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A facile preparation of WS₂ nanosheets as a highly effective HER **catalyst**

Xiangyong Zhang1 · Hao Fei¹ · Zhuangzhi Wu1,2 · Dezhi Wang1,2

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Abstract

Tungsten disulfide (WS_2) has been considered as a promising hydrogen evolution reaction (HER) candidate due to its high activity, robust chemical stability, and earth-abundant resources. However, the inert basal planes and low electrical conductivity greatly hinder its development in HER. Increasingly, the density of active sites through the structural designing is one of the most efective strategies to enhance the HER performance. Herein, we report a facile one-step hydrothermal method for synthesizing flower-like WS_2 nanosheets for highly efficient HER. Besides, the effect of preparation temperature is also been discussed. The optimized WS_2 nanosheets exhibit the enhanced HER activity in strong acidic solutions with a low Tafel slope and a good stability. The improvement of the HER performance can be attributed to sheet-like nanostructures, which greatly increase the edge sites and defects, resulting in a high density of exposed active sites. Besides, these sheet-like nanostructures also can make the acidic electrolyte easily accessible to the surface of $WS₂$ and accelerate the electron transfer rate.

Keywords Tungsten disulfde · Nanosheet · Electrocatalyst · Hydrogen evolution reaction · Hydrothermal method

1 Introduction

Exploring renewable carbon-free energy alternatives is one of the most promising pathways for alleviating the energy and environmental crisis [[1](#page-6-0), [2](#page-6-1)]. Hydrogen has been considered as one of the most promising clean energy carriers because of its high energy density and no pollutant product [[3\]](#page-6-2). Moreover, electrochemical water splitting is a highly efficient sustainable hydrogen production route $[4]$ $[4]$. However, the corresponding hydrogen production process always requires excellent catalysts to achieve fast kinetics and lower the overpotential for HER. Up to now, although platinum (Pt) and other Pt-group metals exhibit the best catalytic activity for HER, the scarcity and high cost impede their widespread applications [[5\]](#page-6-4). Thus, there is still an urgent

 \boxtimes Zhuangzhi Wu zwu2012@csu.edu.cn

 \boxtimes Dezhi Wang dzwang@csu.edu.cn

Key Laboratory of Ministry of Education for Non-ferrous Materials Science and Engineering, Changsha 410083, China demand for developing earth-abundant catalysts to replace these noble metal catalysts for efective HER.

Recently, a large variety of non-precious metal candidates, including transition metal carbides [[6–](#page-6-5)[9\]](#page-6-6), phosphides [\[10–](#page-6-7)[13\]](#page-6-8), chalcogenides [[14–](#page-6-9)[20](#page-7-0)] and so on, have been investigated and shown striking HER performances. Among these alternatives, tungsten disulfide (WS_2) has received persistent interest for its high activity, robust chemical stability, and earth-abundant resources [\[21\]](#page-7-1). However, both theoretical and experimental studies revealed that the edge state possesses a lower hydrogen adsorption Gibbs free energy, which means that HER performance mainly arises from the edge site, while the large-area basal planes are catalytically inert and useless for HER [\[22](#page-7-2)[–24](#page-7-3)]. Moreover, as a semiconductor material, the activity of WS_2 is primarily limited by its low electrical conductivity, which restricts charge transfer kinetics for HER [[25\]](#page-7-4). Considering the above two factors, there are generally two routes to enhance the catalytic activity of $WS₂$. One is to increase the density of the exposed active sites. Among large amounts of methods, structural designing is one of the most efective strategies for increasing the number of active sites [\[26](#page-7-5), [27\]](#page-7-6). In our previous work [\[28](#page-7-7)], the WS_2 nanosheets (NSs) with loosely stacked layers were successfully obtained by a mechanical activation strategy. This special nanostructure provides highly exposed rims and

¹ School of Materials Science and Engineering, Central South University, Changsha 410083, China

edges for HER. Cheng et al. [\[29\]](#page-7-8) synthesized the ultrathin $WS₂$ nanoflakes by the high-temperature solution-phase method. The obtained catalyst possessed abundant edges and the ultrathin thickness. By directly vulcanizing the WO_3 nanosheets, Shang et al. $[30]$ $[30]$ fabricated a WS₂/WO₃ heterostructure which could expose abundant active sites for HER. The other route is to improve the electrical conductivity. Sun et al. $[31]$ $[31]$ synthesized the N-doped WS₂ nanosheets by onestep sol-gel process and found that N doping in WS₂ might be an effective way to improve the intrinsic conductivity of WS₂. Duan et al. [\[32\]](#page-7-11) fabricated 3D WS₂ nanolayers@ heteroatom-doped graphene flms via a vacuum-fltration process, and the conductive network of graphene sheets greatly accelerates the charge transfer kinetics. Despite these developments, the HER activity of WS_2 is still much lower than that of Pt. Thus, it is still a big challenge to further improve the HER performance of $WS₂$.

Herein, the flower-like WS_2 nanosheets were synthesized via a facile one-pot hydrothermal method. This unique nanostructure not only can greatly increase the edge sites and defects, resulting in a high density of exposed active sites, but also can make the acidic electrolyte easily accessible to the surface of WS_2 and accelerate the electron transfer rate. Moreover, the HER performance of WS_2 is improved by controlling the reaction temperature. As a result, the optimized WS_2 NSs exhibit a good HER activity in an acidic solution with a small Tafel slope of 70 mV dec⁻¹ and a good stability.

2 Experimental

2.1 Materials

Sodium tungstate dihydrate $(Na_2WO_4.2H_2O)$, sodium hypophosphite $(NaH_2PO_2·H_2O)$, thioacetamide (CH_3CSNH_2) and bulk WS_2 were purchased from Aladdin. The Pt/C catalyst (20 wt.%) was purchased from the Johnson Matthey and Nafon solution (5 wt.%) was purchased from the DuPont.

2.2 Synthesis of WS₂

Typically, 0.99 g Na₂WO₄.2H₂O, 0.32 g NaH₂PO₂.H₂O and 1.13 g CH_3CSNH_2 were dissolved into 50 mL distilled water. After being stirred to form a transparent solution, 0.75 mL concentrated HCl was dropped into the solution and continuously stirred for 30 min. Then, the solution was transferred into a 100 mL Teflon-lined stainless-steel autoclave, and heated in an oven maintained at 180–220 °C for 24 h. After being naturally cooled to room temperature, the asprepared samples were obtained by centrifugation, washed by absolute ethanol and deionized water for several times, and dried at 80 °C overnight.

2.3 Preparation of working electrodes

In a typical procedure, 3 mg of the catalyst was added into the solution containing 80 μ L Nafion solution (5 wt.%), 0.2 mL absolute ethanol and 0.8 mL deionized water. After being sonicated for 30 min, 5 mL of the formed homogeneous slurry was dropped onto a smooth glassy carbon electrode (GCE) of 3 mm in diameter, generating a working electrode with a loading of 213 μ g cm⁻².

2.4 Physical measurements

The powder XRD patterns were recorded using the Mini Flex 600 system, and the Raman spectroscopy was obtained on the LabRAMHR-800 (French company HRIBA). The transmission electron microscopy (TEM) images and the scanning electron microscopy (SEM) images were acquired on the Tecnai G^2 F20 and the FEI Sirion 200, respectively. The X-ray photoelectron spectroscopy (XPS) was conducted on the ESCALAB 250Xi and the correspond binding energy values were corrected via the C1s.

2.5 Electrochemical measurements

A CHI660E workstation (CH Instruments, China) with a standard three-electrode system was used to perform all the electrochemical tests. The modifed GCE was used as the working electrode, and the carbon rod and the saturated Hg/ Hg_2Cl_2 electrode (SCE) were served as the counter electrode and the reference electrode, respectively. The linear sweep voltammetry (LSV) was carried out in 0.5 mol L^{-1} H₂SO₄ using a sweep rate of 2 mV s⁻¹ from 0.1 to – 0.4 V (vs. SCE). The long-term cycling test was performed via the cyclic voltammetry (CV) measurements with a sweep rate of 50 mV s⁻¹ from 0.1 to $-$ 0.4 V for 2000 cycles. In addition, the electrochemical impedance spectroscopy (EIS) was conducted between the frequency range of 10^6 Hz and 1 Hz with a sinusoidal perturbation of 5 mV at an overpotential of -300 mV.

2.6 Electrochemical measurements

The electrochemically capacitance measurements were conducted by cyclic voltammograms from 0.00 to 0.20 V with various scan rates (20, 40, 60, 80, 100, 120, 140 mV s−1). The capacitive currents were measured in a potential where no faradic processes were observed, i.e., at 0.1 V versus RHE. According to the previous report [[14](#page-6-9)], the specific capacitance, a flat standard with 1 cm^2 of the real surface area, is approximately 60 μ F cm⁻². Thus, the electrochemical active surface area can be calculated by the following Eq. (1) (1) :

$$
A_{\text{ECSA}} = \frac{\text{electrochemical capacitance}}{60 \,\mu\text{F cm}^{-2} \,(\text{per cm}_{\text{ECSA}}^2)}
$$
(1)

3 Results and discussion

3.1 Microstructural characterization

The XRD pattern of WS_2 NSs is shown in Fig. [1a](#page-2-1). The peaks at 14.2°, 32.1°, 35.4°, 43.4° and 56.9° can be assigned to the (002) , (100) , (102) , (006) , and (110) planes of WS₂ (JCPDS) No. 08-0237), respectively. There are no major impurities detected, suggesting the high purity of the obtained WS_2 . Besides, the XRD pattern of bulk WS_2 is also given for a comparison, which is well consistent with the standard WS_2 . Interestingly, compared with the bulk WS_2 , all peaks of WS_2 NSs are weaker and broader, indicating a poor crystallinity and disordered structure [[33\]](#page-7-12). Raman spectroscopy was conducted to further confirm the structure of WS_2 NSs. As shown in Fig. [1b](#page-2-1), there are two pronounced peaks at \sim 50 and 420 cm⁻¹, which are assignable to the E_{2g}^1 modes and the A_{1g} modes of $2H-WS₂$, respectively [[34,](#page-7-13) [35](#page-7-14)]. Compared with the bulk WS_2 , the Raman peaks of WS_2 NSs show slight shifts, which can be attributed to the possible changes in the bond polarization [\[36](#page-7-15)].

The morphologies of as-prepared WS_2 and bulk WS_2 are shown in Fig. [2.](#page-3-0) It can be seen from Fig. [2a](#page-3-0), d that the as-prepared WS_2 exhibits a flower-like morphology, which is consisted of a large number of loosely assembled thin nanosheets. The high-resolution TEM (HRTEM) image (Fig. [2e](#page-3-0)) of the WS_2 NSs reveals that the interlayer spacing of the WS_2 WS_2 (002) planes is ~ 0.62 nm. In Fig. 2b, the

pristine bulk WS_2 shows large and thick bulk morphology. The HRTEM image (Fig. [2](#page-3-0)c) suggests that the basal plane of the bulk WS_2 possesses high crystallinity and no defects. As for the basal plane of WS_2 NSs (Fig. [2](#page-3-0)f), many defects (the yellow selection in Fig. [2f](#page-3-0)) can be observed, which may be beneficial for HER as possible active sites.

To get insights into the surface chemical valence states of WS_2 NSs, the XPS analysis was further performed (Fig. [3](#page-3-1)). As shown in Fig. [3a](#page-3-1), the high-resolution *W* 4*f* spectra can be ftted into two doublets. The doublets located at binding energies of 38.2 eV (W $4f_{5/2}$) and 36.1 eV (W $4f_{7/2}$) are originated from W^{6+} in WO_3 , which can be ascribed to surface oxidation of WS_2 [\[25](#page-7-4), [30](#page-7-9), [37](#page-7-16)]. The peaks located at 34.4 eV and 32.3 eV are assigned to $4f_{5/2}$ and $4f_{7/2}$ of W⁴⁺, respectively, which are consistent with the presence of WS_2 [[38,](#page-7-17) [39\]](#page-7-18). Correspondingly, the doublets at around 163.0 eV and 161.8 eV (Fig. [3b](#page-3-1)) are indexed to $2p_{3/2}$ and $2p_{1/2}$ of *S*, respectively, which can be attributed to S^{2-} species in WS₂ [[40,](#page-7-19) [41\]](#page-7-20).

3.2 Insights into the HER kinetics

The electrochemical performance of the bulk WS_2 and $WS₂$ NSs for HER was detected by the LSV measurement in 0.5 mol L^{-1} H₂SO₄. The polarization curves of different catalysts are shown in Fig. [4](#page-4-0)a, and the key electrochemical results are also displayed in Table [1.](#page-4-1) The commercial Pt/C catalyst (20 wt.%, Johnson Matthey) was also measured as a reference. The bulk WS_2 exhibits a poor HER performance with a high onset potential (the current density reaches 1 mA cm−2) of − 396 mV, and low current density (*j−*400) of 1.1 mA cm⁻² at a specific overpotential of $-$ 400 mV. In comparison, the WS_2 NSs exhibits a good HER activity with a low onset potential of − 306 mV, and a high *j−*⁴⁰⁰

Fig. 1 a XRD patterns and **b** Raman spectra of the WS_2 NSs and bulk WS_2

Fig. 2 a SEM image of WS₂ NSs; **b**, **c** TEM and HRTEM images of bulk WS₂; **d**–**f** TEM and HRTEM images of WS₂ NSs

Fig. 3 XPS spectra of WS₂ NSs: **a** W 4 f and **b** S 2 p

of 8.5 mA cm−2, which is ~ 7 times higher than that of the bulk WS_2 .

To further study the catalytic activity of the WS_2 catalysts, the Tafel slopes were extracted from the corresponding LSV curves through the Tafel equation $(\eta = b \log j + a$, where *a* is the onset potential and *b* is the Tafel slope) (Fig. [4](#page-4-0)b). The WS₂ NSs exhibit a Tafel slope of 70 mV dec⁻¹, which is remarkably smaller than that of bulk WS₂ (112 mV dec⁻¹), indicating a more efficient HER process $[42, 43]$ $[42, 43]$ $[42, 43]$ $[42, 43]$. Generally,

in acid media, it is accepted that the HER process involves two steps. The electrochemical adsorption of hydrogen (Volmer reaction, $H^+ + M + e^- \rightarrow M - H^*$, where M is the catalysts) is the frst step. While, the desorption of adsorbed hydrogen is the second step, which includes the electrochemical desorption (Heyrovsky reaction, $M - H^* + H^+ + e^- \rightarrow M + H_2$ or the chemical desorption (Tafel reaction, $M - H^* + M - H^* \rightarrow 2M + H_2$). According to the classic theory, the Tafel slope can be used to evaluate

Fig. 4 a Polarization curves, **b** Tafel slopes and **c** Nyquist plots of the WS₂ NSs and the bulk WS₂; **d** Polarization curves of the WS₂ NSs and after 5000 cycles

ECSA electrochemically active surface area

the rate-determining step in the HER process [\[44–](#page-7-23)[46\]](#page-7-24). Correspondingly, the Tafel slope values will reach 118, 29 and 39 mV dec^{−1} if the Volmer, Tafel, and Heyrovsky reactions are the rate-determining steps, respectively. The small Tafel slope of 70 mV dec⁻¹ denotes that the WS₂ NSs undergo a mixed Volmer-Heyrovsky mechanism and the Heyrovsky reaction acts as the rate-determining step.

To get insights into electrode kinetics of the HER process, the EIS measurements were performed at a constant voltage of − 300 mV. A simplifed equivalent circuit (inserted in Fig. [4c](#page-4-0)) was utilized to ft the Nyquist plots. The charge transfer resistance (R_{ct}) can be determined from the semicircle. The $R_{\rm ct}$ value of WS₂ NSs (244 Ω) is much smaller than that of bulk WS₂ (5231 Ω), suggesting a faster electron transfer rate and superior kinetics of WS_2 NSs in HER pro-cess [[47\]](#page-7-25). The small R_{ct} value of WS₂ NSs can be attributed the sheet-like nanostructure, which can make the acidic electrolyte easily accessible to the surface of WS_2 , resulting in a faster electron transfer rate [\[48,](#page-7-26) [49](#page-7-27)]. On the contrary, the bulk structure of bulk WS_2 slows down the electron transfer rate, thus, leading to a large $R_{\rm ct}$ value.

Besides, the stability of catalysts is an important indicator for evaluating the electrocatalytic performance. The longterm cycling measurement of the WS_2 NSs was adopted using the CV. As shown in Fig. [5](#page-4-2)a, only a negligible shift of the current density can be observed after scanning for 2000

Fig. 5 a Polarization curves of the WS₂ NSs and after 2000 cycles; **b** *I*–*t* curve of the WS₂ NSs for 50 h

cycles, implying the good catalytic stability of WS_2 NSs. The time-dependent current density curve (*I*–*t* curve) was also obtained (Fig. [5](#page-4-2)b). Only a slight degradation of the current density over 50 h further proves its good stability. The SEM and TEM images of the WS_2 NSs after scanning for 2000 cycles are shown in Fig. [6](#page-5-0), and there are few changes in the morphology after the stability test.

The ECSA of WS_2 catalysts was estimated by testing the double-layer capacitance (C_{dl}) at non-faradaic potentials (Fig. [7](#page-5-1)). A higher ECSA value means a larger density of active sites for WS_2 . Figure [7](#page-5-1)c is the measured current density plotted as a function of the scan rate, in which the C_{dl} value is equivalent to the slope value. A direct comparison of these two samples can be obtained because the ECSA value is proportional to the $C_{\rm dl}$ value. One can see that the $C_{\rm d}$ value of the WS₂ NSs is 11.2 mF cm⁻², whereas the $C_{\rm d}$ value of the bulk WS₂ is 0.2 mF cm⁻². Accordingly, the WS₂ NSs exhibits a high ECSA value of 190 cm²_{ECSA} ~ 57 times as high as that of the bulk WS_2 (3.33 cm²_{ECSA}), suggesting that the WS_2 NSs possess plenty of edge sites and defects that will greatly increase the density of active sites.

Due to the hydrothermal reaction being sensitive to the preparation temperature, the effect of the preparation

temperature is also been discussed. All the as-prepared WS_2 NSs are named as WS_2 -*XXX* for convenience, where *XXX* represents the preparation temperature in the unit of °C. In Fig. [8a](#page-6-10), one can see that the HER activity of WS_2 NSs firstly increases with the increase of the preparation temperature and reaches the maximum at 200 °C, and then decreases at higher temperatures, which means that 200 °C is the most suitable temperature for the formation of the high-efficiency HER catalyst in WS_2 NSs. Similar regulations can also be found in Tafel and Nyquist plots (Fig. [8](#page-6-10)b, c). In general, as the preparation temperature increases, the crystallinity of the as-prepared catalysts is improved, leading to a decrease in the number of defects and a corresponding decrease in the number of active sites. While, on the other hand, a higher crystallinity of the $WS₂$ NSs could result in a better conductivity [\[50](#page-7-28)]. Thus, we believe that a balance is achieved between the number of active sites and conductivity at 200 °C, resulting in the best HER performance.

As a result, the good HER activity of the optimized WS_{2} -200 can be ascribed to a sheet-like structure, which not only can greatly increase the number of exposed active sites, but also can make the acidic electrolyte easily accessible to the surface of WS_2 and accelerate the electron transfer rate.

Fig. 7 Electrochemical capacitance measurements: **a** WS₂ NSs and **b** bulk WS₂; **c** Measured capacitive current densities (at 100 mV vs. RHE) plotted as a function of the scan rate for the WS_2 NSs and bulk WS_2

Fig. 8 a Polarization curves, **b** Tafel slopes and **c** Nyquist plots of the WS₂ NSs obtained at various temperatures

Besides, the appropriate preparation temperature is also important, which makes the WS_2 -200 possess a relatively large number of active sites and high conductivity, in the same time.

4 Conclusion

In summary, a flower-like WS_2 catalyst is successfully synthesized via a facile one-pot hydrothermal method. The unique nanostructure can greatly increase the edge sites and defects, resulting in a high density of exposed active sites. Compared with bulk WS_2 , the WS_2 NSs show a high ECSA value of 190 cm²_{ECSA}, ~7 times as high as that of bulk WS_2 ; and a high *j−*400 of 8.5 mA cm−2,~7 times higher than bulk WS_2 and a lower Tafel slope of 70 mV dec⁻¹. Moreover, the good stability also makes it a promising alternative to replace precious-metal HER catalysts. Besides, the efect of preparation temperature is also discussed, and 200 °C is found to be the best preparation temperature for WS_2 NSs.

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Dr. Dezhi Wang born in 1968, obtained his Doctor degree in 1998 supervised by Tieyong Zuo, an academician of the Chinese Academy of Engineering. He joined the School of Materials Science and Engineering in Central South University (CSU) as an Associated Professor in 2000, and was promoted to a full professor in 2005. He has engaged in the development of rare-metal-based materials for more than 20 years. Until now, he has published more than 80 papers and 15 patents, which are supported by National Science Foundations, 863 programs and military projects.