A coupled model on fluid flow, heat transfer and solidification in continuous casting mold

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Abstract: Fluid flow, heat transfer and solidification of steel in the mold are so complex but crucial, determining the surface quality of the continuous casting slab. In the current study, a 2D numerical model was established by Fluent software to simulate the fluid flow, heat transfer and solidification of the steel in the mold. The VOF model and k- ε model were applied to simulate the flow field of the three phases (steel, slag and air), and solidification model was used to simulate the solidification process. The phenomena at the meniscus were also explored through interfacial tension between the liquid steel and slag as well as the mold oscillation. The model included a 20 mm thick mold to clarify the heat transfer and the temperature distribution of the mold. The simulation results show that the liquid steel flows as upper backflow and lower backflow in the mold, and that a small circulation forms at the meniscus. The liquid slag flows away from the corner at the meniscus or infiltrates into the gap between the mold and the shell with the mold oscillating at the negative strip stage or at the positive strip stage. The simulated pitch and the depth of oscillation marks approximate to the theoretical pitch and measured depth on the slab.

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In the continuous casting process of the slab, many complex phenomena^[1-3] occur in the mold, including</sup> fluid flow, heat transfer, steel solidification, interaction between steel and slag^[4], etc. The liquid steel flows into the mold cavity through the submerged entry nozzle (SEN) and then solidifies against the four walls of the water-cooling copper mold. The solidified shell is continuously withdrawn downward at the casting speed. The slag powder is continuously added on the top of the steel, then melts or sinters to form a slag bed, and liquid slag infiltrates into the gap between the shell and the mold^[5, 6]. Due to the interfacial tension, a curved meniscus^[7] forms between liquid steel and liquid slag, and solidifies with the heat transfer from the steel to the mold. Heat transfer and solidification of the liquid steel at the meniscus as well as the mold oscillation affect the formation of the oscillation marks^[8], which may cause some defects in the slab, including slag entrapment^[9], bubble and inclusion capture, element segregation, transverse cracks^[10], etc.

In the current study, a 2D model was established to simulate the multiphase fluid flow of steel-slag-

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air, heat transfer and solidification in the mold with mold oscillation. The model considered the interfacial tension between different phases and shell movement at the casting speed. The phase distribution and flow field in the mold were simulated, and phenomena near the meniscus were presented, including meniscus solidification, slag penetration and movement of the mold and the shell. The comparison of simulated pitch and depth of oscillation marks with theoretical pitch and measured depth on the slab surface was conducted to validate the accuracy of the simulation.

1 Model description

1.1 Governing equations

The model developed in the study couples a multiphase steel-slag-air flow approach with heat transfer, mold oscillation and solidification in the mold during continuous casting of steel slabs. The model is based on the commercial software ANSYS-Fluent 14.0, and solves the Navier-Stokes equations^[11] coupled with the volume of fluid (VOF) method for calculation of phase fractions and the continuum surface force (CFS) method for tracking the steel-slag-air interfaces. The k- ε viscous model and the standard wall functions are used to account for the turbulence in the system. The energy equation and Fourier equation are solved, and the solidification model is applied to consider the heat

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transfer and solidification in the mold.

In the VOF scheme, the effective density (ρ_{mix} , kg· m⁻³) and the effective viscosity (μ_{mix} , kg·m⁻¹·s⁻¹) are given by:

$$\rho_{\rm mix} = \alpha_{\rm p} \rho_{\rm p} + (1 - \alpha_{\rm p}) \rho_{\rm q} \tag{1}$$

$$\mu_{\rm mix} = \alpha_{\rm p} \mu_{\rm p} + (1 - \alpha_{\rm p}) \mu_{\rm q} \tag{2}$$

where α is the phase fraction and the subscripts p and q represent any two of the three phases present in the cell (steel, slag or air). Thus, the continuity equation for the volume fraction of any of the phases can be derived by:

$$\frac{\partial}{\partial t}(\alpha_{q}\rho_{q}) + \nabla(\alpha_{q}\rho_{q}\vec{v}) = \sum_{p=1}^{n} (\dot{m}_{pq} - \dot{m}_{qp})$$
(3)

where \vec{v} is the overall velocity vector (m·s⁻¹), \dot{m}_{pq} is the mass transfer from phase p to phase q (kg·s⁻¹), \dot{m}_{qp} is the mass transfer from phase q to phase p (kg·s⁻¹). The volume fraction for the primary phase is computed based on the following constraint:

$$\sum_{p=1}^{n} (\alpha_q) = 1 \tag{4}$$

Energy equation for fluid:

$$\frac{\partial}{\partial t}(\rho_{\min}E) + \nabla(\vec{v}(\rho_{\min}E + P)) = \nabla(K_{\text{eff}}\nabla T + (\tau_{\text{eff}}\cdot\vec{v}))$$
(5)

$$E = \frac{\sum_{q=1}^{n} (\alpha_q \rho_q E_q)}{\sum_{q=1}^{n} (\alpha_q \rho_q)}$$
(6)

where E is the enthalpy (J), K_{eff} is the effective thermal conductivity (W·m⁻¹·K⁻¹), E_q depends on the specific heat for the q phase.

Heat conduction for solid:

$$\frac{\partial}{\partial t}(\rho h) = \nabla (K_{\rm m} \nabla T) \tag{7}$$

where *h* is the sensible enthalpy (J) and K_m is the effective thermal conductivity (W·m⁻¹·K⁻¹).

1.2 Model domains and simulation conditions

The 2D domain consists of half SEN, half slab mold, copper plate and 0.5 m of slab length after the mold exit. The mesh of 2D mold is exhibited in Fig.1(a). Due to the simulation of an initial solidification and oscillation mark with the size of 0.2– 0.5 mm, the refined mesh locates at the interface of steel and slag, especially at the meniscus near the corner. The minimum size of the cell is 50 μ m, and the total cell number is 151,934. In order to investigate the formation of a curved meniscus and the slag infiltration between the shell and mold, the slag of 50 mm thickness is considered on the top of the steel in the mold, and the interfacial tension and contact angle between liquid steel and liquid slag is set as 1.3 N·m⁻¹ and 46°, respectively. The dynamic mesh is applied to the mold, and the mold oscillates with the frequency of 182 cycles per minute (cpm). The expressions for determining the mold position and mold velocity are as follows:

$$s = A \sin\left(\frac{2\pi f}{60}t\right) \tag{8}$$

$$v_{\rm m} = \frac{2\pi f}{60} A\cos\left(\frac{2\pi f}{60}t\right) \tag{9}$$

where s is mold position (m), A is oscillation amplitude of

the mold (m), *f* is oscillation frequency (cpm) and v_m is mold velocity (m·s⁻¹).

The model considers the interaction among three phases of steel, slag and air, and the material parameters are exhibited in Table 1. The solidification of the steel is considered, and the solidification coefficient is set as 5×10^8 . The thermal conductivity of steel has a relation to the steel temperature. After the steel solidification, the casting speed of $1.45 \text{ m}\cdot\text{min}^{-1}$ is applied to the solidified shell. In case of the application of the casting speed to the solidified slag, the model neglects the formation of the solidified slag. However, the viscosity and thermal conductivity of liquid slag are related to the temperature, and the liquid slag has the same property as the solidified slag when the temperature is lower than the break temperature of the slag^[12]. The model can therefore predict the movement and heat

Table 1: Material parameters

Parameter	Value
Density of liquid steel, $ ho_{ m steel}$ (kg·m ⁻³)	7,020
Viscosity of liquid steel, μ_{steel} (kg·m ⁻¹ ·s ⁻¹)	0.0063
Specific heat of steel, $C_{p, \text{ steel}} (J \cdot kg^{-1} \cdot K^{-1})$	750
Latent heat of steel, H_{steel} (J·kg ⁻¹)	270,000
Liquidus of steel, T_{L} (K)	1,805
Solidus of steel, T_{s} (K)	1,790
Density of liquid slag, $ ho_{ m slag}$ (kg·m ⁻³)	2,500
Specific heat of slag, $C_{p, slag}(J \cdot kg^{-1} \cdot K^{-1})$	830
Air density, ρ _{air} (kg⋅m⁻³)	1.225
Air viscosity, $\mu_{ m air}$ (kg·m ⁻¹ ·s ⁻¹)	1.8×10 ⁻⁵
Interfacial tension of slag-steel, γ (N·m ⁻¹)	1.3 ^[10]

transfer of the slag.

The simulation conditions are shown in Fig. 1(b). The initial temperatures of the fluids (steel, slag and air) and the mold are 1,805 K and 405 K, respectively. The thickness of the slag and air at the top of the mold are both 50 mm, and the thickness of the copper mold is 20 mm. The conditions of velocity inlet and outflow are applied to the nozzle top and the outlet at the model bottom, respectively. The temperature at the mold top is considered as room temperature (300 K), the heat transfer coefficients^[13] on the cold face (h_c) of the mold and the side face of the steel below the mold (h_s) are calculated from the cooling water flow of the plant. The detailed simulation parameters are shown in Table 2. Since the multiphase fluid flow, heat transfer and solidification with the mold oscillation are considered through the 2D model, and the refined mesh is used, the simulation is time-consuming. With the time step of 0.00001 s and six eight-core computers, the calculation time every day can be about 0.5 s, and the results are calculated for several weeks.

2 Results and discussion

The phase distribution, streamlines and velocity contour are



Fig. 1: Model mesh (a) and simulation conditions (b) of 2D model

Parameter	Value / Type
Mold width (mm)	1,300
Mold length (mm)	1,000
Simulation length (mm)	1,500
Slag thickness (mm)	50
Air thickness (mm)	50
Mold thickness (mm)	20
Mold oscillation mode	Sinusoidal
Oscillation frequency, $f(cpm)$	182
Oscillation stroke, $s_{\rm osc}$ (mm)	6
Casting temperature, T_{p} (K)	1,830
Casting speed, v_c (m·min ⁻¹)	1.45

 Table 2: Simulation parameters

illustrated in Fig. 2. The liquid steel out of the SEN flows as the upper backflow and lower backflow, and the velocity of the backflows are $0.4-0.8 \text{ m} \cdot \text{s}^{-1}$ and $0.5-1.1 \text{ m} \cdot \text{s}^{-1}$. Due to the upper backflow flowing to the nozzle, the steel level near the mold is



Fig. 2: Phase distribution, streamlines and velocity contour

higher than the one near the nozzle, and the thickness of the slag bed near the nozzle is also greater. At the center of the half mold, the liquid slag depresses the steel level, and due to the washing of the liquid steel with high speed, slag entrapment may occur. A small circulation flowing to the steel-slag interface exists at the meniscus, and the velocity of the fluid there is lower than $0.2 \text{ m} \cdot \text{s}^{-1}$. The circulation may be affected by natural convection, mold oscillation and pressure from the slag pool^[6].

In Fig. 3, the two curves represent the mold position $(m_{\rm disp})$ and velocity $(V_{\rm m})$, respectively, the horizontal line represents the casting speed $(V_{\rm c})$, and the positions of the points a–d in the curve are corresponding to the simulation results. The calculated temperatures of steel and mold at four oscillation speeds (a–d) of the mold are demonstrated. The negative stage $(t_{\rm n})$ as the gray part in the curve zone is between point b and c, which represents the beginning and end of the negative stage, respectively. In Fig. 3(a), with the mold oscillating upward, the liquid slag infiltrates into the gap between the shell and the mold, which may be caused by the reduced pressure of the slag pool at the meniscus ^[8]. With the cooling of the water in the mold, the meniscus begins to solidify, and the mushy zone between the temperatures of liquidus (1,805 K) and solidus (1,790 K) appears.

From point a to point b (positive strip stage), the mold oscillates upward (Fig. 3a) and then downward (Fig. 3b), the meniscus moves to the mold, and the speed of the liquid slag flowing into the gap decreases. From point b to point c (negative strip stage), the speed of the mold oscillating downward exceeds the casting speed, then the slag flows into the gap and the meniscus moves away from the mold (Fig. 3c), which may be caused by the growing pressure of the slag pool^[8]. Due to the movement downward and heat transfer of the cooling mold, the meniscus solidifies more quickly. From point c to point d (positive strip stage), with the mold oscillating from downward to upward and the solidified shell moving at the casting speed, the meniscus moves to the mold (Fig. 3d) and the oscillation mark forms. Then the new meniscus solidifies again. The cycle is repeated and periodic oscillation marks form.

The distribution of practical and simulated oscillation marks with the same casting parameters in Table 1 is illustrated in



Fig. 3: Temperatures of steel and mold during one cycle

Fig. 4. Figure 4(a) shows two adjacent oscillation marks on the longitudinal section vertical to slab surface, and the slab sample including the oscillation marks was low-carbon steel (content of carbon was 0.03wt.%), and cast at the casting speed of 1.45 m·min⁻¹ and the casting temperature of 1,883 K, with the mold oscillation of 182 cpm (*f*) and 6 mm (s_{osc}). The sample was taken from the narrow face of the slab, and the details of sample taken were introduced in literature^[15]. Figure 4(b) exhibits the distribution of simulated ones. The depression length of the simulated oscillation marks is larger than that of practical ones, because the simulation does not consider the strength of



Fig. 4: Distribution of oscillation marks, (a) practical ones; (b) simulated ones

the solidified shell, and the oscillation marks are depressiontypes. According to the oscillation frequency (182 cpm) and casting speed (1.45 m·min⁻¹), the theoretical oscillation pitch is 7.97 mm, while after measuring the simulated oscillation mark spacing, the average is 7.66 mm. The oscillation depth measured from the slab samples is 0.2–0.5 mm and the average is 0.299 mm, while the simulated oscillation depth is 0.3–0.5 mm and the average is 0.384 mm. Therefore the simulation in the study can explain the phenomena occurred at the meniscus and the formation of oscillation marks.

3 Conclusion

In the current study, a 2D model was established to simulate the multiphase fluid flow of steel-slag-air, heat transfer and solidification in the mold with the mold oscillation. The conclusions are as follows:

(1) In the mold, the liquid steel flows as upper backflow and lower backflow, and the small circulation forms at the meniscus. The upper backflow can increase the risk of slag entrapment.

(2) At the positive strip stage, the mold moves upwards with respect to the mold, and the slag infiltrates into the gap between the shell and the mold as lubricant. At the negative strip stage, the mold moves downwards relative to the mold, and the slag flows out of the gap. With the cooling of the mold with a lower temperature, the meniscus solidifies further, and then deeper oscillation marks may form.

(3) The simulated pitch and depth of oscillation marks approximate to the theoretical pitch and measured depth from the slab samples, respectively. Therefore, the simulation in the study can explain the phenomena occurred at the meniscus and the formation of oscillation marks.

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