ORIGINAL ARTICLE



Synthesis and characterization of titania nanotubes by anodizing of titanium in fluoride containing electrolytes

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Received: 25 July 2017/Accepted: 22 September 2017/Published online: 6 October 2017 © The Author(s) 2017. This article is an open access publication

Abstract Titania nanotubular structure was prepared by anodizing titanium metal in the fluoride containing electrolytes and studied for hydrogen reduction using photo electrochemical cell. Potentiodynamic scan was performed before actual anodizing to optimize the anodizing conditions. The morphology of the TiO2 nanotubes was investigated by SEM and the presence of TiO2 nanotubes was confirmed. Raman spectroscopy was done to confirm the different phases present. Hydrogen generation capability was revealed by electrochemical testing in three-electrode system in dark and in visible light at 200 W power using Gamry Potentiostat. The corrosion potential of TiO₂ nanotubes produced was found to be more active side in potassium hydroxide solution under visible light than in the dark condition. Cathodic polarization behavior of specimens in the presence of light showed more activity towards hydrogen generation than in dark condition. In comparison, the hydrogen generation capability of specimen anodized in 2H15 electrolyte was higher than specimens anodized in other electrolytes. Electrochemical impedance spectroscopy was used to study the charge transfer resistance of the nanotubes produced. The results showed that TiO₂ nanotubular structure is a promising material for photoelectrochemical cell. Low-charge transfer resistance also depicts that it can be efficiently used to harvest solar energy.

Introduction

Nanotubular materials are of great importance due to their exceptional physical properties and potential applications. In terms of surface activity, open mesoporous structure offers highest surface area-to-volume ratio among many other nanostructured materials (Cochran et al. 2007; Lee et al. 2011). Moreover, titania nanotubes have attracted considerable attention because of their energy applications like in photo-catalysis, photo-splitting of water, gas-sensing devices, photovoltaic devices such as dye-sensitized solar cells, and photo electrochemical cells (Lee et al. 2014). Titanium metal has an excellent corrosion resistance due to its inherent tendency to form passive oxide film. The passive oxide film formed is a semiconductor with a wide bandgap (Dobrzański et al. 2016). Techniques used for the synthesis of nanostructured titania, ranging from nanotubes to nanowires and nanopores, include hydrothermal method (Kasuga et al. 1999; Nian and Teng 2006; Tsai et al. 2006; Zhang et al. 2003), solgel technique (Momeni and Ghayeb 2016), atomic layer deposition (Knez et al. 2007; Zazpe et al. 2016; Shin et al. 2004), and anodization (Endut et al. 2013; Macak et al. 2007; Salari et al. 2011). Anodization has received attention as it is comparatively cost-effective and represents an ideal solution for developing nanostructured TiO2 for use in the variety of fields (Bozzini et al. 2008; Ellingsen 1990).

The self-organized nanotubes or nanoporous morphology of TiO₂ on titanium surface can be produced in fluoride containing electrolyte by electrochemical anodization. The configuration, dimensions, shape, and activity of the



Keywords Titania nanotubes · Anodizing · Potentiodynamic scan · Harvest solar energy · Electrochemical

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grown nanotubes are affected by the process parameters. By varying the electrolyte composition and the potential applied, the diameter of the nanotubes in the desired narrow range can be achieved (Marichy et al. 2012). The length of the nanotubes can be increased or decreased by changing the pH of the electrolyte and/or by anodizing time (Sreekantan et al. 2009; Zwilling et al. 1999; Lim et al. 2012). To grow nanotubes having specific properties, the optimization of the anodizing condition is very critical. The pre-surface condition, nature of electrolyte (aqueous/organic), voltage ramp, fluoride ion concentration, and potential are all of great concern. The qualification of process parameters by potentiodynamic polarization of CP titanium in the desired electrolyte would be beneficial for optimization. Due to the pollution caused by and the depletion of fossil fuels, harnessing solar energy to achieve hydrogen as green energy carrier may be an alternative method to resolve the energy crisis. Since photo catalytic water splitting using TiO₂ as photo anode was successfully reported by Fujishima and Honda (Fujishima and Honda 1972), TiO₂ has been considered as the best hopeful material for this use owing to its extraordinary photo catalytic activity, low cost, nontoxicity, and photochemical stability (Nada et al. 2005; Kitano et al. 2007; Mishra et al. 2003; Paulose et al. 2006; Yi et al. 2008).

In the present work, TiO₂ nanotubes were synthesized using different electrolytes containing varying amounts of fluoride ions by keeping voltage and anodization time constant. A very novel technique in this current research was devised in order to lower the possible amounts of NH₄F (expensive compared to diammonium phosphate) using (NH₄)₂-HPO₄. These reduced amounts of NH₄F were proved to be beneficial in hydrogen generation and also in the application of Photo Electrochemical Cell. The morphology of synthesized nanotubes was investigated by using scanning electron microscopy. Raman spectroscopy was done to confirm the different phases of TiO₂ nanotubes present. As the nanotubes were synthesized for the application in photo electrochemical cell, synthesized titania nanotubes were characterized for their electrochemical performance. Three-electrode system was employed for the testing like corrosion potential, potentiodynamic scan, and electrochemical impedance spectroscopy. A comparison of the electrochemical trends was done by testing the synthesized nanotubes in the dark and in the presence of illumination source of known intensity.

Experimental

Materials

Commercially pure titanium metal was selected as substrate material in this research work. The chemical



composition of the material was determined by X-ray fluorescence (Table 1). The analytical grade chemicals, i.e., ammonium fluoride (NH₄F), diammonium phosphate ((NH₄)₂HPO₄), and ethylene glycol (C₂H₆O₂), were used for electrochemical anodization. Ethanol and acetone were used for cleaning. All the electrolytes were prepared in deionized water. Rinsing of specimens before and after anodizing was also carried out. (Ghicov et al. 2005; Tan et al. 2012; Lim et al. 2012).

Sample preparation for anodizing

The test specimens were prepared by the following method: titanium strip of 0.8 mm thickness was used as substrate material. The samples were cut into sizes of $25 \text{ cm} \times 12.5 \text{ cm} \times 0.8 \text{ mm}$. After cutting, the specimens were cold mounted and ground with emery papers of 100, 120, 180, 220, 320, 400, 600, 800, 1000, 1200, 1500, 2000, and 2500 grit size. The ground specimens were degreased in acetone and rinsed with deionized water. The samples were labelled as 2H, 2H15, and 2H75. After grinding and rinsing, one side of the specimen was insulated by the mounting material to make it nonconductive.

Electrochemical anodizing

The specimens were anodized in different electrolytes. The composition of prepared electrolytes and anodizing conditions are given in Table 2.

The anodizing of the specimens was performed in a twoelectrode cell system attached with a power supply (Matrix-MPS-3003L-3). Before connecting the sample to the anode, the power supply was turned on and required voltage was set by turning the voltage knob. Then, it was turned off after the voltage was set to the correct value. The working sample was made anode and graphite rod cathode. Properly clamped electrodes were immersed into electrolyte. Then, power supply was turned on and samples were anodized for required time. After anodizing, the samples were rinsed with deionized water and washed with ethanol. The specimens were then dried in the air.

The electrochemical optimization was investigated by means of potentiodynamic scans in electrolytes of different composition as given in Table 2. The specimens of titanium sheet were cut into $1.0 \text{ in} \times 0.5 \text{ in} \times 0.8 \text{ mm}$ for electrochemical testing. The surface of the titanium was prepared before electrochemical testing. The primary objective of surface preparation was that the surface should response similar to commercially produced materials in electrochemical analysis. The working electrodes were ground with emery papers of up to 2500 grit size. The ground working electrodes were degreased in acetone, rinsed with deionized water, and then dried in air. The

Table 1 Chemical composition of commercially pure titanium

Elements	Atomic%
Iron (Fe)	0.01
Oxygen (O)	0.30
Titanium (Ti)	Balance

Table 2 Electrolytes composition and anodizing parameters for anodization of specimens

Chemicals	2Н	2H15	2H75
$C_2H_6O_2$ (ml)	97	97	97
NH ₄ F (gm)	0.30	0.15	0.075
H_2O (ml)	3	3	3
$(NH_4)_2HPO_4$ (gm)	0.01	0.01	0.01
Voltage (V)	30	30	30
Time (min)	120	120	120

electrochemical study was conducted in three-electrode cell in which Ti sample was used as working electrode, graphite rod as counter/auxiliary, and standard calomel electrodes as reference electrode. The samples were electrically connected with wire. Equivalent weight and exposed surface area were calculated. The experimental parameters adjusted for optimization before anodizing for the bare specimens for Electrolytes 2H, 2H15, and 2H75 are given in Table 3.

Characterization

Anodized samples were observed at high magnification to reveal surface morphology at nanoscale by scanning electron microscope (Hitachi VP SEM S3400 N). The surface morphology of the anodized specimens was investigated by scanning electron microscope (SEM) (JSm-5600LV). The anodized specimens were investigated for the composition using energy dispersive X-ray spectroscopy (EDS).

Table 3 Input parameters for potentiodynamic scans

Parameters	Values
Initial potential (vs. EOC)	0 V
Final potential (vs. EOC)	6 V
Scan rate	10 mV/sec
Sample period	1 s
Sample area	2.58 cm^2
Equivalent weight	11.97 g
Density	4.5 g/cm^3

The anodized specimens were tested in dark as well as in the presence of light for the comparison. A bulb of 200 W power was used for illumination purpose. Corrosion potential test, photodynamic polarization test, and electrochemical impedance spectroscopy (EIS) were performed in 0.5 M KOH solution.

To investigate the corrosion potential $(E_{\rm corr})$ of the anodized specimens, corrosion potential test was performed. The input parameter for the test is given in Table 4.

The anodized specimens were cathodically polarized in three-electrode cell system to measure corrosion potential $(E_{\rm corr})$, and corrosion current $(I_{\rm corr})$. Potentiodynamic polarization scan can also help us to validate the passivity of the samples in the particular environment. The initial delay for open circuit potential was 300 s and stability 0.1 mV/s. During potentiodynamic polarization, the initial potential was -0.5 V versus open circuit potential, the final potential was 2 V versus open circuit potential and scan rate of 10 mV/s.

EIS was performed to observe the charge transfer behavior in that particular electrolyte. Tests were performed in dark as well as in the presence of light in 0.5 M KOH solution to observe the performance of the synthesized TiO₂ nanotubular structure. The test conditions and input values are given in Table 5.

Results and discussion

Electrochemical anodizing and optimization of anodizing conditions

Table 1 shows the chemical composition of as-received Titanium metal. The chemical composition was determined by X-ray fluorescence. The polished samples ground up to 2500 grit size were anodized in three different electrolytes as mentioned in Table 2, i.e., 2H, 2H15, and 2H75. The observed color of the anodized specimens was light grey with golden shade. The organic-based electrolytes were used for the anodization.

The potentiodynamic trends (current versus potential) of as-received samples were obtained to study the variation in current under constantly increasing potential at the rate of 10 mV/s up to potential of 6 V.

As evident in Fig. 1, the potentiodynamic scan for 2H sample shows that at low potential, there is an increase in value of current which depicts the dissolution of the oxide layer already formed on the surface of the substrate. After that the value of the current is slightly decreased which shows the formation of oxide layer. The value of current is then gradually increased to a maximum value, and after 1.5 V, value of current remained almost constant.



Table 4 Input parameters for corrosion potential

Parameters	Values
Initial potential (vs. open circuit potential)	0 V
Final potential (vs. open circuit potential)	6 V
Scan rate	10 mV/sec
Sample period	0.51 s
Total time	600 s
Sample area	2.3 cm^2
Equivalent weight	11.97 g

The low current value at higher potential suggested more dissolution rate of ${\rm TiO_2}$ than its growth and results in a nanotubular structure as confirmed by SEM analysis shown in Fig. 2. The presence of oxygen in the EDS analysis also confirms the formation of ${\rm TiO_2}$ layer (Diebold 2003; Mo and Ching 1995; Tang et al. 1994; Li et al. 2007; Yin et al. 2003; Kang 2003). The EDS spectrum of the anodized specimen in 2H is shown in Fig. 3 and elemental analysis is shown in Table 6.

In 2H15 electrolyte, the value of current reaches to $190 \mu A$ at low voltage and decrease gradually, and after that, the value of current becomes constant up to 6 V. Potentiodynamic curve of the sample is shown in Fig. 4.

SEM micrograph of the specimen has been shown in Fig. 5, after anodizing, and confirmed the formation of nanotubular structure. The EDS spectrum of 2H15 specimen is shown in Fig. 6. The weight% and atomic% composition of the anodized specimen is given in Table 7. The presence of 35.77 weight% oxygen confirmed the oxide formation after anodizing.

Potentiodynamic scan for 2H75 sample is shown in Fig. 7. As evident from the figure, the current initially decreases near zero potential and then increases to $82~\mu A$ which is then became constant up to 6 V. This shows dissolution and growth of oxide layer, respectively (Kim et al. 2003; Kolen et al. 2003; Hoyer 1996).

Anodizing results in almost uniform porous structure with a little dissolution were confirmed by SEM analysis in

 Table 5 Input
 parameters
 for
 electrochemical
 impedance

 spectroscopy

эресиозгору		
Parameters	Values	
Initial frequency	10,000 Hz	
Final frequency	0.01 Hz	
Points/decade	10	
AC voltage	10 mV	
DC voltage	0 V	
Sample area	2.3 cm^2	
Equivalent weight	11.97 g	



Fig. 8. The EDS spectrum of sample is shown in Fig. 9 which shows the presence of oxygen and confirmed the oxide formation. The EDS analysis for 2H75 is shown in Table 8.

Raman spectra

The anodized samples were characterized using Raman spectroscopy. Figure 10 shows Raman spectra of anodized samples. The spectra shows different phases of titania present having different intensities in different electrolytes. Some of the peaks were observed slightly shifted from the standard pattern, while some peaks were broadened as observed by the previous research workers (Gong et al. 2001; Ghicov et al. 2005; Bauer et al. 2006).

Electrochemical analysis

Potentiodynamic, corrosion potential, and electrochemical impedance spectroscopy analysis were examined to characterize the hydrogen reduction behavior of anodized samples in 0.5 M KOH solution in dark and in the presence of light. For illuminated characterization, the samples were irradiated by Philips Electric Fluorescent Bulb of 200 W power having 2725 Lumens intensity. The cell was placed in box to insulate it from surrounding light.

The electrochemical behavior of 2H, 2H15, and 2H75 samples was characterized by open circuit potential in 0.5 M KOH solution. The behavior of corrosion potential of 2H sample, in dark and in the presence of light, is shown in Fig. 11. The scans were obtained by exposing the surface area of 2.3 cm² in 0.5 M KOH solution for 600 s using Gamry Potentiostat (PC14/750).

The corrosion potential of first sample is initially around -0.298 V which gradually decreases to -0.322 V when tested in the absence of any illumination. The potential of the sample, when tested in the presence of light, was around -0.431 V which remains constant for the period of 600 s and reached to -0.435 V. The testing shows that the 2H sample has more negative potential in the light than dark which shows that more energetic electrons are available for reduction of H_2 .

As shown in Fig. 12, the corrosion potential for 2H15 in dark is the same as for 2H, but in the presence of light, it is more negative than 2H. The corrosion potential of sample is initially around -0.325 V which gradually decreases to -0.273 V when tested in the absence of any illumination. The potential of the sample, when tested in the presence of light, was around -0.508 V which remains constant for the period of 600 s and reached to -0.525 V. This indicates rate of H_2 reduction is higher.

The corrosion potential for 2H75 in KOH solution is shown in Fig. 13. The corrosion potential for 2H75 is

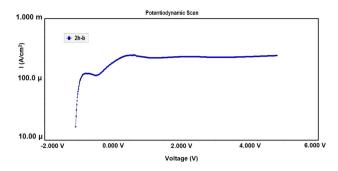


Fig. 1 Potentiodynamic scan for 2H sample

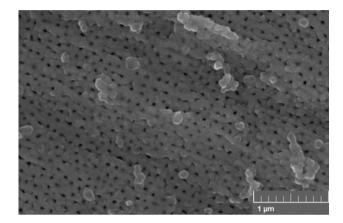


Fig. 2 SEM image of 2H sample

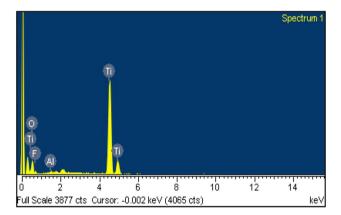


Fig. 3 EDS analysis of 2H sample

positive both in dark and in the presence of light as

Table 6 Elemental analysis of 2H sample

Elements	Weight%	Atomic%
Iron (Fe)	9.02	12.30
Oxygen (O)	35.44	57.44
Aluminium (Al)	0.45	0.43
Titanium (Ti)	55.09	29.83
Total	100.0	100.0

compared to other two samples (2H and 2H15). This shows that $\rm H_2$ reduction rate is lower in case of 2H75. The corrosion potential of the first sample was initially around $-0.129~\rm V$ which gradually decreases to $-0.124~\rm V$ when tested in the absence of any illumination. The potential of the sample, when tested in the presence of light, was around $-0.353~\rm V$ which remains constant for the period of 600 s and reached to $-0.391~\rm V$. The graph in Fig. 14 shows the trend of the corrosion potential of all the specimens in dark as well as in irradiated conditions and is in confirmation with the previous researchers (Ratner et al. 2008; Yao and Webster 2006; Marcus 2002).

The anodized samples were studied by potentiodynamic scans. The samples were cathodically polarized to see the capability of the specimen for hydrogen reduction. The potentiodynamic scan of specimen 2H is shown in Fig. 15. The specimen shows increase in the current density in the presence of light. The light promotes the reduction and, hence, the current density as a facilitating source for the voltage which imparts energy to overcome the reduction potential barrier. The value was 29.70E–06 A/cm² in the absence of light. This value increases to 71.78E–06 A/cm² which showed increase in the activity towards hydrogen reduction.

In case of the specimen 2H15 (the potentiodynamic scan has been shown in Fig. 16), the activity of the anodized sample was observed to be increased from 13.88E-06 A/cm² in dark to 78.56E-06 A/cm² in the presence of light. This specimen shows an increase in activity to split the water in the presence of light as found by the previous researchers (Taveira et al. 2006; Sul et al. 2001; Meekins 2012; Yao and Webster 2006).

The specimen 2H75 (the potentiodynamic scan has been shown in Fig. 17) showed a different behavior than the other two samples. It shows a decrease in the current density when compared its performance in dark with that in the presence of light. The value of current was 74.02E-06 A/cm² in dark which is decreased to 9.252E-06 A/cm² when tested in the presence of light.

The Nyquist plot [Impedance (real) vs. Impedance (imaginary)] for specimen 2H is shown in Fig. 18. The anodized specimen showed an increase in $Z_{\rm real}$ component in the presence of light than in the absence of light. This depicts an increase in charge transfer resistance of the anodized sample. The value of $R_{\rm ct}$ was 9.38E–05 S*s^a initially which slightly increase to 9.56E–05 S*s^a.

In case of 2H15 specimen, the value of $R_{\rm ct}$ was $2.86E-04~{\rm S^*s^*a}$ in dark which then decreased to a value of $1.11E-04~{\rm S^*s^*a}$. This recession in the value of $R_{\rm ct}$ depicted a decrease in the charge transfer resistance which means more capability to reduce the hydrogen in the presence of light. The EIS scan is shown in Fig. 19.



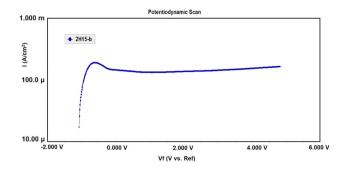


Fig. 4 Potentiodynamic scan for 2H15 sample

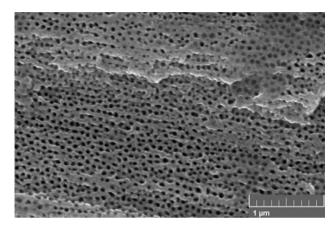


Fig. 5 SEM image of 2H15 sample

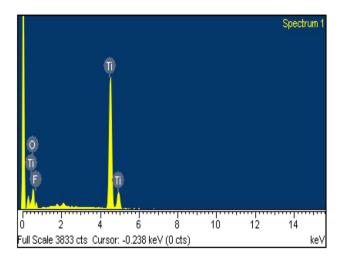


Fig. 6 EDS analysis of 2H15 sample

Table 7 Elemental analysis of 2H15 sample

Weight%	Atomic%
10.39	14.21
35.52	56.96
54.09	28.83
100.0	100.0
	10.39 35.52 54.09

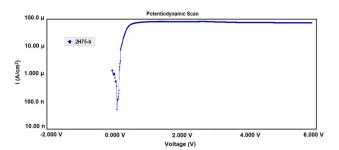


Fig. 7 Potentiodynamic scan for 2H75 sample

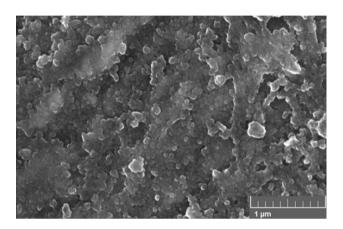


Fig. 8 SEM image of 2H75 sample

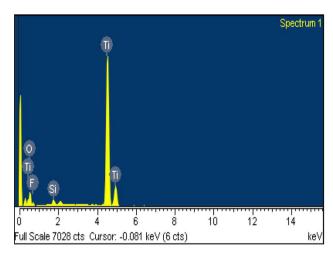


Fig. 9 EDS analysis of 2H75 sample

Table 8 Elemental analysis of 2H75 sample

Elements	Weight%	Atomic%
Iron (Fe)	5.30	9.42
Oxygen (O)	29.28	51.98
Silicon (Si)	0.96	0.96
Titanium (Ti)	64.46	37.64
Total	100.0	100.0



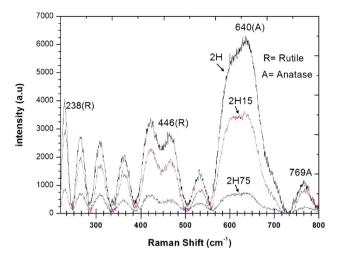
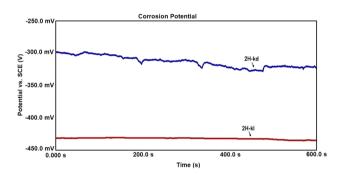


Fig. 10 Raman spectra of anodized samples



 $\begin{tabular}{ll} Fig. \ 11 & Corrosion potential of 2H sample in KOH solution in dark and in light \\ \end{tabular}$

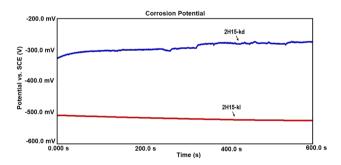


Fig. 12 Corrosion potential of 2H15 sample in KOH solution in dark and in light

The specimen 2H75 showed a decrease in the charge transfer resistance in the presence of light, as shown in Fig. 20. The value of $R_{\rm ct}$ was initially $8.01E-05~{\rm S*s^a}$ in dark which decreases to a value of $1.04E-04~{\rm S*s^a}$. This trend shows that the specimen anodized in 2H57 electrolyte has less charge transfer resistance and, hence, more potential to split the water when tested in the presence of light. Figure 21 summarizes EIS trends of all anodized samples in dark and in the presence of light.

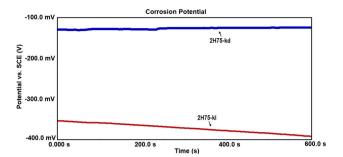


Fig. 13 Corrosion potential of 2H75 sample in KOH solution in dark and in light

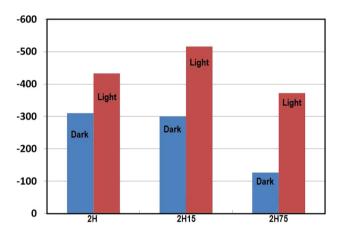


Fig. 14 Trend of CP of specimens in dark and light

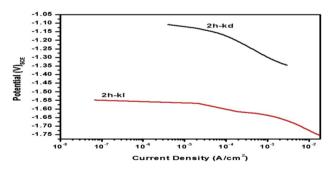


Fig. 15 Potentiodynamic scan of 2H sample in dark and in light

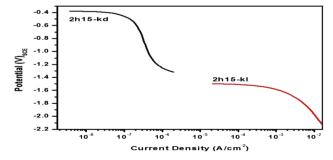


Fig. 16 Potentiodynamic scan of 2H15 sample in dark and in light



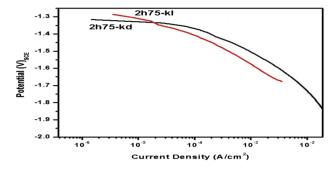


Fig. 17 Potentiodynamic scan of 2H75 sample in dark and in light

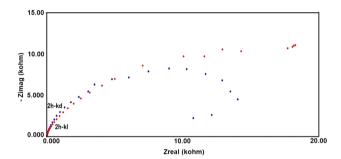


Fig. 18 EIS scan of 2H sample in dark and in light

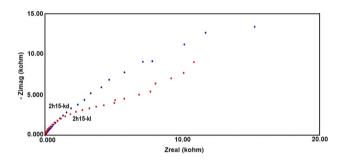


Fig. 19 EIS scan of 2H15 sample in dark and in light

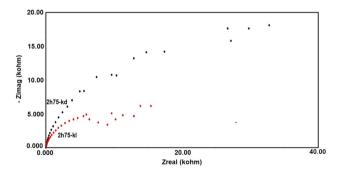


Fig. 20 EIS scan of 2H75 sample in dark and in light

Conclusions

The conditions to produce self-organized nanotubular structure on titanium were optimized by anodic potentiodynamic polarization scan. The formation of ${\rm TiO_2}$ nanotubes in fluoride-based electrolytes was found to be a complex phenomenon. It



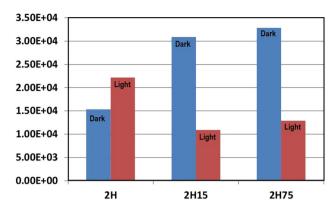


Fig. 21 EIS trends of anodized samples

was found that both the voltage and fluoride ions concentration affect the growth of TNT's equally. The conditions to produce self-organized nanotubular structure on titanium at 30 V and for 120 min were optimized by potentiodynamic polarization scan. It is found that the concentration of fluoride ions affects the growth of TNTs. The current research work is emphasizing the use of (NH₄)₂HPO₄ to modify the extent of self-organization of titania nanotubes. The use of (NH₄)₂HPO₄ also reduces the amounts of NH₄F to be used in electrolyte and enhanced the capability of hydrogen generation. It is found that the concentration of fluoride ions affects the growth of TNT's. The extent of self-organization was modified with the little addition of (NH₄)₂HPO₄ in the electrolyte. Raman spectroscopy confirms the presence of nanotubular structure. The presence of oxygen in the EDS analysis shows the formation of oxide layer. The more negative potential in light shows increase in surface activity which depicts higher hydrogen reduction than in dark. The cathodic polarization trend showed increase in hydrogen reduction reaction in the presence of light. The electrochemical impedance spectroscopy validated the decrease in charge transfer resistance on exposure to light. The electrochemical behavior depicting in hydrogen generation tendency trend was exceptionally lower on sample anodized in 2H solution compared to 2H15 and 2H75 solution which may be related to surface morphology obtained. Comparatively higher efficiency of hydrogen reduction at the surface of 2H15 was observed as validated by corrosion potential, potentiodynamic, and electrochemical impedance spectroscopy. It is concluded from the results that optimum conditions for the formation of nanotubular structure and its performance towards hydrogen generation were relatively higher for the specimen anodized in 2H15 electrolytes.

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