LETTER

Hydropathy modulation on $Bi₂S₃$ for enhanced electrocatalytic $CO₂$ reduction

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Electrochemical CO_2 reduction reaction (CO_2RR) is a promising way to achieve carbon neutrality. However, the activity and selectivity of $CO₂RR$ are limited by not only the development of earth-abundant catalysts but also the $CO₂$ mass transfer during the $CO₂RR$ process. Herein, $Bi₂S₃$ nanorods were synthesized under a relatively mild route. Furthermore, benefitting to the modulation of the hydropathy, the optimized sample (BS-P1) achieved a Faradaic efficiency of $HCOO^{-}$ ($> 90\%$) in the range from -0.9 to -1.2 V, a high current density of HCOO⁻ $(2.29$ times larger than that of BS-P0 at -1.2 V) and a prolonged stability from 12 to 20 h at -1.1 V. When the temperature decreased from 25 to 0° C and eventually to $-$ 20 °C, the reaction kinetics of CO₂RR was slowed down, the distribution of products was changed and hydrogen evolution reaction (HER) was inhibited. This work provides a facile synthesis for $Bi₂S₃$ and highlights the importance of triple-phase interfaces in $CO₂RR$.

To solve the issue of excessive emission of $CO₂$, many strategies have been proposed $[1-5]$. Electrochemical $CO₂$

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Herein, we synthesized $Bi₂S₃$ nanorods under relatively mild conditions and tuned their hydropathy by adding polytetrafluoroethylene (PTFE) nanoparticles into catalyst ink. The optimized sample (BS-P1) achieved a FE_{HCOO-} of 93.4% at -1.1 V, and the J_{HCOO} was 2.29 times larger than that of BS-P0 (without modification) at -1.2 V.

products [\[30](#page-6-0)].

Also, the stability of Bi_2S_3 was extended from 12 to 20 h by tuning hydropathy. In flow cells, $FE_{HCOO-} > 90\%$ at from -0.5 to -0.75 V with increased current density. Hydrophobicity of interfaces could accelerate $CO₂$ mass transfer, thus enhancing $CO₂RR$ performance. The influence of temperature for $CO₂RR$ was also investigated.

Universally, the synthesis conditions of $Bi₂S₃$ were generally complex (e.g., strong acids and high temperatures, etc.) (Table S1). Then we prepared $Bi₂S₃$ under mild conditions. The $Bi₂S₃$ nanorods were synthesized from BiCl₃ and TAA in ethanol at 150 °C (Figs. 1a and S1). First, we explored the ratio of $BiCl₃$ to TAA, called as BS-5/2, BS-3/1, BS-10/3 (BS-P0) and BS-11/3, respectively. X-ray diffraction (XRD) was used to investigate the crystal structure for the catalysts. The main peaks of BS located at 24.9 \degree and 28.6 \degree were consistent with (130) and (211) planes in standard card of Bi_2S_3 (PDF No. 17-0320), respectively (Fig. 1b). Then PTFE nanoparticles were added in catalyst ink (see the detail in SI). According to the amounts of PTFE added, the catalysts were called as BS-Px ($x = 0, 0.5, 1, 2$). Peak at 18.1[°] was a characteristic peak of PTFE (PDF No. 47-2217) [\[31](#page-6-0)]. As the amount of added PTFE increased, characteristic peak of F was enhanced. Then the Bi 4f spectra of BS-P0 and BS-P1 are shown in Fig. 1c, and two peaks at 158.2 and 163.6 eV corresponded to Bi $4f_{7/2}$ and Bi $4f_{5/2}$, respectively. Also, peaks at 160.9 and 162.0 eV were related to Bi–S vibration [\[32](#page-6-0)]. Peaks at 158.6 and 164.3 eV would correspond to Bi–O vibration, which was mainly due to oxidation in air [\[33](#page-6-0)]. After adding PTFE, the peaks of FTFE were be detected, but the peaks intensity of Bi 4f spectrum become weaker (Fig. S2). X-ray photoelectron spectroscopy (XPS) is a surface detection

Fig. 1 a Schematic illustration for synthesis of Bi₂S₃; **b** XRD patterns; **c** Bi 4f XPS result; **d** FTIR of BS-P0; **e** Raman spectrum of BS-
P0 (inset: Raman spectrum in range of 100–500 cm^{−1}); f EPR of BS-P0; **g** elemental mappings images of BS-P1

Fig. 2 a FE_{HCOO}. of BS with different amounts of TAA; b J_{HCOO} and c FE_{HCOO}. of different samples; d comparison of FE_{HCOO}. with different PTFE particle sizes; e stability at -1.0 V of BS-P1 and BS-P0 in H-type cells

technique. The addition of PTFE reduced the amount of $Bi₂S₃$ detectable on the electrode surface. XPS results showed that PTFE was successfully loaded on the surface of Bi_2S_3 nanorods.

In the Fourier transform infrared (FTIR) spectra pre-sented in Fig. [1](#page-1-0)d, the bands at 617 and 1114 cm^{-1} could be assigned to Bi–S vibration. The band at 1642 cm^{-1} was assigned to O–H stretching vibrations due to the adsorption of water on surface of BS-P0 [[34\]](#page-6-0). The Raman spectrum of BS-P0 is provided in Fig. [1e](#page-1-0), and Raman bands located at 230, 246, 300, 424, 608 and 962 cm^{-1} could be observed, matching with the Raman feature of $Bi₂S₃$ [[35–37](#page-6-0)]. No carbon species were found in BS-P0. The isotropic electron paramagnetic resonance (EPR) signals at $g = 2.003$ in Fig. [1](#page-1-0)f were observed. They could be ascribed to the formation of sulfur vacancies, which may be good for $CO₂RR$ [\[7](#page-5-0)].

The crystal structure was revealed by high-resolution transmission electron microscope (HRTEM) [\[38\]](#page-6-0). In Fig. [1](#page-1-0)g, the crystalline phase with interatomic distance of 0.360 nm can be related to $Bi₂S₃$ (130) plane. The result of HRTEM was consistent with that of XRD patterns. Energydispersive X-ray spectroscopy (EDS) elemental mappings revealed that the width of the Bi₂S₃ nanorod was ~ 200 nm. The morphology of $Bi₂S₃$ did not change with the

addition of PTFE. Also, EDS elemental mappings confirm a relatively uniform distribution of PTFE (Figs. [1h](#page-1-0) and S3).

The electrochemical performance of $CO₂RR$ was then investigated in H-type cells with $0.5 \text{ mol} \cdot \text{L}^{-1} \text{ KHCO}_3$ [\[39](#page-6-0)]. The final products (included gas products and liquid products) were measured by gas chromatography and nuclear magnetic resonance (NMR) spectroscopy (Fig. S4) [\[40](#page-6-0)]. Considering that the optimal ratio of TAA and $BiCl₃$ was 10/3 (Fig. 2a), the BS-P0 (10/3) was selected as the optimized sample. The current density in $CO₂$ was larger than that in Ar, indicating the electrocatalytic activity toward $CO₂RR$ (Fig. S5) [\[41](#page-6-0)]. To compare their selectivity in $CO₂RR$, the constant voltage electrolysis methods were used (Fig. $S6$). The final products were only $HCOO^-$, CO and H_2 . Sum of FE was close to 100%, and no other products were detected. With adding PTFE nanoparticles (200 nm), J_{HCOO} increased. Excessive PTFE led to the reduction of J_{HCOO} . The maximum J_{HCOO} of BS-P1 could reach -41.08 mA \cdot cm⁻² at -1.2 V, but J_{HCOO} of BS-P0 only was -17.45 mA \cdot cm⁻² (Fig. 2b). BS-P1 has the largest J_{HCOO_2} in the range from -0.9 to -1.2 V with FE_{HCOO} above 90%. Then FE_{HCOO} of BS-P1 could reach 93.4% at -1.1 V. In comparison, for BS-P0, the maximum FE $_{\text{HCOO}}$ was 84.64% at -1.0 V (Fig. 2c). When the size of PTFE nanoparticles changed from 200 to 100 nm,

Fig. 3 a LSV curves of BS-P1 under different atmospheres and b FEs under different voltages of BS-P1 in standard three-electrode flow cells

Fig. 4 a ECSA measurement; b EIS results; c contact angle measurements on BS electrode with different PTFE loadings before and after $CO₂$ electrolysis at -1.0 V in H-type cells

 FE_{HCOO} reduced (Figs. [2](#page-2-0)d and S7). Then FE_{HCOO} of BS-P1 still remained 88.5% at -1.0 V after 20 h, while the FE_{HCOO}- of BS-P0 was only 70.5% at -1.0 V after 12 h (Fig. [2](#page-2-0)e). BS-P1 had a relatively good $CO₂RR$ performance in the H-type cells (Fig. S8 and Table S2) [[42\]](#page-6-0).

Then we assembled the standard three-electrode flow cells with 1.0 mol \cdot L⁻¹ KOH as electrolyte [[43\]](#page-6-0). A higher current density (200 mA·cm⁻² at $-$ 0.69 V) was obtained than that in H-type cells (Fig. $3a$). FE $_{HCOO}$ was larger than 93% in the potential range from $-$ 0.50 to $-$ 0.75 V. The

Fig. 5 a LSV curves (inset: optical photograph of low-temperature control device) and b FE of BS-P1 at different temperatures in twoelectrode flow cells

maximum FE $_{\text{HCOO}}$ reached 95.[3](#page-3-0)8% at - 0.6 V (Figs. 3b and S9).

As shown in Figs. [4](#page-3-0)a and S10, adding 0.5% PTFE nanoparticles had a little effect on the electrochemical active surface areas (ECSA), which was evaluated by the electric double-layer capacitance (C_{d1}) method [\[44](#page-6-0)]. Compared to BS-P0 (581 μ F·cm⁻²), ECSAs of BS-P1 and BS-P2 were significantly reduced to 526 and 503 μ F·cm⁻², respectively, because of hydropathy modulation of triplephase interfaces by adding PTFE, which could retain gas and reduce contact between electrolyte and electrode surface [\[45](#page-6-0)]. On the other hand, the ECSA of BS-P2 reduced nearly 13.4% of BS-P0. It meant that less electrolyte contacted with the electrode surface, which was the reason of the reduced $CO₂$ activity compared with BS-P1 (Fig. S5).

To investigate the intrinsic impedance of the catalyst, we tested the electrochemical impedance spectroscopy (EIS) without $CO₂$. EIS results showed that adding PTFE increased charge transfer resistance due to the poor conductivity of PTFE [[46\]](#page-6-0). However, PTFE could change the microenvironment of the triple-phase interfaces on the electrode, thus influencing the $J_{\text{HCOO-}}$ (Figs. [4](#page-3-0)b, [2](#page-2-0)b). A lower Tafel slope of 159.82 mV \cdot dec⁻¹ on BS-P1 than that on BS-P0 $(209.69 \text{ mV-dec}^{-1})$ suggested the accelerated reaction kinetics (Fig. S11) [\[47](#page-6-0)].

In Fig. [4c](#page-3-0), the initial contact angles of BS-P0 (140.79°), BS-P0.5 (144.74 $^{\circ}$), BS-P1 (144.47 $^{\circ}$) and BS-P2 (143.82 $^{\circ}$) were similar. When BS with different PTFE amounts were electrolyzed at -1.0 V for 1 h, the contact angle of BS-P0 decreased from 140.79° to 91.05° and the contact angle with 1% PTFE decreased from 144.47° to 130.36° . And with the increase in PTFE amounts, the contact angle after electrolysis also showed an increasing trend, which was larger than that of BS-P0. It meant that moderate hydrophobicity of triple-phase interfaces increased the $CO₂RR$ performance. The hydrophobicity of interfaces accelerated $CO₂$ mass transfer and reduced the availability of electrolyte [[48\]](#page-6-0). When 2% PTFE was added, the current density was dropped because of the lost balance between gas and liquid phases (Fig. S5).

Considering future extraterrestrial planet exploration, such as Mars, the performance of $CO₂RR$ at low temperature is worth studying. However, the research of $CO₂RR$ at low temperature was rare [\[49](#page-6-0)]. Here, as temperature decreased (25 to $-$ 20 °C), the current density decreased (Fig. 5a). Compared with BS-P1 at 25° C, the onset potential of BS-P1 at $-$ 20 °C shifted to a more positive value, implying a slower $CO₂RR$ kinetics [\[50](#page-6-0)]. In Fig. 5b, the peak FE_{C1} of BS-P1 reached 99.42% at 2.8 V, while the competing hydrogen evolution reaction (HER) on BS-P1 was substantially suppressed. The inhibition of HER may be due to the inhibition of the Volmer reaction of HER under alkaline conditions $[51]$ $[51]$. FE_{CO} of BS-P1 was increased with temperature decreasing (Figs. S12, S13).

In summary, $Bi₂S₃$ nanorods were synthesized under a mild condition. Then we added PTFE nanoparticles with different sizes (200 and 100 nm) into the catalyst ink to regulate the hydropathy of electrode surface. The J_{HCOO} . and stability of $Bi₂S₃$ were improved more than two folds. Also, a high FE_{HCOO} was achieved (93.4% at -1.2 V). Thus, this work not only provides a relatively mild route to synthesize $Bi₂S₃$, but also further improves the current density and stability of $Bi₂S₃$ through hydropathy modulation. By adding PTFE nanoparticles, the transition from hydrophobic to hydrophilic of electrode surface could be effectively slowed down during $CO₂RR$, which was the reason of enhanced $CO₂RR$ performance. In addition, the

influence of temperature in the $CO₂RR$ was studied, which could change the distribution of products. However, research on low-temperature $CO₂RR$ is rare. The mechanism of inhibition of HER at low temperature is explored underway.

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Declarations

Conflict of interests The authors declare that they have no conflict of interest.

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