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Effect of Nanosilica on the Mechanical Properties, Compression Set, Morphology, Abrasion and Swelling Resistance of Sulphur Cured EPDM/SBR Composites

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Abstract

The main objective of the current research work is to explore the effect of nanosilica particles on the compound EPDM/SBR-SiO₂ (ethylene-propylene-diene monomer/styrene-butadiene rubber-nanosilica). The composite EPDM/SBR with and without silane coupling agent was processed using an open mill mixer. The nanosilica particles are prepared in the laboratory and were used as the reinforcing material in EPDM/SBR rubber composites. The cure characteristics, mechanical properties, hardness, rebound resilience, swelling resistance, abrasion resistance and compression set of the composites are completely analyzed and studied. Nanosilica particles are produced in the laboratory and used as reinforcement material in EPDM/SBR rubber compounds. Fully analyzed and examined are the cure characteristics, mechanical properties, hardness, rebound resilience, swelling resistance and compression collection of the composites. It was also evident from the result that with the inclusion of nanosilica particles in the EPDM/SBR rubber composites, the mechanical properties, swelling resistance, hardness, abrasion resistance and compression set of the composites. It was also evident from the result that with the inclusion of nanosilica particles in the EPDM/SBR rubber composites, the mechanical properties, swelling resistance, hardness, abrasion resistance and compression set properties improved.

Keywords EPDM/SBR · Silica nanoparticles · Mechanical properties

1 Introduction

Rubber and its blends are commercially used in industries for engineering applications. In particular, the rubber blends exhibits inimitable tailor made properties required for engineering materials. It also decreases the total cost of the product and, to a greater degree, increases its processability [1]. Ethylene propylene diene monomer (EPDM) was a saturated and non-polar rubber with high ozone, weathering and thermal resistance among the various forms of rubber materials. EPDM was generally used in automotive industries and construction industries. Generally, EPDM exhibits a lesser adhesion property but its cost was comparatively at the high side as compared with other rubber compounds. To surmount the shortcomings in EPDM rubber, it was blended with styrene-

G. Anand anand.12ng@gmail.com butadiene rubber (SBR) material [2]. There are a broad variety of engineering applications for the EPDM/SBR blend, such as conveyor belts, seals, electrical insulators, gaskets, tires, tubing, etc. [3]. The blended rubber compounds thus ensures higher mechanical properties and abrasion resistance, when it was mixed in the ratio 80: 20 of EPDM/SBR [4]. These rubber blends are also appropriately used as the matrix material for various industrial and engineering applications. The schematic representation of the EPDM/SBR rubber blends is shown in Fig. 1.

In recent times, carbon black (CB) and amorphous silica have been widely used to enhance the mechanical properties of rubber blends in the manufacturing of materials [5–7]. Silica has been abundantly present in nature and can easily be extracted from natural agricultural sources such as groundnut shell, bagasse, rice husk, etc. Amongst the different sources of natural silica, rick husk was the most promising one used in the manufacturing of nanosilica. Rice husk was an agricultural byproduct that was abundantly available in rice-producing countries like India, China, and other Asian countries [8]. The removal of rice husk debris was a dreary procedure since it prompts the antagonistic impact to nature such nursery impact during the emanation procedure, air

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Fig. 1 The proposed interaction of EPDM/SBR blends

contamination, inhabitance of landfill space and vitality squander, and so forth [9, 10]. The preparation of nanosilica from the rice husk therefore reduces the antagonistic ecological effect, furthermore financially helpful considered to industrially accessible silica nanoparticles. The significant elements of rice husk are natural materials (i.e., cellulose and lignin) and hydrated silica; this sort of silica was normally undefined in nature.

The researchers have detailed the various techniques to get ready silica from rice husk debris previously, for example, soluble base and sulfuric corrosive treatment [11], sol-gel strategy [12], calcination process [13] and compound precipitation amalgamation [14]. In this research work, antacid and sulfuric corrosive treatment technique dependent on disintegration process and followed by precipitation process was utilized to extricate silica from the agronomically squander rice husk. The silica got through this method was of high immaculateness with the width of 20-40 nm in run. Silica nanoparticles are one of the generally utilized as viable strengthening filler in the rubber blends due to its high perspective proportion. The nanofillers utilized in the composite equally used to enhance the different properties of matrix material in nanocomposites. These nanofiller materials are additionally utilized as the adsorbing activators and crosslinking density in the matrix, that guarantees the cross-linking density of the matrix material in addition to the nanofiller materials [15, 16].

Due to its prodigious surface property, silica particles are widely used to boost the mechanical properties of the elastic material. The surface bound hydroxyl groups prompts a strong polarity of the material [17]. Then again, the poor compatibility with the rubber and interaction between the exceptionally polar filler (nanosilica) and non-polar (like EPDM/SBR, NR/ SBR, EPDM/NR, and so on.) rubber matrix was unacceptable for achieving wanted fortifying reinforcing effectiveness [18]. Henceforth, the surface of nanosilica should be changed by silane coupling agent (SCA) which will in general improve the increases the rubber compatibility and interaction among filler and rubber compound which finally leads in the improvement of the mechanical properties of rubber composites [19]. As the surface-topping specialist for nanosilica and as the vulcanizing agent for rubber matrix blend, bis[3-(triethoxysilyl)propyl]tetrasulphide (Si-69) can be used effectively by different SCAs. It offered the unique significance for rubber reinforcement [20, 21]. In this work, we have well focused on the modification of EPDM/SBR blend by bis[3-(triethoxysilyl)propyl]tetrasulphide (Si-69) using a melt blending process in an open mill mixer. The product was characterized by field-emission scanning electron microscopy, cure characteristics, mechanical, abrasion, compression set and swelling properties.

2 Experimental

2.1 Materials

Styrene butadiene rubber (SBR, Grade: 1502), 23.5% styrene content, Mooney viscosity 52 M, 0.93 g/cm³ density and ethylene propylene diene monomer (EPDM Grade: KEP 270), 68% ethylene content, 4.5% termonomer content (ethylidene norbornene) 4.5%, 0.86 g/cm³ density was procured from Arihant Reclamation Private Limited, New Delhi, India. Rice husk was gathered from Bahour, Puducherry, India. Silica nanoparticles were integrated in our research center by disintegration and precipitation procedure and then stored in desiccators. From Vignesh Chemicals, Ambattur, Chennai, India, the chemicals such as zinc oxide, stearic acid, bis[3-(triethoxysilyl)propyl]tetrasulphide (Si-69), mercaptobenzothiazyl disulphide (MBTS), tetramethylthiuram disulphide (TMTD) and business grade sulphur were procured for the experimental purpose.

2.2 Preparation of Nanosilica

In the hot plate, the rice husk was processed so as to expel the dampness present in it. After that, it was steadily and uninterruptedly heated up to 900 °C for seven hours in a heater and the rice husk debris was collected in a measuring glass. The ash obtained from the rice husk was used to prepare nanosilica through a different processes (disintegration procedure and precipitation process). At first, the collected rice husk ash was mainly dissolved (made of 1 mol of sodium hydroxide (NaOH) response) in the alkaline medium to break down the carbon - based materials present in it. In this way, the acquired sodium silicate arrangement (Na₂SiO₃) was washed (sifted) and desiccated in the oven for 24 h at 100 degree celcius for the dampness material present in it to be removed. At that point, silica precipitation from the sodium silicate solution (Na_2SiO_3) was treated with 6 mol of sulphuric acid (H_2SO_4) at a pH of 7 for 24 h. The silica precipitate was centrifuged and

then cleaned in warm water. At 80 °C for 24 h, the last particles present in it are dried and the silica particles are removed from the rice husk [22].

Dissolution Process

$$SiO2 + 2NaOH \rightarrow Na2SiO3 + H2O$$
(1)

Precipitation Process

$$Na2SiO3 + H2SO4 \rightarrow SiO2 + Na2SO4 + H2O$$
(2)

2.3 Composite Preparation

Table 1 tabulates the composition of the prepared EPDM/SBR rubber composites. In a two-roll open mixing mill with an operating temperature of 80 °C, compounding of the material was achieved. The EPDM was fully mixed with SBR and the necessary amounts of nanosilica, Si-69 and curatives were then applied to it. Finally, prior to sheeting off, 10 end-move passes were made. Before the vulcanization process, the aggravated rubber compound was left overnight for several periods. The vulcanization procedure was performed at the optimum cure time in a semi-robotic hydaulic press at 160 °C under a pressure of 40 MPa and a composite rubber sheet of 15 cm \times 15 cm \times 0.2 cm was made.

Table 1 Formulation for EPDM/SBR-SiO₂ composites

2.4 Characterization

The rheological properties of the rubber compounds are tested by the oscillating disc rheometer at 160° C, oscillation arc 0.5° , and 100 cycles per minute (1.66 Hz) test conditions in accordance with the ASTM D-2084 standard. The different output characteristics such as minimal torque (M_l), maximum torque (M_h), etc. are measured. Using the Eq (1) below, CRI was determined.

$$CRI(min^{-1}) = \frac{100}{t_{C90} - t_{S2}} \tag{1}$$

Where, CRI = Cure Rate Index, t_{C90} = Optimum cure time, t_{S2} = Scorch time.

The tensile properties of the composite samples were measured (using type-C die) and was tested at a temperature of 23 °C on a universal testing machine (UTM) with crosshead speed of 500 mm/min as per ASTD 412. The tear characteristics are conducted at 23 °C on un-nicked 90° angle testshaped specimens as per ASTM D 624-B standards on a UTM. The hardness was measured in Durometer with Shore A scale (ASTM D 2240). As per ASTM D 2632, the rebound resilience of the composite materials was determined.

The swelling resistance was conducted as per ASTM D471 by immersion technique. The swelling test was performed on various chemical substances such as aromatic, aliphatic and chlorinated hydrocarbons. The relative differences in the weight of the rubber compounds shall be assessed before and after immersion in the solvent. The immersed sample was to be kept at the dark room for 3 days and then to be taken away and cleaned for the removal of excess solvent over the surface, then weighed immediately in the precision weighing

Sample code	Compounds (phr)										
	EPDM	SBR	SiO ₂	Si-69	Zinc oxide	Stearic acid	MBTS	TMTD	S		
S ₀	80	20	0	_	4	1.5	1.2	1	2.5		
S_2	80	20	2	-	4	1.5	1.2	1	2.5		
S_4	80	20	4	-	4	1.5	1.2	1	2.5		
S ₆	80	20	6	-	4	1.5	1.2	1	2.5		
S ₈	80	20	8	-	4	1.5	1.2	1	2.5		
S ₁₀	80	20	10	-	4	1.5	1.2	1	2.5		
SS_0	80	20	0	3	4	1.5	1.2	1	2.5		
SS_2	80	20	2	3	4	1.5	1.2	1	2.5		
SS_4	80	20	4	3	4	1.5	1.2	1	2.5		
SS_6	80	20	6	3	4	1.5	1.2	1	2.5		
SS_8	80	20	8	3	4	1.5	1.2	1	2.5		
SS ₁₀	80	20	10	3	4	1.5	1.2	1	2.5		

balance. The mole percent absorption of Q_t for solvent was calculated by using the Eq (2) shown below.

$$Q_t(mol\%) = \frac{(M_f - M_i)/W}{M_0} \times 100$$
(2)

where, $M_f = Final$ mass of the rubber compound, $M_i = initial$ mass of the rubber compound W = Molecular weight of the solvent.

The following equation estimated the amount of crosslinking density (3) [23–25]:

$$\vartheta \left(\frac{mol}{cm^3}\right) = \frac{1}{2M_c} \tag{3}$$

where, $M_c = Molar$ mass between crosslinks.

The molar mass was determined from the Flory-Rehner Eq (4) shown below between the crosslinks of the composites [23–26]:

$$M_{c}\left(\frac{g}{mol}\right) = \frac{-\rho_{p}V_{s}V_{r}^{1/3}}{\ln(1-V_{r}) + V_{r} + \chi V_{r}^{2}}$$
(4)

where, ρ_p = Density of the material, V_s = Molar volume of the solvent, V_r = Volume fraction of polymer in the solvent-swollen filled compound, χ constant (0.3) [27], and The following equation can be used to measure Vr (5) [28]:

$$V_r = \frac{1}{1 + Q_m} \tag{5}$$

where, Q_m = Weight swell of the composites in toluene.

The abrasion resistance of the rubber samples is measured according to ASTM D 5963 in a DIN abrader. Using the following equation the abrasion loss of the rubber samples are determined (6):

Abrasion loss
$$(mm^3) = \frac{\Delta m \times S_0}{\rho \times S}$$
 (6)

where, $\Delta m = Mass loss in mg$; $\rho = density in mg/mm^3$; $S_0 = Nominal abrasive power (200 mg)$; and S = Average abrasive power in mg.

The compression set test was performed in accordance with ASTM D395. The compression set was determined using formula shown in Eq (7) below:

Percentage of compression set
$$(\%) = \frac{t_0 - t_1}{t_0 - t_s} \times 100$$
 (7)

where, $t_o = Original$ thickness, $t_1 = Final$ thickness and $t_s = Thickness$ of the spacer bar.

The FESEM was used to analysis the fractured composite specimen surfaces in Hitachi model S-4160. The fractured

samples were coated with gold layer prior to FESEM examination and the micrograph are taken at 3 kV acceleration voltage.

3 Results and Discussions

3.1 Rheometric Characteristics

The cure characteristics, such as minimum torque (M₁), maximum torque (M_h), delta torque ($\Delta M = M_H - M_L$), scorch time (t_{S2}) , optimum cure time (t_{C90}) and cure rate index (CRI) for different compounds are clearly shown in Fig. 2a-f. The cure characteristics of EPDM/SBR composites are affected by the concentration of the filler and its surface properties [29]. Figure 2a-c represents the minimum torque, maximum torque and delta torque of the nanosilica reinforced EPDM/SBR composites. With an increase in nanosilica loading, the minimum torque, maximum torque and delta torque of EPDM/SBR composites have been found to increase. This was due to the silica nanoparticles' adsorption of curatives [30]. The minimum torque value was expressed by the nanosilica dispersion and the processing characteristics of the composites of EPDM/SBR. The composite of EPDM/SBR has the maximum for minimum torque due to the increasing agglomerates of nanosilica. But, adding of silane couping agent, the minimum torque of composites decreased noticeably to a greater extent. The maximum torque of silane coupling agent composites is much higher than that of silane coupling agent-free composites, which may be due to the breakdown of agglomerates by silane coupling agent. The delta torque indicates that the cross-linking amid the reinforcing material and the matrix. The interactions at the nanosilica-EPDM/SBR interface are higher with the silane coupling agent.

Figure 2d-f shows the graph of the nanosilica-filled EPDM/SBR composites for scorch time, optimum cure time and cure rate index. The scorch time and optimum cure time can be used to estimate the vulcanization rate of the composite. As the concentration of nanosilica increases, the scorch time and optimum cure time of the composites decreases. This was due to the ethoxy-groups present in it and reacts with the surface of nanosilica. In addition, the reduction in scorch time and optimum cure time was due to the strong polar interplay of nanosilica with ZnO2 during the process of vulcanization and the forming of ZnO₂ bound nanosilica. This decreases ZnO₂ activity in order to increase the accelerator during the vulcanization process [31]. The nanosilica filled EPDM/SBR composite cure rate index increases with increased nanosilica concentration. This was due to the effects of silane and ethoxy groups that react effectively on the surface of nanosilica with the hydroxyl group. With the presence of silane coupling agent, scorch time, optimal cure time decreases and the cure rate index rises for the composites. This Fig. 2 Cure characteristics versus Nanosilica Loading with and without silane coupling agent. **a** Minimum torque Versus Nanosilica Loading, **b** Maximum torque versus Nanosilica Loading, **c** Delta torque versus Nanosilica Loading, **d** Scorch time versus Nanosilica Loading, **e** Optimum cure time versus Nanosilica Loading, **f** Cure rate index versus Nanosilica Loading



confirms that the crosslinking among nanosilica and the matrix is on the higher side in comparison.

3.2 Mechanical Properties

The compounds are cured for the optimum cure time and the procedure is carried out in the vulcanizing press, and their mechanical properties are studied. The mechanical properties are represented in Fig. 3a–d. The tensile strength of EPDM/SBR composites was shown in Fig. 3a. With an increase in the nanosilica material concentration of up to 6 phr,

the tensile strength of the EPDM/SBR composites increases and then decreases gradually. This may be due to the poor interaction of nanosilica-EPDM/SBR caused by the hydrophilic nature of silica nanoparticles that leads to higher loading conditions in the formation of aggregates. In the EPDM/SBR matrix with more agglomerates, nanosilica (without silane coupling agent) was non-uniformly distributed due to the formation of hydrogen bonds between the silanol groups and the nanosilica surface The composites containing silane coupling agent shows 18% increase in tensile strength as compared with the composites without silane coupling agent and Fig. 3 Mechanical properties of the composites versus Nanosilica Loading with and without silane coupling agent. **a** Tensile strength versus Nanosilica Loading, **b** Elongation at break versus Nanosilica Loading, **c** 100% modulus versus Nanosilica Loading, **d** Tear strength versus Nanosilica Loading



increased to 134% as compared with the neat rubber. Most of the silanol groups on the nanosilica surface react with silica nanoparticles to be compatible with the EPDM/SBR and get well distributed in it due to the improvement in the tensile strength due to the incorporation of silane coupling agent. In Fig. 3b, c, respectively, elongation at break and 100% modulus of EPDM/SBR composites were shown. Elongation at break and 100% modulus increase with an increase of up to 6 phr in nanosilica loading further reduces with an increment in the loading of EPDM/SBR composites. The increase in elongation at break suggests a good interfacial adhesion of the filler-rubber and proper wetting of the nanosilica. The reduction in the elongation at break indicates the restriction of movement in polymeric molecular chains. The elongation at break of EPDM/SBR composites with SCA is obviously decreased compared with composites without SCA, indicating the dispersion of nanosilica particles in the rubber matrix has improved. In addition, SCA and nanosilica particles are nondeformable and non-flexible and it restricts the free movement of molecular chains [32-34]. 100% modulus was influenced by concentration of the nanosilica loading and the particle size in the EPDM/SBR composites. The composites with SCA shows 5% decrement in elongation at break and 4% rise in 100% modulus compared to that of composite without SCA. The composites with SCA shows 16% and 51% increase in elongation at break and 100% modulus, respectively as related to the neat rubber. Tear strength the rubber composites was shown in Fig. 3d. Tear strength of the composite rubber steps up with an increasing concentration of nanosilica in the matrix. The tear strength of the composite with SCA was higher than that of the composite without SCA. This was due to the rise in the crosslink density of the rubber composite. In general, the nanosilica particles are well dispersed at lower loading, but in our case of tear strength especially for 4 phr of nanosilica filled composites with SCA have deviated from the position. The poor dispersion of nanosilica or SCA material possesses a larger particle size (agglomerated), which promotes flaws for crack propagation, resulting in premature failure upon uni-axial loading.

3.3 Hardness and Rebound Resilience

The hardness of the composites with respect to the loading of nanosilica has been shown in Fig. 4. The composite hardness increases with an increase in initial amount of nanosilica. The composite with SCA indicates that the increase in hardness was due to improved crosslink density and enhanced silica dispersion. Higher crosslinking density was the reason the enhanced hardness. As the composite's crosslink density increased, the softer matrix turns to a harder one. The rebound resilience of the composites was shown in Fig. 5. The rebound resilience drops with an increase in concentration of



Fig. 4 Hardness of the EPDM/SBR-SiO $_2$ composites with and without silane coupling agent

nanosilica, which also improves the rigidity and stiffness of the composites. The composite's rebound resistance was inversely proportional to the material's hardness, thus decreasing with an increase in the nanosilica content. By adding nanosilica, the decrease in rebound resistance was attributed to an improvement in the reinforcing effect [33]. Decreased tendency of rebound resilience may be attributed to improved interaction between matrix and reinforcement. EPDM/SBR composites with and without SCA rebound resistance are comparative. As the content of nanosilica increases in the matrix material, the elasticity of the rubber chains decreases, leading to poor rebound resilience [20]. The rebound resilience of the composite reduces by converting the energy consumed to heat. Generally, the resilience of the composite was



Fig. 5 Rebound resilience of the EPDM/SBR-SiO $_2$ composites with and without silane coupling agent

inversely proportional to the heat generated during the process.

3.4 Abrasion Resistance

Abrasion resistance was quite crucial when considering the operational life of the rubber material and was analyzed as volume loss during the DIN test [35]. In Fig. 6, the abrasion loss of the composites was shown. The resistance to abrasion of composites increases with the rise in nanosilica concentration. The abrasion resistance of composites with silane coupling agent was obviously greater than that of composites without SCA at the given nanosilica loading point. Improved abrasion resistance properties could be due to improved nanosilica dispersion in the EPDM/SBR matrix. As the reinforcement material, nanosilica has a stronger interfacial relationship with the EPDM/SBR matrix and the reduction of abrasion loss in the EPDM/SBR composites was clearly evident.

3.5 Compression Test

Table 2 shows the compression set values of the composites. It is evident from Table 2 that the compression set is low value in the case of neat rubber, while the compression set increases concurrently with the rise in nanosilica concentration. Even after removal of the applied load, the mobility of the rubber chain was limited as the composite module increased. As the concentration of nanosilica increases, the compression set and cross-link density of the composite increase, while the mobility of the polymer chains decreases, causing composite stiffness. As expected, the rise in crosslinking density results in a set value of high compression. For EPDM/SBR-SiO₂ composites with the presence of SCA, the crosslinking density was slightly increased compared to composites without



Fig. 6 Abrasion loss Vs Nanosilica Loading

Table 2Compression set forEPDM/SBR-SiO2 composites

Sample code	Compression set (%)								
	For 1 day at 23 °C	For 2 day at 23 °C	For 3 day at 23 °C	For 1 day at 70 °C	For 1 day at 100 °C				
S ₀	4.1	9.74	13.14	21.67	25.23				
S_2	5.75	11.45	15.76	24.32	29.32				
S_4	6.06	11.69	16.17	26.78	30.67				
S ₆	6.5	12.55	16.64	29.45	31.76				
S_8	6.67	12.93	17.06	30.05	32.54				
S_{10}	6.79	13.34	17.18	30.21	32.84				
SS_0	4.18	9.83	13.36	21.89	25.54				
SS_2	6.08	12.68	16.89	25.17	30.94				
SS_4	6.55	11.94	17.56	28.45	32.34				
SS_6	7.03	13.76	18.07	31.65	33.81				
SS_8	7.16	14.05	18.73	33.39	34.34				
SS_{10}	7.22	14.28	19.12	33.58	34.93				

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SCA. As time and temperature rises, the compression set percentage also rises, and Table 2 shows it clearly. The temperatures have a significant influence on the rubber material's compression set property. The worst retainable elastic properties of the rubber composite material are indicated by the higher compression set. The lower the compression set percentage, the more the engineering composite material is used.

3.6 Swelling Properties

The composites' swelling behavior was affected by the form of matrix, filler, filler geometry (size, shape, orientation, distribution and concentration), solvent and matrix reaction, temperature, penetrants, etc. In order to understand the interfacial interaction of the system, the analysis of the swelling characteristics of the composites can therefore be used effectively. Rubbers are mainly used in manufacturing applications and are exposed throughout their lifespan to a range of chemical environments. The effects of nanosilica loading and the existence of the penetrants were studied via EPDM/SBR composites with and without SCA on mole percent absorption. The swelling curves are represented as a graph (i.e., composite vs $t^{1/2}$ (where, t = time) mole percent uptake (Q_t)) and are shown in Figs.7, 8, 9.

Table 3 shows the swelling properties in terms of mole percent uptake at different nanosilica loading in rubber composites with and without SCA for different penetrants. The penetrants used are aromatic such as benzene, toluene, etc., aliphatic materials such as n-petane, n-hexane, etc., and chlorinated chemicals such as dichloromethane, chloroform, etc. The mole-percentage absorption of benzene by nanosilica loading is clearly shown in Fig. 7. As nanosilica concentration increases, the mole percent absorption of nanosilica filled EPDM/SBR composites with and without silane coupling agent decreases. With the rise in nanosilica loading, the improvement in mole percent absorption of both EPDM/SBR composites could be explained as a significance of the offered pressure involved between the rubber polymeric chain network and the penetrant that acts to shrink the rubber chain network. In fact, with an increase in nanosilica concentration, the crosslinking density in rubber composites increases dramatically, resulting in an increase in network elasticity. In addition, crosslinks restrict the swelling-induced extensibility of the rubber polymeric chains (i.e., macromolecular chains) and make it more difficult for solvent (penetrant) to disperse in the gaps between rubber content molecules and reduce the



Fig. 7 Mole percent uptake (mol %) Vs Time (min)



Fig. 8 Mole percent uptake of pentane, hexane, heptane and octane Vs Time (\min)

swelling percentage [36–38], thus combating the dissolution tendency [39]. As a result, the swelling decreases with network augmentation. It was evident from Table 3 that the swelling properties of EPDM/SBR-SiO₂ composites with SCA were comparatively lower than those of a composite without SCA for similar nanosilica loading. This was due to the higher crosslinking density of EPDM/SBR-SiO₂ composites with SCA compared to those composites without SCA. Thus, as seen in Table 3, a similar pattern was achieved with aromatic, aliphatic and chlorinated solvent.

The effect of penetrating size on the mole percent uptake of EPDM/SBR composites filled with pentane,



Fig. 9 Mole percent uptake Vs Time (min) with and without SCA and crosslinked with mixed systems

hexane, heptane and octane through 10 phr nanosilica is presented in Fig. 8. The higher solvent molecular weight shows the lowest mole percentage uptake, while the lower solvent molecular weight shows the higher mole percentage uptake for the rubber composite. From Table 3, it was evident that the pattern was as follows: a) aromatic: mesitylene < xylene < Toluene < Benzene b) aliphatic: n-octane < n -heptane < n-hexane < n-pentane c) chlorinated: carbon tetrachloride < chloroform < dichloromethane. Thus, both the penetrating solvent and the molecular mass are inversely proportional to each other. Finally, greater composite swelling suggests that it was not best suited to be used in industrial applications. The effect of the SCA on the mole % uptake of dichloromethane through 10 phr nanosilicafilled EPDM/SBR composites is shown in Fig. 9. Therefore, EPDM/SBR-SiO2 composites with higher nanosilica loading and SCA were practically preferable to SCA-free composites for industrial applications.

3.7 Crosslinking Density

The crosslinking density values for EPDM/SBR-SiO₂ composites are shown in Fig. 10. The crosslinking density was calculated using the swelling data. As the volume of nanosilica content increases, the composite cross-linking density increases. The larger the crosslinking density was mainly accountable for enhanced mechanical properties, durability, swelling and abrasion resistance.

3.8 Morphology

The ultimate properties of EPDM/SBR composites are determined by the dispersion of nanofillers in the rubber matrix. Uniform dispersion of nanofillers enhances the properties of the composite. The aggregated nanofillers develop a stress concentration point which results in drop down of properties in the composite materials [40]. As shown in Fig. 11a, the tensile fractured surface of EPDM/SBR-SiO₂ composites clearly indicates the undulations and little roughness with slight agglomerates of silica nanoparticles. The Fig. 11b shows the EPDM/SBR-SiO₂ composites with silane coupling agent. From Fig. 11b it was clear that irregularity over the fractured surface was observed with undulation. The homogeneous dispersion of silica nanoparticles results in effective load transfer which leads to the improvement in mechanical properties of composites with silane coupling agent.

Table 3 Mole percent uptake for different penetrant of hybrid composites material

Sample code	Mole percent uptake (mol%)										
	Aromatic				Aliphatic				Chlorinated		
	Benzene	Toluene	Xylene	Mesitylene	n- pentane	n- hexane	n- heptane	n- octane	Dichloromethane	Chloroform	Carbon tetrachloride
So	3.95	3.57	3.46	3.03	2.42	2.34	2.28	2.24	5.42	4.56	2.26
S_2	3.7	3.34	3.23	2.9	2.36	2.17	2.04	2.08	5.1	4.28	2.03
S_4	3.57	3.14	3.04	2.78	2.18	2	1.87	1.84	4.83	4	1.9
S ₆	3.4	2.9	2.76	2.65	1.96	1.84	1.76	1.72	4.6	3.83	1.81
S ₈	3.33	2.84	2.7	2.6	1.9	1.8	1.7	1.64	4.54	3.76	1.75
S ₁₀	3.3	2.78	2.66	2.55	1.84	1.72	1.66	1.6	4.5	3.72	1.71
SS_0	3.92	3.54	3.44	3	2.39	2.31	2.25	2.21	5.39	4.53	2.23
SS_2	3.61	3.27	3.15	2.82	2.29	2.09	1.97	1.91	5.04	4.13	1.96
SS_4	3.52	3.05	2.97	2.66	2.04	1.92	1.82	1.77	4.76	3.81	1.78
SS_6	3.28	2.77	2.59	2.5	1.84	1.75	1.68	1.63	4.52	3.66	1.73
SS_8	3.23	2.75	2.55	2.46	1.81	1.71	1.6	1.58	4.44	3.59	1.64
SS ₁₀	3.19	2.71	2.52	2.41	1.79	1.68	1.58	1.54	4.35	3.55	1.61

4 Conclusions

In the current study, we prepared sulphur cured EPDM/SBR composites by introducing nanosilica particles into the EPDM/SBR matrix. The comparison between composites with and without silane coupling agent on the cure characteristics, mechanical properties, hardness, rebound resilience, compression set, swelling and abrasion resistance of EPDM/SBR composites.



Fig. 10 Crosslinking density Vs Nonosilica Loading

(b) An increase in torque values was achieved for the nanosilica filled composites, that clearly indicates better crosslinks in the system.

without SCA.

were derived:

(a)

(c) The tensile characteristics such as strength, elongation at break, etc., of the EPDM/SBR composites steps up with increase in the content of nanosilica up to 6 phr and then again drops. The composites with silane coupling agent shows 134%, 16% and 51% increase in tensile strength, elongation at break and 100% modulus, respectively as compared to the neat rubber.

From the experimental findings, the following conclusions

The minimum torque and the other related components

are increased, whereas the scorch time and optimum cure

time decreases with the rise in the weight content of silica

nanoparticles in the EPDM/SBR composites with and

- (d) The addition of nanosilica in the rubber blend enhances the other properties such as tear strength, hardness, swelling and abrasion resistance but at the same time there is a drop in the rebound resilience property of the material.
- (e) Mechanical, hardness, swelling and abrasion resistance of the EPDM/SBR-SiO₂ composites with silane coupling agent showed better result than the composites without silane coupling agent. Silane coupling agent performed as a excellent compatibilizer in the rubber recipe containing EPDM/SBR matrix and nanosilica particles.

Fig. 11 SEM Micrograph of tensile fractured specimen with the Magnifications: X = 20,000 a S_6 and b SS_6



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Availability of Data and Material All data generated or analyzed during this investigation are included in this published article.

Author Contributions 1. S. Vishvanathperumal – First Author -Fabricated the samples and conducted the testing as per the ASTM standard and helped in writing the manuscript, both authors read and approved the manuscript.

2. G. Anand – Corresponding Author – Helped in Fabrication of composite samples and conducted the testing as per the ASTM standard and have written the manuscript, both authors read and approved the manuscript.

Declarations No, all testing are conducted in this study as per ASTM standards.

Conflict of Interest The authors declare no conflict of interest.

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