



Flexible and porous Co₃O₄-carbon nanofibers as binder-free electrodes for supercapacitors

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Abstract

Herein, flexible and porous Co₃O₄-carbon nanofibers (Co₃O₄-CNFs) were fabricated by electrospinning technique combining with the followed carbonization process. The effects of material composition and calcination temperature on morphology, pore structure, and electrochemical properties of the Co₃O₄-CNFs were systematically investigated. Results indicated that the obtained Co₃O₄-CNFs exhibited high porosity, high mechanical strength, and superior electrical conductivity. Electrochemical characterization results showed that the optimized Co₃O₄-CNFs as binder-free electrodes exhibited a specific capacitance of 369 F g⁻¹ at the current density of 0.1 A g⁻¹. Even at a high current density of 2 A g⁻¹, the specific capacitance still remained at 181 F g⁻¹, with the capacitance retention rate of 49%. Intriguingly, the prepared Co₃O₄-CNF film could recover to its original state easily after folding for three times, indicating good mechanical flexibility for free-standing electrodes. Coupled with the excellent mechanical flexibility, high specific capacitance, and simple fabrication process, the flexible and free-standing Co₃O₄-CNFs with hierarchical porous structure could be promising electrode materials for energy storage applications.

Flexible and porous Co₃O₄-carbon nanofibers were prepared by electrospinning and carbonization, which can be used as free-standing electrodes for supercapacitors.

Keywords Electrospinning · Lignin · Carbon nanofibers · Co₃O₄ · Supercapacitors

1 Introduction

Much attention has been focused to supercapacitors (SCs) because of their fascinating power density, long life cycles, fast charge and discharge rates, and high efficiency [1, 2]. However, the low energy-power ratio of SCs has been

restricted to their applications. In recent years, numerous materials, such as carbonaceous materials [3, 4], metal oxides or hydroxides [5, 6], and conducting polymers [7, 8], have been widely used in SCs. Among the electroactive materials, Co₃O₄ has been extensively reported because of its high theoretical specific capacitance up to 3560 F g⁻¹, high redox performance, controllable size and shape, relatively cost-effective, and widespread applications [9–12].

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Recently, much progress has been made in the development of energy storage device [13–17]. For example, Pal et al. have demonstrated a single-step solvothermal method to synthesize self-supported ultra-small Co_3O_4 nanocubes over Ni foam, which exhibited pseudocapacitive performance of 1913 F g^{-1} at current density of 8 A g^{-1} [18]. Jiang et al. reported a technique combining hydrothermal and calcination for the synthesis of two-dimensional Co_3O_4 thin sheets, which showed a specific capacitance of 1500 F g^{-1} at 1 A g^{-1} [19]. Besides, the Co_3O_4 has also widely been used as wave-absorbing material, as they have good impedance matching [20–24]. For example, the hierarchically porous Co/C nanocomposites were fabricated by freezing dry and carbothermic reduction process with excellent absorption performance of its minimum reflection loss could reach to 34.2 dB, which was profit from the optimization of hierarchical porous microstructures and impedance matching [25]. However, the low electrical conductivity and inferior rate capacity of Co_3O_4 have greatly limited their practical application [26–28]. To address these drawbacks, the carbon-based Co_3O_4 composites for SCs electrodes have been investigated in various forms [29–31].

Carbon matrixes, such as carbon nanotube (CNT), graphene, and carbon nanofibers (CNFs), have been regarded as potential candidates for SC electrodes [32–34]. Recently, Zheng et al. constructed 3D hierarchical N-doped carbon-based Co_3O_4 nanopillar arrays as a binder-free electrode, which exhibited a maximum specific capacity of 978.9 F g^{-1} at 0.5 A g^{-1} [35]. Fan and co-workers reported the 2D thin Co_3O_4 nanosheets anchored on 3D porous graphene/nickel foam substrate through hydrothermal synthesis [36]. Because of the synergy of thin Co_3O_4 nanosheets and conductive grapheme layer, the composites displayed a high specific capacitance of 3533 F g^{-1} at a current of 1 A g^{-1} . Besides, Su et al. prepared two-ply yarn SCs by electrochemical deposition of MnOx or Co_3O_4 materials on CNT/stainless steel (SS), the $\text{Co}_3\text{O}_4/\text{CNT}/\text{SS}$ yarn electrode exhibited volumetric capacitance of 82.94 F/cm^3 at 0.02 V/s [37]. Nevertheless, the complicated synthesis process and high cost of graphene and CNT may prevent their widespread application in SCs [38]. Fortunately, CNFs have been scaled up to industrial production by electrospinning technique, with a much lower cost than graphene and CNT, which have high mechanical strength, excellent chemical resistance, and superior electrical conductivity [39–42].

Traditional CNFs are mostly made from polyacrylonitrile (PAN), in which the high cost and environmental concerns limit its extensive applications. As a green and sustainable material, lignin is regarded as one of the most attractive precursors for CNFs because of its high carbon content (up to 60%) and aromatic monomers [43–53]. However, most of the lignin-based CNFs show lower mechanical strength as compared to PAN-based CNFs [54–58]. Although lignin-based

CNFs may not meet the high requirement standard for construction materials, they can be promising electrode materials for energy storage applications. In addition, it is reported that the flexibility and electrochemical performances of lignin-based CNFs can be improved by rational design of the porous structure [59–61]. For example, Liu et al. reported the fabrication of free-standing hierarchical porous CNFs films by a metal ion-assistant acid corrosion process [62]. Results indicated that the porous structure could increase ion transportation path and facilitate the accessibility between CNFs and electrolyte. Samuel and co-workers demonstrated the preparation of flexible core-shell SnOx/CNF composites using poly (methyl methacrylate) as pore-forming agent. It was found that the SnOx/CNF-based symmetric SCs exhibited a specific capacitance of 289 F g^{-1} at a scan rate of 10 mV s^{-1} [63].

Inspired by the above-mentioned studies, we propose to synthesize lignin-derived porous Co_3O_4 -CNFs by using electrospinning technique and the subsequent carbonization, in which the advantages of porous structure of CNFs with good mechanical performance, high theoretical specific capacitance of Co_3O_4 , and the low cost of lignin can be integrated. The surface chemistry, crystalline structure, and morphology of the as-produced Co_3O_4 -CNFs were characterized by FTIR, XPS, XRD, EDS, SEM, and TEM analyses. Moreover, the electrochemical behaviors of the Co_3O_4 -CNFs electrodes were measured by cyclic voltammetry (CV), galvanostatic charging-discharging (GCD), and electrochemical impedance spectroscopy (EIS) tests. This work will provide a facile and versatile strategy to prepare porous CNF-based composite electrode materials for high-performance supercapacitors.

2 Experimental section

2.1 Materials

Cobalt nitrate hexahydrate, acetic acid, and *N,N*-dimethylformamide (DMF) were purchased from Beijing Chemical Works. PAN (Mw = 150 000) and terephthalic acid (TPA) were bought from Macklin. Lignin (Mw = 10,000) was purchased from Sigma-Aldrich. All chemicals used were analytical grades and utilized without further purification.

2.2 Synthesis of porous Co_3O_4 -CNFs

A total of 0.50 g of PAN and 0.30 g of TPA were mixed and dissolved in 5 mL DMF at 60°C forming a homogeneous colloidal solution. Meanwhile, 0.20 g lignin and 0.291 g cobalt nitrate hexahydrate were dissolved in the mixed solution of 0.5 mL acetic acid and 1 mL DMF. Then, the above two solutions were mixed together by stirring. The mixed

solution was transferred into a plastic syringe (10 mL) equipped with capillary needle (0.42 mm in diameter). The solution was electrospun at a feeding rate of 0.5 mL h⁻¹ under high voltage of 20 kV. Then, the fibers were collected with an aluminum foil sheet, which distance between the needle and collector was 18 cm. The fibers were then dried in fume hood. Then, the dried nanofibers were thermostabilized in tube furnace to 280 °C at 1 °C min⁻¹ and kept at 280 °C for 2 h in air environment. Subsequently, they underwent a pyrolysis process in tube furnace at three different temperatures, 700, 800, and 900 °C for 2 h in N₂ atmosphere with a heating rate of 3 °C min⁻¹. Finally, the carbonization nanofibers were annealed at 300 °C for 60 min in air to obtain Co₃O₄ nanoparticles (NPs). The schematic illustration of the experimental process is shown in Scheme 1. The corresponding samples were labeled as Co₃O₄-CNFs-700, Co₃O₄-CNFs-800, and Co₃O₄-CNFs-900, respectively.

2.3 Characterization

X-ray powder diffraction (XRD) detection was analyzed in the 2θ range from 10 to 80° on a Rigaku DMax-RB 91–0459 diffractometer with Cu Kα radiation at a scan rate of 4° min⁻¹. The morphology of nanofibers was investigated by the high-resolution images, which were measured by FESEM (SU8010, Japan). Besides, the structures of CNFs were conducted by transmission electron microscopy (TEM, JEOL, JEM-1010). The X-ray photoelectron spectroscopy (XPS) date was performed on an ESCALAB 250Xi X-ray photoelectron spectroscope with a monochromatic Al Kα radiation. And the spectral elemental analysis was examined by EDS (Thermo Scientific attached with Hitachi S-4800

operated at 15 kV). The pore structure of Co₃O₄-CNFs was tested by N₂ adsorption/desorption isotherms using Belsorp-max surface area detecting instrument. Thermogravimetric analysis (TGA) of the precursors was measured (SDT Q600) in air atmosphere with the temperature range of 30–600 °C at a heating rate of 10 °C min⁻¹. FTIR spectra of lignin, precursor, and CNFs were collected on Thermo Nicolet (Nicolet iN10) with a resolution of 4 cm⁻¹. Raman spectroscopy was recorded from 100 to 2000 cm⁻¹ using a LabRAM HR Evolution by wavelength of 523 nm under ambient conditions.

2.4 Electrochemical measurement

Electrochemical properties were performed using CHI 660D electrochemistry workstation (Shanghai Chenhua Instrument Co., China) in three-electrode setup, where CNFs were served as working electrodes without any additional binders or conductive additives, Hg/HgO as reference electrode and platinum mesh as counter electrode, respectively. Electrochemical tests including CV, GCD, and EIS were performed in 3 M KOH aqueous electrolyte at room temperature.

3 Results and discussion

Figure 1a presents the electrospinning precursor before calcination, in which the obtained nanofibers are smooth and randomly ordered with a network structure and the average nanofiber diameter of 600 nm. After carbonization and oxidation process, the TPA component sublimated, generating many pores in the carbonized fibers. Furthermore, the appearance of Co₃O₄-CNFs became rough, and Co₃O₄ NPs

Scheme 1 Schematic illustration of porous Co₃O₄-CNFs prepared by electrospinning and carbonization

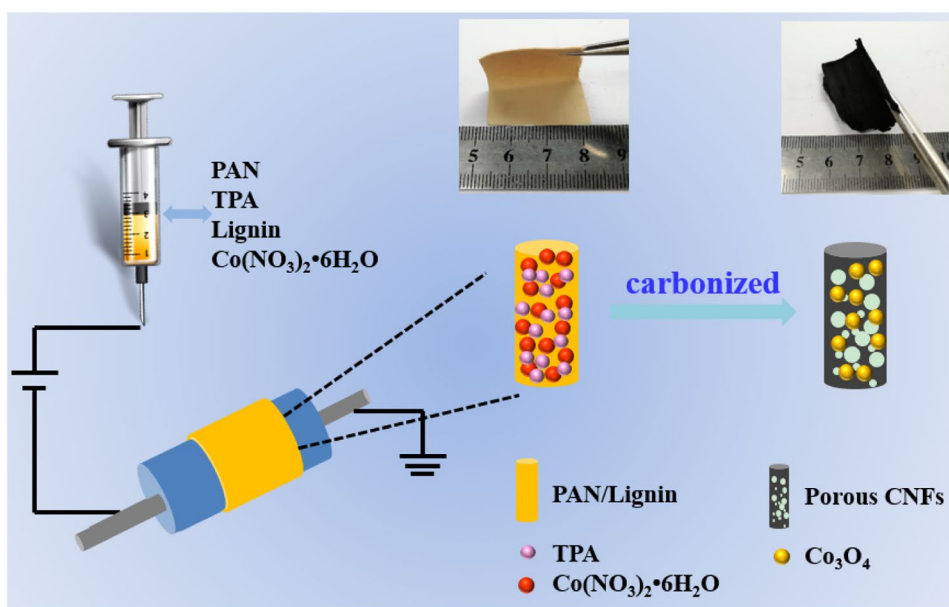
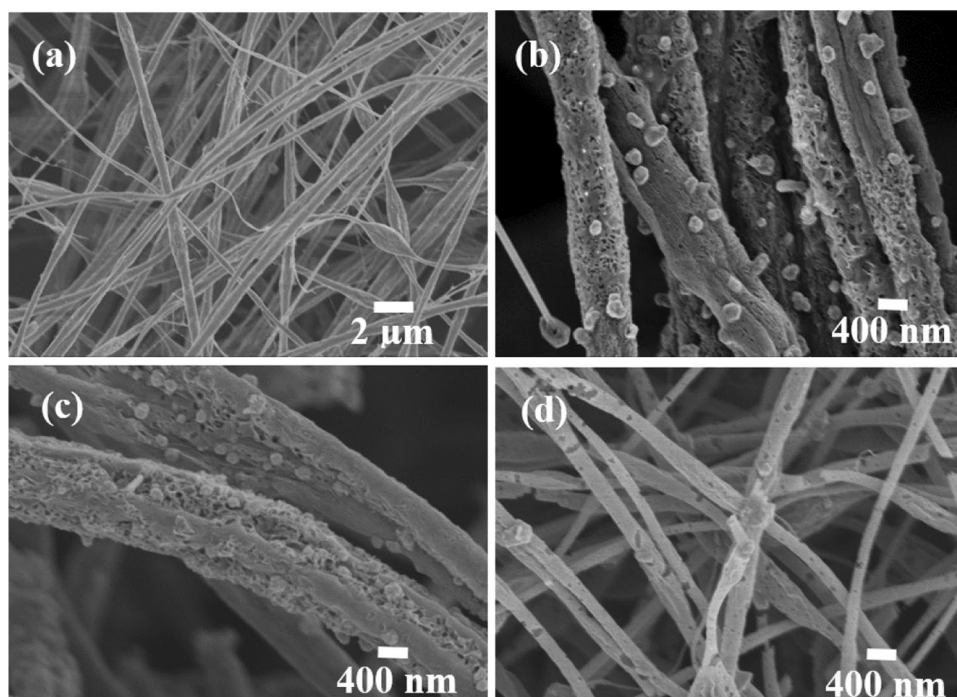


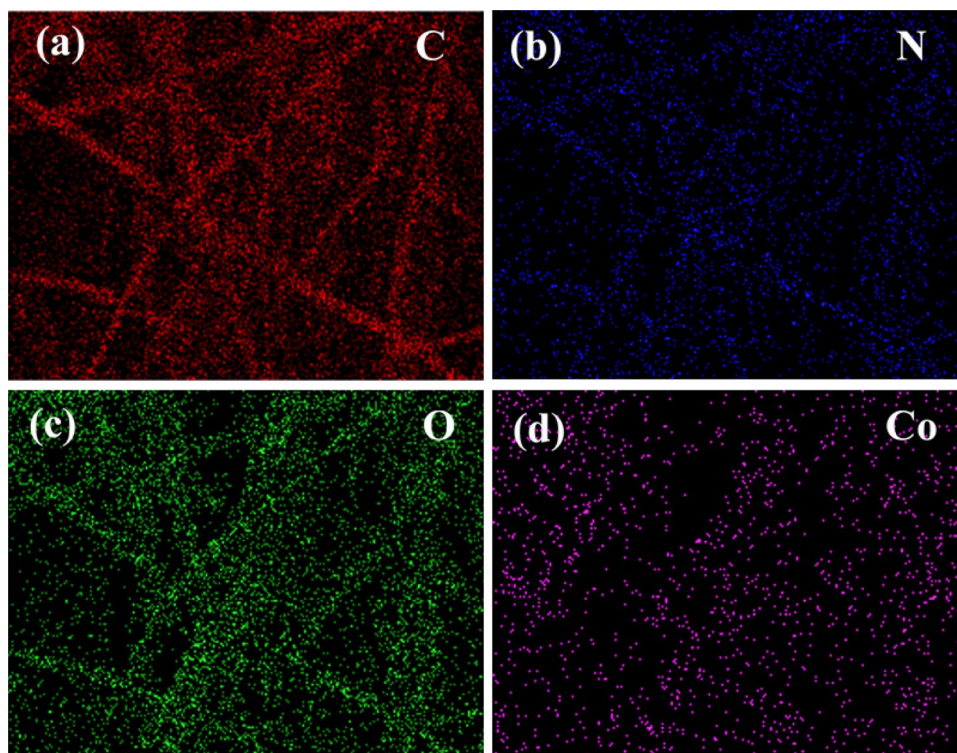
Fig. 1 SEM images of **a** PAN-cobalt salt precursors, **b** Co_3O_4 -CNFs-700, **c** Co_3O_4 -CNFs-800, and **d** Co_3O_4 -CNFs-900, respectively



were also found on the surface of the CNFs uniformly. It is probably because of the formation of porosity and Co_3O_4 NPs, the mean diameter of Co_3O_4 -CNFs-700 shows little change after carbonization (Fig. 1b). While the calcination temperature increased to 800 °C, the mean diameter of

Co_3O_4 -CNFs-800 reduced to 450 nm (Fig. 1c), which could be due to the further decomposition of polymer during the carbonization process. Additionally, the average diameter of Co_3O_4 -CNFs-900 continues to reduce to 250 nm, and in the meantime, the quantity of cobalt nanoparticles decreases

Fig. 2 The EDS element mappings of Co_3O_4 -CNFs-700: **a** C; **b** N; **c** O; **d** Co

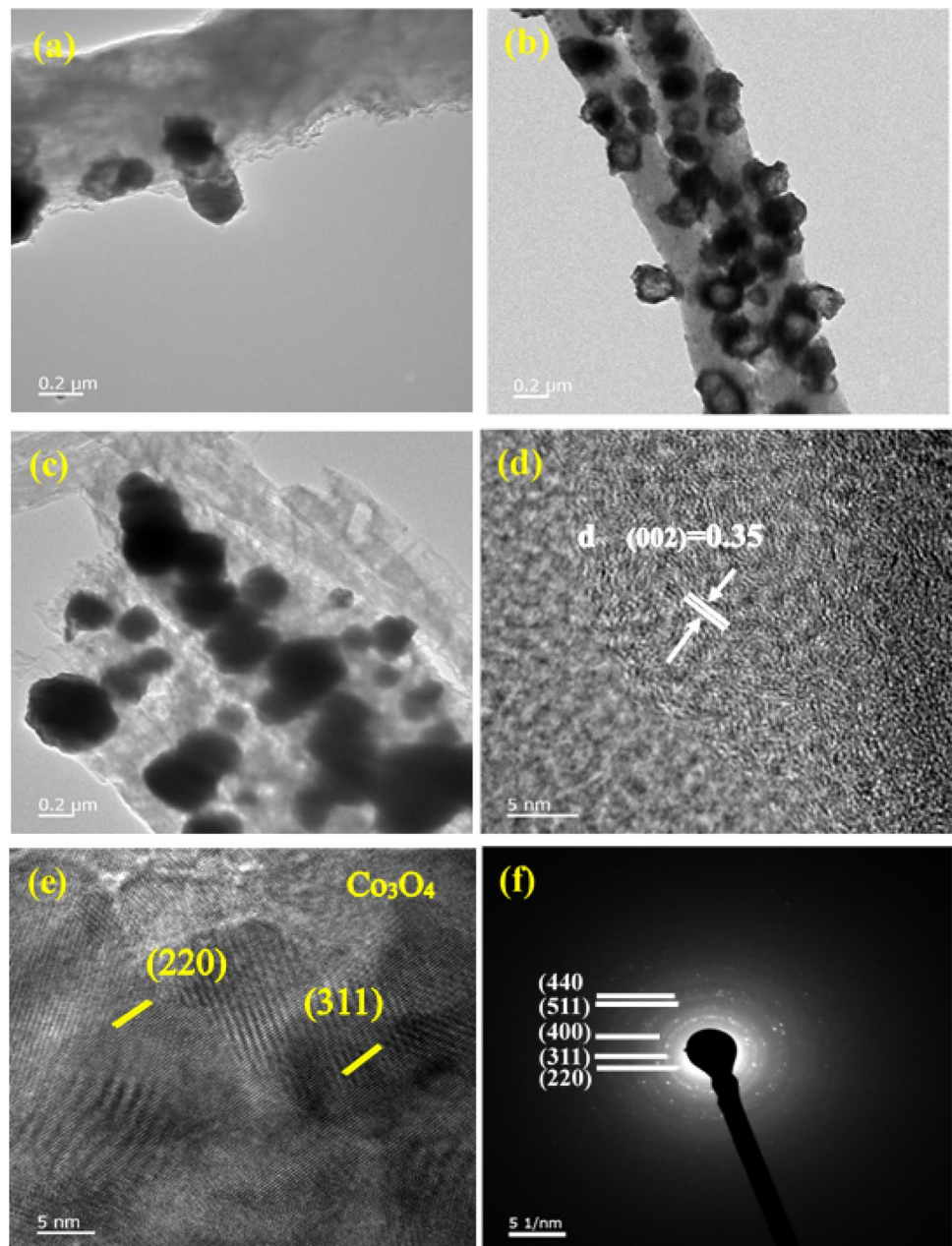


with the increase of particle size (Fig. 1d). Moreover, the element mappings in Fig. 2 demonstrate the existence and uniform dispersion of Co_3O_4 NPs. The energy-dispersive X-ray spectroscopy (EDS) analysis also proves the existence of C, N, O, and Co in the samples, which could be attributed to carbon, Co_3O_4 , and the retention of N from PAN.

The transmission electron microscopy (TEM) images (Fig. 3) further reveal the detailed morphologies and structures of Co_3O_4 -CNFs. Figure 3a, b show that the Co_3O_4 NPs exhibit a hollow structure with diameters ranging from 120 to 600 nm. As displayed, the hollow Co_3O_4 NPs are formed by Kirkendall effect, which refers to higher diffusion rate of

outward Co than that of inward O [64]. In consequence, it enabled hollow Co_3O_4 NPs to form. When the calcining temperature was up to 900 °C, the hollow Co_3O_4 NPs turned into solid ones, and the size of these nanoparticles became large. It is suspected that Co/CoO is not transformed into Co_3O_4 crystals completely after oxidation process. In the meantime, the amorphous carbon transformed to graphitic carbon partially (Fig. 3c, d). More detailed structures of Co_3O_4 -CNFs are demonstrated through the high-resolution TEM (HRTEM) and selected area electron diffraction (SAED). The measured d-spacings distance of carbon layers is 0.35 nm, assigning to the (002) plane of carbon (Fig. 3d). And the lattice fringe

Fig. 3 TEM images of **a** Co_3O_4 -CNFs-700, **b** Co_3O_4 -CNFs-800, and **c** Co_3O_4 -CNFs-900, respectively; **d**, **e** HRTEM images of Co_3O_4 -CNFs-800, and **f** the corresponding SAED pattern



d-spacings of 0.285 and 0.243 nm related to the (220) and (311) planes of Co_3O_4 (Fig. 3e). The corresponding selected-area electronic diffraction (SAED) pattern shows a polycrystalline nature of the Co_3O_4 , and the diffraction rings can be indexed to (220), (311), (400), (511), and (440) crystallographic planes of Co_3O_4 crystal (Fig. 3f).

The formation of Co_3O_4 and graphitic carbon could be confirmed by X-ray diffraction (XRD), Raman, and X-ray photoelectron spectroscopy (XPS). Figure 4a shows the X-ray diffraction (XRD) pattern of Co_3O_4 -CNF products. As shown, the six characteristic peaks of Co_3O_4 -CNFs-900 at 31.6, 36.8, 45.2, 55.4, 59.5, and 65.1° can be corresponded to the lattice planes of (220), (311), (400), (422), (511), and (440), respectively, which belong to the JCPDS card no. 42–1467 of Co_3O_4 . The broad diffraction peak located at 24.7° index to the (002) diffraction plane of carbon fibers. The diffraction peaks of Co_3O_4 -CNFs-700 are not apparent, which may be due to the trace amount of Co_3O_4 . From the XRD pattern of three different calcination temperatures, one can see that with the increase of temperature, the carbon and Co_3O_4 diffraction peak intensity increases. The XRD results reveal that Co_3O_4 -CNFs include an amorphous carbon and Co_3O_4 , which could greatly affect their electrochemical performance.

Raman spectroscopy is the standard technique to investigate the structure and graphitic degree of carbon materials (Fig. 4b). Raman spectra of Co_3O_4 -CNFs exhibit two visible peaks at around 1330 and 1590 cm^{-1} , corresponding to the D (ascribed to defects and disordered carbon at the edges of the sp^2 domain) and G (related to the E_{2g} in-plane vibration of the graphite lattice of the C sp^2 atom) bands, respectively [65–68]. The degree of graphitization could be calculated by the intensity ratio of D and G band (ID/IG).

As we know, the higher the ratio is, the more disordered the carbon material will be. For Co_3O_4 -CNFs-700, the ID/IG value is 1.13. With increase of calcination temperature, the ID/IG of Co_3O_4 -CNFs-800 decreased to 1.05, indicating the increase content of graphite carbon. Furthermore, the Raman spectra of Co_3O_4 -CNFs display characteristic peaks at 189, 466, 517, and 673 cm^{-1} , corresponding to F_{2g}^1 , E_{2g} , F_{2g}^2 , and A_{1g} vibration modes of Co_3O_4 , respectively, which is in good agreement with XRD results [69, 70].

The chemical composition of Co_3O_4 -CNFs was further examined by X-ray photoelectron spectroscopy (XPS) (Fig. 5). Three typical peaks corresponding to the binding energies of C 1s, N 1s, and Co 2p are observed. As shown in Fig. 5, the XPS spectrum of C 1s can be deconvoluted into four peaks with binding energies at 284.7, 285.4, 286.6, and 289.5 eV, corresponding to sp^2 graphitic carbon, C–C, C–O, and O–C=O, respectively [71, 72]. The graphitic carbon was increased with the calcination temperature, which could enhance the electron transfer efficiency of CNFs. And the import of doped nitrogen from pyrolysis of PAN is also conducive to promote the conductivity of CNFs, resulting in the improvement of electrochemical performance [73–75]. The high-resolution N 1s spectrum can be further fitted into three individual peaks, which are centered at 398.6, 400.2, and 401.4 eV, corresponding to the pyridinic N, pyrrolic N, and graphitic N, respectively [76–78]. At the same time, the peak at 401.4 eV of Co_3O_4 -CNFs-900 is the strongest, and the corresponding graphitic N can improve the conductivity of CNFs, which indirectly indicates that the electrochemical performance of Co_3O_4 -CNFs-900 sample is optimal. What is more, the high-resolution spectra of Co 2p of the three samples displayed two similar XPS peaks with two weaker satellite peaks. The spectra of Co $2p_{3/2}$ peak reveal binding

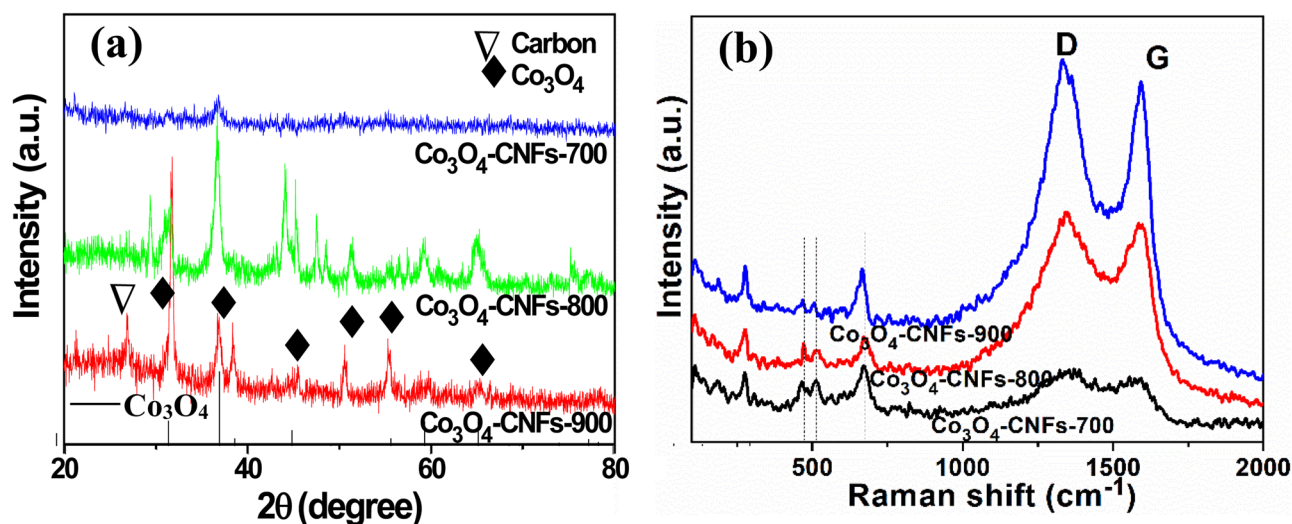


Fig. 4 a XRD patterns of Co_3O_4 -CNFs samples and b Raman spectrum.

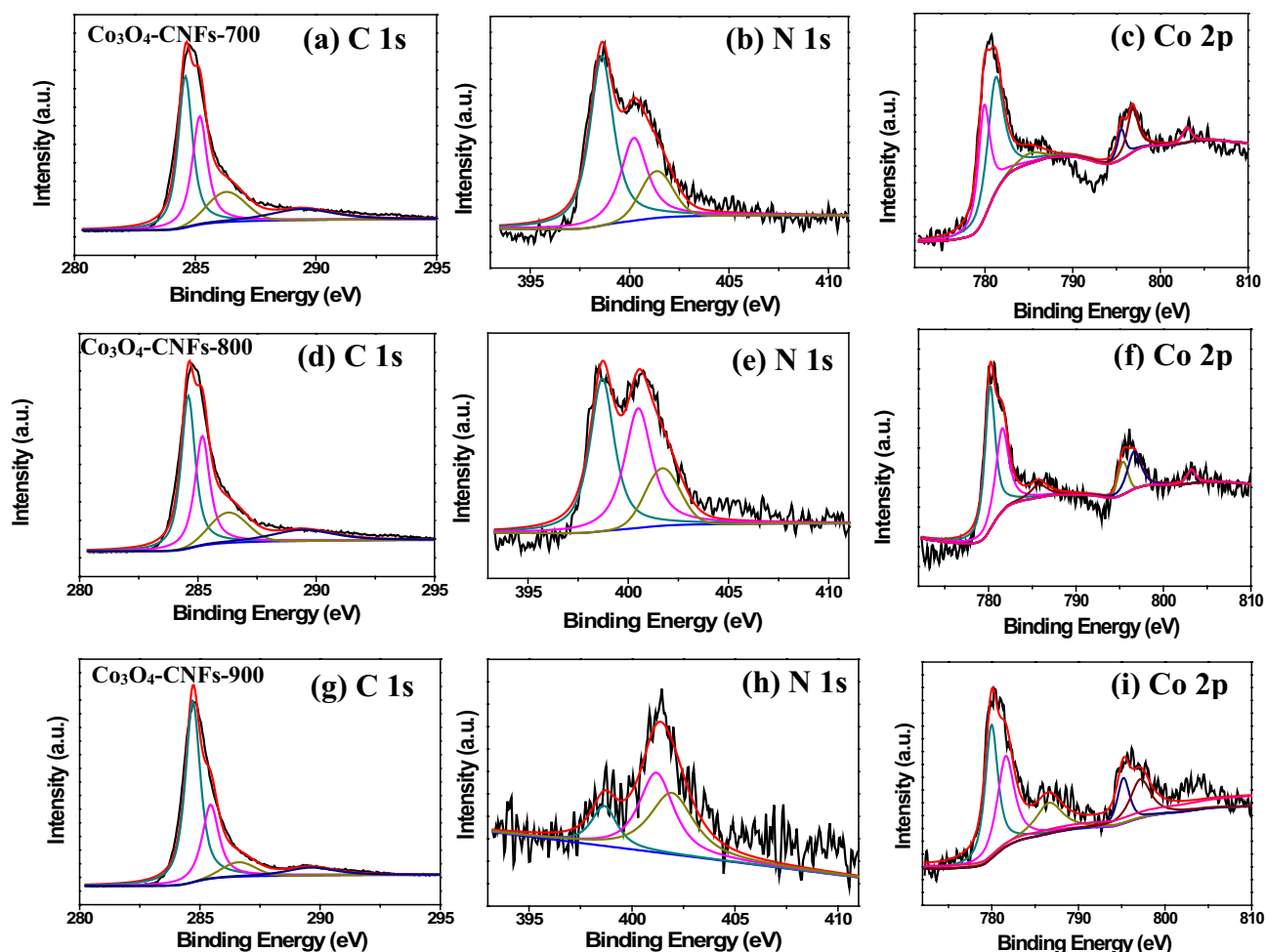


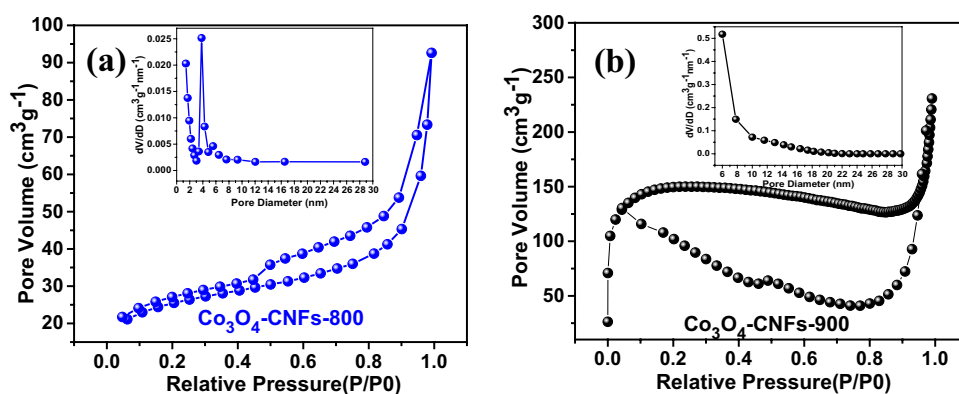
Fig. 5 XPS spectrum of Co_3O_4 -CNFs, high-resolution C1s, N1s, and Co2p spectra

energy at 795.6 and 780.0 eV, which is in good agreement with the reported data on Co_3O_4 , further confirming the presence of Co_3O_4 in Co_3O_4 -CNFs composites [79]. The principle peaks can be deconvoluted into four sub peaks, which the fitting peaks located at 781.3 and 796.9 eV are assigned to Co^{2+} , and the 779.9 and 795.6 eV peaks are indexed to Co^{3+} [80]. The integrated intensity ratio of Co^{2+} to Co^{3+} of Co_3O_4 -CNFs decreased at first and then increased with the increasing calcining temperature, which indicated that more Co^{2+} ions are generated accompanying with the generation of oxygen vacancies [81]. Benefiting from the advantage of high calcination temperature, high graphitic carbon and more oxygen vacancies will improve the specific capacity of the samples. Therefore, TEM, XRD, and XPS demonstrated that the homogeneously dispersed nanocrystalline Co_3O_4 and conductive carbon species existed in Co_3O_4 -CNFs, which is desired for high-performance electrochemical materials.

The nitrogen adsorption desorption measurement has been taken as represented in Fig. 6. The Co_3O_4 -CNFs

present a type-IV isotherm with a distinct hysteresis loop, showing the characteristics of a mesoporous material. Furthermore, the isotherm shows a sharp increase at high relative pressures ($p/p_0 > 0.9$), demonstrating the macroporous characteristics [82–85]. Meanwhile, the corresponding pore size distribution curves (inset figures) also show the hierarchical pore structure of the Co_3O_4 -CNFs. The total BET surface areas can be determined to be $86.31 \text{ m}^2 \text{ g}^{-1}$ and $347.62 \text{ m}^2 \text{ g}^{-1}$, respectively. The specific surface areas increase with the increasing calcination temperature, which can be ascribed to the high calcination temperature that offered more specific surface and pore structure by sublimation of the TPA component and degradation of lignin. That could be beneficial for achieving higher electrochemical performance due to the moderate BET surface areas by providing passageways for the ion to pass through with the hierarchical pore structure. TGA measurement was performed in air from 30 to 600 °C to confirm the content of Co_3O_4 in Co_3O_4 -CNF sample (Fig. 7). The TGA curve of the Co_3O_4 -CNFs-900 sample presented a sharp

Fig. 6 N_2 absorption and desorption isotherms and pore size distribution of the Co_3O_4 -CNFs (inset figures)



weight loss at 260–510 $^{\circ}C$, which corresponds to the burning of carbon matrix [86–89]. Correspondingly, the DTG curve showed the main peaks at 395 $^{\circ}C$ and 468 $^{\circ}C$. Therefore, the TGA-DTG analysis determined that the content percentage of Co_3O_4 was 54.9% in the sample.

At the same time, the flexibility and foldability of the prepared Co_3O_4 -CNFs were studied. As shown in Fig. 8, after three folds, Co_3O_4 -CNFs could fully recover to their initial state, indicating their excellent flexibility and high mechanical strength. Such outstanding flexibility and bending of Co_3O_4 -CNFs give them a superior benefit for applications in flexible devices. The results also revealed that the porous structure of as-obtained CNFs could conduce to improve the flexibility of CNFs by using TPA as a soft template, because the sublimation of TPA in carbonization process resulted in the formation of porous structure. The porous morphology of obtained CNFs could also be observed in TEM images (Fig. 3). When Co_3O_4 -CNFs were bent, the flexibility of CNFs can be enhanced. Because the pore structure can expand stress distribution, thus reducing the stress, then its flexural performance can be optimized. At the same time, the existence of micro- and mesoporous can enhance ion transport rate and strengthen the charge

accommodation, resulting in excellent electrochemical properties of the samples.

The FTIR spectra of lignin PAN, PAN/cobalt salt precursors, and Co_3O_4 -CNFs-700 fibers are shown in Fig. 9a. In lignin, the characteristic peaks at 1607 cm^{-1} and 1411–1587 cm^{-1} correspond to C=C in aromatic ring, which is prevalent in lignin structure [90–94]. The band at 1037 cm^{-1} represents the absorption peak of carbonyl group [95–98]. The peak at 1456 cm^{-1} can be assigned to $-CH_2$ bending vibration, and peak at 2245 cm^{-1} is ascribed to $C\equiv N$ stretching vibration in PAN chain. As for the spectra of PAN-cobalt precursor composite fibers, the C=C peak in aromatic ring is still observed. Compared with lignin spectrum, the peak strength of carbonyl group at 1037 cm^{-1} is significantly reduced in lignin, which indirectly indicates that lignin interacts with the nitrile group in PAN. The peaks at 1594 and 1296 cm^{-1} can be attributed to N–O stretching vibration of PAN. Compared with PAN, the $C\equiv N$ peak strength of PAN/cobalt precursor at 2245 cm^{-1} was also significantly reduced, indicating that with the introduction of lignin, part of $C\equiv N$ bond is converted to $C=N$ in PAN, and the N–O bond is formed with $-O-H$ of lignin [99–106]. The possible connections between PAN and lignin are shown in Fig. 9b.

Fig. 7 TGA and DTG curves of the Co_3O_4 -CNFs-900 sample in air

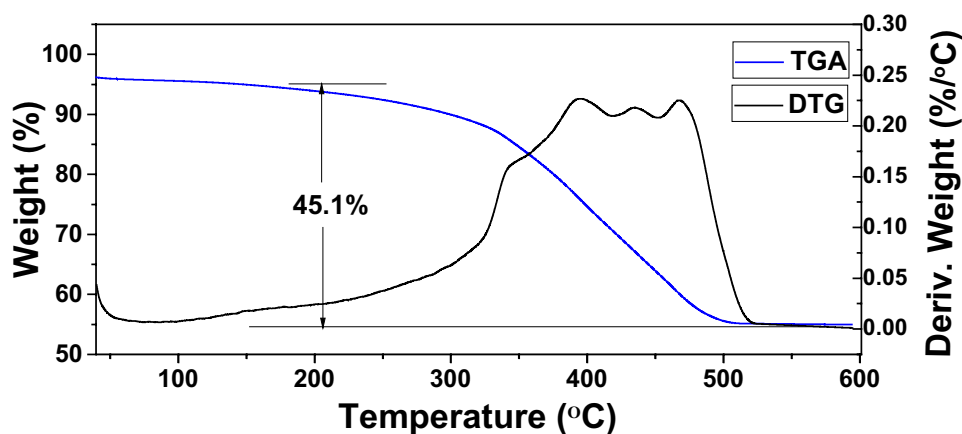
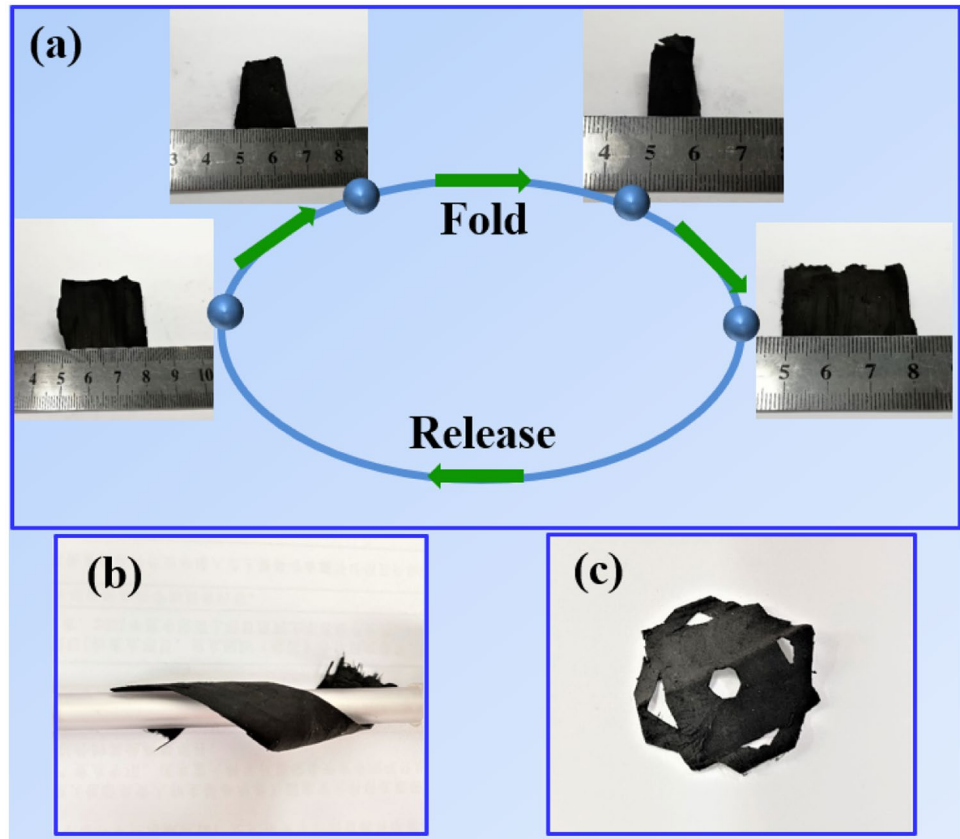


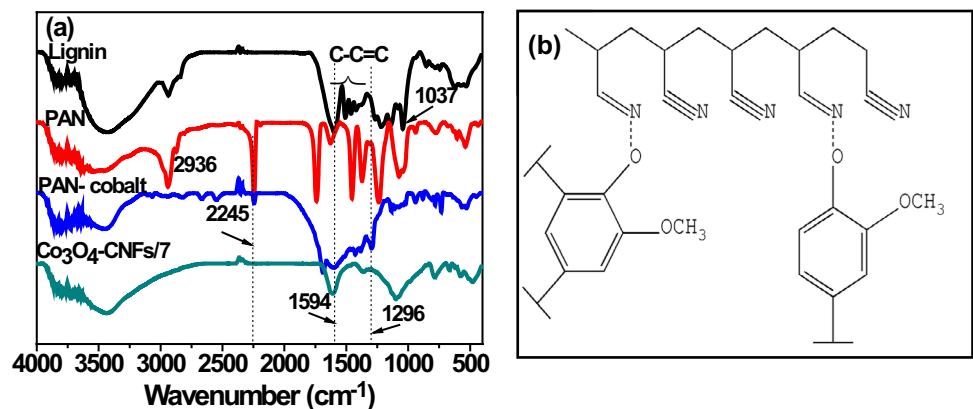
Fig. 8 **a** The digital photographs of Co_3O_4 -CNFs sample under 3-folded and could recover its initial state. **b** Digital photo images of the flexible Co_3O_4 -CNFs sample. **c** The paper cutting cut by Co_3O_4 -CNFs sample.



The obtained flexible Co_3O_4 -CNFs were directly used as working electrode without adding binder or conducting agent, and the electrochemical properties were evaluated by CV, GCD, and EIS. Figure 10 shows the CV curves of different Co_3O_4 -CNFs samples at scan rates ranging from 5 to 500 mV/s. As can be seen, the CV curves of three samples maintain a rectangular shape in the operating voltage range of -0.2 – 0.8 V, indicating their ideal electrochemical performance with a fast charging-discharging process. Among them, Co_3O_4 -CNFs-900 has the largest CV curve area, indicating its maximum capacitance. At the same time, Co_3O_4 -CNFs have an arch curve at a scanning rate of

5 mV/s, indicating that it also has pseudocapacitive properties (Fig. 10d). The specific capacitance of Co_3O_4 -CNFs-900 is the largest, possibly because the graphitization degree of carbon fiber increases with the increase of calcination temperature [107–117]. Besides, Co_3O_4 particles increase and their size becomes larger, which may be caused by the increase in the number of micropores. The CV curves of Co_3O_4 -CNFs-700 and Co_3O_4 -CNFs-800 have larger deformation than Co_3O_4 -CNFs-900 at high scanning rates, indicating that their impedance is larger, and there is a charge transfer resistance, which mainly comes from the resistivity of samples and the ion diffusion resistance in micropores [111, 112].

Fig. 9 **a** FTIR spectra of lignin, PAN, PAN-cobalt salt precursors, and Co_3O_4 -CNFs-700 and **b** possible lignin/PAN interaction.



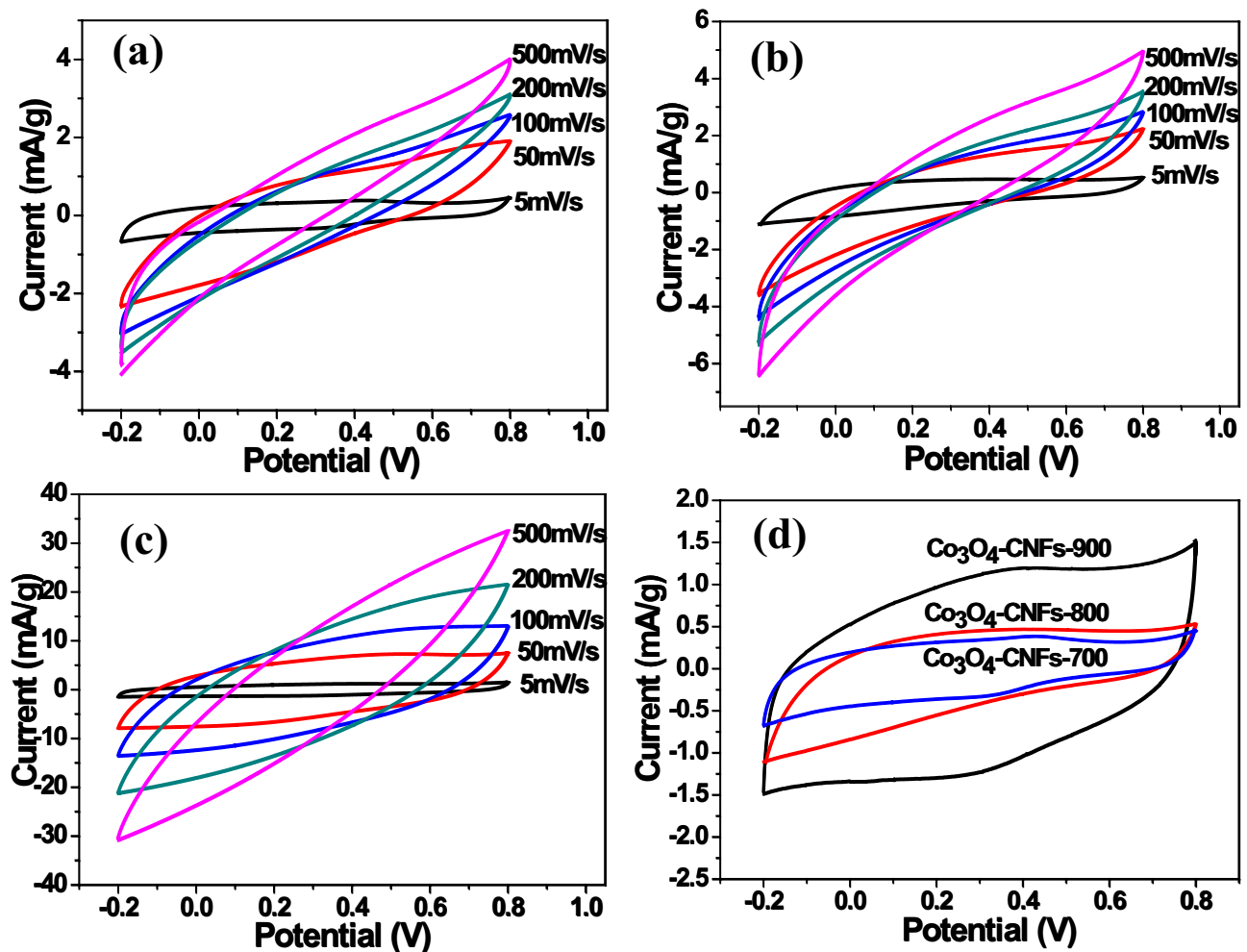


Fig. 10 CV curves of **a** Co₃O₄-CNFs-700, **b** Co₃O₄-CNFs-800, and **c** Co₃O₄-CNFs-900 at different scan rates, i.e., 5, 50, 100, 200, and 500 mV s⁻¹; **d** CV curves of different Co₃O₄-CNFs samples at scan rates of 5 mV s⁻¹

In addition, Fig. 11 shows the GCD curves of Co₃O₄-CNFs at various current densities ranging from 0.1 to 2 A g⁻¹. It can be observed that the GCD curves exhibit a similar triangle shape in the range of -0.2–0.8 V, indicating that the double-layer capacitance is reversible at wide range of current density. Meanwhile, the GCD curves show a small deviation, which also indicates that the Co₃O₄ has pseudocapacitance. At higher calcination temperature, Co₃O₄-CNFs-900 has higher graphitization and more micro- and mesopores. The unique 3D Co₃O₄-CNFs network pore structure could reduce the electron pathways between electrode and electrolyte, and the pores could serve as electrolyte reservoir. Thus, it shows the longest discharge time, indicating that they have the highest specific capacitance, which is in good agreement with the CV results. Figure 12a compares the capacitance values of different samples at different current densities. When the current density is 0.1 A g⁻¹, the specific capacitance of Co₃O₄-CNFs-900,

Co₃O₄-CNFs-800, and Co₃O₄-CNFs-700 is 369, 125, and 119 F g⁻¹, respectively. When the current density is 2 A g⁻¹, the specific capacitance of Co₃O₄-CNFs-900 is 181 F g⁻¹, and its capacitance retention rate is 49%, exceeding most other flexible or self-standing similar carbon electrodes (Table 1). Besides, after the cycling measurements, the hierarchical porous CNFs network, as well as the morphology of Co₃O₄, was almost maintained, indicating the excellent structural stability of the electrode during the electrochemical process (Fig. 12a). Its good rate performance is mainly due to the fact that the porous CNF network structure can shorten the electronic pathway between electrode and electrolyte, and the doping of N and Co₃O₄ can further improve its electrical conductivity. However, when the current density increases, the specific capacitance of Co₃O₄-CNFs also tends to decrease gradually, which is due to the space limitation of CNF itself, so that only part of ions can penetrate the micropores.

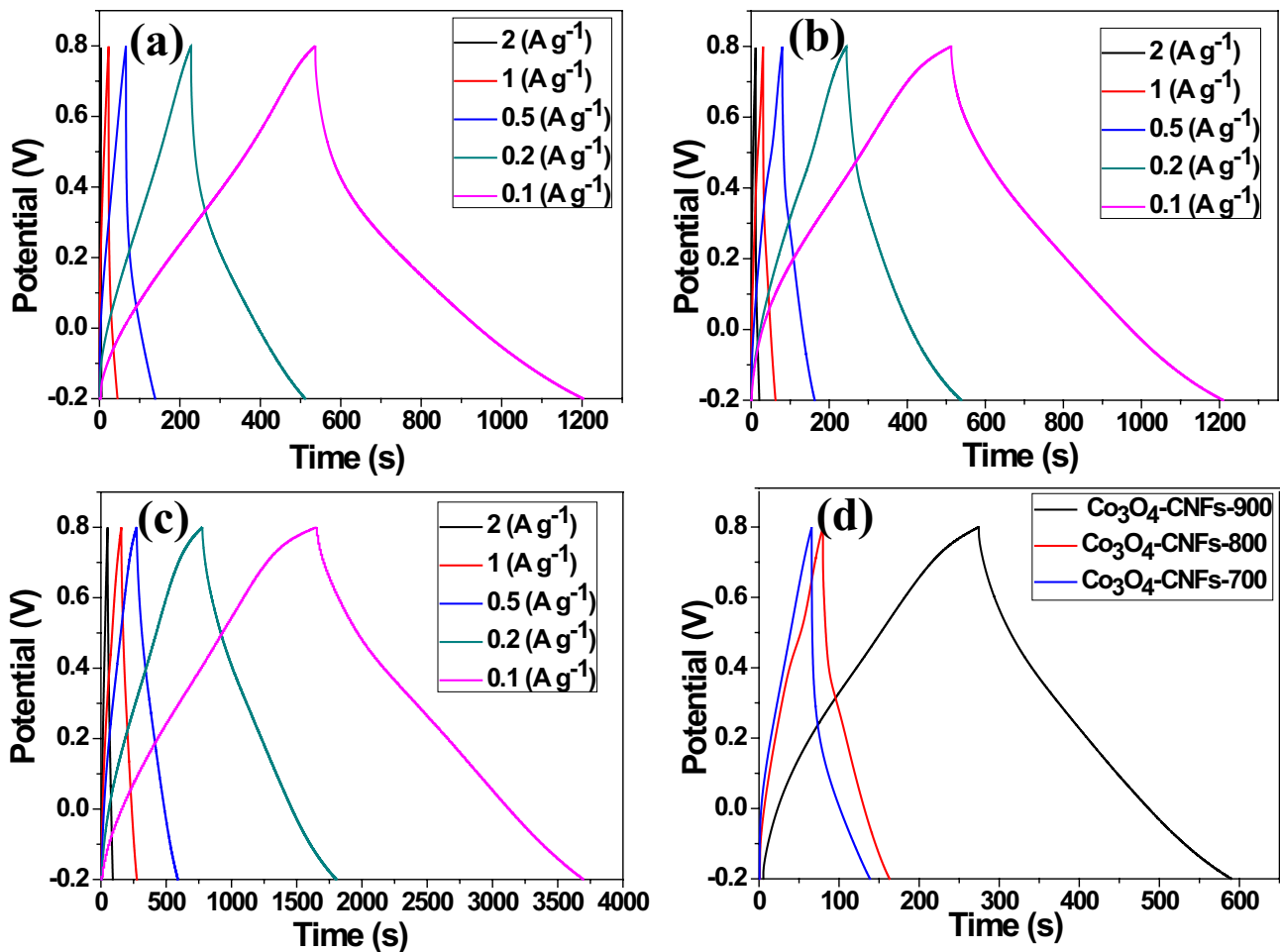


Fig. 11 GCD curves of **a** Co₃O₄-CNFs-700, **b** Co₃O₄-CNFs-800, and **c** Co₃O₄-CNFs-900 under different current densities; **d** GCD curves of different Co₃O₄-CNFs samples at the current density of 1 A g⁻¹

Figure 12b shows the Nyquist plot of Co₃O₄-CNFs in the frequency range of 100 kHz to 0.01 Hz. Obviously, the Nyquist diagram of Co₃O₄-CNFs-900 consists of semi-circular arc in the high-frequency region and double-layer capacitance response diagonal in the low-frequency region. The equivalent series resistance (*R_s*) of Co₃O₄-CNFs samples is 3.9, 2.53, and 2.63 Ω, respectively, indicating that the three samples have good conductivity. The interfacial charge transfer resistance (*R_{ct}*) of Co₃O₄-CNFs-900 can

be obtained from the diameter of solid axis semicircle in the high-frequency region, which is 6.93 Ω, while the *R_{ct}* of Co₃O₄-CNFs-800 and Co₃O₄-CNFs-700 is higher. The results further indicate that the improvement of graphitization degree, the increase of Co₃O₄ particles, and the hierarchical pore numbers can improve the electronic conductivity of CNFs. In addition, the Warburg diffusion line of Co₃O₄-CNFs-900 is smaller, while the Warburg resistance region of Co₃O₄-CNFs-800 and Co₃O₄-CNFs-700 is more

Table 1 Comparison of specific capacitive performances of Co₃O₄-CNFs in this work and similar materials reported in the literature

Electrode materials	Voltage window	Electrolyte	Specific capacitance	Reference
Tubular porous Co ₃ O ₄ /carbon	0–0.5 V	6 M KOH	284.2 F/g at 1 A/g	[116]
NFC/porous Co ₃ O ₄	0–0.35 V	6 M KOH	594.8 mF cm ⁻² at 5 mV s ⁻¹	[117]
Porous carbon/Co ₃ O ₄	0–0.4 V	3 M KOH	423 F/g at 1 A/g	[118]
CNF/Co(OH) ₂	–1.0–0 V	6 M KOH	135 F/g at 2 A/g	[119]
Co ₃ O ₄ /graphene	–0.2–0.4 V	2 M KOH	60 F/g at 2 A/g	[120]
Co ₃ O ₄	–0.1–0.5 V	6 M KOH	162 F/g at 2.75 A/g	[121]
Co ₃ O ₄ -CNFs-900	–0.2–0.8 V	3 M KOH	181 at 2 A/g	This work

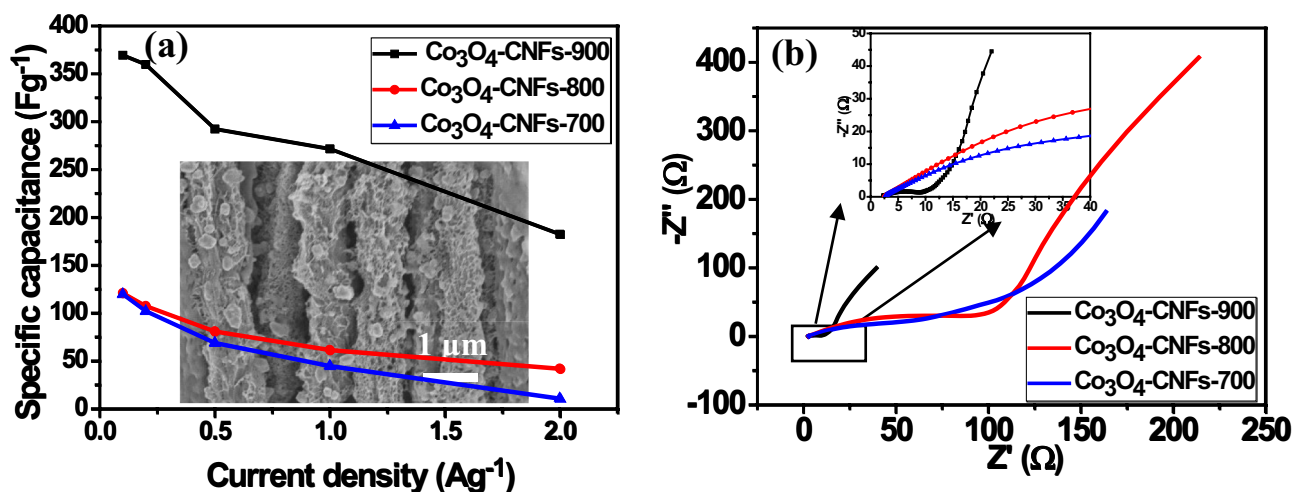


Fig. 12 **a** Specific capacity diagram at different current densities (the inset is the SEM image after cycling), and **b** alternating current impedance graph of Co_3O_4 -CNF samples.

obvious. Warburg resistance region is short, indicating a high ion adsorption efficiency, and electrolyte ions can effectively diffuse at the electrode interface [122–128]. At lower frequency, Co_3O_4 -CNFs-900 also shows a higher slope, demonstrating good double-layer capacitance behavior. It can also be concluded that Co_3O_4 -CNFs-900 has high specific capacitance performance from EIS analysis.

4 Conclusions

In summary, we have successfully fabricated the flexible, free-standing, and porous Co_3O_4 -CNFs by electrospinning technique and the followed carbonization process. TPA was used as a soft template during the synthesis process to form hierarchical pore structure, which significantly improved the porosity and flexibility of the as-prepared Co_3O_4 -CNFs. The calcination temperatures played an important role in the electrochemical properties of the obtained Co_3O_4 -CNFs. For example, the Co_3O_4 -CNFs-900 showed the highest specific capacitance of 369 F g^{-1} at the current density of 0.1 A g^{-1} , which is about three times higher than that of the Co_3O_4 -CNFs-700 at the same testing condition. Also, the Co_3O_4 -CNFs-900 electrode exhibited good rate capability, i.e., the specific capacitance retained to 181 F g^{-1} at a high current density of 2 A g^{-1} . The superior electrochemical performance was mainly attributed to the high electrical conductivity, the extra pseudocapacitance contributed by Co_3O_4 and N heteroatom, and the hierarchical pore structure (providing fast transport channels for ions). With the advantages of excellent mechanical flexibility, high conductivity, and high specific capacitance, the free-standing Co_3O_4 -CNFs

could be a promising candidate for the development of high-performance flexible energy storage devices.

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Declarations

Conflict of interest The authors declare no competing interests.

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