STRUCTURE OF INORGANIC COMPOUNDS

Crystal Growth, Structural Elucidation, and Accurate (CHARDI/BVS) Investigations of Zn₃Fe₄(PO₄)₆ with Howardevansite-like Structure

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Received June 30, 2020; revised June 30, 2020; accepted August 3, 2020

Abstract—Zinc and iron-based phosphate $Zn_3Fe_4(PO_4)_6$ was synthesized by solid-state reaction and characterized by single-crystal X-ray diffraction and energy dispersive X-ray spectroscopy. The structure of the title compound belongs to the Howardevansite family and crystallizes in the triclinic system, space group PI with the following unit cell parameters a = 6.3447(2) Å, b = 7.8838(2) Å, c = 9.3019(2) Å, $\alpha = 104.726(1)^\circ$, $\beta = 108.408(1)^\circ$, $\gamma = 101.245(1)^\circ$, and Z = 1. The proposed structural model was supported by both charge distribution and bond valence sum validation methods. The crystal structure is built up from kinked chains of edge-sharing polyhedra following the sequence $[Fe(1)_2O_{10}]-[Zn(1)O_5]-[Fe(2)_2O_{10}]$. These chains are held together through PO₄ tetrahedra forming a stacked layers perpendicular to [010]. The resulting 3D framework defines tunnels running along [100], in which some Zn²⁺ cations are located.

DOI: 10.1134/S1063774521070087

INTRODUCTION

The research of new mixed metal phosphate with open-framework architectures is currently a subject of perpetual development in solid-state chemistry. Especially iron phosphate which are of particular interest because of their highly attractive potentials as a promising functional material for many practical applications including phosphate glass fibers (PGFs) engineering [1], catalysis [2], corrosion inhibition [3], biosensing [4], composites materials [5], electrode for Liion batteries [6], and immobilization of nuclear waste [7]. Moreover, these kinds of compounds exhibit fascinating structural chemistry, with a large variety of chemical compositions due to the tendency of iron (Fe²⁺or Fe³⁺) to adopt different coordinate geometries FeO_n (n = 4, 5, or 6), as well as a versatile structural arrangements [8]. In fact, their impressive crystal structures lead generally to noteworthy, physical and/or chemical properties [9]. Typical examples are phases with an original magnetic behavior [10] or characterized by a spin-order-induced ferroelectricity and magnetoelectric effect [11], important electronic and optical attributes [12], Intervalence Charge Transfer (IVCT) phenomenon [13], and offering a facile redox reactions [14].

Based on our previous studies concerning iron phosphate materials, a number of new compounds with distinctive topological networks have been elabo-

rated and structurally characterized. For instance we cite Ag_{1.655}Co_{1.64}Fe_{1.36}(PO₄)₃[15], Na₂Co₂Fe(PO₄)₃[16], $Na_{1.67}Zn_{1.67}Fe_{1.33}(PO_4)_3$ [17], and $(M,Mn)_3Fe(PO_4)_3$ (M = Ca, Cd) [18] with an alluaudite-like structure or $M'Ni_2Fe(PO_4)_3$ (M' = Ca [19], Sr [20]) adopting the α -CrPO₄ structure type. Additional phases such $M''Mn_2Fe(PO_4)_3$ (M'' = Ba, Sr) [21] and CaZn₂Fe(PO₄)₃ [22] were also reported. During the investigation of BaO-ZnO-Fe₂O₃-P₂O₅ pseudo-quaternary system, and in an attempt to obtain the new compound $BaZn_2Fe(PO_4)_3$ belonging to one of the latter structure types, $Zn_3Fe_4(PO_4)_6$ was successfully isolated. Its structure is similar to that of the natural Howardevansite NaCuFe₂(VO₄)₃ [23]. Since the first description of the crystal structure of mixed-valence Fe-phosphate $Fe_3^{2+}Fe_4^{3+}(PO_4)_6$ [24], the number of the papers devoted to the Howard evansite-type $A_{3+n}^{2+}B_{4-n}^{3+}(PO_4)_{6-n}(HPO_4)_n$ $(n \le 4)$ [25], has been increased significantly. As a result, multiple compounds covering a wide range of substitution exist e.g., $A_x B_v (XO_4)_6$ with $(A = Fe^{3+})$, Mg²⁺, Cu²⁺, Zn²⁺, Co²⁺, Na⁺, and Ag⁺; $B = Fe^{3+}$, Ti⁴⁺, Mg²⁺, Cu²⁺, Zn²⁺, Co²⁺; $X = P^V$, V^V, Mo^{VI}) and $6 \le x + y \le 8$ [26], and $M_7^{II}(PO_4)_2(HPO_4)_4$ ($M^{II} = Mg$ [27], Fe [28], Mn [29], Co [30]).

Herein we report the synthesis of $Zn_3Fe_4(PO_4)_6$ via the classical solid-state route and its characterization,



Fig. 1. SEM micrographs and EDS spectrum of $Zn_3Fe_4(PO_4)_6$.

through single-crystal X-ray diffraction (XRD) supported by the energy-dispersive X-ray spectroscopy (EDS). The resulting structural model has been supported by means of bond-valence-sum (BVS) [31] and charge-distribution (CHARDI) [32] validation tools.

EXPERIMENTAL

Synthesis and Chemical Composition

Single crystals of $Zn_3Fe_4(PO_4)_6$ were grown by conventional solid-state reaction in air. In the first step, appropriate amounts of the starting precursors: $Ba(NO_3)_2$, $Zn(NO_3)_2 \cdot 6H_2O$, $Fe(NO_3)_3 \cdot 9H_2O$, and NH₄H₂PO₄ were weighted in molar ratios Ba: Zn: Fe: P = 1:2:1:3 and subsequently dissolved in concentrated nitric acid. The resulting mixture was stirred at room temperature for 24 h and then evaporated under moderate stirring until drvness. The dry was carefully milled in an agate mortar, placed in a platinum crucible and preheated at 723 K for 24 h, to remove gaseous products. After regrinding, the residue was melted for 1 h at 1150 K and cooled down slowly to 950 K at rate of 5 K/h and then rapidly to room temperature by turning off the furnace. Finally, green crystal of $Zn_3Fe_4(PO_4)_6$ with parallelepiped shape and optimal size have been picked up under polarizing microscope for the X-ray diffraction (XRD) and the energy dispersive X-ray spectroscopy (EDS) experiments.

The qualitative analysis and morphology of the synthetized single crystals were performed using JEOL JSM-IT100 InTouchScopeTM scanning electron microscope (SEM) equipped with energy dispersive X-ray spectrometer (EDS), operating under 20 kV accelerating voltage. The EDS analysis unambiguously confirms the exclusive presence of Zn, Fe, P, and O atoms. Furthermore, the Zn/P and Fe/P ratios are in agreement with its nominal composition resulting from single crystal X-ray diffraction study. The SEM micrograph of $Zn_3Fe_4(PO_4)_6$ crystal and EDS spectrum are given in Fig. 1. Elemental mapping SEM-EDS analysis was also carried out to appraise the elemental spatial repartition. As shown in Fig. 2 Zn, Fe, P, and O are evenly distributed in the sample.

X-ray Diffraction Study

A suitable single crystal of dimensions $0.36 \times 0.15 \times$ 0.08 mm³, was selected for X-ray data collection at 296 K. The specimen was glued to a thin glass fiber and mounted on a Bruker X8 APEXII diffractometer equipped with an Apex II CCD area detector and a sealed-tube X-ray source (Mo K_{α} radiation, $\lambda =$ 0.71073 Å). The APEX2 software was used for data collection over a full sphere of reciprocal space with φ and ω scan. Cell refinement and data reduction were carried out using SAINT program. A total number of 25997 reflections were measured in the range of $2.5^{\circ} \leq$ $\theta \leq 35.6^{\circ}$ of which 3740 were independent and 3638 reflections with the statistical criterion $I > 2\sigma(I)$. Complementary absorption corrections were introduced semi-empirically from equivalent reflections on the basis of multi-scans using SADABS program [33]. The structure was solved by Direct Methods in the space group $P\overline{1}$ (no. 2) and refined anisotropically by full-matrix least-squares on F^2 using SHELXT 2014/7 [34] and SHELXL2018/3 [35] programs implemented in the software package WinGX [36]. Following the proposed structural model, the final refinement cycle including all atomic positions and their anisotropic displacement parameters converged to the reduced reliability factors $R_1 = 0.017$ and $wR_2 = 0.043$. At this stage the remaining electron densities in the last Fourier-difference map are $\Delta \rho_{min} = -1.195$ and $\Delta \rho_{max} =$ 0.615 $e^{A^{-3}}$ located at 0.30 Å from the Zn(2) site and 0.73 Å from the P(2) one, respectively.

Crystallographic data, X-ray data collection and structure refinement details are compiled in Table 1. The coordinates of the basic atoms and their displacement parameters are listed in Table 2. Selected interatomic distances are summarized in Table 3.

RESULTS AND DISCUSSION

Validation of Structural Model

The powerful analysis tools (BVS) and (CHARDI), were chosen to confirm the validity as well as to prove the adequacy of the proposed structural model of $Zn_3Fe_4(PO_4)_6$. Firstly, the (BVS) con-



Fig. 2. EDS elemental mapping images showing the distribution of chemical elements in $Zn_3Fe_4(PO_4)_6$ crystal.

cept was introduced successfully to predict the bond lengths, estimates the quality of the crystal structure determination as well as the bond strain. In this model all, the neighboring cation-anion distances are taken into account and considered as bonds, with unequal strengths. The bond valence or bond strength S_{ij} for a bond set up between two ions *i* and *j* can be calculated by the following relation:

$$S_{ij} = \sum_{j=1}^{k} \exp((r_0 - t_{ij})/B),$$

where t_{ij} is the bond length between two ions *i* and *j*, while r_0 and *B* (usually B = 0.37 Å), are empirical constants tabulated and determined for each pair of cation-anion. In practice, the valence or oxidation state V(i) of ions *i* coordinated by the *k* nearest counter-ions *j* obey to the relation [37]:

$$V(i) = \sum_{j=1}^{k} S_{ij} = \sum_{j=1}^{k} \exp((r_0 - t_{ij})/B).$$

The CHARDI analysis is fundamentally differs from BVS calculations by exploiting the true bond distances in a self-consistent computation, instead of employing empirical curves (R - s) requiring empirical parameters. This method adopt a Madelung-type description of crystal structures in which each atom is described as a point charge, where the formal charge corresponds to the oxidation number [38]. The CHARDI analysis method gives the distribution of calculated ECoN (effective coordination number) of a central atom among all the neighbouring ligands [39]. The calculation of this number is directly correlated to the distribution of atomic charges in crystalline structures. The correctness of the structure (cation ratio) and of the degree of over or under bonding (OUB) (anion ratio) is performed via the evaluation of the internal criterion q(i)/Q(i) (where q(i) is the formal oxidation number, and Q(i) represents the computed charge) [40]. For a correctly refined structure, the calculated valences V(i) giving by the (BVS) approach and the computed charge Q(i) according to the CHARDI concept must be in close agreement with the oxidation number of each atoms. The CHARDI computations were carried out with the CHARDI2015 Fortran program [41], while the BVS calculation was performed using EXPO2014 program [42] (Table 4). The crystal structure of the title compound shows no statistical disorder of the seven metallic sites. Further-

Table 2. Atomic coordinates and their equivalent thermal parameters in $Zn_3Fe_4(PO_4)_6$ structure

Chemical formula	$Zn_3Fe_4(PO_4)_6$	Atom	x/a	y/b	z/c		$U_{\rm eq}, {\rm \AA}^2$
М	989.33	Zn1	0.78569(3)	0.30909(2) 0.78921	(2) ().00725(4)
Crystal system, space	Triclinic, $P\overline{1}$, 1	Zn2	0.50000	0.50000	0.50000).01051(5)
group, Z		Fe1	0.45501(3)	0.77678(2	2) 0.97212	(2)).00463(4)
<i>a</i> , <i>b</i> , <i>c</i> , Å	6.3447(2), 7.8838(2),	Fe2	0.88374(3)	0.951650	2) 0.61508	(2)).00445(4)
	9.3019(2)	P1	0.26965(5)	0.35085(4	4) 0.73002	(3)).00310(5)
$\alpha, \beta, \gamma, deg$	104.726(1), 108.408(1),	P2	0.89680(5)	0.72636(4	(1) 0.86963		0.00363(5)
VÅ ³	101.243(1) 407.31(2)	P3	0.40946(5	0.90689(4) 0.66746	(3)).00292(5)
V, A	407.31(2)	01	0.2892(1)	0.551(1)	0.7947($\hat{\mathbf{n}}$).0086(1)
$D_x, g/cm$	$M_{0} K = 0.71073$	02	0.0412(1)	0.2323(1)	0.7263($\dot{\mathbf{n}}$).006(1)
mm^{-1}	8 54	03	0.2724(1)	0.3014(1)	0.5578(1) ().0066(1)
μ , mm $T K$	296	04	0.4792(1)	0.3082(1)	0.8392($\hat{\mathbf{D}}$	0.0067(1)
Sample size mm^3	$0.36 \times 0.15 \times 0.08$	05	0.7743(1)	0.5342(1)	0.7385($\dot{1}$).008(1)
Diffractometer	Bruker X8 APEXII	O6	0.9477(1)	0.8636(1)	0.7869(1) 0).0095(1)
Scan mode	ϕ and ϕ	07	0.7484(1)	0.7751(1)	0.9639($\hat{1}$).0099(1)
T_{\min}, T_{\max}	0.3910, 0.7470	08	1.1312(1)	0.7308(1)	0.9929(1) 0).0072(1)
$\theta_{\rm max}$, deg	35.595	09	0.4368(1)	0.9514(1)	0.8469(1) 0).0061(1)
Ranges of indices h, k, l	$-10 \le h \le 10, -12 \le k \le 12,$	O10	0.1897(1)	0.96(1)	0.5839(1) 0).0058(1)
	$-15 \le l \le 15$	O11	0.6233(1)	1.0366(1)	0.6611(1) ().0059(1)
Number of reflections:	25997/3740,0.030/3638	O12	0.3911(1)	0.7123(1)	0.5851(1) ().0093(1)
measured/unique,				,	,	,	
$R_{\rm int}$ /with $I > 2\sigma(I)$		Table 3	Selected	hand length	s (Å) in the	cryst	al structure
Refinement technique	Full-matrix least squares method on F^2	of $Zn_3Fe_4(PO_4)_6$				ui structure	
Number of refined parame-	170	Zn(1)	05	1.958(1)		06	1.862(1)
ters	170		O2 ⁱ	2.0343(9)		O 11	2.0238(9)
Weighting scheme	$w = 1/[\sigma^2(F_o^2) + (0.0145P)^2$		O11 ⁱⁱ	2.0388(9)		O3 ^{iv}	2.0341(9)
	+ 0.5521P],		O8 ⁱⁱⁱ	2.0491(9)		O 10 ⁱ	2.0424(9)
	where $P = (F_o^2 + 2F_c^2)/3$		04	2.1371(9)		O2 ^{viii}	2.0698(9)
R_1/wR_2	0.017/0.043	$\langle Zn(1) \rangle$	$ -0\rangle$	2.0435	$\langle Fe(2)-O \rangle$		2.0202
S	1.08	Zn(2)	O12 ^{iv}	2.0071(9)	P(1)	01	1.503(1)
$\Delta \rho_{\rm min} / \Delta \rho_{\rm max}$, $e {\rm \AA}^{-3}$	0.615/-1.195	. /	012	2 0071(0)		02	1 5 402(0)

more, the repartition of the electric charges at the nineteen independent crystallographic sites of the asymmetric unit shows that the Zn²⁺, Fe³⁺, and P⁵⁺ cations occupy only 7 sites. Otherwise, balancing electrical charges requirement imply the repartition of additional 12 oxygen atoms in the remaining 2i sites. A close examination of the obtained results (Table 4), reveals that the obtained values of Q(i) charges and V(i) bond-valence-sums of all cations are very close to the corresponding weighted oxidation number q(i).sof(i). Some minor fluctuations were also reported, revealing the distortion of the accommodate sites. Despite these irregularities, all the values of the internal criterion q(i)/Q(i) are very close to unity, supporting the validity of the proposed structural model.

	011 ⁱⁱ	2.0388(9)		O3 ^{iv}	2.0341(9)
	O8 ⁱⁱⁱ	2.0491(9)		O10 ⁱ	2.0424(9)
	O4	2.1371(9)		O2 ^{viii}	2.0698(9)
(Zn(1)-O)		2.0435	$\langle Fe(2)-O \rangle$		2.0202
Zn(2)	O12 ^{iv}	2.0071(9)	P(1)	O 1	1.503(1)
	O12	2.0071(9)		O2	1.5492(9)
	O3	2.2054(9)		O4	1.5516(9)
	O3 ^{iv}	2.2054(9)		O3	1.5563(9)
	O5 ^{iv}	2.2575(9)	$\langle P(1)-O\rangle$		1.5400
	O5	2.2575(9)	P(2)	O 7	1.514(1)
(Zn(2)-O)		2.1567		O6	1.520(1)
Fe(1)	O 7	1.8896(9)		O5	1.546(1)
	O 1	1.9182(9)		O 8	1.5521(9)
	O4 ^v	1.9864(9)	$\langle P(2)-O\rangle$		1.5330
	O9	2.0141(9)	P(3)	O12	1.494(1)
	O8 ^{vi}	2.0946(9)		O 11	1.5586(9)
	O9 ^{vii}	2.1819(9)		O10	1.5599(9)
$\langle Fe(1)-O \rangle$		2.0141		09	1.5605(9)
Fe(2)	O 10 ^{ix}	2.0892(9)	$\langle P(3)-O\rangle$		1.5433

Table 4. CHARDI and BVS analysis for the principal cations in the title compound Cation q(i).sof(i) CN(i) ECoN(i)V(i)Q(i)q(i)/Q(i)5 2.022 Zn1 2 4.86 2.01 0.99 Zn2 2 6 5.42 1.846 2.05 0.98 3 5.49 3.118 3.01 Fe1 6 1.00 3 Fe₂ 6 5.67 3.029 2.96 1.01

q(i) is formal oxidation number; sof(i) is site occupancy; CN(i) is	s
classical coordination number; $Q(i)$ is calculated charge; $V(i)$ is	s
calculated valence; ECoN(<i>i</i>) is effective coordination number.	

3.97

3.98

3.95

4.933

5.022

4.899

4.95

5.03

5.02

1.01

0.99

1.00

For a structure described as built on classical cation-centered, the Mean Absolute Percentage Deviation (MAPD) was successfully introduced to evaluate the convergence of the CHARDI model. The MAPD measures the agreement between q(i) and Q(i) for the whole sets of PC atoms (polyhedron-centering atoms) and of V atoms (the vertex atoms) [43].

MAPD =
$$\frac{100}{N} \sum_{i=1}^{N} \left| \frac{q(i) - Q(i)}{q(i)} \right|$$

where N is the number of polyhedron-centering or vertex atoms in the asymmetric unit. For the PC atoms a MAPD > 10% results in an incorrect analysis or bad accuracy of refinement, while for the V atoms a MAPD < 5% indicates a low OUB effect, which can increase proportionally to MAPD. In the structure, such effect is noticeable for some oxygen atoms labelled O(1), O(3), O(5), O(6), O(8), O(9), and O(11), which deviate slightly from the ideal value of -2 (Table 5). Following the current distribution scheme, the obtained value of MPAD for the positive charges is only 0.8 and 2.9% for the negative ones, which proves once again the applicability and adequacy of the experimental model. In order to verify the chemical plausibility of the crystal structure we have also looked into the Global Instability Index (GII) [44]:

$$\text{GII} = \sqrt{\sum_{i=1}^{N} \left(\left(\sum_{j} S_{ij} - V_{i} \right)^{2} / N \right)}.$$

The GII index evaluates the rationality of the structure and measures the deviation of the bond-valence sums from the formal valence V(i) averaged over all N atoms of the asymmetric unit.

Indeed, for an unstrained structure GII < 0.1 v.u. and may approach 0.2 v.u. in a structure with lattice-induced strains. Values larger than 0.2 v.u. indicates

Table 5.	CHARDI	calculation	for the	oxygen	anions	in t	he
title com	pound						

Anion	q(i).sof(i)	Q(i)	q(i)/Q(i)
01	-2	-2.08	0.96
O2	-2	-2.04	0.98
O3	-2	-1.93	1.04
O4	-2	-2.02	0.99
O5	-2	-1.92	1.04
O6	-2	-2.07	0.97
O 7	-2	-2.04	0.98
O 8	-2	-1.92	1.04
O9	-2	-1.91	1.05
O10	-2	-2.02	0.99
O11	-2	-2.07	0.97
O12	-2	-1.98	1.01

intrinsic strains strong enough to involve instability of the crystal structure. In our case, we found a very good (GII) index converging to 0.03 v.u, which proves the high stability and rigidity of the structure.

Crystal Structure Description

The basic building units of $Zn_3Fe_4(PO_4)_6$ belonging to the Howardevansite are more or less distorted polyhedra (FeO₆, ZnO₆, ZnO₅) and nearly regular PO₄ tetrahedra, as depicted in Fig. 3. In the current structural model, all atoms are located in Wyckoff position 2i, except Zn(2) which is on Wyckoff position 1h of the $P\overline{1}$ space group. This structure consists of infinite kinked chains of edge-sharing [FeO₆] octahedra, linked into two kinds of dimers ($[Fe(1)_2O_{10}]$ and $[Fe(2)_2O_{10}]$) which are in turn interconnected via a common edge to $[Zn(1)O_5]$ group, as shown in Fig. 4. These chains are held together by the apices of the PO_4 tetrahedra giving rise to a stacked layers perpendicular to [010] (Fig. 5). Inside of the layers the junction of $Fe(2)O_6$ and $Zn(1)O_5$ polyhedra by common vertices of $P(2)O_4$ and $P(3)O_4$ tetrahedra built an open threedimensional framework that delimits tunnels parallel to [100], hosting the $Zn(2)^{2+}$ cations (Fig. 6).

As we have pointed out, the (CHARDI) method extends to the analysis of the geometric aspect of each hetero polyhedron building a crystal structure using the number (ECoN). A careful inspection of the ratios ECoN (Fe(1))/CN(Fe(1)) = 5.49/6 and ECoN (Fe(2))/CN(Fe(2)) =5.67/6 reveals an increased distortion in the FeO₆ units ranging from the Fe(2)O₆ to Fe(1)O₆ polyhedra. Indeed both Fe(1)O₆ and Fe(2)O₆ octahedral groups are slightly distorted and weakly compressed. This tendency to deviates from the regularity is due to the junction based on edge sharing within the [Fe₂O₁₀] dimers, which induces metal-

P1

P2

P3

5

5

5

4

4

4



Fig. 3. The principal building units in the structure of the title compound. Displacement ellipsoids are drawn at the 60% probability level. Symmetry codes: (i) x + 1, y, z; (ii) x, y - 1, z; (iii) -x + 2, -y + 1, -z + 2; (iv) -x + 1, -y + 1, -z + 1; (v) -x + 1, -y + 1, -z + 2; (vi) x - 1, y, z; (vii) -x + 1, -y + 2, -z + 2; (viii) x + 1, y + 1, z; (ix) -x + 1, -y + 2, -z + 2; (viii) x + 1, y + 1, z; (ix) -x + 1, -y + 2, -z + 1; (x) x - 1, y - 1, z; and (xi) x, y + 1, z.



Fig. 5. A view along [010], showing a layer resulting from the connection of chains via vertices of $[PO_4]$ units.

metal repulsion forces. Despite this electrostatic stress, the extremely large void space ($V_{\text{Fe}(1)} \simeq 10.50 \text{ Å}$ and $V_{\text{Fe}(2)} \simeq 10.61 \text{ Å}^3$) in the unit cell fully supports such deformations of both octahedral units. The interatomic distances Fe–O are ranging from 1.8896(9) to 2.1819(9) Å for Fe(1) Å and varying between 1.862 (1) and 2.0892 (9) Å for Fe(2). The corresponding average



Fig. 4. Edge-sharing [Zn(1)O5] polyhedral and $[Fe_2O_{10}]$ dimers forming an infinite zigzag chain running parallel to [101].



Fig. 6. Three-dimensional view of $Zn_3Fe_4(PO_4)_6$ crystal structure showing tunnels parallel to [100].

distances $\langle Fe(1)-O \rangle = 2.0141$ Å and $\langle Fe(2)-O \rangle = 2.0202$ Å are close to 2.03 Å, reported by Shannon for Fe³⁺ cations within octahedral coordination [45]. The first zinc atom is surrounded by five neighboring O atoms, resulting from four long Zn(1)-O $\simeq 2$ Å and a short one Zn(1)-O(5) (Zn(1)-O(5) = 1.958(1) Å), forming a highly deformed trigonal bipyramid. The

related distortion was evidenced trough the ECoN(Zn(1))/CN(Zn(1)) = 4.86/5 ratio as well as by measuring the degree of trigonality using the τ_5 index which reach 0.504 v.u. [46]. The octahedral coordination sphere of the Zn²⁺(2) cation is quite distorted as seen from the Zn(2)–O bond lengths in the range 2.0071(9)–2.2575(9) Å with a mean value of 2.1567 Å. The remaining P–O interatomic distances within the PO₄ tetrahedra units vary between 1.494(1) and 1.5605(9) Å. Their mean distances $\langle P(1)-O \rangle = 1.5400$ Å, $\langle P(2)-O \rangle = 1.5330$ Å, and $\langle P(3)-O \rangle = 1.5433$ Å, are in a good accordance with the $\langle P-O \rangle$ length often encountered in monophosphate groups [47].

ACKNOWLEDGMENTS

The authors thank the Unit of Support for Technical and Scientific Research (UATRS, CNRST) for the X-ray measurements.

FUNDING

Funding for this research was provided by Mohammed V University, Rabat, Morocco.

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