.

In summary, the family of the langbeinite-type structures was complemented by a novel formula type  $\text{A}_{5/3}\text{E}_{17/3}[\text{M}^{2+}\text{E}^{4+}(\text{PO}_4)_3]$ :  $\text{A}_{5/3}\text{MgE}_{4/3}(\text{PO}_4)_3$  (A = K, Rb; E = Ti, Zr) phosphates. We studied their phase formation, refined the crystal structure, and investigated the thermal stability as well as thermal expansion. It was shown that the structure was based in a three-dimensional framework constituted by the PO<sub>4</sub> tetrahedrons and (Mg/E)O<sub>6</sub> octahedrons linked via the vertexes, whereas potassium (rubidium) and zirconium atoms populated the void positions. The heating of the phosphates from 25 to 800°C led to isotropic expansion and did not induce any polymorph transitions. Linear thermal expansion coefficients of  $\text{K}_{5/3}\text{MgZr}_{4/3}(\text{PO}_4)_3$  and  $\text{K}_{5/3}\text{MgTi}_{4/3}(\text{PO}_4)_3$  were typical of intermediate and strongly expanding substances, respectively.

#### EXPERIMENTAL

The  $\text{A}_{5/3}\text{MgE}_{4/3}(\text{PO}_4)_3$  (A = K, Rb; E = Ti, Zr) phosphates were prepared via a sol-gel method followed by heat treatment. The synthesis was performed as follows: stoichiometric amounts of aqueous solutions of the alkali metal (K or Rb) chloride, magnesium chloride, titanium or zirconium oxychloride, and ammonium

dihydrophosphate (all of “chemical pure” grade) were mixed. The mixture was then dried at 90 and 150°C (12 h at each stage) and subject to stepwise heat treatment in air at 600–900°C (20–26 h at each stage). After each heating step, the mixture was homogenized by dispersion. The powder was pressed in disc to ensure more complete conversion.

The samples were characterized by means of electron microscopy and local microanalysis, X-ray diffraction analysis, and IR spectroscopy. Chemical composition and uniformity of the phosphates were monitored using a JEOL JSM-7600F scanning electron microscope equipped with a dark-field electron gun (the Schottky cathode). The microscope was equipped with an OXFORD X-Max 80 energy-dispersive spectrometer (Premium) with semiconducting silicon-drift detector

**Table 4.** Selected interatomic distances (*d*) in (Mg/Zr)O<sub>6</sub>, (Rb/Mg)O<sub>6</sub>, and PO<sub>4</sub> polyhedrons of the  $\text{Rb}_{5/3}\text{MgZr}_{4/3}(\text{PO}_4)_3$  structure

Bond	<i>d</i> , Å
Rb <sup>1</sup> /Zr <sup>1</sup> –O <sup>2</sup> ( <i>x</i> = 3)	3.192(27)
Rb <sup>1</sup> /Zr <sup>1</sup> –O <sup>4</sup> ( <i>x</i> = 3)	3.209(26)
Rb <sup>1</sup> /Zr <sup>1</sup> –O <sup>3</sup> ( <i>x</i> = 3)	3.481(28)
Rb <sup>2</sup> –O <sup>1</sup> ( <i>x</i> = 3)	2.735(24)
Rb <sup>2</sup> –O <sup>3</sup> ( <i>x</i> = 3)	3.392(31)
Rb <sup>2</sup> –O <sup>4</sup> ( <i>x</i> = 3)	3.036(29)
Mg <sup>1</sup> /Zr <sup>2</sup> –O <sup>2</sup> ( <i>x</i> = 3)	1.918(27)
Mg <sup>1</sup> /Zr <sup>2</sup> –O <sup>3</sup> ( <i>x</i> = 3)	1.995(31)
Mg <sup>2</sup> /Zr <sup>3</sup> –O <sup>1</sup> ( <i>x</i> = 3)	1.909(24)
Mg <sup>2</sup> /Zr <sup>3</sup> –O <sup>4</sup> ( <i>x</i> = 3)	2.309(29)
P–O <sup>4</sup>	1.455(32)
P–O <sup>3</sup>	1.539(33)
P–O <sup>2</sup>	1.658(31)
P–O <sup>1</sup>	1.689(28)



allowing local microanalysis of the specimen. The error in determining the elemental composition of the samples was no more than 2 at %.

X-ray diffraction analysis was performed using a Shimadzu XRD-6000 diffractometer ( $\text{CuK}\alpha$  radiation,  $\lambda = 1.54178 \text{ \AA}$ ,  $2\theta$  range  $10^\circ\text{--}60^\circ$ ). The analysis was performed upon each heat treatment step to gain information on the phase composition of the substances. High-temperature measurements were performed using the same instrument equipped with an HA-1001 Shimadzu attachment allowing discrete heating at  $20\text{--}800^\circ\text{C}$  in  $100^\circ\text{C}$  step. Temperature was measured using a platinum-rhodium thermocouple with accuracy of  $\pm 2\%$ . Unit cell parameters were determined by the least squares method from the indicated diffraction patterns.

Structure of the  $\text{Rb}_{5/3}\text{MgZr}_{4/3}(\text{PO}_4)_3$  sample was investigated by recording the X-ray diffraction pattern over  $2\theta = 10^\circ\text{--}110^\circ$  with  $0.02^\circ$  step (15 s per a point). The structure was refined by means of full-profile analysis of the powder diffraction data using RIETAN-97 software [22, 23]. The peaks were fitted using modified pseudo-Voigt function [24].  $\text{K}_{1.75}\text{Ti}_2(\text{PO}_4)_3$  was used as the model for the structure refinement [25].

IR spectroscopy studies were performed to confirm the functional composition of the samples. The spectra were recorded using an FSM-1201 spectrometer ( $400\text{--}1400 \text{ cm}^{-1}$ ).

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#### CONFLICT OF INTEREST

No conflict of interest was declared by the authors.

#### REFERENCES

- Sarrign, M.L.M., Clemente, A.R., and Vila, L.M., *J. Solid State Chem.*, 1990, vol. 84, p. 308.  
[https://doi.org/10.1016/0022-4596\(90\)90329-V](https://doi.org/10.1016/0022-4596(90)90329-V)
- Zatovskii, I.V., Slobodyanik, N.S., Ushchapivskaya, T.I., Ogorodnik, I.V., and Babarik, A.A., *Russ. J. Appl. Chem.*, 2006, vol. 79, p. 10.  
<https://doi.org/10.1134/S1070427206010034>
- Pet'kov, V.I., Shchelokov, I.A., Surazhskaya, M.D., Palkina, K.K., Kanishcheva, A.S., and Knyazev, A.V., *Russ. J. Inorg. Chem.*, 2010, vol. 55, p. 1352.  
<https://doi.org/10.1134/S0036023610090044>
- Souamti, A., Martín, I.R., Zayani, L., Hernández-Rodríguez, M.A., Soler-Carracedo, K., Lozano-Gorrín, A.D., Lalla, E., and Ben Hassen Chehimi, D., *J. Luminescence*, 2016, vol. 177, p. 160.  
<https://doi.org/10.1016/j.jlumin.2016.04.045>
- Battle, P.D., Gibb, T.C., Nixon, S., and Harrison, W.T.A., *J. Solid State Chem.*, 1988, vol. 75, p. 21.  
[https://doi.org/10.1016/0022-4596\(88\)90299-X](https://doi.org/10.1016/0022-4596(88)90299-X)
- Ogorodnyk, I.V., Zatovsky, I.V., Slobodyanik, N.S., Baumer, V.N., and Shishkin, O.V., *J. Solid State Chem.*, 2006, vol. 179, p. 3461.  
<https://doi.org/10.1016/j.jssc.2006.07.015>
- Sadhasivam, S., Manivel, P., Jeganathan, K., Jayasankar, C.K., and Rajesh, N.P., *Mater. Lett.*, 2017, vol. 188, p. 399.  
<https://doi.org/10.1016/j.matlet.2016.11.091>
- Liu, J., Duan, X., Zhang, Y., Li, Z., Yu, F., and Jiang, H., *J. Alloys Compd.*, 2016, vol. 660, p. 356.  
<https://doi.org/10.1016/j.jallcom.2015.11.147>
- Pet'kov, V.I., Asabina, E.A., Markin, A.V., Alekseev, A.A., and Smirnova, N.N., *J. Therm. Anal. Cal.*, 2016, vol. 124, no. 3, p. 1535.  
<https://doi.org/10.1007/s10973-016-5319-8>
- Souamti, A., Kahlaoui M, Mohammed, B., Lozano-Gorrín, A.D., and Chehimi, D.B.H., *Ceram. Int.*, 2017, vol. 43, no. 14, p. 10939.  
<https://doi.org/10.1016/j.ceramint.2017.05.132>
- Asabina, E.A., Pet'kov, V.I., Gobechiya, E.R., Kaba-lov, Yu.K., Pokholok, K.V., and Kurazhkovskaya, V.S., *Russ. J. Inorg. Chem.*, 2008, vol. 53, p. 40.  
<https://doi.org/10.1134/S0036023608010075>
- Tsyrenova, G.D. and Pavlova, N.N., *Inorg. Mater.*, 2011, vol. 47, no. 7, p. 786.  
<https://doi.org/10.1134/S0020168511070235>
- Pet'kov, V.I., Asabina, E.A., Lukuttsov, A.A., Korchemkin, I.V., Alekseev, A.A., and Demarin, V.I., *Radiochemistry*, 2015, vol. 57, no. 6, p. 632.  
<https://doi.org/10.1134/S1066362215060119>
- Pratheep Kumar, S. and Gopal, B., *J. Alloys Compd.*, 2016, vol. 657, p. 422.  
<https://doi.org/10.1016/j.jallcom.2015.10.088>
- Glogarová, M., *Phys. Status Solidi (A)*, 1974, vol. 22, no. 1, p. 69.  
<https://doi.org/10.1002/pssa.2210220159>
- Hikita, T., Kitabatake, M., and Ikeda, T., *J. Phys. Soc. Japan*, 1980, vol. 49, no. 4, p. 1421.  
<https://doi.org/10.1143/JPSJ.49.1421>
- Trussov, I.A., Male, L.L., Sanjuan, M.L., Orera, A., and Slater, P.R., *J. Solid State Chem.*, 2019, vol. 272, p. 157.  
<https://doi.org/10.1016/j.jssc.2019.02.014>

18. Zaripov, A.R., Orlova, V.A., Pet'kov, V.I., Slyunchev, O.M., Galuzin, D.D., and Rovnyi, S.I., *Russ. J. Inorg. Chem.*, 2009, vol. 54, p. 45  
<https://doi.org/10.1134/S0036023609010112>
19. Pet'kov, V.I., Alekseev, A.A., Asabina, E.A., Borovikova, E.Y., and Koval'ski, A.M., *Russ. J. Inorg. Chem.*, 2017, vol. 62, no. 7, p. 870  
<https://doi.org/10.1134/S0036023617070178>
20. Filatov, S.K., *Vysokotemperaturnaya kristalloghimiya. Teoriya, metody i rezul'taty issledovaniy* (High Temperature Crystal Chemistry. Theory, Methods, and Research Results), Leningrad: Nedra, 1990.
21. Krivovichev, S.V. and Filatov, S.K., *Kristalloghimiya mineralov i neorganicheskikh soedinenii s kompleksami anionotsentrirovannykh tetraedrov* (Crystal Chemistry of Minerals and Inorganic Compounds with Complexes of Anion-Centered Tetrahedra), St. Petersburg: SPbGU, 2001.
22. Rietveld, H.M., *Acta Crystallogr.*, 1967, vol. 22, no. 1, p. 151.  
<https://doi.org/10.1107/S0365110X67000234>
23. Kim, Y.I. and Izumi, F., *J. Ceram. Soc. Japan*, 1994, vol. 102, p. 401.  
<https://doi.org/10.2109/jcersj.102.401>
24. Izumi, F., *The Rietveld Method*, New York: Oxford University Press, 1993, ch. 13, p. 236.  
<https://doi.org/10.1002/crat.2170300412>
25. Leclaire, A., Benmoussa, A., Borel, M.M., Grandin, A., and Raveau, B., *J. Solid State Chem.*, 1989, vol. 78, p. 227.  
[https://doi.org/10.1016/0022-4596\(89\)90101-1](https://doi.org/10.1016/0022-4596(89)90101-1)