

## Preparation and performance of dye-sensitized solar cells based on ZnO-modified TiO<sub>2</sub> electrodes

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**Abstract:** The ZnO-modified TiO<sub>2</sub> electrode was prepared by adding Zn(CH<sub>3</sub>COO)<sub>2</sub>·2H<sub>2</sub>O to the TiO<sub>2</sub> colloid during the sol-gel production process, and was used in dye-sensitized solar cells (DSCs). The open circuit voltage ( $V_{OC}$ ) and fill factor (ff) of the cells were improved significantly. The performances of the ZnO-modified TiO<sub>2</sub> electrode such as dark current, transient photocurrent, impedance, absorption spectra, and flat band potential ( $V_{fb}$ ) were investigated. It is found that the interface charge recombination impedance increases and  $V_{fb}$  shifts about 200 mV toward the cathodic potential. The effect mechanism of ZnO modification on the performance of DSCs may be that ZnO occupies the surface states of the TiO<sub>2</sub> film.

**Keywords:** dye-sensitized solar cells; TiO<sub>2</sub> electrode; modification; photoelectrochemical properties

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### 1. Introduction

Dye-sensitized solar cells (DSCs) based on TiO<sub>2</sub> nanocrystalline electrodes have attracted the attention of scientists because of their high performance and low-cost production [1-5]. Although DSCs based on other nanocrystalline semiconductors such as ZnO and SnO<sub>2</sub> are also prepared, most of them do not have the conversion efficiency as high as the efficiency of TiO<sub>2</sub> cells [6-8]. The major parameters of DSCs are the short circuit current density ( $J_{SC}$ ), open circuit voltage ( $V_{OC}$ ), and fill factor (ff) of the photocurrent-voltage ( $I$ - $V$ ) curves. The charge recombination at the TiO<sub>2</sub>/dye/electrolyte interface has a serious effect on  $J_{SC}$ ,  $V_{OC}$ , and ff of the solar cells [9-10]. To reduce the recombination reactions, several research groups modify TiO<sub>2</sub> with metal oxides such as Nb<sub>2</sub>O<sub>5</sub>, MgO, ZnO, and so on [11-15].

Kim *et al.* [15] obtained ZnO-modified TiO<sub>2</sub> by spraying

the ethanol solution of TiO<sub>2</sub> powder and ZnCl<sub>2</sub> into liquid N<sub>2</sub>, and used the nanocrystalline electrode in the flexible dye-sensitized solar cells. The conversion efficiency ( $\eta$ ) increased from 0.71% to 1.21%. Kim *et al.* [16] prepared ZnO-coated TiO<sub>2</sub> by the thermal chemical vapor deposition method. The  $J_{SC}$  and  $V_{OC}$  of the cells were improved by 12% and 17%, respectively. Roh *et al.* [17] improved the conversion efficiency by the chemically deposited ZnO layer on TiO<sub>2</sub> electrodes.

In this paper, ZnO modified TiO<sub>2</sub> electrodes were prepared by adding Zn(CH<sub>3</sub>COO)<sub>2</sub>·2H<sub>2</sub>O to the TiO<sub>2</sub> colloid during the sol-gel production process. Some experiments, such as dark current, transient photocurrent, impedance, absorption spectra, and flat band potential, were carried out to explore the causes for the enhancement of the photoelectrochemical properties of DSCs.

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## 2. Experimental

### 2.1. Preparation of ZnO-modified TiO<sub>2</sub> electrodes

TiO<sub>2</sub> colloids were prepared by hydrolyzing the mixture of titanium (IV) isopropoxide (Aldrich) and isopropyl alcohol in the presence of distilled acetic acid (pH=2) followed by autoclaving at 250°C for 12 h. After the hydrothermal treatment, different contents of Zn(CH<sub>3</sub>COO)<sub>2</sub>·2H<sub>2</sub>O were added to the TiO<sub>2</sub> colloidal solution. The colloidal solution was condensed in a rotary evaporator. The Zn/Ti molar ratio was controlled to be 0.01, 0.025, 0.5, and 0.1, respectively. After the concentration, the TiO<sub>2</sub> colloidal paste was doctor-bladed onto the clean transparent conduction glass sheets (FTO, 30 Ω-square<sup>-1</sup>, F-doped SnO<sub>2</sub>). To control the thickness of the film, two edges of the transparent conduction glass substrate were covered with adhesive tapes. After drying in air, the samples were heated for 30 min at 450°C, resulting in anatase TiO<sub>2</sub> nanoparticle film electrodes. The prepared nanocrystalline TiO<sub>2</sub> films were conserved in a desiccator.

### 2.2. Characterization of the TiO<sub>2</sub> film

The crystalline phases of the samples were characterized by a Rigaku D/max2500 X-ray powder diffractometer (XRD) with a monochromatized Cu K<sub>α</sub> irradiation. The morphologies of the samples were determined using a Hitachi S-4300F field emission scanning electron microscope (SEM).

### 2.3. Solar cell fabrication

Nanoporous TiO<sub>2</sub> films were dipped into the 5×10<sup>-4</sup> mol ethanol solution of *cis*-bis(thiocyanato)-*N,N'*-bis(2,2'-bipyridyl-4,4'-dicarboxylate) ruthenium (II) for about 12 h. The counter electrode was Pt foil. The mixture of ethylene carbonate (EC, Acros) and propylene carbonate (PC, Acros) (EC:PC=5:5, *V/V*) containing 0.5 mol KI and 0.05 mol I<sub>2</sub> expressed as EC/PC/KI/I<sub>2</sub> was selected to be the electrolyte of DSCs. The counter electrode and the dye-sensitized TiO<sub>2</sub> electrode were clamped firmly together. The redox electrolyte solution was introduced into the porous nanocrystalline TiO<sub>2</sub> film by capillary action [18].

### 2.4. Photoelectrochemical and electrochemical measurements

The photoelectrochemical characteristics of DSCs were measured by the photocurrent-voltage curve (*I-V* curves) measurement with a potentiostat (EG&G Princeton Applied Research, Model 273) under the simulated solar light. A 250-W tungsten halogen lamp was used as the light source. The incident light intensity was 100 mW·cm<sup>-2</sup> measured by

350 linear/log optometer (UDT instruments) and the active cell area was 0.2 cm<sup>2</sup>. The dark current and transient photocurrent pattern of DSCs were measured using the same potentiostat as the *I-V* curves.

The electrochemical impedance spectra (EIS) and the flat band potential measurement were carried out with a potentiostat (Solartron SI1287) and a frequency response analyzer (Solartron 1255B). The impedance measurement of DSCs was recorded over a frequency range of 0.01-1 MHz with the AC amplitude of 10 mV. The measurement was performed at the open circuit voltage under the same light intensity as the *I-V* curve measurement. The flat band potential was measured by the Mott-Schottky analysis method with the potential range from 0 to -1.0 V [19-20].

## 3. Results and discussion

Fig. 1 illustrates the XRD patterns of the bare TiO<sub>2</sub> and 5at% ZnO-modified TiO<sub>2</sub> thin films. The bare TiO<sub>2</sub> and 5at% ZnO-modified TiO<sub>2</sub> thin films both consist of anatase titania crystals. The diffraction peaks of the 5at% ZnO-modified TiO<sub>2</sub> thin film are almost the same as the pure TiO<sub>2</sub> film except for a little decrease in the peak intensity. There are no diffraction peaks of ZnO structure, which means that the addition of a small amount of ZnO does not change the structure of anatase-TiO<sub>2</sub>.

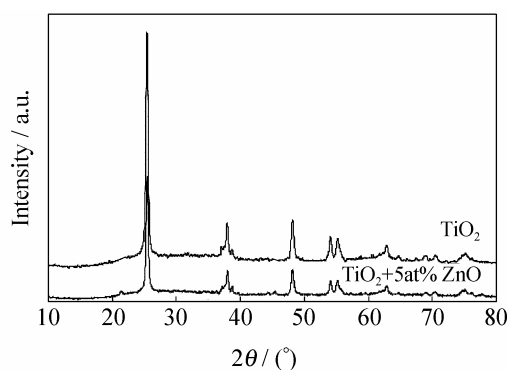


Fig. 1. XRD patterns of bare TiO<sub>2</sub> and 5at% ZnO-modified TiO<sub>2</sub> thin films.

Fig. 2 shows the SEM micrographs of pure TiO<sub>2</sub> and 5at% ZnO-modified TiO<sub>2</sub> thin films. After the modification of 5at% ZnO, the morphology of the TiO<sub>2</sub> thin film has no significant change.

The electron dispersive X-ray (EDX) spectrum of the ZnO-modified TiO<sub>2</sub> film is shown in Fig. 3. The Zn element is found in the sample, but the quantity of Zn is small which is consistent with the conclusion of the XRD patterns (Fig. 1) and the SEM micrographs (Fig. 2).

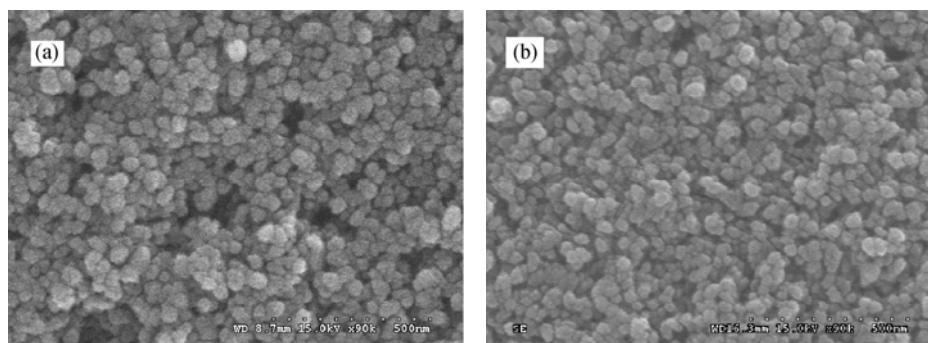


Fig. 2. SEM micrographs of bare TiO<sub>2</sub> (a) and 5at% ZnO-modified TiO<sub>2</sub> thin films (b).

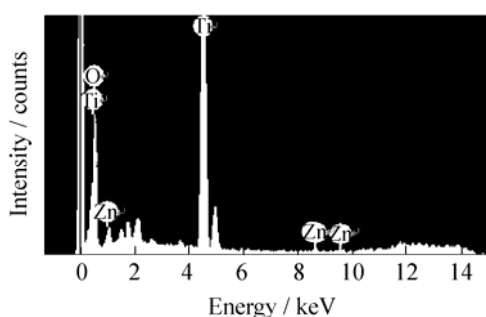


Fig. 3. EDX spectrum of the 5at% ZnO-modified TiO<sub>2</sub> thin films.

The photoelectric data of the dye-sensitized solar cells using ZnO-modified TiO<sub>2</sub> nanoporous electrodes are summarized in Table 1. The open circuit voltage and fill factor keep increasing with the increase of the Zn content. However, the short circuit current decreases with the increase in Zn/Ti (at%). The conversion efficiency can be enhanced about 0.4% when the Zn content is 5at%. The short circuit current decreases sharply when the Zn content is higher than 5at%, and the conversion efficiency begins to decline.

Table 1. Photoelectrochemical parameters of DSCs using ZnO-modified TiO<sub>2</sub> nanoporous electrodes

Zn:Ti / at%	$J_{sc} / (\text{mA}\cdot\text{cm}^{-2})$	$V_{oc} / \text{mV}$	ff	$\eta / \%$
0.000	17.0	539	0.55	4.98
0.010	16.2	559	0.56	5.08
0.025	15.7	572	0.59	5.28
0.050	13.0	641	0.65	5.39
0.100	9.3	664	0.67	4.11

The restriction of charge recombination at the TiO<sub>2</sub>/dye/electrolyte interface is one important factor for the improvement of  $V_{oc}$  and ff of DSCs [9, 12]. The dark current arises from the reduction of triiodide and oxidized dye molecule by conduction band electrons of TiO<sub>2</sub> films. The dark current measurement is an apparent analysis on the interface charge recombination. Fig. 4 addresses the dark cur-

rent for the dye-sensitized solar cells using pure TiO<sub>2</sub> and ZnO-modified TiO<sub>2</sub> nanoporous electrodes. The dark current is significantly reduced by the modification of ZnO. This observation of the reduction of charge recombination is consistent with the reports of Kim *et al.* [16].

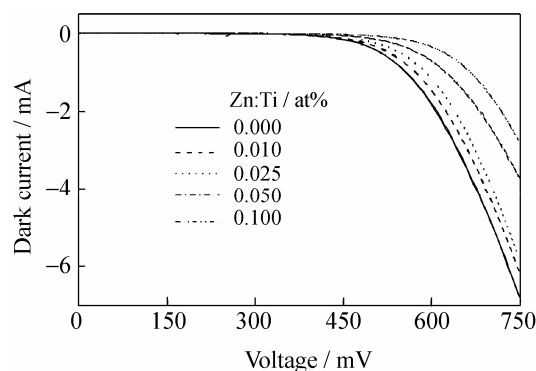


Fig. 4. Dark current characteristics of DSCs using pure TiO<sub>2</sub> and ZnO-modified TiO<sub>2</sub> nanoporous electrodes.

Fig. 5 shows the electrochemical impedance spectra (EIS) of pure TiO<sub>2</sub> and 5at% ZnO-modified TiO<sub>2</sub> electrodes, where the  $\text{Re}Z$  and  $\text{Im}Z$  correspond to the real and imaginary part of the electrochemical impedance, respectively. The illumination condition of the impedance measurement was the same as that of the  $I$ - $V$  measurement. Three arcs appear in the impedance spectra of the dye-sensitized solar cell using pure TiO<sub>2</sub> which are expressed as their characteristic frequencies  $\omega_1$ ,  $\omega_2$  and  $\omega_3$ . The arcs appear around 2000, 100, and 0.5 Hz, respectively. The three semicircles are separately attributed to the redox reaction at the platinum counter electrode ( $\omega_1$ ), the electron transfer at the TiO<sub>2</sub>/dye/electrolyte interface ( $\omega_2$ ), and ion transport within the electrolyte ( $\omega_3$ ) [21-22]. However, only one arc is observed in the impedance spectra of the dye-sensitized solar cell using the 5at% ZnO-modified TiO<sub>2</sub> electrode. This arc appears around 50 Hz which should be caused by the recombination reactions at the TiO<sub>2</sub>/dye/electrolyte interface

( $\omega_2$ ). The other two arcs ( $\omega_1$  and  $\omega_3$ ) may be hidden by the large arc ( $\omega_2$ ). It is noticeable that the recombination reaction impedance ( $\omega_2$ ) has been increased by the modification of 5at% ZnO. It can also be found that the characteristic frequency of ( $\omega_2$ ) of the DSCs using the 5at% ZnO-modified TiO<sub>2</sub> electrode becomes smaller than that of the DSCs using the pure TiO<sub>2</sub> electrode. The inverse of the characteristic frequency of  $\omega_2$  has been expressed to be the recombination lifetime of the electron in the conduction band of TiO<sub>2</sub> [23-24]. Therefore, the decrease in the charac-

teristic frequency of  $\omega_2$  means that the injected electrons in the TiO<sub>2</sub> conduction band can stay steadily for a longer time in the ZnO-modified TiO<sub>2</sub> electrode. Both the larger impedance and the smaller characteristic frequency of the arc  $\omega_2$  should be favorable for the high performance of DSCs. It can be seen that ZnO modification can passivate the surface of the TiO<sub>2</sub> film. The normalized transient photocurrent pattern and absorption spectra of bare TiO<sub>2</sub> and ZnO-modified TiO<sub>2</sub> films were carried out to explore the causes for the changes of photoelectric and impedance performance.

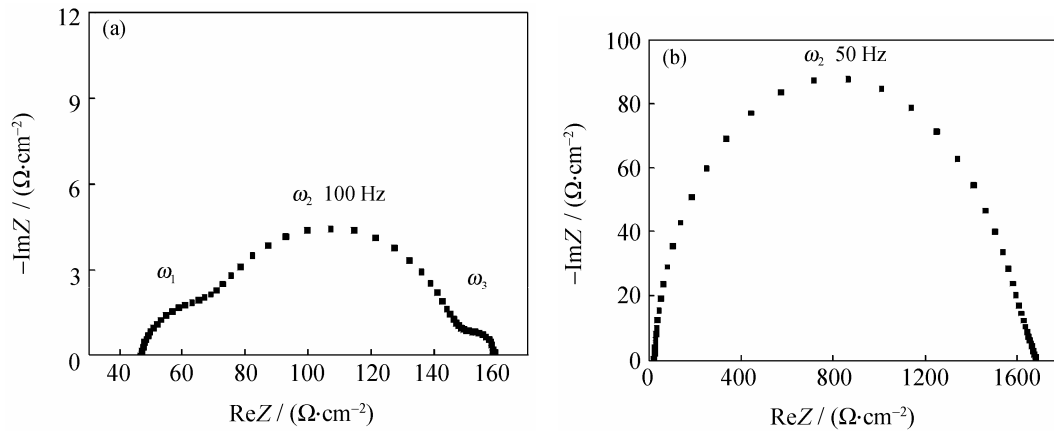


Fig. 5. EIS of DSCs using the bare TiO<sub>2</sub> (a) and 5at% ZnO-modified TiO<sub>2</sub> thin films (b).

Fig. 6 shows the normalized transient photocurrent patterns of DSCs using pure TiO<sub>2</sub> and ZnO-modified TiO<sub>2</sub> nanoporous electrodes. The photocurrent response of the ZnO-modified TiO<sub>2</sub> electrode becomes more rapid than the pure TiO<sub>2</sub> electrode. When the light is switched off, the photocurrent of the ZnO-modified TiO<sub>2</sub> electrode falls faster than the pure TiO<sub>2</sub> electrode. The cause may be that ZnO occupies the surface states of the TiO<sub>2</sub> electrode. The bondage of the electron in the TiO<sub>2</sub> conduction band is reduced and the recombination reaction between the electron and triiodide ion (or oxidized dye molecular) is decreased.

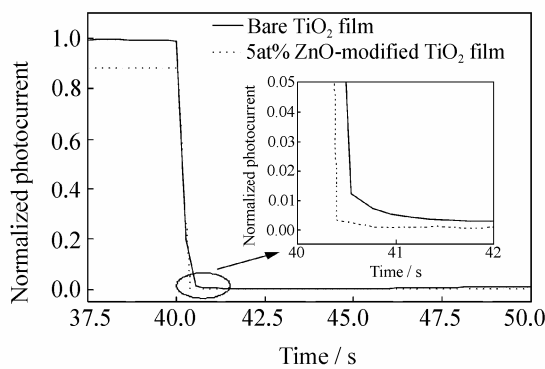


Fig. 6. Normalized transient photocurrent patterns of DSCs using the bare TiO<sub>2</sub> and 5at% ZnO-modified TiO<sub>2</sub> thin films.

ZnO-modified TiO<sub>2</sub> films. After ZnO modification, the absorption band edge of TiO<sub>2</sub> is blue-shifted which means that the band gap energy of the ZnO-modified TiO<sub>2</sub> film is improved. However, the band gap energy of ZnO is 3.2 eV which is almost the same as that of anatase TiO<sub>2</sub> [25]. The change of the absorption band edge can be that ZnO occupies the surface states of TiO<sub>2</sub> film.

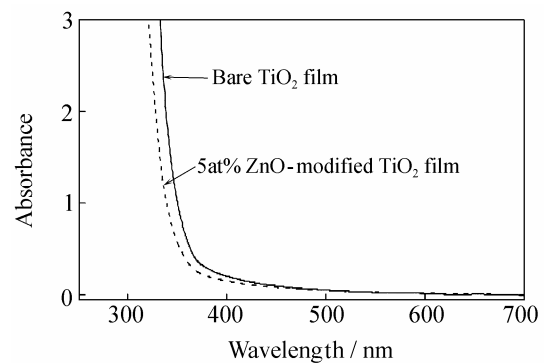
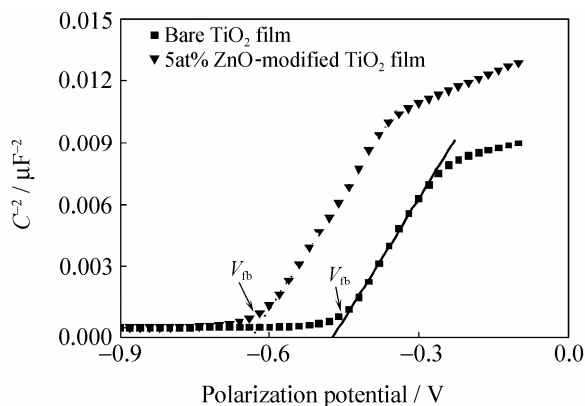


Fig. 7. Absorption spectra of bare TiO<sub>2</sub> and 5at% ZnO-modified TiO<sub>2</sub> thin films.

Fig. 8 shows the Mott-Schottky curves of pure TiO<sub>2</sub> and ZnO-modified TiO<sub>2</sub> films, where  $C$  is the space-charge capacitance,  $V_{fb}$  is the flat band potential. The plot of  $C^{-2}$  vs. polarization potential shows an X-intercept corresponding to  $V_{fb}$ . The  $V_{fb}$  shifts about 200 mV toward the cathodic poten-

Fig. 7 shows the absorption spectra of bare TiO<sub>2</sub> and

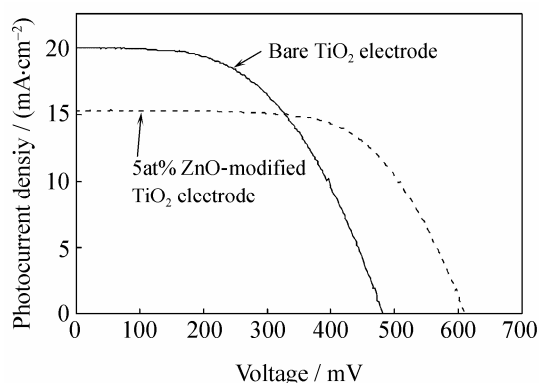
tial after the ZnO modification of the TiO<sub>2</sub> film which is also one important cause for the improvement of  $V_{OC}$  of the DSCs [26]. However, when  $V_{fb}$  shifts negatively, the driving force decreases for the charge transfer from the excited dye molecular to the conduction band, resulting in the decline of the photocurrent of DSCs.



**Fig. 8.** Mott-Schottky plots of bare TiO<sub>2</sub> and 5at% ZnO-modified TiO<sub>2</sub> thin films.

To avoid the decline of the photocurrent,  $V_{fb}$  of the TiO<sub>2</sub> film should be shifted positively. Li<sup>+</sup> is a strong electropositive cation. After the addition of LiI in the electrolyte, Li<sup>+</sup> will be absorbed on the surface of the TiO<sub>2</sub> electrode and its radius is so small that Li<sup>+</sup> will penetrate into TiO<sub>2</sub> lattice [27-28]. As a result, the energy level of the conduction band of the TiO<sub>2</sub> electrode is enhanced. In this experiment, 0.05 mol LiI was added into the EC/PC/KI/I<sub>2</sub> electrolyte.

Fig. 9 and Table 2 show the photoelectric performance of the pure TiO<sub>2</sub> and ZnO-modified TiO<sub>2</sub> electrodes using the electrolyte (EC:PC=1:1,  $V/V$ )+0.5 mol/L KI+0.05 mol/L I<sub>2</sub>+0.05 mol/L LiI.



**Fig. 9.** Photocurrent-voltage curves of DSCs using the bare TiO<sub>2</sub> and 5at% ZnO-modified TiO<sub>2</sub> electrodes in the EC/PC/KI/I<sub>2</sub> electrolyte with 0.05 mol LiI.

The photocurrent is increased significantly for all the

DSCs after the addition of LiI. For the pure TiO<sub>2</sub> electrode, the  $V_{OC}$  and ff are decreased seriously, and the conversion efficiency has almost no change after the addition of LiI in the electrolyte. However, for the ZnO-modified TiO<sub>2</sub> electrodes, the conversion efficiency is improved from 5.39% to 5.89%.

**Table 2.** Photoelectrochemical parameters of DSCs using the ZnO-modified TiO<sub>2</sub> nanoporous electrode in the EC/PC/KI/I<sub>2</sub> electrolyte with 0.05 mol LiI

Zn:Ti / at%	$J_{SC}$ / (mA·cm <sup>-2</sup> )	$V_{OC}$ / mV	ff	$\eta$ / %
0.00	20.0	484	0.52	4.97
0.05	15.3	611	0.63	5.89

#### 4. Conclusion

The ZnO-modified TiO<sub>2</sub> electrode was prepared by adding Zn(CH<sub>3</sub>COO)<sub>2</sub>·2H<sub>2</sub>O to the TiO<sub>2</sub> colloid during the sol-gel production process. The crystalline phases and morphologies of TiO<sub>2</sub> electrodes have almost no change after ZnO modification. However, the  $V_{OC}$  and ff of DSCs are improved significantly. The conversion efficiency can be enhanced from 4.98% to 5.39% when the Zn content is 5at%. After ZnO modification, the dark current of DSCs declines and the interface charge recombination impedance increases significantly. On the other hand,  $V_{fb}$  shifts about 200 mV toward the cathodic potential which is also one important cause for the improvement of  $V_{OC}$ . The effect mechanism of ZnO modification on the performance of DSCs may be that ZnO occupies the surface states of the TiO<sub>2</sub> film. In the electrolyte (EC:PC=1:1,  $V/V$ )+0.5 mol/L KI+0.05 mol/L I<sub>2</sub>+0.05 mol/L LiI, the conversion efficiency of the cell using ZnO-modified TiO<sub>2</sub> is 5.89%, which is improved by about 20%.

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