#### **ORIGINAL PAPER**



# **Morphological, Optical, and Crystalline Analysis of ZnTiO3 Nanostructures Deposited on Porous Silicon Substrate**

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#### **Abstract**

Zinc titanate ( $ZnTiO<sub>3</sub>$ ) was grown on silicon and porous silicon.  $ZnTiO<sub>3</sub>$  layers were prepared by sol-gel method. Porous silicon was fabricated by electrochemical etching of silicon in HF solution. The efect of substrate porosity on morphology, structure and otpical properties of  $ZnTiO<sub>3</sub>$  nanostructures has been studied. Theses properties were investigated using XRD, Ultraviolet–Visible spectroscopy, and HRTEM. Some important parameters (absorption, refectivity (R (%) and grain size) were studied. It was found that the Structural, morphology and optical properties of ZnTiO<sub>3</sub> layers are dependent strongly on the type of substrates. The crystalline size decreased for  $ZnTiO<sub>3</sub>$  layers deposited on PS substrate. The average grain size is about 80 nm for  $ZnTiO<sub>3</sub>$  grown on porous silicon. The surface morphology of films was also found to be uniform and homogeneous. ZnTiO<sub>3</sub>-PS shows enhancing photon absorption compared to ZnTiO<sub>3</sub>-Si.

Keywords ZnTiO<sub>3</sub> · Porous silicon · UV-Vis · XRD · HRTEM

# **1 Introductuion**

Metal oxide semiconductors have recently shown great infuence in the fields of photocatalysis  $[1–3]$  $[1–3]$  $[1–3]$  $[1–3]$ , optoelectronics [\[4](#page-5-2)[–6](#page-5-3)], and solar cells [\[7](#page-5-4)]. Amongst the studied semiconductors, Zinc oxide (ZnO) has received much attention. It is a wide-bandgap oxide semiconductor with a direct energy gap of about 3.37 eV. ZnO has high chemical and mechanical stability; furthermore, it is nontoxic and widespread in nature  $[8, 9]$  $[8, 9]$  $[8, 9]$  $[8, 9]$ . TiO<sub>2</sub> has also shown promise in the areas of photocatalysis [\[2](#page-5-7)], solar cells [[10](#page-6-0)], gas sensor and other optical applications [\[11](#page-6-1), [12\]](#page-6-2). Enormous increase of applications based on  $ZnO$  and  $TiO<sub>2</sub>$  has been caused not only by the improvements

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of their intrinsic properties but also by the achievements of transition metal doping and mixed oxides formations.

So, TiO<sub>2</sub> and ZnO have been used to fabricate ZnO-TiO<sub>2</sub> composite materials, annealing conditions and  $ZnO/TiO<sub>2</sub>$ molar ratio were found have signifcants efects on the phase formation [[13](#page-6-3)]. The coupling  $TiO<sub>2</sub>/ZnO$  gives a material named zinc titanate that enhances the properties of ZnO and  $TiO<sub>2</sub>$ , for example, by widening its light absorption spectrum [\[14\]](#page-6-4). Additionally, the photocatalytic activity of oxides may help reduce the susceptibility of pollutants to form aggregate structures [\[15\]](#page-6-5).

However, zinc titanate system has three compounds that exist in the:  $ZnTiO<sub>3</sub>$ ,  $Zn2Ti<sub>3</sub>O<sub>8</sub>$ , and  $Zn<sub>2</sub>TiO<sub>4</sub>$ , which it remains as a structural defect.  $ZnTiO<sub>3</sub>$  is a perovskitetype oxide  $(ABO<sub>3</sub>)$  whose perovskite structure endows the flexibility to alter the arrangement of the A- and B-sites, and incorporates cation combinations at the A- and B-sites to assemble substituted perovskites.  $\text{ABO}_3$  have been considered as materials with various applications in solid oxide fuel cell electrodes (SOFC) [[16](#page-6-6)], metal barriers [\[17\]](#page-6-7), sensors [[18](#page-6-8)], electronics [[19\]](#page-6-9) and catalysts [[20](#page-6-10)]. As a well-known member of this family, zinc titanate  $(ZnTiO<sub>3</sub>)$  has been used as pigments [[21\]](#page-6-11), dye adsorbents [[22](#page-6-12)], sensors such as NO and CO gases [\[23\]](#page-6-13), microwave resonator materials [[24\]](#page-6-14), heat reflective pigments [[25\]](#page-6-15) and Photocatalyst [[26](#page-6-16)].

Several methods are used to synthesize Zinc titanate  $(ZnTiO<sub>3</sub>)$ , as solid state reaction [[27](#page-6-17)], Sol-gel [[28](#page-6-18), [29](#page-6-19)], Pechini process [\[30](#page-6-20)], Hydrothermal method [\[31\]](#page-6-21), Sputtering [\[32](#page-6-22)], Microwave heating [\[33](#page-6-23)], and molten salt Method [\[34\]](#page-6-24). Zinc titanate  $(ZnTiO<sub>3</sub>)$  thin films deposited on different substrates. The investigation of the infuence of various substrates such as Si (100) [[35](#page-6-25)], glass [\[36](#page-6-26)], quartz substrate [\[37\]](#page-6-27), and ITO coated glass [[38](#page-6-28)] has also been studied.

In This work, porous silicon (PS) is another candidate as substrates due to adjustable roughness (large internal surface), high resistance, strong absorbability, and potential for the development of silicon-based optoelectronic devices. Porous silicon has been used to deposit metal oxides and obtain good quality thin layers [\[39–](#page-6-29)[41\]](#page-7-0). we report the growth of  $ZnTiO<sub>3</sub>$  hexagonal and nanoscale on silicon and porous silicon substrates by spin coating. Optical, morphology and crystal structure of hexagonal and nanoscale were studied by UV-Vis, XRD and TEM techniques. The experimental fndings on the efect of the sponge- like structure of the porous silicon substrates on the crystallinity properties of the  $ZnTiO<sub>3</sub>$  thin films are presented and the causes are discussed.

# **2 Experimental Details**

For the fabrication of nanostructures, porous Si (PSi) substrates were obtained by electrochemical etching [\[40](#page-7-1), [41](#page-7-0)] of p-type, (100) oriented silicon (Si) wafers with a resistivity of 10–20  $\Omega$  cm. For the preparation of the porous substrates, c-Si wafers were cleaned with 2-propanol (under sonication) for 4 min. The Si wafers were rinsed with deionized water, dried with N2 flux, and immersed in hydrofluoric acid aqueous solution (2%) for 4 min, followed by rinsing with deionized water and ethanol and dried under N2 fux. The etching process was carried out for 15 min in hydrofuoric acid (HF, 40 wt%) and ethanol (1:1 volumetric ratio) solution at a constant current density of 10 mA/cm2. After the etching process, the substrates were rinsed with ethanol and dried with N2 fux.

On the other hand,  $ZnTiO<sub>3</sub>$  nanostructured films on PSi and Si substrates were obtained by spin coating deposition. The formation of  $ZnTiO<sub>3</sub>$  and the corresponding characterization have been widely studied in the literature [[28,](#page-6-18) [29,](#page-6-19) [35\]](#page-6-25). Zinc titanate thin flms was prepared by the sol-gel method. In general, Zinc acetate dihydrate (Zn (CH3COO)2·2H2O, Sigma Aldrich, reagent grade, 99% purity), and Titanium (IV) n-butoxide (Ti(O(CH2)3CH3)4, Sigma Aldrich, reagent grade, 99% purity) were used as Zn and Ti source materials, respectively. The Zinc acetate and tetrabutyl titanate were dissolved into ethylene glycol monoethyl ether and acid acetic acid solvent at 80 °C and stirred for 30 min to form clear solution.

For the deposition, PSi and Si substrates with a dimension of 20 mm $\times$  20 mm were used. The substrates were washed successively with acetone, hydrochloric acid aqueous solution, deionized water and absolute ethanol in an ultrasound bath. Then, dried at 100 °C for 10 min before coating. ZnTiO3 precursor solution was spin-coated on the sustrates at 3000 rpm during 30 sec. The as-prepared flms were annealed at 120 °C for 10 min to remove organic materials and then at 700 °C for 2 hours to crystallize them into a perovskite structure in a rapid thermal annealing furnace. The process from coating to annealing was repeated 2, 4, 6 and 8 times to produce diferent thickness of the flms. The as prepared and sintered samples were subjected to various analyses by suitable analytical technique.

The structural properties of the prepared materials were studied by XRD analysis using (BRUKER D8 advance model, at room temperature). The morphology of  $ZnTiO<sub>3</sub>$ was analyzed by high resolution transmission microscopy (TEM), using a HRTEM JEOL2100F microscope. UV-vis absorbance and refectance spectroscopy analysis (UV-vis) was carried out using Perkin Elmer Lamda 950 spectrophotometer.

## **3 Result and Discussion**

#### **3.1 XRD Analysis**

The thicknesses of the flms resulting of the 2, 4, 6 and 8 repetitions of spin coating and annealing were found to be 80 nm, 120 nm, 160 nm and 240 nm.

To study the efect of substrate (Silicon, Porous silicon) on crystal structure of ZnTiO<sub>3</sub>, we performed XRD experiments on the samples. The XRD analysis of  $ZnTiO<sub>3</sub>/Si$  and  $ZnTiO<sub>3</sub>/PSi$  were reported as in Fig. [1](#page-2-0). The crystal planes (104), (110), (024), (116), (214), for hexagonal  $ZnTiO<sub>3</sub>$ (JCPDS card No.26–1500) in the difractograms can be indexed [[30,](#page-6-20) [42,](#page-7-2) [43\]](#page-7-3).

The (104) peak is more intense in the  $ZnTiO<sub>3</sub>/PSi$ compare to ZnO/Si. The peak (110) peak appeared in the  $ZnTiO<sub>3</sub>/PSi$  compare to  $ZnO/Si$ . Consequently, there traduces better crystallinity of  $ZnTiO<sub>3</sub>$  formed on PSi than on Si. The diference in intensity can be explained by higher absorption rate of the capillary effect presented in  $\text{ZnTiO}_3$ / PSi sample and the high adhesion due to the high specifc surface area in the case of PSi.

The average crystallite sizes of the nanoparticles  $ZnTiO<sub>3</sub>$ can be estimated using Scherrer equation which is defned as  $d = (0.94 \text{ k})/(b \cos(\theta))$ , where d is the average grain size, k is the X-ray wavelength (0.15406 nm), b is the full-width at half maximum (FWHM), and  $\theta$  is the diffraction angle (32.9°). The strongest peaks (104) in XRD were used to calculate the grain size for  $ZnTiO<sub>3</sub>$ . Applying Scherrer's

<span id="page-2-0"></span>**Fig. 1** X-ray difraction of  $ZnTiO<sub>3</sub>$  nanostructures grown on conditions on (**a**) Si substrate, (**b**) PSi substrate



formula, the grain sizes were found to be 45 nm and 20 nm for  $ZnTiO<sub>3</sub>$  formed on PSi and  $ZnTiO<sub>3</sub>$  formed on Si, respectively. The obtained values indicate that the porous layer has a significant effect on the synthesis mechanism of  $ZnTiO<sub>3</sub>$ nanostructures and can serve as the starting point for the growth of nanostructures. Therefore, the rough surface morphology of PSi plays a major role in controlling the growth of the wettinglayer [\[44](#page-7-4)]. Due to its special surface morphology, the porous layer is a good substrate for latticemismatched heteroepitaxy. The surface of the porous silicon layer is composed of many nanocrystals. These Si nanocrystals maintain (100) orientation with the outer surface of the silicon wafer. These randomly distributed Si crystallites on the surface act as nucleation sites and induce the growth of  $ZnTiO<sub>3</sub>$  nanostructures along the preferred orientation.

## **3.2 Optical Properties**

Optical properties of the  $ZnTiO<sub>3</sub>/Si$  and  $ZnTiO<sub>3</sub>/PSi$  samples have been investigated using UV-Vis spectroscopy. Figure [2](#page-2-1) compares the absorbance of  $ZnTiO<sub>3</sub>/Si$  and  $ZnTiO<sub>3</sub>/PSi$ . The layer thickness of  $ZnTiO<sub>3</sub>$  for two samples is on the order of 120 nm. As shown in Fig. [2,](#page-2-1) the absorbance recorded in the spectral range 250–900 nm. The spectrum a corresponding to the samples reveals three bands. A first UV absorption band extending from 250 to 270 nm with a sharp band located at about 260 nm, second band with a centered at about 350 nm, and three absorption band extending from 350 to 900 nm.

The band edge is observed at  $\sim$ 280 (first band) and  $\sim$  400 (second band) nm for all samples. Those bands edge absorption intensity show its absorbance capacity in UV light of



<span id="page-2-1"></span>**Fig. 2** UV-Vis spectra of  $ZnTiO<sub>3</sub>$  nanostructures grown on conditions on (**a**) Si substrate, (**b**) PSi substrate

 $ZnTiO<sub>3</sub>$  it is similar as reported in the literature [\[45](#page-7-5), [46](#page-7-6)]. The absorbance increases with the porous layer due to the increasing to optical path in porous silicon specifc surface. The increase in the visible absorbance (three band) confirmed that the  $ZnTiO<sub>3</sub>$  nanoparticles are emerged in the pores and therefore are deposited on the specifc surface. We deduce that the  $ZnTiO<sub>3</sub>/PS$  acts as an efficient solar absorber.

Optical absorbance spectra of multilayer  $ZnTiO<sub>3</sub>$  films on porous silicon are shown in Fig. [3.](#page-3-0) It is seen from this Figure; the spectrum is similar to Fig. [2](#page-2-1) with variation of absorbance intensity. The results show the strongest absorbance in the UV region of 250–400 nm increase with layer A(u.a.)

 $0<sub>0</sub>$ 0,1 0,2  $0.3$  $0.4$ 0,5 0,6 0,7 0,8 0,9 1,0

<span id="page-3-0"></span>

300 400 500 600 700 800 900 1000

 $\lambda$ (nm)

 $\mathsf{E}_3$ 

PSi<br>E<sub>1</sub> = ZnTiO<sub>3</sub>/PSi (80nm)<br>E<sub>2</sub> = ZnTiO<sub>3</sub>/PSi (120nn = ZnTiO<sub>3</sub>/PSi (120nm)

= ZnTiO<sub>3</sub>/PSi(160nm)  $E_4 = ZnTiO_3$ 

/PSi(240nm)

thickness of  $ZnTiO<sub>3</sub>$ . Figure [3](#page-3-0) shows the absorption intensity increase of  $E_1$ , compared to porous silicon and the other samples. The enhanced absorption of  $ZnTiO<sub>3</sub>$  due to the Surface Plasmon Resonance (SPR) of the free electrons and  $ZnTiO<sub>3</sub>$  incorporated in pores.

Figure [3](#page-3-0) shows the absorption intensity in visible region decrease with layer thickness of  $ZnTiO<sub>3</sub>$ . It is because the layer become thick, getting larger and smoother, causing a reduction in the absorption optical.

The sample  $E_4$  has high UV absorbance, which indicates this sample has synergistically enhanced UV absorption behavior. This is very beneficial for enhanced anti-UV aging performance. These conclusions are consistent with the statistical results as shown in Fig. [4](#page-3-1).

Figure [4](#page-3-1) shows UV–vis refectance spectra of PS (black line),  $E_1$  (red line),  $E_2$  (green line),  $E_3$  (blue line) and  $E_4$ (light blueline) structures for the wavelength range of 250 nm–900 nm. The reflectivity of the porous silicon surface without  $ZnTiO<sub>3</sub>$  was around 14.3% and decreased to around 13.4% after layer of  $ZnTiO<sub>3</sub>$  deposed on porous silicon at around 450 nm. This is due to the formation of needle-like structures that result in enhanced light trapping. The decreased refectivity results in signifcant increase of absorbance in  $E_1$ . It can be observed from the spectra that the optical reflectance spectra of  $nZnTiO<sub>3</sub>/PS$  increased signifcantly with thickness layer. The increased refectance results in significant decrease of absorbance in  $E_2$ ,  $E_3$  and  $E_4$ .

### **3.3 HRTEM Study**

The microstructural information of the samples was obtained using transmission electron microscopy (TEM). Figure [5a](#page-4-0) shows a TEM image of  $ZnTiO<sub>3</sub>$  grown on Si (a, c) (A) and



<span id="page-3-1"></span>**Fig. 4** Reflectivity of  $ZnTiO<sub>3</sub>$  nanostructures with different thickness grown on PSi substrate

PS (b) (B). As can be seen, the  $ZnTiO<sub>3</sub>$  was fully crystalline at the nanoscale and the formation of irregular spherical shaped. The TEM images (Fig. [5](#page-4-0)) suggest that the submicrosized particles of A and B are both crystalline with no apparent defects and dislocations. The average diameters of the smallest visible isolated particle/crystallite agglomerate were found to range between 100 nm and 120 nm for A and the particles B with the size of about 80 nm. These results are in excellent agreement with the experimental of values obtained from XRD.

Figure [6](#page-4-1) shows surface morphology of  $\text{ZnTiO}_3$  particle. Note that the microstructure of B becomes a sponge. This last composed of nanopores. On the other hand, the structure is smooth and lacks pores for A. These results are in parallel with UV-Vis results. The porous structure of  $ZnTiO<sub>3</sub>$  influences the increases light absorption (UV-Vis) compared to A (Fig. [2\)](#page-2-1).

The HRTEM image of the sample is presented in Fig. [7.](#page-5-8) The distance between the adjacent lattice fringes is 0.38 and 0.1 nm for A and B, which can be assigned to the interplanar distance of the hexagonal phases of A and B. Compared with compound A, the crystallinity of B is better. Comprehensively, the results confrmed that the  $ZnTiO<sub>3</sub>$  layers were successfully prepared at annealing temperatures at 700 °C.

# **4 Conclusion**

In summary, we have synthesized nanoscale  $ZnTiO<sub>3</sub>$ on PS and Si. The dependence of the structural and optical properties of these nanostructures on the different substrates was investigated systematically. The

<span id="page-4-0"></span>





<span id="page-4-1"></span>**Fig. 6** TEM image of the surface ZnTiO<sub>3</sub> grown on Si (**a**) and PS (**b**)



<span id="page-5-8"></span>



nanostructures were polycrystalline in nature, (410) plane was the preferred orientation, and showed decreasing crystal grain size with porous silicon substrates. A strongest UV-Vis absorption intensity for  $ZnTiO<sub>3</sub>$ nanoscale have been obtained on PS substrate compared to  $ZnTiO<sub>3</sub>$  as grown on Si substrate. This is due to the capillary effect and its high specific surface area of PS. With HRTEM, We compared the structure and morphology of  $ZnTiO<sub>3</sub>$  nanostructures grown on porous silicon and silicon substrate. The average diameter and the distance between the adjacent lattice fringes of these  $ZnTiO<sub>3</sub>$  on porous silicon decreased from 120 to 80 nm and from 0.38 to 0.1 nm, respectively.

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**Data Availability** Not applicable**.**

## **Declarations**

**Ethics Approval** Not applicable.

**Consent to Participate** Not applicable.

**Consent for Publication** The Author hereby consents to the publication of the work in the "Silicon" journal.

**Competing Interests** The author declares that there is no confict of interest in the printing of this manuscript.

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