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Potato starch‑based bioplastics synthesized using glycerol–sorbitol blend as a plasticizer: characterization and performance analysis

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Abstract

The demand for petroleum-based plastics is a major threat and has had several negative impacts on the environment. Biodegradable plastics are becoming increasingly attractive due to their cost-efectiveness and eco-friendly properties. This study aimed to develop and evaluate the properties of biodegradable plastic flms produced using a low-cost starch source (potato) with a blend of glycerol and sorbitol as the plasticizers. A total of 10 bioplastic flms were prepared by varying the type and concentration of plasticizers at a ratio of 0%, 10%, 20%, 30%, and the glycerol–sorbitol blend mixture at a ratio of 2:1, 1:2, and 1:1 (wt%) using the solution casting technique. Several characterization techniques were used to determine the morphological, structural, and chemical properties of the as-synthesized bioplastic flms. Fourier transform infrared spectroscopy and morphology measurements verifed the confrmation of potato starch bioplastics. Bioflms with glycerol: sorbitol blend plasticizer exhibited higher thermal stability and maximum tensile stress than flms plasticized with glycerol. The results also showed that the flms with a blend plasticizer exhibited better compatible morphology than those plasticized with glycerol. The results indicate that the synthesized bioplastic samples have great potential for various applications, especially as packaging materials.

Keywords Bioplastic · Potato starch · Plasticizer · Glycerol–sorbitol blend · Biodegradability

Introduction

Globally, single-use plastics are used for a plethora of purposes. To meet the ever-increasing demand, plastic production increased from 1.5 million tons in 1950 to 335 million tons in 2016 (Plastics Europe [2018](#page-16-0)). Subsequently, global carbon dioxide emissions have increased due to the

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production of synthetic and semi-synthetic plastic polymers (Statista [2020\)](#page-16-1). Moreover, synthetic plastics have introduced potentially hazardous pollutants into the environment, and they enter the food chain in the form of microplastics (MPs) (Sarker et al. [2020\)](#page-16-2). Approximately 20–42% of total global plastics are currently deposited on the land (Zhang et al. [2019](#page-16-3)). Another 10% flow into aquatic environments, resulting in some ocean currents being termed 'marine garbage belts" (Thompson [2006\)](#page-16-4). These environmental concerns have led many researchers to develop biodegradable plastics. Agricultural waste has very often been indicated as a low-cost and renewable raw material alternative (Jain and Tiwari, [2015](#page-15-0)).

Starch is a semi-crystalline polymer consisting of two polysaccharides, amylopectin, and amylose. (Dufresne [2012](#page-15-1)). Amylopectin is a large branched polymer composed of α-1,4 and α-1,6 bonds located in the main polymer and side polymer chains. Amylose, on the other hand, is a linear or slightly branched polymer composed of glucose molecules interconnected mainly by α -1,4 glycosidic bonds. (Eliasson [2004](#page-15-2); Perez et al. [2009](#page-16-5)). Starch is a natural, renewable, and sustainable polymer isolated from many potential botanic

resources such as wheat, corn (maize), rice, potato, and tapioca (Perez et al. [2009\)](#page-16-5). During photosynthesis, starch is produced in plants as the reserve food supply (Chen et al. [2013](#page-15-3)). Therefore, the beneft of starch-based biodegradable plastic is its abundance of raw materials and easy manufacturing process with low cost (Avella et al. [2009](#page-15-4)). Starch also has favorable thermoplastic properties and is biodegradable (Imre and Pukánszky [2015\)](#page-15-5). Furthermore, starch-based bioplastic flms showed tremendous optical (transparent, colorless), organoleptic (favorless, odorless), and barrier properties $(O_2$ and CO_2 permeability) (Thakur et al. [2019](#page-16-6)). On the other hand, the strong intermolecular hydrogen bonding between amylopectin and amylose macromolecular network chains in native starches makes them stable and brittle (Ma and Yu [2004](#page-15-6)). Therefore, starch-based bioplastics could be a promising alternative to hazardous petroleum-based plastics (Shafqat et al. [2020\)](#page-16-7).

However, the efficiency of starch as a raw material for bioplastics depends on its specifc structure and composition (Kaseem et al. [2012](#page-15-7)). Therefore, it is crucial to optimize the starch-based bioplastic flms so that they could satisfy the requirements of industrial applications. The addition of plasticizers increases the versatility of the starch-based bioplastics and, therefore, can be combined with several polymeric substances that ensure plentiful utility (Amin et al. [2019](#page-15-8)). However, the nature and constituents of plasticizers are precarious determinants that signifcantly impact bioflm properties (Vieira et al. [2011;](#page-16-8) Jost et al. [2014\)](#page-15-9). These plasticizers are usually small molecules such as polyols (sorbitol and glycerol) that scatter and intercalate among and between polymer chains (Khoramnejadian et al. [2013\)](#page-15-10), disrupting hydrogen bonding and spreading the chains apart, which escalate the bioflm fexibility, as well as intensifying water vapor and gas permeability (García et al. [2000\)](#page-15-11). Therefore, this study was conducted with two main objectives—(a) the preparation and characterization of bioplastics using potato starch and plasticizers (glycerol and sorbitol) and (b) to examine the infuence of plasticizers (glycerol and sorbitol) concentrations and/or combinations on the bioflm's functional bioflm properties.

Materials and methods

Materials

Fresh potatoes were obtained from a local market in Patan, Gujarat, India. Glycerol (G5516) and sorbitol (240,850) were purchased from Sigma Aldrich. Sodium hydroxide (MB095), chloroform (AS040), diethyl ether (AS126), methanol (AS059), and ethanol were purchased from HiMedia.

Extraction of starch from potato

Due to its higher suitability for alkali medium (Xu et al. [2018\)](#page-16-9), water was chosen as the medium for potato starch extraction. Briefy, 1 kg of fresh potato was thoroughly washed with tap water, peeled, grated, and then ground to a paste using a mortar pestle. The potato paste was volumed up to 1 L using distilled water and fltered twice using a cheesecloth. Then, this flm-forming solution was allowed to settle at room temperature for 2 h. After discarding the excess water, the residue was collected and dried in an oven at 35 °C for 5 h. Subsequently, the dried residue was powdered with a commercial blender, sieved, and stored at room temperature.

Preparation of bioplastic flms

Potato starch-based bioplastic flms were prepared by following the protocols described by Farahnaky et al. ([2013\)](#page-15-12) with slight modifcations. Briefy, 10 g of potato starch powder was mixed with 100 ml of distilled water and stirred mechanically for 15 min at room temperature (25 °C). Then, after adding plasticizer (glycerol or sorbitol) at diferent concentrations (0–30% w/w starch basis, these solutions were heated on a hot plate at 90 ± 2 °C with constant stirring (500 rpm) for 20 min to ensure complete gelatinization. Again, the glycerol–sorbitol blend was added at the ratios of 2:1, 1:2, and 1:1 (wt%), and these mixers were heated following the similar procedures described above. Later, the mixture was evenly spread on a Petri plate and allowed to dry in a hot air oven at 50 °C. The dried bioplastic films were peeled off from the plate and stored in desiccators until further study. The list of synthesized bioplastic samples is shown in Table [1.](#page-2-0)

Characterization

Fourier transform infrared (FTIR) spectra of prepared flms were recorded by a Perkin Elmer spectrophotometer at wavenumbers ranging from 4000 to 500 cm^{-1} . There were peaks at diferent positions on the graph, which confrmed the presence of diferent functional groups in the synthesized bioplastic samples. The surface of the bioflms was studied using a scanning electron microscope (SEM) at an accelerated voltage of 15 kV and up to 2000X magnifcation. For the SEM analysis, the bioflm samples were cut into 1×1 cm and dried in a hot air oven at 50 °C for 5 h. Thermogravimetric analysis (TGA) was executed using the Autograph AG100KNG universal testing machine. The analysis was carried out in a temperature

Sample ID	Plasticizer	Concentration $(\%)$	Appearance Brittle, hard, cracks on plastic surface, non-transparent		
$F-0$		0			
$FG-10$	Glycerol	10	Less surface cracks, slightly sticky, non-transparent		
$FG-20$	Glycerol	20	Semi-transparent, cracks on the surface, easy to peel, sticky		
$FG-30$	Glycerol	30	Semi-transparent, cracks on the surface, easy to peel, sticky		
$FS-10$	Sorbitol	10	No surface cracks, more transparent than FG-10, slightly sticky, semi-transparent		
$FS-20$	Sorbitol	20	Non-transparent, cracks on the surface, easy to peel, non-sticky		
$FS-30$	Sorbitol	30	Non-transparent, cracks on the surface, easy to peel, non-sticky		
$FG2-S1$	$Glycerol + Sorbitol$	2:1	Brittle, fragile, hard, cracked		
$FG1-S2$	$Glycerol + Sorbitol$	1:2	Brittle, fragile, non-transparent, cracks on the surface		
$FG1-S1$	$Glycerol + Sorbitol$	1:1	No surface cracks, leathery, transparent, non-sticky, elastic		

Table 1 List of synthesized bioplastic flms with various combinations and concentrations of plasticizers

range of 25–800 °C at a 10 °C/min rate and a flow rate of 20 ml/min.

The tensile test was performed using the Autograph AG100KNG universal test machine (Shimadzu, Kyoto, Japan) according to the standard test method for the tensile properties of plastic sheets (ASTM D882-02). The protocol described by (Krishnamurthy and Amritkumar [2019](#page-15-13)**)** was followed to adjust machine parameters, specimen settings, and conditioning. The bioflm samples were cut into $100 \text{ mm} \times 25 \text{ mm}$ in dumbbell shape with a gauge length of 50 mm. The cross-head speed was modifed and set at 3 mm/min. Three specimens were tested for each sample.

Evaluation of bioplastic properties

The thickness of the bioplastic films was determined according to Oluwasina et al. ([2017](#page-16-10)) with slight modifcations. Briefy, the bioflms were cut into small pieces $(2 \text{ cm} \times 2 \text{ cm})$, and then, 15 measurements were taken at diferent points in the bioplastic flms using a digital Vernier caliper. The thickness of the bioflms was calculated using the following equation.

$$
Thickness = \frac{Sum of measured values}{15}
$$

The mass (*M*), area (*A*), and thickness (*d*) of the bioplastic flms were determined. Then, the density of the bioflms was calculated using the equation below (Salgado et al. [2010\)](#page-16-11).

Density =
$$
\frac{M}{A \times d}
$$

The moisture content was calculated according to the procedures described by Oluwasina et al. ([2017\)](#page-16-10). The initial weight (*W*1) of the bioplastic films (2 cm \times 2 cm)

was measured, these films were oven-dried at 105 °C for 3 h, and then, the fnal weight (*W*2) was determined. The moisture content (MC) was determined using the following equation.

$$
MC(\%) = \frac{W1 - W2}{W1} \times 100
$$

To perform water absorption tests on synthesized bioplastic flms, a protocol previously described by Oluwasina et al. [\(2015\)](#page-16-12) was followed by slight modifcations. Briefy, the weight of the oven-dried (for 3 h at 105° C) biofilm samples was measured, and then, the flms were soaked in water for 24 h at room temperature (25 $^{\circ}$ C) with occasional agitation. Then, after the adhered water from the bioplastic flms, the weight (Wt) of these flms was measured again. The moisture absorption by the bioflms was calculated using the equation below.

Moisture Absorption(%) = $\frac{Wt - W0}{W0} \times 100$

Solubility and biodegradability tests

Bioplastic solubility was checked in water, 10% NaOH, chloroform, 10% H₂SO₄, diethyl ether, methanol, and ethanol solvents at room temperature and elevated temperature (60 °C) by following the protocol described by Pimpan et al. with slight modifications (Pimpan et al. [2001](#page-16-13)). Because all samples were soluble in water at both temperatures, the water solubility of the bioplastic samples was quantitatively measured according to the following procedures. Briefy, bioplastic films were cut into 2 cm \times 2 cm, weighed, and immersed in 50 ml of the above-mentioned solutions for 24 h with constant shaking (120 rpm) at room temperature. Subsequently, the flms were separated and kept in the oven for 24 h at 105 °C to dry. The fnal weight was measured,

and the following equation was used to determine the percentage of weight loss.

$$
Weightloss(\%) = \frac{Initialweight - Finallyeight}{Initialweight} \times 100
$$

The biodegradability of the bioplastic samples was determined using the soil burial degradation test. Films were cut into small pieces $(2 \text{ cm} \times 2 \text{ cm})$ and weighed. The films were then buried in garden soil and weighed at regular intervals of 10, 15, 20, 25, and 30 days. Biodegradability was measured by comparing the fnal weight (Wf) with the initial weight (Wi) using the following equation.

$$
\text{Biodegradability}(\%) = \frac{Wf - Wi}{Wi} \times 100
$$

where *Wi* is the initial weight of the sample and *Wf* is the fnal weight after the specifc time.

Water vapor transmission rate (WVTR) test

The water vapor transmission test was conducted by utilizing the desiccant method according to the protocol described by Ng and colleagues (Ng et al. [2022\)](#page-15-14). Briefy, desiccant silica was placed within an impermeable plastic cup, which was then tightly sealed with plastic flm. A cup of distilled water and this dish were then placed inside a desiccator. Every 24 h, the sealed cups were weighed. The water vapor transmission rate (WVTR) was calculated using the following equation:

$$
WVTR = \frac{G}{tA}
$$

where *A* was the test area (m^2) and *G*/*t* was the slope of the straight line (g/h) calculated using linear regression.

X‑ray difraction (XRD) test

A Shimadzu 7000 X-ray difractometer was used for the XRD test, which was carried out using CuKα radiation $(\lambda = 1.5405 \text{ Å})$, recorded between 0° and 60°, and operated at 30 kV and 10 mA.

Diferential scanning calorimetry (DSC) test

The DSC test was carried out in accordance with the procedure given by (Yang et al. [2022\)](#page-16-14), with minor adjustments. Dry samples (approximately 5–10 mg) were scanned by a DSC 1/500 (Mettler-Toledo (M) Sdn Bhd) after being held at 45 °C for 1 min and then heated from 45 to 220 °C at a heating rate of 10 °C/min in a sealed aluminum pan. A flow rate of 10 mL/min of high-quality nitrogen gas was employed. The DSC thermograms were used to calculate

the glass transition temperature (Tg), melting temperature (Tm), and crystallization temperature (Tc).

Antimicrobial activity test

The disk difusion technique (Kirby Baeur) was used to assess antimicrobial activity according to the methodology provided by Agustin and Padmawijaya ([2017\)](#page-15-15). In a brief, 1-cm-diameter bioplastics were placed on tryptone soya agar media [g/L—tryptone 17, soya peptone 3.0, sodium chloride 5.0, dextrose (Glucose) 2.5, dipotassium hydrogen phosphate 2.5, agar 15, final pH (at 25° C) 7.3 ± 0.2]. The surface of this nutritional media was uniformly swabbed with the bacterial suspensions. Two gram-negative (*Salmonella typhi*, *Escheritia coli*) and three gram-positive (*Bacillus subtilis*, *Bacillus megaterium*, *Bacillus haynesii*) bacterial isolates (collected from the microbial culture stock of the Department of Life Sciences, Hemchandracharya North Gujarat University) were used in this test. The samples were then incubated for 24 h at 28 ± 2 °C. The inhibitory zone (clear zone) around the bioplastics samples was determined after incubation.

Results and discussion

Physicochemical properties of bioplastic flms

To evaluate the efect of various combinations of plasticizers in the prepared bioflms, F-0 (containing 0% plasticizer), FG-10 (containing 10% glycerol), FS-10 (containing 10% sorbitol), and FG1-S1 (containing glycerol: sorbitol ratio = 1:1) samples were selected. FTIR analysis helps to determine any potential chemical alterations in the bioplastic samples. Also, additionally, it allows for quick, authentic, and efficient determination of functional groups (Pavia et al. [2001\)](#page-16-15). In all samples, characteristic peaks (between 2900 and 3000 cm^{-1}), which signify the (C-H) stretching due to the presence of starch, were observed (Figs. [1](#page-4-0), [2,](#page-4-1) [3](#page-5-0), [4](#page-5-1) and Table [2](#page-6-0)). Previous studies have also reported similar peaks in starch-based bioplastic samples (Yin et al. [2005](#page-16-16); Shafqat et al. [2021\)](#page-16-17). In the case of the sample FG1-S1, peaks were observed that signify functional (O–H) groups at wavelength 1418 cm−1 (Fig. [4](#page-5-1)). This is because FG1-S1 was plasticized with both glycerol and sorbitol; both plasticizers are polyols (Ano et al. [2017\)](#page-15-16) that contain a huge number of (O–H) groups (Chen et al. [2015a,](#page-15-17) [b](#page-15-18)). The four bioplastic samples showed peaks $(C=C)$ at the wavelength between 892 and 1664 cm^{-1} 1664 cm^{-1} 1664 cm^{-1} 1664 cm^{-1} (Figs. 1, [2](#page-4-1), [3](#page-5-0), 4 and Table [2\)](#page-6-0), similar to a previous report in which bioplastic flms were generated using modifed cassava starch (Harsojuwono et al. [2018\)](#page-15-19). The 1,2,4-trisubstituted functional group was present only in the case of the samples plasticized with glycerol: sorbitol

combinations (Table [2](#page-6-0)). The change in specifc starch conformations induced by specifc plasticizer combinations, which leads to polymer disorder, could be attributed to the altered intensity of specifc peaks (Warren et al. [2016\)](#page-16-18).

Thermogravimetric analysis (TGA) of potato starchbased bioplastic samples is shown in Fig. [5.](#page-6-1) The TGA diagram shows that the bioplastic samples have a continuous degradation pattern with a two-step process of decompositions. In the frst step of decomposition, the light volatile compounds are lost and the evaporation of the water occurs at a temperature between 150 and 200 °C, which is similar to a previous report by Amin et al. (Amin et al. [2019](#page-15-8)). The decomposition graphs show that the T50% (the temperature at which 50% of weight loss occurred) for the flm F-0 is 250–270 °C (Fig. [5a](#page-6-1)), for FG-10 it is 240–250 °C (Fig. [5](#page-6-1)b), and for FS-10 and FG1-S1 it is 300–320 °C (Fig. [5c](#page-6-1) and d). For the FS-10 and FG1-S1 flms, a drastic weight loss was observed between temperatures 300 and 350 °C. In their study, Nurul et al. have also mentioned that for potato bioplastics, T50% are at 310 °C (Nurul et al. [2016\)](#page-16-19). However, all bioplastic samples become fully decomposed at 450–500 °C. The overall results of the thermal analysis have

shown that sorbitol-plasticized bioplastics have greater heat stability compared to those glycerol-plasticized bioplastics.

4000

3500

3000

2500

Wavenumber cm-1

DSC was used to investigate the thermal characteristics of starch bioplastic and bioplastic composites such as melting temperature (Tm) as illustrated in Figs. [6](#page-7-0), [7,](#page-7-1) [8](#page-8-0), [9.](#page-8-1) The F-0 sample had the highest melting temperature (141.61 $^{\circ}$ C) (Fig. [6\)](#page-7-0), followed by the FS-10 (138.46 \degree C, Fig. [8](#page-8-0)), FG1-S1 (135.11 °C, Fig. [9](#page-8-1)), and FG-10 (134.49 °C, Fig. [7\)](#page-7-1). The crystallization temperature in these four bioplastic samples ranged from 147.69 °C to 156.66 °C (Figs. [6](#page-7-0), [7](#page-7-1), [8,](#page-8-0) [9\)](#page-8-1), indicating that the starch-based bioplastics are semi-crystalline in nature (Amin et al. [2019\)](#page-15-8). We were unable to identify any distinct Tg in the DSC thermograms for the tested bioplastic samples. Previous research had demonstrated that the Tg of starch-based bioplastics was seldom detected by DSC (Yokesahachart and Yoksan [2011](#page-16-20)). As a result, our fndings are corroborated by previous reports.

1500

1000

500

2000

Evaluation of the mechanical properties of bioplastics is crucial for assessing their utility (Azahari et al. [2011;](#page-15-20) Spiridon et al. [2013\)](#page-16-21). Plasticizers significantly afected the tensile strength of the bioplastic samples (Fig. [10\)](#page-9-0). The highest elongation (74.38%) was found in

Sample name	Func- tional groups	Wave- length (cm^{-1})	Name of the functional groups
$F-0$	$C = C$	1663	Alkene
	$C=O$	1152	Tertiary alcohol
	C–Cl	948	Halo compound
$FG-10$	$C = C$	1664	Alkene
	$C-O$	1241	Alkyl aryl ester
	$C = C$	847	Alkene
$FS-10$	$S=O$	1376	Sulfate
	$C=O$	1087	Secondary alcohol
	$C-O$	1053	Primary alcohol
	$C = C$	892	Alkene
$FG1-S1$	O-H	1418	Alcohol
	$C = C$	998	Alkene
	$C-H$	862	1,2,4-Trisubstituted

Table 2 FTIR absorption peaks for the selected bioplastic samples

the FS-10 samples where sorbitol was used as a plasticizer (Fig. [10A](#page-9-0)). In FG1-S1 samples, the elongation was reduced to 13.56%, but the tensile strength was highest in this samples (4.12 MPa) (Fig. [10B](#page-9-0)). The results of the potato starch-based bioplastics tensile test have shown that the samples that have greater mechanical strength have less fexibility. Previous fndings on plasticized bioflms of glycerol and sorbitol have shown similar trends for the tensile test (Ooi et al. [2012](#page-16-22); Tapia-Blácido et al. [2013](#page-16-23)). The higher plasticization capacity of glycerol over sorbitol can be attributed to the fact that the molecular mass of glycerol is almost half that of sorbitol, therefore allowing the molecules of glycerol and starch to interact more efficiently (Shafqat et al. [2021](#page-16-17)). However, in the sample where a combination of the two plasticizers is added, both impact the bioplastic tensile simultaneously (Sanyang et al. [2016](#page-16-24)).

The XRD was employed to investigate the crystalline structures of the starch-based bioplastic samples. The results obtained from XRD analysis indicate the major pick at near 20° for all four bioplastic samples (Fig. [11](#page-10-0)A–D). Starch is a semi-crystalline material consisting of crystal units and amorphous units (Zhong and Xia [2008\)](#page-17-0). There are two components in starch, namely amylose, which has a linear chain structure, and amylopectin, which has a branch chain structure (Zhaosheng et al. [2008](#page-17-1)). Therefore, our fndings on the XRD analysis of the bioplastic samples are standard for starch-based materials (Jangong et al. [2019;](#page-15-21) Sangian et al. [2021](#page-16-25)).

Fig. 5 TGA diagram of the potato starch-based bioplastics: **A** F-0, **B** FG-10, **C** FS-10, and **D** FG1-S1

Fig. 6 DSC thermograms of F-0 bioplastic samples

Fig. 7 DSC thermograms of FG-10 bioplastic samples

Morphology

Figure [12](#page-11-0)A–D represents the digital images of the bioplastic flms. The physical appearance of the bioplastics with diferent plasticizer concentrations is shown in Supplementary Figure S1. However, bioplastic should be tough enough to resist load-bearing pressure. The toughness could be assessed by the physico-tensile properties of bioplastics

Fig. 8 DSC thermograms of FS-10 bioplastic samples

Fig. 9 DSC thermograms of FG bioplastic samples

such as thickness, density, and tensile strength. As can be seen in the results from Fig. [12](#page-11-0)E, the bioplastic flms formulated by the glycerol:sorbitol combination have the highest thickness followed by the sorbitol and glycerol-based samples. The lowest thickness was found in the samples without any plasticizer. The thicknesses obtained from the flms in

Fig. 10 A Stress vs strain curve, and **B** tensile strength of the bioplastic samples. Here, Mpa=Megapascal

this work were greater than the 0.087–0.091 mm of potato starch nano-silica bioplastics reported by the Zhang et al. group Zhang et al. ([2018\)](#page-16-26). However, the diferences in the thickness values recorded in this study may be related to the type and percentage concentration of added plasticizers (Oluwasina et al. [2021\)](#page-16-27).

Furthermore, a directly proportional relationship between the thickness of the bioplastic flm and the density was found (Fig. [12](#page-11-0)E). The high densities of bioplastic flms may be related to the larger particle size of the bioplastic flms (Reddy and Rhim [2014](#page-16-28)). Other fndings have shown a correlation between increased density and the concentration of added plasticizers (Slavutsky et al. [2012](#page-16-29); Abdullah et al. [2019](#page-15-22)).

SEM micrographs are shown in Fig. [13](#page-12-0)A–H, which help to visually analyze the surface morphology and microstructure of the potato starch-based bioplastic samples. The study anticipated a comparison of bioplastics developed using diferent combinations of plasticizers. The results showed that the surface features consist of granules (the remaining part of the starch particle), which means that the starch was not fully gelatinized during the formation process (Fig. [13](#page-12-0)). These fndings are similar to previous reports describing SEM images showing the presence of insoluble remains (i.e., ghosts) from the swelling starch granules (Hern'andez et al. [2017\)](#page-15-23). The glycerol-based bioplastic sample (FG-10)

contained some voids (Fig. [13](#page-12-0)D) that may have contributed to lower impact and tensile strength (Amin et al. [2019](#page-15-8)). However, the SEM images show that the potato starch and sorbitol plasticizers were relatively well homogenized, as the samples with sorbitol (FS-10 & FG1-S1) showed no phase separation, voids, or cracks (Fig. [13](#page-12-0)E, F). However, the incomplete gelatinization in bioplastic samples could be linked to the molecular rearrangements initiated by diferent starch: plasticizer ratios, which can induce amorphous, semi-crystalline, and crystalline arrangements of starch in the bioplastics (Capron et al. [2007](#page-15-24)).

Moisture content and water absorption study

The moisture content of the bioplastic samples increased when the plasticizers were added. From Fig. [14A](#page-13-0), it can be found that the sample F-0 has a minimal value of moisture content. Bioplastic samples synthesized with glycerol as a plasticizer had the maximum values of moisture content, and bioplastics with sorbitol had the lowest value. However, in bioplastic samples where the glycerol–sorbitol combination was used, the moisture content had values lower than those of the glycerol samples but higher than those of the sorbitolbased samples. A previous study has also shown that the glycerol-based bioplastic formulation had a higher value than the sorbitol formulation, where two bioplastic samples were prepared from banana peel starch (BPP) and banana peel/corn starch/rice starch composite (COM) (Shafqat et al. [2021](#page-16-17)). This could be related to the fact that glycerol consists of a hydroxyl group that shows an afnity for water molecules that allows them to form hydrogen bonds and contain water in the structure (Cerqueira et al. [2012\)](#page-15-25). However, sorbitol develops substantial hydrogen bonds with starch molecules, and therefore, the affinity to water molecules is reduced (Sanyang et al. [2016\)](#page-16-24).

The water absorption values of the prepared bioplastic flms are presented in Fig. [14B](#page-13-0). As observed, flms with no plasticizer (F-0) had the highest values $($ ~ 80%) of water absorption. Among the plasticized samples, glycerol-based plasticization showed the highest water absorption, followed by a glycerol–sorbitol combination and then sorbitol. These fndings are consistent with previous reports (Shafqat et al. [2021](#page-16-17)). Previous fndings have also shown that glycerol has a relatively higher affinity for water compared to sorbitol (Cerqueira et al. [2012](#page-15-25); Sanyang et al. [2016](#page-16-24)). This is because the hydroxyl group in starch has a strong affinity for water molecules and gelatinization also breaks up starch granules that allow water to difuse (Azahari et al. [2011](#page-15-20)). Several reports have described a directly proportional relationship between the starch content and water uptake capacity of bioplastic flms (Azahari et al. [2011;](#page-15-20) Aranda Garcia et al. [2015](#page-15-26); Sujuthi and Liew [2016](#page-16-30)). Therefore, the addition of plasticizers reduces the absorption of water.

Fig. 11 Result of the XRD analysis of the bioplastic samples

In the current work, the water vapor permeability of potato-based biodegradable plastic flms made with various kinds and concentrations of plasticizers was tested using the WVTR method. Figure [15](#page-13-1) displays the outcomes of the WVTR test. Due to their hydrophilic characteristics, biodegradable polymers derived from potatoes often have poor water vapor barriers (Othman et al. [2017](#page-16-31)). However, we discovered that the WVTR value is proportional to the concentration of plasticizers; when plasticizer concentration increased, WVTR increased as well, regardless of the type of plasticizer (Fig. [15](#page-13-1)). WVTR increased for sorbitol-based bioplastics from 76 to 104%, whereas it increased from 76 to 115% for glycerol-based bioplastics (Fig. [15\)](#page-13-1). High plasticizer concentration increased the fexibility and mobility of starch polymer chains, resulting in a looser network through the structural change of starch–starch molecular interactions, could be one of the most plausible explanations for our fndings (Sanyang et al. [2016;](#page-16-24) Ng et al. [2022\)](#page-15-14).

Solubility and biodegradation investigation

Figure [16A](#page-13-2) and Table [3](#page-14-0) show the solubility test results on bioplastic samples. Water solubility increased upon adding plasticizer. The reason why F-0 samples show the lowest value is that hydrogen bonds of the starch molecules' crystalline structure of starch molecules signifcantly reduce their water solubility (Sarker et al. [2013](#page-16-32)). Similar to the moisture content, the water solubility of starch-based bioplastics was higher in samples with glycerol as a plasticizer, followed by the combination of glycerol–sorbitol and lowest in samples with sorbitol (Fig. [16A](#page-13-2)). Compared to sorbitol, glycerol has a higher affinity for water, and its molecular weight is also lighter, which allows water molecules to easily penetrate into polymer chains (Ghasemlou et al. [2011\)](#page-15-27). These results are consistent with previous reports showing that it afects the solubility of bioplastic in water (Chiumarelli and Hubinger [2014](#page-15-28); Shafqat et al. [2021](#page-16-17)).

Fig. 12 Selected bioplastic samples prepared from potato starch. **A** F-0, **B** FG-10, **C** FS-10, and **D** FG1-S1. **E** Thickness and density of bioplastic samples

From Table [3,](#page-14-0) it can be seen that all samples were soluble in water at both room temperature and elevated temperature (60 °C). At room temperature, bioplastic samples with sorbitol and glycerol–sorbitol combinations were insoluble in 10% NaOH and 10% H_2SO_4 . However, with an increase in temperature, some samples became soluble in these solvents, and some showed partial swelling. All of these samples were insoluble in both methanol and ethanol, while some of them showed partial solubility and/ or swelling in diethyl ether at elevated temperature. The current fndings are also supported by previous reports by Pimpan et al. ([2001](#page-16-13)).

The biodegradation ability of bioplastics is attributed to their physicochemical properties such as molecular weight, chemical structure, affinity to water, surface area, etc. (Tokiwa et al. [2009](#page-16-33)). The biodegradation ability of the bioplastic samples was evaluated in garden soil for up to 30 days. After every 5 days, the samples were observed, weighted, and the percent weight loss was calculated. From the time series data presented in Fig. [16A](#page-13-2), the percentage of weight loss of all samples was observed to increase. Films with glycerol showed the highest percentage of biodegradation, after which samples with sorbitol and glycerol–sorbitol combination act as a plasticizer, respectively. The lowest weight loss

Fig. 13 SEM images showing the surface morphology of potato starch-based bioplastic samples. **A** & **B** F-0; **C** & **D** FG-10; **E** & **F** FS-10; **G** & **H** FG1-S1. Red arrow=starch; yellow arrow=non-melted, non-ruptured granule/granule fragments; white arrow=void

was observed in the F-0 samples (Fig. [16](#page-13-2)B). However, all bioplastic samples degraded within 30 days and weight loss reached 100%. The addition of plasticizers that increases the biodegradation of samples can be attributed to the higher affinity of both the plasticizers (glycerol and sorbitol) toward water molecules (Shafqat et al. [2021\)](#page-16-17). In this study, we excluded any fllers in the formulation of the bioplastic samples. A recent study has shown that the addition of filler to corn starch bioplastics reduces biodegradation (Mohan et al. [2016\)](#page-15-29). However, the presence of fller provides a huge surface contact area through which degrading agents can enter the polymer (Rutkowska et al. [2002\)](#page-16-34). The application of natural fllers has been reported to accelerate the biodegradation of bioplastics (Egute et al. [2009](#page-15-30); Kumar and Sekaran [2014\)](#page-15-31). Fillers demonstrate hydrophilic properties, and microorganisms can easily absorb them, resulting in enhanced adhesion of microorganisms to the surface of the polymer and its resultant biodegradation (Shah et al. [2008](#page-16-35)). Fillers that contain protein or easily soluble and hydrolyzed components improve biodegradation ability (Mastalygina et al. [2016\)](#page-15-32). Therefore, as a future strategy, it is recommended to consider diferent kinds of waste materials to determine their efficiency as reinforcing filler in bioplastics.

Fig. 14 Data on the **A** moisture content, and **B** percentage water adsorption of the bioplastic samples

Fig. 15 WVTR of the bioplastic samples

The results of the antimicrobial test are shown in Table [4](#page-14-1) and Fig. [17.](#page-14-2) The inhibitory response was calculated using the clear zone surrounding the bioplastic flm disc. Measurement of clear zone diameter was started from the outside diameter of flm bioplastics.

The absence of the clear zone indicates no inhibitory zone, and the diameter was valued as zero. The bioplastics

Fig. 16 A Time-series data on the biodegradability test on the garden soil; **B** data on the water solubility of the bioplastic samples

have shown fair antimicrobial activity against the both gram-positive and gram-negative bacteria except *B. haynesii* where no clear zone was found. Also, FG1-S1 samples did not show any inhibitory activity against the *S. typhii*. A similar trend in data was also reported by Agustin and Padmawijaya [\(2017\)](#page-15-15).

Conclusion

This study shows that potato starch, with or without the addition of diferent plasticizers, can be used to produce bioplastics with diferent physical and chemical properties. Diferences in these properties will enhance the suitability of these bioplastics for various applications. The thickness values of the bioplastic flms obtained in this study suggest that the bioplastic flms have the stress-bearing strength needed for good packaging material. These potato starchbased bioplastics were biodegradable and environmentally friendly, making them a good substitute for petroleumbased plastics and an efective way to alleviate the problem **Table 3** Results of the solubility test of the starch-based bioplastic flms prepared using glycerol as plasticizers

RT Room temperature; *ET* Elevated temperature $(60 °C)$; $+=$ Soluble; \sim = Swelling; \times = insoluble; $\sqrt{\frac{1}{2}}$ Some part is soluble and some part swells

Table 4 Antimicrobial activity test of the bioplastic samples

Bacteria	Bacteria type	Inhibition zone diameter (mm)			
		$F-0$			FG-10 FS-10 FG1-S1
Salmonella typhi	Gram-negative 32 0			36	0
Escherichia coli	Gram-negative 30		32	40	32
Bacillus subtilis	Gram-positive 29		27	32	30
Bacillus megaterium	Gram-positive 22		18	35	33
Bacillus haynesii	Gram-positive	θ	0	0	θ

of plastic pollution. However, further studies are needed to identify suitable raw materials for the synthesis of biodegradable biopolymers. It would be interesting to include diferent fllers obtained from a variety of waste materials and determine their efficiency as reinforcing fillers in bioplastics. The use of cheap raw materials for the production of bioplastics also increases their economic feasibility. Moreover, further research is needed to fnd agents that can impart hydrophobic properties to bioplastic flms and enhance their moisture resistance.

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Data availability The datasets generated during and/or analyzed during the current study are available from the corresponding author on reasonable request.

E Bacillus haynesii

Fig. 17 Antimicrobial activity test. Red triangle=Inhibition zone

Declarations

Conflcit of interest The authors declare that they have no known competing fnancial interests or personal relationships that could have appeared to infuence the work reported in this paper.

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