Preparation of Nanostructured Tin(IV) Oxide and Supported Platinum Electrocatalysts Based on It

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Abstract—We have synthesized tin dioxide with a large specific surface area ($122 \text{ m}^2/\text{g}$), prepared platinum materials supported on it, and characterized them. The results demonstrate that $Pt/SnO_2 + C$ composites can be used as catalysts for oxygen electroreduction. The stability of such a catalyst containing 50% carbon and 15 wt % Pt is better than that of commercially available Pt/C electrocatalyst containing 20% Pt.

Keywords: tin(IV) oxide, platinum nanoparticles, electrocatalyst, oxygen electroreduction reaction, noncarbon support

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INTRODUCTION

The engineering of environmentally safe chemical energy sources, such as fuel cells, is an important issue among modern research directions [1-4]. Efficient electrocatalysts for low-temperature fuel cells (FCs) include nanostructured composite materials containing platinum-based nanoparticles supported on nanoor microparticles of a carbon support [5-10]. A key issue pertaining to polymer membrane FCs is their limited lifetime, primarily because of the degradation of the membrane and catalytic layer [11-16]. Therefore, the ability to optimize the support for platinum nanoparticles is an important issue [17]. The use of some types of carbon supports more oxidation-resistant than pyrolytic carbon black, for example, carbon nanotubes, makes it possible to extend the life of membrane-electrode units [18, 19], thereby raising their efficiency. At the same time, the nature of the support material, capable of being oxidized, remains unchanged.

In recent years, a great deal of attention has been paid to degradation-resistant noncarbon catalyst supports. The use of various methods for the synthesis of inorganic materials [20, 21] makes it possible to considerably extend the range of systems suitable for practical application and stimulates a search for new approaches to modifying existing materials.

Tin dioxide is a promising support for Pt-containing electrocatalysts owing to a number of its advantages (stability, availability, simple synthesis, and others). The development of relatively simple methods for the synthesis of SnO_2 supports and SnO_2 -based platinum electrocatalysts for polymer membrane Cs is a topical issue. In particular, in a study of the effects of solution pH and synthesis time (at a temperature of 190°C) on the size of SnO₂ nanoparticles Jiang et al. [22] suggested a possibility of obtaining crystallites one to tens of nanometers in size, without however BET surface area measurements. Interesting results were reported by Kuriganova and Smirnova [23] and Kuriganova et al. [23] about the electrochemical synthesis of Pt/SnO_x-C and Pt/MO_x-C materials. Unfortunately, despite the possibility of controlling the crystallite size of SnO_2 and the relative simplicity of the method, they reported a number of difficulties in controlling the chemical composition of SnO_2 [23]. Besides, the characteristics of the supports were not described in sufficient detail [24]. Antoniassi et al. [25] studied an electrocatalyst of complex composition, consisting of Pt, SnO₂, and carbon. They heralded the possibility of synthesizing a $Pt/SnO_2 + C$ composite with cubic platinum nanoparticles without using organic additives as stabilizers. Note that they paid most attention to ways of increasing the fraction of the (100) faces of Pt nanoparticles, without describing in detail SnO₂ synthesis conditions.

Interesting results were reported by Zhang et al. [26], who obtained SnO_2 -supported platinum electrocatalysts offering enhanced stability in comparison with commercially available Pt/C materials. Results reported by Frolova and Dobrovolsky [27] and Frolova et al. [28] are also of obvious interest because they suggest the possibility of further improving functional characteristics of the Pt/(Sn,Sb)O_x materials obtained. Note that, in the above-mentioned studies [26-28], SnO₂ was prepared from SnCl₄ using rather complex sol–gel processing. Elezović et al. [29] studied Sb–SnO₂ and Ru–SnO₂ materials and related electrocatalysts, and Gutsche et al. [30] examined the effect of Ag⁺ impurities on the growth of platinum nanoparticles on the surface of SnO₂ and found a direct relation: the fraction of faceted Pt nanoparticles was shown to increase with increasing Ag⁺ concentration.

Ruiz-Camacho et al. [31] studied $Pt-Ag/SnO_2-C$ composites as electrocatalysts for methanol oxidation, which is rather interesting from the viewpoint of enhancing the catalytic activity of Pt owing to the presence of SnO_2 . At the same time, it is clear that the addition of other metals to Pt/SnO_2 increases the probability of contaminating the polymer components of the membrane-electrode unit with metal cations.

On the whole, the previously reported results on the behavior of Pt/SnO₂ electrocatalysts demonstrate that SnO₂ nanoparticles have a positive effect on their electrochemical performance and stability compared to Pt/C. At the same time, the problem of producing relatively simple, well-reproducible and scalable methods for the synthesis of supports and impurityfree Pt/SnO₂ catalysts capable of ensuring stable functional characteristics remains to be resolved. Another important issue is the ability to enhance the mass activity of such catalysts. The composition of Pt/SnO_2 composites with carbon remains to be optimized, and it is not yet clear whether or not carbon has a negative effect on the stability of electrocatalysts. Given that a porous catalytic layer should be produced, it is necessary to continue a search for optimal compositions of $Pt/(SnO_2 + C)$ composites capable of ensuring a combination of effective electron transport and contact of Pt nanoparticles with an ionomer and reactant molecules during the operation of the electrocatalyst.

The objectives of this work were to obtain highly dispersed SnO_2 with a particle size in the nanometer range and, hence, a large surface area (at least 100 m²/g) and to use it for producing a stable Pt/SnO₂ electrocatalyst for the electroreduction of oxygen. Given the low electronic conductivity of tin dioxide, it was important to study the electrochemical behavior of composite electrocatalysts based on mixtures of Pt/SnO₂ with fine-particle carbon and assess their catalytic activity and stability in comparison with Pt/SnO₂ and Pt/C catalysts.

EXPERIMENTAL

The support was synthesized using tin(IV) chloride dihydrate (SnCl₂ · 2H₂O), microcrystalline urea (NH₂CONH₂), saturated aqueous ammonia (NH₄OH, 35 wt %), and a concentrated aqueous hydrogen peroxide solution (H₂O₂, 30 wt %, pure grade). To synthesize sample 1 (SnO₂), to a 10% SnCl₂ solution was added with constant stirring an excess of a NH₃ solution (to adjust the solution to pH 12) and then a tenfold excess of a H₂O₂ solution. After that, stirring was continued for an hour at a temperature of 60°C. To prepare sample 2, an aqueous 10% SnCl₂ solution was mixed with a tenfold excess of urea (under neutral conditions) in a round-bottomed refluxing flask, and the resultant suspension was then heated at 90°C in air on a water bath with constant stirring (15 h). The precipitate was repeatedly washed and separated by centrifugation (10 000 rpm, 5 min). In both cases, the product (SnO₂) yield was 98–100%. Pt-containing materials were prepared via the chemical reduction of Pt(IV) in a SnO₂ suspension. To prepare the suspension, we used samples 1 and 2 obtained in the preceding step, which were dispersed in a water + ethylene glycol mixture. Next, an appropriate volume of an aqueous solution of chloroplatinic acid ($H_2PtCl_6 \cdot 6H_2O_1$) Aurat, Russia) was added and the suspension was adjusted to pH 10 with aqueous ammonia (35 wt %). After that, a fourfold excess of a sodium borohydride $(NaBH_4)$ solution was added with vigorous stirring to the suspension, and the solid phase was then separated by centrifugation. The resultant samples were washed repeatedly (eight to ten times) with double-distilled water and ethanol. The product was dried to constant weight over P_2O_5 . The Pt-containing materials thus prepared will hereafter be denoted as IPt and 2Pt. $Pt/SnO_2 + C$ composites were prepared by adding Vulcan XC72 carbon black to Pt/SnO₂ in the process of preparing "electrochemical ink"-a catalyst suspension to be applied to an electrode [32].

X-ray diffraction patterns were collected on an ARL X'TRA automatic diffractometer with CuK_{α} radiation ($\lambda_{av} = 1.5418$ Å). The average crystallite size D_{av} was evaluated as described elsewhere [33]. The specific surface area and pore size distribution of the samples were determined using an ASAP 2020 surface area and porosimetry system (Micromeritics, USA). In data processing, we used BET analysis [34].

The samples were examined by transmission electron microscopy (TEM) on a FEI Tecnai G2 F20 S-TWIN TMP analytical (scanning) transmission electron microscope equipped with an EDAX energy dispersive X-ray spectrometer system.

The electrochemically active surface area (*S*), activity, and stability of the electrocatalysts were assessed in a standard three-electrode cell using a Pine AFCBP1 bipotentiostat system (Pine Research Instrumentation, USA). The technique used to apply a catalyst layer to electrodes and the electrochemical measurement procedure were similar to those described previously [32].

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Table 1. Structural characteristics of the synthesized SnO₂ materials

| Sample | 1 | 2 |
|--|----------------------|----------------------|
| Space group | P4 ₂ /mnm | P4 ₂ /mnm |
| Number of reflections indexed | 9 | 5 |
| a, Å | 4.757(2) | 4.736(9) |
| <i>c</i> , Å | 3.182(3) | 3.169(4) |
| Full width at half maximum (FWHM) of the peak, deg | 2.2 | 4.4 |
| Average crystallite size D_{av} , nm | 3.9 ± 0.1 | 2.8 ± 0.1 |

Table 2. Some characteristics of the Pt/SnO_2 and $Pt/SnO_2 + C$ materials

| Sample | 1Pt | 1PtC50 | <i>2</i> Pt | 2PtC5 | 2PtC15 | 2PtC25 | 2PtC33 | 2PtC50 | JM20 |
|--------------------------------|----------------------|----------------------|----------------------|----------------------|----------------------|----------------------|----------------------|----------------------|---------------|
| Support | SnO ₂ , 1 | SnO ₂ , 1 | SnO ₂ , 2 | Vulcan-72 |
| wt % Pt in Pt/SnO ₂ | 30 | 15 | 30 | 28.5 | 25.5 | 22.5 | 20.1 | 15 | 20 |
| wt % C in $Pt/SnO_2 + C$ | — | 50 | — | 5 | 15 | 25 | 33 | 50 | 80 |
| D _{av} (Pt), nm | 5.1 ± 0.2 | 5.1 ± 0.2 | 7.3 ± 0.3 | 7.3 ± 0.3 | 7.3 ± 0.3 | 7.3 ± 0.3 | 7.3 ± 0.3 | 7.3 ± 0.3 | 2.0 ± 0.1 |
| $S, m^2/g Pt$ | 7 ± 0.7 | 30 ± 3 | 17 ± 2 | 24 ± 3 | 34 ± 5 | 42 ± 5 | 47 ± 6 | 47 ± 6 | 94 ± 7 |
| Stability, % | — | 95 | — | — | _ | — | _ | 90 | 37 |

RESULTS AND DISCUSSION

In preparing SnO₂ from SnCl₂, ammonia or urea was used as a precipitant. The oxidant used was H₂O₂ in the former case and atmospheric oxygen in the latter. The X-ray diffraction patterns of the materials obtained show reflections characteristic of the rutile structure, without reflections attributable to impurity phases (Fig. 1, Table 1). The observed diffraction line broadening is caused by the small crystallite size (D_{av}). We obtained two materials, similar in chemical composition (SnO₂) but differing in average crystallite size (Table 1).

Pt nanoparticles were applied to the surface of SnO_2 particles by a technique capable of ensuring reproducible nanoparticle size distributions and shapes.

Given the low electronic conductivity of SnO₂, platinum electrocatalysts were produced by adding Vulcan XC72 carbon black to Pt/SnO₂. The amount of the carbon black was such as to ensure electronic conductivity of the composite as a whole, as well as electron supply to and removal from the Pt nanoparticles. The electrocatalysts produced by adding 5 to 50 wt % carbon black to samples *I*Pt and *2*Pt will hereafter be denoted as *I*PtC and *2*PtC (Table 2).

The results of the *S* measurements in the Pt/SnO_2 materials and the $Pt/SnO_2 + C$ composites differing in carbon content demonstrate that increasing the carbon content of the $Pt/SnO_2 + C$ electrocatalysts based on sample *2*Pt to about 33% is accompanied by an increase in S (Table 2). At larger percentages of carbon in the composite, S stabilizes (Table 2). It seems likely that carbon content of 33% or above corresponds to "conversion" of all the Pt nanoparticles into an electrically active state.

Taking into account the above results, we also measured the S of the IPtC50 electrocatalyst, containing 50% carbon (Table 2). Its S turned out to be smaller: 30 m²/g Pt. This can be the result of the poorer accessibility of catalytic centers because of the specific morphological features of the particles in sample I as a support. Note that the "working" surface area of the Pt in each carbon-containing composite considerably



Fig. 1. Typical X-ray powder diffraction pattern of the synthesized tin dioxide samples.



Fig. 2. X-ray powder diffraction patterns of the synthesized Pt/SnO₂ materials (samples *1* and *2*).

exceeded the S of the Pt in the carbon-free Pt/SnO_2 materials (Table 2).

The parent supports (SnO_2) have a large specific surface area. In particular, the surface area of sample 2 is 122 m²/g (BET measurements), the average total pore area in it is 35.5 m²/g Pt, and the average pore diameter is 4.1 nm. According to the TEM results, the composite *I*PtC50 contains individual SnO₂ nanoparticles 4 nm in size, agglomerates of such particles, and well-crystallized Pt nanoparticles about 5 nm in size (Fig. 3). Unfortunately, since the Pt and SnO₂ nanoparticles differ little in contrast in TEM images, histograms of size distributions for the Pt and SnO₂ nanoparticles are difficult to construct. X-ray element mapping results for the *I*PtC50 electrocatalyst (Fig. 4) demonstrate that the Pt is evenly distributed over the surface of the SnO₂ grains.

The smaller *S* of sample *I*Pt50, in which the Pt crystallite size is smaller than that in *2*Pt50 (Table 2), is attributable to both the presence of pores similar in size to the Pt nanoparticles (so that some of the nanoparticles reside in pores) and the higher degree of agglomeration of the platinum nanoparticles in *I*Pt than in *2*Pt.

Figure 5 compares cyclic voltammograms of sample *I*PtC50 and a commercially available electrocatalyst. The shape of the voltammograms for oxygen reduction reaction (ORR) on the *2*PtC50 electrocatalyst is typical of supported platinum electrocatalysts (Fig. 6). The half-wave potential in ORR at a rotation rate of 1600 rpm was 0.88 eV, and the number of electrons involved in the reduction of an O₂ molecule, evaluated using the Koutecky–Levich equation, was determined to be 3.9. This confirms that oxygen electroreduction on the composite electrocatalyst follows a four-electron mechanism. At a potential of 0.85 V, the activity of the electrocatalyst is 1.73 A/m² Pt, which is somewhat lower than that of the commercially available Pt/C analogue.







Fig. 3. TEM images of parts of sample *I*PtC50 (the arrows mark platinum nanoparticles).

The stability of the 2PtC50 electrocatalyst (Fig. 7) was evaluated in comparison with a commercially available Pt/C electrocatalyst (JM20). The Pt/C elec-

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Fig. 4. X-ray element maps of a portion of the surface of sample *1*PtC50.

trocatalyst was found to undergo catastrophic degradation: its *S* dropped from 93 to 11 m²/g Pt in 500 cycles. At the same time, the *S* of the 2PtC50 composite electrocatalyst decreased by just about 10% relative to its original level.

CONCLUSIONS

The optimization of conditions for the synthesis of nanostructured SnO_2 from a $SnCl_2$ solution has made it possible to obtain a material in the form of nanopar-



Fig. 5. Cyclic voltammograms of sample 2PtC50 and the commercially available Pt/C electrocatalyst JM20 (RHE = reversible hydrogen electrode).

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ticles with an average crystallite size near 3-4 nm and a large specific surface area ($122 \text{ m}^2/\text{g SnO}_2$). The structure of the SnO₂ nanocrystallites has been confirmed by direct TEM examination, electron diffraction data, and X-ray diffraction characterization results. The crystallite size agrees well with the nanoparticle size determined by TEM.

 Pt/SnO_2 electrocatalysts containing 30 wt % Pt and having a crystallite diameter of 5 and 7 nm have been prepared by chemical reduction. The addition of a Vulcan-XC72 fine-particle carbon material leads to an increase in the electronic conductivity of the compos-



Fig. 6. Voltammograms for ORR.



Fig. 7. Accelerated stress test results for electrocatalyst stability (500 cycles) at potentials in the range 0.6-1.4 V (ECASA = electrochemically active surface area).

ite. The largest S of the platinum, up to 47 m²/g Pt in the case of the 2PtC50 electrocatalyst, is observed at a carbon content of the composite above 33%.

The mass activity of the $Pt/SnO_2 + C$ electrocatalyst for ORR is slightly inferior to that of a commercially available Pt/C electrocatalyst containing considerably smaller Pt nanoparticles. The stability of the $Pt/SnO_2 + C$ electrocatalyst under harsh stress test conditions is significantly better than that of Pt/C. This confirms that $Pt/SnO_2 + C$ composites have considerable potential for use as electrocatalysts for polymer membrane hydrogen—air FCs. Reducing the size of Pt nanoparticles and optimizing the carbon content of such composites can improve the functional characteristics of the electrocatalysts.

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