



# Effect of $B_2O_3$ on the Melting Temperature and Viscosity of $CaO-SiO_2-MgO-Al_2O_3-TiO_2-Cr_2O_3$ Slag

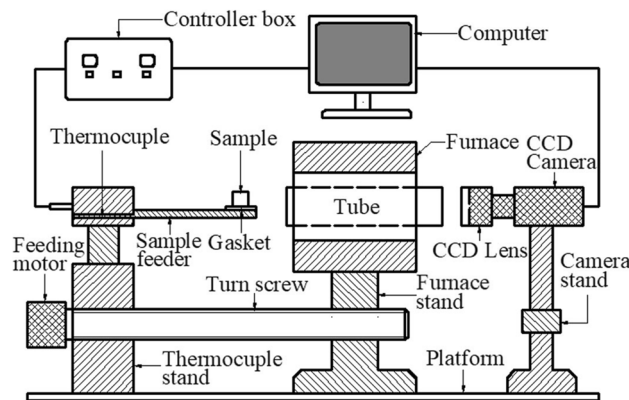
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Received: 18 April 2021 / Accepted: 18 July 2021 / Published online: 28 July 2021  
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## Abstract

Fluorite is widely employed as a fluxing agent in metallurgical processes, inevitably leading to severe pollution. In this study, to promote the development of sustainable metallurgy,  $B_2O_3$  is used as a fluxing agent of  $CaO-SiO_2-MgO-Al_2O_3-TiO_2-Cr_2O_3$  slag, acting as a replacement for  $CaF_2$ . The effect of  $B_2O_3$  on the melting temperature and viscosity of  $CaO-SiO_2-MgO-Al_2O_3-TiO_2-Cr_2O_3$  slag was investigated using a melting-point and melting-rate measurement instrument and a melting physical property comprehensive measurement instrument. Scanning electron microscopy was used to observe the microstructure of the precipitated phase, and X-ray diffraction was used to identify its phase composition. The results indicated that both the melting temperature and viscosity decreased with increasing  $B_2O_3$  content. Moreover,  $B_2O_3$  had a significant influence on the morphology of the precipitated phase, particularly for the shape and size of the precipitated particles. Accordingly, with the increase in the  $B_2O_3$  content, the composition of the precipitated phase and the corresponding diffraction peak intensities changed. The results of this study provide a theoretical and technical basis for the comprehensive utilization of Cr-containing high-titanium melting slag.

## Graphical Abstract



**Keywords**  $B_2O_3$  · Cr-containing high-titanium melting slag · Melting temperature · Viscosity · Phase composition

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## Introduction

Vanadium titanomagnetite (VTM) is a multi-component mineral that contains Fe, Ti, V, and various rare metals. In addition to Fe, Ti, and V, Hongge VTM (HVTM) found in the deposits in the Panzhuhua-Xichang area also has a high chromium content and is the largest VTM resource in China [1, 2]. Currently, two processes are mainly used for

the utilization of HVTM, namely, the blast furnace (BF) process and coal-based direct reduction process. However, for both the BF and coal-based direct reduction processes, the recovery rates of titanium, vanadium, and chromium are still low. In addition, it should be noted that it is difficult to deal with BF slag while the coal-based direct reduction process has a relatively low efficiency due to its high energy consumption and high operating temperature [3–5]. None of the above techniques can be easily implemented in industrial production and commercial use, hindering the development of sustainable metallurgy.

Therefore, for efficient HVTM utilization, a novel and sustainable smelting process has been proposed by our laboratory that obtains a significant increase in the recovery rates of valuable elements. In this process, HVTMs were first pelletized and oxidized by roasting (HVTMP) and then were reduced in a shaft furnace. Subsequently, the reduced HVTMPs were separated by melting for the comprehensive recovery of iron, titanium, vanadium, and chromium [6]. It is clear that the melting separation is an essential procedure in this novel and sustainable smelting process. It is known that in the melting separation process, the fluidity of the titanium slag at high temperature can affect the separation of iron and slag, thus, affecting the efficiency of the subsequent titanium extraction. Moreover, the extraction efficiency is closely related to the mineral composition and the microstructure of the slag which depend on the chemical composition, initial melting state, and cooling conditions. Therefore, to achieve high extraction efficiency of titanium, it is necessary to improve the fluidity of the slag.

In the melting separation process,  $\text{CaF}_2$  and  $\text{B}_2\text{O}_3$  are commonly used as additives due to their remarkable effect on the fluidity of the slag.  $\text{CaF}_2$  breaks up into small particles at high temperature and then melts quickly, decreasing the melting temperature and viscosity of slag. Although  $\text{CaF}_2$  plays an important role in the metallurgical industry, its use leads to significant environmental pollution that is inconsistent with the development of sustainable metallurgy [7–10]. In the past decades, many studies have proposed to replace  $\text{CaF}_2$  with  $\text{B}_2\text{O}_3$  that not only avoids the disadvantages of  $\text{CaF}_2$  but also achieves a better metallurgical effect than  $\text{CaF}_2$ . The acidity of  $\text{B}_2\text{O}_3$  is clearly stronger than that of  $\text{SiO}_2$ . Therefore, the  $\text{CaO}$  in the slag reacts preferentially with  $\text{B}_2\text{O}_3$  to form  $n\text{CaO}\cdot\text{B}_2\text{O}_3$  with a low melting point and then promotes the melting of lime. At the same time, the slag viscosity decreases with increasing  $\text{B}_2\text{O}_3$  content. This improves the diffusion and mass transfer in the melting separation process, which is beneficial for the slag system and the melting separation process [9, 11–18].

The melting temperature and viscosity are crucial physical properties for the fluidity of slag. The effect of  $\text{B}_2\text{O}_3$  on the slag properties has been examined in previous studies [19–25]. Wang et al. investigated the influence of  $\text{B}_2\text{O}_3$  on

the melting temperature and viscosity of refining flux. It was demonstrated that  $\text{B}_2\text{O}_3$  was beneficial for decreasing the melting temperature and viscosity [20]. Ren et al. investigated the effect of  $\text{B}_2\text{O}_3$  on the viscosity of Ti-bearing blast furnace slag and showed that the addition of  $\text{B}_2\text{O}_3$  decreased the viscosity and improved the fluidity of the slag [21]. Li et al. found that for the  $\text{CaO-SiO}_2\text{-Al}_2\text{O}_3\text{-Cr}_2\text{O}_3$  slag,  $\text{B}_2\text{O}_3$  behaved as a network former to decrease the number of non-bridging oxygen atoms per Si atom (NBO/Si). Unfortunately, the addition of  $\text{B}_2\text{O}_3$  also resulted in the formation of low-melting-point eutectics and decreased the structural strength of the obtained metal [23]. To summarize, the studies performed to date have mainly focused on the effect of  $\text{B}_2\text{O}_3$  on the blast furnace slag, mold flux, or refining slag. However, there have been few reports on the effect of  $\text{B}_2\text{O}_3$  on the properties of high-titanium melting slag, particularly in the presence of chromium. In the Cr-containing high-titanium melting slag ( $\text{CaO-SiO}_2\text{-MgO-Al}_2\text{O}_3\text{-TiO}_2\text{-Cr}_2\text{O}_3$ ), the  $\text{TiO}_2$  content is higher than that in the Ti-bearing blast furnace slag, and considerable utilization can be obtained if the titanium can be extracted effectively. Therefore, it is urgently necessary to improve the fluidity of the Cr-containing high-titanium melting slag.

In this study,  $\text{B}_2\text{O}_3$  was introduced as a fluxing agent into the Cr-containing high-titanium melting slag, and the effect of  $\text{B}_2\text{O}_3$  on the melting temperature and viscosity of this slag was investigated. In particular, the evolution of the phase composition and structure was examined. This study provides reference data for an environmental-friendly metallurgical process and also provides the basis for the subsequent extraction of titanium and other valuable elements. Thus, this work contributes to the development of sustainable metallurgy.

## Experimental

### Sample Preparation

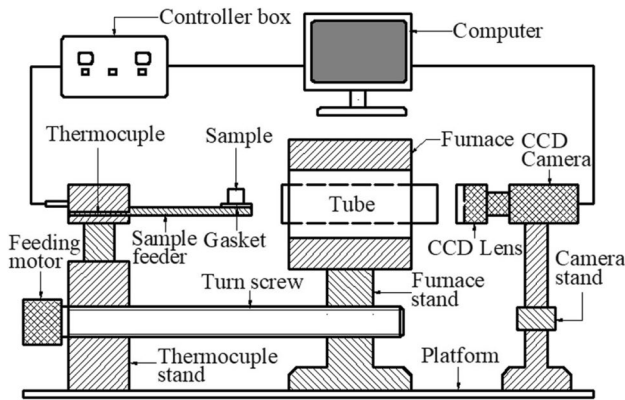
To ensure the accuracy of the experiments, the samples were pre-melted to form a homogeneous slag. Certain amounts of oxides (Table 1) were proportionally weighed, mixed, and placed into a graphite crucible lined with molybdenum flakes and then were placed in a  $\text{MoSi}_2$  resistance furnace. The mixture was melted under argon atmosphere at 1550 °C. After full stirring, the slag was removed, cooled, and crushed for further experiments.

### Experimental Apparatus

Figure 1 shows a schematic diagram of the instrument used for melting-point and melting-rate measurements that consists of three components, namely a light source, a heat

**Table 1** Chemical composition of Cr-containing high-titanium melting slag (wt%)

No	CaO	SiO <sub>2</sub>	MgO	Al <sub>2</sub> O <sub>3</sub>	TiO <sub>2</sub>	Cr <sub>2</sub> O <sub>3</sub>	B <sub>2</sub> O <sub>3</sub>
1	14.71	21.02	11.22	10.42	40.63	2.00	0.00
2	14.30	20.43	11.22	10.42	40.63	2.00	1.00
3	13.89	19.84	11.22	10.42	40.63	2.00	2.00
4	13.48	19.25	11.22	10.42	40.63	2.00	3.00
5	13.07	18.66	11.22	10.42	40.63	2.00	4.00

**Fig. 1** Schematic diagram of the experimental apparatus for melting temperature

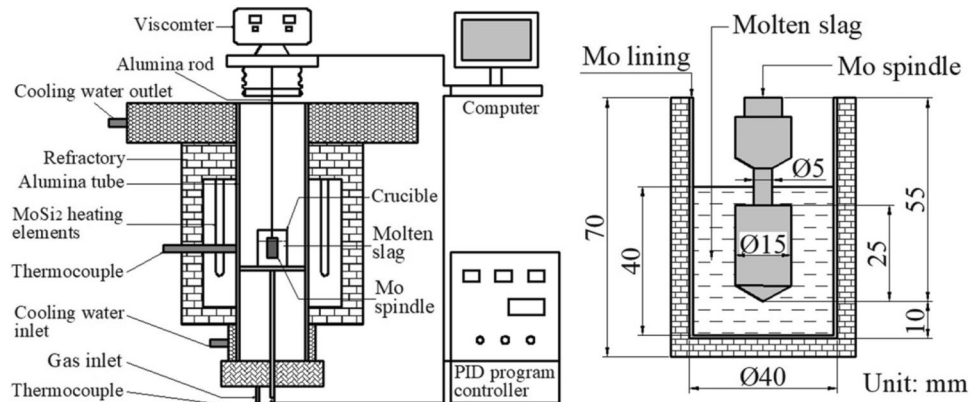
source (an electric furnace with an alumina tube), and a camera system. A U-shaped MoSi<sub>2</sub> is used as the heating element, and the highest operating temperature is 1550 °C. The temperature control precision is  $\pm 2$  °C and the heating thermocouple is located at the bottom of the sample and can display the temperature change. To reduce the experimental error, two sets of temperature readings were obtained for the same sample. If the difference between the two readings exceeded 3 °C, the sample was tested again.

The viscosity was measured using an RTW-10 melting physical property comprehensive measurement instrument designed by Northeastern University, as shown in Fig. 2. This system enables the continuous measurement of the

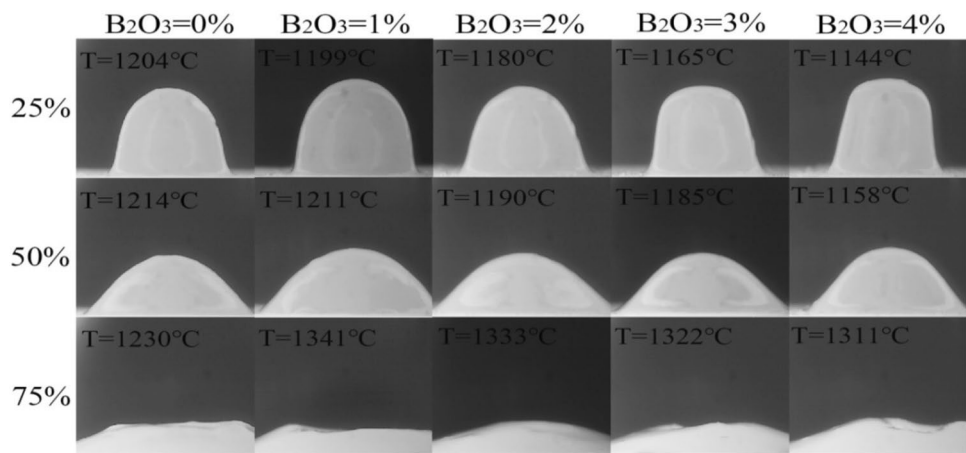
viscosity during the cooling process and the fixed-point measurement of the viscosity at a constant temperature. The heating element consists of a U-shaped MoSi<sub>2</sub> and a Pt–Rh thermocouple to ensure that the temperature deviation in the constant zone is less than 1 °C. The spindle is made of molybdenum and the connecting rod is made of corundum.

### Experimental Procedure

The slag was ground to 74 μm in an agate mortar and the standard sample was prepared by pressing the slag into the cylinder. Then, the standard sample was placed on a corundum sheet in the middle of the tube. The shape change of the standard sample was observed and the melting behavior was described by three characteristic temperatures. When the height of the standard sample decreased one fourth of its original height, the corresponding temperature was recorded as the softening temperature (ST). The hemispherical temperature (HT) was the temperature at which the height of the standard sample decreased to one half of its original height. In addition, this temperature was defined as the melting temperature and this method of temperature measurements is called the hemispherical method [25]. The flow temperature (FT) was the temperature at which the sample was liquefied and its height decreased three fourths of its original height. Figure 3 shows the morphology and characteristic temperature of the standard samples with different B<sub>2</sub>O<sub>3</sub> contents in the heating process.

**Fig. 2** Schematic diagram of viscosity furnace and detailed crucible size

**Fig. 3** Morphology and characteristic temperature of standard sample with different  $B_2O_3$  contents



Prior to carrying out the viscosity experiment, the viscometer was calibrated using castor oil at room temperature [26–28]. The distance between the spindle and the bottom of the crucible was maintained at 10 mm. When the furnace temperature reached 1550 °C, the slag was kept at this temperature for 1 h, and then the viscosity measurement was carried out as the temperature decreased. To avoid the measurement interruption caused by the overload of the torque sensor, the viscosity measurement was terminated immediately when the viscosity reached 5 Pa s. During the viscosity measurement, the slag was fully liquid and acted as a Newtonian fluid. Each experiment was conducted twice to ensure the reliability of the results. Argon was used as the protective gas throughout the experiment.

When the viscosity measurement was completed, the furnace temperature decreased to 1100 °C, and the slag solidified. Then, the slag was removed from the furnace and was air cooled to room temperature. The slag sample was broken, and a uniform size was selected. To obtain a smooth surface, the sample was first polished using sand paper with different roughness grades from coarse to fine and then polished using a polishing machine. To improve its electrical conductivity, gold was sprayed onto the detection surface prior to scanning electron microscopy (SEM) observation. The other part was ground in an agate mortar and then filtered by a 200-mesh filter sieve. Then, the obtained powder was used for X-ray diffraction (XRD) analysis.

## Characterization

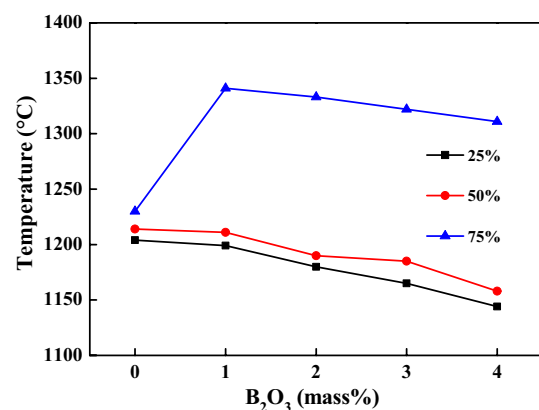
The composition of the precipitated phase was identified by XRD (PANalytical) using Cu-K $\alpha$  radiation with the voltage and current of 40 kV and 40 mA, respectively. The morphology of the precipitated phase was observed by SEM (Ultra Plus; Carl Zeiss GmbH, Jena, Germany) with energy dispersive spectroscopy (EDS) using a Schottky-type

field emission electron source and resolution ratios of 0.8 nm/15 kV and 1.6 nm/1 kV at 20 V to 30 kV.

## Results and Discussion

### Effect of $B_2O_3$ on the Melting Property

The results of the effect of  $B_2O_3$  on the characteristic temperature of Cr-containing high-titanium melting slag are shown in Fig. 4. With the increase of  $B_2O_3$  content from 0 to 4%, the ST and the HT clearly decreased from 1204 and 1214 °C to 1144 and 1158 °C, respectively. It should be noted that the FT of the slag with  $B_2O_3$  was higher than that without  $B_2O_3$ , which may be due to the expansion of the slag after the addition of  $B_2O_3$ . This gives rise to a severe flow deformation. On one hand, it is well known that  $B_2O_3$  is an acid oxide. When  $B_2O_3$  was added to the slag, according to ionic theory, the anionic structure became complicated, and the electrostatic attraction acting on the cations decreased,



**Fig. 4** Characteristic temperature of Cr-containing high-titanium melting slag with different  $B_2O_3$  contents

decreasing the surface tension. Therefore, the volume of the slag increased. On the other hand, at higher temperature, when  $B_2O_3$  was not added, the fluidity of the slag is poor due to its complex structure. However, when  $B_2O_3$  was added, it gave rise to a “flooding phenomenon” and increased the volume of the slag. The volume and height of the slag decreased only after the “bubble” was broken. Therefore, the flow temperature of the slag with  $B_2O_3$  was higher than that without  $B_2O_3$ . When the temperature reached FT, the slag melted. However, when  $B_2O_3$  content increased from 1 to 4%, the FT decreased.

Since  $B_2O_3$  is a typical acid oxide with a low melting point ( $\sim 450^\circ\text{C}$ ) [29], the presence of  $B_2O_3$  is conducive to the fusing of  $CaO$ ,  $Al_2O_3$ , and other high-melting-point components into slag and decreases the melting point of the slag. Additionally,  $B_2O_3$  can easily combine with various oxides, forming low-melting-point eutectic crystals such as  $MgO \cdot B_2O_3$  ( $988^\circ\text{C}$ ) and  $CaO \cdot B_2O_3$  ( $1100^\circ\text{C}$ ). This is also conducive to the decrease of the melting temperature of the slag.

### Effect of $B_2O_3$ on the Viscosity

Figure 5 shows the viscosity of the Cr-containing high-titanium melting slag with different  $B_2O_3$  contents. It is clear that the viscosity increased with decreasing temperature. Another important finding was that the viscosity decreased significantly with the increase of  $B_2O_3$  content, and this trend became more pronounced with higher  $B_2O_3$  content. Generally, in a certain range of  $B_2O_3$  content, a slag with high fluidity can be obtained at high temperature and high  $B_2O_3$  content.

According to a previous study,  $B_2O_3$  is an acid oxide and acts as a network former [30]. In the slag, boron is found in the form of  $[BO_4]^{5-}$  that gives rise to a more complex structure of the silicate network. This increases the degree

of polymerization (DOP) and decreases the NBO/Si [31–38]. While the presence of  $B_2O_3$  generally leads to the increase of slag viscosity, the opposite trend was observed for the high-titanium melting slag. This unusual behavior can be explained as follows. First, B–O bond is weaker than Si–O bond and breaks more easily, reducing the polymerization strength [17]. Second, although boron forms  $[BO_4]^{5-}$  with a complex network structure, the increase in the content of this structure is equivalent to the dilution of the complex silicate network structure [18, 39, 40]. Finally,  $B_2O_3$  has a low melting point and can easily combine with many different oxides to form a low-eutectic mixture. This not only decreases the slag viscosity but also significantly decreases the melting temperature and finally improves the slag fluidity [15, 16, 29].

### Crystallization Morphology

The crystallization morphology of the Cr-containing high-titanium melting slag with different  $B_2O_3$  contents are shown in Fig. 6. It was observed that with increasing  $B_2O_3$  content, the morphology and size of the precipitated phases presented dense and sparse distribution. When  $B_2O_3$  was not added, two types of precipitated phases could be found, namely dark-gray matrix phase and light-gray phase. The EDS results show that the matrix phase and the light-gray long-stripe phase were silicate and titanium bearing, respectively. For the  $B_2O_3$  content of 1%, the light gray phase was almost entirely found in the form of small particles and the long-stripe phase disappeared. The EDS results presented in Fig. 6g indicated that some boron-containing phases precipitated and the dark gray phase was still the silicate matrix.

When the  $B_2O_3$  content increased to 2%, the small light-gray phase gradually grew to long stripe and almost disappeared. However, when the  $B_2O_3$  content reached 3%, some new boron-containing phase precipitated. Moreover, based

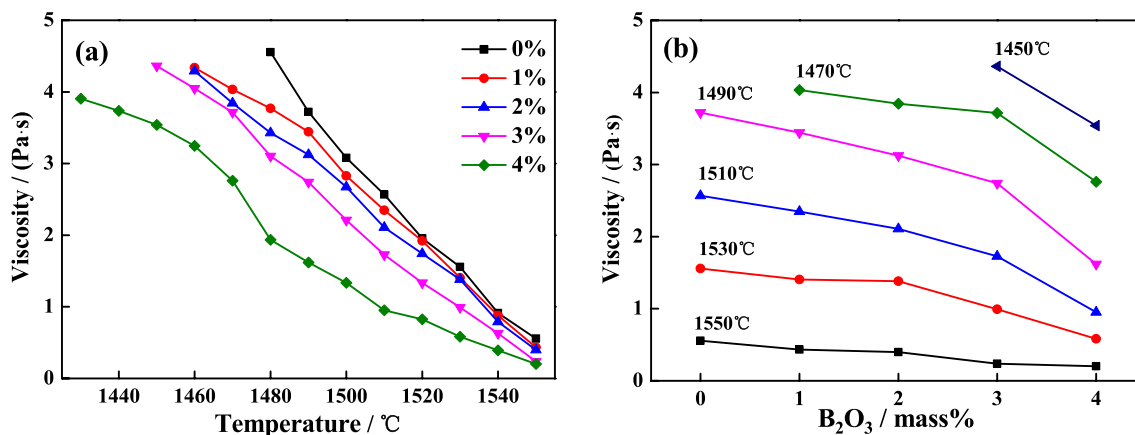
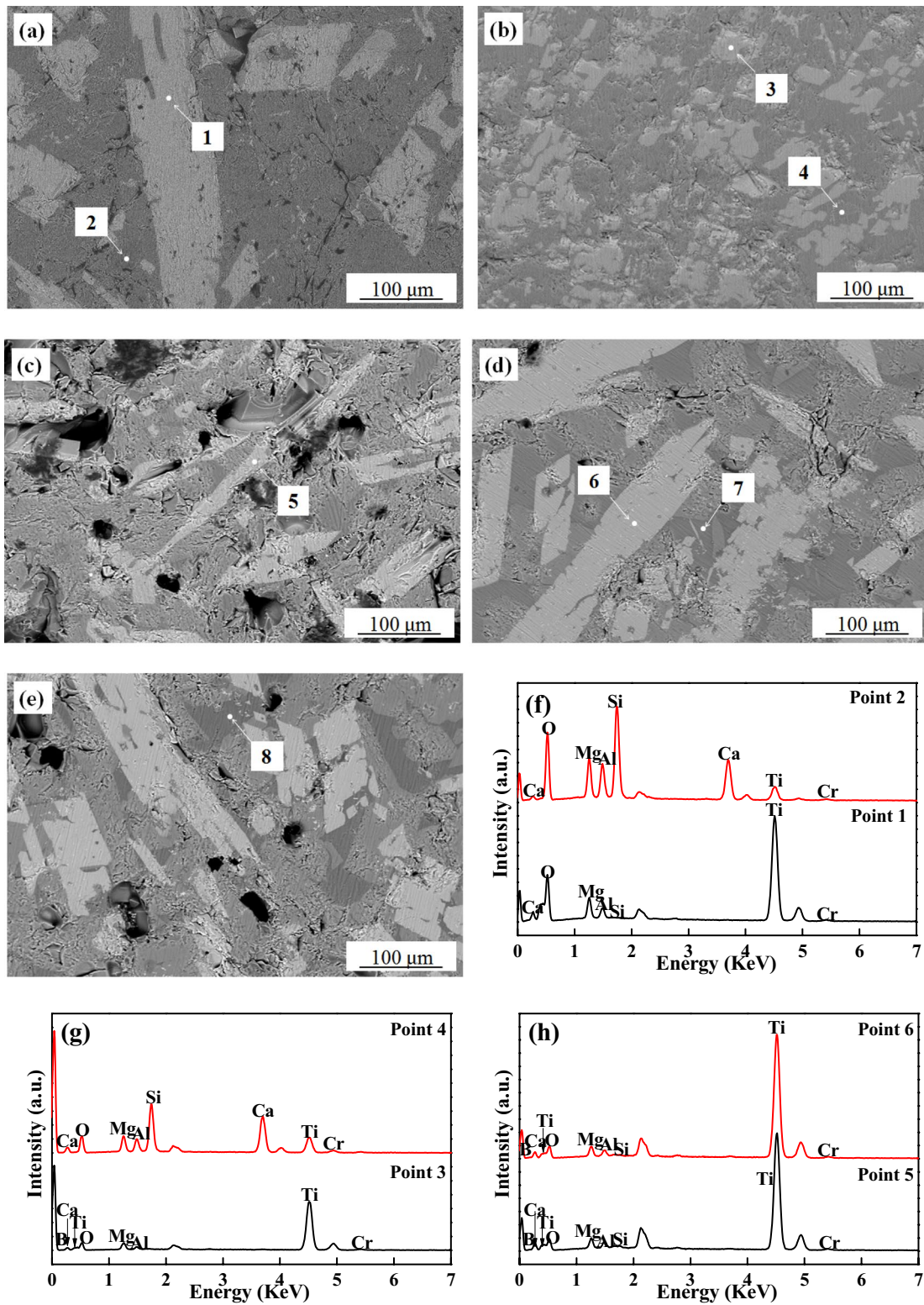


Fig. 5 Change of the viscosity of Cr-containing high-titanium melting slag with different  $B_2O_3$  contents





**Fig. 6** SEM-EDS analysis of Cr-containing high-titanium melting slag with different  $B_2O_3$  contents. **a** 0%, **b** 1%, **c** 2%, **d** 3%, **e** 4%, and **f-i** EDS analysis

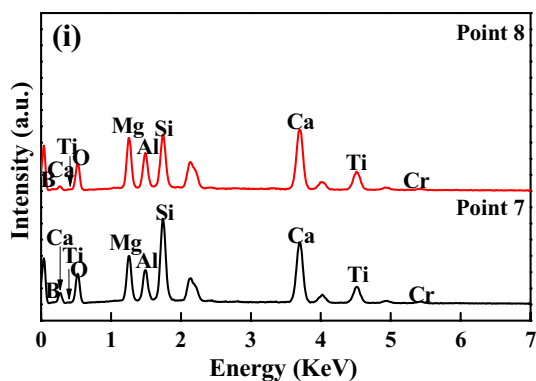


Fig. 6 (continued)

on the EDS results, it was determined that the new precipitated phase was borate. To determine the elemental distribution, EDS surface scanning was carried out, and the results are shown in Fig. 7. It was observed that Ca, Si, and Al were mainly concentrated in the matrix phase, while Ti was mainly concentrated in the long-stripe phase, and the other elements were evenly distributed. When the  $B_2O_3$  content was 4%, the morphology and elemental composition did not

change significantly, but the diffraction peak intensity of the precipitated borate phase increased.

### Crystallization Phase Composition

The crystallization phase composition of the Cr-containing high-titanium melting slag was identified by XRD, and the results are shown in Fig. 8. In the absence of  $B_2O_3$ , six precipitated phases were found, namely anosovite ( $MgTi_2O_5$ ), sphene ( $CaTiSiO_5$ ), pyroxene ( $CaMgSi_2O_6$ ,  $CaTiSi_2O_6$ ), perovskite ( $CaTiO_3$ ), anorthite ( $CaAl_2Si_2O_8$ ), and spinel ( $MgAlCrO_4$ ,  $MgCr_2O_4$ ). After 1%  $B_2O_3$  was added, the final precipitated phases contained several boron-containing phases such as kotoite ( $Mg_3B_2O_6$ ), suanite ( $Mg_2B_2O_5$ ), clinokurchatovite ( $CaMgB_2O_5$ ), sinhalite ( $MgAlBO_4$ ), warwickite ( $MgTiBO_4$ ), and danburite ( $CaB_2Si_2O_8$ ). When  $B_2O_3$  content increased to 4%, the types of the precipitated phase did not change significantly, but the diffraction peak intensities did change. The spinel phase disappeared completely, and the diffraction peak intensity of the perovskite with a high melting point also decreased. Moreover, based on the diffraction peak intensities, danburite became the main phase, verifying the reliability of the SEM results.

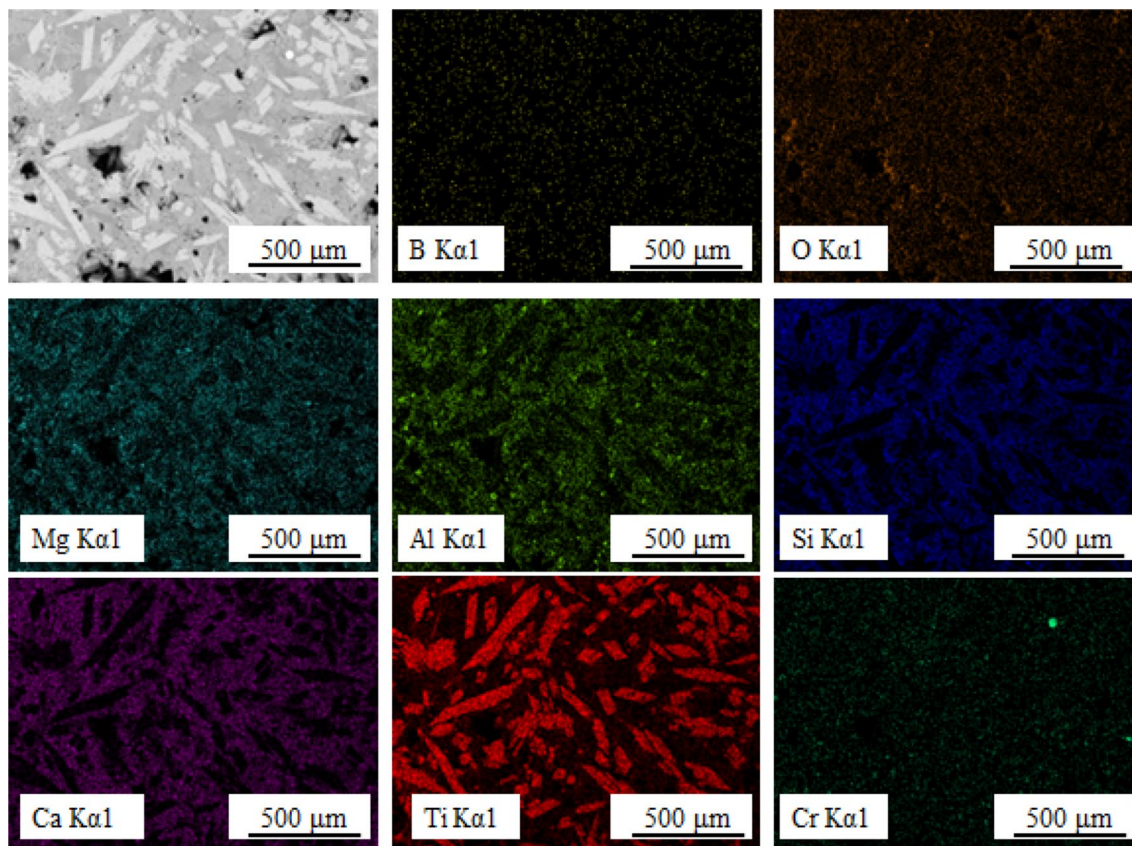


Fig. 7 Surface scanning of Cr-containing high-titanium melting slag with 3%  $B_2O_3$

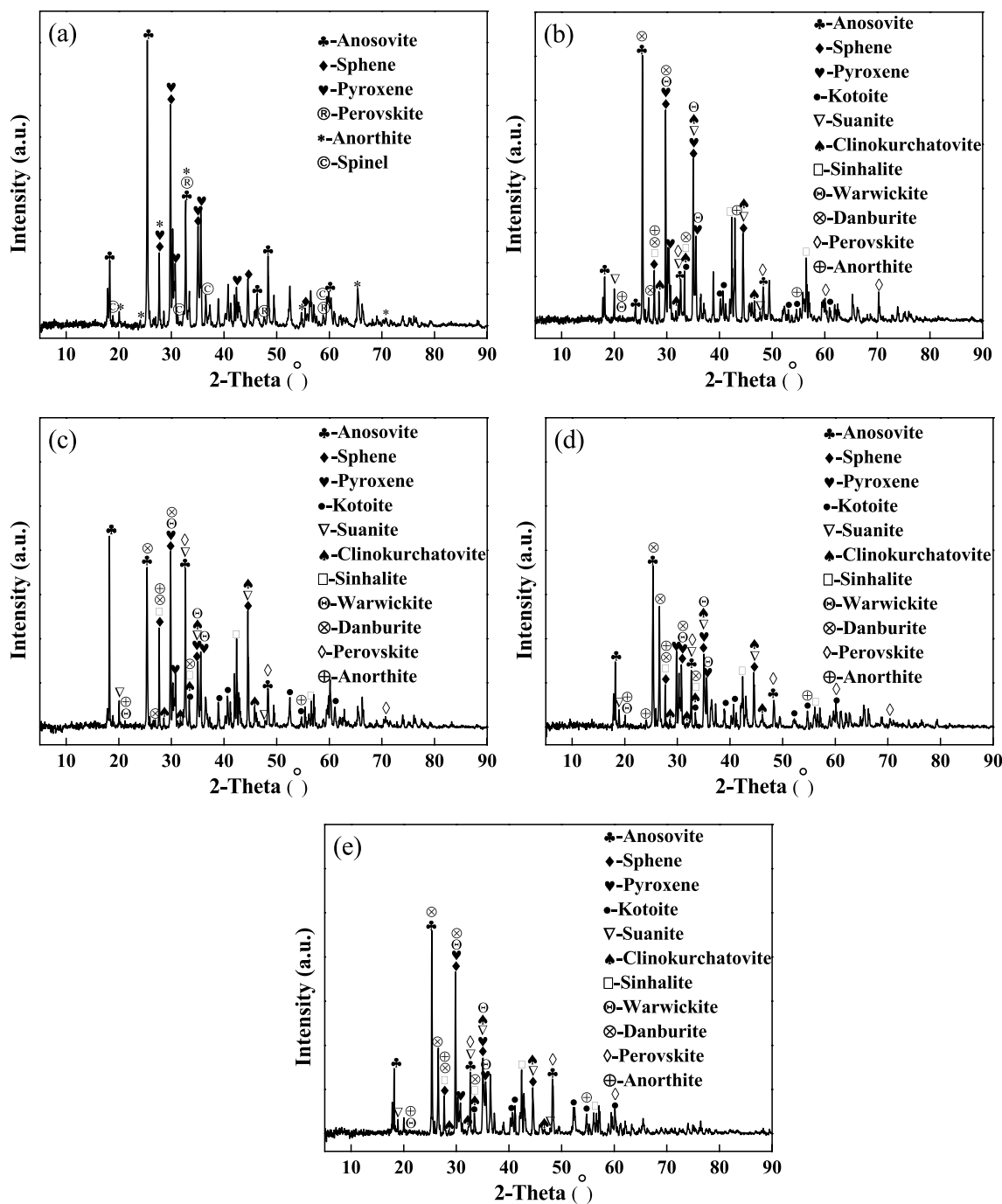


Fig. 8 XRD patterns of precipitated phases with different B<sub>2</sub>O<sub>3</sub> contents. **a** 0%, **b** 1%, **c** 2%, **d** 3%, and **e** 4%

**Discussion**

Enrichment and extraction of titanium are important for the utilization of the Cr-containing high-titanium melting slag. However, the viscosity of the slag will increase due to the presence of chromium and this may greatly increase the difficulty of titanium extraction. Therefore, to improve the fluidity of the slag, B<sub>2</sub>O<sub>3</sub> was introduced into the Cr-containing

high-titanium melting slag. As different components have different effects on the slag system due to their different internal structures. Therefore, the elucidation of the relationship between the slag composition and its properties is a key goal of our investigations. Both the phase composition and microstructure of the slag directly reflect the internal structure and affected the melting and viscous properties of the slag. According to our experimental results, when the B<sub>2</sub>O<sub>3</sub>



content increased from 1 to 4%, the melting temperature decreased significantly. In addition, the viscosity of the slag gradually decreased with the increase in the temperature and the  $B_2O_3$  content. This can greatly improve the fluidity of the slag, facilitate the separation of slag and iron, and decrease the iron loss. Moreover, the method of titanium enrichment plays a key role in the extraction of titanium.

As an acid oxide,  $B_2O_3$  acts as a network former but still improves the fluidity of the slag at high temperature.  $B_2O_3$  easily combined with other oxides to form low-melting-point eutectics, which was more effective in reducing slag viscosity. Addition of  $B_2O_3$  depolymerized the complex 3-D  $[BO_4]^{5-}$  into a simple 2-D  $[BO_3]^{3-}$  structure, meanwhile, the bridging oxygen  $O^0$  transformed into non-bridging oxygen  $O^-$  during depolymerization, depolymerizing the chain/molecule of the slag. These changes led to a decrease in the viscosity [41]. In addition, according to previous studies, the addition of  $B_2O_3$  reduced the activation energy  $E_A$  [42, 43]. The decrease of  $E_A$  indicated that the  $(BO_3)^{3-}$  of the planar triangle was the main boron-containing unit in the melt that was looser than that of  $(BO_4)^{5-}$  of the tetrahedral structure, implying that the addition of  $B_2O_3$  was beneficial for improving the fluidity of slag [44]. In the absence of  $B_2O_3$ , the main enrichment phase of titanium was anosovite. However, after the addition of  $B_2O_3$ , the amount of the precipitated perovskite decreased and the titanium-bearing phase was mainly concentrated in anosovite, which was conducive for the subsequent titanium extraction process.

## Conclusion

In this study, the effect of  $B_2O_3$  on the melting temperature and viscosity of  $CaO-SiO_2-MgO-Al_2O_3-TiO_2-Cr_2O_3$  slag was studied. The following conclusions were drawn:

- (1) The addition of  $B_2O_3$  is beneficial for decreasing the melting temperature. When the  $B_2O_3$  content increased from 0 to 4%, the melting temperature decreased from 1214 to 1158 °C. In addition, the softening temperature and the flow temperature also decreased by 60 °C and 30 °C, respectively.
- (2) The viscosity decreased with increasing  $B_2O_3$  content, particularly when the  $B_2O_3$  content increased from 3 to 4%. Thus, an increase of  $B_2O_3$  content in a certain range greatly improves the fluidity of the slag.
- (3)  $B_2O_3$  had a strong effect on the morphology of the precipitated phase as manifested in the size and shape of the precipitate particles. The morphology of the precipitated phase changed from small sheets to long stripes. In addition, the increase in the  $B_2O_3$  content led to anosovite and danburite becoming the main precipitated phases and inhibited the precipitation of perovskite,

which was beneficial for the subsequent titanium extraction process.

- 4) The addition of  $B_2O_3$  to  $CaO-SiO_2-MgO-Al_2O_3-TiO_2-Cr_2O_3$  slag is highly great significant for the improvement of slag fluidity and environmentally friendly, as well as for the development of sustainable metallurgy.

**Acknowledgements** This work is financially supported by the National Natural Science Foundation of China (Grant No. 51904066), Liaoning Revitalization Talents Program (Grant No. XLYC1802032), Fundamental Research Funds for the Central Universities (Grant No. N182503032), Postdoctoral Foundation of Northeastern University (Grant No. 20190201), and Postdoctoral International Exchange Program (Dispatch Project, 20190075).

## Declarations

**Conflict of interest** On behalf of all authors, the corresponding author states that there is no conflict of interest.

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