

Tool Wear and Surface Roughness in M 35 Single Point Cutting Tool Steel Under Non‑cryogenic and Cryogenic Condition

P. I. Patil¹ · M. M. Patil1 [·](http://orcid.org/0000-0002-5535-3079) P. R. Baviskar¹

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Abstract

This paper presents the comparative analysis of M-35 high-speed steel tool with the cryogenic and conventional heat treatment process. Phase transformation is an efective way to improve the tool material properties. The investigation observes the infuence of the cryogenic process on roughness and tool wear of M35 single point cutting tool. It observed that for a cryogenic tool due to an increase in the hardness, the resistance of the work piece become low. Moreover, scanning and microscopy performed using a scanning electron microscope to analyses the variation in microstructure characteristics. The analysis of variance (ANOVA) technique has employed to investigate the highest contributing factor process parameters with experimental validation. The tool steel electrodes are carried out with heat treatment that gives the desired results. The treatment at a very low temperature at cryogenic state is carried out.

Keywords High-speed steel · Deep cryogenic · Wear resistance · Austenite · Martensite

1 Introduction

The increasing demand for higher productivity at an afordable cost has formulated for tool materials like cemented carbide, cermets, ceramics, and ultra-hard materials (CBN, PCBN, and PCD). High-speed steel serves this purpose hence widely preferred material for tools in the industry [\[1](#page-9-0)]. It has various applications such as, broaches, milling cutters, taps, drills where it has low cutting speed than carbide tools. Modern machining methods evolved for high accuracy of machining, lower surface roughness with increment in chip volume. Presently, hard machinability materials were machined with the regular manufacturing methods that are non-feasible due to greater tool wear and high machining time [\[2](#page-9-1)]. The increasing tool life for cutting tools is an important economic factor in the manufacturing industry. Conventional heat treatment is applied to tool steel and high-speed steel (HSS) for increasing the cutting tool life of drills and improving their properties [\[3](#page-9-2)]. After conventional heat treatment, the austenite phase is retained in the material's microstructure [[4\]](#page-9-3). This retained austenite has a very soft phase in the microstructure that hampers wear and fatigue life [[5\]](#page-9-4) while cryogenic treatment observed beneficial to reduce the residual stresses and improve fatigue life and wear behavior [\[6](#page-10-0)]. Cryogenic treatment becomes a supplementary process to the conventional heat treatment process [[7](#page-10-1)]. Cryogenic treatment performed in two temperatures were used−63 °C as shallow cryogenic temperature (SGT) and−185 °C as deep cryogenic temperature (DCT) [[8\]](#page-10-2). Because of this, the formation of carbide particles and their homogeneous distribution takes place to reduce the retained austenite. It observed that that 6.5% austenite retained in the conventional heat treatment. It was reported that the retained austenite reduces to 5.1% after SCT while 2.7% after DCT from 6.5% [\[9](#page-10-3)]. Due to this process, the capacity of wear resistance and hardness of HSS tools increases [[10](#page-10-4), [11](#page-10-5)]. Implementation of cryogenic treatment of tools for diferent materials and cutting tools has been increased signifcantly in recent years due to their additional properties like stress relieving and increasing tool life [\[12–](#page-10-6)[14\]](#page-10-7). Das et al. [[15\]](#page-10-8) proposed the study of wear resistance of cryogenically treated with diferent soaking times (0–132 h) for steel [AISI D2]. It observed that for a soaking time of 36 h, the maximum wear resistance of cold-worked steel is reported. Huang, et al. found that wear resistance increases due to the homogeneous distribution of carbide particles that formed after cryogenic treatment [[16\]](#page-10-9). It is found recorded that M2 HSS drills while drilling on carbon

 \boxtimes M. M. Patil milind.patil@sitrc.org

¹ Department of Mechanical Engineering, Sandip Institute of Technology and Research Centre, Nashik, MH 42213, India

steel, shows 77% improvement in tool life and 126% improvement in wear resistance for deep cryogenic treatment [\[6](#page-10-0)]. In another study, the cryogenic treatment was done on En31 steel that has reported a decrease in wear by 75% for diferent working conditions [\[17\]](#page-10-10). Also, it is found that tool life is improved by 110% for cryogenic treatment which is better than TiN coatings [\[18](#page-10-11)]. In a study, it is reported that retained austenite along with the formation of eta-carbide/nano-scale carbides refects in terms of increase in wear resistance. When cryo-soaked at−185 °C for 4–48 h and soft tempered to 100 °C, several mechanical properties such as hardness, wear loss, residual stresses, precipitation of carbide, and homogeneous carbide distribution are studied by observing the microstructure of samples. The infuence of these parameters on wear behavior is investigated to understand the wear mechanism. The cryotreated specimens exhibited low wear and subsequent wear stabilization for higher cryo-soaking intervals [\[19\]](#page-10-12). Developed the crystal structure with improved adhesive, erosive wear, and corrosion resistance [\[20\]](#page-10-13). If the parts are treated up to−84 °C, it is known as shallow cryogenic treatment and from−84 °C up to−196 °C, the process is called deep cryogenic treatment. Deep Subzero treatment is the stress-relieving technique for metals and alloys. The materials are exposed to low temperatures up to−185 °C for a prolonged period leading to thermal equilibrium as well as equilibrium in phase concentration conditions. The specimens made up of M35 were treated to 1200 °C for hardening, triple tempered at 400 °C, cryo-soaked to−185 °C for 4 to 48 h, and soft tempered (100 °C), and the wear behavior was investigated [\[21](#page-10-14)]. The diferent characterization techniques namely hardness, residual stresses, wear rates and SEM images were considered for investigation for fnding out the cause of failure. It was observed that the amount of cryo-soaking time (4–48 h) governs the kinetics of carbide precipitation. Excessive accumulation of residual stresses at cryogenic temperature is the factor for cracking [\[22](#page-10-15)].

The present study reports the effect of cryogenic treatment on M35 HSS single point cutting tool for surface roughness and tool wear in turning of AISI 1018 work piece, as it preferred in industry and exhibits good machinability characteristics. The cryogenic treatment reduces the stresses as per the third law of thermodynamics as the material is subjected to very low temperature and achieves the equilibrium conditions and attains the minimum entropy.

2 Experimental Method

Experiments were performed on the Smarturn CNC lathe machine. It has a maximum turning length of 262 mm and a maximum turning diameter of 200 mm. M35 non-cryo and cryo tools used for the experiment. The work material is a low carbon steel (AISI 1018) of a bar (45 mm \times 6 m). Design of experiment plans, design matrix with the desired level of parameters, and ANOVA analyses the infuence of parameters on performance parameters with validation of experiment [[23,](#page-10-16) [24\]](#page-10-17). After the experimentation, surface roughness and tool wear are measured, also SEM analysis has been performed. Finally, an empirical model was proposed and confrmation of the experiment was carried out.

2.1 Sample Preparation

Duly Conventional Heat-treated (CHT) HSS M35 steel tool was selected for experimentation. HSS M35 tool steel contains molybdenum (Mo) as a primary alloying element. Table [1](#page-1-0) shows the chemical composition of M35 tool steel.

Tool bits of grades conventionally heat-treated (CHT) HSS M35 steel tool was procured from Miranda Tools, Ankleshwar, Gujarat, (India). All tool bits have standard size $\frac{1}{2}$ " × $\frac{1}{2}$ " × 4" [12.7 mm × 12.7 mm × 101.6 mm]. Pilot experiments were conducted with non-cryo tools (Fig. [1](#page-2-0)a) and cryo tools (Fig. [1](#page-2-0)b). As tool bits were procured referring to the standard grades, the heat treatment and the chemical compositions were maintained during the manufacturing stages. The cutting tools used in the machining tests are: These tool bits were subjected to grinding operations for providing the standard single-point tool signature for turning operation. The dimensions of tool geometry provided to the specimen are, back rake angle: 10°, side rake angle: 10°, end relief angle: 10°, side relief angle: 5°, end cutting-edge angle: 8°, side cutting-edge angle: 0°, nose radius: 0.5 mm.

Commonly used work piece material in turning is the steel from the revised literature, having low or medium carbon content due to its abundance. However, it was decided to select a work piece of low carbon steel material corresponding to AISI 1018. From the available diameter of bar 45 mm \times 6 m AISI 1018, the specimens for machining of original diameter of 45 mm and length of 260 mm are prepared on the power hacksaw machine as shown in Fig. [1](#page-2-0)c.

Fig. 1 a Non-cryo, **b** cryo tool and **c** work piece for experimentation

Table 2 Process parameters for cryogenic treatment

Sr. No.	Factors	Assigned values for HSS M 35 tool steel			
	Cooling rate	$(1-2.5 \text{ °C/min}) 2 \text{ °C/min}$			
	Soaking temperature	$-185 °C$			
	Soaking time	8 h, 12 h, 14 h and 16 h			
$\overline{4}$	Soft tempering temperature	100 °C			
	Soft tempering time	1 h			

2.2 Cryogenic Tool Preparation

Table [2](#page-2-1) shows the process parameters for cryogenic treatment. Soaking time has varied for each factor of given tool steel.

An additional fourth factor in the case of cryogenically treated tools is soft tempering temperature of 100 °C and soft tempering time of 1 h. By selecting the said values of cryogenic temperature, soaking time, cutting speed, feed, and depth of cut, trials had been conducted using non-cryo tools as well as cryo-treated tools. The complete setup for cryogenic treatment tools has been depicted in Fig. [2](#page-2-2). A treatment of cryogenic is performed at − 185 °C with a soaking time in the steps of 8 h, 12 h, 14 h, and 16 h were adopted. It allows a phase transformation followed by tempering at 100 °C. The internal stresses creep in due to the transformation of retained austenite as a result of cryogenic treatment that induces brittleness and volumetric expansion. The brittleness is alleviated to relieve internal stresses and fne carbide specimens are tempered. All the cryo-treated tools were invariably subjected to soft tempering in the

Fig. 2 Schematic diagram for cryogenic process

muffle furnace (having specifications 3 kW , 230 V AC , 600 ± 5 °C) at 100 °C for 1 h to relieve the cooling stresses. Specimens were cooled at the rate of 20C/min until they reach the fnal soaking temperature of−185 °C. Thereafter, the cycle is reversed and the temperature is allowed to ramp up at the rate of 2 °C /min to room temperature.

2.3 Design of Experiments (DOE)

The experiments were designed using the mixed factorial design of the experiment (DOE). DOE has become a useful method for obtaining high-reliability search results, especially because it saves a lot of time and material costs [[25,](#page-10-18) [26](#page-10-19)].

Table [3](#page-3-0) depicted the levels assigned to various factors. The cutting speed, feed, and depth of cut chosen as process parameters for non-cryo tools. Whereas, the soaking time has been considered the fourth process parameter for cryo tools. The individual values of these factors to be assigned at each level are, in general, equispaced. The ratio between the consecutive values of cutting speed has been kept as 1.06 as per ISO: 3685-1993 [[27\]](#page-10-20). The values selected for feed and depth of cut are based on the grades of the material. The ratio of feed to the depth of cut is kept as 4 as per ISO: 3685-1993 [\[27](#page-10-20)]. The surface roughness is an indication of close-spaced irregularities present on surface texture [[28\]](#page-10-21) and interpreted in microns (µm) and denoted by Ra. If this value drops, the surface becomes smooth, and if it increases it means that the surface is rough. The surface roughness (*Ra*) value is measured using the surftest sj-210 machine. Toolmakers microscope with magnification \times 30 has been used to measure the tool wear (Tw).

3 Results and Discussion

3.1 Analysis of Non‑cryo HSS M35 Tools

Analysis of non-cryo HSS M35 tools by using Minitab 17 software empirical models are developed with the data obtained by experimentation with non-cryo- tools. For furnished result, experiment has been performed thrice and average values are calculated. Analysis of variance (ANOVA) formulated for identifying signifcant factors. The results of performance parameters Tw and Ra, obtained for eight experiments along with the design matrix are summarized in Table [4](#page-4-0)**.**

The results of the quadratic model for Tool wear in the form of ANOVA given in Table [5.](#page-5-0) The ft summary recommended that the quadratic model is statistically signifcant for the analysis of tool wear. The values of R^2 and adjusted $R²$ are over 95%. This means the regression model focuses on the relationship between process parameters and performance parameters. When the associated *P*-value for the model is lower than 0.05, indicates that the model is considered to be statistically signifcant. It generates the equation that describes the relationship between performance and predictor parameters called regression. The effects of various process parameters on the surface roughness (*Ra*) are analyzed based on Eqs. ([1\)](#page-3-1) and ([2](#page-3-2)).

(1) *Tw* = −0.0364 + 0.000312 Speed + 0.0380 Feed + 0.21050 DOC

(2) *Ra* = 2.975 − 0.01452 Speed + 13.080 Feed + 0.105 DOC

Cutting speed has a considerable infuence on tool temperature. The feed has a little efect on temperature because a greater area of the tool is used which will dissipate additional heat. Figure [3](#page-5-1) shows the normal probability plot of the residuals for (a) *Tw* and (b) *Ra*. It is noticed that the residuals are falling on a straight line, which means that errors are normally distributed.

Figure [4](#page-5-2)a shows the microstructure image of noncryogenically treated H.S.S and M35 material with \times 10,000 magnification. It can be seen the elongated carbides and not abnormal distribution with large carbide size. Figure [4](#page-5-2)b shows the exact structure of tool wear (0.05405 mm). It can be concluded that tool wear is uneven and afects the tool life.

3.2 Analysis of Cryo‑Treated HSS M35 Tool

The results obtained from the design of experiments were analyzed using ANOVA to fnd the signifcance of each input factor on the measures of process performances. The sixteen experiments were conducted in a triplicate manner and the average values of *Tw* and *Ra* along with the design matrix were tabulated in Table [6](#page-6-0).

ANOVA models are formulated in Table [7](#page-6-1). The results of factor responses are considered by using 'lower the

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better' expectation. ANOVA results show that the confdence level for the Cryo M35 tool is more than 95% [for *Tw*—99.51%, *Ra*—99.30%]. *P*-value represents that Soaking time, cutting speed; feed, and depth of cut are parameters that have a signifcant contribution. *F* value demonstrates that the signifcant parameters are having greatest value, 1342.87, and 606.88 for *Tw* and *Ra*. The mathematical relationship was thus obtained for analyzing the infuences of the dominant process parameters on performance parameters. Equations ([3](#page-4-1)) and ([4](#page-4-2)) show the regression equation for this model (Cryo M35).

 $Tw = 0.08333 + 0.001157$ Soaking Time H.

Ra = 2.525 + 0.01808 Soaking Time H.

− 0.01067 Cutting Speed + 16.156 Feed

(4) − 0.871 Depth of Cut

A straight line in a normal probability plot indicates that data does ft a normal probability distribution. Figure [5](#page-7-0)a and b illustrate that residuals follow an approximately straight line in a normal probability plot for tool wear and surface roughness.

Figure [6](#page-7-1) shows a microstructure image of Cryogenically Treated H.S.S. M35 material with soaking time 8, 12, 14, and 16 h with \times 10,000 Magnification. Most uniformly distributed spherical-shaped carbides are observed in the microstructure of the cryo-treated tool. Little Spherical shaped carbide is seen attached to some irregularly shaped carbide. The carbide size seems to increase at and above 12 h soaking temperature. It has been observed that a homogeneous distribution of carbides due to secondary carbide precipitation. Therefore, the hardness and wear resistance of M35 HSS tools improve signifcantly. When the specimens were cooled at−185 °C and tempering at about 100 °C is done, the optimum wear resistance is observed. It is supposed that a large amount of retained austenite is transformed into martensite during cryogenic treatment. Due to the dispersion strengthening mechanism as a result of the formation of fne precipitates of carbides and their uniform distribution in martensite matrix the wear resistance of deeply cryo-treated cutting tool had increased. Figure [7](#page-8-0) clearly shows that wear of tool subjected to cryo-treatment with soaking period of 16 h has been seen comparatively less than untreated form as well as 14 and 8 h cryo- treated. However, cry-treated samples for 14 h and 8 h have almost the same volume fraction of retained austenite being transformed as the 16 h treated sample. Nevertheless, the wear resistance of later was superior to the former. It should be inferred that the precipitation of the carbides resulted from the cryogenic treatment is

Table 4 Response of non-cryo M35 tool **Table 4** Response of non-cryo M35 tool

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Fig. 3 Normal probability plot tool M35 non-cryo **a** tool wear **b** surface roughness

responsible for the improvement of wear resistance. However, the non-cryo-treated single point tool has more tool wear observed.

3.3 Validation

Figure [8](#page-9-5) depicts the comparison of predicted and experimental performance parameters. Predicted values are calculated by ANOVA-generated regression equations. The reading obtained by machining is in good agreement with the predicted and actual parameters. The genuine results predicted values and calculated percentage error

Fig. 4 CHT AISI M35 HSS (Non-Cryo) **a** SEM image **b** optical view of tool wear

(Flank)

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Expt. No	Process parameters				Observed values							
	Cutting speed m/min	Feed mm/rev	DOC mm	Soaking time Hr	Tw (mm)				Ra (µm)			
					I	\mathbf{I}	Ш	Average		\mathbf{I}	Ш	Average
1	75	0.075	0.3	8	0.048	0.0463	0.0473	0.0472	2.7565	2.9122	2.8105	2.8264
2	80	0.075	0.3	8	0.0402	0.0405	0.0387	0.0398	2.7603	2.7686	2.6674	2.7321
3	85	0.125	0.5	8	0.069	0.0645	0.0705	0.068	3.5655	3.4895	2.713	3.256
4	90	0.125	0.5	8	0.0638	0.0601	0.0597	0.0612	3.3122	3.2565	3.2893	3.286
5	75	0.075	0.5	12	0.0725	0.0696	0.0715	0.0712	2.7126	2.6505	2.6904	2.6845
6	80	0.075	0.5	12	0.0662	0.064	0.0642	0.0648	2.5807	2.6128	2.68	2.6245
7	85	0.125	0.3	12	0.0596	0.058	0.0582	0.0586	3.5905	3.5125	3.592	3.565
8	90	0.125	0.3	12	0.0506	0.0539	0.0524	0.0523	3.5784	3.6144	3.4935	3.5621
9	75	0.125	0.3	14	0.0796	0.0772	0.0778	0.0782	3.8603	3.7969	3.8118	3.8230
10	80	0.125	0.3	14	0.0739	0.0708	0.0728	0.0725	3.7863	3.802	3.6695	3.7526
11	85	0.075	0.5	14	0.0658	0.0682	0.0664	0.0668	2.8458	2.7564	2.8146	2.8056
12	90	0.075	0.5	14	0.0596	0.0602	0.056	0.0586	2.6905	2.6404	2.6485	2.6598
13	75	0.125	0.5	16	0.0996	0.0975	0.0993	0.0988	3.7905	3.8124	3.6933	3.7654
14	80	0.125	0.5	16	0.0918	0.0936	0.0918	0.0924	3.7455	3.7128	3.5886	3.6823
15	85	0.075	0.3	16	0.0602	0.0569	0.0575	0.0582	2.9685	2.9355	2.9256	2.9432
16	90	0.075	0.3	16	0.0501	0.0479	0.0508	0.0496	2.9864	2.9464	2.9292	2.9540

Table 7 ANOVA results for cryo M35 tool

of verifcation experiments conducted with the process parameters for diferent responses individually that the prediction error is about less than 5%. It has been discovered from validation experiments that there is a minor percentage error between the predicted and experimental values therefore, the model can be validated.

4 Conclusions

It can be concluded that the cryogenic treatment is found to have a signifcant infuence on the formation of martensite from retained austenite that provides a homogeneous distribution of carbides due to secondary carbide precipitation. Hence, wear-resistance and hardness for M35 HSS tools improve signifcantly. M35 cryo-treated tool steel shows a reduction in the tool wear as compared with

Fig. 5 Normal probability plot tool M35 cryo **a** tool wear, *Tw* **b** surface roughness, *Ra*

Fig. 6 SEM of Cryo treated M35 **a** 8 h **b** 12 h **c** 14 h **d** 16 h

Fig. 7 Observed tool wear with soaking period **a** 16 h (*Tw* 0.07475 mm) **b** 14 h (*Tw* 0.069025 mm) **c** 12 h (*Tw* 0.061575 mm) **d** 8 h (*Tw* 0.05405 mm)

Fig. 8 Validation with experimental and predicted values **a** and **b** M35 Non-Cryo, **c** and **d** M35 Cryo

M35Non-cryo treated tools. Overall, the cryogenic treatment shows a favorable infuence on the performance of tools. The proposed treatment would be a suitable alternative that enhances productivity however further optimization of various parameters of the thermal cycle may give better results. M35 cryo-treated tool steel shows a reduction in the tool wear and improvement in surface roughness as compared with M35 non-cryo treated tools. The signifcant levels of process parameters can be identifed for further optimization.

Declarations

Conflict of interest There is no confict of interest.

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