ORIGINAL RESEARCH

Highly sensitive and selective 2‑nitroaniline chemical sensor based on Ce‑doped SnO2 nanosheets/Nafion‑modified glassy carbon electrode

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Abstract

In this paper, pure SnO₂ and Ce-doped SnO₂ nanosheets were synthesized through a facile hydrothermal method. The synthesized materials were characterized by diferent techniques for their physico-chemical properties. The XRD data indicated the characteristic tetragonal rutile crystal phase for SnO₂. Ce doping was ascertained by the presence of the diffraction peaks of $CeO₂$ in all the doped samples of the SnO₂ nanosheets. FESEM images revealed highly rough surfaces as well as the agglomeration of a large number of small nanoparticles of multiple shapes to form nanosheets like morphologies for pure $SnO₂$ and Ce-doped SnO2. Electrochemical techniques like cyclic voltammetry (CV), electrochemical impedance spectroscopy (EIS), and chronoamperometry were applied to demonstrate the electrochemical performances of the pure $SnO₂$ and Ce-doped SnO₂ nanosheets/Nafion-modified glassy carbon electrode (GCE). The 3% Ce-doped SnO₂ nanosheet/Nafion-modified GCE showed a remarkable sensitivity of 0.9986 $\mu A \mu M^{-1}$ cm⁻² over a linear dynamic range of 0.5–20.3 μ M. The corresponding linear regression equation was Ip (μ A)=0.0709 [2-nitroaniline (μ M)]+0.1385 with R^2 =0.99325. The LOD of the modified sensor was found to be 6.3 ± 0.1 nM at the signal-to-noise ratio of $S/N = 3$. The newly developed sensor electrode exhibited good selectivity toward 2-nitroaniline in the presence of common interfering species.

Keywords Ce-doped \cdot SnO₂ \cdot Nanosheets \cdot Electrochemical sensor \cdot 2-Nitroaniline

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1 Introduction

Tin dioxide $(SnO₂)$ due to its excellent optical and electrical properties has been reported for applications such as supercapacitors [[1\]](#page-10-0), gas and chemical sensors [\[2\]](#page-10-1), lithium-ion batteries [\[3](#page-10-2)], dye-sensitized solar cells [[4\]](#page-10-3), antimicrobial agents [\[5\]](#page-10-4), electrochemical biosensors [\[6](#page-10-5)], and electrocatalysis [\[7](#page-10-6)]. A wide variety of applications associated with $SnO₂$ nanostructures are due to the ease of synthesis of $SnO₂$ nano/ microstructures with versatile morphologies, n-type semiconducting nature, a wide-band gap of 3.6 eV, high carrier density, and excellent chemical and thermal stabilities [\[8](#page-10-7)]. However, for further improvement for specifc applications, many methods have been adopted. Among the various methods adopted, doping is considered as one of the best methods for altering the microstructural, surface morphology, electronic, optical, and bandgap properties of the $SnO₂$ nano/ microstructures [[9](#page-10-8)].

Cerium-a lanthanide element has received extensive attention as a dopant material for semiconductor metal oxides due to its mixed valences of $+3$ and $+4$, availability of vacant 4f orbitals, redox characteristics ($Ce^{4+} \leftrightarrow Ce^{3+}$), rapid oxygen ion mobility, and electrocatalytic properties [\[10–](#page-10-9)[13\]](#page-10-10). The reduction of Ce^{4+} to Ce^{3+} ion has been found to initiate the generation of lattice mobile oxygen, which in turn results in the formation of surface oxygen vacancy defects. These defects act as hot spots for the synergistic binding of other nanomaterials especially the metal oxides [\[14](#page-10-11), [15\]](#page-10-12). Ce-doped metal oxides such as Ce-doped ZnO nanoparticles $[16, 17]$ $[16, 17]$ $[16, 17]$ $[16, 17]$, Ce-doped SnO₂ thin films $[18]$, Ce-doped NiO nanoparticles [\[19](#page-10-16)], Ce-doped In_2O_3 nanostructure [\[20](#page-10-17)], Ce-doped hierarchical flower-like In_2O_3 microspheres [\[21](#page-10-18)], Ce-doped $Fe₃O₄$ magnetic particles [[22](#page-10-19)], Ce-doped NiO micro-flowers $[23]$ $[23]$, Ce-doped TiO₂ nanoparticles $[24]$ $[24]$ $[24]$, Cedoped WO_3 with raspberry-like architecture [[25\]](#page-10-22), Ce-doped CuO nanoparticles [\[26](#page-10-23)], and many more have been explored and reported for a variety of applications.

Among the various applications, the electrochemical sensors based on Ce-doped semiconductor metal oxide nanomaterials have extensively been fabricated to detect hazardous and toxic chemicals. One such class of hazardous materials is of nitroanilines which are the derivatives of aniline and exist in diferent isomeric forms. These chemicals are widely used as raw materials in industries such as dyes, pharmaceuticals, polymers, developing agents, gasoline, insecticides, pesticides, rubber, photo-stabilizers, antioxidants, explosives, anti-aging agents, and paints [\[27,](#page-10-24) [28\]](#page-10-25). Excess release of these chemicals as effluents into the water bodies has a devastating efect not only on the environment but also on human health. In living beings, these chemicals can potentially cause damage to the liver, kidneys, DNA mutations upon inhalation, and even skin diseases. As far as environmental efects are concerned, nitroanilines are non-biodegradable and hence are considered as typical organic pollutants [[29](#page-10-26)]. Many metal oxide semiconductor materials have been explored as transducers for the sensing of nitro-anilines. Bitter gourd–shaped Yb_2O_3 -doped ZnO nanostructures [30], ZnO nanoflower $CeO₂$ nanoparticles [\[31\]](#page-11-1), ZnO nanorods on fluorine-doped $SnO₂$ [[27\]](#page-10-24), $Sm₂O₃$ -doped ZnO beech fern hierarchical structures [[32\]](#page-11-2), novel sphere-like $Co₂SnO₄$ [\[33\]](#page-11-3), K⁺ intercalated $MnO₂-rGO$ composite [[34\]](#page-11-4), Yb-doped ZnO nanopencils [[35](#page-11-5)], CdO-ZnO hexagonal nanocones [[36\]](#page-11-6), etc*.* have been reported as electrochemical sensor materials towards nitroaniline isomers. Since Ce-doped $SnO₂$ nanomaterials show improved electric and electronic properties as compared to pure $SnO₂$, they can be potential materials for electrochemical sensing applications. Further, limited studies have been conducted to use Ce-doped $SnO₂$ nanomaterials for 2-nitroaniline detection.

Therefore, the main focus of the present research conducted herein was to synthesize pure $SnO₂$ and Ce-doped $SnO₂$ nanosheets with different Ce- concentrations through a facile hydrothermal method. As-synthesized materials were characterized through diferent techniques to confrm the composition, morphology, crystallinity, and optical and vibrational properties. The electrochemical sensing behavior of Ce-doped $SnO₂$ nanosheets was compared with that of pure $SnO₂$ nanosheets for 2-nitroaniline. It was found that the 3% Ce-doped $SnO₂$ nanosheets/Nafion-modified GCE exhibited.

2 Experimental details

2.1 Materials

To synthesize Ce-doped $SnO₂$ nanosheets, all chemicals of AR grade were procured from Sigma-Aldrich and used as obtained without any further purifcation. For the synthesis, various chemicals, i.e., stannous chloride dihydrate $[SnCl₂·2H₂O]$, cerium(III) chloride heptahydrate $[CeCl₃·7H₂O]$, hexamethylenetetramine $[HMTA; C₆H₁₂N₄],$ and ammonium hydroxide [NH₄OH], were used. For the sensing purpose, 2-nitroaniline (2-NA) was also procured from Sigma-Aldrich and used as obtained. To prepare all solutions for the synthesis and sensing applications, deionized (DI) water was used.

2.2 Synthesis of pure SnO₂ and Ce-doped SnO₂ nanosheets

A facile hydrothermal process was used to synthesize pure and Ce-doped $SnO₂$ nanosheets. For the synthesis of $SnO₂$ nanosheets, 100 mM equimolar aqueous solutions of $SnCl₂·2H₂O$ and HMTA (40 mL each) were mixed well under stirring for 30 min. Consequently, several drops of $NH₄OH$ were added, to maintain the solution $pH=11$, into the resultant mixture and stirred again for further 30 min. After vigorous stirring, the obtained solution was transferred to a Tefon-lined stainless steel autoclave which was sealed and heated at 160 °C for 8 h. After the desired reaction time, the autoclave was cooled to room temperature and the obtained precipitate was centrifuged and washed several times with DI water and ethanol. Finally, the obtained white precipitate was dried at room temperature and annealed at 450 \degree C for 3 h.

Ce-doped $SnO₂$ nanosheets were prepared following the same synthetic procedure of $SnO₂$ nanosheets; however, for Ce doping, different concentrations of $CeCl₃·7H₂O$ (1 mM for 1%, 3 mM for 3% and 5 mM for 5%), made in 20 mL DI water, were added in the mixture of $SnCl₂·2H₂O$ and HMTA, then completed the same hydrothermal procedure of $SnO₂$ nanosheets as described above.

2.3 Characterizations

The synthesized pure $SnO₂$ and Ce-doped $SnO₂$ materials were analyzed by several techniques to examine the morphologies, crystal structures and phases, composition, and optical and vibrational properties. Morphology, composition, purity, and elemental distribution in the material matrix were analyzed through feld-emission scanning electron microscopy (FESEM: JEOL-JSM-7600F, Hitachi, Japan) attached with EDS for elemental mapping and composition. The crystallinity, crystal size, and crystal phases were determined by X-ray difractometer (XRD; PANalyticalX'Pert PRO; Cu-K α radiations source with $\lambda = 0.1542$ nm) in the range of 20–80° with a scan speed of 2°/min. Bandgap energies were determined by UV–vis spectroscopy (Perkin Elmer-UV/VIS, Lambda 950). Vibrational properties were examined by FTIR spectroscopy (Perkin Elmer-FTIR Spectrum-100). FTIR analysis of the synthesized nanomaterials was carried out by initially mixing the product homogeneously with powdered KBr (1% by weight of the sample) followed by palletization at high pressure. The scattering properties of the synthesized nanomaterials were investigated by Raman scattering (Perkin Elmer-Raman Station-400 series) at room temperature. Raman analysis was carried out by directly placing the dry powders of the synthesized nanomaterials below the probe.

2.4 Fabrication of 2‑nitroaniline chemical sensors

Initially, the glassy carbon electrode (GCE) with apparent surface area 0.071 cm^2 (BAS Inc. Japan) was thoroughly polished with a 1-μm polishing diamond followed by polishing with 0.05 μm alumina slurry. After that, the polished electrode was washed several times with distilled water and dried in an air oven at 100 °C for 1 h. The surface of the GCE was coated with a thin layer of hydrothermally synthesized sensor materials. A homogeneous suspension from the sensor materials was made using Nafon solution as a conducting binder to provide stability and frmness to the coated material on the surface of the electrode. Typically, 5.0 mg of active material was dispersed in 0.45 mL of propan-2-ol containing 0.05 mL of 5% Nafon, followed by ultra-sonication for 20 min to get a homogeneous suspension. Of this suspension, 1.5 μl was gently coated onto a clean GCE surface and dried at ambient conditions for 5 min, followed by drying at 60 °C for 20 min to get a homogeneous layer of active material on the GCE surface. A three-electrode electrochemical cell connected to an electrochemical workstation (Zahner Zennium, Germany) was used for various electro-analytical measurements. A Pt wire as a counter electrode, modifed GCE as a working electrode, and an Ag/AgCl (saturated KCl) electrode as a reference electrode were used. All the electro-analytical observations were recorded in 0.1 M

phosphate-buffered solution (PBS) of pH 7.0 at room temperature. The sensitivity of the fabricated sensor was determined from the slope of the calibration curve divided by the active area of the modifed electrode.

3 Results and discussion

3.1 Properties of the pure SnO₂ and Ce-doped SnO₂ nanosheets

The diffraction peaks of the $SnO₂$ lattice planes are shown in the form of an XRD spectrum in Fig. [1.](#page-2-0) Well-defned diffraction peaks corresponding to (110), (101), (200), (111), (211), (220), (002), (310), (112), (301), (202), and (321) diffraction planes are well-matched with the reported literature [[37–](#page-11-7)[39\]](#page-11-8) and JCPDS: 41–1445. The XRD data indicated the characteristic tetragonal rutile crystal phase of $SnO₂$. The presence of $CeO₂$ in all the prepared doped samples of the $SnO₂$ nanosheets is confirmed by the presence of a small but well-defned peak corresponding to (111) difraction plane which indicated the cubic fluorite phase of $CeO₂(JCPDS)$ card no. $81-0792$) $[40, 41]$ $[40, 41]$ $[40, 41]$ $[40, 41]$ $[40, 41]$. No other peak was detected, except for $SnO₂$ and $CeO₂$ in the XRD spectra of un-doped and doped samples which further confrmed the high purity of the samples. The crystallite size was calculated using the *Scherrer* equation (Eq. [1](#page-2-1)) [\[42](#page-11-11), [43\]](#page-11-12).

$$
d = \frac{0.89\lambda}{\beta \cos \theta} \tag{1}
$$

Here, λ = wavelength of X-rays (1.542 Å), β = full width half maximum (FWHM), and θ = Bragg angle of diffraction.

Fig. 1 XRD spectra of the pure $SnO₂$, 1%, 3%, and 5% Ce-doped SnO2 nanosheets

For analysis, the three most intense peaks corresponding to (110), (101), and (211) difraction planes were considered. The resulting parameters are tabulated in Table [1.](#page-3-0) The average crystallite sizes for pure $SnO₂$, 1%, 3%, and 5% Ce-doped $SnO₂$ nanosheets were found to be 31.07, 30.19, 32.10, and 33.86 nm, respectively. Nominal change in the diffraction angles for Ce-doped $SnO₂$ nanosheets as compared to pure $SnO₂$ further confirmed the replacement of the Sn^{4+} ions by Ce^{4+} ions from some of the lattice points from the tetragonal rutile crystal phase of $SnO₂$. However, for pure $SnO₂$ and 1% Ce-doped $SnO₂$ nanosheet diffraction angles were almost the same due to the very low concentration of the dopant. With the increase in dopant concentration difraction angles were found to increase.

The FESEM images of the pure $SnO₂$ and Ce-doped $SnO₂$ nanosheets are shown in Fig. [2a](#page-4-0)–p. Panoramic images reveal highly rough surfaces and illustrate that the pure, as well as Ce-doped $SnO₂$ nanosheets, are formed by the agglomeration of a large number of small nanoparticles of multiple shapes. High-resolution FESEM images showed that the thickness of the 3% Ce-doped $SnO₂$ $SnO₂$ $SnO₂$ nanosheets (Fig. 2j–l) was more as compared to 5% Ce-doped $SnO₂$ nanosheets (Fig. [2n](#page-4-0)–p). The cross-edge sections FESEM image of pure $SnO₂$ and Ce-doped $SnO₂$ nanosheets displayed a dense distribution and close packing of the nanoparticles to form layered nanosheets. For 3% Ce-doped SnO₂ nanosheets, a comb-like morphology was also observed from cross-edged FESEM images. Such small nanoparticle-layered nanosheets of pure $SnO₂$ and Ce-doped $SnO₂$ provide a large surface area for the chemisorption of analytes, required as an important pre-requisite to perform the role of electron mediator, by synthesized nanosheets for efficient and reliable electrochemical sensing.

EDS-selected area and the corresponding EDS spectrum of pure $SnO₂$ are shown in Fig. [3](#page-5-0)a and $(a₁)$, respectively. The EDS spectrum showed peaks for Sn and O only. The EDS spectrum without any other peaks corresponding to any impurities confirms the fact that the synthesized $SnO₂$ nanosheets have appreciable purity. To further confrm the EDS results, the composition and element distribution of the constituent elements of $SnO₂$ nanosheets were studied by the FESEM-elemental mapping technique (Fig. [3](#page-5-0) (a_2, a_3)). The uniform distribution of Sn and O content in the entire structure of the nanosheets lattice can be seen. Similar to pure $SnO₂$ nanosheets, the Cedoped SnO₂ nanosheets were also analyzed by EDS selected area, EDS spectra, and FESEM-elemental mapping technique. The doping of the Ce⁴⁺ into the crystal lattices of the $SnO₂$ nanosheets was confrmed for all the doped samples. The EDS spectra for all the doped nanosheets exhibited peaks for Sn, O, and Ce only. The fact was further supported by the FESEM-elemental mapping images as portrayed in Fig. [3](#page-5-0) (b_2-b_4) for 1% Ce-doped, Fig. [3](#page-5-0) (c_2 – c_4) for [3](#page-5-0)% Ce-doped and Fig. 3 (d_2-d_4) for 5% Ce-doped SnO₂ nanosheets. The Sn, Ce, and O contents were uniformly distributed in doped $SnO₂$ nanosheets.

The molar ratio of $[Sn^{2+}]$ and $[OH^-]$, pH of the growth solution, and the presence of directing agents like HMTA significantly affect the shape, size, and aspect ratio of the $SnO₂$ nanomaterials $[44]$ $[44]$. NH₄OH present in the reaction mixture provides a sufficient amount of HO⁻ ions which control the nucleation and crystal growth process which then controls the shape, size, and morphology of the $SnO₂$ nanostructures. The presence of the HMTA further adds to the concentration of HO– ions by hydrolysis (Eqs. [2](#page-3-1) and [3](#page-3-2)).

$$
(CH2)6N4 + 6H2O \rightarrow 6 HCHO + 4NH3
$$
 (2)

$$
NH_4\text{OH} \rightarrow \text{NH}_4^+ + \text{HO}^- \tag{3}
$$

During the hydrothermal growth at 160 °C for 8 h, initially, there is the formation of diferent hydroxylated species like

Table 1 Various XRD parameters for pure $SnO₂$ and different Ce-doped $SnO₂$ nanosheets

Fig. 2 FESEM images for (**a**–**d**) pure SnO2, (**e**–**h**) 1% Ce-doped SnO2, (**i**–**l**) 3% Ce-doped SnO2, and (**m**–**p**) 5% Ce-doped SnO2 nanosheets

 $Sn(OH)₃⁻, Sn(OH)₄²$, and $Sn(OH)₂$ in alkaline medium with pH 11 (Eqs. [4](#page-4-1)[–6\)](#page-4-2) [[45](#page-11-14), [46\]](#page-11-15).

$$
Sn^{2+} + 3HO^- \rightarrow Sn(OH)_3^-
$$
 (4)

$$
\text{Sn}^{2+} + 4\text{HO}^- \rightarrow \text{Sn}(\text{OH})_4^{2-} \tag{5}
$$

$$
Sn^{2+} + 2HO^- \rightarrow Sn(OH)_2 \tag{6}
$$

During the nucleation, these hydroxides are converted to SnO at high growth temperature and pressure conditions in the autoclave. High-temperature annealing in the presence of air at 450 °C for 3 h oxidizes SnO to $SnO₂$ nanoparticles which aggregate together to result in the formation of layered nanosheets like morphologies (Eqs. [7–](#page-4-3)[9\)](#page-4-4) [\[39](#page-11-8)].

$$
\text{Sn}(\text{OH})_3^- \rightarrow \text{SnO} + \text{HO}^- + \text{H}_2\text{O} \tag{7}
$$

$$
Sn(OH)42- \to SnO + 2 HO- + H2O
$$
 (8)

$$
2SnO + O_2 \rightarrow 2SnO_2 \tag{9}
$$

The presence of $CeCl₃·7H₂O$, during the hydrothermal synthesis of Ce-doped $SnO₂$ nanosheets, results in the formation of Ce(OH)₃ and Ce(OH)₄ in strong alkaline medium (Eqs. [10,](#page-4-5) [11\)](#page-4-6). Due to electrostatic interactions, Ce(OH)₃ and Ce(OH)₄ are mixed easily with $Sn(OH)_3^-, Sn(OH)_4^2$, and $Sn(OH)_2$ in the alkaline medium during the nucleation and growth process. High-temperature annealing in the presence of air oxidizes $Ce(OH)₃$ and $Ce(OH)₄$ hydroxides into $CeO₂$ (Eqs. [12,](#page-5-1)[13\)](#page-5-2).

$$
Ce^{3+} + 3HO^- \rightarrow Ce(OH)_3
$$
 (10)

$$
Ce^{3+} + 4HO^- \rightarrow Ce(OH)_4^-
$$
 (11)

Fig. 3 EDS selected area, EDS spectra, and FESEM-elemental mapping images for (a, a_3) pure SnO₂, (b, b_3) 1% Ce-doped SnO₂, (c, c_3) 3% Cedoped SnO₂, and $(\mathbf{d}, \mathbf{d}_3)$ 5% Ce-doped SnO₂ nanosheets

$$
2 \text{Ce(OH)}_3 + \frac{1}{2}\text{O}_2 \rightarrow 2 \text{ CeO}_2 + 3\text{H}_2\text{O}
$$
 (12)

$$
2 \text{Ce(OH)}_4^- + \frac{1}{2}\text{O}_2 \rightarrow 2 \text{ CeO}_2 + 2 \text{ HO}^- + 3 \text{ H}_2\text{O}
$$
 (13)

Figure [4a](#page-6-0) represents the FTIR spectra of as-synthesized pure $SnO₂$ and Ce-doped $SnO₂$ nanosheets. The fingerprint region of pure as well as Ce-doped $SnO₂$ nanosheets showed vibrational peaks in the region $604–614$ cm⁻¹ which may be assigned to the vibration of constituent ionic species in the crystal lattices and are due to the M − O stretching vibra-tions [\[47\]](#page-11-16). A large intensity band near 3430 cm⁻¹ may be attributed to the stretching vibrational mode of O–H bonds of the H₂O molecules adsorbed on the surface of pure $SnO₂$ and Ce-doped $SnO₂$ nanosheets during sample preparation.

Figure [4b](#page-6-0) shows the Raman spectra of the pure $SnO₂$ and Ce-doped $SnO₂$ nanosheets. The rutile phase crystal structure of $SnO₂$ consists of two $SnO₂$ formula units per unit cell. In a single unit cell, each Sn atom is surrounded by six O-atoms whereas each O-atom is surrounded by three Sn-atoms. Symmetry point group and space group of the rutile phase of $SnO₂$ are D_{4h} and p42/mnm, respectively, which have 15 optical phonons with symmetry symbols $A_{1g} + A_{2g} + A_{2u} + B_{1g} + B_{2g} + 2B_{1u} + E_g + 3E_u$. Among these, only the A_{1g} , B_{1g} , B_{2g} , and E_g optical phonons are Raman active, and the remaining are Raman inactive [\[48](#page-11-17)]. The Raman spectra for the pure, as well as Ce-doped $SnO₂$ nanosheets, show three fundamental Raman-active peaks situated at 464, 630, and 766 cm⁻¹.

The peak at 464 is assigned to the vibration of oxygen and corresponds to E_g Raman active mode. Peaks centered at 630 and 766 cm⁻¹ are assigned to the expansion and contraction vibrational modes of Sn–O bonds and correspond to the A_{1g} and B_{2g} optical Raman active modes, respectively [\[49](#page-11-18), [50\]](#page-11-19). Since the main Raman active peak corresponding to the F₂g mode of CeO₂ cubic structure is ~463 cm⁻¹ which is close to the E_g Raman active mode of the SnO₂, no separate Raman active peak for $CeO₂$ in Ce-doped SnO₂ nanosheets was observed.

3.2 2‑Nitroaniline chemical sensor applications of pure SnO₂ and Ce-doped SnO₂ nanosheets

Cyclic voltammograms (CV) were recorded and compared to analyze the electro-catalytic performance of bare GCE and modified GCE by pure $SnO₂$ and Ce-doped $SnO₂$ nanosheets/Nafon towards 10 μM 2-nitroaniline within the potential range $+0.5$ to $+1.5$ V at a scan rate of 50 mV/s in PBS at 7.0 pH vs. Ag/AgCl (saturated KCl) (Fig. [5a](#page-6-1)).

Fig. 4 (a) FTIR and (b) Raman spectra of pure $SnO₂$ and Ce-doped $SnO₂$ nanosheets

As compared to bare GCE, the pure $SnO₂$ and Ce-doped $SnO₂$ nanosheets/Nafion modified GCE showed better electrocatalytic performances as shown by distinct anodic peaks in diferent voltammograms. The well-shaped reversible 2-nitroaniline oxidation peak appeared on bare as well as different modified GCEs. For bare GCE, pure $SnO₂/GCE$, 1% Ce-doped SnO₂/GCE, 3% Ce-doped SnO₂/GCE, and 5% Ce-doped $SnO₂/GCE$ oxidation peak currents of 0.6165, 1.646, 1.704, 3.057, and 2.779 μA were observed at potentials 1.291, 1.496, 1.335, 1.316, and 1.336 V, respectively for 10.0 µM 2-nitroaniline at 50 mV/s in PBS. The Ce-doping into the crystal lattice of the $SnO₂$ nanosheets affects the microstructures, optical band gap energies, electrochemical, and redox properties of the Ce-doped $SnO₂$ nanosheets. Further, the unique two-dimensional sheet-like morphology, comprising very small nanoparticles, of the synthesized Ce-doped $SnO₂$ nanomaterials provides a large surface to

Fig. 5 (**a**) CV representing the electro-catalytic performance of bare GCE and modified GCE/Nafion electrodes based on pure $SnO₂$ and Ce-doped $SnO₂$ nanosheets towards 10 μ M 2-nitroaniline within the potential range $+0.5$ to $+1.5$ V and at a scan rate of 50 mV/s in PBS with pH 7.0 and (**b**) CV for 3% Ce-doped $SnO₂$ nanosheets in the absence and presence of 10 μM 2-nitroaniline

volume ratio for the greater extent of adsorption of analyte species onto the surface.

Among all the modifed GCEs, the electrode comprising 3% Ce-doped SnO₂ nanosheets exhibited maximum electrocatalytic activity for 10 μ M 2-nitroaniline. This can be attributed to the presence of comb-like nanoparticles in case of 3% Ce-doped SnO₂ nanosheets. Therefore, in the subsequent electrochemical analysis, only 3% Ce-doped $SnO₂$ nanosheet/Nafion-modified GCE was studied under different conditions. This 3% Ce-doped SnO₂ nanosheet/ Nafon-modifed GCE showed no distinct anodic peaks in the absence of 2-nitroaniline, whereas, in the presence of 10 μM 2-nitroaniline, amplifed anodic current was observed at the modifed active GCE at a scan rate of 50 mV/s in PBS (Fig. [5](#page-6-1)b). Thus, as fabricated 3% Ce-doped SnO₂ nanosheet/

Nafon-modifed GCE can potentially be used for electrocatalytic oxidation/reduction and hence as electrochemical sensor towards 2-nitroaniline even at very low concentration of 10 μM.

The electro-catalytic sensing characterizations of the 3% Ce-doped SnO₂ nanosheet/Nafion-modified GCE were analyzed as a function of scan rate through CV within the potential range $+0.5$ to $+1.5$ V. Figure [6a](#page-7-0) represents the effect of change of scan rates from 10 to 60 mV/s on the electrochemical behavior toward 5.0 μM 2-nitroaniline. With the increase in the scan rate, a continuous increase in anodic peak current was observed. A high determinant coefficient (R^2) value of 0.9643 was observed from a linearized variation of anodic peak currents with scan rates which further indicate the diffusion-controlled electrochemical oxidation

of 2-nitroaniline [[51](#page-11-20)] (Fig. [6b](#page-7-0)). The corresponding calibration plot for the anodic current responses for $5.0 \mu M$ 2-nitroaniline vs. (scan rate) $1/2$ with the determinant coefficient $R^2 = 0.99344$ (Fig. [6](#page-7-0)c) and the plot representing the Ep vs. log scan rate with the determinant coefficient R^2 = 0.99801 (Fig. [6](#page-7-0)d) for 3% Ce-doped SnO₂ nanosheets/Nafion modified GCEs were also linear confirming the pure diffusion-controlled process.

To further investigate the electrochemical properties of the pure $SnO₂$ and Ce-doped $SnO₂$ nanosheets/ Nafon-modifed GCE, a comparative study of EIS was also carried out. The EIS data was collected in 1.0 mM $K_3Fe(CN)_{6}/K_4Fe(CN)_{6}/0.1$ M KCl solution at +0.6 V. The AC signal amplitude was 10 mV and the frequency range was 0.1 Hz to 100 kHz. The corresponding Nyquist plots drawn between real variables (Z_{Real}) and the negative value

Fig. 6 (**a**) The efect of diferent scan rates on the CV response for 5.0 μM 2-nitroaniline, (**b**) linear plot of scan rate vs. anodic current response, (c) linear plot for (scan rate)^{1/2} vs. anodic current response,

and (**d**) log scan rate vs. Ep using 3% Ce-doped SnO₂ nanosheet/ Nafon-modifed GCE

Fig. 7 Nyquist plots between Z_{Real} and $-Z_{\text{Imag}}$ for different modified electrodes

of the imaginary impedance variable (Z_{Image}) are shown in Fig. [7.](#page-8-0) The semicircle shape of the Nyquist plots for the modifed electrodes represents the electron transfer resistance, which resulted due to the electron transfer by the redox probe $[Fe(CN)_6]^{3-4}$. The diameter of the semicircle plot is correlated to the charge-transfer resistance (R_{ct}) which controls the rate of electron transfer kinetics of the $[Fe(CN)₆]^{3-/4-}$ probe at the interface of the electrode. As revealed, the lowest semicircle's diameter was obtained for the 3% Ce-doped $SnO₂$ -modified GCE, indicating a decreased R_{ct} value for this active electrode than other modified working electrodes including pure $SnO₂/GCE$, 1% Ce-doped SnO_2/GCE , or 5% Ce-doped SnO_2/GCE . This again confrmed an enhanced electron-transfer capability for the 3% Ce-doped $SnO₂$ modified GCE which is advantageous for the electrochemical sensing performance toward 2-nitroaniline, consistent with the above CV results of Fig. [5a](#page-6-1).

The amperometric technique was also explored under constant stirring conditions in PBS at an applied potential of 1.3 V against Ag/AgCl (saturated KCl). Figure [8a](#page-8-1) displays the current real-time response of 3% Ce-doped $SnO₂$ nanosheets/Nafion-modified GCE in the presence of different 2-nitroaniline concentrations ranging from 0.5 to 38.0 µM. In the absence of 2-nitroaniline, a stable background response was observed by the modifed GCE. Amperometric current responses increased as soon as 2-nitroaniline with diferent concentrations was added to the PBS.

To determine the sensitivity, LOD, and LDR, a calibration plot of the amperometric response current vs. [2-nitroaniline] was plotted (Fig. [8b](#page-8-1)). The 3% Ce-doped $SnO₂$ nanosheet-modified/Nafion GCE showed a linear

Fig. 8 (**a**) The amperometric response curve of 3% Ce-doped $SnO₂$ nanosheets/Nafion-modified GCE in the presence of different 2-nitroaniline concentrations ranging from 0.5 to 38.0 µM at an applied potential of 1.3 V in PBS and (**b**) corresponding calibration plot of [2-nitroaniline] vs. amperometric current responses

dynamic range of 0.5–20.3 µM. The corresponding linear regression equation was Ip $(\mu A) = 0.0709$ [2-nitroaniline (μM)] + 0.1385 with R^2 = 0.99325. The LOD of the modified sensor was found to be 6.3 ± 0.1 nM at the signal-to-noise ratio $S/N = 3$. The sensitivity of the sensor is estimated to be 0.9986 μA μM⁻¹ cm⁻².

The effect of some interfering electro-active chemical species on the selectivity of the 3% Ce-doped SnO₂ nanosheet/Nafion-modified GCE towards 2-nitroaniline was also examined. Figure [9](#page-9-0) represents the variations of the amperometric current responses of modifed GCE during the successive addition of 2-nitroaniline $(5.0 \mu M)$, 4-nitroaniline (50 μM), 4-nitrophenol (50 μM), NaCl (50 μM), and CaCl₂ (50 μM) after certain time intervals and at an applied potential of 1.3 V. Signifcant increase in current response upon the addition of even 5.0 μ M 2-nitroaniline as compared to

Fig. 9 Amperometric plot representing the interference behavior of diferent species for the electrochemical sensing of 2-nitroaniline using 3% Ce-doped SnO₂ nanosheets/Nafion modified GCE at 1.3 V in PBS

negligible current changes on the additions of 4-nitroaniline, 4-nitrophenol, NaCl, and $CaCl₂$, is undoubtedly confirming the excellent selectivity of the fabricated 3% Ce-doped SnO₂ nanosheet based sensor for 2-nitroaniline.

3.3 Sensing mechanism

Figure [10](#page-9-1) illustrates the proposed mechanism for the better electrochemical sensing of the 2-nitroaniline by Cedoped $SnO₂$ nanosheets. The presence of $Ce⁴⁺$ ions into the $SnO₂$ crystal lattice promotes the charge transfer from

Fig. 10 Proposed sensing mechanism for 2-nitroaniline by Ce-doped $SnO₂$ nanosheets

the conduction band of $SnO₂$ nanosheets to the surfaceadsorbed O_2 molecules which were present in the PBS. Additionally, Ce-doping also increases the density of oxygen vacancies and roughness on the surface of the nanosheets required to generate sufficiently more adsorption sites for O_2 molecules. Conduction band electrons reduce the adsorbed O_2 molecules to different oxygenated species like O^- , O^{2-} , and O^-_2 [[30\]](#page-11-0). Further, the rates of intermolecular charge transfer process and adsorption of the 2-nitroaniline are facilitated on the surface of the Cedoped $SnO₂$ nanosheets due to the simultaneous presence of electron-rich amino $(-NH₂)$ group and an electron-deficient nitro $(-NO₂)$ group [\[52,](#page-11-21) [53](#page-11-22)]. Different reactive oxygenated species on the surface of the nanosheets initially reduce 2-nitroaniline into 2-hydroxylamine aniline. The reduction process is followed by a fast reversible oxidation process which oxidizes the 2-hydroxylamine aniline into 2-nitrosoaniline. The electrons released during the oxidation process are transferred back to the conduction band of Ce-doped SnO₂ nanosheets which increases the current response and electrical conductivity [[54\]](#page-11-23).

Furthermore, there is signifcant potential for improving the sensitivity of this chemical sensor by applying external photons, which could result in improving its behavior toward 2-nitroaniline [\[55–](#page-11-24)[57](#page-11-25)].

4 Conclusions

Herein, a novel Ce-doped $SnO₂$ nanosheet/Nafion-modified GCE has been prepared to quantify the 2-nitroaniline in PBS at pH 7.0. Among the various modified electrodes, the 3% Ce-doped $SnO₂$ nanosheets exhibited better electrochemical sensing behavior as compared to others. As-fabricated 3% Ce-doped $SnO₂$ nanosheets/ Nafion-modified GCE showed a linear dynamic range of 0.5–20.3 µM, LOD 6.3 ± 0.1 nM at the signal-to-noise ratio $S/N = 3$ and a remarkable sensitivity of 0.9986 μA μ M⁻¹ cm⁻². High sensitivity, very low detection limit, wide dynamic range, quick response time, and excellent selectivity of the fabricated sensor may be attributed to high surface area and easy diffusion of the 2-nitroaniline into the sensor matrix. Hence, the hydrothermal synthesis of Ce-doped $SnO₂$ nanosheets has great sensor potential and can be explored for future sensing applications towards toxic and harmful chemicals.

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Declarations

Conflict of interest The authors declare no competing interests.

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