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A waterborne polyurethane–based leather fnishing agent with excellent room temperature self‑healing properties and wear‑resistance

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Abstract

Water-based polymers have been widely used in coating materials, but their durability and service life are often greatly reduced due to macroscopic damage or cracking caused by external forces. In this paper, poly(propylene glycol) terminated with ortho-amiomethylphenyl diboronic acid (AMPBA-PPG) and cross-linked polymer containing boroxine (CLP-boroxine) were synthesized by nucleophilic substitution. Then the CLP-boroxine was introduced into waterborne polyurethane (WPU) to prepare AMPBA-PPG/WPU composite emulsion for leather fnishing and CLP-boroxine/WPU composite flm. The selfhealing characterization results showed that the self-healing efficiency of the 15 wt% CLP-boroxine/WPU film was as high as 93.6% after only 4 h of water stimulation at room temperature. Compared with neat WPU fnished leather, the leather samples fnished with AMPBA-PPG/WPU exhibited more excellent wear-resistance. More importantly, the leather coating fnished with AMPBA-PPG/WPU still showed nice wear-resistance even after wear and self-healing treatment. These results are attributed to that the hydrogen bonds between AMPBA-PPG and WPU, WPU, and WPU, and the dynamic reversible covalent bonds in AMPBA-PPG can improve/increase the self-healing efficiency of the composite coating. In addition, the boron hydroxyl groups in AMPBA-PPG could also react with the hydroxyl groups on the leather surface, thereby further improving the wear-resistance of the leather coating.

Keywords Waterborne polyurethane · Leather fnishing · Self-healing · Wear-resistance

1 Introduction

Waterborne polyurethane (WPU) is a new type of polyurethane system emerging to meet the increasing demand for environment friendly materials all over the world [[1\]](#page-9-0). WPU is widely used as the fnishing materials for leather, fabric, wood, rubber, and plastic engineering due to its environmental protection, strong adhesion, and flm

Highlights

synthesized.

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forming properties [[2–](#page-9-1)[4\]](#page-9-2). However, due to its poor wearresistance, the durability and service life of WPU coating are dramatically reduced [[5–](#page-9-3)[7](#page-10-0)]. Therefore, it is an urgent requirement to develop a WPU coating with improved wearresistance not only to enhance its function as a protective material but also renders its fnishing products a higher application value.

Introducing materials with self-healing function into WPU system can endow WPU with self-healing property [[8](#page-10-1)–[10](#page-10-2)]. So, its service life can be effectively improved and its coated products can be endowed with higher application value $[11-16]$ $[11-16]$ $[11-16]$. Reversible covalent bond self-healing system is a healing process using reversible chemical reactions of polymer materials [[17\]](#page-10-5), including disulfide bond reaction [[18](#page-10-6)], Diels-Alder reaction [[19](#page-10-7)], imine bond reaction [[20](#page-10-8)], and acylhydrazone bond reaction $[21]$. The bond energy of reversible covalent bonds is second only to that of covalent bonds, so materials based on reversible covalent bond not only have excellent selfhealing properties, but also have superior mechanical

[•] A cross-linked polymer containing boroxine were

[•] The coating fnished with AMPBA-PPG/WPU shows nice wear-resistance.

[•] The CLP-boroxine/WPU flms can efectively self-healing at room temperature.

properties and wear-resistance $[22-25]$ $[22-25]$. Deng et al. $[26]$ $[26]$ prepared a self-healing hydrogel with acylhydrazone bond. The fracture of this hydrogel can be self-healing after 48 h under acidic conditions, and its self-healing efficiency is 90%. Bergman et al. [[27\]](#page-10-13) used heat to make the imine group react with furan group on maleamide for DA reaction. Then, the material underwent reversed DA reaction by cooling down to repair the damage. These results indicate that most self-healing materials based on reversible covalent bonds usually require external stimuli such as pH, heat, and light.

The unique molecular structure of boron-oxygen compounds makes polymer materials have higher crosslinking density, which is very suitable for manufacturing polymer materials with high mechanical strength [[28–](#page-10-14)[30\]](#page-10-15). Boric acid/boronate ester balance is easily changed by temperature or the addition/removal of Lewis base or water, thus providing a self-healing mechanism [[31,](#page-10-16) [32\]](#page-10-17). Lai et al. [[33](#page-10-18)] use dynamic boronate ester bonds to crosslink polydimethylsiloxane chain into the 3D network. The prepared samples could achieve most of the mechanical strength after being heated at 70 °C after wetting and healing for 12 h. Boronate ester is formed by boric acid and diol under alkaline conditions. They are extremely unstable under acidic conditions [\[34,](#page-10-19) [35](#page-10-20)]. Compared with the boric acid center, the presence of an amino group in the ortho position can achieve B-N interactions, thereby reducing the alkalinity of amine and the acidity of boric acid, thus contributing to form boronate ester in a neutral pH value [[28](#page-10-14), [36,](#page-10-21) [37\]](#page-10-22). In addition, boronate ester is a thermally stable dynamic covalent bond. Compared with boronate ester, B–N bonds can reduce the acidity of boric acid; consequently, nitrogen coordinate boronate ester can be broken and re-built reversibly at a relatively low temperature [[38,](#page-10-23) [39\]](#page-10-24). Therefore, the nitrogen-coordinate boronate ester can make up for the shortage of boronate ester. It can achieve positive self-healing ability of material under the stimulation of water at room temperature.

In this paper, poly(propylene glycol) terminated with ortho-amiomethylphenyl diboronic acid and cross-linked polymer containing boroxine (denoted as AMPBA-PPG and CLP-boroxine) were synthesized by nucleophilic substitution. And then, the CLP-boroxine was introduced into waterborne polyurethane (WPU) and used for the constructing of self-healing and wear-resistant leather coating by utilizing the high reversibility, the crosslinking of boroxines, and rich hydrogen bonding networks. Upon the water stimulation, the CLP-boroxine/ WPU composite films could achieve rapid self-healing at room temperature. Meanwhile, the wear-resistance of the CLP-boroxine/WPU composite leather coating had been significantly improved due to the chemical interaction between boron hydroxyl of CLP-boroxine and hydroxyl groups on the surface of the collagen fiber. What is more worthy of attention was that the composite coating still showed well wear-resistance after the treatment of first wear and then self-healing. These results indicate that the introduction of borate-based dynamic covalent bonds in the coating system can provide a new strategy for constructing self-healing and durable wear-resistant coating on the surface of polyhydroxy-containing substrates.

2 Experimental

2.1 Materials

Poly(propylene glycol) bis(2-aminopropyl ether) $(NH_2-PPG-NH_2, Mn \approx 2000)$ and 2-(bromomethyl) phenylboronic acid (Br-PBA) were purchased from Aladdin Industrial Corporation. Isophorone diisocyanate (IPDI), dibutyltin dilaurate (DBTDL), and 2, 2-dihydroxybutyric acid (DMBA) were provided by Shanghai Maclin Biochemical Co., Ltd. and Guangdong Guanghua Sci-Tech Co., Ltd., respectively. Polytetramethylene glycol (PTMG1000) and dichloromethane (CH₂Cl₂, \geq 99.5%) were procured from Jining Huakai Resin Co., Ltd. Ethanol and triethylamine $(TEA, \geq 99.5\%)$ were obtained from Tianjin Fuyu Fine Chemical Co., Ltd. All chemicals were analytical grades and used directly without further processing.

2.2 Preparation of WPU

Firstly, PTMG1000 and IPDI were mixed in a three-neck round-bottom fash (500 mL) and stirred using a mechanical agitator for 30 min. Then, three drops of DBTDL were added to the mixture and continued to stir for 30 min. Afterwards, the water bath was heated to 75 °C for 3 h. Subsequently, DMBA was slowly added to reaction solution as chain extender. The reaction continued for 2 h before being cooled to 50 °C. Finally, TEA (4 mL) was added by pipette and reacted for 20 min, and then deionized water was added and emulsifed for 1 h.

2.3 Synthesis of AMPBA‑PPG and CLP‑boroxine

Br-PBA (1.29 g) and NH₂-PPG-NH₂ (6 g) were first successively added to a three-necked flask containing 60 mL of ethanol and reacted at 60 °C for 6 h under stirring conditions. Next, the above mixture was placed in an ice water bath, and $NabH_4$ (0.37 g) and a drop of acetic acid glacial were slowly added, and the reaction was continued for 2 h. After the reaction was completed,

Scheme 1 Synthesis route of AMPBA-PPG and CLP-boroxine

ethanol was removed by distillation to obtain AMPBA-PPG. Finally, the AMPBA-PPG gradually dehydrated and cross-linked into CLP-boroxine after the drying process was carried out. The synthetic route of AMPBA-PPG and CLP-boroxine is showed in Scheme [1.](#page-2-0)

2.4 Preparation of CLP‑boroxine/WPU composite flms

Under magnetic stirring, ethanol solution of CLPboroxine was added to WPU drop by drop (about a drop in 10 s), and the AMPBA-PPG/WPU composite emulsion was formed by the hydrolysis of CLP-boroxine. CLP-boroxine/WPU composite film was obtained by transferring AMPBA-PPG/WPU composite emulsion into a polytetrafluoroethylene film-forming plate to evaporate the solvent at room temperature for 12 h. During the film formation process, as the solvent evaporates, the hydrogen bond interactions between the secondary amine groups of AMPBA-PPG and the carbonyl groups of the WPU, and the transition from AMPBA-PPG to CLPboroxine structure promote the formation of multiple network structures inside the composite film (As shown in Scheme [2](#page-2-1)).

2.5 Application in leather fnishing

The WPU and AMPBA-PPG/WPU composite emulsion were employed for leather finishing. Before finishing the leather, the amount of spray required for the leather samples area were calculated according to QB/T 2002.2–2018. Then the specified amount of composite emulsion was mixed with water at a mass ratio of 1:1. After mixing uniformly, the air pressure and the discharge amount of the spray gun were adjusted to be sprayed, and each leather sample was sprayed back and forth repeatedly.

2.6 Characterization

The chemical structure of samples was analyzed by Fourier transform infrared spectroscopy (FT-IR) using a Bruker VERTEX-80 spectrometer in the range of 400–4000 cm−1. Nuclear magnetic resonance spectrometer $({}^{1}H NMR)$ of AMPBA-PPG was measured on a 400-MHz NMR spectrometer at room temperature. In order to analyze the thermal stability of CLPboroxine, the thermal properties of CLP-boroxine were tested by differential scanning calorimetry (DSC) using

a NETZSCH DSC200F3 system. In details, the samples were heated from − 80 to 50 °C at a rate of 10 °C/min under a nitrogen flow rate of 50 mL/min. The fracture morphologies of the samples were tested by Hitachi S-4800 field emission scanning electron microscope (SEM). Briefly, the film cross-section was sprayed with gold by JEOL JFC-1600, and the SEM test conditions were an acceleration voltage of 3.0 kV and a current of 10 μA.

The mechanical properties of the CLP-boroxine/WPU composite flms and leather samples were determined using an AI-3000 tensile tester with a stretching rate of 100 mm/ min. The CLP-boroxine/WPU composite flms were cut into dumbbell shapes with 80 mm in full length, 30 mm in effective length, and 5 mm in intermediate width. Each sample was tested three times.

All the wear experiments in this article were carried out on the MMUD-1B friction and wear testing machine. In the experiment of wear, the end friction pair was carried out by dry friction at room temperature without lubricating oil, and the standard wool felt was fxed on the friction head. The parameters of friction and wear test are as follows: test force is 120 N, rotation speed is 50 r/min, and the test time is 60 min. The detailed information of wear index calculation is available in supplementary information fle.

The self-healing ability of CLP-boroxine/WPU composite flm was tested at room temperature. The CLPboroxine/WPU composite flm was frst cut into two separate flms with a bistoury, and the ends of the damaged two sections were immersed in water for 2 min. Then the two sections were butted together at room temperature for 4 h for self-healing. The self-healing efficiency of the

CLP-boroxine/WPU composite flm was evaluated by its tensile strength before and after self-healing.

3 Results and discussion

3.1 Characterization of CLP‑boroxine

Figure [1](#page-3-0) showed the FT-IR results of Br-PBA, CLP-boroxine, and $NH₂-PPG-NH₂$. For Br-PBA, the characteristic peak at 3387 cm−1 belongs to the stretching vibrations of B-OH and the peaks at 1596 and 1349 cm−1 are attributed to Ar $(C=C)$ and B-O bonds, respectively [[40](#page-10-25), [41\]](#page-10-26). With regard to the polymer of NH₂-PPG-NH₂, the peak at 1598 cm⁻¹ is attributed to the bending vibration of $R-NH₂$, while the typical peak at 1105 cm^{-1} is assigned to the formation of C–O–C bonds [\[42](#page-10-27)]. In the spectrum of CLP-boroxine, both the peak at 3387 cm−1 of B–OH stretching vibration in Br–PBA and the peak at 1598 cm−1 of N–H deformation vibration in NH_2 –PPG–NH₂ are disappeared. However, a new peak appears at 750 cm−1, which is attributed to the boroxine [[43](#page-10-28)]. The results confrm that the Br–PBA and $NH₂$ –PPG–NH₂ have been reacted successfully, and the boroxine structure has formed in CLP-boroxine.

¹H NMR was used to further investigate the chemical structure of AMPBA-PPG. As presented in Fig. [2](#page-3-1), the chemical shifts of hydrogen atoms in AMPBA-PPG molecule are in the range of 7.51–1.18 ppm, including the following: δ 7.51 ppm (Ha, 2H, Ar–H), δ 7.22–7.15 ppm (Hb, 6H, Ar–H), δ 4.89 ppm (Hc, 4H, B-OH), δ 4.07 ppm (Hd, 4H, Ar-CH₂-), δ 3.88–3.84 ppm (He, 2H, -NH-), δ 3.75–3.15 ppm (Hf, Hg, 106H, -CH-, -CH₂-), and δ 1.15 ppm (Hh, 106H, $-CH_3$). ¹H NMR spectrum further

Fig. 1 FT-IR spectra of Br-PBA, NH₂-PPG-NH₂, and CLP-boroxine, respectively

Fig. 2 ¹H NMR spectrum of AMPBA-PPG in CD_3OD

Fig. 3 Inclined **a** and inverted **b** views of the CLP-boroxine in vial, and DSC curves **c** of CLP-boroxine at − 80 °C~50 °C

demonstrates that the AMPBA-PPG has been synthesized successfully.

CLP-boroxine was put into a small vial, and its appearance was shown in Fig. [3.](#page-4-0) As can be seen in Fig. [3](#page-4-0), the CLP-boroxine is a transparent yellowish color and shows good fuidity at room temperature. Figure [3](#page-4-0)c is the DSC curve of the CLP-boroxine. The DSC curve shows the glass transition temperature (T_o) of CLP-boroxine is -70 °C. Moreover, the two critical points indicate that it is in glassy state before -75 °C, in viscous flow state after -53 °C [[44,](#page-11-0) [45](#page-11-1)]. Therefore, CLP-boroxine exhibits a viscous fow state at room temperature. This result has an agreement with the observation as shown in Fig. [3b](#page-4-0). All these results indicate that CLP-boroxine is more easily incorporated into emulsion systems.

3.2 Properties of AMPBA‑PPG/WPU composite emulsion and CLP‑boroxine/WPU composite flms

The Zeta potential of AMPBA-PPG and WPU emulsion is characterized by nano particle surface potential analyzer for studying the stability of AMPBA-PPG/WPU composite emulsion. Table [1](#page-4-1) showed that the Zeta potentials of AMPBA-PPG and WPU emulsion are − 11.3 mV and − 34.9 mV at pH =7, respectively. This result indicates that there is a repulsive efect between them. This repulsion separates AMPBA-PPG from WPU, which is conducive to the stability of the emulsion [\[46](#page-11-2)]. Therefore, the addition of AMPBA-PPG to WPU emulsion can maintain the stability of its composite emulsion.

To further illustrate its storage stability, AMPBA-PPG is added into WPU emulsion and the resulted mixture is kept still at room temperature for 20 days to observe the status of the composite emulsion. The results show that the composite emulsion remains stable and shows even dispersion after 20 days at room temperature. Moreover, AMPBA-PPG/WPU composite emulsion (10 g) has been centrifuged at 3000 r/min for 30 min for the purpose of testing its centrifugal stability. The results show that the AMPBA-PPG/WPU composite emulsion has no obvious precipitation, gelation or stratifcation, etc. (As shown in Fig. [4\)](#page-4-2). This phenomenon indicates that the centrifugal stability of AMPBA-PPG/WPU composite emulsion is splendid [\[47,](#page-11-3) [48](#page-11-4)]. In summary, the AMPBA-PPG/WPU composite emulsion has excellent stability.

The FT-IR spectra were used to characterize the chemical structure of CLP-boroxine/WPU composite film (Fig. [5\)](#page-5-0). The FT-IR spectrum of CLP-boroxine/WPU composite film shows that the C=O stretching vibration peak in WPU changes from 1706 to 1745 cm−1, and the peak at 1589 cm−1 corresponding to the -NH group shifts to 1542 cm−1 in the CLP-boroxine. These are attributed to the formation of hydrogen bonds between the carbonyl groups of WPU and the secondary amine groups of CLPboroxine in CLP-boroxine/WPU composite films [[49,](#page-11-5) [50\]](#page-11-6). At the same time, the absorption peak of boroxine

Fig. 4 Visual appearance of AMPBA-PPG/WPU composite emulsion **a** without and **b** after centrifugation at 3000 r/min for 30 min

Fig. 5 FT-IR spectra of WPU, CLP-boroxine, and 10 wt% CLPboroxine/WPU, respectively

appears at 750 cm^{-1} , which indicates that CLP-boroxine/ WPU composite films are formed.

3.3 Mechanical properties of CLP‑boroxine/WPU composite flms

The tensile strength and elongation at break of CLPboroxine/WPU composite films with different CLPboroxine contents as can be seen in Fig. [6.](#page-5-1) It can be clearly seen in Fig. [6](#page-5-1)a that the tensile strength of CLPboroxine/WPU composite films decreases continuously with the addition of CLP-boroxine. This phenomenon that may be attributed to the CLP-boroxine is a flexible

material. The CLP-boroxine may reduce the strength of the films when added to WPU.

As for the elongation at break, Fig. [6](#page-5-1)b shows different tendency compared with tensile strength of CLPboroxine/WPU composite films. It can be seen that a suitable amount of CLP-boroxine can greatly increase the elongation at break of WPU film. Its elongation at break reaches the maximum value of 731.2% with 10 wt% CLP-boroxine, an increase of as much as 23.8% in comparison with that of the neat WPU film (590.5%). This may be attributed to two reasons. On the one hand, rich hydrogen bonds have been formed between the secondary amine groups of CLP-boroxine and the carbonyl groups of WPU, thus effectively enhancing the interaction between molecular chains. On the other hand, multiple network structures can be formed in the composite system through the production of boronate ester bonds between CLP-boroxine, thereby enhancing the strength of the molecular chains of the CLP-boroxine/WPU composite system.

In order to explain this phenomenon, the fracture surface morphologies of neat WPU and CLP-boroxine/ WPU composite films after tensile tests were studied by SEM, which provided immediate proof for the interfacial interactions between CLP-boroxine and WPU (Fig. S1). While with the further increase of the CLP-boroxine addition, the elongation at break of CLP-boroxine/WPU composite film decreases. This may be due to the fact that CLP-boroxine itself is an ultra-low strength material. Too much addition will not only reduce the strength of the CLP-boroxine/WPU composite film but also reduce the interaction between molecules [[30\]](#page-10-15).

Fig. 6 a Tensile strength and **b** elongation at break of CLP-boroxine/WPU composite flms with diferent CLP-boroxine contents

Fig. 7 SEM micrograph of fractured surface of CLP-boroxine/ WPU composite flms obtained with the diferent content of CLP-boroxine. **a** 0 wt%, **b** 5 wt%, (**c**) 10 wt%, **d** 15 wt%, and EDX mapping of **e–h** neat WPU and **i–m** 10 wt% CLP-boroxine/ WPU. Red, green blue, and yellow dots indicate signals of C, N, O, and B, respectively

3.4 Analysis of section morphology of CLP‑boroxine/WPU composite flms

In order to observe the internal structure of CLP-boroxine/ WPU composite flms, the flms with diferent CLP-boroxine contents were brittle fractured with liquid nitrogen (as shown in Fig. [7\)](#page-6-0). It can be seen that the morphology fractured surface of the neat WPU flm is relatively smooth. Cracks appear on the fractured surface of neat WPU, which is the result of liquid nitrogen freezing. As the amount of CLP-boroxine increases, there is no crack on the fractured surface of CLP-boroxine/ WPU composite flm, which indicates that CLP-boroxine has better low temperature resistance. Meanwhile, a large number of white substances appear on the fractured surface of CLP-boroxine/WPU composite flm. These white substances may be the aggregates of precipitated CLP-boroxine. This may be due to the fact that the condensation of water vapor on the fractured surface of composite flm leading to the transformation from CLP-boroxine to AMPBA-PPG and then precipitate from the matrix. As water vapor evaporates, the AMPBA-PPG accumulates and transforms CLP-boroxine compound gel on the fractured surface of composite flm [\[51\]](#page-11-7). The EXD mapping of 10 wt% CLP-boroxine/WPU (Fig. [7e](#page-6-0)–m) is further confirmed that CLP-boroxine is uniformly distributed in the CLP-boroxine/WPU composite flm.

3.5 Self‑healing ability of CLP‑boroxine/WPU composite flms

The CLP-boroxine/WPU composite films possess good self-healing property by virtue of the dynamic reversibility of boronate ester bond in the presence of water at room temperature. As can be seen in Fig. [8,](#page-6-1) the incision in the 15 wt% CLP-boroxine/WPU composite film has been completely healed under water stimulation at room temperature for 4 h. This primary proves that the CLP-boroxine/WPU composite film has self-healing property by direct visual method. Boronate ester group is responsive to water stimulation. So, the water resistance of CLP-boroxine/WPU composite films is discussed (see Fig. S2). Compared with the neat WPU film, the water resistance of the 5 wt% CLP-boroxine/WPU composite film is significantly improved. However, when the amount of CLP-boroxine added exceeds 5 wt%, the water absorption of CLP-boroxine/WPU composite films is increased. This phenomenon that may be attributed to the presence of excessive CLP-boroxine in the system will hinder the interaction between WPU molecules, resulting in changes in the densification of CLP-boroxine/WPU composite films.

In order to further evaluate the self-healing performance of the CLP-boroxine/WPU composite flms, tensile testing

Fig. 8 Comparison of 15 wt% CLP-boroxine/WPU composite flm before **a** and after **b** healing

Table 2 The self-healing efficiency of different content of CLPboroxine/WPU composite flms

Content of CLP- Tensile boroxine $(wt\%)$	strength (MPa)	Tensile strength after self-healing (MPa)	Self-healing efficiency $(\%)$
0	36.10	6.33	17.52
	34.27	23.18	67.63
10	30.00	24.68	82.27
15	18.15	16.99	93.59

is carried out on the self-healed samples with diferent CLP-boroxine content. Table [2](#page-6-2) indicates that neat WPU flm also has self-healing ability, which is mainly due to the large amount of intermolecular hydrogen bonding in WPU. But the healing efficiency of it is only 17.52% . Compared with neat WPU, the self-healing efficiency of CLP-boroxine/WPU increases with the introduction CLPboroxine. When the CLP-boroxine amount is 15%, the self-healing efficiency of CLP-boroxine/WPU composite flm is as high as 93.59%.

The excellent self-healing performance of CLP-boroxine/ WPU composite flm is mainly attributed to three aspects. On the one hand, CLP-boroxine itself has better molecular chain fexibility. The introduction of CLP-boroxine into WPU can improve the molecular motion ability. On the other hand, the hydrogen bond between WPU and CLP-boroxine and the dispersion of WPU itself can promote the rapid aggregation of molecular chains [[52\]](#page-11-8). Therefore, the speed of self-healing is accelerated. On the last aspect, multiple network structure of boronate ester bond is formed with the evaporation of water, thereby realizing self-healing function of CLP-boroxine/WPU. The healing mechanism of it is shown in Fig. [9.](#page-7-0)

Fig. 10 a Interfacial shear strength and **b** wear index of AMPBA-PPG/WPU composite emulsion coated leather samples with diferent AMPBA-PPG content

Fig. 11 SEM images of worn surface of unfnished leather **a–a′**, the leather samples coated by WPU **b–b′**, 15 wt% AMPBA-PPG/WPU (**c–c′**), and the leather sample coated by 15 wt% AMPBA-PPG/WPU after wear and healing treatment **d–d′**, respectively

In addition, in order to further study the healing properties of the composite films, the healing efficiency of 15 $wt\%$ CLP-boroxine/WPU composite flm with diferent healing times was also analyzed. As shown in Fig. S3, the healing efficiency of 15 wt% CLP-boroxine/WPU composite film gradually increases with the extension of healing time. However, when the self-healing time exceeded 4 h, there is no significant change in the self-healing efficiency. This phenomenon that may due to the system has reached a relative balance after 4 h of self-healing treatment.

3.6 Applied properties of AMPBA‑PPG/WPU composite emulsion for leather fnishing

In order to further study the application performance of AMPBA-PPG/WPU composite emulsion, the as-prepared composite emulsion has been applied to leather finishing. The AMPBA-PPG/WPU composite emulsion would form a coating film on the surface of leather after drying, which plays a variety of applications such as beautify leather surface, hide leather surface defects, and improve the performance of leather [[53](#page-11-9)]. The tribological properties of AMPBA-PPG/WPU composite coating were studied by wear-resistance test. Compared with neat WPU (582 mg), the wear index of AMPBA-PPG/WPU finished leather samples reduced by 11.8~15.9% (69~93 mg) with different contents of AMPBA-PPG (As shown in Fig. [10](#page-7-1)b). This result indicates that AMPBA-PPG/WPU coating can effectively improve the wear-resistance of leather.

In order to further evaluate the protection of the AMPBA-PPG/WPU coating, SEM was used to observe the surface of the coated leather after abrasion. As revealed in Fig. [11](#page-8-0), the leather samples coated with WPU and AMPBA-PPG/WPU composite coating agent have a relatively good protection function. The uncoated leather sample is severely worn and exposed to a large amount of collagen fibers, as shown in Fig. [11](#page-8-0)a–a′. However, WPU and 15 wt% AMPBA-PPG/WPU finished leather samples show only slight wear of the coating after friction, and no significant exposure of the collagen fibers are observed (Fig. $11b-c'$ $11b-c'$). It is worth noting that the leather samples finished with 15 wt% AMPBA-PPG/WPU still show excellent wear-resistance after self-healing (Fig. [11](#page-8-0)d–d′). The unique self-healing function of AMPBA-PPG/WPU coating can effectively extend its wear-resistant life. The excellent wear-resistance of AMPBA-PPG/WPU coating that may be attributed to the boron hydroxyl groups in AMPBA-PPG can react with the hydroxyl groups on the surface of leather collagen fiber, thus effectively improving the adhesive property of AMPBA-PPG/WPU coating (Fig. [10a](#page-7-1)).

4 Conclusions

In this paper, a novel of AMPBA-PPG/WPU composite coating with water-induced self-healing function was prepared by a blending method. With the increase of CLPboroxine content, the tensile strength of CLP-boroxine/ WPU composite films gradually decreases, while the elongation at break increases frst and then decreases. With the increase of CLP-boroxine content, the healing efficiency of the CLP-boroxine/WPU composite films increases gradually and reaches 93.6% after 4 h selfhealing at room temperature (15 wt% CLP-boroxine/ WPU). Compared with neat WPU fnished leather samples, the leather samples finished with AMPBA-PPG/WPU exhibit more excellent wear-resistance. This is mainly due to the fact that AMPBA-PPG can form chemical bond with collagen fbers on the surface of leather, thus efectively improving the wear-resistance and adhesion of the composite coating. And even more remarkable, the leather coating fnished with AMPBA-PPG/WPU still show nice wear-resistance even after wear and self-healing treatment. This work provides a new approach to the development of leather coatings with durable wear-resistance.

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Compliance with ethical standards

Conflict of interest The authors declare that they have no confict of interest.

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