RESEARCH ARTICLE



Structural, optical and electrical properties of In_2O_3/α -Fe₂O₃ heterostructure thin films for photocatalytic application

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Abstract In this study, In_2O_3/α -Fe₂O₃ heterostructure thin film was prepared on glass substrate using a low-cost and simple spray pyrolysis technique. After successful deposition, the structure and morphology were investigated by X-ray diffraction, scanning electron microscopy, EDX and Raman microscopy. The X-ray diffraction studies confirm the rhombohedral phase of α -Fe₂O₃ and the cubic phase of In₂O₃ without other impurity peaks, this result also was proved by Raman spectroscopy. The surface morphology of In_2O_3/α -Fe₂O₃ thin film was determined by using scanning electronic microscopy (SEM), which indicates the formation of a very homogenous surface with a spherical grain randomly disturbed with diverse sizes. Hall effect measurements showed that the conduction of In_2O_3/α -Fe₂O₃ was n type, and the most electrons are free charge carriers. The resistivity, hall mobility and charge carriers concentration are equal to $5.7 \times 10^{-4}~\Omega$ cm, 5.82 cm²/V S and 1.87×10^{21} cm⁻³, respectively. On the other hand, the grain size was equal to 28.2 nm for In₂O₃ and 30.61 for In_2O_3/α -Fe₂O₃ thin films. The prepared In_2O_3/α -Fe₂O₃ thin film was used as a photocatalyst against methylene blue, and this layer demonstrates significantly higher photocatalytic efficiency with degradation rate compared to In₂O₃ thin layer which may be related to the surface roughness and crystallite size measured by XRD and SEM.

Keywords In_2O_3/α -Fe $_2O_3$ thin film \cdot Spray pyrolysis \cdot Structural study \cdot EDX \cdot Raman microscopy \cdot Hall effect \cdot Photocatalytic activity

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Introduction

The utilization of solar energy has gained attention recently since it is a sizable, sustainable and clean energy source. The effective exploitation of this energy will offer enough power to fulfill the rising global energy demand. Additionally, solar energy can be used for organic synthesis and environmental cleanup. Of all the options for converting solar energy, photoelectrochemical water decomposition is the most convincing strategy for producing clean and renewable fuels. Due to their stability, natural abundance and suitable band gap that makes use of 40% of electromagnetic radiation, In_2O_3 and Fe_2O_3 have been viewed as an appealing alternative among the many semiconducting materials for photocatalytic application [1, 2].

First of all, In₂O₃ may be made into an n-type semiconductor with a high electrical conductivity. It is a transparent semiconductor with a moderate band gap of 2.8 eV. It also crystallizes in a cubic bixbyite structure and has a transparent nature. Due to its low cost, stable physicochemical and visible light responsive properties, this material has attracted a lot of attention in the photocatalytic destruction of organic contaminants and photo-splitting of water. It is well known that In_2O_3 , either on its own or in combination with other oxides of transition metals, became the most widely used transparent conducting oxide (TCO) in photocatalytic and related applications. Previous research suggests that the strong visible light harvesting capacity and the efficient spatial separation of photogenerated electron hole pairs in In_2O_3 , when combined with other semiconductors, improve the photocatalytic efficiency [3].

Iron oxide Fe_2O_3 , on the other hand, is a conventional form of sensing and has been studied because of its high sensitivity, rapid response, as well as durability. It is also a typical n-type semiconductor with a band gap of 2.1 eV.

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However, only a small number of studies have taken into account the development of a heterojunction system based on In_2O_3 and Fe_2O_3 for photocatalytic application [4, 5].

In order to improve the performance of acetone sensing, Zhang et al. have recently revealed simple techniques to create Fe_2O_3 -In₂O₃ heterostructure nanocomposites. The resulting samples showed good qualities such as a strong response and recovery capacities.

Additionally, the investigation of the Hall effect enables the determination of the concentration and mobility of charge carriers as well as, in some cases, the intrinsic energy gap and crucial data on the electrical characteristics of semiconductors [6].

In this work, we prepared In_2O_3 thin films on glass substrate using the spray pyrolysis technique. Fe_2O_3 was subsequently grown on the In_2O_3 to form heterojunction structure. The phase structure, morphologies, Hall effect and photocatalytic performance of the resultant samples were carried out.

Experimental

Film preparation

In_2O_3 thin film preparation

 In_2O_3 thin film was prepared by spraying an aqueous solution of indium chloride (InCl₃). The concentration of indium chloride is 0.01 M. This solution was sprayed under N₂ gas pressure of 0.5 bar on glass substrates, which were placed on hot plate heated at 350 °C.

In_2O_3/α -Fe₂O₃ thin film preparation

After preparing In_2O_3 thin films, α -Fe₂O₃ thin film was prepared from aqueous solution containing Iron(III) chloride dehydrate (FeCl₃·2H₂O) with concentration equal to 10^{-2} M as a precursor using spray pyrolysis technique on the In_2O_3 substrates at a temperature equal to 350 °C.

Characterization techniques

The spectra of X-ray diffraction of obtained thin films were examined using Cu K_{α} radiation ($\lambda = 0.15418$ nm) and a copper-source diffractometer (Analytical X Pert PROMP D), with 2 θ varying from 0° to 70°. The micro-Raman system from Jobin Yvon Horibra LABRAM-HR, which is visible between 200 and 1200 cm⁻¹, was used to record Raman scattering studies at room temperature. In addition, A JEOL-JSM 5400 model scanning electron microscope (SEM) was used to examine the morphology of the films. For elemental analysis with an electron gun, an energy-dispersive X-ray (EDX) spectrometer coupled to a thermal field emission scanning electron microscope was employed. A Hall measuring system using the van der Pauw method was used to conduct electrical tests, resistivity, Hall mobility and carrier concentration at room temperature with a mild magnetic field of approximately 0.554 T.

Finally, the breakdown rate of methylene blue (MB) aqueous solution under solar illumination was used to determine the photocatalytic activity of the structures based on In_2O_3 and In_2O_3/Fe_2O_3 thin films. MB was initially present in a concentration of 4.5 mg/L. For one hour in the dark, the MB solution was agitated at a magnetic stirrer to achieve equilibrium adsorption. In_2O_3 and In_2O_3/Fe_2O_3 thin films served as photocatalysts in the aqueous solution of MB, which was continuously stirred while exposed to sunlight. Analytical samples were taken from the solution every 1, 2, 3, 4, 5 and 6 h to determine the impact of sun light irradiation on MB solution, and a UV–Vis spectrophotometer was used to determine the amount of MB present in each sample.

The efficiency of MB degradation was calculated based on the following equation:

Degradation efficiency(%) =
$$\left(\frac{A_0 - A}{A_0} \times 100\right) = \left(\frac{C_0 - C}{C_0} \times 100\right)$$
(1)

where A_0 , A, C_0 and C are the absorbance and concentration of MB corresponding to the initial time and the variable time, respectively.

Results and discussion

XRD analysis

Figure 1 shows the XRD patterns obtained for the pure In_2O_3 , pure α -Fe₂O₃ and In_2O_3/α -Fe₂O₃ heterostucture thin film. The primary diffraction peaks for the In_2O_3 thin films were consistent with the conventional cubic structure represented by JCPDS card number 06-0416. These peaks demonstrate the synthesis of the pure phase In_2O_3 [7, 8]. In addition, the XRD pattern obtained for the α -Fe₂O₃ thin films showed the presence of Rhombohedral structure according to the standard card for Fe₂O₃ (JCPDS No. 01-1053) Fig. 1b. Finally, Fig. 1c presents the X-ray pattern of In_2O_3/α -Fe₂O₃ where the main diffraction peaks correspond to the Rhombohedral phase of α -Fe₂O₃ (JCPDS No. 01-1053) and the cubic phase of In_2O_3 (JCPDS No. 06-0416). There were no further impurity peaks or mixed oxides, proving that there was no chemical interaction between the various metal oxides [9].

Debye–Scherrer's equation, which is given in Eq. 2, can be used to determine the average value of In_2O_3/α -Fe₂O₃ thin films crystallite size [4]:



Fig. 1 X-ray diffractogram of **a** In₂O₃, **b** α -Fe₂O₃ and **c** In₂O₃/ α -Fe₂O₃ heterostructure thin films

$$D = \frac{K\lambda}{\beta\cos\theta} \tag{2}$$

where *K* is Scherrer's constant equal to 0.9, λ is the wavelength of incident beam, β is the width at half height intensity or full width half maximum of the peak and θ represents the diffraction angle. The average crystallite size was 28.2 nm for In₂O₃ and 30.61 for In₂O₃/ α -Fe₂O₃ thin films. The larger radius of In³⁺, which has five layers of electrons as opposed to four layers for Fe³⁺, can be attributed to the Debye–Scherrer Formula's rule that a narrower peak results in a bigger grain size. As a result, In₂O₃ has a wider radius than Fe₂O₃, and the little amount of In³⁺ that is replaced by Fe³⁺ results in a reduction in the size of the average unit cell. Due to these factors, In₂O₃/ α -Fe₂O₃ thin film has higher average grains than In₂O₃ [5].

In the same vein, the relationship shown below was used to compute the microstrain (ε_{str}) that was generated in these thin films:

$$\epsilon_{str} = \frac{\beta \cos \theta}{4} \tag{3}$$

This strain, which is of order 10^{-3} , is caused by a tensile stress that due to In^{3+} ions substitution by Fe³⁺ ions in the lattice. Table 1 presents the estimated grain sizes **D** as well as the microstrain (ε_{str}) for various thin films.

Table 1 The grain size and the microstrain of $In_2O_3,\,\alpha\text{-}Fe_2O_3$ and $In_2O_3/\alpha\text{-}Fe_2O_3$ thin films

Samples	In_2O_3	In_2O_3/α -Fe ₂ O ₃	α-Fe ₂ O ₃	
The grain size D (nm)	28.2	30.61	38.11	
Strain $\boldsymbol{\varepsilon}_{str}$ (10 ⁻³)	1.23	1.13	0.91	

Raman measurements

In order to investigate the crystal structure and identify the structure of thin films, Raman spectrum was performed on the In_2O_3/α -Fe₂O₃ thin films as presented in Fig. 2.

The peaks at 243, 292, 404 and 604 cm⁻¹ belong to Eg modes of α -Fe₂O₃, while peaks at 221 and 497 cm⁻¹ relate to A1g modes. These findings supported the creation of pure Fe₂O₃, while the vibrational modes at 507 cm⁻¹ belonged to In₂O₃ [10–12].

EDX measurments

The mapping pattern for the acquired thin films is shown in Fig. 3. The spectrum for these films indicates that they only contained the three components O, Fe and In. No other elements were found in the films, which were made of the material In_2O_3/α -Fe₂O₃. In contrast to the Fe and O signals, which were evenly distributed throughout the whole structure, the In signals were primarily discovered in the core region and only rarely in the outside region. These findings supported the distinctive quality of heterostructure nanocomposites. No other elements could be detected within the sensitivity of the apparatus indicating high purity of In_2O_3/α -Fe₂O₃ thin films which is consistent with the results reported in the earlier literature [11–13].

Morphological properties

The surface morphologies of In_2O_3/α -Fe₂O₃ thin film were determined using scanning electronic microscopy (SEM) and presented in Fig. 4. The SEM images show the development of a fairly homogeneous surface with a spherical grain that is randomly disturbed and has a various of sizes. This surface disturbance is may be caused by very tiny droplets from the spray process that evaporate above the glass substrates and condense as microcrystallites of different sizes.



Fig. 2 The Raman spectra of In_2O_3/α -Fe₂O₃thin films

Fig. 3 EDX spectra from In_2O_3/α -Fe₂O₃thin film





Fig. 4 SEM micrographs of In_2O_3/α -Fe₂O₃ thin films

Table 2 Electrical resistivity ρ , the Hall mobility μ and charge carriers concentration *n* of In₂O₃/ α -Fe₂O₃ thin films

Conductivity- type	Resistivity ρ (10 ⁻⁴ Ω cm)	μ (cm ² /V S)	Carrier con- centration n $(10^{20} \text{ cm}^{-3})$
n	5.71	5.82	- 18.7

Hall effect study

At room temperature, the electrical characteristics of In_2O_3/α -Fe₂O₃ thin films were assessed using the Hall effect. Charge carrier concentration, Hall mobility and electrical resistivity dependency are summarized in Table 2.

Hall measurements show the degenerate n-type semiconductors in the thin layer [14], and the majority of free charge carriers are electrons. Charge carriers concentration, resistivity and Hall mobility are equal to $5.7 \times 10^{-4}\Omega$ cm, 5.82 cm²/V S and -18.7×10^{20} cm⁻³, respectively.

Photocatalytic test

Indium oxide has attracted a lot of attention as a semiconductor photocatalyst because it has demonstrated considerable potential for minimizing energy and environmental issues during the past few decades. To create novel photocatalysts with even improved performance, different photocatalyst systems based on indium oxide heterojunctions have been developed [15-17]. Numerous experiments have been done to alter In₂O₃ thin films to enhance their photocatalytic activity, improve their optical characteristics and extend the absorption edge to visible light. The process of coupling two semiconductors involves combining semiconductor metal oxides, such as In_2O_3/α -Fe₂O₃ has been studied for surface and optical modification, and in this work, we studied the effect In_2O_3/α -Fe₂O₃ heterojunction thin films on the degradation of MB under sun light. Wherefore, a notable way to encourage the migration of photoexcited electrons and holes and to lower its band gap is by combining metal oxides with α -Fe₂O₃ host. The photo-induced holes on α -Fe₂O₃ with higher valence band potential will combine with the photo-induced electrons on In₂O₃ with lower conduction band potential, while the stronger redox-capable electrons and holes remain on α -Fe₂O₃ and, respectively [18].

In addition, the formation of In_2O_3/α -Fe₂O₃ heterojunction can efficiently encourage the separation of photoexcited electron-hole pairs and the interfacial charge transfer; additionally, the appropriate arrangement of combined semiconductors' conduction band (CB) levels can encourage the migration of photoexcited electrons. The CB of In_2O_3 (E_{CB} for $In_2O_3 = -0.67$ V vs. NHE) is lower than that of α -Fe₂O₃ (E_{CB} for α -Fe₂O₃ = 0.32 V vs. NHE). Thus, when In_2O_3 and α -Fe₂O₃ are coupled together, the heterostructure that is generated is an effective junction to facilitate the separation of photo-induced electron-hole pairs [19].

When an active light source and a photocatalyst are present, photocatalytic activity is crucial for the breakdown and elimination of organic pollutants. It depends on an electrical phenomenon that takes place on a catalyst's surface. According to earlier investigations [20], the potential dye degradation mechanism of In_2O_3/α -Fe₂O₃ thin films can be viewed in the following ways. Under solar radiation, In_2O_3/α -Fe₂O₃ absorbs an effective photon, which causes the production of electron-hole pairs. The oxygen radical anion O₂ is created when these charge carriers are moved to the catalyst surface, where the electrons they produce are absorbed by O₂ molecules, although surfacebound entities have imprisoned the holes H₂O to produce hydroxyl radicals, OH^{*}. The generated radical species (O^{*}₂ and OH₂) aid in the degradation of the organic MB dyes [11]. The color of the MB dye solution quickly disappears when our thin layer is present. Figure 5 explains this technique in more detail.

The thin film capacity increases with dye concentration, which might be explained by a higher probability of MB molecules filling the active groups and resting on the thin film active sites given a constant volume of MB solution and adsorbent. The progressive transformation of the aqueous solution from dark blue to light blue may be detected in various stages of the layers' adsorption of MB.

The photocatalytic activity of In_2O_3/α -Fe₂O₃ thin films was assessed using the degradation of methylene blue (MB) dye under sunlight irradiation for 1–2 h Fig. 6.

Figure 7 shows that when compared to In_2O_3 thin films, In_2O_3/α -Fe₂O₃ has the best photocatalytic degradation of MB. This improvement is likely brought about by the surfaces effect and the large crystallite size discovered by SEM analysis. As a result, the thin films photocatalytic activity will be greatly influenced by its surface properties. This study found that thin films of In_2O_3/α -Fe₂O₃ showed good photocatalytic activity, which may have been impacted by the surface roughness and larger particle size.



Fig. 6 Absorption spectra of MB showing its degradation by In_2O_3 and In_2O_3/α -Fe₂O₃ thin film under solar radiation



Fig. 5 Diagram showing the mechanism of photocatalytic degradation of MB by In_2O_3/α -Fe₂O₃ thin films



Fig. 7 Photocatalysis of methylene blue: normalized concentration of MB as a function of time for In_2O_3 and In_2O_3/α -Fe₂O₃ thin film under solar radiation

Conclusion

On glass substrates, we effectively created In_2O_3/α -Fe₂O₃ thin films using an easy spray pyrolysis method. The rhombohedral structure of α -Fe₂O₃ and the cubic phase of In₂O₃ are confirmed by X-ray diffraction experiments; this conclusion was further supported by Raman spectroscopy. Scanning electronic microscopy (SEM) was used to analyze the surface morphology of In_2O_3/α -Fe₂O₃ thin film, and the results show that a fairly homogeneous surface with spherical grains of various sizes that are randomly disturbed has formed. The majority of the electrons in In_2O_3/α -Fe₂O₃ are free charge carriers, according to observations of the Hall effect. Charge carriers concentration, resistivity and Hall mobility are all identical to $5.7 \times 10^{-4} \Omega$ cm, $5.82 \text{ cm}^2/\text{V}$ S and 1.87×10^{21} cm⁻³, respectively. In contrast, for In₂O₃ and In_2O_3/α -Fe₂O₃ thin films, the grain size was equal to 28.2 nm and 30.61 nm, respectively. The photocatalytic activity for MB for these sprayed thin films is then assessed. It was discovered that the photocatalytic activity and degradation mechanism of the MB dye were affected by the paired In_2O_3/α -Fe₂O₃ photoatalysts.

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Data availability We declare that all data are available in the manuscript file.

Declarations

Conflict of interest We declare that this manuscript is original and has not been published elsewhere nor is it currently under consideration for publication elsewhere. We also declare that this work was funded by the Deanship of Graduate Studies and Scientific Research at Jouf University under grant no. (DGSSR- 2023-02-02269).

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