



# Gelatin-Based Microribbon Hydrogels Guided Mesenchymal Stem Cells to Undergo Endochondral Ossification In Vivo with Bone-Mimicking Mechanical Strength

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## Abstract

Most stem cell-based bone tissue engineering strategies to date yield bone through direct bone formation, which mimics intramembranous ossification. However, bone injuries often affect long bones which are formed through endochondral ossification, involving an initial cartilage template formation followed by remodeling to form bones. There remains a critical need to develop scaffolds that enhance stem cell-based bone formation through endochondral ossification with bone-mimicking mechanical strength. Here we evaluated the potential of gelatin-based microribbons ( $\mu$ RBs) as macroporous scaffolds for enhancing human mesenchymal stem cell (MSC)-based bone formation through endochondral ossification. This material platform was compared with conventional gelatin hydrogels (HGs) as controls. MSCs were encapsulated in  $\mu$ RB or HG scaffolds, primed in chondrogenic medium in vitro for 2 weeks, and then implanted in a mouse subcutaneous model with no additional factors.  $\mu$ RB scaffolds supported fast cartilage deposition by MSCs, which was completely remodeled and replaced by mineralized bone. Impressively, the compressive moduli of MSC-seeded  $\mu$ RB scaffolds increased from 10 to 3224 kPa by week 11, a range that mimics native bone. In comparison, while HG supported endochondral ossification, the speed was much slower, with less matrix deposition and only a modest increase in compressive modulus to 269 kPa. These results validate gelatin  $\mu$ RBs as a promising scaffold for repairing long bone defects by guiding robust endochondral ossification.

## Lay Summary

Natural bone development and healing occurs through two distinct pathways: intramembranous ossification and endochondral ossification. Most bone injuries affect long bones, which are formed through endochondral ossification, involving an initial cartilage template formation followed by remodeling to form bones. However, scaffolds that can guide stem cell-based bone formation through endochondral ossification with bone-mimicking mechanical strength remain lacking. Here we report that macroporous gelatin-based microribbons ( $\mu$ RBs) accelerate endochondral ossification by human mesenchymal stem cells (MSCs) in vivo using a mouse subcutaneous model. Impressively, the mechanical properties of MSC-seeded  $\mu$ RB scaffolds

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increased over 300-fold over 11 weeks to bone-mimicking range, whereas conventional gelatin hydrogel controls reached less than 10% of the bone modulus. These results validate gelatin  $\mu$ RBs as a promising novel scaffold for repairing long bone defects by guiding robust endochondral ossification.

**Keywords** Macroporous · Scaffolds · Endochondral ossification · Bone · Stem cells · Gelatin · Microribbon

## Introduction

Bone loss affects millions of patients annually and can be caused by trauma, cancer resections, or degenerative diseases such as osteoporosis and osteonecrosis [1–4]. Bone is the second most commonly transplanted tissue after blood [5]. Given the donor scarcity and potential immunogenicity associated with bone grafts, there remains a significant need for alternative treatment options for repairing critical size bone defects that cannot heal on their own [1]. Depending on the anatomical locations, bones in the human body form and heal through two distinct processes: intramembranous ossification or endochondral ossification [6–8]. Flat bones in skull, mandible, maxilla, and clavicles form through intramembranous ossification, which is characterized by direct bone formation [9]. Long bones that make up most of the skeleton, on the other hand, are formed through endochondral ossification. While intramembranous ossification results in direct bone formation, endochondral ossification involves formation of an intermediate avascular cartilage template, which is subsequently remodeled and replaced by vasculature ingrowth and bone tissue [10–12].

To promote bone regeneration, various biomaterials have been employed including ceramics, poly(ester)-based scaffolds, and hydrogels [13]. Despite the extensive efforts and progress in utilizing biomaterials to enhance bone regeneration, most strategies developed to date resulted in direct bone formation [14–18], which only mimics flat bones formed through intramembranous ossification. The majority of bone fractures, however, occur in long bones, which are formed through endochondral ossification with a cartilage template. To induce endochondral ossification, recent studies primed mesenchymal stem cells (MSCs) in chondrogenic medium containing TGF- $\beta$ 3 *in vitro* for 2 to 5 weeks before *in vivo* implantation using a rodent subcutaneous model. This protocol was shown to induce cartilage template formation *in vitro*, which was subsequently remodeled and replaced by mineralized bone *in vivo* [19, 20]. However, several limitations remain. First, the previous method required additional factors such as cartilage-derived matrix particles or IL-1 $\beta$  [19, 20]. Second, the speed of bone formation remains very slow, requiring up to 14 weeks to achieve only partial mineralization [20, 21]. Third, these studies have been unable to achieve bone-mimicking mechanical strength, characterized by high

stiffness in the range of MPa [22, 23] due to slow and limited mineralization.

To facilitate clinical translation, the ideal scaffold for MSC-based bone regeneration should support direct and homogeneous cell encapsulation, contain macroporosity to accelerate new matrix deposition, and facilitate rapid increase in the mechanical properties of resulting bone. While conventional hydrogels support easy cell encapsulation, they generally contain porosity orders of magnitude smaller than cells, leading to physical restriction and slow tissue formation [24–26]. Although porosity can be introduced into hydrogels using porogens, this further reduces the already weak mechanical properties of hydrogels [27]. Conventional methods for fabricating macroporous scaffolds, such as PLGA sponges, do not support direct cell encapsulation. To overcome the limitations associated with conventional biomaterials for bone regeneration, our lab has recently reported a method to fabricate gelatin into microribbon ( $\mu$ RB)-shaped hydrogel building blocks featuring widths (40–100  $\mu$ m) larger than the size of individual cells, which support homogeneous cell encapsulation, as shown by confocal microscopy, and can subsequently intercrosslink to form 3D macroporous cell-laden scaffolds [25, 28]. Unlike conventional hydrogels, the resulting  $\mu$ RB scaffolds exhibit great shock-absorbing capacity and maintain structural integrity upon cyclic compression [25, 26, 28]. We further demonstrated that the macroporosity of  $\mu$ RB scaffolds made with 7.5% (w/v)  $\mu$ RBs substantially accelerated stem cell-based cartilage regeneration *in vitro* [26] and enhanced stem cell survival and endogenous bone formation *in vivo* [25]. However, the potential of gelatin  $\mu$ RB scaffolds to support MSC-based endochondral ossification has yet to be explored. Here we seek to compare gelatin  $\mu$ RB scaffolds to gelatin hydrogel (HG) scaffolds for their potential to guide MSC-based bone formation via endochondral ossification. We hypothesized that the macroporosity within  $\mu$ RB scaffolds would accelerate and enhance new bone formation and result in a faster increase in mechanical strength of engineered bone tissue. To induce the formation of an intermediate cartilaginous template, human MSCs were encapsulated in both  $\mu$ RB and HG scaffolds and cultured in chondrogenic medium supplemented with TGF- $\beta$ 3 for 2 weeks. The scaffolds were then implanted subcutaneously in nude mice for up to 9 additional weeks *in vivo* without any additional growth factors. The resulting tissues were characterized at multiple time points using mechanical testing, histology, and immunostaining.

## Materials and Methods

### Materials

Dimethyl sulfoxide (DMSO), L-lysine hydrochloride, gelatin type-A (Bloom number range: 90 to 300 g), methacrylic anhydride, and glutaraldehyde were purchased from Sigma-Aldrich (St. Louis, MO). All materials were used as received.

### Scaffolds Fabrication and Characterization

Methacrylated gelatin was synthesized as previously described for forming conventional hydrogels [26]. Gelatin  $\mu$ RBs (40–100  $\mu$ m wide, 2–3 mm long, and about 4  $\mu$ m thick) were fabricated using wet-spinning as previously reported [26]. Both  $\mu$ RB and HG scaffolds (2 mm in height and 4.5 mm in diameter) were formed using the same concentration of gelatin (7.5% w/v) and photocrosslinked, induced by a photoinitiator (LAP, 0.05% w/v) and UV light (365 nm, 2 mW/cm<sup>2</sup>, 4 min) [28]. Scaffold morphology was characterized using scanning electron microscopy with a Hitachi S-3400N variable pressure scanning electron microscope [26]. The sample stage was cooled gradually from room temperature to operating temperature (– 25 °C); simultaneously, the chamber pressure was reduced from 1 atm to 50 Pa, following a pressure-temperature curve at which water stays in the liquid phase, to ensure samples remained hydrated and did not dry or freeze. Images were taken using an electron beam of 15 kV at a working distance of ~ 7 mm.

### Cell Encapsulation in 3D

Human MSCs (Lonza, Basel, Switzerland) were expanded in standard MSC growth medium [26] until passage 6 before use for encapsulation. MSCs were encapsulated in either gelatin  $\mu$ RBs or gelatin HGs at 15 million cells/ml. The same gelatin concentration (7.5% w/v) was used for both  $\mu$ RB and HG groups. The cell/polymer mixture was loaded between two glass slides (with a 2 mm spacer) and crosslinked by UV light (365 nm, 2 mW/cm<sup>2</sup>, 4 min). After 24 h of incubation in MSC growth medium, cylindrical samples (4.5 mm in diameter) were punched out using a biopsy punch and transferred to 24-well plates. Cell viability 24 h after encapsulation was evaluated using a LIVE/DEAD Viability/Cytotoxicity Kit (Invitrogen, Carlsbad, CA). Images were acquired using a Zeiss fluorescence microscope (Axio Observer 3.1, Zeiss, Oberkochen, Germany) and the ZEN 2 (blue edition) software (Zeiss).

### In Vitro Priming and In Vivo Implantation

To induce endochondral ossification, all cell-laden scaffolds were first primed in vitro for 2 weeks in chondrogenic

medium containing high-glucose DMEM, 10 ng/ml of TGF- $\beta$ 3, 100 nM of Dexamethasone, 50  $\mu$ g/ml of ascorbic-2-phosphate, 40  $\mu$ g/ml of proline, 5  $\mu$ g/ml of ITS premix, 100  $\mu$ g/ml of sodium pyruvate, 100 units/ml of penicillin, and 100  $\mu$ g/ml of streptomycin [26], followed by subcutaneous implantation in vivo using an immunocompromised mouse model (NCRNU, female, 9 weeks, Taconic). All animal studies were performed in accordance with the National Institutes of Health guide for the care and use of Laboratory animals (NIH Publications No. 8023, revised 1978); all protocols were approved by the Stanford institutional animal care and use committee. Samples were harvested at multiple time points (2, 5, and 11 weeks from the initial encapsulation) for analyses. Outcomes were analyzed using mechanical testing ( $n = 5$ /group), histology, and immunostaining ( $n = 3$ /group).

### Histology and Immunostaining

In vivo samples showed excellent host-tissue integration and hence scaffolds were harvested with peripheral host tissue, as especially the  $\mu$ RB samples were impossible to separate from the fibrous capsule. At the time of each harvest time point, all samples were fixed in 4% paraformaldehyde (4 °C, 2 h), embedded in optimal cutting temperature (OCT) compound (VWR International, Radnor, PA), and frozen using liquid nitrogen vapor. A Microtome Cryostat (Triangle Biomedical Sciences, Durham, NC) was used to cut 8  $\mu$ m-thick sections from the OCT-embedded samples. Safranin-O staining (Sigma-Aldrich) was used to visualize sulfated glycosaminoglycan (sGAG) deposition. Collagen deposition and tissue morphology was examined with Masson's trichrome staining (Thermo Scientific, Waltham, MA). Alkaline phosphatase (ALP) staining was visualized using Fast Blue RR Salt (FBS25, Sigma-Aldrich) and Naphthol AS-MX phosphate (855, Sigma-Aldrich). Mineralization was stained using Alizarin Red S (ARS) (A5533, Sigma-Aldrich). For immunostaining, sections were blocked with 2% goat serum and 3% bovine serum albumin in 1X PBS for 1 h and incubated overnight at 4 °C in the presence of diluted primary antibodies. The following antibodies were used: rabbit polyclonal antibodies against collagen I (ab34710, 1:100, Abcam, Cambridge, MA), collagen II (ab34712, 1:80, Abcam), collagen X (ab58632, 1:100, Abcam), and osteopontin (ab8448, 1:100, Abcam); and rat polyclonal antibodies against osteocalcin (ab13418, 1:80, Abcam), F4/80 (MCA497GA, 1:50, Bio-Rad, Hercules, CA), or vimentin (ab92547, 1:200, Abcam). Depending on the primary antibodies of choice, the following secondary antibodies were used: Alexa Fluor 488 goat anti-rabbit (1:100, Invitrogen) and Alexa Fluor 488 goat anti-rat (1:100, Invitrogen). Samples were also incubated without primary antibodies and only secondary antibodies as negative controls. Nuclei were counter stained using Hoechst 33342 (Thermo Scientific). Images were taken with a Zeiss

fluorescence microscope (Axio Observer 3.1) and the ZEN 2 (blue edition) software (Zeiss).

## Mechanical Testing

Unconfined compression tests were conducted using an Instron 5944 system (Instron Corporation, Norwood, MA) fitted with a 10 N load cell (Interface Inc., Scottsdale, AZ). Sample constructs were tested on the following timepoints: weeks 0, 2, 5, and 11 ( $n = 3\text{--}5$  per time point). The setup consisted of custom-made aluminum compression platens lined with polytetrafluoroethylene to minimize friction. All tests were conducted in PBS at room temperature. Before each test, a preload of approximately 10 mN was applied. The upper plate was then lowered at a rate of 1% strain/s to a maximum strain of 30%. Load and displacement data were recorded at 100 Hz. The compressive modulus was calculated from a linear curve fit of the stress versus strain curve within the range of 20–30%.

## Statistical Analysis

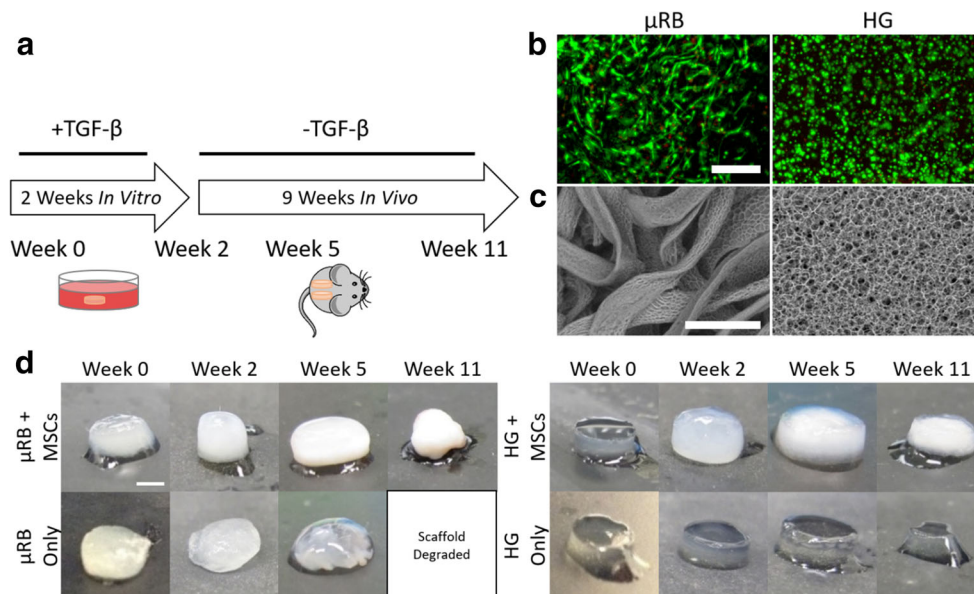
All data were expressed as mean  $\pm$  standard error. Statistical significance was determined via analysis of variance using Student's *t* test with equal variance. *p* values (two-tailed)  $< 0.05$  were considered statistically significant, and *p* values  $< 0.005$  were considered statistically highly significant.

## Results

### $\mu$ RB Scaffolds Accelerated Matrix Deposition by MSCs and Exhibited Faster Degradation In Vivo

To induce endochondral ossification, MSC-seeded  $\mu$ RB and HG scaffolds were first cultured in vitro in chondrogenic medium containing TGF- $\beta$ , then implanted in a mouse subcutaneous model without any growth factors for up to 9 additional weeks (Fig. 1A). Acellular  $\mu$ RB and HG groups were included as negative controls. LIVE/DEAD staining showed high cell viability and homogeneous cell distribution in both groups 24 h after encapsulation (Fig. 1B). Cells encapsulated in the  $\mu$ RB scaffolds showed extensive spreading on the surface of the ribbons but remained round and entrapped in HG (Fig. 1B). The observed cell distribution and morphology corroborated whole sample confocal microscopy results published previously [25]. SEM showed that  $\mu$ RB scaffolds exhibited a highly interconnected macroporous network, whereas HG contained much smaller and restrictive pores (Fig. 1C).

Visual examination of sample gross morphology over time reveals that MSC-seeded  $\mu$ RB scaffolds showed a more rapid increase in opacity than HG over time, suggesting that  $\mu$ RBs supported faster new matrix deposition by MSCs compared with HG (Fig. 1D). MSC-seeded  $\mu$ RB scaffolds became white and shiny by week 5, resembling the appearance of articular cartilage. (Fig. 1D). In contrast, acellular  $\mu$ RB scaffolds



**Fig. 1** Experimental design and characterization of gelatin-based  $\mu$ RB scaffolds vs. conventional HGs as a 3D stem cell niche. (A) Schematic of in vivo experimental design. Human MSCs were encapsulated in gelatin  $\mu$ RB or HG scaffolds and first cultured in chondrogenic medium containing TGF- $\beta$ 3 for 2 weeks in vitro. The samples were then implanted in a mouse subcutaneous model without additional growth factors. Samples were harvested at weeks 5 and 11 for analysis. (B)

Cell viability and morphology in  $\mu$ RB and HG scaffolds 24 h after encapsulation as shown by LIVE/DEAD staining. Green: live cells; red: dead cells; scale bar 200  $\mu$ m. (C) SEM to compare the macroporosity in the  $\mu$ RB and HG scaffolds. Scale bar 100  $\mu$ m. (D) Gross morphology of  $\mu$ RB and HG scaffolds with or without MSCs over time. Increase in sample opacity indicates matrix deposition. Scale bar 2 mm



quickly degraded over time, becoming transparent by week 5. By week 11 they could not be found, suggesting full degradation (Fig. 1D). While the MSC-seeded HGs also showed increasing opacity over time, the degree of opacity was much less compared with the MSC-seeded  $\mu$ RB scaffolds, suggesting HGs induced less and slower matrix deposition (Fig. 1D). In addition, the newly deposited matrix was not as homogeneous in HGs, as the opacity was higher in the center of HGs, leaving a transparent edge. Acellular HG samples remained transparent throughout the duration of the experiment, showing partial degradation by week 11 (Fig. 1D).

### MSC Seeded $\mu$ RB Scaffolds, but Not HGs, Enabled Fast Increase in Compressive Moduli of Engineered Tissues over Time

Given that weight bearing is an important function of bone tissue, we next characterized the changes in compressive modulus of MSC-seeded  $\mu$ RB scaffolds or HG scaffolds over time using unconfined compression testing. At week 0, all samples exhibited low compressive moduli, with  $\sim 6$ – $10$  kPa for  $\mu$ RB scaffolds, and higher for HG scaffolds ( $\sim 23$ – $40$  kPa) (Fig. 2). After 2 weeks of in vitro culture in chondrogenic medium containing TGF- $\beta$ , the compressive modulus of MSC-containing  $\mu$ RB scaffolds increased 21-fold, reaching 210 kPa, whereas MSC-seeded HGs only increased slightly to 47 kPa (Fig. 2). Acellular  $\mu$ RB scaffolds degraded substantially, accompanied by a decrease in compressive modulus to  $\sim 2$  kPa (Fig. 2). No significant changes were observed in compressive modulus of acellular HGs (Fig. 2).

Following subcutaneous implantation, the compressive modulus of MSC-seeded  $\mu$ RBs or HGs showed a minor change between weeks 2 and 5, followed by a huge increase between weeks 5 and 11. By week 11, the compressive modulus of MSC-seeded  $\mu$ RBs reached an impressive 3224 kPa (Fig. 2), which is comparable to the compressive modulus of native bone tissue [22, 23]. In contrast, acellular  $\mu$ RB samples had fully degraded by this point. MSC-seeded HGs showed a smaller increase to 270 kPa by week 11, less than 10% of the

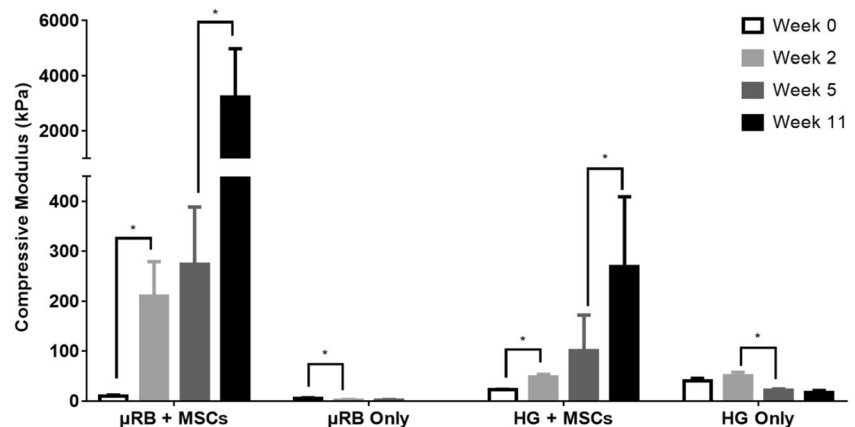
compressive modulus of the MSC-seeded  $\mu$ RBs (Fig. 2). Compared with week 0, MSC-seeded  $\mu$ RB and HG scaffolds exhibited a total of 325-fold and 12-fold increase in compressive modulus, respectively (Fig. 2). Given the same amount of gelatin was used for making  $\mu$ RB or HG scaffolds, macroporous  $\mu$ RB scaffolds were much more efficient in enabling rapid increase of mechanical strength of new bone tissues deposited by MSCs in 3D.

### Macroporous $\mu$ RB Scaffolds Accelerated Cartilage Deposition and Subsequent Remodeling In Vivo

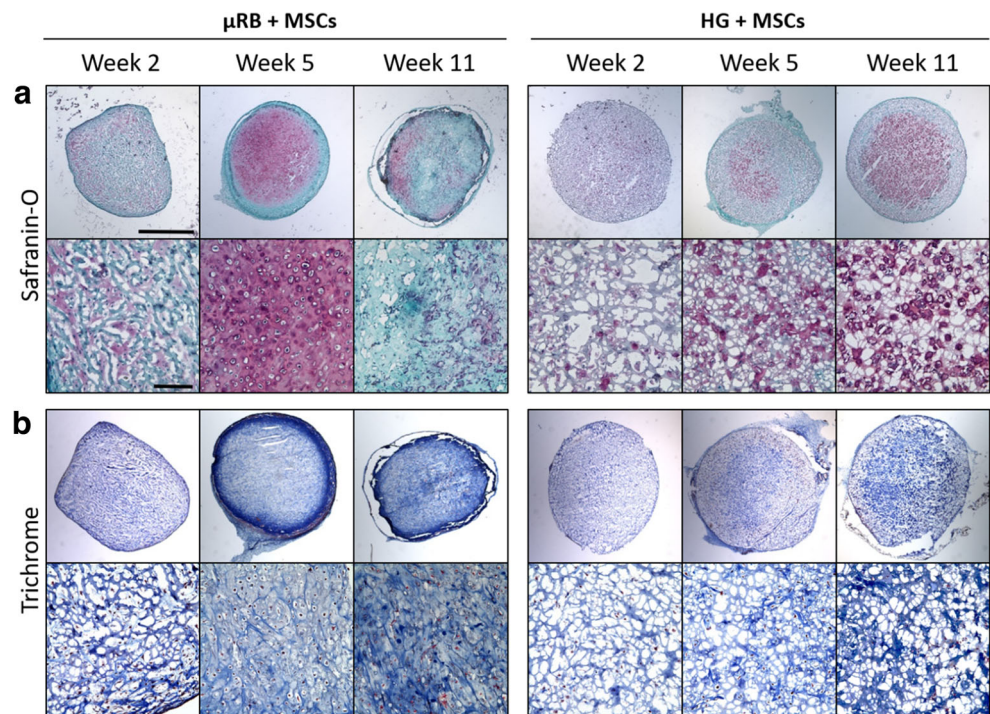
To assess whether  $\mu$ RB and HG scaffolds can support temporary cartilage formation and subsequent remodeling in vivo, samples were harvested at weeks 5 and 11 and stained for cartilage matrix. Safranin-O staining was used to visualize the amount and distribution of sGAGs, a key component of cartilage matrix. Low magnification images were shown to illustrate the matrix distribution throughout the scaffolds, and high magnification images were shown for assessing matrix distribution on the cellular level. By week 5,  $\mu$ RB + MSC scaffolds were completely filled with sGAGs distributed in a highly interconnected manner, but sGAGs were almost completely absent by week 11 (Fig. 3A). In contrast, the HG + MSC group only showed a small amount of sGAG deposition by week 5, limited to the center of the sample, which continued to increase by week 11 (Fig. 3A). Acellular scaffolds were also stained and showed no sGAGs at any time point (Fig. S1A), confirming all sGAGs were deposited by encapsulated MSCs, not host cells.

To view total collagen deposition, Masson's Trichrome staining was performed on all samples. A fibrous capsule can be found around the edge of scaffolds in all groups over time (Fig. 3B, Fig. S1B). Unlike the trend of sGAGs, collagen deposition showed a continuous increase in both  $\mu$ RB + MSC and HG + MSC groups, with the newly deposited matrix filling up the pores in the scaffolds (Fig. 3B). Since the scaffolds are gelatin-based, which stains positive for Masson's Trichrome, we also included acellular samples to show the

**Fig. 2** Characterization of compressive moduli of gelatin  $\mu$ RB scaffolds or gelatin HGs, with or without MSCs over 11 weeks using unconfined compressive testing. The compressive moduli of the acellular  $\mu$ RB group were not performed at week 11 due to complete sample degradation. \* $p < 0.05$



**Fig. 3** Gelatin  $\mu$ RB scaffolds accelerated the speed and amount of new tissue matrix deposition compared with HGs over time. (A) Safranin-O staining for sGAGs, a major component of articular cartilage; (B) Masson's Trichrome staining for total collagen. Low magnification (upper row) and high magnification (lower row) images are provided for both stains. Scale bars in low magnification images: 1 mm; scale bars in high magnification images: 200  $\mu$ m



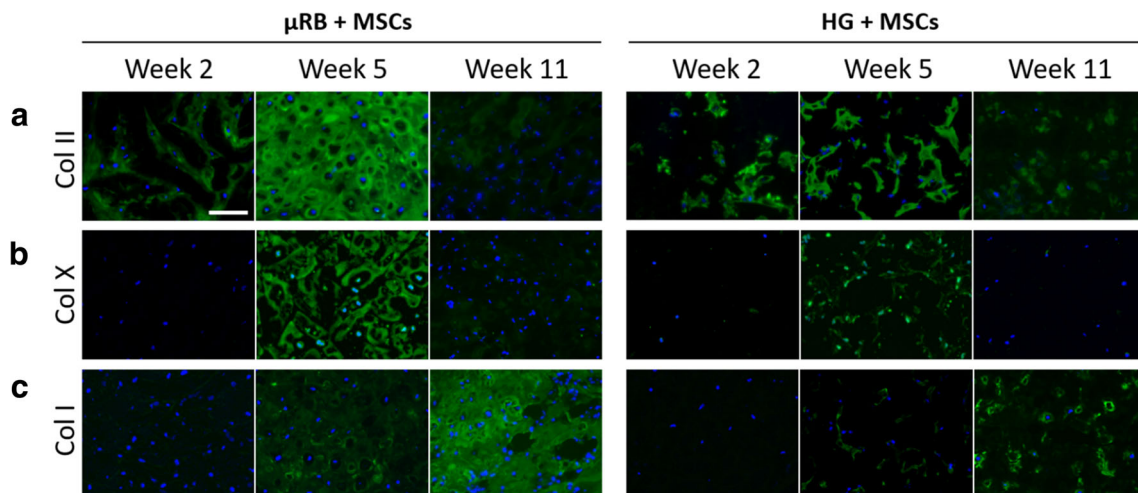
background due to the scaffolds. Only scaffold structures similar to SEM results were observed (Fig. S1B), confirming the majority of collagen filling the spaces throughout the macropores were contributed by MSC-deposited new matrix.

To further characterize the cartilage phenotype, immunostaining was performed for type II collagen (a marker for desirable articular cartilage), type X collagen (hypertrophic cartilage marker) and type I collagen (fibrocartilage or bone marker) (Fig. 4, S2). For MSC-seeded  $\mu$ RBs, type II collagen increased at week 2, peaked at week 5, but largely disappeared by week 11 (Fig. 4A). A similar trend was observed for type X collagen (Fig. 4B). Collagen I started to be positive at week 5, and continued to increase by week 11 (Fig. 4C). A similar

trend of phenotype change was observed in the HG + MSC group, but the amount of matrix was much less than in the  $\mu$ RB + MSC group and limited to only the pericellular regions (Fig. 4). Acellular samples showed no background staining in any sample (Fig. S2), confirming all matrix was contributed by transplanted MSCs, not by host cells.

#### **$\mu$ RB Scaffold Accelerated Endochondral Ossification by MSCs In Vivo**

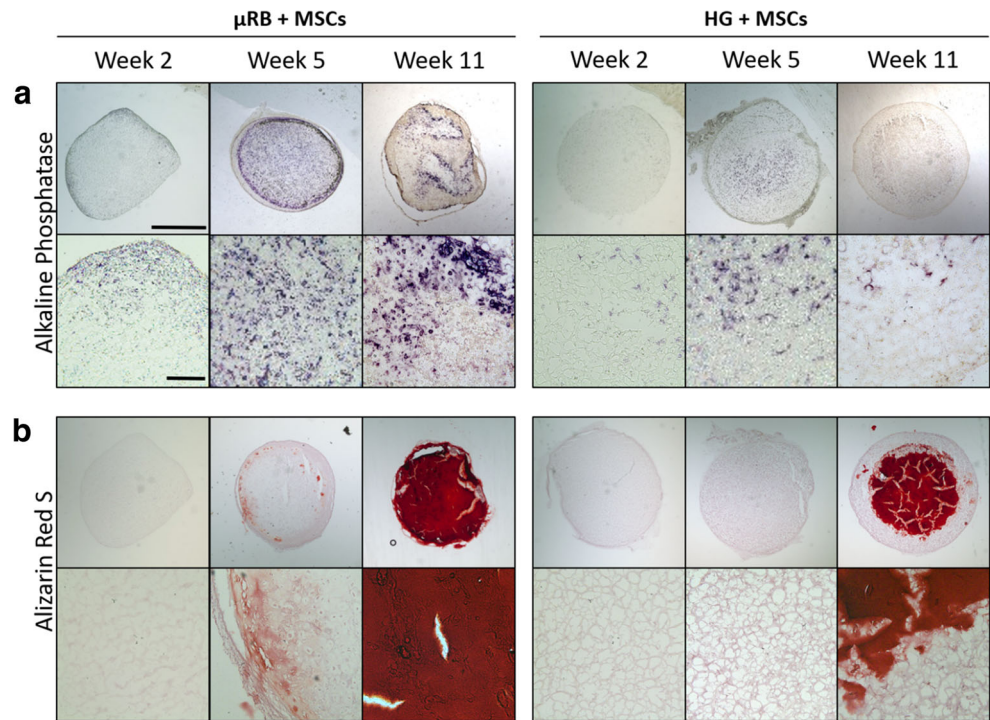
The disappearance of cartilage markers (sGAG and type II/X collagen) after week 5, combined with the increase in type I collagen by week 11, suggested a phenotype transition of



**Fig. 4** Immunostainings of different types of collagen deposited by MSCs showed gelatin  $\mu$ RB scaffolds accelerated endochondral ossification compared with HGs. (A) Type II collagen, (B) type X collagen, (C) type I collagen. Green: specific type of collagen; blue: nuclei. Scale bar 100  $\mu$ m



**Fig. 5** Staining of bone markers showed gelatin  $\mu$ RB scaffolds accelerated mineralized bone formation over time compared with gelatin HG. (A) ALP, an early bone marker; (B) ARS staining for mineralization. Upper row: low magnification, scale bar 1 mm; lower row: high magnification, scale bar 200  $\mu$ m



MSC-deposited tissues from cartilage to bone between weeks 5 and 11 (Figs. 3 and 4). To further confirm the bone phenotype, all samples were stained for ALP, an early bone marker, as well as mineralization using ARS staining. For the  $\mu$ RB + MSC group, ALP increased substantially at week 5 and decreased by week 11 (Fig. 5A). ARS staining shows that mineralized bone started to emerge in the periphery of MSC-seeded  $\mu$ RBs at week 5 and filled the entire scaffold with intense mineralization by week 11 (Fig. 5B). The HG + MSC sample showed delayed bone formation, with no mineralization observed at week 5, and intense mineralization was observed by week 11 but limited to only the center of the sample (Fig. 5A, B). Acellular control samples were negative for ALP and ARS for all samples at all timepoints (Fig. S3).

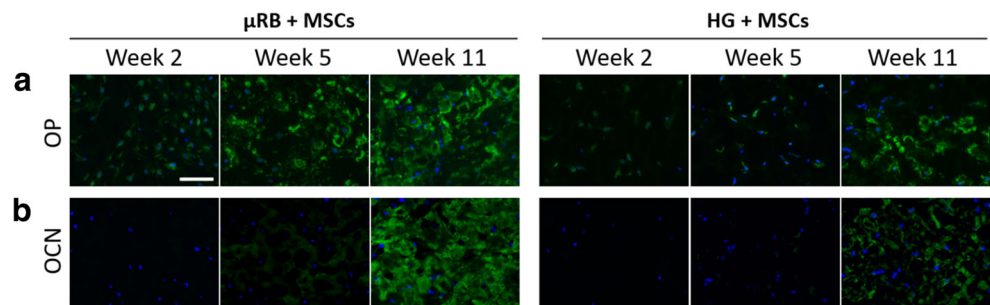
To further characterize the speed of bone formation, we performed immunostaining for osteopontin and osteocalcin, which are intermediate and late bone markers respectively (Fig. 6, S4). Consistent with the histology results, immunostaining shows that  $\mu$ RBs supported a much faster transition to bone phenotype than HGs, as both markers were expressed

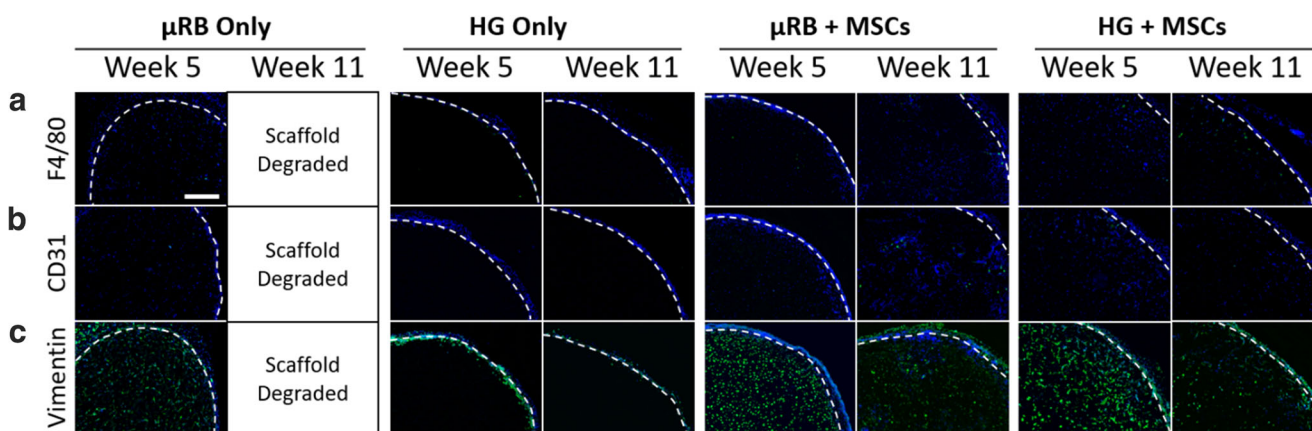
much earlier in the  $\mu$ RB samples (Fig. 6). Acellular samples showed no signals at any timepoints (Fig. S4).

### Characterization of Host Cell Infiltration In Vivo

In vivo samples were harvested with the surrounding host tissues intact as the macroporosity of  $\mu$ RB scaffolds induced rapid host tissue ingrowth, leading to excellent host tissue integration and difficulty in separating the implants from the surrounding host tissue. Histological staining from samples harvested at weeks 5 and 11 demonstrated a very distinctive in vivo response between  $\mu$ RB and HG scaffolds (Fig. 7). Masson’s Trichrome staining showed a fibrous capsule formation around both cellular and acellular  $\mu$ RB and HG scaffolds (Fig. 3B, S1B). To further characterize the phenotype of these cells, samples were stained with F4/80, a marker for macrophages. The border of the  $\mu$ RB and HG scaffolds were marked with a white dotted line (Fig. 7A). While a small number of invading cells in the acellular  $\mu$ RB sample were F4/80 positive, the majority of cells in all groups were not macrophages

**Fig. 6** Immunostainings of osteopontin and osteocalcin showed  $\mu$ RB scaffolds accelerated MSC osteogenesis and bone maturation compared with HGs. (A) Osteopontin; (B) osteocalcin. Scale bar 100  $\mu$ m





**Fig. 7** Characterization of host cell infiltration into gelatin  $\mu$ RB scaffolds and gelatin HGs in a mouse subcutaneous model. Immunostaining was performed for markers for macrophages (F4/80), endothelial cells

(CD31), and fibroblastic/mesenchymal cells (vimentin). White dotted line indicates the boundary of scaffolds. (A) F4/80; (B) CD31; (C) Vimentin. Scale bar 100  $\mu$ m

(Fig. 7A). Staining with the endothelial marker CD31 for blood vessel invasion was also largely negative except for the  $\mu$ RB + MSC group at week 11 (Fig. 7B). Staining with vimentin revealed that the large majority of cells migrating into the  $\mu$ RB scaffold or forming the fibrous capsule around the HG were fibroblastic/mesenchymal cells (Fig. 7C). Endogenous cells were unable to invade into the acellular HG samples, suggesting that the vimentin positive cells in the HG + MSC group are encapsulated MSCs rather than host cells (Fig. 7C). In both cellular samples ( $\mu$ RB and HG), the vimentin signal decreased within the scaffold between weeks 5 and 11. Together, these results indicate that macroporous  $\mu$ RB scaffolds, but not HGs, enabled rapid host cell infiltration into the scaffolds, which is desirable for promoting host tissue integration (Fig. 7).

## Discussion

Here we report that macroporous gelatin  $\mu$ RB scaffolds support MSC-based endochondral ossification *in vivo* with short *in vitro* priming in chondrogenic medium and no additional growth factors *in vivo*. Only gelatin  $\mu$ RB scaffolds, but not conventional HGs, enabled a fast increase in compressive modulus, first to the range of articular cartilage [29] and then to bone-mimicking stiffness ( $\sim 3$  MPa) [30] (Fig. 2). In addition, the macroporous  $\mu$ RB scaffold improved the homogeneous distribution of *de novo* deposited cartilage and bone matrix throughout the scaffold (Figs. 3, 4, 5, and 6). These results support our hypothesis that the macroporosity within the  $\mu$ RB scaffold would accelerate and enhance matrix formation by MSCs in 3D, enabling faster restoration of mechanical strength of engineered tissues. Together, this study validates the efficacy of gelatin-based  $\mu$ RB scaffolds in supporting MSC-based bone regeneration through endochondral

ossification and its substantial advantages over conventional gelatin HGs.

Most bone defects impact long bones, which are formed through endochondral ossification. Ideal scaffolds for stem cell-based bone repair for long bone defects should support the cartilage to bone phenotype transition while supporting rapid restoration of bone-mimicking mechanical strength. While some studies demonstrated endochondral ossification, none have reported achievement of mechanical strength in the range of native bone [19, 31–33]. Hydrogels in particular have shown limited success, with engineered bone tissue demonstrating compressive moduli an order of magnitude lower than that of native bone [34, 35]. This is consistent with our findings, where MSC-seeded HGs achieved less than 10% of the compressive moduli of native bone after 11 weeks (Fig. 2). In contrast, the macroporous gelatin  $\mu$ RB scaffold achieved cartilage-mimicking strength (210 kPa) after 2 weeks of culture *in vitro* and increased to a bone-mimicking compressive modulus by week 11 (3224.3 kPa) (Fig. 2). This increase is remarkable considering that the  $\mu$ RB scaffold alone is characterized by a compressive modulus of only  $\sim 10$  kPa. Given that the HG and  $\mu$ RB scaffolds were made from the same concentration of gelatin, this stark contrast in the quality of new bone highlights the importance of scaffold design. The inherent interconnected macroporosity in the gelatin  $\mu$ RB scaffold reduces the physical restrictions experienced by cells in HGs, supporting fast cell spreading and interconnectivity of newly deposited matrix as shown by histology (Figs. 3, 4, 5, and 6).  $\mu$ RBs not only accelerated the speed of matrix deposition but also promoted a denser and more interconnected matrix network; in contrast, the matrix was limited to isolated pericellular deposition in HGs. Due to the extensive infiltration of host cells into the  $\mu$ RB scaffolds, it was not feasible to separate  $\mu$ RB implants from surrounding host tissues for accurate biochemical quantification, so we chose histology rather than biochemical assays to more accurately characterize the



tissue formation. Consistent with the findings from the present study, our previous *in vitro* study comparing the  $\mu$ RB and HG scaffolds for chondrogenesis also showed that macroporous gelatin  $\mu$ RB scaffolds substantially accelerate cartilage regeneration by MSCs with great interconnectivity compared with gelatin HGs [26]. Importantly, the fold of increase in mechanical property was much higher than the fold of change in total matrix content [26], confirming the improved interconnectivity is critical for improved mechanical function of engineering tissues. Since histology data shows distinct differences in both the amount and distribution of newly deposited tissues (Figs. 3, 4, 5, and 6) between  $\mu$ RB vs. HG groups, it is a much more reliable and accurate way to analyze *in vivo* tissue formation in this study.

To assess the biocompatibility and host integration of  $\mu$ RBs *in vivo*, we characterized host cell infiltration into the scaffolds using immunostainings for markers of macrophages (F4/80), endothelial cells (CD31), and fibroblastic/mesenchymal cells (vimentin). We observed minimal signals for macrophages and endothelial cells across all groups at both time points (weeks 5 and 11) (Fig. 7). The absence of detectable macrophages may be due to the relatively late time points, which passed the duration of the initial acute inflammatory phase. The absence of vasculature at week 5 is not surprising given that cartilage formation peaked at week 5. Cartilage is an avascular tissue and is known to secrete anti-angiogenic factors [36, 37]. With the remodeling of the cartilaginous template and formation of bone by week 11, we observed an increasing number of CD31-positive cells in the  $\mu$ RB scaffold. This is consistent with previous reports that angiogenesis plays a role in bone formation and fracture healing [10]. Endothelial cells were only observed in the  $\mu$ RB + MSC group, suggesting that both macroporosity and the presence of MSCs are necessary to promote vascular ingrowth. MSCs have been known to secrete a broad spectrum of paracrine signals that promote angiogenesis [38–40]. The fact that the HG + MSC group did not show any endothelial cell infiltration suggests the lack of macroporosity in conventional gelatin HGs inhibits host cell infiltration. This is further supported by the results of immunostaining of vimentin, which is an intermediate filament protein that is expressed in mesenchymal cells such as fibroblasts [41]. While acellular  $\mu$ RBs showed extensive staining of vimentin-positive cells throughout the scaffolds at week 5, acellular HGs completely inhibited cell infiltration throughout 11 weeks. For successful cartilage tissue regeneration, host tissue integration is essential, which requires rapid host cell infiltration. Our results validate that gelatin  $\mu$ RB scaffolds, but not HGs, are a much better choice for host tissue integration enhancement.

When characterizing biomaterials for *in vivo* applications, one important factor to consider is degradation. Ideally, the scaffold should degrade over time as the new tissue forms. Acellular  $\mu$ RB scaffolds degraded much faster than HGs and

could not be retrieved by week 11 (Figs. 1D and 7). In contrast, acellular gelatin HGs showed minimal degradation. Given both scaffolds were made from the same concentration and composition (gelatin), the observed drastic difference in degradation is likely a result of differences in the macroporosity and speed of host cell infiltration (Fig. 7). As noted above,  $\mu$ RB scaffolds enabled rapid host cell infiltration and tissue integration while HGs inhibited host cell infiltration. Furthermore, the  $\mu$ RB scaffolds started to lose integrity as the intercrosslinks among  $\mu$ RB building blocks degraded, exposing increased surface area to infiltrated or transplanted cells which further accelerated the degradation of individual  $\mu$ RBs.

Another finding from this study is that transient priming of MSCs in chondrogenic medium containing TGF- $\beta$ 3 (2 weeks *in vitro*) was sufficient to induce MSCs to undergo endochondral ossification *in vivo* without the use of any additional growth factors. The simplicity of the protocol deems it highly desirable for clinical translation as it avoids cost for additional growth factors that suffer from short half-lives *in vivo* and can induce undesirable excessive bone formation [42]. Similar to our findings, previous studies have shown 2 to 5 weeks of *in vitro* priming in chondrogenic conditions was sufficient to induce endochondral ossification in a mouse subcutaneous model without additional growth factors [19, 21].

## Conclusion

In summary, here we demonstrated that macroporous gelatin  $\mu$ RB scaffolds accelerated and enhanced endochondral ossification of human MSCs *in vivo* without the need for additional growth factors *in vivo*. The resulting tissue-engineered bone exhibited a rapid increase in mineralization and reached bone-mimicking mechanical strength. Unlike conventional HGs,  $\mu$ RBs exhibited a toothpaste-like texture when hydrated in PBS for injection, which makes it easier to apply for filling irregular-shaped defects *in vivo* than liquid hydrogel precursors. Given that macroporous  $\mu$ RB scaffolds induced rapid and extensive host cell infiltration, they also show great advantage in promoting host tissue integration, which is highly desirable for tissue engineering scaffolds. In the present study we have used a mouse subcutaneous model to demonstrate proof of principle given it is a more cost-effective and higher throughput choice and allowed testing of samples with larger sizes. Now that we have proven the advantages of gelatin  $\mu$ RBs over conventional HGs, future studies can further validate the potential of  $\mu$ RBs for supporting endochondral ossification using long bone defect models [31, 32, 43, 44].

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## Compliance with Ethical Standards

**Ethical Approval** All applicable international, national, and/or institutional guidelines for the care and use of animals were followed. All procedures performed in studies involving animals were in accordance with the ethical standards of the institution or practice at which the studies were conducted.

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