ORIGINAL ARTICLE

Efect of alkaline treatment on mechanical properties of palmyra and S‑glass fber reinforced epoxy nanocomposites

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Received: 18 January 2023 / Revised: 29 March 2023 / Accepted: 2 April 2023 / Published online: 13 April 2023 © Wroclaw University of Science and Technology 2023

Abstract

Although epoxy resins have many advantages, their use needs to be expanded by improving their mechanical properties, including a wide variety of material quality, easy processing, negligible shrinkage due to curing, and good adhesiveness to many forms of fber materials. The research focuses cost-efective utilization of palmyra fber treated with 5% alkali solution and diferent volume percentages of S-glass fberglass incorporated by epoxy resin developed by hand layup technique. The fnal epoxy hybrid composite consists of diferent weight ratios of palmyra/S-glass fberglass as 25:75, 50:50, and 75:25. Infuences of palmyra (treated) fber dispersion quality on density, voids, mechanical and moisture absorption performance of the epoxy hybrid composite is studied by ASTM rule. The elevated output characteristics performance is compared with untreated fber composite. Based on the rule of mixture, composite density is varied and Archimedes' principle measures voids. The alkali treated composite samples showed good tensile stress, fexural and impact strength. While compared to untreated fber composite, the tensile, fexural, and compressive strength of TPF/GF(25:75) composite was improved by 19.58%, 29%, and 14.3%, respectively. The reduced water absorption behaviour was observed on the treated composites. The efect of fber dispersion on the mechanical failure of hybrid composite is studied by SEM analysis.

Keywords Alkaline treatment · Palmyra fber and glass fber · Hybrid nanomaterial · SEM · Water adsorption studies

1 Introduction

Natural fbers as reinforcement nanocomposites are developed in diferent industries, including marine structures, supporting structures, automobile components, and aviation, which are subjected to various forces and stresses. To replace traditional metal components and synthetic composite materials, many scientists are actively experimenting by

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Department of Mechanical Engineering, Saveetha School of Engineering, SIMATS, Chennai 602105, Tamilnadu, India using various fbers and resins to create essential parts for advanced and complex parts [[1\]](#page-12-0). Numerous studies on fber composites have recently focused on the characteristics of glass fber [[2\]](#page-12-1). Therefore, researchers seek environmentally acceptable alternatives and maximize hybrid performance by incorporating natural fbers to replace or reduce synthetic fabrics. The distinct qualities of natural fbers include their superior physical strength, excellent heat transfer characteristics, low density, good acoustic properties, and biodegradability [[3\]](#page-12-2). Organic fbers, including jute, kenaf, papaya, sisal, and palm, have received increasing attention due to their high interfacial adhesion for establishing organic polymeric materials. [[4\]](#page-12-3). Additionally, extensive investigation is aimed at the compatibility of resin composites with various organic polymers continued in this feld. The fbers are mixed with diferent lengths and physically tested using diferent matrix materials to achieve the maximum stifness of the FRC material [[5\]](#page-12-4).

Due to essential properties such as adhesion, heat resistance, excellent biocompatibility, and thermal properties, epoxy resins have emerged as a widely used resource over the past several decades. However, these technologies have

seen signifcant progress and improvement over the past 20 years, which are signifcant today [[6\]](#page-12-5). Recent research has focused on forming epoxy resin composites, although previously excellent resins have signifcantly been strengthened by using and modifying them as composite matrixes. Among the most versatile thermoplastic groups, epoxy resins are used in painting and surface coating, composite materials, wind turbines, buildings, and electrical systems [\[7](#page-12-6), [8](#page-12-7)]. Due to their unique qualities, including strength properties, chemical stability, high thermal stability, and heat transfer adaptability, epoxy resins have been used in many industries over the past century [[6](#page-12-5)]. Epoxy resins-based composites are the most signifcant materials in thermosetting groups for the applications of building, wind energy, composites, electrical, and coatings [\[9](#page-12-8), [10](#page-12-9)]. Epoxies are often hardened and combined with an organic solvent to solidify wet mixtures. The epoxy resin becomes a more durable, long-lasting liquid when a curing agent is added. [[11](#page-12-10), [12](#page-12-11)]. Because previously special polymers have been greatly strengthened by using materials for customization and as a composite matrix, researchers are currently focusing on improving epoxy resin compositions [[13\]](#page-12-12).

Natural fber-reinforced polymer matrix composite was developed by conventional technique and the effect of chemically treated natural fber on water absorption properties was studied experimentally. The investigational results are compared with untreated fiber. The treated fiber offered less moisture absorption than untreated fber [\[14](#page-12-13)]. Epoxy-based polymer composites' fexural characteristics are studied with and without nano clay. The nanoclay facilitates good fexural response compared to conventional epoxy resin [\[15](#page-12-14)]. However, the properties of the composite are decided by weathering process [[16\]](#page-12-15). The polyester and plastic/ rubber composite is synthesized via 549 and 550 Napier composite and studied its efects on the compressive strength of composite [\[17\]](#page-12-16). Vacuum transfer molding developed glass fber reinforced plastic/aluminum laminate hybrid composite tensile and fexural strength is investigated by ASTM standards and found glass fiber offered good tensile and flexural strength compared to base laminate $[18]$ $[18]$ $[18]$. Natural cellulosic fiber gained great potential in polymer matrix composite fabrication due to its superior stifness, easy availability, and low cost [\[19](#page-12-18)]. The polymer-based hybrid composite is prepared by utilizing diferent weight percentages of palmyra fber bonded with glass fber and its efects on mechanical properties and moisture absorption were studied and the results are compared with diferent fber length mat composites.

The functional palmyra is performed enhanced mechanical and moisture behaviour [[20](#page-12-19)]. Poly (lactic acid) hybrid composite is synthesized using Alkali treated sugar palm and glass fber for motorcycle applications. Developed composites' mechanical and flammable behaviour are studied and compared with conventional Acrylonitrile Butadiene Styrene plastics. It performed higher hardness and impact strength of 88.6 HRS and 3.10 kJ/m^2 [\[21](#page-12-20)]. Recently, the mechanical properties of inset with sisal fber (Alkali treated) ASTM standards and its experimental fndings study reinforced polymer matrix hybrid composite are compared to untreated fber reinforced polymer composite.

The results showed that the 5% alkali-treated composite tensile, fexural, and impact strength was increased by 5.21, 9.25, and 5.98 [\[22\]](#page-12-21). However, the thin wall graphene nanoplatelets bonded composite post-yield region is the strength by vibration technique and the yield stage was found using the hyperbolic diferential quadrature method [\[23](#page-12-22)]. Functionally grade CNT fber-embedded polymeric nanocomposite layers were analyzed by porosity-dependent vibration technique and considered for hygrothermal efect [\[24](#page-13-0)]. Compressive strength of nano silica-reinforced concrete via machine learning technique [[25\]](#page-13-1). Integral higher-order shear deformation theory adopted non-linear cylinder bending analysis of single-walled carbon nanotubes was estimated their functionality and the micro thickness direction was measured by micromechanical method [[26\]](#page-13-2). Carbon nanotube-reinforced polymeric composite was developed for doubly curved micro-shell panel applications. The composite's mechanical behaviour is studied with the deformation theory of shear in a curvilinear coordinate system mixed with the approach of the non-classical system $[27]$ $[27]$ $[27]$. The effect of non-linear functionality grade CNT fber on the mechanical properties of composite is experimentally studied for beam applications [[28\]](#page-13-4).

Similarly, free vibration study is made [[29\]](#page-13-5), dynamic analysis of SW-CNT is studied [[30](#page-13-6)]. The computational framework adopted for sandwich doubly curved nanocomposite panels [[31](#page-13-7)]. Frequency simulation system to be adopted for imperfect honeycomb core sandwich disk with multiscale hybrid nanocomposite [\[32](#page-13-8)]. Framework base sandwich disk composite was proposed with higher order mechanics [[33](#page-13-9)], vibration analysis [[34\]](#page-13-10), and stress and strain response with three-dimensional analysis [\[35](#page-13-11)].

The present study aims to enhance the epoxy matrix's mechanical characteristics and water absorption behaviour by adding (diferent ratios) alkali treated Palmyra and S glass fber synthesized through a cost-efective hand layup route. The effect of material dispersion on epoxy composite mechanical failure is analyzed using scanning electron microscope (SEM). According to ASTM standards, the tensile, compressive and fexural strength is measured by INSTRON 3382' universal tensile testing machine. Archimedes' principle evaluates the density and voids of composites. The water absorption test was carried out with diferent (treated) fber lengths of 2, 4, and 6 mm, producing epoxy composite for 7 days. Based on an integrated design process in the future, polymer composites to be fabricated based on optimized parameters such as fber length, material types, treatment methods, and volume fraction.

2 Materials and methods

Epoxy is a conventional name for the epoxide structural group and the essential features of epoxy resins or cured fnished products. Prepolymers and polymers are major responsive elements of epoxy resins, also called polyepoxides. Through catalytic photopolymerization, epoxy resins can interact (cross-link) with themselves or with a wide range of co-reactants such as reactive functional groups, amines, protons (including acid anhydrides), phenolic compounds, ethyl alcohol, and oxidizing agents. These co-reactions are named hardeners, and the cross-linking response is classifed as a curing process. The thermosetting polymers are formed by polyfunctional hardeners or reactions of polyepoxides, often developing the physical, thermal, and wear behaviour of epoxy composites.

This present research uses the palmyra fiber (LY-556) extracted from leaf sheets of the Palmyra Palm and is produced in southern and eastern India. The use of fruit bunch waste in oil palm can reduce the amount of enzyme production compared to alternative sources. The advantages of palm oil fruit bunch fbers include accessibility, biodegradability, ease of fabrication, and high strength. This type of fber has low viscosity, dimensional stability, high material strength, and corrosion resistance when the hardener Aradur HY-951 is added as a matrix material. Tables [1](#page-2-0) and [2](#page-2-1) show the mechanical properties of epoxy resin and palmyra fber.

2.1 Fiber treatment

Manufactured fber reinforcement is alkali-treated for fbers of three diferent weights and lengths. All three fber types were subjected to several treatment periods with 5% NaOH solvent. The alkaline treatment process was conducted for three diferent length fbers of 2, 4, and 6 cm in epoxy composites. The chemicalized treatment removes contaminants from the fber material and ensures molecular orientation. During the alkali treatment, the fbers are thoroughly cleaned with

Table 1 Epoxy resin physical properties

Properties	Unit	Value
Density	kg/cm ³	1200
Tensile strength	MPa	70
Equivalent epoxy weight	g /eq	187
Viscosity at room temperature	cp	12,600
Appearance		Light yellow
Modulus of elasticity	GPa	20

distilled water and then dried at ambient temperature for one day. The fibers were separated into smaller fibers of 2, 4, and 6 cm lengths to fabricate the nanocomposite. This is done by wetting the fiber with the polymer matrix properly and it has better strength than untreated fbers.

2.2 Polymer composites fabrication

Hand lay-up is the most popular technique for fabricating polymer composite materials during manufacturing. Using this process, the fber pieces were reduced to the desired size. The polymer was processed to a dimension of $80 \times 30 \times 5$ mm using a spray-coated mold. A mixture of resin and hardener was placed onto the fber in stages using a hand lay-up method, and successive layers of fber were poured over the surrounding matrix to create the desired thicknesses. The samples were heated for 5 min at 150 °C under 2.5 MPa of axial pressure. The detailed fabrication process and their testing samples are shown in Fig. [1](#page-3-0). The formation of epoxy composites with their percentage of mixed proportions is shown in Table [3](#page-3-1).

3 Mechanical testing

3.1 Density measurement

The physical properties of composite samples, including density, water absorption, and void content, were evaluated according to ASTM standards. Experimental density was calculated for the proposed hybrid polymer composites according to ASTM D792. A digital vernier caliper with a precision of 10^{-2} mm was used to measure the sample dimensions, while an electronic balance of 10^{-4} kg was used to measure the volume of the samples. The experimental and theoretical densities are calculated according to the composition law for the mathematical expression in Eqs. (1) (1) and (2) (2) .

$$
\rho_{\rm exp} = \frac{\text{Mass}}{\text{Volume}} \tag{1}
$$

(UTPF or TPF/S glass fiberglass/Epoxy)

Hand layup process

Developed samples

Fig. 1 Details fabrication layout and its test samples

Table 3 Formation of epoxy-based composites

$$
\rho_{\rm th} = \rho_{\rm PA} V_{\rm PA} + \rho_{\rm G} V_{\rm G} + \rho_{\rm m} (1 - V_{\rm PA} + V_{\rm G})
$$
 (2)

where ρ_m , ρ_m , ρ_m , density of matrix, Palmyra and glass fiber; V_{PA} and V_G volume fractions of Palmyra and glass fiber.

Also, the void content prepared composites are calculated by following expressions

$$
\left(\%\right) = \frac{\rho_{\text{th}} - \rho_{\text{exp}}}{\rho_{\text{th}}} \times 100\tag{3}
$$

polymer composites of initial weight were noted before being submerged in deionized water. The confdence interval for the test conducted following ASTM D570-98 appears to be $60\times10\times4$ mm. The composite specimens were removed from the water after 24 h and the samples were weighed. The percentage of water adsorption is determined using the following mathematical expression

% of water absorption =
$$
\frac{w_f - w_i}{w_i} \times 100
$$
 (4)

where w_f , w_i final and initial weight of the specimen

3.2 Water adsorption

The water absorption percentage of hybrid composites is measured over a wide range at room temperature. The prepared

3.3 Tensile, fexural and compressive test

The UTM cross-head speed of 2 mm/min was carried out for tensile strength and specimens were dimensioned $(300 \times 30 \times 3)$ as per ASTM D3039 standards. Three specimens of tensile strength average values are taken from each proportion of polymer composites. TM conducted fexural properties with 5 kN load cell, strain rate 0.01 mm/mm/ min, cross-head speed-2.56 mm/min on three-point testing according to ASTM D-790-10 and the temperature range 23 ± 1 °C at relative values humidity of $50 \pm 5\%$. Three samples of average values are taken for the standard specimen dimension is $150 \times 23 \times 6$ mm. Similarly, INSTRON 3382's was tested for compressive strength using a 100 kN load cell following ASTM D695-15 standards. The standard dimension of 12.5×6 mm of three specimens was extracted from each prepared polymer composite. The interfacial bonding between fber/matrix reinforcement and fracture surface was characterized after mechanical testing. Experiments were performed using a Quantum Superpositions 200 model with an operating distance of 9.8–12.3 mm and an accelerating voltage of 15 kV. The degraded fber surface was treated to allow efficient electronic conductivity in the hybrid. The mechanical tensile, fexural and compression test sample fxed with the test machine is shown in Fig. [2](#page-4-0).

4 Results and discussions

4.1 Density and void content

The prepared palmyra fber reinforced composites of experimental densities (ρ_e) agree with the theoretical density (ρ_t) . Table [4](#page-4-1) shows the fber reinforcement on epoxy nanocomposites of measured and calculated density and void fraction. A minor diference has been observed due to void content during the material processing. Adding glass fber to the composite material has resulted in signifcant improvement in density, as predicted because glass fber is denser than polyamide fber.

The amount of void space and porosity present during the processing of fber-reinforced resin polymer composites is used to calculate void content. The efect of void formation reduces the strength of the matrix because increasing the void content increases the porosity level. Table [4](#page-4-1) noted that theoretically, estimated densities are not equal to experimentally observed densities. Increasing glass fber (25%, 50%, and 75%) in epoxy composites conduct to increase in theoretical and experimental density. The hand layup process achieves good fber saturation and produces hybrids with minimum void content (less than 5%). Beyond 75% GF fller content, the test density of composite materials may not be

Tensile test

Flexural test

Compression test

Table 4 Experimental density and theoretical density

Fig. 2 Test sample fxed with

test equipment

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further reduced. Thomason et al. observed the same results when the fber/fller concentration exceeded the optimum level. The physical properties of the composite materials automatically decreased. The palmyra fber obtained a maximum void fraction of 3.56% for unreinforced epoxy composites, whereas a lower void fraction was obtained for $EP+25\%$ PF+75% GF epoxy composites. It increases the test density of 25%, 50%, and 75% GF while reducing the void area when combined with palmyra and GF reinforcement and resin.

4.2 Water absorption test

Dimensional instability, cracking and low strength of fberreinforced epoxy composites are analyzed based on water absorption. As a result of water adsorption with diferent volume fractions concerning different fiber lengths for treated fber as shown in Table [5](#page-5-0). A high-precision weighing and balancing machine estimated and considered the initial weight of the sample during the water absorption test. After that, the samples were placed in fresh water and their mass was recorded as the fnal mass of composite specimens. This test was conducted for 7 days. Due to cellulose and naturally hydrophilic chemical composition, natural fbers are susceptible to water absorption. As the cellulose content rises due to the increase in the amount of free hydrophilic segments in the fber, the water absorption of the produced fber will probably increase. From the experimental results, the water absorption range of prepared mixtures is obtained from 0.08% to 0.5%.

4.3 Mechanical properties

4.3.1 Tensile strength

A comparison between the before and after tensile tests of all prepared samples is shown in Fig. [3,](#page-5-1) and the tensile strength of treated and untreated epoxy-palmyra fber was tested. The TS of epoxy-reinforced nanocomposites is increased with an increase in PF and GF compositions [\[14\]](#page-12-13). The tensile strength of alkali-treated epoxy-fber reinforced composites was 70.64 MPa, higher than untreated and palmyra fberreinforced composites. The GF (100%) has shown that the tensile strength is 58.62 MPa, less than that of hybrid polymer nanocomposites(Table [6\)](#page-6-0). The treated fber-reinforced composite (UTSC) showed a higher value than the epoxy composite without fber reinforcement. It demonstrates the crucial role of palmyra fber reinforcement in enhancing the

Table 5 Water adsorption for diferent volume fractions (treated)

Fiber size	Volume fractions of fiber	Initial weight	Final weight	Percentage of absorption
2	$EP + 75\%$ PF + 25% GF	4.121	7.442	80.5
	$EP + 50\% PF + 50\% GF$	5.122	7.010	36.86
	$EP + 25\%$ PF + 75% GF	5.455	7.624	29.76
4	$EP + 75\%$ PF + 25% GF	5.226	7.132	36.47
	$EP + 50\%$ PF + 50% GF	4.956	6.854	38.29
	$EP + 25\%$ PF + 75% GF	4.772	6.854	43.62
6	$EP + 75\%$ PF + 25% GF	4.152	6.442	55.15
	$EP + 50\%$ PF + 50% GF	5.236	6.325	20.80
	$EP + 25\% PF + 75\% GF$	5.324	7.524	41.32

Fig. 3 Schematic representation of before and after tensile test specimens for both treated and untreated specimens (ASTM D3039)

(a) Before tensile

(b) After tensile (treated)

Table 6 Tensile strength and stress value of epoxy composites with palmyra and Glass fber reinforcement

Percentage of reinforce- ments	Tensile strength (MPa)		Stress (MPa)	
			Untreated Treated Untreated	Treated
GF (100%)	58.62		46.58	
PF (100%)	33.52	39.65	29.34	33.24
$EP + 75\%$ PF + 25% GF	42.51	52.31	38.48	41.26
$EP + 50\%$ PF $+ 50\%$ GF	42.51	55.68	34.25	45.32
$EP + 25\%$ PF + 75% GF	52.14	62.35	47.12	53.76

specimen's tensile strength. The alkali-treated palm rindtreated reinforced composite (TPF) showed a tensile strength value of 30 MPa and the calculated value seems to be almost 80% higher than the untreated composite. The results of all tests revealed that in terms of tensile strength, the treated palmyra fber-reinforced nanocomposite was superior to the untreated fber-reinforced composite.

Epoxy with palmyra FRC showed a tensile strength of 42.51 and 52.31 MPa, which was very low compared to other combinations of composites. Comparatively, composites reinforced with 25% palmyra fber and 75% S-glass fber had a higher tensile strength of 60.25 MPa than composites without adding palmyra fber and glass fber reinforcement, as shown in Fig. [4](#page-6-1). All the test results showed that the treated palm fber-reinforced epoxy composite surpassed the untreated fber-reinforced composite in terms of performance. The stress–strain behaviour of hybrid epoxy composites was recorded during the tensile test, as shown in Fig. [5.](#page-6-2)

The values of tensile strength stress value increased to 35–42% due to expansion before fber content decreased during the treatment process. According to the test results, the

Fig. 4 Glass-fber/palm-fber reinforced composites of tensile strength for treated and untreated

Fig. 5 Tensile stress–strain curve of epoxy-reinforced hybrid composites

tensile strength of the prepared materials decreased up to 50 (PF):50 (GF) wt%, before there was a modest improvement of up to 68% for composites including treated fber composites. Due to the integration of fber waste, there was no increase in tensile strength. The critical size, below which the material does not reinforce the matrix, is the smallest size of fber. Larger stresses develop at moderate strains and the stress concentration is not uniform when there are not enough fbers to control the properties of the matrix. However, once the minimum number of fbers required to constrain the matrix is reached, the stress distribution becomes uniform and the fbers begin to reinforce the matrix. The same trend was observed for using natural rubber composites by Hasan et al. [\[36](#page-13-12)]. On the other hand, the size of fber 20–100 mm may be the reason for achieving low strength for untreated and treated fber polymer composites. Very short fber stress never reaches the fracture point, and its only function is to prevent matrix crack propagation. The failure may be occurred by using a large type of fiber. When a fiber cracks, the growing microcrack can no longer be properly intercepted, which reduces the fber's toughness.

4.3.2 Flexural strength

Figure [6](#page-7-0) displays the treated palmyra and glass fiber reinforced epoxy composites' fexural strength of epoxy hybrid composites. Variations in fexural stress–strain behaviour were observed due to diferent proportions of reinforcement. According to Hook's law, the stress is directly proportional to strain for the observed treated and untreated polymer nanocomposite behaviour [\[15\]](#page-12-14). The curves show that all composites failed after initial cracking due to their elastic deformation. The efect of concentrations on untreated

Fig. 6 Flexural strength of epoxy composites with diferent volume fractions of palmyra (PF) and Glass fber (GF) reinforcement

and treated fbers of fexural strength and fexural stress are shown in Table [7](#page-7-1).

It is anticipated that a fracture will start on the tension relatively close to the beam and slowly spread into the longitudinal beam of the structures. According to the composite specimens, there seem to be diferences in the fexural stress–strain behaviour, as shown in Fig. [7](#page-7-2). The structures display typical polymer composite materials' mechanical and physical properties in a fexural stress–strain pattern.GF was obtained as the maximum fexural stress–strain gradient at 45 MPa, followed by $PF25\% + GF75\%$ at 52.34 MPa [\[16](#page-12-15)]. The alkaline formation in the alkaline group attached to the fber-reinforced composites leads to the development of the bonding strength between the fber and matrix. The results demonstrated that alkali treatment improves the fexural and yield strength of Epoxy/Palmyra/Glass fber reinforced composites.

Figure [6](#page-7-0) shows that the increase in hybridization increases the fexural strength from 43.57 to 75.62 MPa. The experimental results reveal a promising increase in fexural properties of PF/GF/epoxy hybrid nanocomposites

Table 7 Flexural strength and stress value of epoxy composites with palmyra and glass fber reinforcement

Percentage of reinforce- ments	Flexural strength (MPa)		Flexural stress (MPa)	
			Untreated Treated Untreated Treated	
GF (100%)	54.27		45.28	
PF (100%)	39.58	36.64	27.56	32.54
$EP + 75\%$ PF + 25% GF	38.62	43.57	24.52	37.58
$EP + 50\%$ PF + 50% GF	44.26	49.52	35.75	42.78
$EP + 25\%$ PF + 75% GF	58.62	75.62	44.26	52.64

Fig. 7 Flexural stress–strain curve of epoxy composites with diferent volume fractions of palmyra (PF) and Glass fber (GF) reinforcement

between untreated hybrid and treated palmyra-reinforced epoxy hybrid nanocomposites [[17\]](#page-12-16).

An alkali treatment was applied to palmyra fber, which benefts the fexural properties of PF/GF/epoxy composites. Additionally, it was discovered that the fiber-to-fiber ratio impacted the qualities and traits of the composites (PF: GF). GF aims to withstand the high load capacity before torsional failures while mixing with PF entirely randomly to improve the high strength and modulus. This may be due to sufficient fiber effectively transferring the stress between the matrix and the reinforcing fibers $[18]$ $[18]$. Flexural properties are improved with an increase in GF content.

The representation of fractured specimens before and after the fexural test for both treated and untreated epoxy composites is shown in Fig. [8.](#page-8-0) Figure [9](#page-8-1) shows the allowable diference between the maximum fexural load for epoxy/ PF/GF hybrid nanocomposites for untreated and treated palmyra fber-reinforced composites. The fexural load-bearing capacity of PF/GF/epoxy hybrid composites was improved by adding untreated and treated PF [\[14](#page-12-13)]. The TPF/GF/epoxy hybrid composites with TPF25% + GF75% fber-to-fber ratio had a higher fexural load of 394.52N compared to UT PF25%+ GF75% of 384.23N. In addition, the fexural load of TPF (25%): GF (75) was more signifcant than the 338.2N fexural applied load of pure glass fber composites. When using the same fiber–fiber ratio for PF/GF/epoxy mixture nanocomposites with PF (75%): GF (25%) fber–fber ratio, the maximum fexural load of TPF increases by 10%, especially in contrast to UTPF [\[20](#page-12-19)]. Compared to UTSPF, with a relatively similar fiber–fiber ratio for PF/GF/epoxy composite materials with a fber–fber ratio of PF (50%): GF (50%), the maximum fexural load of TPF improved by 6%.

The PF outer layer underwent significant transformations due to NaOH alkaline treatment. This increased

(a) Before flexural

(b) After flexural (treated) (c) After flexural (untreated)

Fig. 9 Flexural test of maximum load for treated and untreated fberreinforced epoxy composites

epoxy interfacial adhesion at the interaction and continued to improve interfacial interaction and structural interlocking generated by various minor void spaces on the fber surface [\[19\]](#page-12-18). Crosslinking interactions may also result in higher strength for composites treated with NaOH. Evans et al. experimented on fber-reinforced composites, fnding similar trends of alkaline-treated fber compared to untreated fiber-reinforced epoxy composites. The effect of treated fiber results and comparison with the maximum fexural load during the fexural test are shown in Table [8.](#page-8-2)

4.3.3 Compressive strength

The compressive test analysis of epoxy composites of the test setup shows in Fig. [10](#page-8-3). Compressive test experimental results demonstrate the compressive and stress–strain behaviour of treated and untreated palmyra glass fiber polymer composites samples. Figure [10](#page-8-3) shows the compressive **Table 8** Flexural load value of epoxy composites with palmyra and glass fber reinforcement

Fig. 10 Compressive strength test of epoxy composites reinforced with palmyra and S-Glass fber

strength of glass fber-reinforced epoxy hybrid nanocomposites. Fartini et al. found the compressive strength results and established the non-linear behaviour. A higher value was placed on the factors and variables for compressive yield strength due to the matrix's highly compressed, cracked, and brittle microparticles.

In addition, treated fber composites were found to have slightly larger values than untreated composites. Figures [11](#page-9-0) and [12](#page-9-1) make it clear that both treated and untreated composites constructed entirely of SPF perform poorly compared to composites made from hybridizing palmyra and S-glass fber. However, the treated hybrid epoxy composites have a high compressive strength compared to untreated ones. To substantially dissolve the hydrogen bonds in the cellulose chains, PF is pre-treated with the solution of NaOH. This increases the exterior area of the fber for more excellent responsiveness to benzoylation. Utilizing benzoylation to reduce palmyra fber's hydrophilicity increases the compressive properties of epoxy hybridized composites. It makes palmyra fber more compatible with hydrophobic matrix [\[37\]](#page-13-13).

On the other hand, a signifcant impact was observed from the hybridization of palmyra fber reinforced with glass fber in epoxy composites. This clearly shows that adding glass fber gradually increases the compressive properties. Also, due to GF's high strength and high modulus, it is superior to SPF. Analysis of the results also indicated that the amount of fber-to-fber content signifcantly improves the compressive properties of composites. Maximum tensile stress tensile strength that hybrid PF/GF/epoxy composite materials can withstand during global testing. The stress on the specimen surface at failure is measured before the natural fbers fully crack. The higher tensile strength, 64.25 MPa, was obtained for PF25%:GF75% reinforcement on epoxy hybrid nanocomposites, which was higher than that of with and without palmyra fber reinforced hybrid composites, as observed from

Fig. 11 Compressive strength epoxy composites with diferent volume fractions of palmyra (PF) and glass fber (GF) reinforcement

Fig. 12 Compressive stress–strain behaviour epoxy composites with diferent volume fractions of palmyra (PF) and Glass fber (GF) reinforcement

Table [9.](#page-9-2) The compressive strength and stress value obtained from the experimental tests is shown in Table [9.](#page-9-2)

4.3.4 Compressive stress–strain behaviour

Figure [12](#page-9-1) shows the compressive stress–strain behaviour of glass fber-reinforced epoxy hybrid nanocomposites. The test was conducted by using UTM with a uniform crosshead speed. Figure [12](#page-9-1) showed non-linear behaviour and is consistent with previously published research. According to Fig. [12,](#page-9-1) the curves' compressive strength, compressive failure strain, and concrete strength vary slightly. Although additives impact composite materials' compressive stress, the polymer matrices' stifness is primarily refected in the composite's longitudinal strength properties. According to the compressive stress test results, the prepared specimens primarily experience shear-compression type brittle fracture. Usually, shear and densifcation failure occur at the junction where the load is proposed in composites formed under high stress. Ultimate strength interface failures are observed due to the matrix/fber. When the fber exceeds the ultimate

Table 9 Compressive strength and stress value of epoxy composites with palmyra and glass fber reinforcement

Percentage of reinforce- ments	Compressive strength (MPa)		Compressive stress(MPa)	
	Untreated		Treated Untreated	Treated
GF (100%)	65.21		62.34	
PF (100%)	45.62	54.27	84.62	158.64
EP+75% PF+25% GF	48.62	60.56	105.26	122.46
EP+50% PF+50% GF	52.21	63.24	75.28	110.26
EP+25% PF+75% GF	56.24	64.25	65.26	118.62

compressive stress, it can withstand and may break in shear failure. The second fragment of the uniaxial compressive curve, which appears after the fnal compressive stress, shows the progression of fber failure. Deformation failure, local fber micro-buckling along the elastoplastic matrix, yield rate related to fber micro-buckling, and actual fber tensile and fexural failure during uniform compression are common causes of nanocomposite failure.

Figure [12](#page-9-1) shows that both treated and untreated composites fully reinforced with PF achieved low strength compared to glass fber composites. In contrast, pure TSPF composites have higher compressive strength (64.25 MPa) than pure PF composite materials made only from PF (54.25 MPa). The ultimate tensile yield strength of the purely treated palmyra fber composites is also 52.27 MPa higher than the pure, untreated palmyra fber composites, as seen in Fig. [12](#page-9-1) (42.62 MPa). PF was pre-treated with sodium hydroxide to partially eliminate the formation of hydrogen bonds in the lignocellulose chains. As a result, the surface of the fber is more alkali-reactive.

The tensile, flexural, and compressive properties are improved by using alkaline to reduce the hydrophilic nature of PF. As a result, the PF and hydrophobic matrix are more uniform. In addition, the hybridization of the PF with the GF should also be considered. Figure [12](#page-9-1) shows that adding GF reinforcement to the composite material signifcantly impacts its compressive yield strength and stress–strain behaviour. The maximum yield strength of epoxy hybrid mixed composites of both treated and untreated composites are shown in Table [10.](#page-10-0) However, the addition of GF is highly advantageous to palmyra fber due to GF leads to increases the compressive strength. The analysis of the results also indicates that the amount of fber-to-fber composition signifcantly impacts the improved performance in the compressive and shear properties of the nanocomposites.

4.3.5 SEM analysis

The fber matrix interfacial bonding strength and fractured surfaces were analyzed using scanning electron microscope (SEM) during the compressive and fexural test. It is clear

Table 10 Flexural load value of epoxy composites with palmyra and glass fber reinforcement

Percentage of reinforcements	Compressive yield strength		
	Untreated	Treated	
GF (100%)	63.21		
PF (100%)	42.62	52.27	
$EP + 75\%$ PF + 25% GF	45.62	59.56	
$EP + 50\%$ $PF + 50\%$ GF	53.21	60.24	
EP+25% PF+75% GF	56.24	62.25	

from all nanocomposite SEM specimens that each exhibits the characteristic features of brittle plastic deformation, including streamlined exteriors and flow cracks, demonstrating the composite's weak resistance to sustained loads or stresses implemented to the specimens.

SEM images of all the compression test specimens were somewhat similar, despite the volume fraction of glass fber in the palmyra composite or the alkaline treatment applied to the palmyra fber. Also, increasing the glass fber density in the palmyra/glass fber hybrid nanocomposites from 25% to 75% makes the composites less brittle, resulting in an occasional surface coating. Figure [13](#page-10-1)a–f shows the SEM images of the surface morphology of epoxy hybrid nanocomposites during the compressive test.

Flexural fracture through the composite is easily visible in the SEM images of the fexural test specimen, demonstrating the matrix control over the strength of the composite material. All analyzed specimens showed evidence of palmyra fber and glass fber deformation. This demonstrates that the matrix and fbers have excellent stress transfer with corresponding load distribution. Thermoplastic sugar starch/agar hybrid reinforced composites with seaweed/sugar palm fber, performance comparable to fber failure mode were noted from previous experimental results. Figure [14a](#page-11-0)–f shows the SEM images of the surface morphology of epoxy hybrid nanocomposites during the compressive test (Fig. [15\)](#page-11-1).

5 Conclusion

Hybrid nanocomposites were efectively prepared using alkaline (NaOH) treated palmyra and glass fiber-reinforced epoxy composites with diferent volume fractions.

Fig. 13 Compressive yield strength epoxy composites with diferent volume fractions of palmyra (PF) and glass fber (GF) reinforcement

Fig. 14 a–**f** Compressive test surface morphology of (Epoxy/PF/GF) using a Scanning Electron Microscope (SEM)

Fig. 15 a–f Flexural test surface morphology of (Epoxy/PF/GF) using scanning electron microscope (SEM)

Comparative analysis of fexural and compressive strength concerning their stress–strain behaviour for treated, untreated epoxy composites. The prepared hybrid nanocomposites show high fexural and compressive strength compared to untreated hybrid epoxy composites (UTPF/GF/ Epoxy). With the addition of GF, the PF composite's compressive and fexural characteristics signifcantly improved. Additionally, it was observed that the nanocomposites'

fexural and compressive characteristics improved as the GF volume fraction increased. It has a 25PF:75GF ratio that offers superior compressive (64.25 MPa) and flexural strength (75.62 MPa), particularly compared to other PF/ GF/epoxy composite materials. The measured compressive and fexural properties of the 25TPF:75GF composite are supported by scanning electron microscope (SEM) analysis. Like pure GF, hybrid composites with a 25PF:75GF ratio have similar fexural and compressive properties. Moreover, the main fndings of the experimental investigation outcome results proved an improved mechanical tensile, fexural, and compressive strength (detailed with stress–strain curves) as well as decreased water absorption of composite developed with Alkali-treated TPF with S-glass fberglass reinforced epoxy composite.

Author contributions All authors contributed to the study's conception and design. Material preparation, data collection and analysis were performed by [NPS], [SS], [RV],and [RM]. The frst draft of the manuscript was written by [NPS] and all authors provided language help, writing assistance and proofreading. All authors read and approved the fnal manuscript.

Funding The authors did not receive support from any organization for the submitted work. No funding was received to assist with the preparation of this manuscript. No funding was received for conducting this study. No funds, grants, or other support were received.

Data availability All the data required are available within the manuscript.

Declarations

Financial interest The authors have no relevant financial or nonfnancial interests to disclose. All authors certify that they have no afliations with or involvement in any organization or entity with any fnancial or non-fnancial interest in the subject matter or materials discussed in this manuscript. The authors have no fnancial or proprietary interests in any material discussed in this article.

Conflict of interest The authors have no competing interests to declare relevant to this article's content.

Ethics approval This is an observational study. Effect of Alkaline pretreatment on Palmyra and S-Glass fber reinforced epoxy Nanocomposites: Mechanical and water adsorption studies, Research Ethics Committee has confrmed that no ethical approval is required.

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