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Near UV-pumped yellow-emitting Sr₉MgLi(PO₄)₇:Eu²⁺ **phosphor for white-light LEDs**

Jianwei Qiao, Zhiguo Xia* [, Zhichao Zhang, Bintao Hu and Quanlin Liu](#page-0-0)

ABSTRACT Laboratory discovery of new phosphors for white-light light-emitting diodes (WLEDs) is still an imperative challenge. A new yellow-emitting $Sr_9MgLi(PO_4)_7:Eu^{2+}$ **phosphor was discovered based on the mineral-inspired prototype evolution and new phase construction strategy pro**posed by our group. Sr₉MgLi(PO₄)₇:Eu²⁺ has been synthesized **by using a high temperature solid-state method, and its phase structure and luminescence properties have been investigated** in detail, and applied in WLED lamp. $Sr_0MgLi(PO_4)_7$ phase is derived from the β -Ca₃(PO₄)₂-type mineral structure. Upon **365 nm UV light excitation, the** $Sr_9MgLi(PO_4)_7:Eu^{2+}$ **phosphor exhibits a broad emission band from 450 nm to 700 nm. The white-light LED lamp was fabricated based on the phosphor** blends of the composition-optimized yellow-emitting Sr₉MgLi $(PO_4)_7$: Eu^{2+} and commercial blue-emitting $BaMgAl_{10}O_{17}$: Eu^{2+} , **and a 365 nm UV chip was used as the excitation source. The** *R***a, CCT value and CIE of the as-fabricated LEDs were found to be 83, 5,612 K, and (0.324, 0.358), respectively. All the results indicate that** $Sr_9MgLi(PO_4)_7:Eu^{2+}$ **could be potential in the development of UV-pumped white-light LEDs.**

Keywords: phosphor, luminescence, rare earth, LEDs

INTRODUCTION

Rare earth (RE) doped photoluminescence materials have been utilized in many different fields including lighting, lasers, displays, biomedicines, and so on, because of their abundant energy levels transitions over a wide range of photon energies [1–[3\]](#page-5-0). This motivates the search for novel RE materials for emerging applications. Recently, phosphor-converted white light-emitting diodes (pc-WLEDs) have attracted much attention due to their high energy efficiency, high material stability, long operational lifetime and environmentally friendly characteristics [4– [7\]](#page-5-1). Typically, the commercial pc-WLEDs are a combination of a blue-emitting InGaN chip and yellow-emitting $Y_3AI_5O_{12}$: Ce^{3+} (YAG: Ce^{3+}) phosphors [\[8\]](#page-5-2). However, the lacking of red spectral contribution in YAG: Ce^{3+} phosphors leads to a poor color-rendering index $(R_{\alpha} < 75)$ and a high correlated color temperature (CCT>7,000 K) [9,[10](#page-5-3)]. To compensate the lacking of red components, red-emitting phosphors or quantum dots have been studied and applied in WLEDs, such as K_2SiF_6 : Mn^{4+} phosphors and $CsPbBr_3$ quantum dots [11[,12\]](#page-5-4). Another alternative way to overcome those disadvantages is to combine the near-ultraviolet (n-UV) LED chip with blue and yellow phosphors [\[4\].](#page-5-5) To create pc-WLEDs with high quality photoluminescence properties, much effort has been devoted to phosphors in different host systems.

Recently, our group has proposed a useful method for the discovery of new LED phosphors by mineral-inspired prototype evolution and new phase construction, and it has been used in many different systems [\[4\].](#page-5-5) Among them, β -Ca₃(PO₄)₂-type compounds have prompted remarkable attention as phosphor host materials because of their versatile structural types and abundant occupation sites for activators $[13,14]$ $[13,14]$ $[13,14]$. There are five independent cationic crystallographic sites in β -Ca₃(PO₄)₂ host and the coordination number is as follows: Ca(1) is 7-coordinated, Ca(2) and Ca(3) is 8-coordinated, Ca(4) is 3 coordinated, and $Ca(5)$ is 6-coordinated $[15]$. A large number of *β*-Ca₃(PO₄)₂-type structure phosphors like $Ca_{10}M(PO_4)_7:Eu^{2+}$ (M=Li, Na and K) [16–[19](#page-6-0)], Ca_9R $(PO_4)_7:Eu^{2+}$ (R=Cr and Rare earth), $Ca_8MgRE(PO_4)_7:Eu^{2+}$ [20[–23](#page-6-1)], $Sr_sMgRE(PO_4)_7:Eu^{2+} (RE=Y, La, Gd, Sc)$ [24–[26](#page-6-2)] have been reported with favorable luminescence properties and versatile applications.

Inspired by the above knowledge, we designed a new yellow-emitting $Sr_9MgLi(PO_4)_7:Eu^{2+}$ phosphor for white light LEDs. In this study, luminescence properties including reflectance spectra and decay time, and thermal stability were discussed in detail. Moreover, a white pc-

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[Figure 1](#page-1-0) Powder XRD patterns of $Sr_{9-x}MgLi(PO_4)_7$: xEu^{2+} ($x=0.02$, 0.04, 0.08 and 0.014) samples (a), and Rietveld profile fitting refinement of the selected sample $Sr_{8.98}MgLi(PO₄)₇:0.02Eu²⁺ (b).$

WLED device was successfully fabricated by combining the blue-emitting BaMgAl₁₀O₁₇:Eu²⁺ and yellow-emitting $Sr₉MgLi(PO₄)$ ₇:Eu²⁺ phosphors with a 365 nm UV chip, and its electroluminescence properties was also reported. All of the results indicate that the $Sr₉MgLi(PO₄)₇:Eu²⁺$ phosphor can serve as a potential candidate for UV WLEDs.

EXPERIMENTAL SECTION

Materials and synthesis

 $Sr_{9-x}MgLi(PO_4)_7:xEu^{2+}$ (0.02≤*x*≤0.14) phosphors were synthesized *via* a high temperature solid-state route. The stoichiometric mixtures of $SrCO₃$ (99.9%), Li₂CO₃ (99.9%), (NH_4) ₂HPO₄ (99%), Mg(OH)₂·4MgCO₃·6H₂O (AR), $Eu₂O₃$ (99.99%) were ground together with a small amount of ethanol using an agate mortar and pestle until the mixtures were almost dry (25 min). Mixtures were shifted to the crucible and preheated at 800°C for 3 h, and then were sintered under a reducing atmosphere $(5\% \text{ H}_{2}/$ 95% N_2) at 1,300°C for 4 h. After firing, the sintered samples were cooled to room temperature and finely ground with a mortar for further characterization.

Characterization

The powder X-ray diffraction (XRD) data were collected on an X-ray diffractometer (Bruker D8 Advance) with Cu Kα radiation $(\lambda=1.5406 \text{ Å})$ operating at 40 kV and 40 mA. The morphology and microstructure of the samples were examined using a scanning electron microscope (SEM, JEOL JSM-6510A). The photographs of microcrystal particle were obtained by a Nikon $LV_{100}ND$ optical microscope. The photoluminescence (PL), photoluminescence excitation (PLE) and temperature dependent luminescence spectra were measured using a Hitachi F-4600 spectrometer equipped with a photomultiplier tube operating at 400 V and a 150 W Xe lamp as the excitation source. The diffuse reflectance spectra were measured on a Hitachi UH4150 ultraviolet-visible-near infrared spectrophotometer. The decay curves were recorded using a spectrophotometer (Edinburgh Instruments, FLS920) equipped with an *n*F900 flash lamp as the excitation source. The electroluminescence spectra, color temperature (CCT) and color-rendering index (R_a) of the asfabricated w-LEDs were measured using a UV-vis-near IR spectrophotocolorimeter (PMS-80, Everfine).

RESULTS AND DISCUSSION

Structure of Sr9MgLi(PO4)7:Eu2+ phosphor

The XRD patterns of the as-obtained Sr_{9−*x*}MgLi(PO₄)₇: xEu^{2+} phosphors with various Eu^{2+} doping concentration (*x*=0.02, 0.04, 0.08, 0.14) are depicted in [Fig. 1](#page-1-0)a. All of the diffraction peaks can be well-indexed to the standard card (PDF 47-1895) of $(Sr_{0.86}Mg_{0.14})_3(PO_4)_2$ phase, which indicates that Sr_{9-*x*}MgLi(PO₄)₇:*x*Eu²⁺ may be a pure phase iso-structural to $(Sr_{0.86}Mg_{0.14})_3(PO_4)_2$ with the space group of *R*3*c* [\[27\].](#page-6-3) According to the enlarged view around 30 degree, the XRD peaks of the series samples shift to larger angle with increasing of Eu^{2+} ions since the ionic radius of $Eu²⁺$ is smaller than that of $Sr²⁺$. TOPAS 4.2 program was used for the Rietveld profile fitting refinement of the selected sample to confirm the phase purity [\[28\].](#page-6-4) The Rietveld refinement results revealed that $Sr₉MgLi(PO₄)₇$: Eu²⁺ has a β-Ca₃(PO₄)₂-type structure with space group of *R*3*c*, and the lattice parameters are $a=b=10.608 \text{ Å}$, $c=$ 39.356 Å, $\alpha = \beta = 90^\circ$, $\gamma = 120^\circ$, *Z*=2, *V*=3,835.269 Å³. Unfortunately, we failed to solve the real crystal structure

[Figure 2](#page-2-0) Optical microscope photographs of the Sr₉MgLi(PO₄)₇:Eu²⁺microcrystal particles without (a) and with 365 nm UV excitation (b). SEM images of the Sr₉MgLi(PO₄)₇:Eu²⁺microcrystal particles (c) and the enlarged one particle (d), element mapping images of Sr (e), Mg (f), P (g), and O (h) for the selected $Sr_9MgLi(PO_4)_7:Eu^{2+}$ particle, and EDX spectrum and content of $Sr_9MgLi(PO_4)_7:Eu^{2+}$ (i).

because of the splitting at the strongest peak (the inset of [Fig. 1b](#page-1-0)). However, this flaw could not prevent the result on the β -Ca₃(PO₄)₂-type structure, and the further studies on the outstanding luminescent properties of $Sr₉MgLi$ $(PO₄)$ ₇:Eu²⁺ phosphors, which will be discussed later.

[Fig. 2](#page-2-0)a and b give the optical microscope photographs of $Sr₉MgLi(PO₄)₇:Eu²⁺$ particles without and with 365 nm UV light source excitation, respectively, and the bright yellow emission in [Fig. 2b](#page-2-0) also predicted the excellent luminescence properties of $Sr₉MgLi(PO₄)$ ₇:Eu²⁺ phosphors. [Fig. 2](#page-2-0)c and d demonstrate the SEM images of the $Sr₉MgLi(PO₄)₇:Eu²⁺$ particles with different scales, and it is found that the microcrystals have well crystallinity with smooth surfaces. The elemental mapping images in [Fig.](#page-2-0) [2e](#page-2-0)–h and the EDX spectrum in [Fig. 2i](#page-2-0) prove that the presence of Sr, Mg, P, O and Eu elements in $Sr₉MgLi$ $(PO₄)$ ₇:Eu²⁺ phosphor. However, Li element was not detected by the elemental mapping because of its small atomic mass. In addition, the detected mole ratio for Sr, Mg, P, O elements approximately equal to 9:0.93:5.7:27, which is close to the stoichiometric ratio of $Sr₉MgLi$ $(PO₄)₇$. Moreover, both the optical microscopy and SEM images in [Fig. 2](#page-2-0) show that the particle sizes of the asprepared samples were in the micrometer range with uniform size distribution, which also benefits the fabrication for white LEDs devices.

Luminescence properties of $Sr_9MgLi(PO_4)_7:Eu^{2+}$ **phosphors** The UV-vis diffuse reflectance spectra of undoped-Sr₉ $Mgli(PO₄)₇$ and $Sr₉$ _r $Mgli(PO₄)₇$ *:x*Eu²⁺ (*x*=0.02, 0.04, 0.08 and 0.14) samples are shown in [Fig. 3](#page-3-0)a. The decreasing reflectance of $Sr₉MgLi(PO₄)₇$ (black solid line) from 200 to 350 nm is ascribed to the host absorption. As a contrast, $Sr₉MgLi(PO₄)₇:Eu²⁺ shows a broad band from 250 to$ 450 nm attributing to the 4f-5d absorption of Eu^{2+} . The band gap of $Sr₉MgLi(PO₄)₇$ can be determined according to the Kubelka-Munk method and Tauc plot [\[29\].](#page-6-5) As displayed in the inset of [Fig. 3a](#page-3-0), the band gap is estimated to be 3.5 eV by extrapolating the linear portion to the photon energy axis.

[Fig. 3](#page-3-0)b and c show the PLE and PL spectra of Sr_{9−*x*}MgLi- $(PO_4)_7$: xEu^{2+} $(0.02 \le x \le 0.14)$ phosphors. As can be seen from the [Fig. 3b](#page-3-0), the PLE spectra display a broad absorption band from 250 to 450 nm corresponding to the diffuse reflectance spectra. The PL spectra excited by 365 nm light are shown in [Fig. 3c](#page-3-0). A broad emission band from 450 to 700 nm is indicative of electron transitions from 5d excited state to 4f ground state of Eu^{2+} . Furthermore, the PL spectra can be at least fitted into three bands using Gauss simulation (green dashed line) attributing to the different occupation of Eu^{2+} ions in *β*-Ca₃- $(PO₄)$ ₂-type phosphor. According to our previous report on $Ca_{10}M(PO_4)_7:Eu^{2+} (M = Li, Na \text{ and } K)$ phosphors [\[15\]](#page-5-7), $Eu²⁺$ ions will occupy parts of the cations' sites. Thus, we can deduce that Eu^{2+} may occupy the Sr1, Sr2 and Sr3 sites equally in $Sr₉MgLi(PO₄)₇:Eu²⁺ phosphory$ with the increase of the Eu^{2+} concentration, the emission intensity increases and reaches a maximum at *x*=0.04, and then decreases due to the concentration quenching. The concentration quenching occurs mainly because of the non-radiative energy transfers between Eu^{2+} ions within a certain distance. To have a better understanding of the concentration quenching mechanism, the critical distance (R_C) is evaluated using the following equation [\[30\]](#page-6-6):

$$
R_{\rm c} = 2 \left[\frac{3V}{4\pi X_{\rm c} N} \right],\tag{1}
$$

where X_c is the critical concentration of activator ions; N is the number of cations which can be substituted by the dopant in per unit cell; and *V* is the volume of the unit cell. In the case of $Sr_{9-x}MgLi(PO_4)_7$: xEu^{2+} phosphors, $X_C =$ 0.04, $V = 3,835.269 \text{ Å}^3$, $N = 18$. The value of R_C is cal-

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[Figure 3](#page-3-0) Absorption spectra of Sr₉−*xMgLi*(PO₄)₇:*x*Eu²⁺ (*x* = 0, 0.02, 0.04, 0.08, 0.14) and calculated energy gap by Tauc equation (a), PLE (b) and PL (c) spectra of $Sr_{9-x}MgLi(PO_4)$;*xEu*²⁺ (0.02 ≤ *x* ≤ 0.14) phosphors, and the relationship between log(*x*) and log(*I/x*) in the $Sr_{9-x}MgLi(PO_4)$;*xEu*²⁺ phosphors (d).

culated to be 21.67 Å. Thus, the energy transfer among Eu²⁺ ions in Sr_{9−*x*}MgLi(PO₄)₇: xEu^{2+} phosphors is not triggered by the exchange interaction type since the corresponding critical distance for exchange interaction is about 3–5 Å $[31]$. According to Dexter's theory, the type of electric multipolar interaction can be calculated by using the following formula [\[32\]](#page-6-8):

$$
I/x = K[1 + \beta(x)^{\theta/3}]^{-1},
$$
 (2)

where I is the integrated emission intensity, x is the activator concentration, K and β are constants for the same excitation condition for a given host lattice; *θ*=3 stands for energy transfer among the nearest neighbor ions, while θ =6, 8 and 10 stands for dipole-dipole, dipolequadrupole and quadrupole-quadrupole interactions, respectively. [Fig. 3d](#page-3-0) shows that the relationship of log(*I/x*) *vs.* log(*x*) is linear and the slope is -0.985 . Thus, the value of *θ* was calculated to be 2.955, which is approximately equal to 3, indicating that the energy transfer among the nearest neighbor ions dominates the concentration quenching of the Eu²⁺ emission in the Sr₉MgLi(PO₄)₇:Eu²⁺ phosphors [\[33\].](#page-6-9)

Fluorescence lifetime properties

[Fig. 4](#page-4-0) presents decay curves of Sr_{9−*x*}MgLi(PO₄)₇: xEu ²⁺ (0.02≤*x*≤0.14) phosphors under excitation at 365 nm, monitored at the peak of 560 nm at room temperature. The corresponding luminescent decay curves could be fitted using a bi-exponential temporal dependence according to Equation $3 \overline{9}$:

$$
I = A_1 \exp\left(-\frac{t}{\tau_1}\right) + A_2 \exp\left(-\frac{t}{\tau_2}\right),\tag{3}
$$

where *I* is the luminescence intensity, *t* is the time after

[Figure 4](#page-4-0) The decay curves of the Eu²⁺ emission in $Sr_{9-x}MgLi(PO_4)_7$: *xEu²⁺* (0.02≤*x*≤0.14) under excitation at 365 nm, monitored at 560 nm.

excitation, and τ_i (*i*=1, 2) is the decay time of the *i* component, A_1 and A_2 are constants. Using these parameters, the average decay time *t* can be calculated by the following formula:

$$
I = \left(A_1 \tau_1^2 + A_2 \tau_2^2\right) / \left(A_1 \tau_1 + A_2 \tau_2\right). \tag{4}
$$

With an increase in the Eu^{2+} concentration, the decay times gradually decrease from 2.768 to 1.886 μs, which directly demonstrates that energy transfer occurs among Eu^{2+} ions.

Thermal stability of $Sr_9MgLi(PO_4)_7:Eu^{2+}$ **phosphor**

Generally, thermal stability is an important character for phosphors applied in WLEDs. [Fig. 5a](#page-4-1) shows the temperature-dependent PL spectra and integrated emission intensity (the inset) of the $Sr_{8.96}MgLi(PO₄)₇:0.04Eu²⁺$ phosphor under 365 nm excitation. Obviously, the emission intensity decreases with an increase in temperature from 30°C to 300°C, and about 50% emission intensity remains when temperature raise up to 150°C. To explain the thermal behavior of $Sr_{8.96}MgLi(PO_4)_7:0.04Eu^{2+}$, the activation energy (ΔE) is obtained from the following formula [\[31\]](#page-6-7):

$$
I_T = \frac{I_0}{1 + A \exp\left(-\frac{\Delta E}{kT}\right)},\tag{5}
$$

where I_0 and I_T is the initial PL intensity of the phosphor at room and experimental temperature, *A* is a constant, *k* is the Boltzmann constant ($k = 8.617 \times 10^{-5}$ eV K⁻¹). By linear fitting the relationship of $\ln[(I_0/I_T)-1]$ *vs.* $1/kT$, the value of Δ*E* is about 0.282 eV, as shown in [Fig. 5b](#page-4-1). The high value of Δ*E* (compared with Chen's report [\[34\]](#page-6-10)) indicates that $Sr₉MgLi(PO₄)₇:Eu²⁺ phosphory$ potential yellow-emitting phosphor in UV chips based WLED applications.

Electroluminescence properties

The CIE chromaticity coordinates of $Sr₉MgLi(PO₄)₇:Eu²⁺$ phosphor and digital images upon 365 nm UV light ex-citation are shown in [Fig. 6](#page-5-9)a. $Sr_9MgLi(PO_4)_7:Eu^{2+}$ takes on homogeneous and bright yellow light, and the CIE coordinate is (0.518, 0.462). Furthermore, the internal quantum efficiency is measured to be 76%. All the above information indicates that $Sr_9MgLi(PO_4)_7:Eu^{2+}$ is suitable to be applied in UV chip pumped WLED. Based on this, we combined blue emitting $BaMgAl_{10}O_{17}:Eu^{2+}$ and yellow-

[Figure 5](#page-4-1) Temperature-dependent PL spectra of the selected S_{r_8} ₆₈MgLi(PO₄)₇:0.04Eu²⁺ phosphor, and inset shows the normalized emission intensity as a function of temperature ($\lambda_{ex}=365$ nm) (a), and Arrhenius fitting of the emission intensity of $Sr_{8.96}MgLi(PO_4)_{7.}0.04Eu^{2+}$ phosphor (b).

[Figure 6](#page-5-9) CIE coordinates and a digital photo of the selected Sr_{8.96}MgLi(PO₄)₇:0.04Eu²⁺ phosphor under 365 nm UV lamp (a), and electroluminescent spectrum of the fabricated WLED (b).

emitting $Sr_9MgLi(PO_4)_7:Eu^{2+}$ with a commercial UV LED chips (*λ*=365 nm) to realize white light emitting. The corresponding CRI, CCT, and CIE chromaticity coordinates of the WLED device were found to be 83, 5,612 K, and (0.324, 0.358), respectively, as shown in [Fig.](#page-5-9) [6b](#page-5-9). The WLED packaging results demonstrate that $Sr₉MgLi(PO₄)₇:Eu²⁺$ is a great potential candidate as UVexcitable white emitting phosphor.

CONCLUSIONS

In summary, new $Sr_9MgLi(PO_4)_7:Eu^{2+}$ phosphors have been successfully synthesized by the high temperature solid-state reaction method. $Sr₉MgLi(PO₄)₇:Eu²⁺ possess a$ *β*-Ca3(PO4)2-type structure with space group of *R*3*c*, and the lattice parameters are $a=b=10.608 \text{ Å}$, $c=39.356 \text{ Å}$, *α*=β=90°, γ=120°, Z=2, V=3,835.269 Å³. Upon 365 nm UV light excitation, the $Sr₉MgLi(PO₄)₇:Eu²⁺ phosphory$ exhibits a broad emission band from 450 to 700 nm, which are ascribed to the 5d-4f transitions of Eu^{2+} . The as-fabricated WLED device gives a high CRI ($R_a = 83$), indicating that $Sr_9MgLi(PO_4)_7:Eu^{2+}$ phosphor might be promising as a candidate for UV WLED lamp.

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近紫外芯片激发的白光LED用黄色荧光粉Sr₉MgLi(PO₄)₇:Eu²⁺

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摘要 发现新型白光LED用荧光粉一直是材料科学领域一项重要的挑战. 采用本课题组提出的矿物结构模型, 高效设计了稀土发光材料新 物相, 并解析、确定新物相的晶体结构. 本论文用黄色荧光粉Sr,MgLi(PO4)7:Eu²⁺设计合成了一种新型白光LED, 并对其结构和发光特性进 行了分析表征. 研究发现: 其物相结构是源于β-Ca₃(PO₄)2矿物模型. 在365 nm近紫外光激发下, Sr₉MgLi(PO₄)₇:Eu²⁺呈现出一个从450 nm到 700 nm的宽带发射. 通过把Sr₉MgLi(PO₄)₇:Eu²⁺黄粉与商用蓝粉BaMgAl₁₀O₁₇:Eu²⁺混合涂敷在蓝光芯片表面, 制作得到了白光LED器件. 测 试结果显示, LED的R_a, CCT, CIE值分别为83, 5612 K, (0.324, 0.358), 表明Sr₉MgLi(PO₄)₇:Eu²⁺可作为白光LED用黄光荧光粉.