**ORIGINAL ARTICLE**



# **Performance and microstructure analysis of high‑strength concrete incorporated with nanoparticles subjected to high temperatures and actual fres**

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## **Abstract**

Currently, nanoparticles are used as admixtures to reduce the thermal deterioration of concrete after exposure to fre. However, the infuence of high temperature on high-strength concrete (HSC) containing silica fume and nanoparticles has not been investigated well. In this study, various HSC mixes incorporated with 1%, 2%, 3% and 4% nanosilica (NS) or 1% and 2% nanoferrite (NF) were prepared to produce HSC with high enduring strength after being subjected to high temperatures of up to 800 °C and actual fres. The specimens were assessed via scanning electron microscopy, compression and splitting tensile tests, modulus of elasticity test, and water permeability coefficient analysis. Results showed that using NS and NF percentages of up to 3% and 2%, respectively, in HSC improved the mechanical properties and water permeability coefficient at elevated temperatures. The compressive strength of the heated specimens with 3% NS was better than those with  $2\%$  NF at temperatures 200 °C–800 °C. With regard to the microstructure feature, the results confirmed that NS acted as an adequate flling material, which produced a condensed microstructure with extra compressed hydration outputs. This may be associated to higher pozzolanic reaction of NS with high distribution that formed additional calcium silicate hydrate gel. The specimens with 3% NS had no cracks until the temperature of 800 °C, but their porosity increased slightly.

**Keywords** Nanosilica · Fire · High temperature · Mechanical properties · Microstructure

## **1 Introduction**

Nanosilica (NS) and nanoferrite (NF) have elicited the most interest amongst all nanomaterials used to improve the properties of concrete  $[1]$  $[1]$ . This interest may be due to their higher reactivity compared to other nanomaterials

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[[2](#page-13-1), [3\]](#page-13-2). NS and NF serve as a nanofller material stufing the pores between calcium silicate hydrate (CSH) gel particles and refning the microstructure [\[4–](#page-13-3)[7](#page-13-4)]. NS has a high pozzolanic activity and reacts with calcium hydroxide (CH), thus generating abundant CSH, resulting in a condensed cement matrix and increasing concrete strength and durability [[8\]](#page-13-5). On the other hand, NF has a physical effect, and it operates as nuclei to speed up the hydration process of the cementitious material to produce CSH; it also reacts with CH and forms calcium ferric hydrate (CFH) gel [[9\]](#page-13-6). The addition of a small amount (up to 3%) of either NS or NF enhances the bonding between the hardened cement paste (CP) matrix and aggregates in the interfacial transition zone (ITZ) [[10](#page-13-7)]. Many researchers have comprehensively investigated their infuence on the hardening properties of concrete. Sumesh et al. [[11](#page-14-0)] examined the efects of nanoparticles on the hardening, durability and bonding properties of cement mortar and geopolymer paste and reported that the use of nanomaterials with efective particle sizes increases the efficiency and strength of cement mortar. Li et al.  $[12]$ 

reported that the addition of nanoparticles can improve the mechanical characteristics of concrete. Furthermore, concrete containing nano-TiO<sub>2</sub> has better abrasion resistance compared with concrete with an equivalent amount of nano-SiO<sub>2</sub> [[13](#page-14-2)]. With the advancement of nanotechnology,  $SiO<sub>2</sub>$  can now be produced in nano-sized particles with predictable size ranges. The efect of the size of  $SiO<sub>2</sub>$  elements on concrete strength was investigated by Schoepfer and Maji [[14](#page-14-3)]. Compressive strength tests revealed a signifcant increase in strength when particle size of  $SiO<sub>2</sub>$  is reduced to 12 nm. Researchers have been working to enhance the durability and long-term sustainability of concrete, and considerable improvement in the mechanical characteristics of cementitious materials has been achieved through the integration of nanoparticles [[15](#page-14-4), [16](#page-14-5)]. Nano-modification may be used to develop a cementitious material that is highly durable, strong and environmentally friendly [[17\]](#page-14-6). Nanoparticles can reduce cement use and convert cement-based goods into electric/ thermal sensors or crack-repairing materials because of their reinforcing action  $[18, 19]$  $[18, 19]$  $[18, 19]$  $[18, 19]$  $[18, 19]$ . The addition of ultrafine nano-elements aids in the reduction of cement content by moderately substituting cement by weight ground to increase the binding impact [\[7](#page-13-4), [17,](#page-14-6) [20\]](#page-14-9). Ultrafne nanomaterial particles also help reduce micropore development by serving as a fller agent, resulting in highly dense concrete that automatically limits micropore growth in ultrahighstrength concrete (UHSC) structures [\[21](#page-14-10), [22\]](#page-14-11). Previous studies demonstrate that nano-silica can improve concrete's resistance to high temperatures, along with increase in the mechanical properties of high-strength concrete at high temperatures [[23](#page-14-12), [24](#page-14-13)]. The presence of nanosilica in mixtures increases resistance to elevated temperatures that is evident as the concrete, after exposure to elevated temperatures maintain most of its compressive strength (*fc*). Inclusion of Nano silica also helps reduce crack length compared with plain or even blended mixtures containing other supplementary materials [\[25](#page-14-14), [26](#page-14-15)]. At temperatures of 200 °C to 400 °C, abundant CSH is produced as a result of the high pozzolanic activity of nanosilica, and anhydrous cement grains are hydrated during the internal autoclaving process [[27,](#page-14-16) [28\]](#page-14-17). Moreover, the pore size distribution in the matrix at high temperatures is reduced by increasing the nanosilica content [[29](#page-14-18)]. The resistance of concrete to elevated temperatures of up to 400 °C can be enhanced by increasing the nanosilica content, but an opposite effect occurs at temperatures exceeding 400 °C [[30](#page-14-19), [31\]](#page-14-20). Furthermore, the incorporation of NS delays the deterioration of fred concrete by reducing the thermal conductivity of concrete [[32\]](#page-14-21). This condition implies that long exposure to fre is essential to attain the required temperature at the core of the concrete members. Thus, the thermal degradation of blended concrete members is delayed [\[33\]](#page-14-22). Spinel  $\text{ZnFe}_{2}\text{O}_{4}$  nanoparticles formulated via the sol–gel process (17.5 nm) have also been added with diferent doses of up to 2% to boost the physico-mechanical and fre resistivity of hardened cement pastes. The optimum addition is 1%  $ZnFe<sub>2</sub>O<sub>4</sub>$  [[34](#page-14-23)].

The previous studies focused only on studying the effects of elevated temperature without analysing real fre situations. Moreover, the literature didn't provide any design and relative cost estimation for a concrete structure incorporating nanoparticles. For the previous issues, the recent study aims to achieve three objectives. The frst is to examine the efect of elevated temperature of silica fume (SF) on high-strength concrete incorporated with NS or NF under elevated temperatures up to 800 °C. The second aim is to investigate the fre resistance of specimens subjected to 600 °C heat under an initial cracking load of control mix for one hour. The third is to make a cost estimation for designing reinforced concrete columns using every mix. Nine mixes were prepared to study the efect of adding diferent percentages of NS and NF. Their results were compared with those of common and SF concrete mixes. Moreover, the efect of increasing the cement content of 3% NS concrete to  $450 \text{ kg/m}^3$  was studied for comparison. Compressive strength  $(f_c)$ , splitting tensile strength  $(f_t)$  and modulus of elasticity  $(E_c)$  were investigated at room temperature. The SEM analysis of the specimens was performed to explain the obtained results and compare it with works of El-Enein et al. and Nili et al. [\[35](#page-14-24), [36](#page-14-25)]

## **2 Experimental work**

#### **2.1 Materials and mix proportion**

Nanosilica (NS) was prepared from rice husk by using hot hydrochloric acid via chemical treatment and burning [\[37](#page-14-26)]. After this treatment, an amorphous powder with 100% purity and an average particle size of 6.25 nm was obtained.  $CoFe<sub>2</sub>O<sub>4</sub>$  nanoparticles (NF) were prepared through the citrate-gel method [[9\]](#page-13-6) with average particle size 29.62 nm and 100% purity. Physical and chemical characterisations of the used nanofllers were performed. Type I ordinary Portland cement (OPC type I 52.5 N) and SF were utilised as binders in the mixes, as shown in Table [1](#page-2-0). The main characteristics of the NS sample were measured using an X-ray difraction (XRD) apparatus, an energy-dispersive X-ray (EDX) spectroscopy device and a transmission electron microscopy (TEM) instrument, as shown in Fig. [1](#page-2-1). Cement was tested in accordance with ASTM C143M-15a [\[38](#page-14-27)]. Natural sand and dolomite were used as fne and coarse aggregates with specifc gravities of 2.67and 2.75 and volume weights of 1.66 and  $1.78 \text{ t/m}^3$ , respectively. The fineness modulus of sand was 2.82, and the NMZ and water absorption of dolomite were 12.5 mm and 1.1%, respectively. The aggregate tests

<span id="page-2-0"></span>**Table 1** Chemical composition and physical properties of OPC, SF, NS and NF

Items	<b>OPC</b>	SF	<b>NS</b>	NF
Chemical composition $(wt\%)$				
SiO <sub>2</sub>	20	97.20	100	$\theta$
$Al_2O_3$	6.25	0.25		
$Fe_2O_3$	3.55	0.54		
CoFe <sub>2</sub> O <sub>4</sub>				100
CaO	62.34	0.21		
MgO	2.12	0.43		
SO <sub>3</sub>	2.42	0.11		
LOI	1.67	0.74		
$K_2O$	0.75	0.45		
Na <sub>2</sub> O	0.81	0.15		
Physical properties				
Specific gravity	3.15	2.15	2.0	5.3
Avg. particle size (nm)	21,000	1000	6.25	29.62



were performed in accordance with ASTM C33/C33M–18 [\[39](#page-14-28)]. In order to achieve good workability of mixes and even distribution of NS in the concrete mix, Sika Visco-Crete 5–930 with a density of 1.08 kg/m<sup>3</sup> was used as the superplasticizer (SP) in accordance with ASTM-C-494 types G and F.

Table [2](#page-3-0) shows the preparation of the nine concrete mixes. The binder (B) involved cement content (C), SF, NS and NF. The cement content and the percentages of sand (S), aggregates (Agg), water (W) and super-plasticizer (SP) "S/ Agg, W/B and SP/B" were kept constant in all the mixes at  $400 \text{ kg/m}^3$  and  $40\%$ ,  $30\%$  and  $4\%$ , respectively. Uniform dispersion of NS and NF in the mixes was not easy because of their high surface energy and low content in the mixture. All these mixes containing NS  $(1-4\%)$  and NF  $(1-2\%)$  were prepared according to the following procedure. First, the Nanoparticles were stirred in water separately in a small mixer at a speed of 120 rpm for 2 min. Next, cement, silica



<span id="page-2-1"></span>

**(c)** Transmission electron microscopy (TEM) Analysis

<span id="page-3-0"></span>**Table 2** Proportions of the designed mixes and their cost ratio with respect to the  $M<sub>o</sub>$  mix



where's  $*$  the content of the Table were in  $(kg/m<sup>3</sup>)$ 

*OPC* cement content (OPC I-52.5 N), *W* water, *SF* silica fume, *D* dolomite as a coarse aggregate, *NS* nanosilica, *S* sand as a fine aggregate, *NF* nano-ferrite, *SP* super plasticizer,  $(Cost/m^3) / M_0$ : cost of 1 m<sup>3</sup> ratio with respect to  $M<sub>O</sub>$  mix, cost of component/OPC: cost of component with respect to local OPC cost (Steel / cement) cost ratio was 13

fume, sand and dolomite were dry mixed at medium speed of 80 rpm for 1 min. Afterwards, keeping the mixer at medium speed, about two thirds of water with nanoparticle was added gradually to the dry constituents for about 30 s. SP was added to the rest of the third mixing water and added gradually while mixing for another 30 s. Finally, the mixture was rested for 30 s and then remixed for 1 min [[37](#page-14-26)]. The specimens were demoulded after 24 h of casting and alleviated in a water tank at ambient temperature until the test dates.

## **2.2 Specimens and testing**

#### **2.2.1 Specimens**

Nine mixes were designed as illustrated in Table [2](#page-3-0). For each mix, three cubes of dimensions  $100 \times 100 \times 100$  mm were prepared for each of the six curing ages to test for compressive strength. Three cylindrical specimens of height 300 mm and diameter of 150 mm were cast for each mix to test modulus of elasticity and tensile strength. For water permeability test, three cylindrical specimens of height 150 mm and diameter of 150 mm for each mix.

#### **2.2.2 Mechanical and physical testing**

A compression test was performed after 7–180 days of aging by using  $100 \times 100 \times 100$  mm cubes. The test was carried out in accordance with BS 1881–116: 1983. A splitting tensile test was performed on the 28th day by using  $150 \times 300$  mm cylinders and carried out in accordance with BS 1881–117: 1983. Static modulus of elasticity (E) compression was applied on the 28th day by using  $150 \times 300$  mm cylinders and carried out in accordance with ASTM C469-65 (1975). A water permeability test was conducted on the 28th day. For a five-hour period, the concrete specimens were subjected to a hydrostatic water fow of 30 bars. The permeability coefficient  $(K)$  was calculated based on Darcy's law.

#### **2.2.3 Durability testing**

The durability of concrete mixes was investigated through high elevated temperature testing and actual fire exposure. For elevated temperature, the specimens were heated in an electric furnace at fixed temperatures with exposure time of for 2 h. The volume of electrical furnace was of  $(250 \times 400 \times 350)$  mm with heating rate of 20 °C /min and maximum achievable temperature of 1200 °C. The heated specimens were left to cool down at ambient temperature until it reaches room temperature as shown in Fig. [2](#page-4-0) [[23](#page-14-12)]. A fire test was carried out using  $100 \times 100 \times 100$  mm cubes. Two German thermocouples (type J) were fixed during the casting of the cubes. The first was imbedded in the middle of a cube to measure the inner temperature, and the other was fixed on the surface of the cube to measure the surface temperature. The cubes were cured for 28 days, removed from the curing tank and allowed to dry at ambient temperature for about 7 days before testing. The fire test was conducted whenever the specimens were loaded with the initial cracking load in accordance with ASTM E119 [[40](#page-14-29)]. The firing tool was designed according to the standard fire test curves of ASTM-E119, and its time–temperature relation was compatible with ISO 834–11. All specimens were loaded with a constant load of 70 KN representing the initial



<span id="page-4-0"></span>**Fig. 2** Heating period of specimens per min

crack load of the control mix. Each specimen was loaded using a hydraulic jack machine with a capacity 10,000 psi, an accuracy of 100 psi and a radius of 98 mm. Each specimen was inserted inside the firing tool. The thermocouples were connected to a voltmeter to measure the output voltage, which was converted into temperature in Celsius throughout the thermocouple curve. Firstly, each specimen was loaded with the predetermined load (70 KN) and kept constant during firing. Secondly, the temperature was raised to 600 °C on the surface of the cube at a rate of 50 °C/min then maintained at 600 °C for an hour under loading. Lastly, the fire was extinguished, and the load on the specimen was gradually increased until failure occurred. The  $f_c$  value was recorded, as shown in Fig. [3.](#page-4-1)

#### **2.2.4 SEM analysis**

Samples for SEM analysis were taken from the centre of cubes, grinded and adhered on aluminium stub using adhesive carbon tape. Samples were sputter coated with gold for 1 min. Then, examination was performed using SEM (JEOL, JSM IT-100) at 20 kv.

<span id="page-4-1"></span>The heated specimens were exposed to various temperatures (100 °C—800 °C) for all mixes after curing for 28 days.



**Fig. 3** Firing of a loaded specimen

#### **2.2.5 Cost estimation of concrete column**

An economic assessment of one cubic meter of the concrete mixes was performed and the cost of the materials was obtained from the seller's reports. The following formula was used based on the seller's reports.

$$
Costi = (1 \times Ci) + (5 \times SFi) + (200 \times NSi) + (1000 \times NFi) + (0.00225 \times Wi) + (0.051 \times Si) + (0.093 \times Di) + (75 \times SPi) (1)
$$

where  $Cost_i$  is the cost of 1 m<sup>3</sup> of produced concrete blend *i*,  $C_i$  is the cement content of blend *i*,  $SF_i$  is the SF content,  $NS<sub>i</sub>$  is the NS content,  $NF<sub>i</sub>$  is the NF content,  $W<sub>i</sub>$  is the water content,  $S_i$  is the sand content,  $D_i$  is the dolomite content and  $SP<sub>i</sub>$  is the SP content. All of the components of concrete mix *i* were measured in  $\text{kg/m}^3$ .

Moreover, an economic assessment of the designed column of concrete mixes was performed. The cost of the reinforced column of mix i was evaluated by assuming a load of 10,000 KN. The quality control of nanoparticle concrete mixes must be excellent. Thus, the confidence percentage and coefficient of variance (*V* %) were assumed to be 95% and 5%, respectively. The designed compressive strength  $(f<sub>cu</sub>)$  of mix i at 28 days of age was calculated with Eq. [2](#page-5-0) and designed area of the concrete and the area of the steel column were calculated with Eq. [3](#page-5-1) where  $\alpha_i$  was derived from Eq. [4.](#page-5-2) The  $\beta_i$  and the factor of safety of concrete  $(\gamma_c)$  and steel  $(\gamma_c)$  were taken as 0.67, 1.75 and 1.36. The yield strength of steel  $(f_v)$  and the density of steel  $D_{st}$ ) were 400 MPa and 7.8 ton/ $m^3$ , respectively. The variable parameters in Eqs.  $(2)$  $(2)$ – $(5)$  $(5)$  are tabulated in [Table 3.](#page-5-3)

$$
(f_{cu})_{i} = (f_m)_i * (1 - 1.67 * V)
$$
 (2)

$$
P_u = 0.9 * \left( \alpha_i \beta_i \frac{(f_{cu})_i}{\gamma_c} A_{ci} + \frac{f_y}{\gamma_s} A_{si} \right)
$$
 (3)

<span id="page-5-2"></span><span id="page-5-1"></span>where

$$
\left[\alpha_i = 0.85 - 0.0033 \left(f_{cu} - 60\right)\right] \ge 0.67\tag{4}
$$

$$
Cost_{coll} = (cost_i * A_{ci} + A_{si} *_{st} * 10^3 * cost_{st}) * h_{col*10}^{-6} (5)
$$

## **3 Results and discussion**

#### **3.1 Fresh properties**

Figure [4](#page-6-0) shows the fresh and hard densities of the SF concrete mixes prepared with the incorporation of nanoparticles particles in comparison with the density of the control mix. The figure indicates that the fresh density of the SF concrete mixes incorporated with nanoparticles was less than that of the control mix due to their lesser specifc gravity. Meanwhile, the dry density of all the mixes was greater than that of the control mix due to the high density of the matrix resulting from the highly pozzolanic reaction of NS and the physical role of NF. Both, NS and NF worked as a catalyst for cement hydration and SF pozzolanic manner, thereby producing highly dense concrete [[9](#page-13-6)].

#### **3.2 Water permeability**

<span id="page-5-0"></span>Figure [5](#page-7-0) shows that the water permeability coefficients (K) of  $M_0$  and  $M_{SF}$  mixes were  $4.04 \times 10^{-11}$  and  $1.65 \times 10^{-11}$  cm/sec, respectively. Permeability when



where  $f_{\text{mean}}$ : mean compressive strength at 28 days age,  $f_{cu}$ : characteristic compressive strength, $\alpha$ : factor to substitute slow loading,  $\beta$ : factor to transfer from cube to cylinder,  $A_C$ : cross section of designed column,  $A_S$ : area of steel of designed column (0.01  $A_C$ ), R%: area of concrete and steel of designed column percentage with respect to  $M_0$ , (Cost/ col) /  $M_0$ : cost percentage of designed column with respect to  $M_0$  mix under constant design load 10,000 ton

<span id="page-5-3"></span>**Table 3** Analysis of studied axially loaded short column

<span id="page-6-0"></span>

the NS content was increased up to 4% was found to be  $0.458 \times 10$ –11 cm/sec. The K of  $M<sub>450</sub>$  was greater than that of  $M_{NS4}$ , which means that there is no need to increase the cement content of NS mixes over  $400 \text{ kg/m}^3$ . The compressive strength of the MNS4 mix was better than that of the M450 mix, consequently, confrming the previous result that 4% NS is the best percentage. In addition, the K of 1% NF was larger than that of 1% NS while the K of 2% NF was smaller than that of 2% NS.

#### **3.3 Mechanical properties**

#### **3.3.1 Compressive strength**

Figure [6](#page-7-1) illustrates the  $f_c$  of the nine mixes at various ages between 7 and 180 days. Adding SF improved  $f_c$  up to 35%. Adding NS improved  $f_c$  by up to 4%. This result may be due to the tiny particle size, large surface area, high purity and high pozzolanic reaction of NS. For example, inclusion of NS resulted in enhancement of compressive <span id="page-7-0"></span>**Fig. 5** Water permeability coefficient (K) (cm/sec) at  $28$  days



<span id="page-7-1"></span>**Fig. 6** Compressive strength of NS and NF mixes at 7–180 days of age at room temperature



better improvement with a very small increase in cost compared to the  $M<sub>450</sub>$  mix. Meanwhile, adding 1% and 2% NF increased the improvement to 44.2% and 72.3% at 28 days of age and to 30.12% and 49.5% at 180 days of age, respectively. However, the  $f_c$  of  $M_{NFI}$  was smaller than that of  $M_{NS1}$  at all ages. The  $f_c$  of  $M_{NF2}$  was better than that of  $M_{NS2}$  at early ages up to 56 days of age. On the contrary, the  $f_c$  of  $M_{NF2}$  was smaller than that of  $M_{NS2}$  at late ages up to 180 days of age.

<span id="page-8-0"></span>

<span id="page-8-1"></span>**Fig. 8** Modulus of elasticity at 28 days of age



#### **3.3.2 Splitting tensile and modulus of elasticity**

Figures [7](#page-8-0) and [8](#page-8-1) show the splitting tensile strength  $(f_t)$  and modulus of elasticity  $(E_c)$  of the NS mixes at 28 days of age. Adding NS has enhanced the  $f_t$  and  $E_c$  by up to 81.3% and 60.8% respectively. Meanwhile, the incorporation of 1% and 2% NF improved  $f_t$  and E<sub>c</sub> to 37.6% and 59.5%, respectively. Finally, increasing the cement content with 3% NS (M450) has enhanced the  $f_t$  and  $E_c$  by 74.6% and 59.2%, respectively. In conclusion, the results of MNF1 were smaller than those of MNS1. On the contrary, the results of MNF2 were better than those of MNS2, which agrees with the  $f_c$  results.

#### **3.4 SEM of nano‑specimens at room temperature**

Figure [9](#page-9-0) shows the SEM of the cement paste matrix (CP) and interfacial transition zone (ITZ) with the aggregate of the mixes at 28 days of age. Hydration outputs, such as CSH, ettringite and CH, were observed in the SF mix. Adding 2% NS produced a thick microstructure with extra-dense hydration products due to the highly pozzolanic reaction of NS with high distribution that formed an additional CSH gel. The microstructure improvement increased with increasing NS percentages of up to 4%. These results are in concurrence with the results of Abo El-Enein and Nili [\[35](#page-14-24), [36\]](#page-14-25) which in turn are also consistent with the other recent results. The



<span id="page-9-0"></span>**Fig. 9** SEM of ITZ and CP of the mixes at room temperature [\[9\]](#page-13-6)

SEM analysis revealed that the  $M_{NS4}$  mix had the densest ITZ and contained the largest content of CSH gel combined with the lowest content of calcium aluminate hydrate (CAH) and traces of calcium hydroxide (CH).

The SEM CP and ITZ of  $M_{\text{NF2}}$  were better than those of  $M_{NS2}$  possibly because of the physical effect of NF, which acted as nuclei that accelerated the hydration process at the early ages [[9](#page-13-6)].

#### **3.5 Relative economic index**

Figure [10](#page-10-0) displays the relative economic index ratio (REI) of the mixes incorporated with nanoparticles with respect to the control mix. Clear increments were observed in 1 m3 of NS mixes from 1.73 up to 3.12. Furthermore, increasing the cement content to 450 kg/m3 has raised the REI to 3.26 without a signifcant increase in the mechanical properties or durability. And fnally, the REI of NF mixes has increased up to 3.12. On the contrary, REI of the columns samples was reduced when compared to the 1 m3 ones (Fig. [10](#page-10-0)), which is due to the improvements in the compressive strengths of nano-particles mixes. For instance, the REI of the MNS4 column was 0.85, which improved fc to 1.885% at 28 days of age and reduced the area of the column to 0.65. Furthermore, the mechanical properties and REI of MNS4 mix were better than those of MNF2 mix. In conclusion, the MNS4 mix can be considered the most economical mix with the smallest relative area reduction amongst the other mixes.

## **3.6 Infuence of temperature on compressive strength**

Figure [11](#page-10-1) shows the compressive strengths of the mixes with nano-particles subjected to diferent elevated temperatures (100 °C–800 °C) at 28 days. They were measured and compared with the  $f_c$  of  $M_0$  at the same temperatures. At 100 °C, the  $f_c$  of M<sub>o</sub> and M<sub>SF</sub> mixes has increased to 65 and 77.5 MPa, respectively. For instance, adding 4% NS has improved the  $f_c$  by up to 61.7%. These increments in strength were due to the accelerated hydration of the highly pozzolanic reaction of NS and cementitious materials. At 200 $\degree$  C, the strengths of  $M_{\odot}$  and  $M_{SF}$  have decreased to 60 and 69 MPa, respectively. The compressive strengths of the nano- particle mixes were also reduced but were still greater than the SF mix. For instance, adding 4% NS has improved the strength by up to 59.2%. At 400 °C, considerable reductions in the strengths of  $M_{\odot}$  and  $M_{SF}$  occurred and reached 40 and 36 MPa, respectively. Although the diference in strength between  $M_{NS3}$  and  $M_{NS4}$  mixes was small, the strength of mix  $M_{NS4}$  was still the highest up to exposure temperature of 400 °C, with the improvement reaching up to 104.3%. Moreover, at 600 °C, the strengths of  $M_0$  and  $M_{SF}$  became 20 and 16 MPa, respectively. The strength of the  $M_{NS3}$  mix was the highest, with the improvement reaching 232%. At 800 °C, the strength of the  $M_{NS3}$  mix was still the highest, with the improvement reaching 190%. Thus, the compressive strength values of  $M_{NS3}$  were the largest at 500 °C up to 800 °C. The  $f_c$  of  $M_{450}$  was smaller than that of  $M<sub>NS3</sub>$  at all temperatures, and the difference between them increased with increasing temperature. Hence, increasing the cement content of SF concrete incorporated with NS had reverse effect on the concrete's resistance under high

<span id="page-10-0"></span>**Fig. 10** Cross sec-area ratio and cost ratio of both of  $1 \text{ m}^3$  and designed column relative to  $M_{\Omega}$ 



<span id="page-10-1"></span>**Fig. 11** Compressive strength of concrete mixes at high temperatures up to 800 °C for 2 h

temperatures. The  $f_c$  of  $M_{\text{NF2}}$  was still better than that of  $M<sub>NS2</sub>$  at all temperatures up to 800 °C. At 300 °C, the  $f<sub>c</sub>$  of  $M<sub>NE2</sub>$  was the best amongst all the  $M<sub>NS</sub>$  mixes, except for  $M_{NS3}$  that had the best  $f_c$  up to 800 °C.

## **3.7 SEM analysis of elevated temperature specimens**

Figures [12,](#page-11-0) [13,](#page-11-1) [14](#page-12-0), [15](#page-12-1) show SEM images of ITZ between aggregate and CP matrix of  $M_{SF}$ ,  $M_{NS2}$ ,  $M_{NF2}$  and  $M_{NS3}$ mixes at various temperatures (200 °C–800 °C). At 200 °C, SEM image of  $M_{\rm SF}$  specimen shows shrinkage of cement paste, emergence of micro-cracks and weakened ITZ due to the chemical and physical loss of water and the decomposition of ettringite. On the contrary, SEM images from specimens of all mixes with nanoparticles show that the cement paste has become denser than before due to the accelerated rate of hydration that formed additional hydration products, especially CAH [\[41](#page-14-30)]. This may be due to low thermal conductivity of nanoparticles that delayed the heat transfer of concrete incorporating nanoparticles compared with the heat transfer of normal concrete [\[42](#page-14-31)].

At 400 °C, SEM of  $M_{SF}$  shows that CH and CSH gel starts to decamp and completely lose water leaving some



<span id="page-11-0"></span>**Fig. 12** SEM of ITZ and CP of the mixes at 200 °C

pores in ITZ and a signifcant increase in micro-cracks are also observed. Extra CAH (as a result of the hydration process) appears in the nanoparticles mix specimens and increases with the increase in nano-particles percentage. At 600 °C, more decomposition of CSH and complete decomposition of CH occur in  $M_{SF}$  specimen along with widening of micro-cracks and pores. Micro-cracks begin to appear in  $M_{NS2}$  specimen, but no cracks or pores emerge in  $M_{\text{NF2}}$  and  $M_{\text{NS3}}$  specimens. Extra CAH is also observed in their specimens. At 800  $\rm{^{\circ}C}, M_{\rm{SF}}$  specimen shows fragmentation, severe cracking, ITZ cracks and loss of bonding due to the disappearance of CSH in the specimen. The number of cracks increase in  $M_{NS2}$  specimen but  $M_{NF2}$  and  $M<sub>NS3</sub>$  specimens have no cracks up to 800 °C, but their porosity increases slightly.

<span id="page-11-1"></span>**Fig. 13** SEM of ITZ and CP of the mixes at 400 °C

The fraction surface of the nano-particles mixes is smooth and passed through the aggregate at all temperatures up to 600 °C. SEM images at all temperatures of  $M_{NS3}$  and  $M_{NF2}$ specimens show that they still have dense matrix without signifcant cracks due the increment of CSH and CAH and getting rid of CH [[23](#page-14-12)].

#### **3.8 Fire in the presence of initial cracking load**

Figure [16](#page-13-8) shows the failure compressive strengths of the NS and NF mixes exposed to fre at 600 °C under an initial cracking load of  $M<sub>o</sub>$ . The strengths were tested at 28 days of age. They were compared with strength of  $M_0$  under the same conditions. The results showed that the strengths of the  $M_0$  and  $M_{SF}$  mixes were 36 and 37.3 MPa, respectively.



<span id="page-12-0"></span>

The addition of SF did not improve the residual strength. Residual strength losses were even observed for the same compositions. These results agree with those of Lubloy et al., where the fre-under-load results of NS and NF were almost similar to that of high-temperature exposure results at 600 °C. [[43](#page-14-32)]. For instance, adding NS has improved the fred strength progressively by up to 117% at 3% NS. However, further increasing the NS percentage (MNS4) or the cement content (M450) has reduced the strength improvement to 101.4% and 85% respectively. Thus,  $M_{NS3}$ was the best fre resistance mix because it could enhance the mechanical and durability properties of concrete, similar to the conclusions and the fre test results matched with those of Abhilash et al. [\[44\]](#page-14-33) on exposure to a high temperature of 600 °C.



**Fig. 14** SEM of ITZ and CP of the mixes at 600 °C **Fig. 15** SEM of ITZ and CP of the mixes at 800 °C

# <span id="page-12-1"></span>**4 Conclusions**

This study analysed the compressive strength of SF mixes with a small content of nanomaterials under exposure to high temperatures and fre in the presence of an initial cracking load. The results of all tested mixes were compared to control mix,  $M<sub>0</sub>$ . The following conclusions were obtained.

- 1. At room temperature, the addition of NS can improve the  $f_c$  by up to 45.5%.
- 2. Increasing the cement content of the concrete with 3% NS does not enhance the resistance to high temperatures and fre.
- 3. The concrete mix with  $M_{NS4}$  has demonstrated the lowest relative economic index of the column area, and  $M_{450}$

<span id="page-13-8"></span>**Fig. 16** Compressive strengths of the concrete mixes after fre at 600 °C for 1 h



is the largest relative area reduction with respect to the control mix column area.

- 4. The addition of 4% NS can enhance the compressive strength by approximately 104% under elevated temperature up to 400 °C.
- 5. Using 3% NS has demonstrated the best results since the compressive strength under elevated temperatures has improved by up to 230%.
- 6. The mixes with nanomaterials have demonstrated the best compressive strengths which correspond of the SEM image results.
- 7. The water permeability of 4% NS has the best reduction percentage reaching to 72%. The K of 2% NF was smaller than that of 2% NS that reached 55.5% and 54.1%, respectively.
- 8.  $M<sub>NS3</sub>$  had the largest fc amongst the fired concrete specimens in the presence of initial cracking load at 600 °C. Its addition improved the strength of the  $M_0$  mix by 117%.

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**Data availability** Data can be provided by demand from the corresponding author.

## **Declarations**

**Conflict of interest** All authors claim that they have no confict of interest.

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