

# SCIENTIFIC REPORTS



OPEN

## Coarsening Behavior of Particles in Fe-O-Al-Ca Melts

Lin Zhu Wang<sup>1</sup>, Junqi Li<sup>1</sup>, Shufeng Yang<sup>2</sup>, Chaoyi Chen<sup>1</sup>, Huixin Jin<sup>1</sup> & Xiang Li<sup>3</sup>

The characteristics of particles greatly affect the microstructure and performance of metallic materials, especially their sizes. To provide insight into coarsening phenomena of particles in metallic melts, Fe-O-Al-Ca melt with calcium aluminate particles was selected as a model system. This study uses HT-CSLM, SEM detections and stereological analysis to probe the behavior of particles and their characteristics including size, number density, volume fraction, spreading of particle size, inter-surface distance and distribution of particles. Based on the experimental evidence and calculation of collision, we demonstrate that the coarsening of inclusion particles is not only dependent on the Ostwald growth as studied in previous study, but also on the particle coagulation, and floatation. The collision of particles affects the maximum size of the particles during whole deoxidation process and dominates the coarsening of particles at later stage of deoxidation under the condition without external stirring in Fe-O-Al-Ca melts. The factors influencing collision behaviors and floating properties were also analyzed, which is corresponding to coarsening behavior and change of particle characteristic in the melts with different amounts of Ca addition. Such coarsening mechanism may also be useful in predicting the size of particles in other metallic materials.

Particles are inevitable products in metallic materials which form during metal refining, casting, and thermal processing in liquid or solid metals<sup>1–4</sup>. It has been a common knowledge that particles play a significant role in determining the continuous castability and performances of materials<sup>5–7</sup>. The Al<sub>2</sub>O<sub>3</sub> particles with high hardness and high melting temperature generate during deoxidation process in Al-killed steels may lead to nozzle clogging<sup>8</sup>. Another possible side effect is a decrease in machinability and service life, caused by the nucleation and propagation of voids around precipitate particles on the weak grain boundaries<sup>5,7,9</sup>. However, the effect of particles on the properties of metal is significantly dependent on the particle size distribution, spatial distribution, morphology and composition of particles. Lots of scholars have verified that the particles with certain characteristics can act by pinning grain boundaries, inhibiting grain growth or inducing precipitated phase formation, such as AF (acicular ferrite), thus refining the grain and improving the microstructure<sup>4,10–14</sup>.

Extensive investigations have focused on particle-assisted microstructure control<sup>1,15,16</sup>. A. Mitchell *et al.* pointed out that the particles with distance larger than 10 μm and diameter smaller than 1 μm have no impact on macro-performance of metallic products<sup>17</sup>. The yield strength and tensile strength would increase remarkably for steels with particles less than 0.3 μm<sup>18</sup>. The fine MgO-containing particles were found to have a facilitating effect on the formation of equiaxed crystallization and refinement of microstructure<sup>19–21</sup>. Yang *et al.*<sup>22</sup> reported that the proportion of AF progressively increased with increasing particle size from 1.0 to 1.8 μm and the ability of particles to induce AF was greatly reduced when the particle size reached 7.0 μm. Particles containing Ce with a size of about 4–7 μm can serve as heterogeneous nucleation sites for AF formation<sup>23</sup>. In spite of controversies on relation between particles with various compositions on the microstructure of metallic materials, some oxides, sulfides, nitrides and complex compounds (MnS, Ti<sub>2</sub>O<sub>3</sub>, TiN, VN, TiO·Al<sub>2</sub>O<sub>3</sub>·MnS, ect.) have been displayed promoting intra-granular ferrite nucleation or pinning grain boundaries<sup>24–26</sup>. The dispersed particles containing Ca or Mg are found to serve as heterogeneous nuclei for fine ferrite effectively due to the relatively weak affinity between individual particles and other characteristics<sup>27</sup>. Furthermore, Ca-containing alloy is commonly used to improve the continuous castability of liquid steel by modifying solid alumina particles into liquid calcium aluminates<sup>28–30</sup>. Yet, inappropriate addition of calcium can lead to the formation of calcium aluminate particles with large size. Wang *et al.*<sup>31</sup> found that the stringer shaped particles longer than 150–350 μm in linepipe steel were deformed from calcium aluminate with 10–20 μm in cast slab which deteriorated the properties of low

<sup>1</sup>School of Materials and Metallurgy, Guizhou University, Guiyang, Guizhou, 550025, China. <sup>2</sup>School of Metallurgical and Ecological Engineering, University of Science and Technology Beijing, Beijing, 100083, China. <sup>3</sup>College of Materials & Metallurgical Engineering, Guizhou Institute of Technology, Guiyang, 550003, China. Correspondence and requests for materials should be addressed to J.L. (email: [jqli@gzu.edu.cn](mailto:jqli@gzu.edu.cn)) or S.Y. (email: [yangshufeng@ustb.edu.cn](mailto:yangshufeng@ustb.edu.cn))

temperature toughness and hydrogen induced crack of steel. One of the key factors for decreasing the side effects of particles, or determining in the pining effect of particles or their ability on serving as cores of precipitated phase nucleation is the particle size.

The formation of particles starts with nucleation which plays an important role in determining the structure, shape and size distribution of the particles. Suito and Ohta *et al.*<sup>32</sup> found that the initial size distribution of particles became narrow in the case of high nucleation rate, which they thought was facilitating to obtain fine particles. After that, the particles are deemed to grow and coarsen by the following steps: the diffusion of reactants to the oxide nuclei, Ostwald ripening<sup>33</sup>, collision and subsequent coagulation in liquid metal. Lindberg *et al.*<sup>34</sup> reported that the time for attaining the 90% of the equilibrium value of particle volume is 0.2 s. Suito and Ohta *et al.*<sup>35</sup> investigated that the growth of particles by diffusion is very fast. In their study, Ostwald ripening dominates the growth of particles in deoxidation process under no fluid flow. However, in our previous study<sup>36</sup>, the experimental evidence for particles size distribution in Fe-O-Al-Ca melt corresponds to the theoretical results based on Ostwald ripening at early stage of deoxidation but not at later stage. Therefore, it is necessary to study the change of particle size distribution in liquid metal affected by the collision and subsequent coagulation. Collisions between particles and rapid diffusion in the liquid phase increase the number of large particles and enhance particle removal by floatation<sup>37</sup>. Extensive theoretical studies about the time-dependent particle size distribution and mathematical model have been reported based on collision-coalescence behavior due to turbulent collision<sup>38</sup>, Stokes collision<sup>39–41</sup>, Brownian collision<sup>42</sup>. Furthermore, attractive capillary force acted on particles has been also investigated with consideration on chemical compositions, size and distance between particles which is one mechanism for coagulation<sup>43–46</sup>. It can be concluded that the particle size distribution is effected by nucleation, growth, coagulation due to collision and attractive force, and floatation behavior of particles in liquid metal<sup>47–52</sup>. However, lots of researches are focused on the behavior of solid particles<sup>32,34,53</sup> and there are limited researches on the coarsening of liquid particles in liquid metal. The nucleation and Ostwald ripening of liquid calcium aluminate particles in Fe-O-Al-Ca melt have been investigated in our previous study<sup>36</sup>. In current study, the coarsening mechanism of particles in Fe-O-Al-Ca melt was studied with the consideration of nucleation, coagulation due to collision and floating properties and verified by experimental data. This study will provide information to understand the relations between characteristics, behavior and coarsening of particles in Fe-O-Al-Ca melts, and will be helpful for predicting and controlling size of particles.

## Methods

Materials used in the present study and high temperature experimental processes are described detailedly in our previous study<sup>36</sup>. Characteristics of particles were detected by SEM at an accelerating voltage of 15 KV and the transformation of particle characteristic in three-dimensional from that in two-dimensional based on stereological analysis is the same with our previous study<sup>36</sup>.

Geometric standard deviation of particle size distribution  $\ln \sigma$  values is calculated by Eq. (1)

$$\ln \sigma = \left[ \frac{\sum n_i (\ln r_i - \ln r_{geo})^2}{\sum_{i=1}^n n_i} \right]^{1/2} \quad (1)$$

Where  $r_{geo}$  is the geometric mean radius of particle given by  $(r_1 \cdot r_2 \cdot r_3 \cdot \dots \cdot r_n)^{1/n}$ . The  $\ln \sigma$  values are obtained from Eq. (1) using the values for the size and number density of particles which were measured in the deoxidation experiments.

The inter-surface distance  $D_{ab}$  between two particles can be obtained by measuring the central coordinates and radius of particles by Image-Proplus software and the illustration is shown in Fig. 1.

$$D_{ab} = \sqrt{(X_b - X_a)^2 + (Y_b - Y_a)^2} - r_a - r_b \quad (2)$$

Where  $X_i$  and  $Y_i$  are the central coordinates of particles in the cross section,  $r_i$  is the equivalent radius and  $D_{ab}$  is the inter-surface distance between two particles.

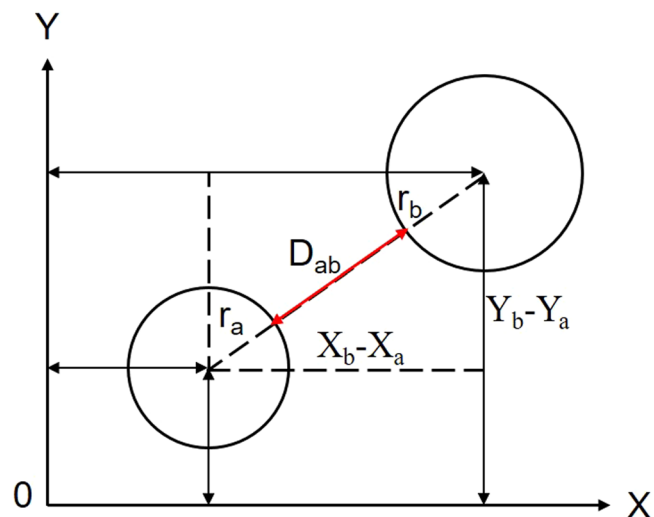
By calculating the inter-surface distances of certain particle with all others, the inter-surface distance between this certain particle with the nearest particle  $D_{mi}$  can be obtained by calculating the minimum value of  $D_{ab}$  as Eq. (3) and  $D_{mi}$  is defined as the inter-surface distance of a pair of adjacent particles in this paper. The average inter-surface distance of particles in certain region of sample  $D_{AV}$  is the arithmetic mean value of  $D_{mi}$  calculated as Eq. (4).

$$D_{mi} = \text{MIN}(D_{i1}, D_{i2} \dots D_{ik}) \quad (3)$$

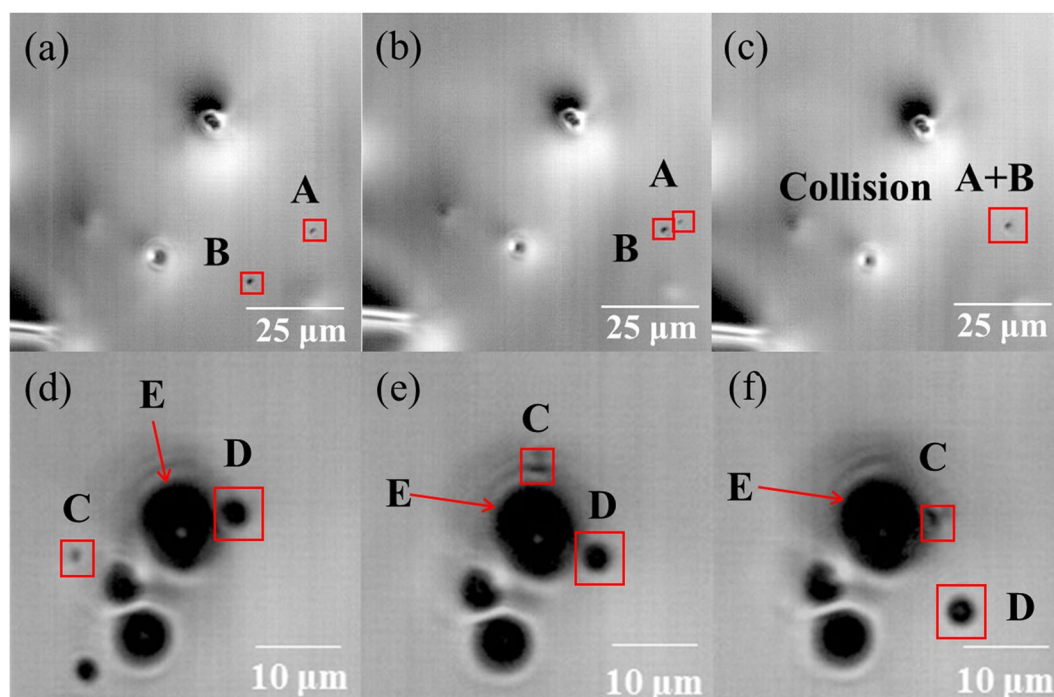
$$D_{AV} = \frac{\sum_{i=1}^k D_{mi}}{k} \quad (4)$$

## Results

***In-situ* observation of liquid particle behavior.** The behavior of particles in Fe-O-Al-Ca melt was *in-situ* observed using high-temperature confocal scanning laser microscope (HT-CSLM) as shown in Fig. 2. Most aggregation and coagulation between liquid calcium aluminates were caused by collision as Fig. 2(a–c). The attraction force was hardly found between most liquid calcium aluminate particles at gas/molten steel interface, even at very small separation (particle C/D moved to particle E and then passed away) as shown in Fig. 2(d–f). The same phenomenon was also observed by Hongbin Yin<sup>54</sup>, in which they found that the liquid calcium aluminate particles could separate freely after getting in touch with each other at 1/6 seconds.

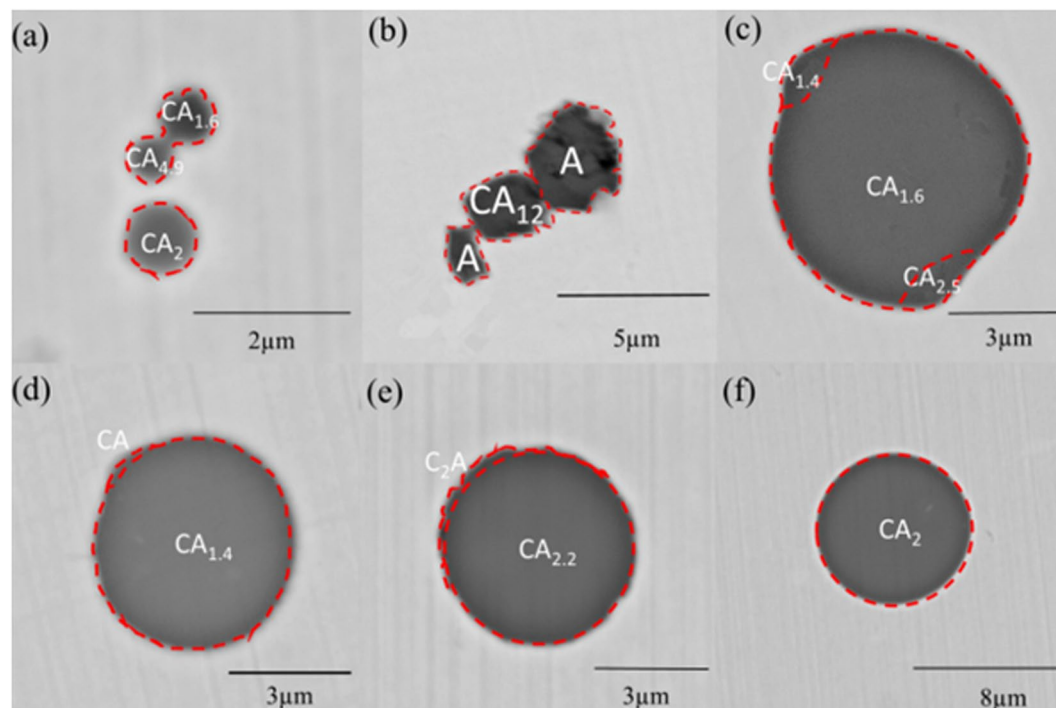


**Figure 1.** Illustration of inter-surface distance between particles.



**Figure 2.** *In-situ* observation of calcium aluminate inclusions on the surface of Fe-O-Al-Ca melt by HT-CSLM. (a) Particles A and B at 0 s in field 1; (b) particles A and B at 0.22 s in field 1; (c) particles A and B at 0.66 s in field 1; (d) particles C, D and E at 0 s in field 2; (e) particles C, D and E at 0.55 s in field 2; (f) particles C, D and E at 1.21 s in field 2.

**Characteristics of particles.** Morphologies and compositions of particles in steels deoxidized by Al and Ca alloys SEM-EDS micrographs are displayed in Fig. 3. It can be seen that there were lots of calcium aluminate particles in collision and coagulation. Few particles with similar size were found jointed as Fig. 3(a). According to the low melting point diagram Ca-Al-S<sup>55</sup>, the particles with mole ratio of Al<sub>2</sub>O<sub>3</sub> to CaO in the range of 0.15–1.5 are in liquid or partially liquid state which are regarded as “liquid particles” in this paper. Some solid particle merged with each other and formed into an irregular aggregate by high temperature sintering which was hard to deform and densify as Fig. 3(b). The coagulations between the liquid particles were observed as Fig. 3(c–e) and these aggregates seem susceptible to deform into spherical body. Such difference is thought to be attributed to the difference of inter-diffusion of composing elements and contact area, as the liquid particles are prone to spread on the surface of the other one<sup>54</sup>. Therefore, it can be concluded that the particles with large discrepancy in size tend to collide and merge, and the deformation as well as densification proceed easily for liquid calcium aluminates particles.

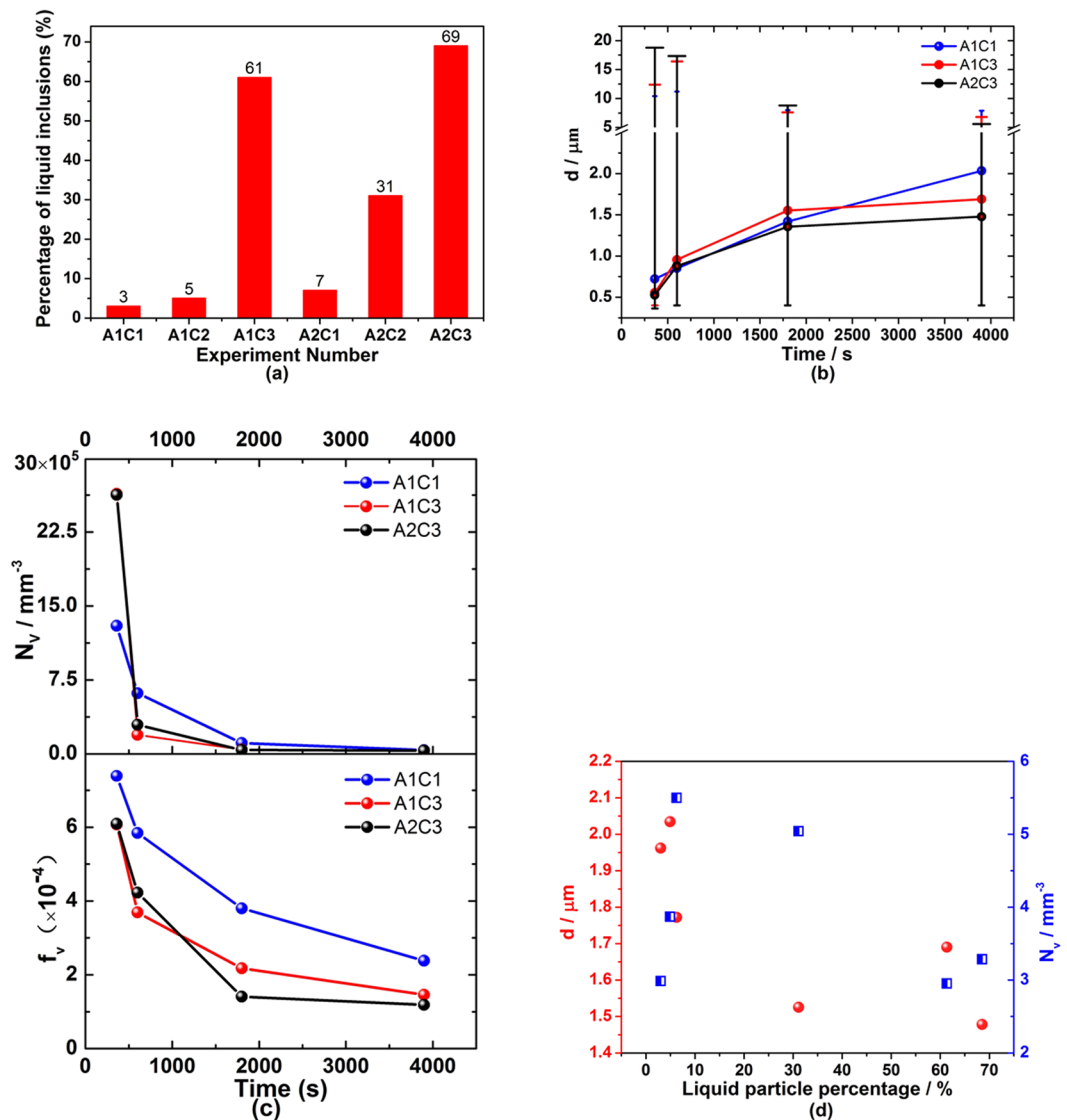


**Figure 3.** SEM-EDS analysis of typical calcium aluminate particles in Fe-O-Al-Ca melts after deoxidation at 1600 °C for 360 s.

In order to study the effect of liquid particle on their characteristics, the percentage of liquid particles in Fe-O-Al-Ca melts is illustrated in Fig. 4(a) (C1/2/3 represents that initial adding amount of Ca is 0.25%/0.4%/0.78%; A1/2 represents that initial adding amount of Al is 0.05%/0.25%). The experimental condition and chemical compositions of samples were depicted in previous study<sup>36</sup>. Liquid particle percentage increased with increasing amount of calcium addition in melts. The steels with high calcium addition ([%Ca] = 0.78) after deoxidation 3900 s have an extremely higher percentage of liquid particles than those with low calcium addition ([%Ca] = 0.25, 0.4).

A few hundreds of particles were observed by SEM-EDS in each sample and the planar particle size distribution was transformed into the size, number density and volume fraction of particles in three-dimensional based on stereological analysis. The average size of particles increased, and their number density and volume fraction decreased significantly with holding time as illustrated in Fig. 4(b,c). The error bars in Fig. 4(b) indicates that the particles with 18 μm formed in experiment A2C3 ([% Al]<sub>i</sub> = 0.25 and [% Ca]<sub>i</sub> = 0.78) during the first 360 s of deoxidation and subsequently those with 16 μm formed in experiment A1C3 ([% Al]<sub>i</sub> = 0.05 and [% Ca]<sub>i</sub> = 0.78), but no particles with size larger than 12 μm were observed in experiment A1C1 ([% Al]<sub>i</sub> = 0.05 and [% Ca]<sub>i</sub> = 0.25). The size of the largest particle in each experiment was larger in the steel with higher Ca addition during the first 360 s of deoxidation, and decreased with holding time due to the rapid floatation of large particle<sup>56</sup> which was explained in DISCUSSION part. The change of number density and volume fraction for calcium aluminates in Fig. 4(c) suggests that the ascending velocity of particles in the steel containing more liquid particles (A1C3 and A2C3) was larger than that containing more solid particles (A1C1) at early stage due to the relatively larger size and fractal dimension of liquid particles in the steels with high calcium (In spite of larger density for solid calcium aluminate particles, the liquid particles in the steel containing high calcium were larger in size and fractal dimension relatively at the initiation of deoxidation, which accelerated the floatation of these liquid particles). It is reported that the ascending velocity of the condensed particles with fractional characteristic was smaller than that of isometric three dimension spherical particles<sup>43</sup>, and it decreased with the decreasing value of  $D_f$  (fractal dimension)<sup>57</sup>. Based on the expression of  $D_f$  by Lech Gmachowski<sup>58</sup>, the liquid aggregates with spherical shape have larger  $D_f$  than the irregular solid aggregates.

With the rapid rise of large particles, the average size of particles increased slightly and their number density decreased slowly after deoxidation for 1800 s. Furthermore, the change of characteristics for particles in the steels containing high calcium (with low number density of particle) was smaller than that in the case with low calcium (with high number density of liquid particle) at later stage of deoxidation. It is thought to be caused by the difference of collision rate affected by the number density which can be verified in DISCUSSION part. Therefore, as illustrated in Fig. 4(d), the average diameter tended to decrease with an increased percentage of liquid particles due to the rapid rise of particles in high Ca containing steel at early stage of deoxidation and less collision at later stage. The number density of particles changed irregularly which is affected by their aggregation and floatation behaviors.

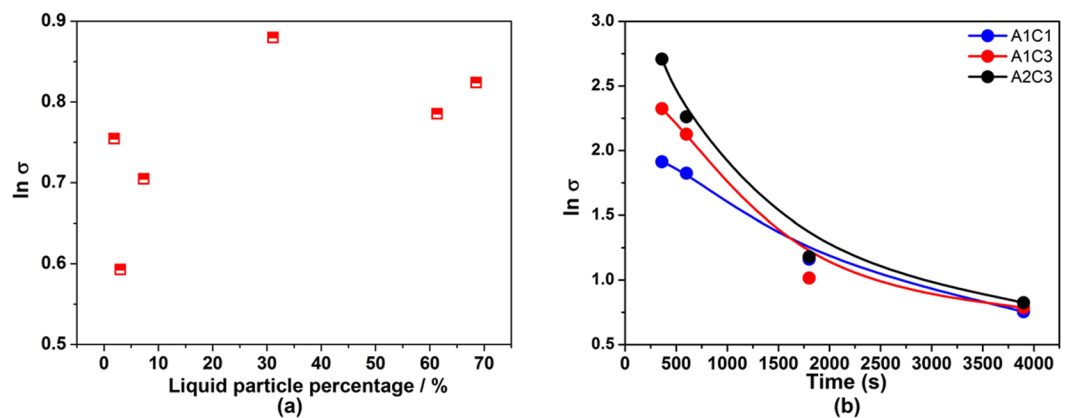


**Figure 4.** Percentage of liquid particles, diameter, number and volume fraction of particles in samples. (a) Percentage of liquid particles in each sample is observed by SEM-EDS in samples after deoxidation for 3900 s and it is calculated based on Ca-Al-S phase diagram<sup>55</sup>; (b) effect of holding time on particle diameter in three-dimensional based on stereological analysis and the error bars represent the maximum and minimum values of particle size; (c) effect of holding time on number and volume fraction of particles in three-dimensional; (d) change of average diameter and number density of particles in three-dimensional with liquid particle percentage, and the red circle and blue square represent average diameter and number density of particles, respectively.

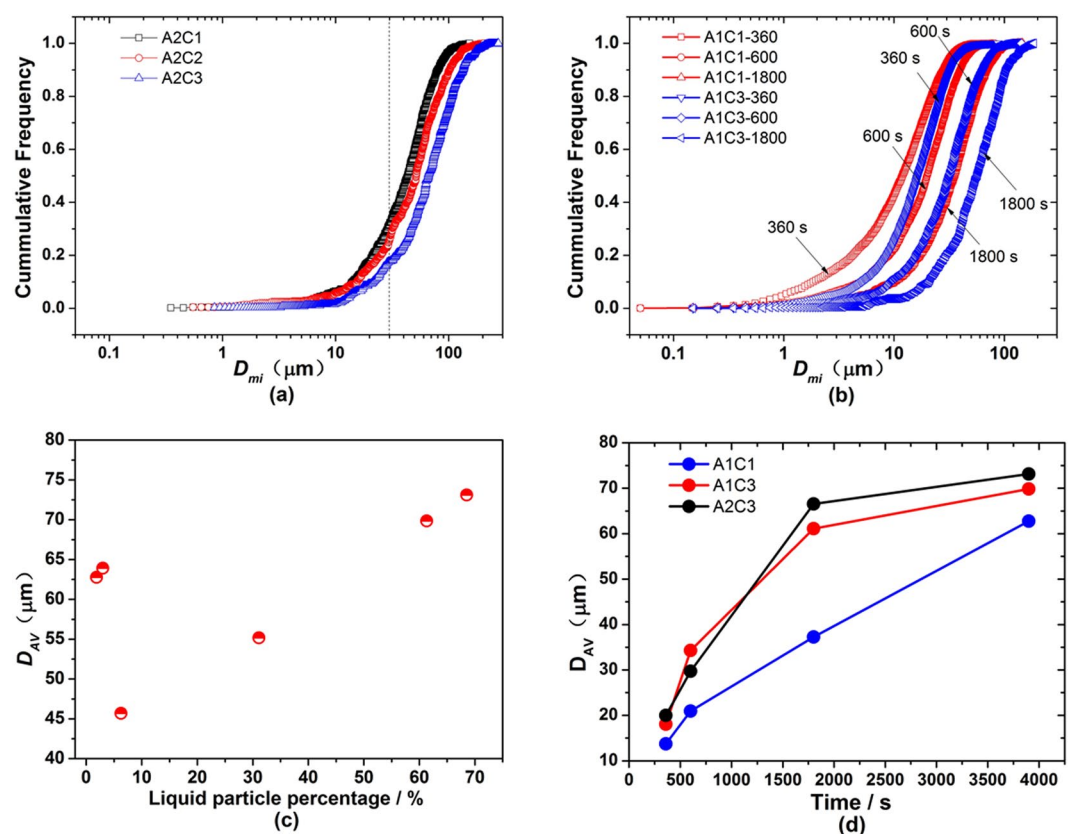
**Spreading of particle size.** The geometric standard deviation of particle size distribution  $\ln \sigma$  value which represents the spreading of particle size distribution in each experiment was measured in the deoxidation experiments. It is illustrated in Fig. 5(a) that the  $\ln \sigma$  values had an increasing trend with the increase of liquid particle percentage after deoxidation for 3900 s and they decreased with time elapsed. As the spreading of a size distribution becomes narrower with a decrease in  $\ln \sigma$ , it means that discrepancy in particle size is greater in the case with more liquid particles. Ohta *et al.*<sup>35</sup> reported that  $\ln \sigma$  values were dependent on the nucleation rate in the early stage. It can be seen that in Fig. 5(b) that the  $\ln \sigma$  values increased in the order of Exp. A1C1 < Exp. A1C3 < Exp. A2C3 at the first 600 s of deoxidation process, in which the theoretical nucleation rates  $\ln I$  were 484, 313, and 309, respectively (as reported in our previous study<sup>36</sup>). This result is in agreement with the conclusion that in the case of low nucleation rate, the particle size distribution becomes broader<sup>35</sup>. The  $\ln \sigma$  values of calcium aluminate particle at 3900 s changed non-monotonically with the increase of liquid particle percentage due to the hereditary of particle size distribution from early stage of deoxidation and the change of particle number density.

**Inter-surface distance between particles.** The cumulative frequency curves of  $D_{mi}$  (inter-surface distance of adjacent particles) in Fig. 6(a) change little, and the particles with inter-surface distance of 30–100  $\mu\text{m}$  were in larger proportion. The curves in Fig. 6(a) move toward right with increasing addition of calcium at 3900 s



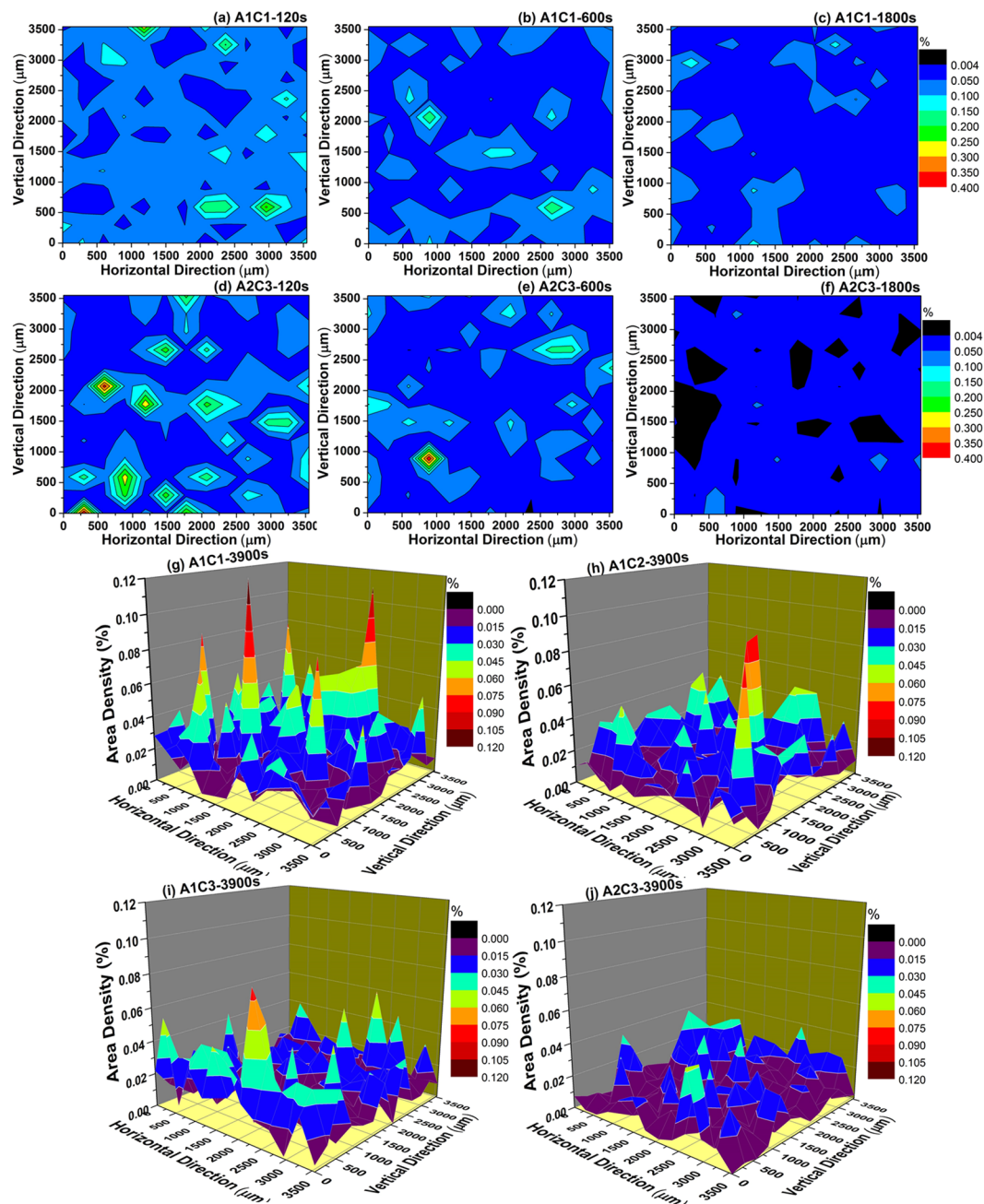


**Figure 5.** Geometric standard deviation of particles as a function of liquid particle percentage and holding time in Fe-O-Al-Ca melts. (a)  $\ln \sigma$  values of particle as a function of liquid particle percentage in Fe-O-Al-Ca melts after deoxidation at 1600 °C for 3900 s; (b)  $\ln \sigma$  values of particle as a function of holding time in experiment A1C1 ([% Al]<sub>i</sub> = 0.05 and [% Ca]<sub>i</sub> = 0.25), A1C3 ([% Al]<sub>i</sub> = 0.05 and [% Ca]<sub>i</sub> = 0.78) and A2C3 ([% Al]<sub>i</sub> = 0.25 and [% Ca]<sub>i</sub> = 0.78).



**Figure 6.** Inter-surface distance of particles in Fe-O-Al-Ca melts. (a) Cumulative frequency of inter-surface distance of adjacent particles  $D_{mi}$  in steels with different amount of Ca addition after deoxidation 3900 s and  $D_{mi}$  is obtained based on Eqs (7 and 8). (b) Cumulative frequency of  $D_{mi}$  in steels with holding time in experiments A1C1 ([% Al]<sub>i</sub> = 0.05 and [% Ca]<sub>i</sub> = 0.25) and A1C3 ([% Al]<sub>i</sub> = 0.05 and [% Ca]<sub>i</sub> = 0.78). (c) Average distance of particles  $D_{AV}$  as a function of liquid particle percentage in Fe-O-Al-Ca melts after deoxidation for 3900 s and  $D_{AV}$  is obtained based on Eqs (7-9). (d)  $D_{AV}$  values of particles as a function of holding time in experiments A1C1 ([% Al]<sub>i</sub> = 0.05 and [% Ca]<sub>i</sub> = 0.25), A1C3 ([% Al]<sub>i</sub> = 0.05 and [% Ca]<sub>i</sub> = 0.78) and A2C3 ([% Al]<sub>i</sub> = 0.25 and [% Ca]<sub>i</sub> = 0.78).

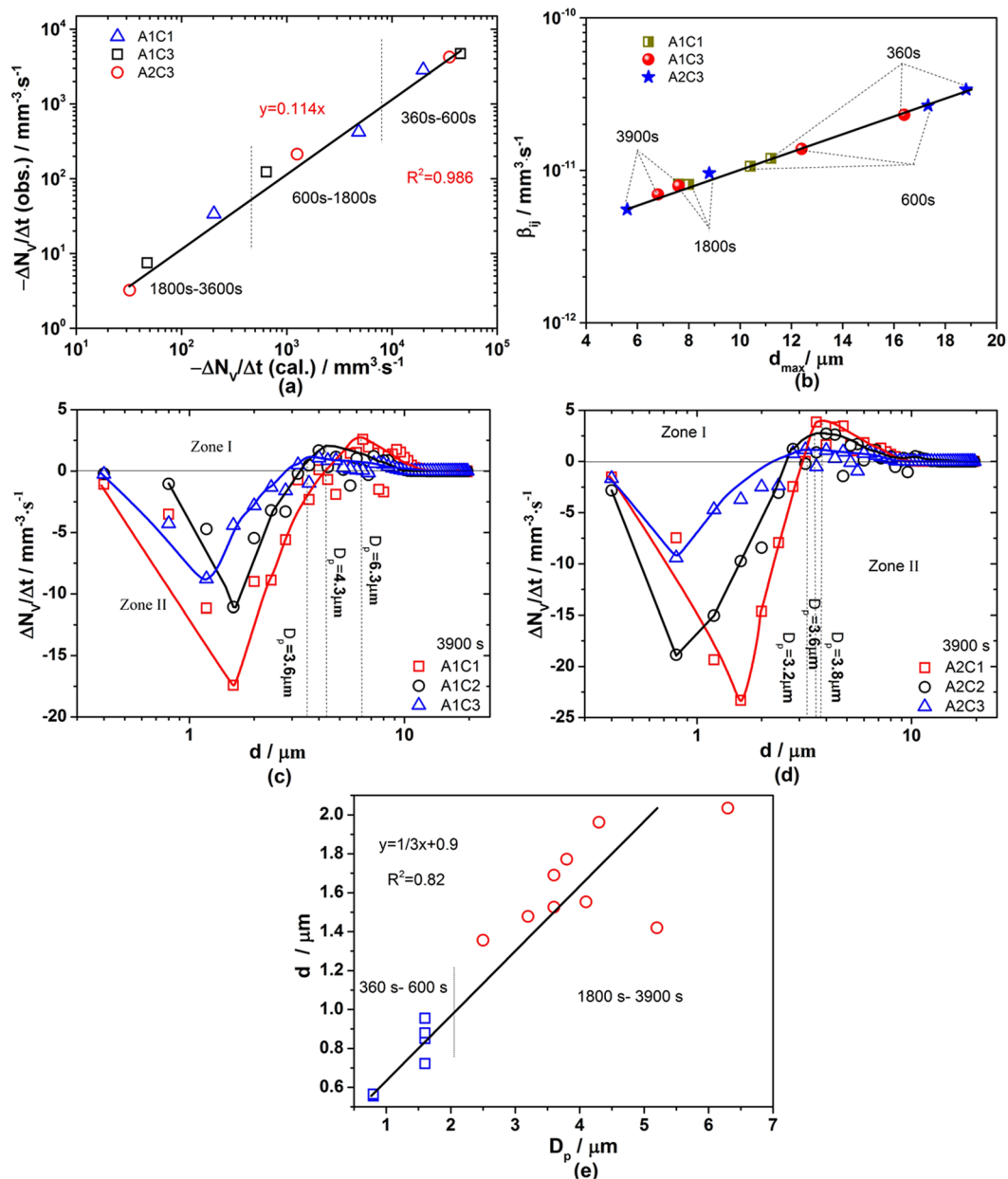
which means that the particles were in larger inter-surface distance with more calcium addition. Figure 6(b) shows that the proportion of particles with close inter-surface distance ( $<10 \mu\text{m}$ ) accounted for 40%, 25% after deoxidation for 360 s in experiments A1C1 and A1C3, and it reduced to 20% and 8% after deoxidation for 600 s.



**Figure 7.** Distribution of area density for particles on the cross section in Fe-O-Al-Ca melts, counting from SEM images.

The inter-surface distance between farthest particles increased with time elapsed. The average inter-surface distances of particles in Fig. 6(c,d) show that the  $D_{AV}$  values (average inter-surface distance of particles in certain region of sample) decreased with the increasing liquid particle percentage when it was larger than 5%, and they increased with time elapsed. It is noteworthy that the change rule of particle inter-surface distance is contracted with that of particle number density, indicating that the larger the number density of particle is, the closer the particles are.

**Distribution of particles.** The distribution of area density for particles in Fe-O-Al-Ca melts as a function of holding time and amount of Al and Ca addition is displayed in Fig. 7. The segregation of particles were more serious in Fe-O-Al-Ca melts containing high Ca at early stage of deoxidation due to high number density which enhanced the collision and coagulation of particles, resulting in larger particles. It can explain the phenomenon that the size of largest particle increased in the order of A2C3 > A1C3 > A1C1 as shown in Fig. 4(b). With time elapsed, the area density for particles in steel decreases due to the floatation of particles. As the floatation of particle in the liquid steel following stokes behavior<sup>56</sup>, the particles with larger size have higher ascending velocity,



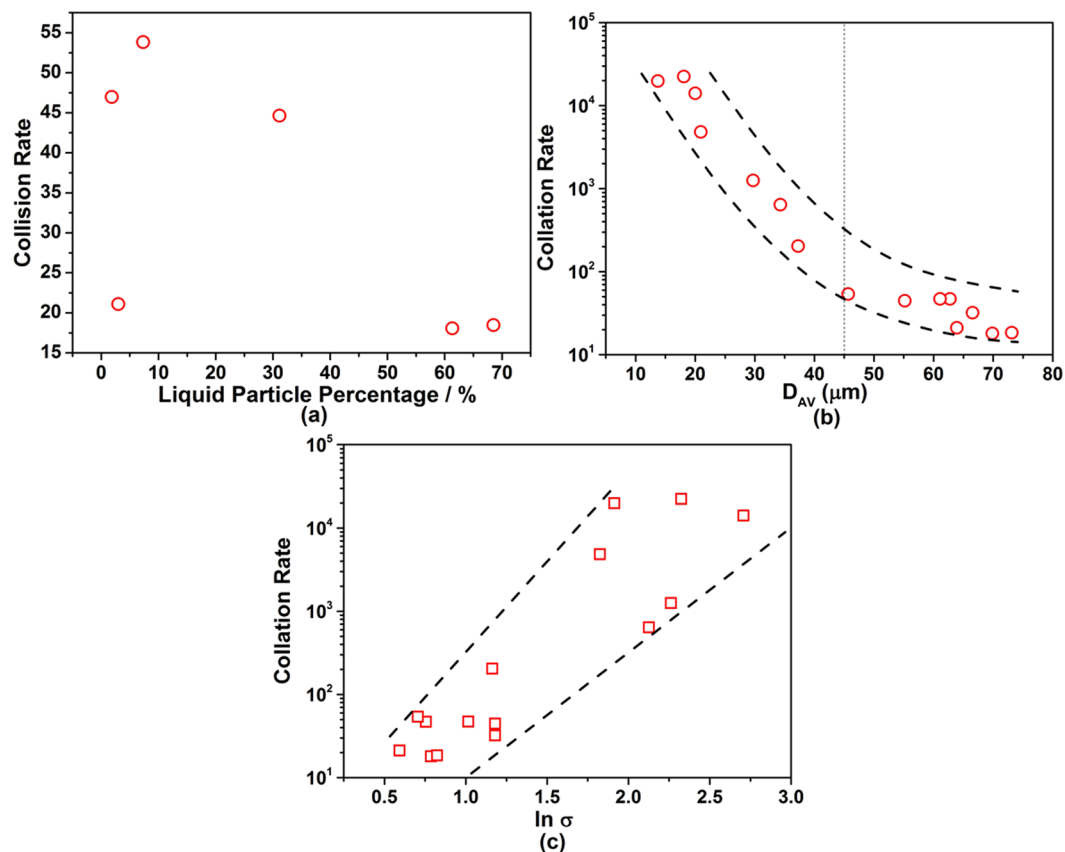
**Figure 8.** Collision rate of particles in Al-Ca deoxidation steel. (a) Experimental change rate of number density is obtained by measuring the total number of particles in Fe-O-Al-Ca melts with holding time and plotted against calculated collision rate of particles based on population balance model; (b) collision frequency for particles as a function of maximum size of particle in Fe-O-Al-Ca melts during deoxidation process based on Eq. (3); (c) Collision rate of particles with different size in melts with 0.05% Al addition after deoxidation for 3900 s; (d) Collision rate of particles with different size in melts with 0.25% Al addition after deoxidation for 3900 s; (e) Relationship between arithmetic mean diameter of particles with  $D_p$  value and  $D_p$  value is the size of particles corresponding to the peak value of curves in Fig. 6c,d.

and thus, in the high Ca containing steel, more particles with large size were removed after deoxidation for 1800s which resulted in lower area density for particles at later stage of deoxidation. It can also be verified in Fig. 7(g-j) and it indicates that the particles distribute more homogeneously and their area density decreased with the increasing amount of Ca addition, resulting in relatively fine particle in melts with high Ca addition, corresponding to Fig. 4(d).

## Discussion

**Collision of particles.** As result shows, the characteristics of particles in Fe-O-Al-Ca melts are dependent on their collision and coagulation behavior. The collision rate of particles in the liquid steel can be estimated as population balance model for collision<sup>44,59</sup>:





**Figure 9.** Change rule of collision rate for particles in Fe-O-Ca-Al melts. (a) Collision rate for particles as a function of liquid particle percentage; (b) Collision rate for particles as a function of average inter-surface distance between particles; (c) Collision rate for particles as a function of geometric standard deviation of particle size distribution.

$$\frac{dn_i}{dt} = \frac{1}{2} \sum_{k=1}^{i-1} (1 + \delta_{k,i-k}) \beta_{k,i-k} n_k - n_i \sum_{k=1}^{i-1} (1 + \delta_{i,k}) \beta_{i,k} n_k \quad (5)$$

Where  $dn_i/dt$  is collision rate of particles ( $\text{mm}^{-3} \cdot \text{s}^{-1}$ ),  $n_i$  is the number of size  $i$  particles per unit volume ( $\text{mm}^{-3}$ ) and  $\beta_{ij}$  is the collision frequency between size  $i$  and size  $j$  particles ( $\text{m}^3 \cdot \text{s}^{-1}$ ).  $\delta_{i,k}$  is the Kronecker delta<sup>60</sup>,  $\delta_{i,k} = 1$  for  $i = k$ , and  $\delta_{i,k} = 0$  for  $i \neq k$ . When  $i = 1$ , the Equation simplifies to

$$\frac{dn_1}{dt} = -n_1 \sum_{k=1}^{i_M} (1 + \delta_{1,k}) \beta_{1,k} n_k \quad (6)$$

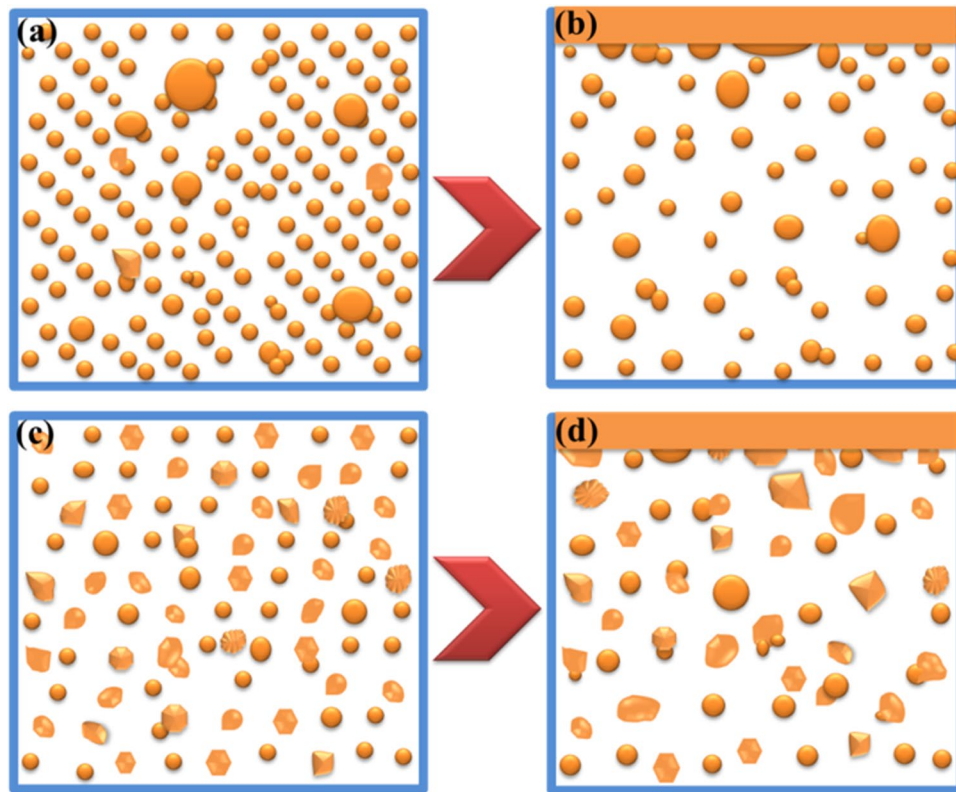
In this experiment, only Brownian collisions and Stokes collisions happened among the particles in Fe-O-Al-Ca melts, and it's not necessary to consider the turbulent collisions without external stirring condition. Therefore, the collision frequency  $\beta_{ij}$  between size  $i$  and size  $j$  particles can be estimated as:

$$\beta_{ij} = \beta_{ij}^S + \beta_{ij}^B \quad (7)$$

$$\beta_{ij}^S = \frac{2g\pi(\rho_{Fe} - \rho_{MxOy})(r_i + r_j)}{9\mu} \quad (8)$$

$$\beta_{ij}^B = \frac{2kT(r_i + r_j)^2}{3\mu r_i r_j} \quad (9)$$

Where  $\beta_{ij}^S$  is Stokes collisions as a result of the difference in ascending velocity of particles and clusters in the liquid steel ( $\text{m}^3 \cdot \text{s}^{-1}$ ),  $\beta_{ij}^B$  is Brownian collisions as a result of random movements of particles in the melt ( $\text{m}^3 \cdot \text{s}^{-1}$ ), and  $\mu$  is the dynamic viscosity of steel ( $=0.006 \text{ kg/m} \cdot \text{s}$ ).



**Figure 10.** Schematic diagram of particle coarsening in Fe-O-Al-Ca. (a) Characteristics of particles for melts containing high Ca at early stage of deoxidation; (b) Characteristics of particles for melts containing high Ca at later stage of deoxidation; (c) Characteristics of particles for melts containing low Ca at early stage of deoxidation; (d) Characteristics of particles for melts containing low Ca at later stage of deoxidation.

The experimental change rate of particle number density ( $-\Delta N/\Delta t$ ) increases monotonically with collision rate of particles in Fig. 8(a). The observed values of  $-\Delta N/\Delta t$  are about 1/9 of calculated collision rate which indicates that not all the particles will coagulation after collision. Compared with the total collision rate in the steel containing low calcium, it is higher in the case of high calcium during the first 600 s, while becomes lower at the later stage of deoxidation process. Moreover, the collision rate of particles decreases with time elapsed which is attributed to a decrease of number density. Figure 8(b) illustrates that the size of largest particle in each sample increases with an increased collision frequency  $\beta_{ij}$  which decreases with time elapsed. It is verified that the collision behavior of particles affects their size significantly.

Figure 8(c,d) represents the collision rate of particles with different size at 3900 s. Zone I and zone II in Fig. 8(c,d) represent the particles with certain size increase and decrease, respectively. The particles with small size ( $<3 \mu\text{m}$ ) decrease and those with large size ( $>3 \mu\text{m}$ ) increase; the particle change rate of number density for particles, *i.e.* absolute value  $\Delta N/\Delta t$ , increases with the increasing size of particles, and then decreases with further increase of particle size in both zone I and zone II. Comparing the curves in Fig. 8(c,d), it is found that the peak values of curves and the size of particles corresponding to those increase with the decrease of the valley values, which means that, the more the small particles reduce, the more the large particles form by collision and the bigger the produced particles with largest number density are. Furthermore, it is found that the size of particles corresponding to the peak value of curves,  $D_p$ , decreases with the increase of Ca addition, which is the same with the changing trend of arithmetic mean diameter of particles observed in experiments. Hence,  $D_p$  is plotted with arithmetic mean diameter of particles in Fig. 8(e). It seems that the  $D_p$  values go up with an increase of the arithmetic mean diameter of particles linearly, especially at later stage of deoxidation process, while they change little with the increase of during the first 600 s of deoxidation process. It indicates that the collision between particles affects the coarsening of the particles at later stage of deoxidation but not early stage under the condition of no stirring, which is in agreement with the conclusion in previous study<sup>36</sup>.

**Influencing factors on collision of particles.** The effect of liquid particle percentage, average inter-surface distance between particles and  $\ln \sigma$  values on the collision rate for particles are summarized in Fig. 8. The changing trend of collision rate with liquid particle percentage is the same with number density in Fig. 4(d) and contrary to inter-surface distance in Fig. 6(c), indicating that the collision rate is mainly affected by particle number density and the distance of particles; as particle number density increases, the inter-surface distance between particles decreases and then the collision rate increases. Figure 9(b) shows an obvious decrease in collision rate with increasing  $D_{AV}$  values from 10 to 45  $\mu\text{m}$ ; however, with further increasing  $D_{AV}$  values, the collision rate exhibits a slight decrease. The change of collision rate with  $\ln \sigma$  values in Fig. 8c shows that the collision rate

increases with the increase of  $\ln \sigma$  values, which means that the collision rate of particles with broad spreading of size distribution is higher than that with narrow spreading of size distribution.

**Coarsening mechanism of particles.** The coarsening mechanism of particles in Fe-O-Al-Ca melts can be summarized in Fig. 10. The collision and coagulation of particles start after their nucleation and continue during whole deoxidation process. In the case of low nucleation rate and small inter-surface distance, the collision and coagulation tend to occur more easily, and hence, the maximum size of particle is larger in early stage of deoxidation as shown in Fig. 10(a) which is in agreement with the experimental result that the size of largest particle is larger in the steel containing higher Ca. Nevertheless, it is verified in our previous study<sup>36</sup> that the average size of particles is mainly dependent on the Ostwald growth. As the rapid rise of liquid particles with large size and fractal dimension in Fig. 10, the inter-surface distance between particles in high Ca-containing melts becomes large and it is larger than that in low Ca-containing melts as in Fig. 10(b–d). With the consumption of Ca, Al and O, the coarsening of particles is mainly affected by their collision, but not Ostwald growth at later stage of deoxidation process. Therefore, the size of particles decreased with an increase of Ca addition after deoxidation for 3900 s.

## Conclusion

The behavior and characteristics of particles in the Fe-O-Al-Ca melts under the condition of no external stirring at 1600 °C was investigated using HT-CSLM, SEM-EDS detection. Most aggregation and coagulation observed between calcium aluminate particles were caused by collision. The characteristics of particles in three-dimensional, *i.e.* size, number density, volume fraction, spreading of particle size, inter-surface distance and distribution based on stereological analysis indicate that their coarsening is not only dependent on Ostwald growth as studied in previous study, but also collision and coagulation, and floatation. The collision of particles affects the maximum size of particle during whole deoxidation process and dominates the coarsening of particles at later stage of deoxidation. The calculated result based on population balance model indicates that the collision rate of particles increases with an increase of their number density, *i.e.* decrease of inter-surface distance, and it is high in the case for particles with narrow spreading of size distribution which is affected by nucleation rate. The particles with relatively larger size and fractal dimension have higher ascending velocity, resulting more fine particles with large inter-surface and low collision rate left in the melts. This mechanism can be used to explain that the collision, coarsening behavior and characteristic change of particles in melts with different amounts of Ca addition.

## Data Availability

The data that support the findings of this study are available from Linzhu Wang upon reasonable request.

## References

1. Wu, C. *et al.* Precipitation phenomena in Al-Zn-Mg alloy matrix composites reinforced with B4C particles. *Scientific Reports* **7**, 9589 (2017).
2. Gao, Q. *et al.* Precipitates and Particles Coarsening of 9Cr-1.7W-0.4Mo-Co Ferritic Heat-Resistant Steel after Isothermal Aging. *Scientific Reports* **7**, 5859 (2017).
3. Godec, M. & Skobir Balantič, D. A. Coarsening behaviour of M23C6 carbides in creep-resistant steel exposed to high temperatures. *Scientific Reports* **6**, 29734 (2016).
4. Wang, Q., Zou, X., Matsuura, H. & Wang, C. Evolution of Inclusions during 1473K Heating Process of EH36 Shipbuilding Steel. *Metall. Mater. Trans. B* **49**, 18–22 (2018).
5. Murakami, Y. Effects of Small Defects and Nonmetallic Inclusions on the Fatigue Strength of Metals. *JSME*. **32**, 167–180 (1989).
6. Hossein Nedjad, S. & Farzaneh, A. Formation of fine intragranular ferrite in cast plain carbon steel inoculated by titanium oxide nanopowder. *Scripta Mater.* **57**, 937–940 (2007).
7. Wang, L., Yang, S., Li, J., Liu, W. & Zhou, Y. Fatigue Life Improving of Drill Rod by Inclusion Control. *High Temp. Mater. Processes* **35**, 661–668 (2016).
8. Zhang, L., Wang, Y. & Zuo, X. Flow Transport and Inclusion Motion in Steel Continuous-Casting Mold under Submerged Entry Nozzle Clogging Condition. *Metall. Mater. Trans. B* **39**, 534–550 (2008).
9. Li, W., Wang, P., Lu, L. & Sakai, T. Evaluation of gigacycle fatigue limit and life of high-strength steel with interior inclusion-induced failure. *Int. J. Damage Mech.* **23**, 931–948 (2014).
10. Kikuchi, N., Nabeshima, S., Kishimoto, Y. & Sridhar, S. Micro-structure Refinement in Low Carbon High Manganese Steels through Ti-deoxidation&mdash;Inclusion Precipitation and Solidification Structure. *ISIJ Int.* **48**, 934–943 (2008).
11. Shim, J. H. *et al.* Ferrite nucleation potency of non-metallic inclusions in medium carbon steels. *Acta Materialia* **49**, 2115–2122 (2001).
12. Babu, S. S. & David, S. A. Inclusion Formation and Microstructure Evolution in Low Alloy Steel Welds. *ISIJ Int.* **42**, 1344–1353 (2002).
13. Zou, X., Sun, J., Matsuura, H. & Wang, C. *In Situ* Observation of the Nucleation and Growth of Ferrite Laths in the Heat-Affected Zone of EH36-Mg Shipbuilding Steel Subjected to Different Heat Inputs. *Metall. Mater. Trans. B* **49**, 2168–2173 (2018).
14. Zou, X., Zhao, D., Sun, J., Wang, C. & Matsuura, H. An Integrated Study on the Evolution of Inclusions in EH36 Shipbuilding Steel with Mg Addition: From Casting to Welding. *Metall. Mater. Trans. B* **49**, 481–489 (2018).
15. Ma, X. *et al.* A novel Al matrix composite reinforced by nano-AlN(p) network. *Scientific Reports* **6**, 34919 (2016).
16. Yaws, C. L. *Yaws' Thermophysical Properties of Chemicals and Hydrocarbons.* (William Andrew, New York, 2008).
17. Lowe J. H. M. A. Zero inclusion steels. *Clean Steel-Super Clean Steel*, 1995 (1995).
18. D. L. Nonmetallic inclusions in steels. (Science Press, Beijing, 1983).
19. Isobe, K. Effect of Mg Addition on Solidification Structure of Low Carbon Steel. *ISIJ Int.* **50**, 1972–1980 (2010).
20. Sakata, K. & Suito, H. Dispersion of fine primary inclusions of MgO and ZrO<sub>2</sub> in Fe-10 mass pct Ni alloy and the solidification structure. *Metall. Mater. Trans. B* **30**, 1053–1063 (1999).
21. Park, J. S. & Park, J. H. Effect of Mg-Ti Deoxidation on the Formation Behavior of Equiaxed Crystals During Rapid Solidification of Iron Alloys. *Steel Res. Int.* **85**, 1303–1309 (2014).
22. Gao, X. *et al.* Effects of MgO Nanoparticle Additions on the Structure and Mechanical Properties of Continuously Cast Steel Billets. *Metall. Mater. Trans. A* **47**, 461–470 (2016).
23. Adabavazeh, Z., Hwang, W. S. & Su, Y. H. Effect of Adding Cerium on Microstructure and Morphology of Ce-Based Inclusions Formed in Low-Carbon Steel. *Scientific Reports* **7**, 46503 (2017).

24. Guo, A. M. *et al.* Effect of zirconium addition on the impact toughness of the heat affected zone in a high strength low alloy pipeline steel. *Mater. Charact.* **59**, 134–139 (2008).
25. Talas, S. & Cochrane, R. C. Effects of Ti on the morphology of high purity iron alloys. *J. Alloy Compd.* **396**, 224–227 (2005).
26. Díaz-Fuentes, M., Madariaga, I. & Gutiérrez, I. Acicular Ferrite Microstructures and Mechanical Properties in a Low Carbon Wrought Steel. *Materials Science Forum* **284–286**, 245–252 (1998).
27. Bin, W. & Bo, S. *In Situ* Observation of the Evolution of Intragranular Acicular Ferrite at Ce - Containing Inclusions in 16Mn Steel. *Steel Res. Int.* **83**, 487–495 (2012).
28. Ren, Y., Zhang, Y. & Zhang, L. A kinetic model for Ca treatment of Al-killed steels using FactSage macro processing. *Ironmak. Steelmak.* **44**, 497–504 (2017).
29. Park, J. H. & Kim, D. S. Effect of CaO-Al<sub>2</sub>O<sub>3</sub>-MgO slags on the formation of MgO-Al<sub>2</sub>O<sub>3</sub> inclusions in ferritic stainless steel. *Metall. Mater. Trans. B* **36**, 495–502 (2005).
30. Lind, M. & Holappa, L. Transformation of Alumina Inclusions by Calcium Treatment. *Metall. Mater. Trans. B* **41**, 359–366 (2010).
31. Wang, X., Huang, F., Qiang, L. I., Haibo, L. I. & Yang, J. Control of stringer shaped non-metallic inclusions of CaO-Al<sub>2</sub>O<sub>3</sub> system in API X80 linepipe steel plates. *Steel Res. Int.* **85**, 155–163 (2014).
32. Suito, H. & Ohta, H. Characteristics of Particle Size Distribution in Early Stage of Deoxidation. *ISIJ Int.* **46**, 33–41 (2006).
33. Voorhees, P. W. The theory of Ostwald ripening. *J. Stat. Phys.* **38**, 231–252 (1985).
34. Ohta, H. & Suito, H. Effects of Dissolved Oxygen and Size Distribution on Particle Coarsening of Deoxidation Product. *ISIJ Int.* **46**, 42–49 (2006).
35. Suito, H. & Ohta, H. Characteristics of Particle Size Distribution in Early Stage of Deoxidation. *ISIJ Int.* **46**, 33–41 (2006).
36. Wang, L. *et al.* Nucleation and Ostwald Growth of Particles in Fe-O-Al-Ca Melt. *Scientific Reports* **8**, 1135 (2018).
37. Guo, L., Wang, Y., Li, H. & Ling, H. Floating Properties of Agglomerated Inclusion in Liquid Steel. *J. Iron Steel Res. Int.* **20**, 35–39 (2013).
38. Saffman, P. G. & Turner, J. S. On the collision of drops in turbulent clouds. *J. Fluid Mech.* **1**, 16–30 (1956).
39. Lindborg, U. K. T. A collision model for the growth and separation of deoxidation products. *Trans. Metall. Soc. AIME* **242**, 94 (1968).
40. Zhang, L. & Thomas, B. G. State of the art in the control of inclusions during steel ingot casting. *Metall. Mater. Trans. B* **37**, 733–761 (2006).
41. Zhang, J. & Lee, H. Numerical Modeling of Nucleation and Growth of Inclusions in Molten Steel Based on Mean Processing Parameters. *ISIJ Int.* **44**, 1629–1638 (2004).
42. Binder, K. & Heermann, D. W. In *Scaling Phenomena in Disordered Systems*. (Springer US, Boston, MA, 1991).
43. Guo, L., Li, H., Wang, Y. & Ling, H. Applying Fractal Theory to Study Agglomeration of Solid Inclusion Particles in Liquid Steel and Floating Characteristics. *Physics Examination and Testing* **4**, 22–26 (2012).
44. Xu, K. & Thomas, B. G. Particle-Size-Grouping Model of Precipitation Kinetics in Microalloyed Steels. *Metall. Mater. Trans. A* **43**, 1079–1096 (2012).
45. Xuan, C., Karasev, A. V. & Jönsson, P. G. Evaluation of Agglomeration Mechanisms of Non-metallic Inclusions and Cluster Characteristics Produced by Ti/Al Complex Deoxidation in Fe-10mass% Ni Alloy. *ISIJ Int.* **56**, 1204–1209 (2016).
46. Lei, H., Nakajima, K. & He, J. Mathematical Model for Nucleation, Ostwald Ripening and Growth of Inclusion in Molten Steel. *ISIJ Int.* **50**, 1735–1745 (2010).
47. Lei, H. & He, J. Nucleation and Growth Kinetics of MgO in Molten Steel. *J. Mater. Sci. Technol.* **28**, 642–646 (2012).
48. Du Gang, L. J., Wang, Z. B. & Shi, C. B. Effect of Magnesium Addition on Behavior of Collision and Agglomeration between Solid Inclusion Particles on H13 Steel Melts. *Steel Res. Int.* **88**, 1600185 (2016).
49. Mu, W., Dogan, N. & Coley, K. S. Agglomeration of Non-metallic Inclusions at Steel/Ar Interface: *In-Situ* Observation Experiments and Model Validation. *Metall. Mater. Trans. B* **48**, 2379–2388 (2017).
50. Mu, W., Dogan, N. & Coley, K. S. Agglomeration of Non-metallic Inclusions at the Steel/Ar Interface: Model Application. *Metall. Mater. Trans. B* **48**, 2092–2103 (2017).
51. Khurana, B., Spooner, S., Rao, M. B. V., Roy, G. G. & Srirangam, P. *In situ* Observation of Calcium Oxide Treatment of Inclusions in Molten Steel by Confocal Microscopy. *Metall. Mater. Trans. B* **48**, 1409–1415 (2017).
52. Wang, L., Yang, S., Li, J., Zhang, S. & Ju, J. Effect of Mg Addition on the Refinement and Homogenized Distribution of Inclusions in Steel with Different Al Contents. *Metall. Mater. Trans. B* **48**, 805–818 (2017).
53. Ohta, H. & Suito, H. Characteristics of Particle Size Distribution of Deoxidation Products with Mg, Zr, Al, Ca, Si/Mn and Mg/Al in Fe-10mass%Ni Alloy. *ISIJ Int.* **46**, 14–21 (2006).
54. Yin, H., Shibata, H., Emi, T. & Suzuki, M. Characteristics of Agglomeration of Various Inclusion Particles on Molten Steel Surface. *ISIJ Int.* **37**, 946–955 (1997).
55. Verma, N. *et al.* Inclusion Evolution During Modification of Alumina Inclusions by Calcium in Liquid Steel: Part I. Background, Experimental Techniques and Analysis Methods. *Metall. Mater. Trans. B* **42**, 711–719 (2011).
56. Joo, S., Han, J. W. & Guthrie, R. I. L. Inclusion behavior and heat-transfer phenomena in steelmaking tundish operations: part III. applications—computational approach to tundish design. *Metall. Mater. Trans. B* **24**, 779–788 (1993).
57. Yao, W. (University of Science and Technology Beijing, 2016).
58. Gmachowski, L. Calculation of the fractal dimension of aggregates. *Colloids and Surface A*. **211**, 197–203 (2002).
59. Nabeel, M., Karasev, A. & Jönsson, P. G. Formation and Growth Mechanism of Clusters in Liquid REM-alloyed Stainless Steels. *ISIJ Int.* **55**, 2358–2364 (2015).
60. Kelly, J. L. & Hughes, D. S. Second-Order Elastic Deformation of Solids. *Phys. Rev.* **92**, 1145–1149 (1953).

## Acknowledgements

Support of this work by the National Science Foundation of China (Nos 51804086, 51574190, 51704085, 51474079, 51774102, 51574095 and 51564003), Program Foundation for Talents of Guizhou University (No. 2017(05)), Science and technology planning project of Guizhou (No. [2017]5626, [2017]5788), Project with Guizhou Education Department (KY (2015) 334), Talent team of Cooperation Project with Guizhou Technology Department ((2015) 4005), the National Natural Science Foundation of Guizhou Province (No. [2018]1060) and Program Foundation for Talents of Education department of Guizhou Provinc (No. Qian Jiao He [2018]105) are gratefully acknowledged.

## Author Contributions

Lin Zhu Wang and Junqi Li wrote the main manuscript text. Shufeng Yang supervised the investigation and revised the paper. Chaoyi Chen, Huixin Jin and Xiang Li contributed to discussions and analysis of the data.

## Additional Information

**Competing Interests:** The authors declare no competing interests.

**Publisher's note:** Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.



**Open Access** This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license, and indicate if changes were made. The images or other third party material in this article are included in the article's Creative Commons license, unless indicated otherwise in a credit line to the material. If material is not included in the article's Creative Commons license and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this license, visit <http://creativecommons.org/licenses/by/4.0/>.

© The Author(s) 2019