ORIGINAL ARTICLE

Extraction and characterization of fiber from the flower stalk of the Agave plant for alternative reinforcing biocomposite materials

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Abstract

To achieve durability and biodegradability in the composite materials sector, natural fbers ofer a preferable alternative to synthetic fbers. The morphological, physical, thermal, and mechanical features of the fower stalk fbers extracted from the *Agave americana* L. plant (FSAA) were investigated for the frst time. Scanning electron microscopy and thermal analysis, Weibull distribution, Fourier transform infrared spectroscopy, and X-ray difraction were all employed in the investigation. According to an XRD examination, the (FSAA) fbers contained crystallites that were 2.53 nm in size and had a crystallinity index of 29.15%. While the fibers' average strength in tension was $(64.34 \pm 11.43 \text{ MPa})$, their modulus of Young was $(77.61 \pm 15.90 \text{ GPa})$ and their strain at failure was $(1.5 \pm 0.31\%)$, their distribution required statistical Weibull analysis with two parameters (ML technique) to be performed. The fndings of the thermal investigation carried out by TGA demonstrate that the fbers of (FSAA) exhibit thermal stability up to 339.57 °C and energy of kinetic activation of 68.1 kJ/mol. The signifcance of this research highlights the possible uses of this fber in the feld of biocomposites for industrial use such as construction, aerospace, and automotive industry.

Keywords Flower stalk of *Agave americana* · Thermophysical characterization · Mechanical test · Natural fber · SEM observation

Highlights

• The frst investigation of fber properties extracted from fower stalks of *Agave americana.*

- The FSAA fibers are extracted using the water-retting method. • Using the peak deconvolution technique, a crystallinity index of
- 29.15% was found.
- The average tensile strength and Young's modulus of the FSAA fbers are 64.34 MPa and 77.61 GPa, respectively.

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1 Introduction

Over the past few decades, natural fbers have attracted the interest of numerous researchers in a variety of industries, including textile, furniture, packaging, recreation, automotive, sports, construction, electronics, decorating, aerospace, kitchenware products, and others [[1\]](#page-14-0). Natural fbers have many benefts over synthetic fbers, including biodegradability, accessibility, low density, non-toxicity, lightweight, reasonably priced, great thermal characteristics, and reduced equipment wear [[2](#page-14-1)]. Due to their environment-friendly production and consumption practices, recycling capabilities, and value addition in the circular economy [\[3\]](#page-14-2), natural fbers played an increasingly signifcant role in the biocomposite manufacturing and packaging sectors [[4\]](#page-14-3). Glass, carbon, ceramic, and aramid synthetic fbers are particularly pricey, toxic, and unsustainable; they also pose concerns to human health when exposed for an extended period of time [[5](#page-14-4)]. Natural fbers are predominantly derived from cellulose-rich plants, such as hemp $[6]$ $[6]$ $[6]$, jute $[7]$ $[7]$, kenaf $[8]$, and banana pseudo-stem fber [[9\]](#page-14-8). The fact that these natural fbers are less costly,

widely accessible, produced with renewable resources, and possess a high specifc strength gives them an edge over glass fbers [\[10,](#page-14-9) [11\]](#page-14-10). Cellulose, among natural polymers, is by far the most common. Nature produces over 10^{10} cellulose annually, which doubles the amount of biomass on Earth. Cellulose is a linear homopolymer composed of glucose residues in a D confguration with glycosidic bonds in the β-(1–4) position [\[12,](#page-14-11) [13\]](#page-15-0). Cellulosic natural fibers can be extracted from various sources, including higher plants such as algae. These fbers can also be obtained from plant fruits, seeds, stalks, and leaves. They consist mainly of cellulose, lignin, pectin, and hemicellulose wax [[14\]](#page-15-1). The most typical process for removing premium fbers is water retting. The presence of bacteria and moisture during the retting process enables the breakdown of the plant's cellular tissues and the degradation of pectins, waxes, and other adhesive substances around the fbers, facilitating the splitting apart of individual fbers [[15](#page-15-2)]. Cellulose remains the most abundant and important natural material due to its exceptional mechanical strength, making it suitable for a diverse array of applications. This has inspired researchers to look into novel fbers that could strengthen polymers and potential substitute dangerous man-made synthetic fbers [[16](#page-15-3)]. According to Marcelet et al.'s morphological study of *Ficus religiosa* root fber using electronic scanning microscopy, they found that the raw surface of the fber was semi-lisse and consists of cellulose content, cire, and hemicellulose in signifcant percentages [\[17\]](#page-15-4). Kathirselvam et al. conducted an investigation on fbers from *Thespesia populnea* (TP) that has been exposed to various sodium hydroxide (NaOH) solution concentrations for varying lengths of time. In particular, measurements were done and reports were written about elements like the percentage of cellulose, crystallinity, thermal stability, tensile strength, and surface shape [[18\]](#page-15-5). A study was carried out by Dao Cheng et al. to investigate the potential of natural *Lufa vine* (LV) fber as a cellulose reinforcement for light, decay-resistant composite materials. The investigation examined the microstructure, as well as the morphological, thermal, chemical, and physical characteristics of the LV fber. According to the results, the LV presents itself as a lightweight natural fber with a porous structure and is distinguished by its low density, distinctive microscopic characteristics, and outstanding mechanical strength [[19\]](#page-15-6). Babu et al. conducted a comprehensive analysis of an original natural fber obtained from *Phaseolus vulgaris* (PVF). The presence of cellulose, which has a 43.01% crystallinity index, was eventually shown by PVFs' XRD analyses. The PVFs' surface morphology was investigated using various magnifcation levels, revealing the presence of cell formations with tiny voids on one side of the fber. In contrast, the opposing side appeared smooth, indicating the presence of microfbrils [\[20\]](#page-15-7). The application of natural fber composites is frequently constrained by their susceptibility to elements like acidic/alkaline agents, moisture absorption, high temperatures, and salty conditions. Therefore, in order to speed up the development of viable products for both industrial and societal usage, researchers are anxious to examine the robustness of biofber composites. In this regard, a recent investigation was carried out by Kumar et al. on a newly discovered natural fber that was taken from the stems of the *Myriostachya wightiana* (MW) plant. This perennial grass, also known as Nalia grass in Odisha, India, is widely distributed in Bangladesh, Myanmar, Malaysia. Although MW stems are often used by artisans to make baskets, toys, ropes, and diferent storage bags, there has not been much substantial scientifc research on MW fber until this study. The study found that this grass contains a higher percentage of cellulose than other natural fbers, suggesting potential use in the paper and packaging industries as well as for cellulose nanofber producers to create useful goods. This fber can also be used in lightweight structural applications like composites due to its low density and strong strength [[21\]](#page-15-8). Maran et al. conducted an analysis to explore the prospective application of freshly isolated cellulosic fbers from the *Sida mysorensis* plant in polymer composites as reinforcement. FTIR spectroscopy was employed to identify the various functional groups of SMF. *Sida mysorensis* fber is the ideal material to replace the synthetic fbers in the fberreinforced plastics, according to all the aforementioned fndings [\[22](#page-15-9)]. A research by Mansouri et al. completed a study on the burying *Agave americana* leaves in the soil to extract buried fbers (BF) as a composite polymer reinforcement. The fndings revealed that lignin, hemicelluloses, and cellulose were the primary constituents of these buried fbers (BF). This study set out to investigate an innovative technique for removing buried fber (BF) from the *Agave americana* L. plant. The obtained fbers underwent chemical, ATR-FTIR, XRD, SEM, and thermal tests after being buried with the plant leaves in the soil. According to the fndings, BF extraction using a soil burying strategy could be efective and proftable. It was discovered to enhance the extracted fbers' quality, making them ideal for polymer composite reinforcement [[23](#page-15-10)]. A study on fbers taken from the prop root of *Fragrant screw pine* (FSP) plant's prop root was done by Gerald et al. They discovered that alkali treatment considerably improved the hydrophobic and surface properties of these *Fragrant screw pine* root fbers (FSPRFs). The research team thoroughly characterized the alkali-treated FSPRFs (AFSPRFs), examining their mechanical, thermal, and chemical characteristics in order to foresee their possible use as reinforcement in polymer composites. The functional group, crystal structure, and surface characteristics of the fber were determined by the team using Fourier

transform infrared spectroscopy (FTIR), X-ray difraction (XRD), and scanning electron microscopy (SEM). Their research suggested that adding AFSPRFs as reinforcement to polymer composites could signifcantly enhance the thermomechanical characteristics of bio-based polymer matrix composites [[24](#page-15-11)].

The succulent *Agave americana*, which is indigenous to Mexico but is produced all over the world, including the semi-arid and arid Ain Achir region of Annaba, Algeria, shows great promise for sustainable material development. The *Agave* plant is a separate species of the Agavaceae family and the Agavoideae subfamily. Its distinctive characteristics are its sturdy, elongated fower stalk, which may soar to a dizzying height of 30 feet and support clusters of vivid yellow or greenish-yellow blossoms. It should be noted that the plant dies after producing its blooms, leaving behind the sturdy stalk, which is frequently thrown away. This quality raises the Agave's signifcance as a source for the creation of natural fabric. These stalks are the subject of our study because they are renowned for their toughness, which encourages efective resource usage. We encourage sustainability and help to lessen our impact on the environment by reusing this otherwise useless part of the plant. These fower stalks' strong, biodegradable fbers can be used as a benefcial alternative to synthetic fbers in the production of composite materials. *The Agave* plant's signifcance goes beyond its inherent toughness; it has the ability to transform industries looking for sustainable substitutes, signaling a movement in material science towards environmental consciousness.

Agave americana fbers were also known as Ramban [[25](#page-15-12)], *Agave* fber-reinforced biocomposites were green materials that had already been reprocessed, and employed in the automotive industry [[26](#page-15-13)] and other industries like epoxy [\[27](#page-15-14)], epoxy, and matrix [[28\]](#page-15-15), biopolymer [[29\]](#page-15-16). *Agave* fbers, on the other hand, had good mechanical qualities, such as strong toughness, robust adhesion to the fber matrix, and minimal damageability [\[30](#page-15-17)].

This work offers a fresh investigation of the underutilized and abandoned resource of the *Agave* plant's flower stalk for fber extraction and characterization in biocomposites. By flling a research void, it enhances understanding of environmentally friendly materials alternatives, specifcally the potential of *agave* fiber to replace traditional synthetic materials, this study seeks to show a unique fber that had been recovered from the fower stalk of *Agave americana* (FSAA), in contrast to previous studies that had explored fbers taken from the leaves and skin of leaves and others parts of *Agave americana*. Reinforcing bio-composite materials with (FSAA) fbers had never been described before. The crystallographic, thermal, mechanical, morphological, and physicochemical properties were evaluated using Fourier-transform infrared spectroscopy, scanning electron microscopy (SEM), thermogravimetric analysis (TGA and DSC), single fber stress testing, and X-ray difraction (XRD).

2 Materials and methods

2.1 Extraction of fbers

After peeling, the fower stalks of *Agave americana* (FSAA) (Fig. [1](#page-3-0)a) were cut into 50-cm (Fig. [1c](#page-3-0), b) lengths and soaked in water (Fig. [1](#page-3-0)d) for 30–45 days at 30–35 °C to remove the fbers (Fig. [1](#page-3-0)e). The biological breakdown of the fower stem is sped up considerably by the anaerobic bacterial activity and in the process, allowing for fber extraction. Fibers were separated from their stems by hand harvesting, washing, and air drying as shown in (Fig. [1f](#page-3-0)).

2.2 Characterization of the fower stalk of *Agave americana* **fbers**

2.2.1 Determination of density

The pycnometer method was widely used to determine how dense natural fbers are. It involved measuring the volume of submerged fbers using a water-flled container with a known volume. Pycnometer was the accepted practice when using the mass diference method to calculate the density of natural fbers. Density analysis was performed using liquid acetone, with a known density of $(0.790 \text{ to } 0.792 \text{ g/cm}^3)$ at normal temperature. Short segments of the natural fbers were separated and put inside a pycnometer with a volume of 25 ml (V_{pv} = 25 ml). The pycnometer's initial weight was noted as m_1 , while the weight following the addition of the fibers was noted as $m₂$. Consequently, $m₃$ represents the weight of the FSAA fbers when combined with acetone. The evaluation was conducted five times, and the average value of each repetition was recorded as the density of the FSAA fbers, then Eq. [\(1](#page-2-0)) was used to obtain the density [[31](#page-15-18)]:

$$
\rho_{FSAA} = \left[\frac{m_2 - m_1}{V_{py} - \frac{(m_3 - m_2)}{\rho_{action}}} \right]
$$
\n(1)

2.2.2 Linear density

Each fber's length was measured and then weighed using the single-fber weighing method. It meets the requirements of ASTM D 2257, with the amount of removed fber measured in Tex units. This method was widely used in the textile industry due to its reliability in determining fber quality. The linear density was calculated using this data [\[31\]](#page-15-18). The following Eq. [\(2\)](#page-2-1) to contrast the fber's extracted fneness with that of other vegetable fbers:

$$
Tex = \frac{weight(g)}{length(m)} \times 1000\tag{2}
$$

Fig. 1 a Flower stalk of *Agave americana* plant, **b**–**d** diferent steps for the immersion process of the FSAA, **e** biodegradation of fbers after 30 to 45 days at temperatures above 35 °C, **f** the FSAA fiber obtained

2.2.3 Moisture content and moisture regains

Working with natural fbers like cotton, wool, and fax requires careful consideration of several factors, including moisture content. These fbers had a propensity to take in moisture from the environment, which may have a substantial impact on their physical and mechanical characteristics. For this study, moisture content and moisture regains were determined according to test standard ASTM 2654 [\[32\]](#page-15-19). The usual weight-loss approach served as instrumental in determining the level of moisture and fber recovery of FSAA fbers. Samples were weighed outside the oven before drying them at 100 ± 5 °C for 2 h.

To obtain a precise result, three samples were examined and calculated $[31]$ $[31]$ using Eqs. ([3\)](#page-3-1) and ([4\)](#page-3-2).

$$
moistureregains(\%) = \frac{originalweight - overdryweight}{Weight of dry fibers in the oven} \times 100
$$
\n(3)

$$
Moisturecontent(\%) = \frac{originalweight - overallryweight}{originalweight} \times 100
$$
\n(4)

2.2.4 Fiber diameter

The width of each fber was assessed employing an optical magnifier from ZEISS with a Plan-Achromat \times 400 lens. Three random locations over the length of the fber were measured from each of the 30 samples and the single fber's typical diameter was identifed [[31\]](#page-15-18). Based on these readings, a calculation was made to determine the typical diameter of the individual fbers with ZEN (Zeiss software). This methodology made sure that the fber diameter was accurately and fairly analyzed for all samples.

2.2.5 Scanning electron microscopy (SEM) analysis

Micrographs were captured using a scanning electron microscope (SEM), the MEB-FEG Quattro S with a range of magnifcation settings and electron-accelerating voltages varying from 3 to 5 kV. To enhance electrical conductivity, improve image quality, protect the sample from potential damage, and reduce surface charging effects, the fiber had a gold covering applied to it. SEM is an efective imaging technique that can be used to analyze the microstructure of FSAA fbers and better understand their impact on mechanical characteristics. The resulting micrographs provided us with important new information about the morphological characteristics of the FSAA fbers, including their size, shape, and physical properties. Utilizing SEM allowed for a thorough analysis of the fbers at various magnifcations, giving a thorough grasp of due to the morphology. Meb Quattro s in topographic mode was the instrument used to analyze fber morphology.

2.2.6 Fourier transform infrared (FTIR) spectroscopic analysis

The existence of functional groups on the FSAA fbers was determined using an FTIR spectrometer, which recorded the IR spectra of the samples. PerkinElmer used infrared spectroscopy using a Fourier transform with a Spectrum RX1 to analyze the chemical features of the samples. The test was carried out by scanning 32 times over a spectral range of (4 cm^{-1}) , and in transmission mode 500 to 4000 cm−1, according to the specifed parameters. The Origin 10.0 software resolution application was used to carry out band deconvolution methods.

2.2.7 Thermogravimetric analysis (TGA) measurement

The ability of natural fbers to strengthen matrix polymers with high treatment temperatures depends on their ability to withstand thermal degradation. Traditional methods for instance, thermogravimetric (TG) and derived DTG, or thermogravimetric analysis, are often employed to assess the heat resistance of fibers. Analyses of TG and DTG are conducted to examine how the bulk and its amount of change vary as the temperature increases in FSAA fbers [\[33](#page-15-20)]. This study examined thermal stability using TGA/DSC $3+1600$ °C (METTLER TOLEDO), for the fibers. An alumina beaker was placed in the oven with FSAA fbers inside. The test was conducted at an ambient temperature of 800 °C with a heating rate of 10 °C per minute. Another critical factor used to evaluate bio-fbers' thermal stability is their kinetic activation. The activation energy of motion is typically calculated using Broido's formula [\(5](#page-4-0)) [[21\]](#page-15-8).

$$
\ln\left[\ln\left(\frac{1}{y}\right)\right] = -\left(\frac{E_a}{R}\right)[\left(\frac{1}{T}\right) + K] \tag{5}
$$

R is for the standard gas constant (8.32 J/mol K); T stands for Kelvin temperature; *y* is the standardized weight (w_t/w_i) , and w_t and w_i stand for the sample's weight at different times and temperatures, respectively.

2.2.8 DSC (diferential scanning calorimetry)

Diferential scanning calorimetry was a common method to examine how materials react to heat. For natural fbers, DSC was employed to investigate their distinct thermal characteristics, including crystallinity, and transition of glass, temperature, melting point. Using this technique, the cellulose and lignin contents could be noted in the fbers' glass transition point temperature (T_g) and destruction temperature. The fbers were subjected manage cycles of heating and cooling to examine their thermal properties. METTLER TOLEDO

was the analysis using a thermogravimetry device used to capture the thermal events that took place when it is being heated, in the fber.

2.2.9 X‑ray difraction (XRD) analysis

Using X-ray difraction (XRD), the crystallinity index (CI) and crystallite size (CS) of FSAA fbers were calculated. When the FSAA fbers were exposed to X-rays, the crystals within the fbers scattered the X-rays, resulting in a distinct difraction pattern that could be analyzed to determine the crystalline properties of the FSAA fber. With a Bruker D8 ADVANCE Difractometer for X-rays, radiation from a Cu-K source $(\lambda = 1.54184 \text{ Å})$. The experiments were carried out at room temperature utilizing a generator using a 40 kV voltage and a 20 mA current. The samples, in powdered form, were examined within a specimen container by scanning them at 2 values between 10° and 70° with a 2°/ min progressive rate. The crystal size is determined by the following Eq. (6) (6) [\[34](#page-15-21)]:

$$
CS = \frac{K.\lambda}{\beta \cdot \cos(\theta)}\tag{6}
$$

The Scherrer constant is denoted by the sign K (0.9) (In Angstroms) represents the wavelength. β, which is the peak's overall breadth when it is half its greatest (in radians); and θ, which stands for the difraction angle. Equation [\(7\)](#page-4-2) is utilized to compute the crystallinity index [\[35](#page-15-22)]:

$$
CI\% = \left[\frac{I_{002} - I_{am}}{I_{002}}\right] \times 100\tag{7}
$$

In Eq. [7](#page-4-2), I_{002} stands for the crystalline region's greatest intensity, and I_{am} for the amorphous region's lowest intensity.

2.2.10 Water absorption

The study consists in tracking the samples changing cumulative mass after immersion in seawater and distilled water for 720 h (equivalent to 30 days) at an ambient temperature. The mass measurements were carried out at irregular intervals throughout a one-month duration following samples being submerged in seawater and distilled water. This approach was implemented to ensure a comprehensive assessment of the temporal changes in mass, thereby providing useful details about the efects of water immersion on the samples, without being restricted to predetermined time intervals for each measurement. Before weighing, the samples were carefully wiped with absorbent paper, and their mass was measured at regular intervals. An electronic scale of the OHAUS type was used for weighing (measurement

 $accuracy = 0.0001$ g). The bulk gain associated with the aging of the fbers in the time % representation of t was used. The ASTM standard method D1037-99 was used to perform the water absorption tests $[36]$ $[36]$. The percentage change in weight of the FSAA fbers calculations was made at a specified time (t) using Mt $(\%)$, representing the weight change, expressed with the following equation:

$$
M(t) = \frac{(m_t - m_0)}{m_0} \times 100
$$
 (8)

According to a Peleg-proposed mathematical model, the fbers of FSAA's water adsorption curves were modifed (Eq. [9](#page-5-0)).

$$
g(t) = a + \frac{t}{b * t + c}
$$
\n(9)

where the difusion parameters, are stated in minutes, and a, b, and c are constants (*a*=291.7; *b*=0.7457; *c*=0.2138).

2.2.11 Tensile test of FSAA fbers

All experiments were performed at 25 °C with a relative humidity of about 62%. Tensile strength of a single fiber strength testing was performed using the Test112 global testing equipment after the fbers were glued to cardboard for stability. In 1 min, the crossbar moved 1 mm. Each sample was measured using a 100 mm GL (gauge length). Thirty diferent samples of FSAA fber were analyzed to evaluate their elastic zone tensile properties. These properties included tensile strength, deformation, and young module. ASTM D3822-07, a method for measuring the tensile strength of textile fbers, was employed for the testing.

For natural fbers, statistical analysis plays a signifcant part in examining the fbers and composites. Through statistical analysis, mean values and variability of characteristics of FSAA fbers were determined. Fragile materials, where this technique had been previously applied multiple times, tended to fail due to the distribution of faults [\[37\]](#page-15-24). A two-parameter Weibull distribution was used to statistically assess the tensile mechanical characteristics of FSAA fbers, and the maximum likelihood technique (ML) was employed to estimate a 95% confdence interval for Young's modulus, fber toughness, and strain. Minitab 19 [[21](#page-15-8)] was employed to conduct the study's statistical analysis.

3 Results and discussion

3.1 Fiber density of FSAA fbers

An important instrument for assessing the strength, functionality, and condition of the fbers and tissues is the analysis of fber density. Natural fbers' density is impacted by a multitude of environmental and cultural factors, including plant age, fber extraction method, soil quality, geographical area, and also plant space. Fiber density was 980 kg/m^3 by using the overarching principle of the methodology (Eq. [1\)](#page-2-0), the density of FSAA fbers is less than other fbers such as *Yucca treculeana* (1330±39 kg/m3) [\[38](#page-15-25)], *Eleusine indica* (1143 kg/ m³) [[39\]](#page-15-26), *Trachelospermum jasminoides* (1398 kg/m³) [\[40](#page-15-27)], *Acacia concinna* (1365 kg/m3) [\[41](#page-15-28)], *Typha augustata grass* (1015 kg/m3) [\[42](#page-15-29)], *Cocos nucifera* (1360 kg/m3) [[43](#page-15-30)], *Cyperus dichrostachus* (1010 kg/m³) [\[44](#page-15-31)], *Fragrant screw pine* (13,852 kg/m3) [\[45](#page-15-32)], Citrullus *lanatus climber* (1227±371 kg/ m³) [[46](#page-16-0)], *Ficus religiosa* root fibers (1246 kg/m³) [[17](#page-15-4)], *Albizia amara* (1043 kg/m³) [\[47](#page-16-1)], and the FSAA fibers have a density greater than other fbers like vascular bundles stand of *Phoenix dactylifera* (914 kg/m3) [[48\]](#page-16-2), *Cypreus papyrus* plant (950 kg/m³) [\[31](#page-15-18)], *Furcraea selloa* k.koch Plant (810 kg/m³) [\[49](#page-16-3)], *Catalpa bignonioides* fruit (713 kg/m³) [\[14\]](#page-15-1). This may be due to the distinctive qualities of the fower stalk of *Agave americana* and the environmental circumstances in which it is grown. The physical characteristics of recently invented natural fbers are replicated in Table [1](#page-6-0).

3.2 Linear density measurement

The FSAA fiber linear density is influenced by several factors, including the maturity of fber included, growth circumstances, and extraction conditions. These elements contribute to defning how fne the fbers are overall, which in turn, afects the quality and potential applications of the material, the linear density calculation is an essential measure to assess the quality and properties of fbers. The linear density of the fower stalk of *Agave americana* fbers is determined in Tex units using Eq. [\(4\)](#page-3-2). The typical fneness of the extracted fbers was 3.06 Tex. The linear density of FSAA fbers is superior to that of *Cyperus dichrostachus* fbers $(0.51 \pm 0.244$ Tex) [[44\]](#page-15-31), and lower than that of *Cyperus papyrus* plant fibers $(8.38 \pm 0.6 \text{ Tex})$ [[31\]](#page-15-18).

3.3 Moisture regains and moisture content analysis

The moisture regain and moisture content of these fibers were estimated using Eqs. ([2\)](#page-2-1) and ([3\)](#page-3-1) at a temperature of $(100 \pm 5 \degree C)$ for 2 h, and the resulting value was 9.80%. The kiln dry weight method was employed for this calculation. There is 8.92% moisture in the material, FSSA fbers have a high moisture content than *Typha augustata* grass fbers (6.56%) [\[42\]](#page-15-29), and lower than that of *Cyperus papyrus* (9.86%) [[31\]](#page-15-18). The inherent hydrophilicity of natural fibers, which causes them to attract moisture from the environment, poses a challenge for composite reinforcement due to the impairs of the hydrophobic matrix and the fiber's interfacial adhesion [[31\]](#page-15-18).

Table 1 Comparison of the FSAA's physical characteristics with other cellulosic fbers

3.4 Diameter determination

The results of the analysis using an optical microscope to determine how large the fbers are, the microstructure and diameter distribution of the FSAA fber are displayed in Fig. [2](#page-6-1), and the fiber micrograph at $400 \times$ magnification in Fig. [2](#page-6-1) reveals the fber diameter. The fber diameter was calculated to be three locations on the FSAA fber, and it was found that the mean diameter was 208 ± 35 µm, this diameter measurement offers crucial details about the size and the fbers' morphology. The distribution of the observed fber diameter is depicted as micrographs in Fig. [2](#page-6-1), some other fbers have a diameter lower and higher than the FSAA fber like *Citrullus lanatus climber* (210 \pm 4.85 µm) [[46\]](#page-16-0), *Myriostachya wightiana* (102±278 µm) [[21](#page-15-8)], *Yucca treculeana* (171±36 µm) [\[38](#page-15-25)], *Eleusine indica* (315.4±10 µm) [[39](#page-15-26)], and *Cyperus dichrostachus* $(9.85 \pm 7.16 \text{ µm})$ $(9.85 \pm 7.16 \text{ µm})$ $(9.85 \pm 7.16 \text{ µm})$ [\[44\]](#page-15-31). Table 1 displays the comparison with other fbers.

3.5 Scanning electron microscopy analysis (SEM) of FSAA fbers

The shape of the FSAA fber's surface contained apparent imperfections, possibly due to wax or lignin, as determined by longitudinal and transverse examinations. Because of the roughness they add to the fber's surface, these impurities can afect the material's or product's mechanical and physical properties. Increased friction between fbers is one efect of a fber's rough surface, and contaminants can also

Fig. 2 Sample micrograph of an individual FSAA fber

cause the fber's surface to become rough and even hollow. Comparing the FSAA fber with other fbers such as *Yucca trucelana* and fber strands of *Phoenix dactylifera*, *Myriostachya wightiana* and fber from *Juncus efusus*, it was found that they share similar surface characteristics and imperfections. This suggests that the presence of wax or lignin impurities, as well as the resulting roughness, may be common features among these fbers [[21,](#page-15-8) [38,](#page-15-25) [48,](#page-16-2) [53\]](#page-16-4). Additionally, it was discovered that composite materials' mechanical toughness reinforced by natural fbers can be diminished because FSAA fbers had porosity along the surface of the

fber that can make it more fragile and likely to break. The durability of a material is diminished because pores act as stress concentration areas and can lead to cracks or breaks. In addition to the width of the lines on the fbers' surface, the fber arrangement displays alignment with the fber axis's direction in the form of a consistently scattered square and semi-rectangular tray. The FSAA fber exhibits a distinctive surface morphology characterized by irregularly distributed micropores or pits interspersed with microfbrils. Moreover, the surface shape of the fber closely resembles an elliptical contour with an almost circular appearance [[48\]](#page-16-2). This observation aligns with fndings from previous investigations done on fbers originating from sources like *Catalpa bignonioides* fruits, *Washingtonia flifera*, and *Eleusine indica* grass [\[14,](#page-15-1) [39](#page-15-26), [54\]](#page-16-8). These fbers also displayed similar features, including the presence of pores and microfbrils. The fber diameter that was measured is 200 µm. As observed in the fber photograph Fig. [3](#page-7-0)a, the surface contains certain substances that could be contaminants, lignin, or wax as well as pores, the roughness of the fber surface may be increased by these contaminants and imperfections [\[38](#page-15-25)], as shown in Fig. [3](#page-7-0).

3.6 Fourier transform infrared (FTIR) spectroscopic analysis

The FTIR spectra for the FSAA fbers are shown in Fig. [4](#page-8-0)a. To analyze the FSAA fbers, the FTIR vibration spectrum was used to examine the absorption bands associated with functional chemical families, including cellulose, hemicelluloses, and lignin [[55](#page-16-9)]. To resolve overlapping peaks

Fig. 3 SEM micrographs of the Longitudinal view of FSAA fber with diferent magnifcations

Fig. 4 FSAA FTIR analysis: **a** FTIR spectra and **b** deconvolution of peaks

in the complex spectra, peak deconvolution was applied specifically in the range of $500-4000$ cm⁻¹ (Fig. [4b](#page-8-0)). This technique allows for the separation and analysis of individual peaks, providing a more detailed understanding of the molecular components and functional groups present in the FSAA fbers. The stretching vibrations of the hydroxyl groups (O–H) linked by hydrogen generated from cellulose cause the peak at 3342 cm^{-1} [\[21\]](#page-15-8), whereas the maximum at 2904 cm−1 encoding cellulose and hemicelluloses are present, indicating the stretching vibration of the C-H from CH and CH₂ $[56]$ $[56]$ $[56]$. There are peaks in the frequency of FSAA fiber at 1731 cm⁻¹ correlated to $C = 0$ stretching encoding the hemicellulose [[15](#page-15-2)]. The aromatic ring of the phenyl propane molecule $C = C$ in lignin is what causes the peak at 1591 cm⁻¹ [\[52\]](#page-16-7). The peaks from the spectrum analysis were compared to those from other works, which are listed in Table [2.](#page-8-1) C-H Bending vibration is responsible for the highest point seen at 1377 cm^{-1} [\[57](#page-16-11)]. We notice a peak at 1243 cm^{-1} caused by the C-O extending the vibration of the acetyl group in lignin [[38\]](#page-15-25). The FTIR spectrum showed dual peaks at 1034 cm^{-1} which are about C-O and O–H indicating the occurrence of lignin. On the other hand, the C–OH bending generates an absorb-ance peak of around 597 cm⁻¹ [\[58\]](#page-16-12), and it can be inferred that the CO–H group out-of-plane bending exists.

3.7 Thermogravimetric analysis (TGA) measurement

The thermogravimetric analysis (TGA) and its derivative (DTG) are graphically depicted in Fig. [5](#page-9-0)a. Two separate mass loss phases can be seen in the TGA curves. During the first phase of mass loss, 9% between room temperature and 100 °C is determined by a DTG peak plot at 75.27 °C inferred from the fiber's dehydration [[55\]](#page-16-9). The following stage of mass loss was 56.52% between 200 and 359 °C, meaning hemicellulose and cellulose combined mass loss. The DTG curves display clear peaks at degradation temperature of 339.57 °C, which was found to be the point where fibers begin to fall apart. Figure [5](#page-9-0)b's ln [ln (1/y)] plot vs. 1/T, this straight line

Table 2 Observation of FTIR peaks for FSAA fber

Peak position in this study Range of peak reference (cm^{-1})	values (cm^{-1})	Functional group and chemical composition	Reference
3342	3449-2912.33	Hydrogen-bonded O-H stretching in α cellulose	[21]
2904	2909-2851	the presence of cellulose and hemicelluloses, represents the C-H stretch- ing vibration from CH and CH ₂	[56]
1731	1731-1636	$C = O$ stretching encodes the hemicellulose	$\lceil 15 \rceil$
1591	1591-1629	the aromatic ring of the phenyl propane molecule $C = C$ in lignin	$\left[52\right]$
1377	1379-2936	C-H Bending vibration	$\left[57\right]$
1243	1242	C-O stretching vibration of the acetyl group in lignin	$\left[38\right]$
1034	1035	C-O and O-H indicating the occurrence of lignin	[58]
597	598	C – OH bending	$\left[58\right]$

Fig. 5 a TG and DTG curves of the thermal analysis of the FSAA fber, **b** Broido's plot

obtained from the data, is employed to determine the activation energy (E_a) using Eq. ([5\)](#page-4-0), which was discovered to be 68.1 kJ/mol. This is superior to the fiber of

Table 3 Thermal characteristics comparison between the FSAA and other cellulosic fbers

the *Leucaena leucocephala* tree (64.22 kJ/mol) [[33\]](#page-15-20), and *Typha angustata grass* (67.99 kJ/mol) [[42\]](#page-15-29), (64.38 kJ/ mol) for *Phaseolus vulgaris* [\[20\]](#page-15-7), and less compared to the *Banyan* tree's aerial roots fiber (72.65 kJ/mol) [[59](#page-16-13)], and *Myriostachya wightiana* (72.84 kJ/mol) [[21\]](#page-15-8), *Fragrant screw pine* (83.9 kJ/mol) [[45](#page-15-32)], *Acacia nilotica* (69.739 kJ/mol) [[60\]](#page-16-14). Table [3](#page-9-1) shows the FSAA fiber's thermal characteristics and the kinetics activation energy in comparison with other cellulosic fibers.

3.8 DSC (diferential scanning calorimetry)

DSC analysis enables the identification of thermal transitions by examining physical changes caused by the absorption and release of thermal energy during heating. Figure [6](#page-10-0) shows the DSC analysis of the fbers of FSAA. The initial temperature peak, indicating an endothermic process, was measured at 76 °C and was caused by the moisture evaporating from the fbers. The peak's associated integrated enthalpy was 2015.61 mJ. The second peak, which was reached at 274 °C, revealed that polysaccharides other than lignin, namely a-cellulose, were decomposing. The next endothermic peak is 318 °C in a nitrogen atmosphere, the peak represents α-cellulose degradation in the FSAA fbers [[65\]](#page-16-15). The surface's cellulose content had completely degraded, as evidenced by the subsequent signifcant endothermic peak at 362 \degree C [[66\]](#page-16-16).

3.9 X‑ray difraction analysis (XRD)

Three peaks were observed after the profles of X-ray diffraction were deconvoluted (Fig. [7](#page-10-1)). The cellulose's crystalline phase is shown by the other peaks at 21.67° (002) and 30.12° (004); however, the first peak at 15.57° (110)

Fig. 6 DSC curves of FSAA fber

indicates the fber's amorphous phase. Using Eq. [\(7](#page-4-2)), the crystallinity index (CI) was estimated to be 29.15%, far smaller compared to that of *Zingiber officinale* (78.81%) [[67](#page-16-21)], *Cocos nucifera* (52%) [[43\]](#page-15-30), and *Catalpa bignonioides* fruit (94%) [[14\]](#page-15-1), but larger than that of *Acacia concinna* (27.5%) [[41](#page-15-28)]. Using Scherrer's Eq. [\(6](#page-4-1)) to compute the crystalline size, the result was (2.53 nm), which is lower than other fbers such as *Tithonia diversifolia* (3 nm) [[55](#page-16-9)], *Yucca treculeana* (3.02 nm) [[38\]](#page-15-25), *Cocos nucifera* (6.5 nm) [[43\]](#page-15-30), and higher than that of *Leucaena leucocephala* (2.33 nm) [[33](#page-15-20)]. A comparison with the referenced publications is provided in Table [4](#page-10-2). FSAA fbers have been

Fig. 7 XRD spectrum of FSAA fber

found to have crystalline properties. The fber's properties are sufficiently strong to enable its utilization as reinforcement in composites.

3.10 Water absorption

From Fig. [8](#page-11-0), it appears that natural fibers have a significant capacity to absorb water, regardless of whether they are submerged in distilled water or seawater. After 24 h of submersion (1440 min), it shows that they can adsorb as much as 850% and 868% of their relative dry weights for distilled and sea water which exceeds that of *Washingtonia flifera* fbers $(220\% \text{ of their initial mass})$ [[54\]](#page-16-8). The fibers possess a high absorption capacity due to the presence of cellulose and its porous structure [\[54\]](#page-16-8). It is crucial to remember that the absorption of water by natural fbers can have consequences on their properties, such as their strength, elasticity, and softness. This must be considered when they are used for the production of textiles and other products [\[70\]](#page-16-22). After 360 and 720 h of water immersion, the value of absorption did not change. It is possible to say that the mathematical formula which allows the explanation for the kinetics of the body absorbing water in the FSSA fbers is given by the mathematical model found in the literature and was used to study the distilled water absorption and seawater absorption kinetics in the FSAA fbers. Better statistical parameters for sampling areas can be found in the pelagic model [[71\]](#page-16-23). We can see from Fig. [8a](#page-11-0) and b that the proposed model maximizes the number of experimental points such closely as is physically possible [\[72](#page-16-24)].

Table 4 The comparison of the crystallographic qualities of FSAA fber with other cellulosic fbers

Fiber	Crystalline properties		Reference	
	Ci (%)	Cs (nm)		
Flower stalk of Agave americana	29.15	2.53	This work	
Yucca treculeana	48.85	3.02	[38]	
Eleusine indica	45		$\left[39\right]$	
Leucaena leucocephala	63.10	2.33	$\left[33\right]$	
Trachelospermum jasminoides	87.68	3.9	[40]	
Cocos nucifera	53	6.5	[43]	
Ageratina adenophora	68.98	16.28 ± 1.55	[68]	
Catalpa bignonioides fruit	94		$\lceil 14 \rceil$	
Typha augustata grass	65.16	6.40	$\lceil 42 \rceil$	
Albizia amara	63.7		[47]	
Citrullus lanatus climber	35.70	16.30	[46]	
Acacia nilotica	44.82	3.21	[60]	
Acacia concinna	27.5	4.17	[41]	
Zingiber officinale	78.81		[67]	
Tithonia diversifolia	45	3.00	$\left[55\right]$	
Yucca gloriosa	37.74		[69]	
Myriostachya wightiana	54.799	3.24	$\lceil 21 \rceil$	

Fig. 8 FSAA water absorption curves: **a** FSAA water absorption behavior, **b** Peleg modeling for distilled water, and **c** Peleg modeling for seawater water

3.11 Tensile test of FSAA fbers

Quasi-static tensile testing of the FSAA fbers was completed at a constant 1 mm per minute speed and a length of 100 mm. A collection of 30 carefully selected specimens composed of these fbers was created. Figure [9a](#page-12-0) illustrates the standard stress–strain curve of individual FSAA fbers. The FSAA fber exhibits exceptionally brittle behavior and a rapid low-load failure in the case of a fber failure. Tensile assessment of a single FSAA fber shows a two-phase response to stress and strain (Fig. [9a](#page-12-0)), exhibiting brittle behavior similar to that of the *Yucca treuculana* [\[38\]](#page-15-25). The following phase was reached when the fber ruptured or when the strain suddenly decreased to an average value of $(64.34 \pm 11.43 \text{ MPa})$. The fiber's tensile strength surpassed that of regularly used fbers, including *Myriostachya wightiana* 46.554 MPa [\[21](#page-15-8)], and lower than that of *Ageratina* phase was characterized by a linear zone with a strain ranging from 0 to 1.57%. Young's modulus for strain levels ranging from 0 to 0.8% was calculated using the elastic zone with a very steep slope of the curve $(77.61 \pm 15.90 \text{ GPa})$ (Fig. [9a](#page-12-0)). The FSAA fbers had a signifcantly higher Young's modulus than other types of fbers than that of the *Jancus efusus* $(4.38 \pm 1.37 \text{ GPa})$ [[53\]](#page-16-4), and *Yucca treculeana* $(16.85 \pm 10.52$ GPa) [[38](#page-15-25)]. Table [5](#page-12-1) presents an overview of the mechanical characteristics of the tested fbers and contrasts them with FSAA fbers. Furthermore, it should be noted that for some experiments in the second phase, the fiber occasionally exhibited discontinuities that were mostly caused by the rupture of some microfbrillar structures. The same behavior for diferent GL between 10 and 40 mm was observed for *Agave americana* fbers [\[57](#page-16-11)]. Young's modulus, strain at failure, and strength under tension are displayed in Fig. [9](#page-12-0)b, c, and d, respectively. As is typical of lignocellulosic fbers, the data demonstrate a signifcant degree of dispersion, and the conclusions must therefore be properly understood through statistical examination of the data. Furthermore, other factors that afect experimental testing include test settings and characteristics such as test speed, machine accuracy, machine grip style, and measurement of fber cross-sections to gauge strength [\[54](#page-16-8)].

The FSAA fber exhibits exceptionally brittle behavior and a rapid low-load failure in the case of a fber failure. Figure [10a](#page-13-0)–c displays the Weibull distribution plot curves for the FSAA fbers' Young's modulus, tensile strength, and strain at failure. Each fber's values for tensile strength, strain at failure, and Young's modulus were determined to fall within the range and to perfectly ft the Weibull distribution. The Weibull graphs' scale factors and shape

Fig. 9 Property tensile of FSAA fbers **a** stress/strained behavior, **b** diameter versus tensile strength, **c** failure strain versus fber diameter, and **d** Young's modulus versus fber diameter for $GL = 100$ mm

Table 5 Mechanical properties comparison of FSAA fber with those from other plants reported in published publications

were employed to verify the fndings of the experiments. The homogeneity of the distribution of the experimental data is confrmed by the shape factor utilized in this study [[37](#page-15-24)], (m_E = 5.2 for Young's modulus, m_σ = 6.5 for strain at failure, and $m_g=5.8$). The scale factor values that were anticipated for the present study were 69.14 MPa, 1.64%, and 84.04 GPa. Where, respectively, represent the tensile strength, strain at failure, and Young's modulus. These numbers perfectly match the average observed trial results at 64.34 ± 11.43 MPa, $1.5 \pm 0.31\%$, and 77.61 ± 15.90 GPa. Figure [11](#page-13-1)a–c plots illustrate the probability of survival (P) for the 2-parameter Weibull using FSAA fbers'

mechanical characteristics. The study's fndings demonstrate that the ML approach combined with the two-parameter Weibull distribution yields mechanical properties that exhibit comparability to the median observed fndings of the experiment. Young's modulus, strain at failure, and tensile strength are each depicted in Fig. [11](#page-13-1)a, b, and c, respectively. Lower mechanical property values have a high probability of survival. Tensile strength, strain, and tensile modulus values for the FSAA fibers for $GL = 100$ of 54.13 MPa, 1.24%, and 63 GPa. respectively, are needed to ensure 50% FSAA fbers survival, corresponding to a probability of $P(\sigma) = P(E) = P(E) = 0.5$.

Fig. 10 Weibull distribution with two parameters for mechanical properties of the FSAA for ML method. **a** Tensile strength, **b** elongation at break, and **c** Young's modulus

4 Conclusions

In conclusion, our research aimed to investigate and comprehend the properties of fber obtained from the *Agave* plant's fower stalk, with a particular focus on its viability as a reinforcing component in biocomposite materials. According to our fndings, these fbers have promising qualities that point to their potential for use in these kinds of applications. As a result, this research moves us one step closer to a day when ecologically friendly options are essential in the materials

Fig. 11 Probability of survival graphs of mechanical properties of FSAA with 2-parameter Weibull for ML method

business. The morphological, mechanical, chemical, thermal, and physical characteristics of FSAA fbers, which have a density of 980 kg/m³, were examined in this study.

The average diameter of the FSAA was approximately $(208 \pm 35 \text{ }\mu\text{m})$.

• FSAA fibers are well suited for low humidity applications due to their elevated moisture recapture (9.80%) and content (8.92%). However, the strength of the interfacial adhesion in reinforced composites may be impacted by their hydrophilic character.

- The existence of wax $(C-C)$, lignin $(C=C)$, hemicellulose (O—H), and cellulose (O=C) was confirmed using FTIR analysis.
- The XRD analysis confrmed that the CI of FSAA is 29.15%, the CS is 2.53 nm.
- The kinetic activation energy of the fibers was determined to be 68.15 kJ/mol using Broido's fgure. The degradation temperature of 339.57 °C has been identifed as the threshold at which the fbers begin to disintegrate. The thermal stability of the fbers is maintained between 200 and 250 °C. Beyond this range, the fbers start to deteriorate, leading to changes in their mechanical and physical properties.
- It appears that the fibers can absorb eight times the dry weight in the frst 72 h, reaching 880% and 868% for distilled water and seawater, respectively. The Peleg model was used to simulate the kinetics of distilled water and seawater absorption and provided improved statistical parameters when used with experimental data.
- The mechanical characteristics of thirty sample fibers (FSAA) tested in quasi-static tension showed a high Young's modulus (77.61 \pm 15.90 GPa) and with an average ultimate tensile stress of (σ = 64.34 \pm 11.43 MPa), strain at failure of $(1.5 \pm 0.31\%)$.
- The two specifications for the FSAA fibers, the Weibull modulus was calculated using the ML technique using the strength (m_{σ}) , and Young's modulus $(m_{\overline{E}})$, strain at break (m_{ϵ}) .

The experimental research in this work supported and confrmed the usage of fbers from the fower stalk of the *Agave americana* as reinforcement in novel composite materials used to produce biodegradable and biocompatible such as materials for medical implants, dressings, and drug delivery systems, and reinforced the materials, bumpers, dashboards, and door panels.

Author contribution Imen Lalaymia: this work involved the following processes — conceptualization, inquiry, methodology, writing, review, and editing. Ahmed Belaadi: conceptualization, investigation, methodology, supervision, writing, and review/editing were all integral parts of this work. Azzeddine Bedjaoui: methodology, investigation, and writing, including review and editing, were carried out for this work. Hassan Alshahrani: investigation and writing — review and editing. Mohammad K. A. Khan: investigation and writing — review and editing.

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Data availability Not applicable.

Code availability Not applicable.

Declarations

Ethics approval The work adheres to ethical standards, containing no libelous or unlawful statements. It respects the rights of others and does not include any material or instructions that may cause harm or injury.

Consent to participate The authors consent to participate.

Consent for publication The authors consent to publish.

Competing interests The authors declare no competing interests.

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