ORIGINAL RESEARCH

Computer simulation of the fow of a thermoplastic elastomer vulcanizate melt through an axisymmetric extrusion die using an augmented Navier's slip equation

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Received: 25 February 2023 / Accepted: 7 May 2023 / Published online: 11 June 2023 © Iran Polymer and Petrochemical Institute 2023

Abstract

The pressure fows of a thermoplastic elastomer vulcanizate (TPV) melt through the die region of a single-screw extruder were simulated using the fnite-element method. The COMSOL software was utilized to develop the fnite-element model of the die and solve the working equations. ACross-WLF rheological equation of state was used for the description of the rheological behavior of the polymer melt. The numerical results were compared with their associated experimental data. The novel aspect of the present work is the development of a new augmented Navier's slip equation to take the efect of wall slip on solution variables into account. The proposed model consists of a power-law equation, which relates the slip coefficient to the shear rate at wall. It is implemented in the COMSOL code through a simple script. The distribution of the velocity, pressure, and temperature as well as the profle of the fuid velocity at exit region and slip velocity along the fow directions were presented and discussed. It has been shown that using the no-slip condition at wall or employing the Navier's slip equation with a constant slip coefficient leads to notable error in the prediction of mass flow rates and pressure.

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Graphical abstract

Keywords Simulation · Fluid fow · Polymer melt · COMSOL · Navier's slip

Introduction

It is generally known that extrusion is one of the fundamental processes that are extensively used for fabricating of polymers into products like rods, pipes, profles, and sheets. In this process, the fnal form of the product is determined by fow of the polymer melt through a die. Therefore, a good understanding of the state of fow variables, such as velocity, pressure, and temperature, is vital to improve the die design, reduce the number of experimental trials and errors, and control of the fow. Dies are normally employed with an adapter (called a head) that connects to the exit region of the extruder barrel. The assembly of a die and head has usually complicated geometry, and thus, sophisticated mathematical models are required to fully describe the fow inside it. To achieve this task, numerical methods are commonly utilized in conjunction with commercial or in-house developed computer codes. In the past few decades, there has been a surge of interest in developing computer simulation techniques to analyze the flow of polymer melts through various domains, including extrusion dies. Several commercial softwares are available on the markets that are well suited for the modeling of the fow of non-Newtonian

fuids in complex geometries. However, some polymeric fuids like rubbers and thermoplastic elastomers (TPEs/TPVs) present very high viscosity even at elevated temperatures. They also show viscoelastic nature, which means that their fow behaviors are between elastic solid and pure viscous materials. One of the consequences of these characteristics is the slip–stick phenomenon at the interface zone between polymer and solid walls, which makes the conventional no-slip assumption to be invalid. If the shear stress at the wall exceeds a critical value, there is a relative velocity between the fuid velocity at the wall and the wall velocity, which is called slip velocity. Traditionally, the slip of the melt at the solid wall is treated using a mathematical model known as Navier's slip equation $[1–3]$ $[1–3]$. This model suggests that the slip velocity is proportional to the shear stress at the wall. This equation is similar to the convection boundary condition in heat transfer (third type or Robin boundary condition). Navier's slip condition in a general three-dimensional framework is given by Eq. (1) (1) [\[4\]](#page-9-2)

$$
-\frac{\eta}{\beta}S.(u - u_s) = S.(\tau.n).
$$
 (1)

In this equation, **u** and **u**_s are the fluid and solid wall velocity vectors, respectively. τ is the viscous stress tensor of the fluid, **n** is the unit vector normal to the plane in which the slip occurs, η is the shear viscosity, and β is a slip coefficient which is the momentum transfer coefficient at the interface between fluid and wall.**S** is a geometric tensor that projects vectors into the local tangent plane, expressed as

$$
S = I - nn.
$$
 (2)

I is the identity matrix. It has been shown that [[5\]](#page-9-3) for high viscous polymers when the wall slip is not taken into consideration, the predicted velocity profles and the parameters derived from it such as fow rate have notable deviation from experimental data. Moreover, the original form of Navier's slip relationship is linear which fails to predict accurate results, especially at higher shear rates. Consequently, several modifed and new relationships are proposed so far which are briefy reviewed in the background section. The present work is a re-visit of the use of Navier' slip condition during the modeling of a flow of a highly viscous non-Newtonian polymer melt (a commercial thermoplastic vulcanizate, TPV) through the die region of a single-screw extruder. A new relationship is introduced in which the slip coefficient is assumed to be related to the shear rate. It is shown that the suggested model gives rise to the prediction of more accurate results at higher screw speeds that correspond to high shear rates.

In the following sections, we frst give a brief overview of the most important works published on the modeling of slip condition in fuid mechanics. We provide an account of the most recent fndings and developments together with highlighting the novelty of our work. Then, the details of the governing equations, including the fow and energy, constitutive or rheological, and Navier's slip equations are presented. Following that, the developed fnite-element models in this work are presented. Next, the experimental parts including the utilized die and extruder, rheological measurements techniques, and extrusion process are explained. The results and discussion are then presented in detail, and fnally, conclusions are drawn.

Background

A considerable amount of literature has been published on the mathematical description of the slip phenomena during the fow of polymeric fuids in contact with solid walls. We, therefore, consider some of the works that are directly related to our selected approach. Lau and Schowalter [[6\]](#page-9-4) have developed a theoretical relationship for the slip velocity of an ethylene–propylene copolymer. Their model relates the slip velocity to the shear stress and temperature through a complex mathematical model given

$$
v_s = C_1 \left(\tau_w\right)^n \left[1 - C_2 \tanh\left(\frac{E - C_3 \tau_w}{kT}\right)\right],\tag{3}
$$

where v_s is the slip velocity, τ_w is the shear stress at the wall, *E* is the activation energy, *T* is the temperature, and C_1, C_2, C_3 , and *k* are the material constants. The model contains complex terms to consider the onset of the slip phenomenon or separation of the fuid at the wall. The basic idea of the work was on a combination of the concept of junctions at the wall/polymer interface as well as in the bulk of the polymer fuid with a kinetic equation describing a reaction, to which activation rate theory applies. Furthermore, an analogous slip velocity model was developed by Hatzikiriakos and Dealy [[7\]](#page-9-5). Their model includes the frst normal stress diference and molecular characteristic efects. They have experimentally studied the fow of a high-density polyethylene melt and determined the conditions for the onset of slip and the relationship between slip velocity and shear stress. Similar relationships were proposed by other researchers which are reviewed in a comprehensive literature survey by Hatzikiriakos [\[8](#page-9-6)]. In that work, it was stated that the slip happens macroscopically when the shear stress at the wall exceeds a critical value. Moreover, for linear polymers, a second critical shear stress is also observed in which a transition from weak to strong slip occurs. The focus of his work was, however, on the slip flow of molten polymers and no discussion was made for rubbery materials. Another complete review of slip phenomena including slip in polymer solutions, suspensions, and other complex fuids was also published by Archer [[9\]](#page-9-7). The subjects of weak and strong slip were taken into consideration. Matthews and Hill [[10\]](#page-9-8) proposed a nonlinear Navier's slip model based on the work carried out by Thompson and Trojan [\[11\]](#page-9-9). They tested it for three simple pressure-driven fows using the Newtonian model without experimental verifcation. The main drawback with these models is the difficulty of imposing the developed equation into fnite-element working equations and obtaining stable and convergent results. In addition, the applicability of the model for highly viscous rubbery materials has not been tested. Recently, Pérez-Salas et al. [[12\]](#page-9-10) developed a semi-analytic solution for the pressure fow of a Phan–Thien–Tanner fuid through an axisymmetric domain with Navier's slip condition at wall. They showed that the results of their model are in good accordance with those predicted by ANSYS Polyfow commercial code. Wilms et al. [[13\]](#page-9-11) studied the wall slip of non-Brownian suspensions in pressure-driven fows. They modifed the classical equations by taking the efect of domain geometry on the slip into account.

From the practical point of view, there are also several published works in which computer simulations have been performed with diferent degrees of complexity to study the efects of wall slip on the fow variables. Ghoreishy

and Nassehi [[3\]](#page-9-1) developed a fnite-element code in which a novel method was introduced for the imposition of Navier's slip condition (Eq. [1](#page-1-0)). They have used their code for the simulation of the fow of rubbers in internal mixers. Later, Ghoreishy et al. [\[5](#page-9-3)] extended this code for the simulation of a TPE fow through the die region of a single extruder with axisymmetric geometry. They have shown that the ignoring slip wall leads to errors in the prediction of the fow rate. In the present research, we have proposed a work and an augmented form of Navier's slip condition for the numerical analysis of the fow of a highly viscous polymer melt through a die. Yang and Li $[14, 15]$ $[14, 15]$ $[14, 15]$ $[14, 15]$ considered the effect of wall slip for a CB/silica flled triple blend rubber (NR/SBR/ BR) compound during oscillatory and steady shear rheometry. It was shown that the slip generally does not afect the oscillatory shear, but in the steady shear flow, it tends to make the measured shear viscosity to be lower than the actual value. Moreover, they have simulated the viscoelastic flow of the mentioned compound through an extrusion die by Polyfow software in which the wall slip was taken into consideration by a linear Navier's slip condition, given as

$$
v_s = -\frac{\tau_w}{k},\tag{4}
$$

where k is the slip coefficient. Moreover, a 5-mode Phan–Thien–Tanner (PTT) model was also used as the rheological model. Having changed the slip coefficient, different profles were predicted for the extrudate. It is reported that increasing the slip coefficient enlarges the predicted cross-sectional shape of the extrudate. Rippl [[16](#page-9-14)] has used a power-law form for the slip velocity in an in-house developed fnite-element code as

$$
\tau_w = k_0 |v_s|^m. \tag{5}
$$

They have simulated the flow of a rubber compound through a two-dimensional channel and three-dimensional die. The efects of three cases including no-slip, free wall slip (perfect slip), and partial slip on velocity and viscosity profles were investigated. Mitsoulis et al. [[17](#page-9-15)] studied the fow behavior of an SBR/CB rubber compound with a hardness of Shore A 70 in capillary and injection molding dies. They also considered the wall slip using a power-law temperature-dependent form similar to Eqs. ([4\)](#page-3-0) and ([5\)](#page-3-1) for slip velocity at walls. Both a pure viscous rheological model (Carreau) and a viscoelastic (K-KBZ) model were utilized in their simulations, which were carried out by their developed codes. They compared the predicted pressure drop with the experimental data and showed that accurate results can only be obtained by considering both viscoelasticity and wall slip. In summary, it can be deduced that due to the complex rheological behaviors of polymer melts, no single form for the slip velocity and its relation to shear stress at the wall can

be used. In this work, we present a new form of slip coeffcient in Navier's slip equation and verify it by comparing the results of the fow of a highly viscous polymer melt in an extruder die with their associated experimental data. Specifcally, the novelty of our work is to propose a new form for the slip coefficient in Navier's slip equation in which it is assumed that the slip coefficient is related to the shear rate through a power-law relationship.

Governing equations

The governing equations of the steady-state, non-isothermal, and laminar fow of an incompressible non-Newtonian fuid in an axisymmetric coordinate system (*r*,*z*) are given as follows [[18\]](#page-9-16):

The equation of continuity:

$$
\frac{\partial v_r}{\partial r} + \frac{v_r}{r} + \frac{\partial v_z}{\partial z} = 0.
$$
 (6)

The equation of motion in r-direction:

$$
\rho \left(v_r \frac{\partial v_r}{\partial r} + v_z \frac{\partial v_r}{\partial z} \right) = -\frac{\partial p}{\partial r} + \frac{1}{r} \frac{\partial}{\partial r} \left(r \tau_{rr} \right) + \frac{\partial \tau_{zr}}{\partial z} - \frac{\tau_{\theta\theta}}{r} + \rho g_r. \tag{7}
$$

The equation motion in z-direction:

$$
\rho \left(v_r \frac{\partial v_z}{\partial r} + v_z \frac{\partial v_z}{\partial z} \right) = -\frac{\partial p}{\partial z} + \frac{1}{r} \frac{\partial}{\partial r} (r \tau_{rz}) + \frac{\partial \tau_{zz}}{\partial z} + \rho g_z. \tag{8}
$$

The equation of energy:

$$
\rho C_p \left(v_r \frac{\partial T}{\partial r} + v_z \frac{\partial T}{\partial z} \right) = \frac{1}{r} \frac{\partial}{\partial r} \left(r k \frac{\partial T}{\partial r} \right) + \frac{\partial}{\partial z} \left(k \frac{\partial T}{\partial z} \right) + \dot{S}.
$$
\n(9)

In these equations, v_r and v_z are the components of velocity vector in r and z directions, respectively, ρ is the material density, *p* is the pressure, g_r and g_z are the components of the gravity vector, C_p is the specific heat, k is the thermal conductivity, *T* is the temperature, and *Ṡ* is the heat generated due to the viscous dissipation efect. The rheological behavior of the polymer melt is assumed to be described by the generalized Newtonian fuid as

$$
\tau = \eta \Delta,\tag{10}
$$

where τ and Δ are the viscous stress and rate-of-deformation tensors, respectively, expressed by the following relationships:

$$
\boldsymbol{\tau} = \begin{bmatrix} \tau_{rr} & \tau_{rz} & 0 \\ \tau_{rz} & \tau_{zz} & 0 \\ 0 & 0 & \tau_{\theta\theta} \end{bmatrix} \tag{11}
$$

$$
\Delta = \begin{bmatrix} 2\frac{\partial v_r}{\partial r} & \frac{\partial v_r}{\partial z} + \frac{\partial v_z}{\partial r} & 0\\ \frac{\partial v_r}{\partial z} + \frac{\partial v_z}{\partial r} & 2\frac{\partial v_z}{\partial z} & 0\\ 0 & 0 & 2\frac{v_r}{r} \end{bmatrix}.
$$
 (12)

Viscosity η (Eq. [10](#page-3-2)) in the present study is given by a modifed Cross rheological model [[19\]](#page-9-17) expressed as

$$
\eta = \eta_0 \left(1 + \left(\frac{\eta_0 \dot{\gamma}}{\tau_{cr}} \right)^{1-n} \right)^{-1},\tag{13}
$$

where *n* and τ_{cr} are two material constants. η_0 is dependent on temperature through the Williams-L and el-Ferry(WLF) equation given as

$$
\eta_0 = D_1 e^{\left[\frac{-A_1(T - T_c)}{A_2 + (T - T_c)}\right]}.
$$
\n(14)

In this equation, A_1 , A_2 , and T_c are three material constants that should be determined by an appropriate experiment. In addition, \dot{y} is the shear rate which is related to the second invariant of the rate-of-deformation tensor by the following relationships:

$$
\dot{\gamma} = \sqrt{\frac{1}{2}I_2} \tag{15}
$$

$$
I_2 = \left(2\frac{\partial v_r}{\partial r}\right)^2 + \left(2\frac{\partial v_z}{\partial z}\right)^2 + 2\left(\frac{\partial v_r}{\partial z} + \frac{\partial v_z}{\partial r}\right)^2 + \left(2\frac{v_r}{r}\right)^2.
$$
\n(16)

Slip boundary condition

As stated previously, the slip of the polymer melt at the solid wall is considered by Navier's slip equation. The two-dimensional form of this equation (derived from Eq. [1\)](#page-1-0) is given as

$$
-\frac{\eta}{\beta}\mathbf{t}.\left(\mathbf{u}-\mathbf{u}_{\mathrm{s}}\right)=\mathbf{nt}:\tau,\tag{17}
$$

where **t** is the unit vector tangent to the boundary. Additionally, as there is no penetration (cross-fow) at the wall, thus, the condition $\mathbf{u} \cdot \mathbf{n} = 0$ should also be taken into consideration in conjunction with the above equation. Two forms for the slip coefficients β were considered. In the first form, a constant value has been assumed, while in the second form, which is also the novelty of this work, the following nonlinear equation has been proposed:

$$
\beta = \beta_0 \dot{\gamma}^m,\tag{18}
$$

where β_0 and *m* are two material constants. The details and justifcation of this new model will be given in the Results and discussion section.

Finite element model and fow simulation

The above-mentioned equations were solved using a fniteelement model in an axisymmetric two-dimensional framework. The details of the working equations and solution algorithm can be found in the literature (for example [\[1](#page-9-0)]) and are therefore not repeated here. The COMSOL Multiphysics (laminar fow with heat transfer in fuids) v. 6.0 [\[20\]](#page-9-18) was used for the simulation of flow in the die region. A mixed (also called u–v–p) fnite-element method was used to solve the Navier–Stocks equations. A second-order discretization was used for the velocity, while a frst-order one was employed for the pressure. The heat transfer was solved using the frst-order discretization in conjunction with an upwinding technique to obtain stable results. The sketch of the flow domain is shown in Fig. [1](#page-4-0). The flow domain was discretized into 99,279 triangular elements. An extra fne meshing size with controlled minimum and maximum element sizes was selected to achieve high-accuracy fnite-element approximation. Figure [2](#page-5-0) shows the selected mesh for diferent parts of the fow domain. Two series of simulations were carried out. In the frst series, an input pressure was applied at the inlet of the domain (Fig. [1](#page-4-0)) and the associated flow field parameters, including the components of velocity vector, pressure, and temperature, were calculated as the primary variables. The mass fow rate was then calculated by integration of the velocity component (v_r) over the surface at the exit region. On the other hand, in the second series, a known value of the mass flow rate was assumed and the corresponding entry pressure was computed. It should be noted that, in each series, three diferent slip conditions including

Fig. 1 Full shape of the die with the fow domain (axisymmetric coordinate system)

$$
\mathbf{E}^{\mathbf{D}}_{\mathbf{D}} \mathbf{P} \mathbf{P} \mathbf{P} \quad \text{where} \quad
$$

Fig. 2 Finite element mesh of the fow domain

no-slip ($\beta = 0$), linear Navier's slip equation ($\beta = constant$), and the proposed form of the Navier's slip equation (Eq. [18\)](#page-4-1) were taken into account.

Experimental

To verify the developed model, the fow of a thermoplastic vulcanizate (TPV) through the die region of a laboratory single-screw extruder at diferent speeds was performed. The details of the materials and process are given below.

Materials

A commercial TPV (Elastron RD.121.A60.914) was selected in this work. It was a blend of PP (polypropylene) as the thermoplastic with a partially vulcanized EPDM as the elastomer parts. Density of the polymer was determined using an ADAM density measuring system. The specifc heat and

Fig. 3 Measured and ftted values of the viscosity versus shear rate at diferent temperatures

Table 2 Rheological properties (parameters of the equations)

$\tau_{cr}(Pa)$		$D_1(Pa.s)$	$A_1(K^{-1})$	$A_2(K)$	$T_c(K)$
13,500	0.28	3.2×10^{14}	11.26	52.5	275

thermal conductivity were measured using Mettler Toldeo DSC and TCA200 (Taurus instrument), respectively. These physical and thermal properties are given in Table [1.](#page-5-1)

Rheometry

The rheological behavior of the material was determined using a MCR501s, Anton Paar, Austria. The experiments were carried out in a shear rate sweep mode covering a range of 0.01–100 s⁻¹. A 25 mm parallel plate geometry with a gap of 2 mm, and various temperatures of 170–200 °C were applied. The Origin software v. 6.1 [21] was used to fit these data into the rheological Eqs. [\(13\)](#page-4-2) and ([14\)](#page-4-3) by a nonlinear curve ftting technique. Figure [3](#page-5-2) shows the variations of the experimentally measured and numerically ftted values of the viscosity versus shear rate at diferent temperatures. The predicted parameters are recorded in Table [2.](#page-5-3)

Extrusion process

The extrusion process of the selected TPV was carried out using a laboratory single-screw extruder equipped with a rod die, as shown in Fig. [1.](#page-4-0) Six screw speeds (10, 20, 30 40, 50, and 60 RPM) were selected. For each test run, the pressure at the entrance zone and the mass fow rate were measured. The temperatures of the die and the last zone of the barrel near the die were set at 190 °C.

Results and discussion

Simulations of the fow of the polymer melt were performed using the above-mentioned methods and process conditions. A uniform temperature of 190 °C was considered as the fuid entered the die. As it was pointed out in the previous sections, two series of simulations were carried out. In the frst series, the pressure measured at the entrance zone was imposed as the boundary condition and the corresponding mass fow rates were computed, which are given in Fig. [4](#page-6-0) with their associated experimentally measured data. Similarly, in the second series, the measured mass fow rates were considered as the boundary condition and the pressures at the inlet were computed, which have been compared with their experimental data, as depicted in Fig. [5.](#page-6-1) In both cases, the no-slip boundary conditions were applied at the wall. Comparing the results in each series, it can be seen that increasing the entry pressure or assuming a higher mass fow rate gives rise to an increase in the deviation of the computed

Fig. 4 Computed mass fow rate versus entry pressure with no-slip boundary condition

Fig. 5 Computed entry pressure versus mass fow rate with no-slip boundary condition

values from their experimental counterparts. This result supports our previous assumption [[5](#page-9-3)] that ignoring the wall slip, especially at higher shear rates leads to the computation of a lower value for mass fow rate than the experimentally measured value. The distributions of the computed temperatures are shown in Fig S1a–f in the Supplementary Materials. It is worth noting that the maximum diference between inlet and outlet temperatures is within $3 \degree C$, which means that although a non-isothermal analysis is performed in this work, an almost isothermal condition is obtained. Consequently, the remaining fnite-element analyses have been carried out based on an isothermal assumption. Moreover, Fig S2a-f in the Supplementary Materials show the computed pressure and velocity felds inside the fow domain for diferent entry pressures corresponding to the selected six screw speeds. As it can be seen and expected, the pressure gradually decreases along the fow direction from its maximum value (entry pressure) to nearly zero at the exit region, which confrms the existence of a typical and classic pressure fow. In addition, the velocity feld shows that by decreasing the cross-section area of the die in z-direction, the velocity vector increases.

To study the efect of the wall slip on the results, the increase in observation in the diference between predicted and experimental values (pressure and mass flow rate) could be attributed to the lower average velocity with the no-slip condition. In fact, with increasing the entry pressure (or mass flow rate), the boundary layer diminishes, leading to the generation of a higher component of the velocity along the fow direction. To further verify and quantify this

Fig. 6 Computed entry pressure versus experimental data at diferent slip coefficients

Fig. 7 Computed mass fow rate versus experimental data at diferent slip coefficients

statement, the simulations were repeated with Navier's slip wall condition with different values for slip coefficients (β) . The computed entry pressure and mass fow rates against dif-ferent slip coefficients are plotted and shown in Figs. [6](#page-7-0) and [7](#page-7-1), respectively. The presented computational results in these

Fig. 8 Slip velocity along flow direction at different screw speeds

Fig. 9 Average velocity along flow direction at different screw speeds

graphs are also compared with experimental data. As it can be seen, the discrepancy between the predicted entry pressure and mass fow rates with experimental data decreases with increasing of the slip coefficient (β) . In addition, the profle of the computed slip velocity and average velocity along the fow direction are shown in.

Figures [8](#page-7-2) and [9](#page-7-3), respectively. As it can be seen, there is a signifcant increase in slip velocity when the polymer melt enters the fnal section, which is also the narrowest part of the die. It should be pointed out that the jumps in the velocity profles at two points at the entry and exit region of the

fnal section are due to the singular points at these regions and attributed to the numerical instabilities created. To further consider this point, the distributions of the shear rates for these simulations are plotted as shown in Fig. S3a–f in the Supplementary Materials. It is apparent from these fgures that by increasing the entry pressure, the shear rates signifcantly increase (as expected) especially at the narrowest section. Therefore, we may conclude that there should be a direct relation between slip coefficient and shear rate, which could be attributed to the complicated interactions between polymer melt and the inner surface of the die. This reveals the need for a new format for the slip coefficient, which is related to the state of rate-of-deformation. Consequently, we have assumed in the present work that the slip coefficient (β) can be related to the shear rate by a power-law relationship as given in Eq. [\(18](#page-4-1)). To fnd the parameters for this equation, the best value of the slip coefficient for each entry, as shown in Figs. 6 and 7 , was chosen in conjunction with its associated average value of the shear rate at the exit zone. It should be explained that, as it can be seen in Fig. [8,](#page-7-2) the slip velocity at the fnal zone of the die is much higher than its corresponding value at the other zones. Therefore, the average value of the shear rate was calculated at the exit region. Using a simple curve ftting approach, the values of β_0 and *m* in this equation are found and given in Table [3.](#page-8-0)

The associated variations of the computed mass fow rate versus experimentally determined entry pressure and computed entry pressure versus experimentally measured mass flow rates are shown in Figs. [10](#page-8-1) and [11](#page-8-2), respectively. It can be seen that, by assuming a variable slip coefficient, very close agreements are obtained between the computed variable and experimental data which confrms our assumption and the proposed model.

The distributions of the pressure and velocity for the flow with variable slip coefficient are shown in Fig. S4a–f in the Supplementary Materials. Compared with their corresponding distributions (no-slip boundary conditions) as shown in Fig S2a–f) in the Supplementary Materials, it can be seen that there are slight diferences between the predicted pressure and velocity between two series of simulations. To further study the efect of the wall boundary conditions, the velocity profle at the exit region, with and without slip at wall for the six screw speeds, is plotted and shown in Fig S5a–f in the Supplementary Materials, respectively. Here, it can be seen that the velocity profle for the no-slip case is lower than the computed velocity variations when the wall slip is taken into account. This is the reason that the computed mass fow rates for no-slip boundary conditions are lower than in the case

Fig. 10 Computed mass fow rate versus entry pressure with modifed Navier's slip equation

where the slip condition is imposed into the working equations. Besides, as expected, by increasing the screw speed (or entry pressure), the diference between velocity profles, with and without wall slip conditions, becomes more prominent, which is in agreement with our previous finding.

Fig. 11 Computed pressure versus mass flow rate with modified Navier's slip equation

Conclusion

The flow of a polymer melt with a generalized Newtonian temperature-dependent rheological model through the die of a single-screw extruder was simulated using the COMSOL software. Diferent slip conditions at solid walls including the no-slip, Navier's slip equation with constant coefficient and a newly proposed model based on Navier's slip equation with variable coefficient were considered. It has been shown that due to the complex interaction between polymer melt and the inner surface of the wall, the assumption of a constant slip coefficient is not valid. Consequently, we have proposed a new phenomenological form in which the slip coefficient is related to the shear rate at wall by a power-law model. Having compared mass fow rate and entry pressure computed from the simulation results with their corresponding experimentally measured data, the applicability and accuracy of this new model was proved. It is recommended that further research should be undertaken to assess the proposed model for three-dimensional problems and check its validity for more complicated frameworks.

Supplementary Information The online version contains supplementary material available at<https://doi.org/10.1007/s13726-023-01190-9>.

Acknowledgements The authors would like to acknowledge the Iran Polymer and Petrochemical Institute for permission to publish the results presented in this paper.

Data availability Data sets generated during the current study are available from the corresponding author on reasonable request.

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