Research Article

Rapid synthesis of nanocrystalline SnO₂ by a microwave-assisted **combustion method**

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Abstract: A facile and rapid microwave-assisted combustion method was used to synthesis nanocrystalline $SnO₂$ powders, through dissolution of tin nitrate (as oxidant) and glycine (as fuel) as starting materials and water as solvent and then heating the resulting solution in a microwave oven. The study suggested that application of microwave heating to produce the nanosize $SnO₂$ was achieved in a few minutes. The structure and morphology of the as-prepared combustion products were investigated by means of powder X-ray diffraction (PXRD) and scanning electron microscopy (SEM). Fourier transform infrared spectroscopy (FTIR) and Raman spectra confirmed the formation of tetragonal rutile structure $SnO₂$, and the SEM results indicated the surface characteristic of the products. The as-prepared powders have larger band gap energy as 3.67 eV.

Keywords: nanoparticles; tin oxide; glycine; stoichiometric; combustion; optical

1 Introduction

Semiconducting nanoparticles have been extensively studied from both experimental and theoretical viewpoints because of their potential applications in solar energy conversion, photo catalysis and optoelectronics industry. Nanostructured semiconductors have unusual optical, electronic and chemical properties and wide potential uses as nanoscale devices. The desire for miniaturization is the driving force behind nanoparticle synthesis efforts. Research on the preparation of oxide nanoparticles is an active area being pursued all over the world. Tin oxide $(SnO₂)$ is an important n-type semiconductor with large band gap and high sensitivity to various toxic or flammable gases, and has been used as one of the promising materials for gas sensors, optoelectronic devices and negative electrodes for lithium batteries [1]. Its ability to transmit visible light or to absorb infrared makes $SnO₂$ a very attractive material for light conductive films and solar cell coatings [2]. For such applications, nanosize particles are required. Recent research shows that the properties of $SnO₂$ depend greatly on product's crystallite size and specific surface area [3]. Compared with bulk $SnO₂$, nanosize $SnO₂$ has especially good properties and outstanding advantages of low operating temperature and high sensitivity for gas-sensing applications [4].

Many methods have been developed to synthesize $SnO₂$ nanoparticles, including evaporation of $SnO₂$ or SnO powders at elevated temperatures [5], solvothermal and hydrothermal routes [6],

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Compared to the conventional methods, microwave synthesis has the advantages of producing small and uniform particle size of metal oxides with high purity owing to short reaction time. Microwave-assisted synthesis is a novel method to produce materials, since microwave heating is an *in situ* mode of energy conversion and the microwave heating process is fundamentally different from conventional heating processes. Heat will be generated internally within the material, instead of originating from external sources. By means of this method, many functional materials and compounds with novel structures and properties have been obtained [12].

As for as the literature survey, microwave-assisted combustion synthesis of tin dioxide colloidal suspensions from aqueous solutions of tin nitrate in the presence of glycine is reported for the first time. This method is simple, and the combustion synthesis process is fast, mild, energy-efficient and environmentally friendly to produce $SnO₂$ nanoparticles using glycine as fuel.

2 Experimental

All the reagents used in the experiments were analytically pure, and were purchased from MERCK Company and used as received without further purification. Nanocrystalline $SnO₂$ was synthesized by a microwave-assisted combustion synthesis process using glycine as fuel. A schematic representation of the synthesis process used in the current study is graphically shown in Fig. 1. Stoichiometric amounts (based on the thermochemical concepts from propellant chemistry calculation of fuel-to-oxidant mixtures) of tin nitrate and glycine were dissolved in deionized water and poured into a quartz container. Further they were mixed well by magnetic stirring for 1 h, making them almost as homogeneous mixture which was placed in a domestic microwave oven (BPL India Limited, Bangalore, India, Model No. IFB, 17PG1S, microwave 700 W, input range 210–230 V, AC 50 Hz, microwave frequency 2.45 GHz). Initially, the solution boiled and underwent dehydration followed by decomposition with the evolution of large

Fig. 1 Schematic representation of the synthesis process.

amount of gases (such as $CO₂$, H₂O and N₂ formed directly from the reaction between fuel and oxidant) with white fumes coming out from the exhaust opening provided on the top of the microwave oven. After the solution reached the point of spontaneous combustion, it began burning and released lots of heat, vaporized all the solution instantly and became foamy white solid powders.

The synthesized $SnO₂$ powders' phase formation was identified by powder X-ray diffraction (PXRD) method using X'Pert PRO PANalytical diffractometer with nickel-filtered Cu K α radiation (λ = 0.15418 nm) as source and operated at 40 kV and 30 mA. The sample was scanned in the 2θ angle ranging from 10° to 80° with minimum step size (0.05°) and minimum scan step time (10.1382 s) mode. The observed peak positions were compared with the standard ICDD data, and Miller indices were assigned to the Bragg peaks. The structure of the as-prepared $SnO₂$ powders was solved using PXRD method with refinement by Rietveld method. Scanning electron microscopy (SEM) and composition analysis of the sample were obtained using Hitachi S4800 scanning electron microscope equipped with an energy-dispersive X-ray spectrometer (EDX). SEM measurements were mounted on aluminum studs using adhesive graphite tape and sputter coated with gold before analysis. Infrared spectra were recorded on a Nicolet Avatar 360 FTIR spectrometer using KBr pellets. Raman spectra were taken on a Renishaw model Nicolet FT-Raman 960 Spectrometer $(Ar^+$ laser, 480 nm). UV–Vis (ultraviolet–visible) absorption spectra were recorded on a Shimadzu UV-2550 spectrophotometer.

3 Results and discussion

The phase identification of the as-prepared $SnO₂$ was determined by PXRD. Figure 2 shows the PXRD pattern of the as-prepared $SnO₂$. It is indicated as the diffraction planes (110), (101), (200), (211), (220), (002), (221), (112), (301), (202) and (321) in the pattern, which can be perfectly indexed to the tetragonal rutile structure. No diffraction peak corresponding to Sn and other impurities is observed in the pattern. The values of lattice parameters $(a =$ 4.7710 Å and $c = 3.2087$ Å) were calculated by the PXRD data, which are in good agreement with the standard diffraction pattern (04-008-8133). Using Scherrer's equation $D = 0.9\lambda / (\beta \cos \theta)$ (where *D* is the average crystalline size; λ is the wavelength of Cu Ka radiation; β is the full width at half maximum (FWHM) of the diffraction peak; and θ is the Bragg's angle), all the half peak widths are very broad, which indicates that the obtained crystallites are small in size, i.e., nanometer size. These results were confirmed by transmission electron microscopy (TEM) observation. The average particle size of $SnO₂$ is estimated as around 8 nm.

The Rietveld refinement for the investigated $SnO₂$ is summarized in Table 1. It shows that the refined lattice parameters of stibnite are in good agreement with the reported values. The results of Rietveld analysis demonstrate the tetragonal rutile structure as shown in Fig. 3. In general, R_p (profile factor), R_{wp} (weight

Fig. 2 PXRD patterns of $SnO₂$ synthesized by a microwave-assisted combustion method.

Space group	$P4_2/mnm$ (136)			
Cell parameters (Rietveld)	a(A)		4.7907(2)	
	c(A)		3.2179(3)	
	$V(\AA)$		73.6784 (3)	
Crystal density $(g/cm3)$	5.850			
Atomic positional parameters	x			B (iso)
Sn(2a)	0	0		2.436
O (4f)	0.312	0.312		2.631

The structure was refined with the following R factors: R_{exp} (expected weight profile factor, %) = 16.10, R_{wp} (%) = 10.63, R_p (%) = 10.36, R_{Bragg} (Bragg factor, $\%$) = 1.021, DW(Durban–Watson statistic, $\%$) = 0.13 and $S(\%)=0.66$.

Fig. 3 Observed (blue line), calculated (red line) and difference profiles (black line below) resulting from the adjustment of the de rayos X (DRX) curves of the Rietveld analysis for glycine fuel used SnO₂.

differences between measured and calculated value) and *S* (good average factor) show decreasing trend with the increase of the weight fraction of albite and suggest that the pattern fitting is accurate. The Rietveld fitting of the $SnO₂$ phase was performed on the tetragonal rutile structure with space group $P4₂/mm$ (136). It is formed by a tin atom at the center, surrounded by six oxygen atoms in the vertex. The tin atom is bounded to four oxygen atoms with the same bond length in the basal plane side and to another two apical oxygen atoms with another same bond length. Rietveld refinement is probably the method of choice for determining accurate unit-cell parameters. The refinements of lattice parameters of these phases are $a = 4.7907 \text{ Å}$, $c = 3.2179 \text{ Å}$ and $V = 73.6784 \text{ Å}^3$ for all samples, while the calculated lattice parameters of $SnO₂$ are slightly higher than the bulk values with respect to the expected ones.

Figure 4(a) shows the SEM micrograph of the as-prepared $SnO₂$ synthesized by a microwave-assisted combustion process. It can be observed that ultrafine SnO2 particles with nano sized dimension are interconnected, which shows strong agglomeration accompanied with a lot of small spherically shaped particles. This agglomerate actually consists of much smaller grains of about 10–20 nm in diameter. These morphology results can be attributed to the liberation of a large amount of gases during a strong combustion reaction, i.e., the rapid release of large volume of gaseous products like H_2O vapor, CO_2 and N_2 , and nitrate ion group acts as an igniter during combustion reaction which promotes the disintegration of the precursor, giving nanocrystalline particles. It is helpful for the finest particle size formation. The agglomerated particles have neck with their neighbors, and they form into spherical shapes intermitted with voids ensuring high surface area. The size can not be exactly determined by the image because they are very small. In spite of that, it is still found from the particle-like morphology. In addition, the EDX analysis (Fig. 4(b)) coupled with SEM confirms that the as-prepared nanocrystalline $SnO₂$ is composed of tin and oxygen elements, indicating that pure $SnO₂$ nanoparticles are obtained. The particle size measured by SEM is close to the average particle size obtained from XRD–Rietveld method. The as-synthesized particles are shown in the TEM photograph of nearly tetragonal rutile nanocrystalline $SnO₂$ particles (Fig. 5). These sizes are well agreed with the Rietveld refinement

Fig. 4 (a) SEM micrograph and (b) EDX spectrum of $SnO₂$ powders synthesized by a microwave-assisted combustion method.

method. The corresponding selected area electron diffraction (SAED) image is shown as inset in the TEM image. The diffraction rings clearly confirm that tetragonal structure of $SnO₂$ has been synthesized successfully. These $SnO₂$ particle sizes based on XRD and Rietveld refinement results are in quite good agreement with each other and are further confirmed by the particle size estimated by TEM.

The chemical structure of the as-prepared $SnO₂$ was investigated by FTIR and Raman spectra. Figure 6 shows the FTIR spectrum of the as-prepared nanocrystalline $SnO₂$. The absorption bands located in the spectral region of 400–700 cm^{-1} are caused by different Sn–O or Sn=O stretching modes. The absorption band located around \sim 460 cm⁻¹ has been ascribed to the stretching frequency of oxide group

Fig. 5 TEM image of the synthesized $SnO₂$ nanoparticles.

Fig. 6 FTIR spectrum of $SnO₂$ powders.

($v_{\rm SnO}$). The peaks observed at ~1219 cm⁻¹ in the FTIR spectra may be assigned to a vibration of hydroxyl–tin bonds of different organic types of surface hydroxyl groups. A broad peak in the range between 1230 cm^{-1} and 1430 cm^{-1} demonstrates that a small amount of adsorbed water and $NO³$ ions exist in the powder particles. A strong peak centered at \sim 2355 cm⁻¹ indicates that the tin complex transforms into tin oxide.

It is known that Raman scattering is very sensitive to the microstructure of nanocrystalline materials, and it is also used here to clarify the structure of the $SnO₂$ nanopowders. As shown in Fig. 7, the first-order Raman spectrum of the as-prepared $SnO₂$ shows the sharpest and strongest peaks at about $473-475$ cm⁻¹, 631–633 cm⁻¹ and 773–775 cm⁻¹, which are in good agreement with those of tetragonal rutile structure of $SnO₂$ with point group D_{4h} [13], a typical Raman active branch of tetragonal rutile phase of nanocrystalline $SnO₂$ originated from the non-polar optical phonon E2 (high) mode [14].

The UV–Vis spectrum of nanocrystalline $SnO₂$ is shown in Fig. 8. The absorption peak shoulder onset

Fig. 7 Raman spectrum of $SnO₂$ powders.

Fig. 8 UV–Vis spectrum of $SnO₂$ powders.

peaks are located at 340 nm corresponding to the band gap of 3.67 eV. Compared with the other methods (like sol–gel, co-precipitation, etc.), nanocrystalline $SnO₂$ obtained by this microwave-assisted combustion method has larger band gap energy [13,14]. In general, the hexagonal wurtzite $SnO₂$ structure has a direct band gap energy (E_{α}) for the SnO₂ powders and can be determined by extrapolation to the zero absorption coefficient (α) which is calculated from the following equation:

$$
\alpha = \frac{2.303A\rho}{IC} \tag{1}
$$

where *A* is the absorbance of the sample; ρ is the density of $SnO₂$ powders; *C* is the concentration of the particles; and *I* is the optical path length. The optical absorption coefficient (α) of a semiconductor is close to the band edge which is estimated from the Tauc's relationship [15] as follows: 2

$$
(\alpha h v)^2 = A(hv - E_g) \tag{2}
$$

where α is the absorption coefficient; *h* is the Planck's constant; ν is the frequency of the incident photon; E_g is the optical direct band gap energy; and *A* is a constant. The exact value of the band gap is calculated by extrapolating the straight line portion of the $(\alpha h v)^2$ vs. *hv* curve. Plotting $(\alpha h v)^2$ as a function of photon energy and extrapolating the linear portion of the curve to the photon energy axis (absorption equal to zero) are shown in inset of Fig. 8.

The dependence of particle size of the $SnO₂$ powders can be determined experimentally from the band gap energy inferred from the optical absorption spectra, which is expressed from an effective mass model [16]. The quantum size effect in the $SnO₂$ powders can be described from the following equation:

$$
E_{\rm g}^{\rm (nano)} = E_{\rm g}^{\rm (bulk)} + \frac{h}{2D^2} \left(\frac{1}{m_{\rm e}^* + m_{\rm h}^*} \right) - 0.248 E_{\rm RV}^* \tag{3}
$$

where $E_{\rm g}^{\rm (nano)}$ is the band gap energy of the synthesized SnO₂ particles; $E_g^{\text{(bulk)}}$ is the band gap energy of the bulk material as 3.6 eV; *h* is the Planck's constant; *D* is the average crystalline size calculated from PXRD results of $SnO₂$ powders; the electron and hole effective masses are taken as $m_e^* = 0.299 m_0$ and $m_e^* = 0.234 m$; and the hully existen hinding approxy $m_h^* = 0.234 m_0$; and the bulk exciton binding energy E_{RY}^* is 130 meV. After simplification of the equation, the band gap energy value is $E_{\text{g}}^{\text{(nano)}} = E_{\text{g}}^{\text{(bulk)}} +$ 7.0989 D^{-2} . The optical band gap energy (E_g) was calculated using effective mass equation and from

Tauc's relation as 3.67 eV and 3.71 eV, respectively. Hence the optical band gap energy value appears slightly lower than the calculated band gap energy value (effective mass model) due to a tight-binding model used in the experimental data (optical absorption spectra).

4 Conclusions

Nanocrystalline $SnO₂$ has been successfully synthesized by a microwave-assisted combustion method using glycine as fuel. PXRD, FTIR and Raman spectra confirm the formation of a tetragonal rutile phase. The crystal structural parameters are in good agreement with the literature values. Besides, there is no peak other than $SnO₂$ in the synthesized powders. FTIR and Raman spectra confirm the compound formation. SEM micrographs show the uniform particles present in the powders. EDX analyses show that the stoichiometric formation of $SnO₂$ nanopowders. From absorption spectrum, the band gap energy of the material is estimated as 3.67 eV which is comparable with the bulk value.

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