

# **An Investigation on Hot Corrosion Behaviour of Cermet Coatings in Simulated Boiler Environment**

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#### **Abstract**

In this research work, hot corrosion behaviour of different  $Cr_3C_2-NiCr$  coatings has been investigated. The coatings were deposited on commercially available boiler tube steel with high velocity oxy fuel thermal spraying technique. The uncoated and coated specimens were exposed to elevated temperature in a silicon tube furnace at 700 °C in molten salt environment. The thermogravimetric technique was used to compare the corrosion resistance of diferent coatings for 50 cycles at high temperature. The microstructures of exposed specimens were evaluated with X-ray difraction and scanning electron microscopy with elemental compositional analysis.  $10Cr_3C_2-90NiCr$  and  $20Cr_3C_2-80NiCr$  coatings provided the higher resistance to corrosion as compared to 100NiCr and  $35Cr_3C_2-65NiCr$ . The formation of chromium carbide layer on the coated surface and this layer protect them from corrosion.

**Keywords** Corrosion · Boiler · Steel · Coating · Temperature

## **1 Introduction**

The surfaces of alloys and metals are oxidized when they are covered with a thin flm of fused salt at elevated temperature, and a porous non-protective oxide scale is formed at the surface. [\[1](#page-5-0)[–4\]](#page-5-1). This is known as hot corrosion. Due to the presence of salt contaminants such as  $Na<sub>2</sub>SO<sub>4</sub>$ , NaCl,  $V<sub>2</sub>O<sub>5</sub>$  at high temperature, this type of oxidation is accelerated [\[1,](#page-5-0) [5](#page-5-2), [6\]](#page-5-3). Hot corrosion is also form of oxidation and these salt contaminants degrade the protective oxide layer [\[7,](#page-5-4) [8](#page-5-5)]. In 1940, the hot corrosion was frst recognized for the degradation of the boilers tubes and steam generating plants [\[1](#page-5-0), [9](#page-5-6)]. After that, this problem was observed in IC engines, gas turbines, and many industrial waste incinerators. Today the researchers have developed many efficient coatings that help to protect the materials from hot corrosion. In a coal fred boiler, many failures occur due to hot corrosion and erosion. It is reported in the literature that 89 failures occurred in boilers in 1 year, out of which 50 failures occurred due to hot corrosion and erosion [\[10\]](#page-5-7). Problems related to corrosion are not completely solved, but the cost related to corrosion can be decreased by 30% by using the diferent coating techniques.

Coatings are being used to increases the life of the materials, improve the performance, and also give a unique appearance  $[11-13]$  $[11-13]$ . Coatings reduce the corrosion rate to a great extent [\[14](#page-5-10)–[16\]](#page-5-11). It is clear that 80% of the total investment to protect the metal from corrosion is related to coating applications [[17,](#page-5-12) [18\]](#page-5-13). So, it is important to know all types of environment and many other conditions which degrade the material and preventive measure can be taken against them to ensure the reliability and safety in the use of metal components [[19–](#page-5-14)[21\]](#page-5-15). The use of protective coating has been increased to decrease the cost related to the maintenance of the component [[22\]](#page-6-0). Applying coating on the material is the method to improve the life of the material and also balance their mechanical properties while protection against corrosion and erosion [[23](#page-6-1)[–25](#page-6-2)].

High-temperature erosion–corrosion is a serious problem in steam generation plants like gas turbines, IC engines, industrial waste incinerators, and boilers. It has been suggested by some researcher that by using diferent materials for boiler tubes together with Gr A1, T11, T22, T91, T92, Super 304H, and many diferent steels in numerous sections of the boilers in industries, the problem of hot corrosion was reduced

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to some extent. But, the research is constantly being done to enhance the performance of these materials with the aid of diferent coatings by the use of specifc coating techniques or by way of introducing new materials for such applications. Literature review reveals that researchers have developed various types of coatings to improve corrosion resistance of boiler steel  $[26-34]$  $[26-34]$ . But the research of on NiCr–Cr<sub>3</sub>C<sub>2</sub> coating is very limited. Therefore, it was decided to investigate performance of these distinct coatings on T22 boiler tube metal (low value material) in silicon tube furnace at 700 °C. The research was designed to study hot corrosion behaviour of boiler steel "ASTM-SA213-T22" with different  $Cr_3C_2-NiCr$ coatings under molten salt environment in silicon tube furnace at 700 °C, under cyclic conditions, and to investigate the microstructure of corroded specimen by using XRD and SEM/ EDAX techniques.

## **2 Materials and Methods**

## **2.1 Substrate Materials**

For this research work, T22 boiler steel material was selected to study hot corrosion behaviour. This material is currently being used in Guru Nanak Dev Thermal Plant Bathinda (Punjab), India. The spectroscopy of T22 boiler steel was done at Munjal castings Ludhiana (Punjab), India. The actual composition of T22 boiler steel is shown in Table [1](#page-1-0).

Small pieces were cut from the rod of T22 boiler steel. The dimensions of steel specimen were approximately  $20 \text{ mm} \times 15 \text{ mm} \times 5 \text{ mm}$ . The specimens were polished by SiC emery paper 200, 600, 800, 1000.

#### **2.2 Development of Coating**

The material used for coating was  $Cr_3C_2-NiCr$  and this material was available in powder form. The  $Cr_3C_2-NiCr$  coatings were deposited at M/S Metallizing Equipment Co. Pvt. Ltd., Jodhpur (India). Diferent types of coatings were deposited on the specimens. The designation and chemical composition of these coatings is shown in Table [2.](#page-1-1)

Before the deposition of the coating,  $Al_2O_3$  (Grit 60) was used for grit blasting the specimens. Coating deposited at M/S Metallizing Equipment Co. Pvt. Ltd., Jodhpur (India). Liquefed petroleum gas (LPG) was used as a fuel. During and after coating specimens were cooled with the help of compressed air jets.

<span id="page-1-1"></span>



#### **2.3 Exposure in Silicon Tube Furnace Environment**

The coated as well as uncoated specimens were exposed to 700 °C in silicon tube furnace. About 700 °C was the average temperature with the variation of  $\pm$  5 °C. Each cycle consisted of 1 h heating followed by 20 min cooling at ambient conditions. At the end of each cycle, to analyse the surface texture visual observation of each specimen was done. After the visual observation, it has been washed out by acetone and the weight of specimen was taken using Electronic balance (CB-120, Contech Instruments Ltd., accuracy 0.001 g). After the 50 cycles, resulting specimens were analysed by visual examination, weight changes studies, SEM/EDAX, and XRD analysis.

### **3 Results**

#### **3.1 Visual Examination**

The behaviour of uncoated "ASTM-SA213-T22" boiler after exposing to 700 °C for 50 cycles was visually examined. All the specimens were exposed in silicon tube furnace at elevated temperature. After 5th cycle, spalling of specimen was started, which increased with the increase in the number of cycles. At the end of the 20th cycle, peeling started and continued till the end of 50 cycles. At the end of 50th cycle, surface of the piece got distorted. The image of uncoated specimen after 50 cycles is shown in Fig. [1](#page-2-0)a. When coated specimen  $35Cr_3C_2-65NiCr$  was exposed to 700 °C for 50 cycles, at the end of 5th cycle, the shine on the surface was observed and white colour was observed after 15 cycles. The grey spots were seen on the surface after 20 cycles, and spalling of the specimen started at the end of 30th cycle. At the end of 50, the coating of specimen was found completely destroyed. The image of the  $35Cr_3C_2-65NiCr$ -coated specimen is shown in Fig. [1b](#page-2-0). Another coated specimen

<span id="page-1-0"></span>**Table 1** Actual chemical composition (wt%) for T22 boiler tube steel used in the present study

Type of steel		Mn	$\sim$ n,			ັ	Mo	Ni	
T22 1/2	0.20	0.66	$\Omega$ U.J 1	0.033	0.031	0.53	$\sim$ 1.32	0.052	Balance



<span id="page-2-0"></span>**Fig. 1** Macrographs of specimens **a** Uncoated T22, **b** 35Cr<sub>3</sub>C<sub>2</sub>–65NiCr, **c** 20Cr<sub>3</sub>C<sub>2</sub>–80NiCr, **d** 10Cr<sub>3</sub>C<sub>2</sub>–90NiCr, **e** 100NiCr after studied the hot corrosion behaviour in molten salt environment under cyclic conditions

 $20Cr_3C_2$ –80NiCr was also exposed to 700 °C for 50 cycles. Coating started degrading after the 1st cycle and as the number of cycles increased, the spalling of the specimen increased. At the end of the 50 cycles, the coating from the corners of the specimen was found to be destroyed. The image of  $20Cr_3C_2-80NiCr$ -coated specimen is shown in Fig. [1](#page-2-0)c. Another coated specimen  $10Cr_3C_2-90NiCr$  when exposed to 700 °C for 50 cycles turned yellowish in colour after 5th cycle and sides got spalled at the end of the 30th cycle. The image of  $10Cr_3C_2-90NiCr$  is shown in Fig. [1d](#page-2-0). Last coated specimen 100NiCr when exposed to 700 °C for 50 cycles showed better results as compared to other coatings. At the end of 30th cycle, reddish and brownish colour was noted on surface and after the 50th cycle it was observed that the coating of the specimen was safe. The image of 100%NiCr-coated specimen is shown in Fig. [1e](#page-2-0).

#### **3.2 Thermogravimetric Study**

The thermogravimetric study was done for uncoated,  $35Cr_3C_2-65NiCr$ ,  $20Cr_3C_2-80NiCr$ ,  $10Cr_3C_2-90NiCr$ , and 100NiCr for low temperature (700 $\degree$ C) hot corrosion. Weight changes per unit area  $(mg/cm<sup>2</sup>)$  versus time expressed in number of cycles for uncoated and coated T22 boiler steel after exposing to the molten salt environment

 $(60V_2O_5 - 40Na_2SO_4)$  in a silicon tube furnace at 700 °C for 50 cycles is presented in Fig. [2](#page-2-1). From the graph, it is observed that the uncoated,  $35Cr_3C_2-65NiCr$ , and 100NiCr showed the high rate of corrosion as compared to  $20Cr_3C_2-80NiCr$ and  $10Cr_3C_2$ –90NiCr, which means that  $20Cr_3C_2$ –80NiCr and  $10Cr_3C_2$ –90NiCr-coated specimen provided the higher



<span id="page-2-1"></span>**Fig. 2** Weight gain/area versus time (number of cycles) for the uncoated and coated specimen during the experimentation



<span id="page-3-0"></span>**Fig. 3** (Weight gain/area)<sup>2</sup> versus time (number of cycles) for the uncoated and coated specimens during the experimentation where Coating  $1=35Cr_3C_2-65NiCr$  coating, Coating  $2=20Cr_3C_2-80NiCr$ coating, Coating  $3=10Cr_3C_2-90NiCr$  coating, Coating  $4=100NiCr$ coating

resistance to corrosion as compared to uncoated and other coated specimens. The cumulative weight change square per unit area  $(mg^2/cm^4)$  for 50 cycles is shown in Fig. [3.](#page-3-0) The cumulative weight change square after 50 cycles of hot corrosion for uncoated,  $35Cr_3C_2-65NiCr$ , and 100NiCr were found to be 3430, 714, and 469  $mg^2/cm^4$ , respectively, while the weight change square for  $20Cr_3C_2-80NiCr$  and  $10Cr_3C_2$ -90NiCr were 4 and 1 mg<sup>2</sup>/cm<sup>4</sup>.

## **3.3 XRD Analysis of T22 Uncoated and Coated After Exposed at 700 °C in Molten Salt Environment**

T22-uncoated and HVOF-coated specimens after exposed to 700 °C in molten salt environment in silicon tube furnace were analysed by X-ray difraction technique. The XRD analysis of uncoated specimen shows the presence of  $Fe<sub>2</sub>O<sub>3</sub>$ after exposure to molten salt environment for 50 cycles. The XRD analysis of HVOF-coated specimens shows the presence of (a) Ni, (b) Cr, (c) NiC, (d)  $Cr_3C_2$ . The presence of  $Cr_3C_2$  on the surface of the coated specimens helps to resist the corrosion. The XRD analysis of T22 uncoated and coated specimens is shown in Fig. [4.](#page-3-1)

## **3.4 SEM/EDAX Analysis of T22 Uncoated and Coated After Exposed at 700 °C in Molten Salt Environment**

SEM/EDAX analysis of T22-uncoated and coated specimens were done after exposed to 700 °C in molten salt environment. SEM/EDAX analysis of T22-uncoated specimen shows the formation of Fe and O on the surface of the specimen as the main elements and these elements form  $Fe<sub>2</sub>O<sub>3</sub>$ .



<span id="page-3-1"></span>**Fig. 4** XRD analysis of T22-uncoated and HVOF-coated specimens

SEM/EDAX analysis of coated specimens shows the occurrence of Cr, O, and C as main element. SEM/EDAX analysis of uncoated and coated specimens is shown in Fig. [5.](#page-4-0)

## **4 Discussion**

The high rate of corrosion was indicated by the macrographs of T22 uncoated,  $35Cr_3C_2-65NiCr$ , and 100NiCr. The weight changes graphs showed the high change in weight for T22 uncoated,  $35Cr_3C_2-65NiCr$ , and 100NiCr during exposure to molten salt environment in silicon tube furnace at 700 °C. The reason for high change in weight was the formation of  $Fe<sub>2</sub>O<sub>3</sub>$  on the surface of T22-uncoated specimen. The macrographs of uncoated specimen show that the oxide scales which formed on the surface were porous in nature and this porous nature was responsible for the high rate corrosion of T22-uncoated specimen. XRD analysis also confirmed the formation of  $Fe<sub>2</sub>O<sub>3</sub>$  for the T22-uncoated specimen after exposing to molten salt environment in silicon tube furnace at 700 °C and shows  $Fe<sub>2</sub>O<sub>3</sub>$  as the main phase. SEM/EDAX analysis of T22-uncoated specimen shows the formation of Fe and O on the surface of the specimen as the



<span id="page-4-0"></span>**Fig. 5** SEM and EDAX analysis of **a** T22 uncoated, **b** 35Cr<sub>3</sub>C<sub>2</sub>–65NiCr, **c** 20Cr<sub>3</sub>C<sub>2</sub>–80NiCr, **d** 10Cr<sub>3</sub>C<sub>2</sub>–90NiCr, **e** 100NiCr after exposed to 700 °C in molten salt environment

main elements. The formation of oxides of iron has also been reported by various authors [[32,](#page-6-5) [35](#page-6-6)[–37\]](#page-6-7). The EDAX analysis also shows Ni and Cr in the scale, and these elements might have difused from the base alloy at high temperature of exposure.

The coatings of  $20Cr_3C_2-80NiCr\text{-}coated$  and  $10Cr<sub>3</sub>C<sub>2</sub>$ -90NiCr-coated specimens were intact after exposure to molten salt environment in silicon tube furnace at 700 °C for 50 cycles.  $20Cr_3C_2-80NiCr$ -coated and  $10Cr_3C_2$ –90NiCr-coated specimens provided the high resistance to corrosion and did not show oxidation attack. The smooth scale was formed on the surfaces of these two coated specimens. The penetration of corrosive species was resisted by splats parallel to the substrate surface [\[5](#page-5-2), [38](#page-6-8)]. The weight changes graph shows that the  $20Cr_3C_2-80NiCr\text{-}coated$  and  $10Cr_3C_2-90NiCr\text{-}coated$  specimens were capable of reducing the weight change as compared to uncoated,  $35Cr_3C_2-65NiCr$ -coated, and 100NiCr-coated specimen. The lowest weight change was found for  $10Cr_3C_2$ –90NiCr-coated specimen, whereas the highest weight change was found for  $35Cr_3C_2-65NiCr$ among diferent types of coatings used. The XRD analysis of coated specimens shows the occurrence of (a) Ni, (b) Cr, (c) NiCr, and (d)  $Cr_3C_2$  after exposing to molten salt environment in silicon tube furnace at 700 °C.

The coated specimens 100NiCr,  $10Cr_3C_2-90NiCr$ , and  $35Cr_3C_2-65NiCr$  show the presence of (a) Ni and (b) Cr. The formation of NiC and  $Cr_3C_2$  might be the reason for lower corrosion rate of these coatings as compared to T22 uncoated specimen. The development of  $Cr_3C_2$  on the surface of 100NiCr-coated,  $10Cr_3C_2-90NiCr$ -coated, and  $35Cr_3C_2$ –65NiCr-coated specimens was further analysed by

SEM/EDAX and showed Cr, O, and C as the main elements after exposing to molten salt environment in silicon tube furnace at 700 °C. The rate of weight gain of coated specimens was reduced in comparison to uncoated specimen by providing the high resistance to corrosive species. SEM/EDAX analysis traced the presence of Ni and Cr on the surface of coated specimens; this was due to penetration of these elements to the coating surface from substrate at low temperature (700 °C) in molten salt environment.

# **5 Conclusions**

The following conclusions are drawn from this experimental study:

- 35 $Cr_3C$ –65NiCr, 20 $Cr_3C$ –80NiCr, 10Cr<sub>3</sub>C–90NiCr, and 100NiCr coatings were deposited by HVOF-sprayed technique.
- During the coating of T22 boiler steel specimens, oxygen was difused into coating.
- All the coatings used in study provide the better resistance to corrosion in molten salt environment when exposed for 50 cycles in silicon tube furnace. The sequence of coatings provided the resistance to corrosion in decreasing order shown below:
	- $10Cr<sub>3</sub>C<sub>2</sub>$ –90NiCr-coated T22>20Cr<sub>3</sub>C<sub>2</sub>–80NiCr-coated T22 > 100NiCr-coated T22 > 35 $Cr_3C_2$ –65NiCr-coated T22.
- Uncoated T22 specimen shows the highest change in weight during the exposure to hot corrosion.
- Spalling and peeling of uncoated T22 during the exposure to molten salt environment in silicon tube furnace at 700 °C. Formation of  $Fe<sub>2</sub>O<sub>3</sub>$  on the surface of uncoated T22.
- $10Cr_3C_2$ –90NiCr and  $20Cr_3C_2$ –80NiCr coatings provided the higher resistance to corrosion as compared to 100NiCr and  $35Cr_3C_2-65NiCr$ . The formation of chromium carbide layer on the coated surface protects them from corrosion.

## **Compliance with Ethical Standards**

**Conflict of interest** The authors declare that they have no confict of interest.

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