# scientific reports



# **Efect of benzoyl chloride OPEN treatment on morphological, thermal, mechanical, and hydrothermal aging properties of date palm/polyvinyl chloride (PVC) composites**

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**The use of natural fbers has seen a signifcant rise in the composites sector, resulting in the creation of polymer composites with exceptional strength. These environmentally-friendly alternatives ofer a compelling substitute for synthetic composites. This study explores the use of date palm waste as reinforcement for the fabrication of polyvinyl chloride (PVC) composites. A surface modifcation method was essential for improving the binding interaction between palm fbers and PVC composites. The two-hour benzoyl chloride treatment at 140 °C played a crucial role. The study examined the efects of hydrothermal aging on mechanical properties of composites, using various techniques such as surface morphology analysis, Fourier Transform Infrared spectroscopy, and Thermogravimetric Analysis, on composites made of untreated fbers and those treated with benzoyl chloride. Although the treatment of palm fber-reinforced composites with benzoyl chloride improved their mechanical properties, it is crucial to note that hydrothermal aging reduced their tensile strength by 10%. Despite this, these composites prove to be well-suited for applications requiring moderate strength and stifness in mild environmental conditions. These composites, while utilizing benzoyl chloride for surface treatment, still represent a more sustainable alternative to traditional synthetic composites by incorporating renewable date palm waste and enhancing mechanical properties, which potentially reduces overall environmental impact.**

**Keywords** Polyvinyl chloride, Composites, Mechanical proprieties, Hydrothermal aging

The agricultural sector, particularly in the Middle East and North Africa (MENA) region, faces a significant challenge in managing the vast quantities of waste generated from date palm harvests. Annually, the MENA region produces an estimated 2.6 to 2.8 million metric tons of agricultural waste<sup>1</sup>, with a single date palm tree contributing approximately 26 to 35 kg of biomass waste. This waste, comprising rachis, leaflets, spadix stems, mesh, and date pits, represents a substantial environmental burden and an untapped resource<sup>[2,](#page-12-1)[3](#page-12-2)</sup>.

In response to this challenge, a paradigm shif is occurring in materials science and engineering. Researchers are increasingly focusing on valorizing agricultural waste, particularly natural fbers, as sustainable alternatives to traditional synthetic materials<sup>4-[6](#page-12-4)</sup>.

The potential applications of these natural fiber-based materials are vast and diverse. They are making signifcant inroads in the construction and automotive industries, fnding use in outdoor furniture, window parts, door panels, and roof materials<sup>[7,](#page-12-5)[8](#page-13-0)</sup>. Their versatility extends to reinforcing various polymer-based materials,

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including polyvinyl chloride (PVC), polystyrene (PS)<sup>[9,](#page-13-1)10</sup>, polyethylene (PE)<sup>11</sup>, and polypropylene (PP)<sup>12</sup>. The growing interest in these materials stems from their remarkable properties, which in many aspects surpass those of conventional reinforcement materials like fberglass.

Natural fibers, particularly those derived from date palm waste, offer a unique combination of benefits. They boast exceptional mechanical properties, thermal stability, and acoustic insulation capabilities. Furthermore, their biodegradability addresses growing environmental concerns, while their non-abrasive nature and cost-effectiveness make them increasingly attractive to manufacturers<sup>13</sup>. These attributes position natural fiber composites as a promising solution for sustainable material development in the twenty-frst century.

However, the widespread adoption of these materials faces a signifcant hurdle: their susceptibility to water absorption. This characteristic not only affects the mechanical properties of the composites but also limits their applicability in outdoor and moisture-rich environments<sup>14</sup>. The complex relationship between water absorption and the mechanical performance of biocomposites has been the subject of extensive research, with most studies reporting detrimental efects on material integrity and longevity.

To overcome these limitations, innovative fber modifcation techniques have emerged as a critical area of research. These modifications aim to remove hemicelluloses, lignin, and amorphous cellulose regions while introducing hydrophobic propertie[s7](#page-12-5)[,15,](#page-13-7)[16.](#page-13-8) Among these techniques, the treatment of fbers with benzoyl chloride  $(C<sub>6</sub>H<sub>5</sub>ClO)$  has shown particular promise. This method not only increases the cellulose percentage in fibers but also enhances their crystallinity index, thermal stability, and mechanical strength $13,17$  $13,17$  $13,17$ .

Previous studies have demonstrated the efficacy of benzoyl chloride treatment across various natural fibers. Increasing cellulose reduces the composites' afnity for water, concurrently improving the adhesion between natural fbers and the polymeric matrix. Consequently, these yields improved mechanical properties and heightened surface roughness<sup>18</sup>. Jacob et al.<sup>19</sup> reported improved mechanical properties and reduced water absorption in polyethylene composites reinforced with benzoyl chloride-treated plantain peel powder. Ganesh Babu's research<sup>20</sup> highlighted the potential of this treatment in enhancing the performance of brake pads made from Cyperus pandora fibers. Sherwani et al.<sup>[21](#page-13-13)</sup> conducted a comprehensive investigation into the effects of benzoyl chloride treatment on sugar palm fbers, noting signifcant improvements in mechanical and morphological properties. Similarly, Izwan et al[.22](#page-13-14) observed enhancements in physical and mechanical properties of sugar palm fbers following benzoyl treatment, including changes in color, diameter reduction, and increased tensile strength.

Building on this foundation, our groundbreaking study aims to revolutionize composite material production by introducing a novel method of treating date palm fbers with benzoyl chloride. Tis research uniquely addresses the challenges of interfacial adhesion and compatibility with polyvinyl chloride (PVC) matrix, with a specifc focus on applications in automotive interior components, furniture, and building materials. Our comprehensive approach employs a range of analytical techniques, including visual observations of physical changes, Fourier transform infrared (FTIR) analysis for chemical structure characterization, scanning electron microscopy (SEM) for microstructural examination, thermal degradation testing, and mechanical strength assessments.

A key innovation of this study lies in its exploration of how benzoyl chloride treatment afects the water absorption properties and tensile strength retention of date palm fber composites during hydrothermal aging. Tis aspect is crucial for developing materials suitable for use in humid environments or outdoor applications. By addressing these critical issues, our research paves the way for the development of more durable, highperformance natural fiber composites. The implications of this study extend far beyond academic interest. By demonstrating the potential of treated date palm fbers in producing robust, long-lasting composite materials, we open new possibilities for sustainable material use across various industries. Tis research not only contributes to solving agricultural waste management issues, but also aligns with global eforts towards sustainable development and circular economy principles.

This study represents a significant advancement in sustainable material science and engineering. By harnessing the potential of date palm waste and innovating fber treatment methods, we aim to create a new generation of eco-friendly, high-performance materials that can meet the demanding requirements of modern industrial applications while addressing pressing environmental concerns.

## **Materials and methodology Materials**

Tis study involved retrieving date palm leafets (Phoenix dactylifera L.) from the Biskra region in the southeastern part of Algeria were obtained with permission from owner of plantation. The leaflets were immersed in cold and hot water to remove impurities. Subsequently, they were air-dried and then trimmed to lengths of 1 cm. Finally, grinding and sieving were used to obtain fbers with an average width of 100 µm and an aspect ratio of 2.96.

Polyvinyl chloride (PVC) was sourced from the National Cable Industry Unit of Biskra, Algeria. It has a degree of polymerization ranging from 1250 to 1450 and a viscosity range from 120 to 130 mPa s. The K-value of the PVC is 67–72. We utilized a PVC compound containing a plasticizer (Di-Iso-Décyl-Phtalate, DIDP), a stabilizer with a calcium/zinc base (BAEROPAN MC 9917 KA), and a lubricant (stearic acid). This stabilizer is commonly used in PVC molding resins and provides good heat stability and resistance to oxidation.

The chemicals sodium hydroxide, acetic acid, benzoyl chloride acid, and xylene were procured from Sigma Aldrich and employed to treat the fbers' surfaces.

## **Fiber surface treatment**

A quantity of 10 g of palm fber is submerged in a solution containing 2% sodium hydroxide (NaOH) at a ratio of 1:20 for 30 min. Te fber is subsequently subjected to multiple rinses with deionized water containing a concentration of 10−2 mol/L acetic acid to remove any residual soda. Subsequently, a mixture comprising 300 ml of

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xylene and 50 ml of benzoyl chloride acid is introduced to the alkali-treated fiber. The process involves continuous stirring and heating at 140 °C under reflux conditions for 2 h. The palm fiber is subjected to a rinsing process using distilled water to eliminate any remaining inorganic salts. Subsequently, the fber is immersed in ethanol for one hour to eliminate residual benzoyl chloride. Afer a thorough water rinse, the specimen undergoes a 24-h desiccation process in an oven with an 80 °C temperature setting.

# **Fabrication of composites**

Date palm fber-reinforced PVC composites were fabricated using a calendar and a hydraulic press. Initially, the treated (BTF) and untreated (UTF) palm fbers (UTF) underwent a drying phase at 60 °C for 24 h to eliminate moisture. The PVC matrix was subsequently melted within a calendar machine ("Schwabenthan polymix 200 P") at temperatures ranging from 160 to 165 °C. Following this, the treated and untreated palm fbers were introduced into the molten PVC at a rate of 10 wt%. The resulting mixture underwent calendaring into strips and was then subjected to compression in a hydraulic press ("polystat 300 S") at temperatures of 160–165 °C and a pressure of 300 bars for 10 min. A detailed formulation of the composites is presented in Table [1.](#page-2-0)

### **Plants (either cultivated or wild)**

We declare that permission to collect palm Petiole as waste from Biskra district of Algeria with permission from owner of plantation and assure you it complies with relevant institutional, national, and international guidelines and legislation.

# **Characterization**

# **Fourier transform infrared analysis (FTIR)**

The FTIR-8400S Shimadzu spectrometer was used to investigate potential chemical bonding in raw and treated fibers. The samples were tested with 2 wt% of KBr and compressed to form a disc. Spectral data were obtained in transmission mode within the range of 4000–400 cm−1.

# **Morphological analysis**

A Quanta FEG250 scanning electron microscope (SEM) set to 15 kV was used to examine the surface structure of palm fibers and polyvinyl chloride composites, both treated and untreated. The specimens underwent processing using liquid nitrogen.

### **Thermal properties**

The Universal V4.5A TA instrument was used for thermogravimetric analysis (TGA) of fibers and PVC composites. Te analysis involved exposing the samples to a heating rate of 10 °C min−1, commencing at room temperature, and reaching a maximum temperature of 600 °C. Tese experiments were executed in a nitrogen atmosphere.

### **Mechanical analysis**

Tensile tests were performed using Zwick/Roell Z50 testing equipment with a 1 mm/min crosshead speed, following ASTM D638-22. For each sample, test at least five specimens.

Tree-point fexural tests were performed using the INSTRON testing machine, following the standard ISO 178:2001. The sample size was  $100 \times 10 \times 2$  mm<sup>3</sup>, and the crosshead speed was 1 mm/min.

#### **Hydrothermal aging test**

The composites have been inspected for tensile characterization after hydrothermal aging. For this purpose, composites specimens are weighed and immersed in a distilled water bath set at 50 °C for five hours. The samples were taken every hour and dried correctly before the tensile and water uptake tests. The weight uptake (w<sub>uptake</sub>) has been calculated in terms of percentage as per the following formula:

$$
w_{\text{uptake}} = \frac{w_f - w_0}{w_0} \tag{1}
$$

where  $w_f$  is the weight of the sample after hydrothermal aging and  $w_0$  is the weight of the sample before hydrothermal aging.

Similarly, the loss of tensile strength ( $\sigma_{\rm Loss}$ ) in terms of percentage has been computed as:

$$
\sigma_{Loss} = \frac{\sigma_0 - \sigma_f}{\sigma_0} \times 100 \tag{2}
$$



<span id="page-2-0"></span>**Table 1.** Formulation of the composites.

where  $\sigma_0$  is the tensile strength of the sample before hydrothermal aging, and  $\sigma_f$  is the tensile strength of the sample afer hydrothermal aging.

# **Results and discussion**

# **Physical appearance of treated and untreated palm fber**

The physical changes in palm fiber before and after chemical treatment were observed visually. Figure [1](#page-3-0) shows the varied colors of fbers afer treatment. Before treatment with benzoyl chloride, the UTF exhibited a dark brown color that gradually turned to a deep black. The change in color of natural fibers following treatment is associated with the breakdown of hemicellulose and lignin components from the fibers upon NaOH treatment $22,23$  $22,23$ .

# **FTIR analysis**

Fourier transform infrared (FTIR) spectra are a powerful way to see how benzoic acid chloride changes the chemical structure of palm fber. Figure [2](#page-3-1) displays signifcant diferences in the FTIR spectra of untreated palm fber (UTF) and palm fber treated with benzoyl chloride (BTF). It is important to note that a drop in the absorption band is linked to the O–H stretching of hydroxyl groups in cellulose and hemicellulose, which happens around 3422 cm−1. Tis decrease shows that the reaction between these hydroxyl groups and benzoyl chloride went well, which changed the surface chemistry of the fiber<sup>[22,](#page-13-14)23</sup>. At the same time, lower peaks at 2920 cm<sup>-1</sup> and 2850 cm<sup>-1</sup>, which are connected to methylene group vibrations, show that aliphatic chains have changed. These changes, along with changes in the CH2 group bending vibrations (shown by the peak at 1460 cm−1), suggest that the molecular structure of the fiber has been completely changed<sup>24</sup>. Additionally, characteristic bands at 1120 cm<sup>-1</sup> and 1046 cm<sup>-1</sup>, indicative of the elongation vibrations of C–O–C groups, are evident<sup>25</sup>. An indicator of the



**Fig. 1.** Physical appearance of treated and untreated palm fber.

<span id="page-3-0"></span>

<span id="page-3-1"></span>**Fig. 2.** FTIR spectra of untreated fber (UTF) and benzoyl chloride-treated fber (BTF).

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benzoylation process is the increased carbonyl band centered at 1737 cm−1. Tis prominent feature signifes the formation of ester linkages, a direct result of the reaction between hydroxyl groups and benzoyl chloride. The emergence of this band, coupled with new peaks at 1273 cm<sup>-1</sup> and 1312 cm<sup>-1</sup>, provides strong evidence for the successful esterifcation of the palm fbers.

The appearance of a peak at  $1605 \text{ cm}^{-1}$ , due to C=C stretching in aromatic rings, further confirms the introduction of benzoyl groups. This, along with the higher intensity at 710 cm<sup>-1</sup>, which shows aromatic rings with only one substituent, shows that benzene rings are part of the fiber structure. These modifications significantly alter the fiber's surface properties and, potentially, its interactions with other materials<sup>18</sup>. An important consequence of the benzoyl chloride treatment is a reduction in the fiber's hydrophilicity. The decreased band, typically associated with water absorption, evidences this. The lowered hydrophilicity can enhance the fiber's compatibility with hydrophobic matrices, making it more suitable for certain composites applications<sup>[26](#page-13-18)</sup>. Figure [3](#page-4-0) schematizes the benzoyl treatment of palm fbers.

# **Scanning electron microscopy**

Figure [4](#page-5-0)'s scanning electron microscope (SEM) pictures provide a close-up view of palm fbers without treatment (Fig. [4a](#page-5-0)) and those treated with benzoyl chloride (Fig. [4](#page-5-0)b). Tese micrographs clearly demonstrate how the chemical treatment altered the fibers' shape and influenced their surface properties<sup>27</sup>.

Figure [4a](#page-5-0), depicting the untreated palm fiber, reveals a complex and irregular surface topography. The fiber surface is characterized by pronounced roughness, with numerous protrusions and indentations creating a highly-textured landscape. A substantial accumulation of impurities and dust particles, forming a thick, heterogeneous layer across the fiber's exterior, further accentuates this roughness. The fiber's surface structure deeply embeds these contaminants, likely comprising various polysaccharides, extractives, and other non-cellulosic components. Such impurities not only contribute to the fber's irregular appearance, but also play a signifcant role in its physical and chemical properties. The presence of these surface contaminants and structural irregularities in the untreated fber can have a number of negative efects. Firstly, they act as potential stress concentration points, creating weak spots in the fber's structure that could initiate or propagate fractures under mechanical stress. Additionally, these impurities signifcantly impair the fber's ability to form strong, coherent interfaces with polymeric matrices in composite materials. The inconsistent surface chemistry and topography hinder efective wetting and bonding, resulting in poor stress transfer between the fber and the surrounding matrix. Unfortunately, this weak interfacial adhesion can make composites that use these untreated fbers perform badly and last a long time $28,29$  $28,29$ 

In stark contrast, Fig. [4](#page-5-0)b showcases the remarkable transformation achieved through benzoyl chloride treatment. The treated fiber exhibits a dramatically altered surface morphology, characterized by a notably cleaner and more uniform appearance. The most striking change is the significant reduction in visible surface impurities and contaminants. The treatment process has effectively stripped away the layer of non-cellulosic materials, exposing a more homogeneous fber surface.



<span id="page-4-0"></span>**Fig. 3.** Benzoyl treatment of date palm fbers by benzoyl chloride.



<span id="page-5-0"></span>**Fig. 4.** Micromorphology of palm fbers; (**a**) untreated fber, (**b**) treated fber.

The benzoyl chloride treatment appears to have induced a chemical etching effect on the fiber surface. This etching process not only removes impurities but also modifes the underlying cellulose structure, potentially increasing the surface area available for bonding. The resulting surface is markedly smoother, with a more consistent texture that suggests improved uniformity in both physical structure and chemical composition. As Fiore et al.<sup>[30](#page-13-22)</sup> reported, removing lignin, wax, pectin, and hemicellulose contributes to an augmented roughness of the fber surface. Tis increased roughness, in turn, improves the adhesion of polymers within composites materials. After the treatment, the observed phenomenon was a reduction in fiber size. This shrinking will help the fibers reduce moisture absorption $^{22}$ .

Figure [5a](#page-6-0)–c presents SEM micrographs of the neat PVC and their composites. At frst observation, the SEM morphology of all PVC/palm fber composites appeared similar yet signifcantly diferent from pure polyvinyl chloride resin. Adding date palm fbers to PVC composites caused the surface to become more rough and irregular. Figure [5b](#page-6-0) illustrates that including 10 wt% UTF fllers roughens the surface compared to pure PVC resin. In addition, the SEM micrograph of PVC-UTF composites showed clumping, fber pull-outs caused by the incompatible palm fller's poor dispersion and minimal adherence, and the formation of voids that promote the production of fragile interfaces or critical stress intensity points, which allow cracks to begin and spread, resulting in composites failure and a diminution in the value of mechanical properties $31,32$  $31,32$ .

The fracture surface of the PVC-BTF composites, where the fibers were treated with benzoyl chloride (Fig. [5](#page-6-0)c), reveals a pattern of fber breakage rather than pull-out that indicates a more efective adhesion between the surfaces of composites reinforced with treated fbers, with the PVC-BTF composites demonstrating superior performance. In contrast to other composites, the bond strength between the PVC matrix and BTF fbers was significantly higher, and no visible voids were observed at the fracture interfaces. The benzoylation process was crucial in toughening the fber surface and strengthening the bond between the PVC matrix and the fbers. Tis heightened bonding was attributed to decreased hydroxyl groups, fostering improved fiber cohesion<sup>[9](#page-13-1),[33](#page-13-25)</sup>.



<span id="page-6-0"></span>**Fig. 5.** Micromorphology of PVC/palm fber composites; (**a**) neat PVC, (**b**) reinforced with untreated fber (PVC-UTF) and (**c**) reinforced with treated fber (PVC-BTF).

# **Thermal properties**

Figure [6](#page-7-0) presents the thermal analysis (TGA) and derivative (DTG) curves for untreated (UTF) and benzoyl chloride-treated fibers (BTF). The thermal degradation behavior is illustrated through four distinct stages: initial dehydration, removal of hemicellulose and a portion of lignin, alpha-cellulose decomposition, and lignin pyrolysis<sup>34</sup>. The annotated curves clearly indicate the weight loss percentages and temperature ranges for each stage, with treated fbers showing higher thermal stability and shifed degradation temperatures.

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<span id="page-7-0"></span>Fig. 6. Thermogravimetric and derivative thermogravimetric curves of untreated fiber (UTF) and benzoyl chloride-treated fber (BTF).

During the initial Stage of decomposition, occurring between 50 and 150 °C, the fbers undergo dehydration, resulting in a 7.16% mass loss within the temperature range of 50.62 °C to 148.81 °C. The observed mass loss primarily results from the evaporation of volatile extracts within the fibers. The significant rise in moisture loss at elevated temperatures suggests robust hydrogen bonding between cellulose and water, indicating an increased abundance of hydroxyl groups OH. It underscores the inherently hydrophilic nature of natural fber[s18](#page-13-10)[,26.](#page-13-18) Not only that, but the temperature at which benzoyl chloride-treated fbers (BTF) break down is much higher than that of untreated palm fbers (UTF), which causes a slight drop in weight (2.54%). Tis weight reduction indicates a decrease in fber moisture content afer treatment, highlighting the efectiveness of the benzoyl chloride treatment process.

The final three stages are associated with the deterioration of hemicellulose, lignin, and cellulose linkages as the temperature increases<sup>[22](#page-13-14)</sup>. The second Stage, observed at 210 °C to 303 °C for raw fibers and 156 °C to 242 °C for treated fibers, signifies the removal of hemicelluloses and a portion of lignin<sup>[35](#page-13-27)</sup>. Alpha-cellulose decomposition occurs in the temperature range of 320–430 °C for untreated fbers, while this process is observed within the 280-444 °C range for benzoyl-treated fibers. These temperature ranges indicate the thermal stability improvements achieved through benzoyl chloride treatment, as evidenced by the shifed decomposition stages and elevated degradation temperatures. During this degradation phase, the UTF exhibits a weight loss of 57.03%, whereas the BTF experiences a weight loss of 64.02%. The maximum degradation temperature of benzoyl-treated palm fbers exhibits an increase, measuring 390.72 °C, compared to the raw fbers, with maximum degradation temperature of 349.71 °C. Cellulose undergoes a slow decomposition process at lower temperatures. In contrast, cellulose molecules are rapidly disintegrated at high temperatures, resulting in the decomposition of glycosyl units and the subsequent production of levoglucosan $36,37$ .

During the fourth Stage, two minor peaks were detected at temperatures of 409 °C and 470 °C, corresponding to the stages of lignin degradation and pyrolysis, respectively<sup>38</sup>. At temperatures exceeding 500 °C, the molecules undergo decomposition, forming diverse low molecular weight substances, including CO<sub>2</sub>, CO, H<sub>2</sub>O, hydrocarbons, and hydrogen  $(H_2)^{37}$  $(H_2)^{37}$  $(H_2)^{37}$ . Untreated fibers exhibited a residue weight of approximately 28.46% at a temperature of 600 °C, whereas the benzoyl chloride-treated palm fbers showed a residue weight of 27.86%. The benzoyl fiber treatment did not increase thermal stability, resulting in a lower temperature than UTF. Fewer decomposition peaks for treated palm fber demonstrated the elimination of specifc components from the fber during treatment<sup>22</sup>. That is due to the varied degradation tendencies of benzoyl-treated fiber molecules. The dissociation of C–C chain bonds and H-abstraction at the dissociation site<sup>39</sup> contribute to the intricate thermal behavior observed. Providing a comprehensive summary, Table [2](#page-8-0) offers valuable insights into the thermal degradation characteristics of treated (BTF) and untreated (UTF) palm fibers. The tabulated information encompasses critical parameters such as the initial degradation temperature, percentage weight loss, and percentage residue, elucidating the impact of benzoyl chloride treatment on the thermal stability of the fbers.



<span id="page-8-0"></span>**Table 2.** Infection temperatures related to the maximum rate of degradation of treated and untreated palm fbers.

Figure [7](#page-8-1) displays the TGA and DTG curves for neat PVC and PVC composites reinforced with treated and untreated palm fibers. The curves demonstrate two distinct stages of thermal decomposition: the removal of Hydrogen Chloride molecules and the degradation of polyene into a carbon black residue. The annotated curves provide clear demarcations of the onset and maximum decomposition temperatures, with PVC composites showing enhanced thermal stability due to improved adhesion between the matrix and fibers. The corresponding thermal analysis data, summarized in Table [3](#page-8-2), provides valuable insights into the distinctive thermal characteristics



<span id="page-8-1"></span>Fig. 7. Thermogravimetric and derivative thermogravimetric curves of neat PVC and composites reinforced with untreated fber (PVC-UTF) and benzoyl chloride-treated fber (PVC-BTF).



<span id="page-8-2"></span>**Table 3.** Decomposition temperatures and weight loss levels of neat PVC and PVC-treated and untreated palm fbers composites.

of PVC and its composites. Te thermal decomposition of the polyvinyl chloride matrix happens in two distinct stages: the frst one initiates at 257.75 °C and concludes at 367.06 °C, reaching a peak decomposition temperature of 310.75 °C. In this phase, there is a weight loss of 53.60%, attributed to the removal of Hydrogen Chloride molecules and the creation of double bonds of polyene within the PVC chains. The subsequent degradation stage, spanning from 416.55 to 559.22 °C, involves the degradation of polyene into a carbon black residue, which remains stable above 559.22 °C and contributes to 27.15% of the total weight $40,41$  $40,41$  $40,41$ .

Both treated and untreated PVC composites reinforced with palm fber exhibit a marginal weight loss at approximately 100  $^{\circ}$ C, attributed to the evaporation of water present in UTF and BTF fibers. The introduction of palm fber decreases the onset decomposition temperature, estimated at 238.36 °C and 231.76 °C for PVC-UTF and PVC-BTF, respectively. Tis reduction is attributed to the chain scission of composites, leading to increased release of hydrochloric acid (HCl) and the decomposition of holocellulose (cellulose and hemicel-lulose)<sup>[42](#page-13-34)</sup>. Enhanced adhesion between the PVC matrix and cellulose fiber leads to enhanced contacts, reducing the composites' thermal stability. This is corroborated by the degradation process of composites, in which the deterioration of one component in a composite has a multiplicative efect on the deterioration of all other components<sup>41</sup>. The second peak at approximately  $454.36$  °C is evident in treated and untreated composites. This phase is accompanied by a weight loss between 52.01% and 51.18%. Te decomposition process involves the breakdown of lignin, the residual residue, and the production of ashes, indicating the comprehensive combustion of the composites $9,43$ .

# **Mechanical characterization**

#### *Tensile testing*

Figure [8](#page-9-0) illustrates neat PVC and composites' tensile strength and modulus before aging. It is noteworthy that the incorporation of date palm fber into PVC results in a reduction in tensile strength from 12.8±1.12 MPa for pristine polyvinyl chloride to  $11.3 \pm 0.48$  MPa for untreated composites, specifically PVC-UTF. The decrease in tensile strength observed in untreated composites (PVC-UTF), compared to the pristine polyvinyl chloride (PVC), can be ascribed to the compromised interfacial adhesion within the composites. Conversely, reduced adhesion between the matrix and fller at the interface results in their separation, enhancing deformability and causing an increase in deformability<sup>19,44</sup>. The tensile strength trend reveals a notable enhancement in treated PVC-BTF composites compared to untreated PVC-UTF composites. These findings suggest an improved compatibility between hydrophilic cellulose fbers and the hydrophobic PVC matrix following the application of benzoyl chloride treatment. The improvement in tensile strength properties, observed in treated composites (PVC-BTF) attributed to the modification of palm fibers surface. The benzoylation treatment reduces the presence of free OH groups on the fber interface. Consequently, it amplifes the hydrophobic properties of the fbers, promoting improved interaction with the matrix. This, in effect, augments the mechanical properties of the composites, aligning with observations documented in prior literature<sup>[13](#page-13-5)</sup>.



<span id="page-9-0"></span>**Fig. 8.** Ultimate tensile strength and modulus of neat PVC and composites.

The tensile modulus of the composites exhibits an increase, rising from  $112.2 \pm 11.06$  MPa for polyvinyl chloride (PVC) to 304.5±7.26 MPa for the treated composites. It is anticipated that benzoyl modifcation will facilitate the removal of solidifying materials, including lignin and natural oils, from the fiber's surface<sup>[45](#page-14-0)</sup>. This removal process results in enhanced packing of cellulose chains, increased fber surface roughness, and enhanced contact area between the fiber and polyvinyl chloride (PVC)<sup>30</sup>. Therefore, this chemical modification has been found to have a beneficial effect on the mechanical properties of the composites<sup>[46](#page-14-1)</sup>. An elevation in stiffness was also noted in untreated samples, attributed to the inherent rigidity of fbers compared to the polymer. Tis increased stifness contributes to a stiffer composite without altering its strength. These outcomes align with findings reported in<sup>9,[47](#page-14-2)</sup>.

#### *Flexural testing*

Figure [9](#page-10-0) illustrates the fexural strength and modulus variations for PVC composites reinforced with treated and untreated palm fibers. The results display a similar trend to the tensile strength findings. The benzoyl-treated PVC-BTF composites exhibit the highest flexural strength at approximately  $2.63 \pm 0.04$  MPa. The enhanced fexural strength observed in benzoyl-treated PVC-BTF composites can be ascribed to improved homogeneity, adequate wetting, and enhanced interfacial bonding between benzoyl-treated date palm fbers (BTF) and the PVC matrix under compressive loads during bending. Additionally, the fber-matrix interface has increased transfer capacity in the reinforced composites with treated fibers<sup>47[,48](#page-14-3)</sup>.

The flexural modulus, as depicted in Fig. [9](#page-10-0), has experienced a positive increase due to the reinforcement with palm fibers. This increase can be attributed to the enhanced strength of the reinforcement material. The flexural modulus is primarily influenced by the properties of the matrix and the bonding between them and fibers<sup>[49](#page-14-4)</sup>. It has been observed that the treatment of benzoylation significantly influences the flexural modulus. The treated composites PVC-BTF exhibited the highest fexural modulus (88.66±4.10 MPa) compared to the PVC-UTF composites and pure PVC. The application of the treatment resulted in significant alterations on the surface of the palm fiber. These alterations included the elimination of alkali-soluble components and a decrease in the thickness of the palm fber.

Consequently, numerous small voids were formed on the fiber surface<sup>45</sup>. The benzene rings in the benzoyl group ( $C_6H_5O$ ) instead of hydroxyl groups attached to the fibers enhance the hydrophobic nature of the fiber surfaces<sup>17</sup>. Thus, enhancing the adhesion between the surfaces of palm fiber and the PVC matrix contributes to improving the fexural modulus in PVC-BTF composites [50.](#page-14-5)

#### **Hydrothermal aging of composites**

Humidity aging is widely recognized as a signifcant factor contributing to the degradation of organic materials exposed to air or in contact with water over time. Several mechanisms of aging caused by dampness have been



<span id="page-10-0"></span>**Fig. 9.** Flexural Strength and modulus of neat PVC and composites.

acknowledged, including matrix plasticization, diferential enlargement due to concentration gradients, and damage at the matrix/fiber interface $51,52$  $51,52$  $51,52$ .

The curves shown in Fig. [10](#page-11-0) depict the water uptake and the trend of loss of tensile strength under exposure to hydrothermal aging versus the square root of time for the virgin PVC and composites. By comparing the mass gain plots of specimens, it can be seen that the infuence of increased temperature on the difusion characteristics is very apparent. The observed phenomena may be explained by the widely recognized impact of temperature on the thermodynamic contribution, known as the "Soret efect" of difusion substances. Elevated temperatures expedite the process of water absorption<sup>53</sup>. Since the PVC matrix is a hydrophobic material, the rate of water uptake was less than other composites and did not exceed 0.13%<sup>54</sup>. The untreated specimens PVC-UTF show higher absorbed water, about 1.23%, after 5 h of immersion in water at 50 °C. The observed phenomenon can be attributed to the gaps and micro-voids between UTF and PVC composites, which provide the channels through which water can pass, causing plasticization and reducing the glass transition temperature<sup>[55,](#page-14-10)56</sup>. The untreated palm fber shows a signifcant amount of OH groups, which readily interactions with water molecules forming hydrogen bonds, leading to swelling and weakening of the fber structure as described in Fig. [11.](#page-12-6) Tis interaction can also disrupt the fber-matrix interface, leading to debonding and further mechanical degradation. Conse-quently, the increase in the concentration of hydroxyl groups corresponds to an elevated water absorption rate<sup>[24](#page-13-16)</sup>.

The benzoyl treatment significantly reduced water absorption for the PVC-BTF composites, amounting to 0.65%. Using benzoylation treatment enhanced the hydrophobic nature of PVC composites reinforced with palm fiber, decreasing mass absorption. This reduction in weight absorption is explained by the hydrophobic nature imparted by benzoyl groups on the fiber surface<sup>[18](#page-13-10),[21,](#page-13-13)57</sup>. The trend of tensile strength loss (Fig. [10\)](#page-11-0) is the same for PVC, PVC-UTF, and PVC-BTF composites afer hydrothermal aging for 5 h at 50 °C. Observing untreated fber-reinforced PVC composites (PVC-UTF) shows that when specimens are submerged in water, their tensile strength drops sharply, from 11.3 MPa to 8.85 MPa. Tis drop in tensile strength, which is around 21.68% compared to the initial tensile strength, is due to the swelling of palm fbers along water absorption, water gets into composites by difusing into the matrix through capillarity at the fber-matrix interface. Hydrophilic parts of the matrix then absorb the water and move along the inside fibers. Then, palm fibers caused a visible growth because the lumen physically grew and cellulose molecules mixed with water molecules. When it became big enough, matrix cracked around the swollen fbers. Over time, as the material aged, pectins, hemicelluloses, and some celluloses that weren't fully crystallized started to leak out of the palm fbers. Tis caused the fber-matrix interface to delaminate completely. Lastly, the diferent ways that each cell wall layer swelled caused the strengthening fibers to split and peel off, which reduced the fiber/matrix interface and caused further damage<sup>[58](#page-14-13),[59](#page-14-14)</sup>. Nevertheless, treated PVC-BTF composites exhibit a slightly lower tensile strength than untreated composites. Karlsson et al.<sup>[60](#page-14-15)</sup> suggest that fibers tend to swell when water infiltrates the interior of composites materials, leading to matrix structure alterations through chain reorientation and contraction. Tis phenomenon results in decohesion, weakening interfacial adhesion, and ultimately diminishing mechanical properties.



<span id="page-11-0"></span>Fig. 10. Weight uptake and loss of tensile strength after hydrothermal aging of neat PVC and composites.



<span id="page-12-6"></span>**Fig. 11.** Formation of hydrogen bonds between cellulose molecules and water molecules.

Tis drop in tensile strength, which is around 21.68% compared to the initial tensile strength, is due to the swelling of palm fbers along water absorption, which reduced the fber/matrix interface and caused further damage[58,](#page-14-13)[59.](#page-14-14) Nevertheless, treated PVC-BTF composites exhibit a slightly lower tensile strength than untreated composites. Behera et al.<sup>[60](#page-14-15)</sup> suggest that fibers tend to swell when water infiltrates the interior of composite materials, leading to matrix structure alterations through chain reorientation and contraction. This phenomenon results in decohesion, weakening interfacial adhesion, and ultimately diminishing mechanical properties.

# **Conclusion**

Tis research investigates the capacity of date palm fbers to enhance the strength of polyvinyl chloride (PVC) polymer composites, focusing on the effects of benzoyl chloride treatment. The study reveals that treating the fibers with benzoyl chloride significantly improves the mechanical properties of the PVC composites. This improvement is primarily attributed to the enhanced adhesion between the matrix and fbers, which is confrmed through Fourier-transform infrared (FTIR) spectroscopy and surface morphology analysis using scanning electron microscopy (SEM). The tensile strength of the treated composites (PVC-BTF) increased from  $11.3 \pm 0.48$  MPa for untreated composites (PVC-UTF) to  $12.8 \pm 1.12$  MPa, and the tensile modulus rose from 112.2±11.06 MPa to 304.5±7.26 MPa, indicating a substantial enhancement in mechanical performance. Furthermore, thermal analysis shows an improvement in the thermal stability of the composites post-treatment. However, hydrothermal aging tests highlight a signifcant impact on water absorption and tensile strength, with untreated composites showing a 21.68% reduction in tensile strength due to water-induced swelling and interface degradation. Despite this, the benzoyl chloride-treated composites demonstrate a more controlled water uptake and better retention of tensile properties, making them suitable for applications requiring moderate strength and stiffness in mild environmental conditions. These findings underscore the potential of benzoyl chloride-treated date palm fbers as a sustainable reinforcement material for PVC composites, ofering improved mechanical and thermal properties while reducing environmental impact.

# **Data availability**

The data that support the findings of this study are available from the corresponding author, (Jawaid, M.), upon reasonable request.

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# **Author contributions**

Data curation, H.B. and M.J.; formal analysis, H.B. and B.G..; project administration, H.B. and M.J.; writing original draf, H.B. and B.G.; writing—review and editing, M.J. funding, validation and editing, funding: H.F; validation and editing manuscript: R.K All authors have read and agreed to the published version of the manuscript.

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# **Competing interests**

The authors declare no competing interests.

# **Additional information**

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