



# Reduced-sized monolayer carbon nitride nanosheets for highly improved photoresponse for cell imaging and photocatalysis

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**ABSTRACT** Two-dimensional graphitic carbon nitride (g-C<sub>3</sub>N<sub>4</sub>) nanosheets (GCNNs) have been considered as an attractive metal-free semiconductor because of their superior catalytic, optical, and electronic properties. However, it is still challenging to prepare monolayer GCNNs with a reduced lateral size in nanoscale. Herein, a highly efficient ultrasonic technique was used to prepare nanosized monolayer graphitic carbon nitride nanosheets (NMGCNs) with a thickness of around 0.6 nm and an average lateral size of about 55 nm. With a reduced lateral size yet monolayer thickness, NMGCNs show unique photo-responsive properties as compared to both large-sized GCNNs and GCN quantum dots. A dispersion of NMGCNs in water has good stability and exhibits strong blue fluorescence with a high quantum yield of 32%, showing good biocompatibility for cell imaging. Besides, compared to the multilayer GCNNs, NMGCNs show a highly improved photocatalysis under visible light irradiation. Overall, NMGCNs, characterized with monolayer and nanosized lateral dimension, fill the gap between large size (very high aspect ratio) and quantum dot-like counterparts, and show great potential applications as sensors, photo-related and electronic devices.

**Keywords:** graphitic carbon nitrides, monolayer nanosheets, photocatalysis, fluorescence, cell imaging

## INTRODUCTION

Due to the fascinating physicochemical and electronic properties arising from the small dimensions and quantum confinement effects (QCE) [1–3], ultrathin nanosheets have been the focus of substantial research in the light of fundamental studies and their broad applications in various fields, including optoelectronic devices [4], energy

conversion and storage [5], sensors [6–9], biomedicine [10], and catalysis [11–15]. In particular, polymeric graphitic carbon nitride nanosheets (GCNNs), an n-type semiconductor, have been receiving considerable attention due to their many intriguing advantages such as the fact that they contain no metal, and nontoxicity, easy availability, sensitive photo-response, excellent chemical and thermal stability, and unique electronic structures [16–19].

Compared with bulk graphitic carbon nitride (GCN), the remarkable optical and catalytic properties of the GCNNs are mainly associated with the QCE and their larger specific surface area (SSA) [18,20]. On one hand, the wider energy bandgap arising from the QCE accompanied by the different photo-physical behavior of photogenerated electron-hole pairs is very helpful for improving their redox performance. On the other hand, the larger SSA of the ultrathin GCNNs provides abundant exposed surfaces, numerous active sites, and a short transfer length for increasing the separation efficiency of interface photoexcited charge carriers, thus showing a much improved photocatalytic performance. In addition, it has been demonstrated that reduced-sized GCNNs and GCN quantum dots with much smaller lateral dimension can be used for bio-imaging due to their distinct fluorescence with low photo-bleaching and low cytotoxicity [21]. However, multilayer GCNNs reported so far have a low fluorescence quantum yield (less than 20%) [8,9,22], and GCN quantum dots show ignorable photocatalytic activity under visible light irradiation [23]. It is believed that nanosized monolayer graphitic carbon nitride nanosheets (NMGCNs)

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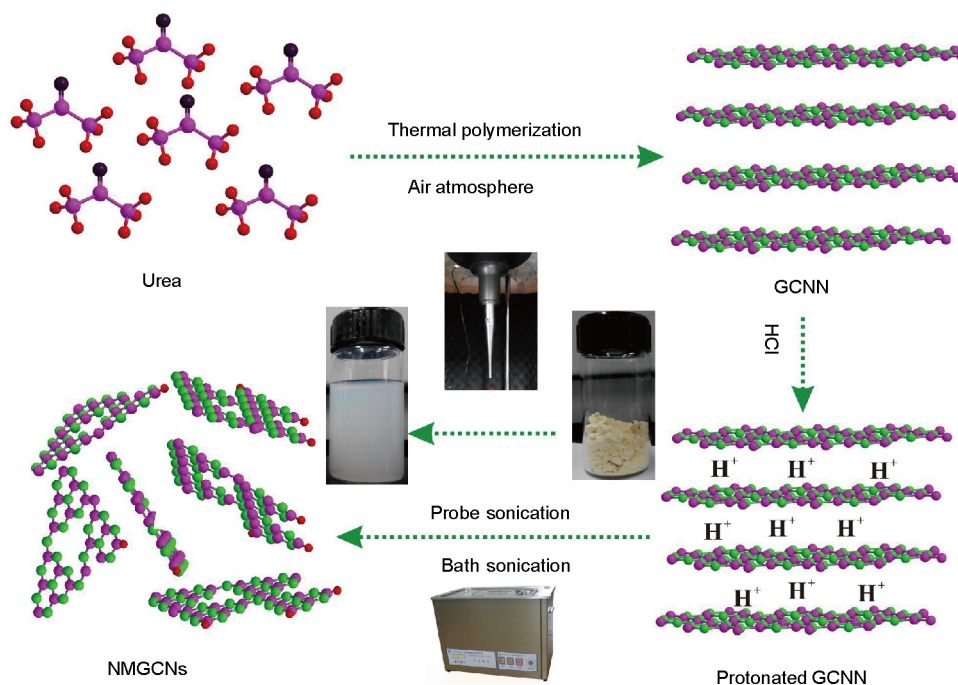
would be highly photo-responsive for both photocatalysis and fluorescence.

Thus far, GCNNs prepared with various techniques, such as mechanical milling [24,25], chemical oxidation thermal or etching [20,26,27], physical and liquid exfoliation [22,28,29], and electrochemical synthesis [30] usually show wide distributions in both the number of layers and lateral size. In particular, due to many advantages of easy operation, low-cost, environmental benignity, and no reason for interfacial defects, liquid-phase exfoliation, usually in conjunction with ultrasonic treatment has been intensively explored for producing ultrathin GCNNs by delaminating bulk GCN [28,31,32]. However, the obtained GCNNs usually have a thickness larger than 2 nm ( $> 5$  layers) [30,33]. It is still a challenge to prepare atom-thin GCNN with a reduced lateral size ( $< 100$  nm) although several research groups have recently reported the preparation of monolayer GCNN that has a lateral dimension larger than  $1.0 \mu\text{m}$  [13,34]. Also, the liquid-phase exfoliation technique is still limited due to its low mass yield (less than 10%) and long time required. Previous results also suggest that protonation before sonication not only significantly enhances the product yield of GCNN but also provides a GCNN with better dispersibility in water at a high concentration since the bridged nitrogen of GCN serves as a unique adsorption site that facilitates water intercalation for exfoliation [7]. Furthermore, it has been reported that

probe sonication treatment by a high-pressure cell cracker is efficient in breaking a large nanosheet into smaller ones without changing the chemical structure [35]. Here we present a preparation technique for NMGCNs that have an average lateral size of 55 nm by probe sonication combined with bath sonication using protonated GCNN as the starting material. These NMGCNs show uniqueness to their counterparts, large size GCNNs and GCN quantum dots, and simultaneously possess highly photo-response for cell imaging and photocatalysis.

## RESULTS AND DISCUSSION

The preparation process of the NMGCNs is illustrated in Fig. 1. This synthesis strategy combines bath and probe sonications to exfoliate and tailor multilayer GCNNs into NMGCNs. GCNNs with an average thickness of about 20 nm were first prepared by the modified thermal poly-condensation of urea in air (Fig. S1, Supplementary information) [36]. It is noted that the high SSA of the GCNNs ( $\sim 102 \text{ m}^2 \text{ g}^{-1}$ ) is conducive to ultrasonic exfoliation since there is a larger contact area with water (Fig. S2). Multilayer GCNNs were then produced by bath sonication in water using proton-functionalized GCNNs as the starting material. Finally, NMGCNs were obtained with a yield of  $\sim 30\%$  after a strong physical cleavage of the multilayer GCNNs by probe sonication in a conical tube (see Supplementary information). It should be noted that the probe sonication yields a

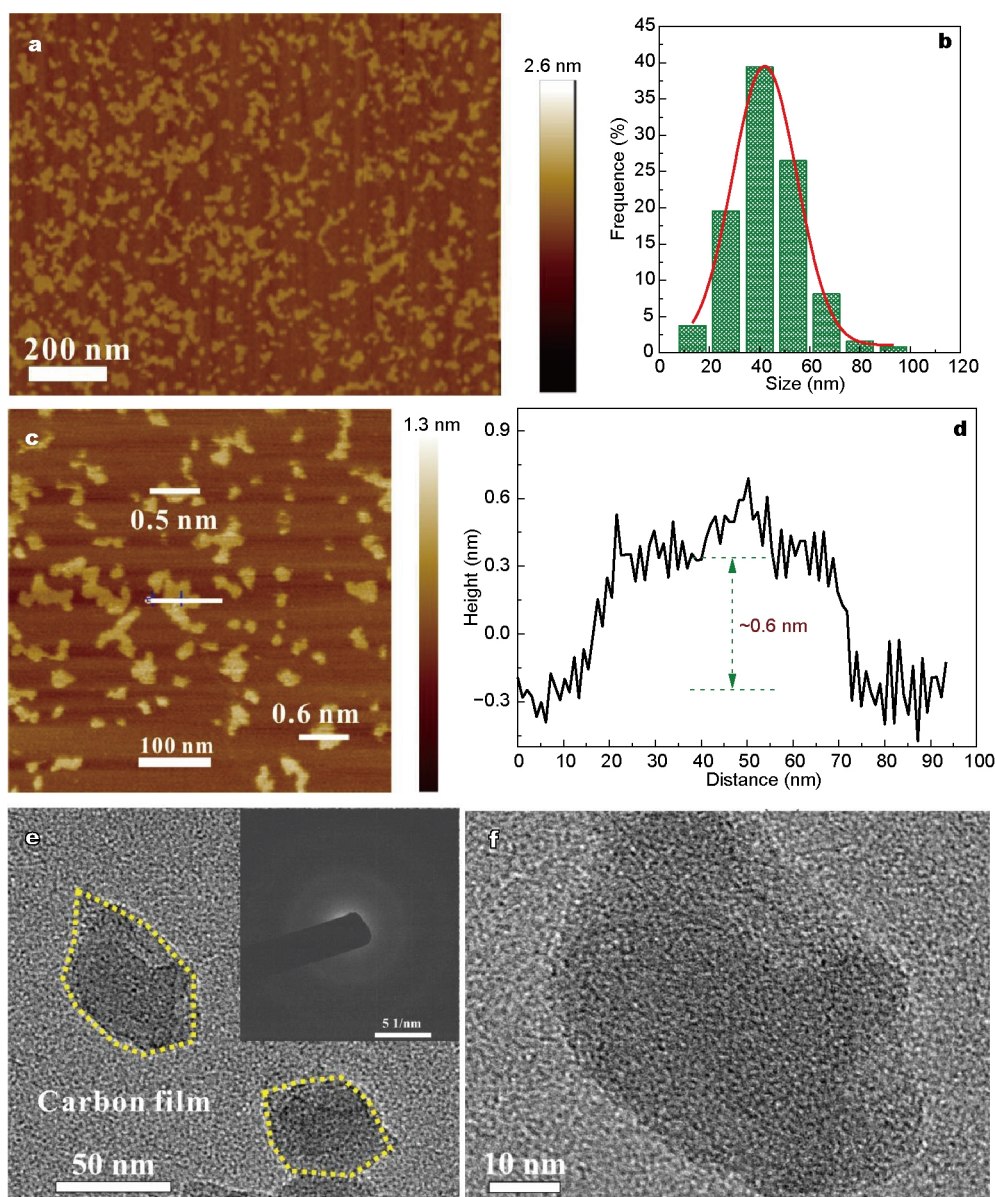


**Figure 1** Illustration for the preparation of NMGCNs.

reduced lateral size for NMGCNs compared with their GCNN counterpart and dot-like NMGCNs with an average size of 3 nm can be also produced by increasing the sonication time up to 9 h (Fig. S3).

Atomic force microscopy (AFM) was first carried out to observe the morphology of the NMGCNs. A typical low-magnification AFM image (Fig. 2a) clearly reveals that the NMGCNs are irregularly shaped nanosheets. Statistical analysis by the ImageJ software reveals that the average lateral size of the NMGCNs is about 55 nm (Fig. 2b). As demonstrated in Fig. 2c, AFM images of the NMGCNs also

reveal freestanding nanosheets with a flat surface. Their average thickness of about 0.60 nm, determined from the fitted height profile measured in AFM topography, verifies their monolayer structure considering the theoretical inter-layer distance of GCN is around 0.35 nm (Fig. 2d). Transmission electron microscopy (TEM) was further conducted to examine the dimensions of the NMGCNs. The low contrast TEM image shown in Fig. 2e confirms their ultrathin nature. The corresponding selected area electron diffraction (SAED) patterns show obscure rings, demonstrating the disordered structures in the NMGCN. High-resolution

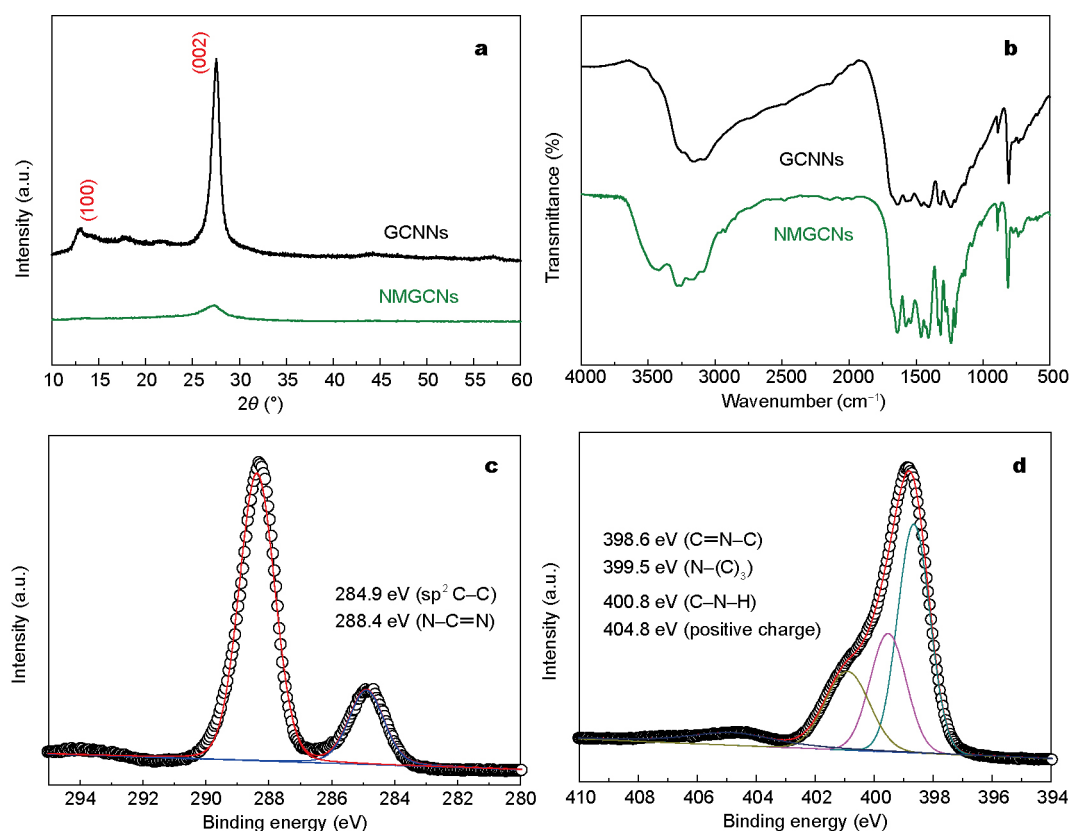


**Figure 2** (a) Low-magnification AFM image, (b) the corresponding lateral size distribution, (c) high-magnification AFM image, (d) the corresponding height profile, and (e) TEM and (f) HRTEM images of the NMGCNs. The inset in (e) is the corresponding SAED pattern of the NMGCNs.

TEM image (HRTEM) also reveals that there is no long-range ordering and continuous lattice fringe in the NMGCNs (Fig. 2f), which is in agreement with the SAED and X-ray diffraction (XRD) results. Both AFM and TEM results confirm the small lateral size and monolayer structure of the NMGCNs.

The layered structure of the NMGCNs was also studied by XRD. Compared with GCNNs (Fig. 3a), the (002) peak at about  $27.5^\circ$  is much weaker and broader, and the (100) peak at  $13.2^\circ$  has disappeared, indicating that the stacking of the conjugated aromatic systems and the in-plane repeat units of GCN are obviously disrupted after the exfoliation [20,26,37]. A Fourier transform infrared (FTIR) spectrum measurement was conducted to verify the chemical structure of the NMGCNs. As shown in Fig. 3b, the FTIR spectra of both GCNN and NMGCN show a sharp peak at  $808\text{ cm}^{-1}$ , five strong peaks at  $1240$ ,  $1320$ ,  $1400$ ,  $1460$ , and  $1550\text{ cm}^{-1}$ , and a board peak ranging from  $3000$  to  $3500\text{ cm}^{-1}$ , respectively corresponding to the typical breathing vibration of the *s*-triazine ring, the characteristic stretching modes of CN heterocycles, and stretching of N-H and O-H bonds [26,29,38]. The FTIR result indicates that the

NMGCNs retain the basic structure of the GCN. Notably, all the FTIR peaks of NMGCNs are much sharper than those of GCNNs, suggesting a larger stacking distance and shorter-range order of the in-plane structure in the NMGCNs. This is in agreement with the above XRD results. Survey X-ray photoelectron spectroscopy (XPS) confirms the presence of C, N, and O in the NMGCNs with a C/N molar ratio of about 0.76 (Fig. S4), indicating that the pristine backbone of GCN is retained in the NMGCNs. The oxygen is ascribed to a small amount of water adsorbed on the surface of the NMGCNs (Fig. S5). The high resolution C 1s spectrum shown in Fig. 3c reveals the existence of two types of C species including the C-C at  $284.9\text{ eV}$  and the C=N at  $288.4\text{ eV}$  [39-41]. The fitted high resolution N 1s spectrum shows the presence of four types of N species, containing C=N-C at  $398.6\text{ eV}$ , N-(C)<sub>3</sub> at  $399.5\text{ eV}$ , C-N-H at  $400.8\text{ eV}$ , and the positive charge of heptazine at  $404.8\text{ eV}$  (Fig. 3d) [39,42,43]. Both C 1s and N 1s spectra firmly demonstrate the presence of basic tri-*s*-triazine units in NMGCNs, in agreement with the FTIR results. The retention of the basic heptazine structure is very important for keeping the unique photo-responsive

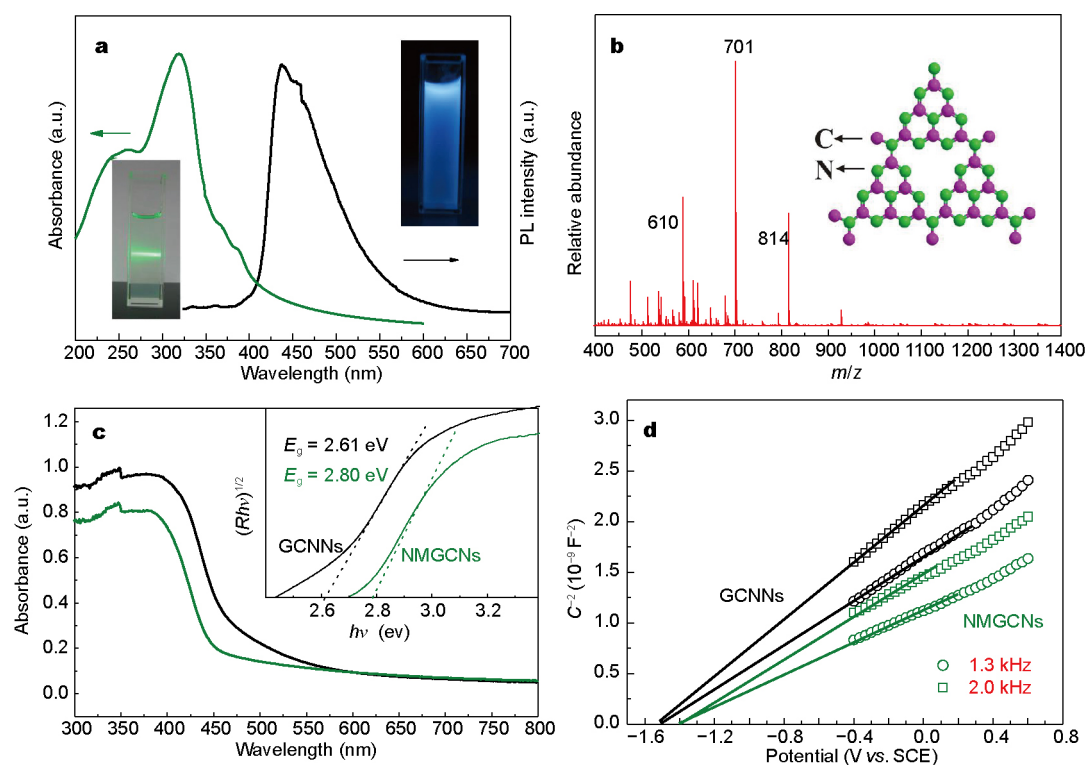


**Figure 3** (a) XRD patterns and (b) FTIR spectra of GCNNs and NMGCNs. (c) C 1s and (d) N 1s XPS spectra of NMGCN.

properties of the NMGCNs [16,39].

The as-prepared NMGCNs are well dispersed in water. A typical Tyndall effect observed in the aqueous solution with a green laser beam indicates a colloidal nature of the NMGCN dispersion (inset in Fig. 4a). The Zeta-potential of the NMGCNs dispersion in water is determined to be about 31 mV. The UV-vis absorption spectrum of the aqueous solution of NMGCNs shows two peaks centered at about 250 and 300 nm, which are ascribed to the absorption from the  $\pi$ - $\pi^*$  of C=N and CN heterocycles, respectively. This is also supported by mass spectrometry (MS) measurements. As shown in Fig. 4b, the MS of the NMGCN dispersion shows a dominant signal at 701 mol  $g^{-1}$ , which is consistent with the relative molecular weight of the basic heptazine in GCN (inset in Fig. 4b). It must be mentioned that, as a result of the positive Zeta-potential (+32.6 mV) and good stability of NMGCN dispersion, the absorption spectrum shows no obvious change after keeping the dispersion for six months at room temperature (Fig. S6). Moreover, the aqueous solution of NMGCNs emits bright blue fluorescence under UV light illumination (inset in Fig. 4a). As expected, a strong emission peak centered at about 432 nm was detected in the photoluminescence (PL)

emission spectrum (Fig. 4a). The fluorescence intensity of the NMGCNs dispersion did not change significantly after continuous excitation with UV light for six hours and keeping it for six months, indicating that it has good structural stability and photostability (Fig. S7). Using quinine sulfate as the standard sample, the fluorescence quantum yield is determined to be about 32%, which is much higher than that of ultrathin GCNNs (usually less than 20%) [22], and similar to the values for the GCN quantum dots [41,44]. Moreover, the maximum PL emission position is independent of the wavelength of excitation light (Fig. S8), suggesting that the PL emission is ascribed to the intrinsic emission of the NMGCNs. To confirm this assumption, we estimated the bandgap of the NMGCNs by diffuse reflectance spectroscopy (DRS) measurements. As shown in Fig. 4c, the absorption edge of the NMGCN powder shows an obvious blue shift to 440 nm compared to that of the pristine GCNNs (475 nm). It was observed that the absorption edge of NMGCNs was close to the wavelength of the maximum PL emission. Accordingly, the energy band gaps of NMGCNs and GCNNs are respectively estimated to be 2.80 and 2.61 eV from the Tauc plots based on the Kubelka-Munk theory (inset in Fig. 4c). The wider bandgap of NMGCNs is ascri-



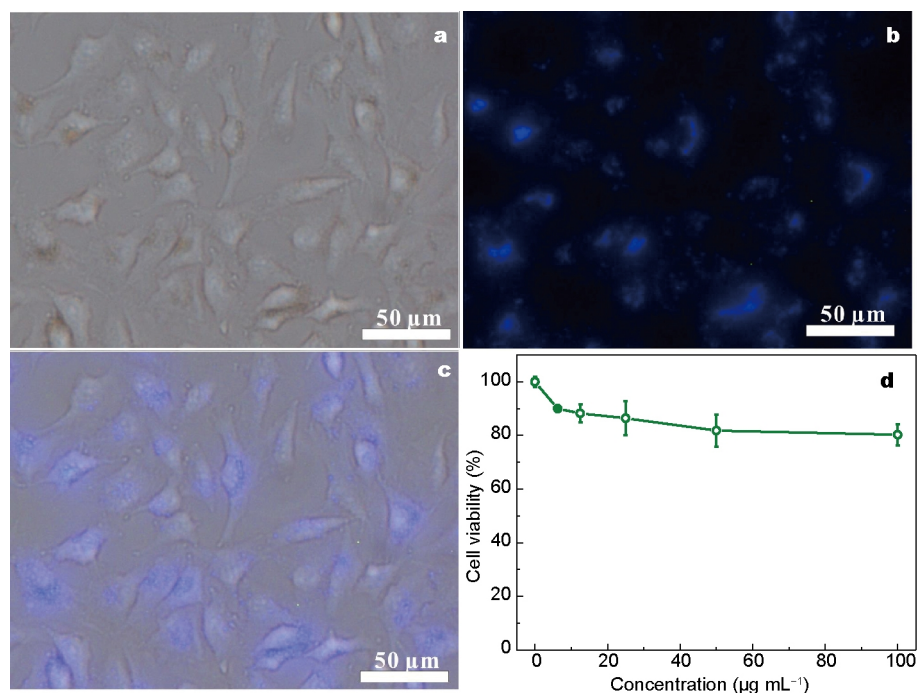
**Figure 4** (a) UV-visible absorption spectra, the Tyndall effect, the fluorescence image under UV light illumination, and (b) mass spectrum of the NMGCNs dispersed in water. (c) UV-visible DRS, and (d) Mott-Schottky plots of GCNNs and NMGCNs. The inset in (b) is the heptazine structure of carbon nitrides.

bed to the QCE arising from the small lateral size and monolayer structure [37]. In order to verify this deduction, we tested the Mott-Schottky plots by electrochemical method in detail. As compared in Fig. 4d, both NMGCNs and GCNNs exhibit positively sloping Mott-Schottky plots for two frequencies (1.3 and 2.0 kHz), suggesting that the NMGCN is still a kind of n-type semiconductor [29]. Furthermore, the derived flat-band potentials for GCNNs and NMGCNs are respectively about 1.50 and 1.39 V *versus* the saturated calomel electrode (SCE). Accordingly, the GCNN and NMGCN valence bands are about 1.11 and 1.41 V *versus* the SCE based on their bandgap values obtained by DRS (Fig. S9) [29]. It is found that both the valence and conduction bands of NMGCNs and GCNNs are significantly different, confirming that the QCE greatly influences the bandgap position. Remarkably, the NMGCNs are still thermodynamically suitable for achieving overall water splitting by photocatalysis since the valence and conduction bands straddle the water splitting redox potentials.

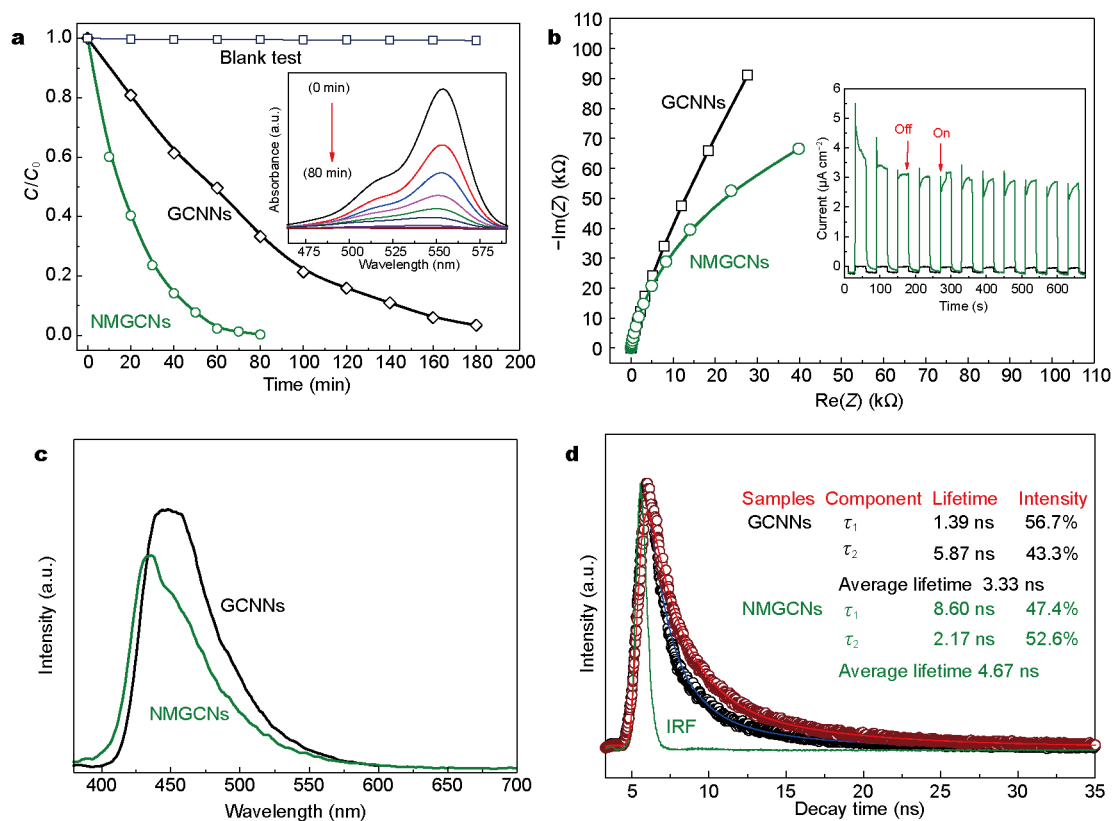
As mentioned above, the resulting NMGCNs have superior fluorescence properties, good dispersibility in water, and small lateral dimensions, enabling them to be used as a potential fluorescent marker for cell imaging. In this regard, a NMGCN aqueous solution was used to incubate A549 cells according to our previous method [45]. As

shown in Fig. 5a, no fluorescence was observed in the bright-field image of the cells; however, they show bright blue fluorescence under UV light irradiation (Fig. 5b). A merging of the bright and dark images under UV light clearly shows that the blue fluorescence is located in the cytoplasm of the A549 cells (Fig. 5c), suggesting that the NMGCNs are able to enter into the cytoplasm of the cells. A standard MTT assay with A549 cells was carried out to evaluate the cytotoxicity of the NMGCNs [45]. As shown in Fig. 5d, more than 80% of the cells were still alive after incubation with the NMGCNs at concentrations ranging from 5 to 100  $\mu\text{g mL}^{-1}$  for 48 h, demonstrating the good biocompatibility of the NMGCNs. Good fluorescence properties and low cytotoxicity enable the NMGCNs to act as a metal-free bioimaging marker.

The NMGCN that has a small lateral size and a monolayer structure are also believed to be very useful for the application in photocatalysis since there are more exposed active sites and edges in contact with the reactant. The visible light driven ( $\lambda \geq 400$  nm) decomposition of a Rhodamine B (RhB) aqueous solution was carried out to compare the photocatalytic activities of the NMGCNs and GCNNs in the same conditions. As shown in Fig. 6a, the blank test indicates that photo-degradation of RhB is hardly seen in the absence of any photocatalysts. With the NMGCNs, the absorbance of RhB decreased quickly under visible light



**Figure 5** (a) Bright field image, (b) fluorescence image under UV light, and (c) the corresponding overlay images of A549 cells incubated with NMGCNs for 6 h. (d) Viability of A549 cells incubated with different concentrations of NMGCNs for 48 h.



**Figure 6** (a) Photocatalytic degradation curves, (b) EIS plots and photocurrent curves under visible light irradiation, (c) PL emission, and (d) the time-resolved fluorescence decay spectra of the NMGCN powder. The inset in (a) is the UV-Vis absorption spectra of RhB after degradation by NMGCNs for different time.

irradiation (inset in Fig. 6a). The time dependence of RhB photo-degradation reveals that almost 98% of the RhB aqueous solution was degraded over NMGCNs and GCNNs after 80 and 180 minutes, respectively. It is noted that, agreeing with previous report, the GCN quantum dots also show negligible photocatalytic activity under visible light, which is attributed to their ignorable photo-response under visible light [23]. The rate constants, quantitatively estimated by the kinetic principle of a pseudo first-order reaction are respectively  $1.10$  and  $3.52 \text{ h}^{-1}$  for the GCNNs and NMGCNs (Fig. S10). Generally, the photocatalytic activity of the NMGCNs is 3.3 times higher than that of the GCNNs.

We conducted photochemical measurements to study the photo-response behaviors of the NMGCNs and GCNNs. Compared with the GCNNs, the NMGCN shows a much smaller semicircle radius of the electrochemical impedance spectroscopy (EIS) plot, suggesting the smaller electronic resistance at the interface of the NMGCN/electrolyte. The time-related photocurrent was also measured to study the photo-response behaviors of the NMGCNs and GCNNs.

As compared in Fig. 6b, both GCNN and NMGCN electrodes show that the photocurrent rapidly increased when the light was turned on and then decreased quickly after turning it off. However, the maximum photocurrent of the NMGCNs ( $\sim 3.2\text{--}4.0 \mu\text{A cm}^{-2}$ ) is up to 20 times higher than that of GCNNs ( $\sim 0.18\text{--}0.2 \mu\text{A cm}^{-2}$ ), confirming the much better photo-response of the NMGCNs. Moreover, the process can be repeated ten times with no significant difference, suggesting the good stability of NMGCNs under visible light irradiation. The better photo-response of the NMGCNs is attributed to the improved separation efficiency of photoexcited charge carriers. As compared in Fig. 6c, the NMGCN powder shows a lower PL emission intensity than the GCNN case, suggesting the lower charge recombination rate in the NMGCNs [14,46,47]. A blue shift of the maximum emission peak was also observed in the PL spectrum of the NMGCN powder, which was in agreement with the above DRS results. Furthermore, time-resolved fluorescence spectroscopy reveals that both NMGCNs and GCNNs show a short ( $\tau_1$ ) and a long ( $\tau_2$ ) lifetime of charge carriers with different proportions (Fig.

6d). Accordingly, the average PL lifetimes calculated by a double exponential fit are respectively 4.67 and 3.33 ns for the NMGCNs and GCNNs (Fig. 6d). The much increased lifetime of charge carriers in the NMGCNs is due to the smaller lateral size and monolayer structure, which result in reduced conjugation system with more exposed edges. Furthermore, the  $N_2$  adsorption measurements (Fig. S2) show that the SSA and pore volume of the NMGCN powder are about  $220 \text{ m}^2 \text{ g}^{-1}$  and  $1.10 \text{ cm}^3 \text{ g}^{-1}$ , respectively. Both values are about two times of those for GCNNs ( $102 \text{ m}^2 \text{ g}^{-1}$  and  $0.55 \text{ cm}^3 \text{ g}^{-1}$ ), confirming the presence of more photocatalytic active sites in the NMGCNs. This is further confirmed by the ferromagnetic measurements because theoretical and experimental studies suggest that the ultrathin GCNNs show intrinsic ferromagnetism at room temperature due to the presence of hydrogen dangling bonds [28,48]. As expected, the  $M$ - $H$  curves of NMGCNs and GCNNs show an obvious hysteresis loop with saturation magnetization (Fig. S11). However, the saturation magnetization ( $M_s$ ) value of the NMGCNs reaches  $0.04 \text{ emu g}^{-1}$ , which is about five times larger than that for the GCNNs ( $0.008 \text{ emu g}^{-1}$ ) and also much higher than previously reported values for the ultrathin GCNNs [28,48]. Notably, this is the highest  $M_s$  value among the previous reports for GCN. Such a high  $M_s$  of the NMGCNs is ascribed to the nanosized and monolayer structures, inducing abundant hydrogen dangling bonds and a higher spin density distribution and magnetic moment [28]. The intrinsic room temperature ferromagnetism with a high  $M_s$  also makes the metal-free NMGCNs a promising nanomaterial for developing portable electronic and spintronic devices. In a word, the good photocatalytic performance and fluorescence of the NMGCNs are attributed to their highly improved photo-response, resulting from the unique monolayer and reduced-sized structures.

## CONCLUSIONS

Nanosized monolayer carbon nitride nanosheets with a lateral size of about 55 nm and a thickness of 0.6 nm were prepared by using an easy yet efficient liquid-phase exfoliation method. As expected, the NMGCNs show excellent photo-responsive properties for both photocatalysis and fluorescence. As a result of the quantum size effect due to the nanosized and monolayer structure, a water dispersion of NMGCNs exhibits strong blue fluorescence with a high quantum yield of 32% under UV light, which is applicable as a good fluorescent marker for cell imaging. With a lateral size apparently larger than GCN quantum dots, NMGCNs also show remarkable photocatalytic

performance under visible light irradiation. Therefore, with combined properties of sheet-like GCNNs and GCN quantum dots, NMGCNs show potential applications as sensors, energy storage and conversion devices, and electronic devices. This study also paves the way, towards some special applications, for the development of other mono-layered or ultrathin nanosheets with small lateral size, which fills the gap between high aspect ratio 2D materials and quantum dot-like counterparts.

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## 具有显著增强光响应的小尺寸单层石墨相氮化碳纳米片用于细胞成像及光催化

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**摘要** 不含金属的二维石墨相氮化碳纳米片由于具有优异的催化、光学及电学性能而受到研究者的广泛关注. 然而制备纳米级尺寸的单层石墨相氮化碳纳米片仍然存在挑战. 本文采用一种高效超声方法制备了横向尺寸约为55 nm, 厚度约为0.6 nm的单层石墨相氮化碳纳米片(NMGCNs). 由于其纳米级尺寸及单层片状结构, NMGCNs表现出与大尺寸纳米片和量子点显著不同的光响应特性. NMGCNs的水分散溶液具有良好的稳定性和优异的荧光性能, 荧光量子产率可达32%, 所以可用于细胞荧光成像. 此外, NMGCNs表现出比多层石墨相氮化碳纳米片更优异的可见光催化性能. 独特的小尺寸及单层超薄结构使得NMGCNs在传感器和光电子等领域都具有潜在应用前景.