High Rate Performance of Nano-Structured LiFePO₄/C Cathode Material Prepared by a Polymer-Assisted Method from Inexpensive Iron(III) Raw Material

Wenjing Pu*a***,** *^b***, *, Wei Lu***a***, **, Zhipeng Chen***a***, Kai Xie***^c* **, and Chunman Zheng***^c*

aInstitutes of Physical Scicence and Information Technology, Anhui University, Hefei, 230601 China b Institute of Applied Physics, PLA Army Academy of Artillery and Air Defense, Hefei, 230031 China c Department of Material Science and Engineering, School of Aerospace Science and Engineering, National University of Defense Technology, Changsha, 410073 China

> **e-mail: wenjingpu@outlook.com **e-mail: weilu0818@ahu.edu.cn* Received April 22, 2019; revised July 2, 2019; accepted July 29, 2019

Abstract—A spherical carbon coated nano-structured $LiFePO₄$ composite is synthesized by a polymerassisted method from inexpensive iron(III) raw material. The synthesis process includes two steps: (1) nano-FePO4/polyaniline composites with core–shell structure are synthesized through the in-situ polymerization of aniline; (2) LiFePO $_4$ /C composites are prepared through carbothermal reduction with the nano-FePO4/polyaniline and sucrose as raw materials. The structure, surface morphology of the materials and the properties of the coated carbon are investigated by X-ray diffraction, Raman spectroscopy, scanning electron microscopy, and high-resolution transmission electron microscopy. The obtained nano-structured LiFePO $_4$ /carbon composite has a spherical morphology compose of ordered olivine structure, which is coated with 2 nm thick amorphous layer of carbon. At the same time, the materials are linked together by amorphous carbon from sucrose decomposition. The aniline plays an important role during the synthesis process. The electrochemical properties of the materials are tested by charge–discharge measurements. The obtained nano-structured LiFePO4/carbon composite shows excellent electrochemical properties, especially its high rate performance. It exhibits initial discharge capacities of 138, 136, 118, 103, and 92 mA h g^{-1} at 0.2, 1, 10, 20, and 30 C rate between 3.65 and 2.0 V, respectively. That makes it a promising cathode material for advanced power Li-ion batteries. The excellent electrochemical properties of the materials can be ascribed to the two different amorphous carbons. The carbon coated on the surface of LiFePO₄ effectively reduces interparticle agglomeration of the LiFePO₄ particles. The carbon interlinked between the composite improve the electronic conductivity. Those shorten the lithium ions diffusion length and improve the electric contact between $LiFePO₄$ particles.

Keywords: LiFePO₄, lithium-ion batteries, high rate, carbon coating **DOI:** 10.1134/S1023193520050092

INTRODUCTION

With the increasing concerns about environmental protection and energy saving, rechargeable lithiumion batteries have been extensively used in a wide variety of portable electric devices owing to their high energy densities, high voltage and portability [1–3]. In recent years, there has been a dramatic increase in research and commercialization activities [3–7]. However, these batteries still need to be improved before they can be used in special applications which need high energy and high power, such as portable power tools, hybrid electric vehicles and other power supplies [8–12]. Olivine-structured lithium iron phosphate ($LiFePO₄$) has many advantages compared with conventional cathode materials, such as $LiCoO₂$, $LiNiO₂$, and $LiMn₂O₄$. The insertion and extraction related to the $LiFePO₄$ electrode involves a two phase mechanism with the reversible reaction expressed as [13–15]:

$LiFePO₄ \leftrightarrow FePO₄ + Li⁺ + e.$

It shows a plateau at around 3.5 V relative to lithium metal and high theoretical discharge capacity of 170 mA h g^{-1} . And it has excellent structural and chemical stability during intercalation and thermal cycling. However, a main obstacle of $LiFePO₄$ is its poor electronic conductivity and low lithium-ion diffusion coefficient, which is the most unfavorable issue

for the rate capability of batteries. And these hinder the application of $LiFePO₄$ in high power field, such as electric vehicles.

As the drawbacks mentioned above, various methods have been adopted to improve the conductivity of electrons in the materials, such as nanonization $[8, 16-20]$, surface coating with carbon $[11, 19-25]$, conductive organic polymer, and $Li⁺$ site doping with Mg^{2+} , Mn^{3+} , Ce^{3+} , Zr^{4+} , Ti^{4+} , Ni^{2+} [12, 26]. In addition, some researches have focused on the electrochemical kinetics and mechanism studies of the lithium intercalation process of $LiFePO₄$ cathode materials, which makes the investigation more deep and extensive. Among these methods, coating nano-sized $LiFePO₄$ crystal with carbon can enhance conductivity and confine the crystal size of $LiFePO₄$ for solving the problem of high power performance [10, 11, 25, 27]. There are many literatures that reported the synthesis of LiFePO₄/C via various methods. However, these methods generally involve a high-temperature treatment to ensure the conductivity of the resulting carbon materials. But the increase of crystallite size is inevitable during the high temperature condition. And the carbon coated on the surface of the material is also incomplete, just like the description at the literature [19, 20, 22, 25, 28, 29].

In addition, to obtain phase-pure $LiFePO₄$, oxidation of Fe^{2+} to Fe^{3+} during the synthesis must be avoided. Since Fe^{2+} can be easily oxidized during the prepare process, it is typically hard to obtain phasepure LiFePO₄ by applying Fe²⁺ raw material. And these salts are much more expensive than $Fe³⁺$ salts. According to [30], H_2 and CO can reduce the iron to a lower oxidation state during the prepare process. This suggests that much cheaper iron (III) material may be used directly for the synthesis of $LiFePO₄$. In 2015, R. Sehrawat and A. Sil have developed a polymer gel combustion method, which use different amount of aniline as the monomer of gel formation. The final particle size of the $LiFePO₄/C$ depends on the initial monomer content used in the synthesis process. Prepared LiFePO₄/C cathode material performs a higher rate capability, due to small particle size, low charge transfer resistance, and higher $Li⁺$ diffusion coefficient [31].

In this work, a spherical carbon coated nano-structured $LiFePO₄$ cathode composite is synthesized by a polymer-assisted method from inexpensive iron(III) raw material. The synthesis process of the composite includes two steps, in which the aniline plays an important role. There are two different amorphous carbons in the as-synthesized LiFePO $_4$ /C composite, which greatly improve the electronic conductivity of the materials. The composite showed a high rate capacity (about 118 mA h g^{-1}) at 10 C rate and excellent cycling performance. That made it a promising cathode

material for advanced electrochemical devices such as power Li-ion battery and supercapacitor.

EXPERIMENTAL

Synthesis of FePO4/*PANI and LiFePO4*/*C Cathode Material*

 $NH_4H_2PO_4$ (Aldrich, 99%) and $(NH_4)_2HPO_4$ (Aldrich, 99%) with molar ratio of 1 : 1 were dissolved in deionized water. Then the aniline (Aldrich, 99.5%) and a little mount of tween-60 were added to the solution, marked as Solution A. The $Fe(NO₃)₃$ (Aldrich, 99.9%) in equimolar amount with $[PO₄]^{3-}$, were dissolved in deionized water, marked as Solution B. Then Solution B was slowly added to Solution A with stirring. Saturated $NH₄ HCO₃$ (Sinopharm Group Co. LTD., 99%) solution was used as pH regulator. The reactive mixture was then stirred for 4 h at room temperature. And the resulting $FePO₄/PANI$ was filtered and washed several times with deionized water.

An equimolar mixture of $CH_3COOLi \cdot 2H_2O$ (Alfa Aesar, 99.9%) and FePO₄/PANI, with a contain amount of sucrose (Sigma-Aldrich, 99.5%) were dispersed in the ethanol as the precursor. After milling for about 1 h, the precursors were heated at 400°C for 4 h under Ar containing 5% H₂ (Jingxiang, 99.999%) in order to decompose the organic composition. The precursors were then remilled for about 1 h and finally calcined at 700°C for 10 h under argon containing 5% $H₂$ to obtain the composite.

Structural Characterization

The DSC–TGA of the precursor was performed by thermal gravimetric analysis (TA SDT Q600, American) under the mixture atmosphere $(Ar : H_2 = 95 : 5)$. Raman spectrum of the samples was performed with Raman Spectrometers (Bruker RAM, Germany).The crystal structure of the obtained $LiFePO₄/C$ powder was identified by X-ray diffraction (XRD) (D/Max-2500, Japan) using Cu K_{α} radiation scanned in the range 20°–45°. The particle size and morphology were observed by a field emission scanning electron microscopy (FESEM) (S-4800, Hitachi, Japan). The nature and thickness of the coated carbon was measured using the images from high resolution transmission electron microscope (HRTEM) (JEM-2100, JEOL, Japan).

Electrochemical Measurements

The electrochemical properties of the products were investigated using a two-electrode test cell with lithium foil as the negative. A positive electrode was made by coating a paste of the nano-structured $LiFePO₄$ composite, acetylene black and polyvinylidene fluoride (PVdF) binder (80 : 10 : 10 wt %) on an alumi-

Fig. 1. The DSC–TGA curve of the precursor under the mixture atmosphere (Ar : $H_2 = 95 : 5$).

num-foil collector. The positive film was subjected to roll press and the electrodes of 10 mm diameter were punched out. The positive electrodes were dried at 120°C for 12 h in a vacuum oven. For comparison, a commercial $LiFePO₄$ powder with micron size (Marked as sample A), provided by Hao Run Technology Co. Ltd., is also used to prepare positive electrode with the same surface density as the synthesized LiFe PO_4/CC composite. The cells were assembled in an argon filled glove box with an electrolyte of 1 mol L^{-1} LiPF₆ in DC–DMC (1 : 1, volume ratio) solution and a separator of Celgard 2400. The charge– discharge measurements were performed using a multi-channel battery tester (LAND CT2001A, China) operating in galvanostatic mode at various rates, between 2.0 and 3.85 V.

RESULTS AND DISCUSSION

Structure, Morphology and Physical Property

Figure 1 shows the DSC–TGA curve of the precursor under the mixture atmosphere $(Ar: H₂ = 95 : 5)$. It can be seen that the weight loss is about 12 wt % from room temperature to 120°C because of the evaporation of water. Accompanied with the decalescence, there is a weight loss of 29 wt % from 120 to 150°C, which is caused by the pyrolysis of sucrose and polyaniline. The Fe³⁺ is reduced to Fe²⁺ and forms the LiFePO₄ with a weight loss of 26 wt % from 360 to 450°C. At the same time, there appears an exothermic peak in the DSC curve during 400 to 450°C. It indicates that the LiFeP O_4 can be formed when the temperature is higher than 450°C. The XRD patterns shows that the element iron exist mainly as the form of $FePO₄$, $Fe_7(PO_4)_6$, $Fe_3(PO_4)_2$ and a small quantity exist as the form of LiFePO₄. The crystalline LiFePO₄ is formed with the increase of the temperature. The carbon,

Fig. 2. XRD patterns of LiFePO₄ powder and nano-structured $LiFePO₄/C$ composite.

which forms because of the pyrolysis of sucrose and polyaniline, has two effects during the process. It can reduce Fe^{3+} into Fe^{2+} . At the same time, it can restrain the increase of the $LiFePO₄$ crystal.

Figure 2 shows XRD patterns of commercial sample A and nano-structured LiFe PO_4/CC composite. It can be found clearly that there are four strong diffraction peaks at $2\theta = 20.68^{\circ}$ (101), 25.50° (111), 29.69° (211), and 35.54° (311) in the XRD patterns of sample A and nano-structured LiFe PO_4/C composite. And there appears another weak peaks at $2\theta = 22.63^{\circ}$, 23.99°, 32.18°, 36.46°, 37.83°, 39.24°, and 42.16°. All the peaks are consistent with the standard $LiFePO₄$ (JCPDS 81-1173), which indicates the presence of the $LiFePO₄$ phase as an ordered olivine structure. According to the Scherrer's equation, the crystal size of the nano-structured LiFePO₄/C composite is about 50.2 nm, indicating the samples have a high crystal degree. At the same time, there is no diffraction peaks attributed to carbon, most likely because there may be amorphous or a low content of crystalline carbon in the nano-structured $LiFePO₄/C$ composite.

Figure 3 shows the Raman spectrum of the nanostructured $LiFePO₄/C$ composite. Raman spectroscopy is an important method for the investigation of the properties of the carbon. The typical characteristic of carbon in the Raman spectrum are the two broad bands at 1350 and 1582 cm^{-1} , which are called as D-band (disorder/defect band) and G-band (graphitic band), respectively. These two bands are observed in Raman spectrum of the nano-structured $LiFePO₄/C$ composite. It indicates that the carbon is composed of amorphous phase and a little amount of graphitic carbon. The electronic conductivity of the composite has a complicated correlation with the peak intensity ratio and the bands are a ratio of the D vs. G band. The ration of I_D/I_G of the synthesized powder is 0.92, indicating that the coated carbon might be amorphous phase. The result is consistent with that of

Fig. 3. Raman spectrum of nano-structured LiFe PO_4/C composite.

XRD analysis, which shows that no diffraction lines caused by graphite have been found in XRD patterns of the nano-structured $LiFePO₄/C$ composite.

The surface morphology of the precursor particles and the LiFePO₄/C sample as well as the shape of the carbon coating is investigated by SEM and HRTEM, as shown in Fig. 4. Figure 4a shows that the precursor powder FePO4/PANI has a spherical morphology and its diameter is about 30 nm. As shown in Fig. 4b, the diameter of nano-structured LiFePO₄/C composite is about 55 nm. In contrast to the precursor $FePO₄/PANI$, the diameter of nano-structured $LiFePO₄/C$ composite becomes bigger. The HRTEM images of Fig. 4c show that the $FePO₄$ is amorphous and there is no clear interface between the $FePO₄$ and PANI. In Fig. 4d, it is clearly founded that regular crystal plane of $LiFePO₄$ particle and coating layer with little crystallinity. It shows that the thickness of the coated carbon layer is about 2 nm. These findings combined both XRD and Raman spectra can help us to understand these grains have a core-shell structure with $LiFePO₄$ crystallite as the core and carbon coating as the shell. The LiFePO₄ grains are sphere-like and surrounded completely by carbon layers. The resultant $LiFePO₄/C$ composites show an enhanced rate property due to the improve-

Fig. 4. SEM images of (a) the precursor particles and (b) the nano-structured LiFePO₄/C composite, HRTEM images of (c) the precursor powder $FePO_4/PANI$, and (d) the Li $FePO_4$ particle and carbon layer.

RUSSIAN JOURNAL OF ELECTROCHEMISTRY Vol. 56 No. 8 2020

Fig. 5. Cyclic voltametry test of nano-structured LiFePO₄/C composite at different scanning rate.

ment of electronic conductivity from the effective carbon coating.

The EDX analysis shows that the content of the carbon is different in the different region of the nanostructured $LiFePO₄/C$ composite. As shown in Fig. 4b, the carbon content of in the dot *b* region is much higher than that in the dot *a* region. The ratio is about 4 : 1, most likely because the carbon in the dot *b* region come from the decomposition of sucrose and the carbon in the dot *a* region come from the decomposition of PANI. These results can help us to understand the carbon distribution in nano-structured $LiFePO₄$ and the reason why the composites having excellent rate properties.

Electrochemical Performance

In order to evaluate the potential application of the nano-structured LiFe PO_4/C composite as cathode materials for lithium-ion batteries, the electrochemi-

Table 1. Discharge capacities and the discharge capacity retention s at various values of $Q_{nC}/Q_{0.2C}$ for nano-structured LiFePO₄/C composite at different rates

Discharge rate	Discharge capacity, $mA h g^{-1}$	$Q_{nC}/Q_{0.2C},$ %
0.2C	138	100
1 C	136	98.6
10 C	118	85.5
20C	103	74.6
30 _C	92	66.7

Fig. 6. The 4th charge–discharge curves of nano-structured $LiFePO₄/C$ composite at different rates.

cal performances with respect to Li insertion/extraction have been investigated. Figure 5 shows the cyclic voltametry test of nano-structured $LiFePO₄/C$ composite at different scanning rate. It is can be seen that a couple of redox peaks are observed between 3.3 and 3.5 V (vs. Li/Li^{+}) in the cyclic voltametry curve obtained at a scanning rate of 0.1 mV s^{-1} . These peaks correspond to the extraction and insertion of lithium ions. These redox peaks can still be clearly observed at a scanning rate of 0.9 mV s^{-1} . It indicates that the composite has good high power performance. The result is consistent with that of the two-electrode test cell as shown in Fig. 6 and Table 1.

To test the working voltage and rate properties, the $LiFePO₄/C$ electrodes have been discharged at different rates, as shown in Fig. 6. The electrode is charged at 0.2 C for each charging step and then discharged at gradually increased rates. At low discharge rate (0.2, 1 C), the nano-structured LiFe PO_4/CC composites has the similar charge–discharge curves with close discharge capacity (138.0, 142.5 mA h g^{-1}) and flat plateaus. And the electrode exhibits flat voltage plateaus at each C-rate. Even at a 20 C rate, the discharge voltage plateau is still higher than 3.0 V (vs. Li⁺/Li), and the delivered capacity is more than 103 mA h g^{-1} . Moreover, the charge–discharge plateau voltage differences are 0.1, 0.15, 0.5 and 0.7 V at 0.2, 1.0, 10, and 20 C-rate, respectively. The results indicate that the nano-structured LiFe PO_4/C electrode has low polarization, good electrical conductivity, as well as good rate capability.

The retention of discharge capacity ($Q_{nC}/Q_{0.2C}$, *n* = 0.2, 1, 10, 20, 30) for LiFePO₄/C at different discharge currents are listed on Table 1. It can be seen that the nano-structured LiFe PO_4/C composites has good rate capability. The discharge capacity retention is

Fig. 7. Cycling performance of the nano-structured $LiFePO₄/C$ composite at different rates compared to a commercial sample A.

85.5, 74.6, 66.7% at 10, 20, and 30 C-rate respectively, while the discharge capacity is 118, 103 and 92 mA h g^{-1} . From the results, it could be concluded that the nanostructured $LiFePO₄/C$ composites present better rate performance, which can be attributed to the structure of the LiFePO₄/C composites.

Figure 7 shows the discharge capacity at various rates (0.2, 1.0, 10, 20, 30, and 40 C-rate) as a function of cycle number for the nano-structured LiFe PO_4/C composites and a commercial sample A. At 0.2 C rate, the nano-structured $LiFePO₄/C$ composites and sample A reached a reversible capacity of 138 and 158 mA h g^{-1} , respectively. After 50 cycles at various

rates, the reversible capacity of the two materials maintained at about 138 and 158 mA h g^{-1} , which indicated a favorable reversible capacity and cycling stability retention even under high-rate discharge conditions. But from the figure, it can be seen that the reversible capacity of the nano-structured $LiFePO₄/C$ composites is higher than that of sample A at high discharge rate. It can be seen the nano-structured $LiFePO₄/C$ composites has good rate-cycle performance.

This excellent high rate performance of the nanostructured LiFe PO_4/CC composite is ascribed to the carbon coated on the surface of the $LiFePO₄$ and the carbon interlinked between the particles, as shown in Fig. 8. According to the literature, although the conductive carbon is added to the electrode film during the preparation process, it cannot form a convenient conductive network because of dispersion reasons. Thus, during the intercalation process of $LiFePO₄$, the electrons cannot reach all the positions where Li⁺ ion intercalation takes place, resulting in polarization of the electrode, especially at high rate circumstance. In the composite prepared by a polymer-assisted method, the $LiFePO₄$ is completely coated with conductive carbon that comes from PANI. Lithium ions can easily intercalate into the $LiFePO₄$ through the carbon shell (about 2 nm). Moreover, the completely carbon coated-LiFe $PO₄$ is interlinked by the carbon that comes from sucrose, which form a convenient conductive network in the composite. When the $Li⁺$ ion intercalation takes place, the electrons can reach from all the positions with the conductive network. Thus it could further alleviate this polarization phenomenon and present excellent high rate performance.

Fig. 8. (a) HRTEM images of the carbon coated LiFePO₄ particles and (b) electron-transfer pathway for LiFePO₄ particles with nano-size and two type of carbon.

RUSSIAN JOURNAL OF ELECTROCHEMISTRY Vol. 56 No. 8 2020

CONCLUSIONS

In summary, a spherical carbon coated nanostructured $LiFePO₄$ composite is synthesized by a polymer-assisted method from inexpensive iron(III) raw material. The cathode material shows a discharge capacity of 138, 136, 118, 103 and 92 mA h g^{-1} at 0.2, 1, 10, 20, and 30 C-rates. The nano-structured LiFePO₄ composite provides an excellent rate performance and a good cycling stability at high rates. This excellent behavior of the nano-structured $LiFePO₄$ composite is ascribed to the carbon coated on the surface of the $LiFePO₄$ and the carbon interlinked between the particles. Such a synthesis using Fe(III) cheap raw material in combination with sucrose as an organic carbon source is a facile and energy-saving way for the synthesis of high-rate performance $LiFePO₄/C$ composite for high power lithium-ion batteries.

FUNDING

This work is supported by the Natural Science Foundation of Anhui Province with Grants no. 1708085QB32, the Natural Science Foundation of China with Grants nos. 51576208, 11505290, the National Megnitic Confinement Fusion Science Program of China (no. 2018YFE0310400) and the National Key R&D Program of China (2017YFE 0300603).

CONFLICT OF INTEREST

The authors declare that they have no conflict of interest.

REFERENCES

- 1. Chen, S.e, Zheng, J., Mei, D., Han, K.S., Engelhard, M.H., Zhao, W., Xu, W., Liu, J., and Zhang, J., High-Voltage Lithium-Metal Batteries Enabled by Localized High-Concentration Electrolytes, *Adv. Mater.*, 2018, vol. 30(21), p. 1706102.
- 2. Liu, Y., Xie, K., Pan, Y., Li, Y., Wang, H., Lu, W., and Zheng, C., LiPON as a protective layer on graphite anode to extend the storage life of Li-ion battery at elevated temperature, *Ionics*, 2018, vol. 24, p. 723.
- 3. Lu, W., Xiong, S., Pu, W., Xie, K., and Zheng, C., Carbonate-Grafted Polysilane as a New Additive for Elevated-Temperature Lithium-Ion Batteries, *ChemElectroChem*, 2017, vol. 4, p. 2012.
- 4. Lu, W., Xie, K., Chen, Z., Pan, Y., and Zheng, C., Preparation and characterization of trifluoroethyl aliphatic carboxylates as co-solvents for the carbonatebased electrolyte of lithium-ion batteries, *J. Fluorine Chem.*, 2014, vol. 161, p. 110.
- 5. Lu, W., Xiong, S., Xie, K., Pan, Y., and Zheng, C., Identification of solid electrolyte interphase formed on graphite electrode cycled in trifluoroethyl aliphatic carboxylate-based electrolytes for low-temperature lithium-ion batteries, *Ionics*, 2016, vol. 22(11), p. 2095.
- 6. Lu, W., Xie, K., Pan, Y., Chen, Z., and Zheng, C., Effects of carbon-chain length of trifluoroacetate co-sol-

vents for lithium-ion battery electrolytes using at low temperature, *J. Fluorine Chem.*, 2013, vol. 156, p. 136.

- 7. Lu, W., Xie, K., Chen, Z., Xiong, S., Pa,n Y., and Zhen,g C., A new co-solvent for wide temperature lithium ion battery electrolytes: 2,2,2-Trifluoroethyl n-caproate, *J. Power sources*, 2015, vol. 274, p. 676.
- 8. Chen, Y., Xiang, K., Zhou, W., Zhu, Y., Bai, N., and Chen, H., LiFe PO_4/C ultra-thin nano-flakes with ultra-high rate capability and ultra-long cycling life for lithium ion batteries, *J. Alloys Compounds*, 2018, vol. 749, p. 1063.
- 9. Tsuda, T., Ando, N., Matsubara, K., Tanabe, T., Itagaki, K., Soma, N., Nakamura, S., Hayashi, N., Gunji, T., Ohsaka, T., and Matsumoto, F., Improvement of high-rate charging/discharging performance of a lithium ion battery composed of laminated $LiFePO₄$ cathodes/graphite anodes having porous electrode structures fabricated with a pico-second pulsed laser, *Electrochim. Acta*, 2018, vol. 291, p. 267.
- 10. Feng, J. and Wang, Y., High-rate and ultralong cyclelife $LiFePO₄$ nanocrystals coated by boron-doped carbon as positive electrode for lithium-ion batteries, *Appl. Surf. Sci.*, 2016, vol. 390, p. 481.
- 11. Guo, H. and Gao, Q., High-performance LiFe PO_4/CP nanocomposites prepared from a micro-reactor based on an unusual water–oil system, *RSC Advances*, 2013, vol. 3, p. 7245.
- 12. Kim, S., Mathew, V., Kang, J, Gim, J., Song, J., Jo, J., and Kim, J., High rate capability of $LiFePO₄$ cathodes doped with a high amount of Ti, *Ceramics Int.*, 2016, vol. 42, p. 7230.
- 13. Takahashi, I., Mori, T., Yoshinari, T., Orikasa, Y., Koyama, Y., Murayama, H., Fukuda, K., Hatano, M., Arai, H., Uchimoto, Y., and Terai, T., Irreversible phase transition between $LiFePO₄$ and $FePO₄$ during high-rate charge–discharge reaction by operando X-ray diffraction, *J. Power Sources*, 2016, vol. 309, p. 122.
- 14. Kuei-Feng Hsua, B. S. H. B. Tsay, S., Chou, T., Sheu, H., Lee, J., and Hwang, B., Formation mechanism of LiFe PO_4/CC composite powders investigated by X-ray absorption spectroscopy, *J. Power Sources*, 2009, vol. 192, p. 660.
- 15. Yi, X., Zhang, F., Zhang, B., Yu, W., Dai, Q., Hu, S., He, W., Tong, H., Zheng, J., and Liao, J., (010) facets dominated $LiFePO₄$ nano-flakes confined in 3D porous graphene network as a high-performance Li-ion battery cathode, *Ceram. Internat.*, 2018, vol. 44, p. 18181.
- 16. Shang, H., Chu, W., Cheng, J., Pan, F., Cheng, D., Xia, D., Wang, W., and Wu, Z., Surface Phase Composition of Nanosized LiFePO₄ and Their Enhanced Electrochemical Properties, *J. Materi. Chem. A*, 2013.
- 17. Zhang, L. and Liang, H., Rapid Synthesis of LiFePO₄ Nanoparticles by Microwave-Assisted Hydrothermal Method, *Russ. J. Electrochem.*, 2013, vol. 49(5), p. 492.
- 18. Liu, J., Zhang, X., Wang, R., and Zhang, J., Facile Synthesis of $LiFePO₄$ Nanoparticles Coated by Few Layers of PAS with Quasi-Graphene Structure, *Int. J. Electrochem. Sci.*, 2012, vol. 7, p. 12983.
- 19. Chen, M., Kou, K., Tu, M., Hu, J., Du, X., and Yang, B., Conducting reduced graphene oxide wrapped $LiFePO₄/C$

nanocrystal as cathode material for high-rate lithium secondary batteries, *Solid State Ionics*, 2017, vol. 310, p. 95.

- 20. Xie, G., Zhu, H., Liu, X., and Yang, H., A Core–shell LiFePO₄/C nanocomposite prepared via a sol–gel method assisted by citric acid, *J. Alloys Compounds*, 2013.
- 21. Hu, Z., Yang, D., Yin, K., Liu, J., Li, F., Gao, W., Qin, Y., and Liu, H., The effect of Lithium source on the electrochemical performance of $LiFePO₄/C$ cathode materials synthesized by Sol–gel method, *Advanced Mater Res*, 2013, vol. 669, p. 311.
- 22. Cech, O., Thomas, J.E., Sedlarikova, M., Fedorkova, A., Vondrak, J., Moreno, M.S., and Visintin, A., Performance Improvement on LiFePO₄/C Composite Cathode for Lithium-ion Batteries, *Solid State Sci.*, 2013.
- 23. Li, X., Shao, Z., Liu, K., Zhao, Q., Liu, G., and Xu, B., Influence of Li:Fe molar ratio on the performance of the $LiFePO_4/C$ prepared by high temperature ball milling method, *J. Electroanal. Chem.*, 2017, vol. 801, p. 368.
- 24. Shangguan, E., Fu, S., Wu, S., Wan, Q., Wu, C., Li, J., Cai, X., Chang, Z., Wang, Z., Li, Q., and Jiang, K., Evolution of spent $LiFePO₄$ powders into LiFePO₄/C/FeS composites: A facile and smart approach to make sustainable anodes for alkaline Ni–Fe secondary batteries, *J. Power Sources*, 2018, vol. 403, p. 38.
- 25. Zhou, N., Uchaker, E., Liu, Y., Liu, S., Liu, Y., and Cao, G., Effect of Carbon Content on Electrochemical

Performance of LiFePO $_4$ /C Thin Film Cathodes, *Int. J. Electrochem. Sci.*, 2012, vol. 7, p. 12633.

- 26. Zhao, N., Li, Y., Zhi, X., Wang, L., Zhao, X., Wang, Y., and Liang, \dot{G} , Effect of Ce^{3+} doping on the properties of LiFePO4 cathode material, *J. Rare Earths*, 2016, vol. 34(2), p. 174.
- 27. Zhao, C, Wang, L., Chen, J., and Gao, M., Environmentally benign and scalable synthesis of $LiFePO₄$ nanoplates with high capacity and excellent rate cycling performance for lithium ion batteries, *Electrochim. Acta*, 2017, vol. 255, p. 266.
- 28. Li, X., Shao, Z., Liu, K., Zhao, Q., Liu, G., and Xu, B., A facile ultrasound assisted high temperature ball milling synthesis of LiFePO₄/graphene with enhanced electrochemical performance, *Int. J. Hydrogen Energy*, 2018, vol. 43, p. 18773.
- 29. Wang, Q., Peng, D., Chen, Y., Xia, X., Liu, H., He, Y., and Ma, Q., A facile surfactant-assisted self-assembly of LiFe PO_4 /graphene composites with improved rate performance for lithium ion batteries, *J. Electroanal. Chem.*, 2018, vol. 818, p. 68.
- 30. Wang, Y., Wang, Y., Hosono, E., Wang, K., and Zhou, H., The Design of a LiFePO $_4$ /Carbon Nanocomposite With a Core–Shell Structure and Its Synthesis by an In Situ Polymerization Restriction Method, *Angew. Chem.*, 2008, vol. 47, p. 7461.
- 31. Sehrawat, R. and Sil, A., Polymer gel combustion synthesis of $LiFePO₄/C$ composite as cathode material for Li-ion battery, *Ionics*, 2015, vol. 21(3), p. 673.