RESEARCH ARTICLE



# **Structural, optical and electrical properties of**  $In_2O<sub>3</sub>/\alpha$ **-Fe<sub>2</sub>O<sub>3</sub> heterostructure thin flms for photocatalytic application**

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**Abstract** In this study,  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> heterostructure thin flm was prepared on glass substrate using a low-cost and simple spray pyrolysis technique. After successful deposition, the structure and morphology were investigated by X-ray difraction, scanning electron microscopy, EDX and Raman microscopy. The X-ray difraction studies confrm the rhombohedral phase of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and the cubic phase of  $In_2O_3$  without other impurity peaks, this result also was proved by Raman spectroscopy. The surface morphology of In<sub>2</sub>O<sub>3</sub>/α-Fe<sub>2</sub>O<sub>3</sub> thin film was determined by using scanning electronic microscopy (SEM), which indicates the formation of a very homogenous surface with a spherical grain randomly disturbed with diverse sizes. Hall effect measurements showed that the conduction of  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> was n type, and the most electrons are free charge carriers. The resistivity, hall mobility and charge carriers concentration are equal to  $5.7 \times 10^{-4}$  Ω cm, 5.82 cm<sup>2</sup>/V S and  $1.87 \times 10^{21}$  cm<sup>-3</sup>, respectively. On the other hand, the grain size was equal to 28.2 nm for  $In_2O_3$  and 30.61 for In<sub>2</sub>O<sub>3</sub>/α-Fe<sub>2</sub>O<sub>3</sub> thin films. The prepared In<sub>2</sub>O<sub>3</sub>/α-Fe<sub>2</sub>O<sub>3</sub> thin flm was used as a photocatalyst against methylene blue, and this layer demonstrates signifcantly higher photocatalytic efficiency with degradation rate compared to  $In_2O_3$  thin layer which may be related to the surface roughness and crystallite size measured by XRD and SEM.

**Keywords** In<sub>2</sub>O<sub>3</sub>/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin film · Spray pyrolysis · Structural study  $\cdot$  EDX  $\cdot$  Raman microscopy  $\cdot$  Hall effect  $\cdot$ Photocatalytic activity

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## **Introduction**

The utilization of solar energy has gained attention recently since it is a sizable, sustainable and clean energy source. The effective exploitation of this energy will offer enough power to fulfll the rising global energy demand. Additionally, solar energy can be used for organic synthesis and environmental cleanup. Of all the options for converting solar energy, photoelectrochemical water decomposition is the most convincing strategy for producing clean and renewable fuels. Due to their stability, natural abundance and suitable band gap that makes use of 40% of electromagnetic radiation,  $In_2O_3$  and  $Fe<sub>2</sub>O<sub>3</sub>$  have been viewed as an appealing alternative among the many semiconducting materials for photocatalytic application  $[1, 2]$  $[1, 2]$  $[1, 2]$  $[1, 2]$ .

First of all,  $In_2O_3$  may be made into an n-type semiconductor with a high electrical conductivity. It is a transparent semiconductor with a moderate band gap of 2.8 eV. It also crystallizes in a cubic bixbyite structure and has a transparent nature. Due to its low cost, stable physicochemical and visible light responsive properties, this material has attracted a lot of attention in the photocatalytic destruction of organic contaminants and photo-splitting of water. It is well known that  $In_2O_3$ , either on its own or in combination with other oxides of transition metals, became the most widely used transparent conducting oxide (TCO) in photocatalytic and related applications. Previous research suggests that the strong visible light harvesting capacity and the efficient spatial separation of photogenerated electron hole pairs in  $In_2O_3$ , when combined with other semiconductors, improve the photocatalytic efficiency  $[3]$  $[3]$ .

Iron oxide  $Fe<sub>2</sub>O<sub>3</sub>$ , on the other hand, is a conventional form of sensing and has been studied because of its high sensitivity, rapid response, as well as durability. It is also a typical n-type semiconductor with a band gap of 2.1 eV.

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However, only a small number of studies have taken into account the development of a heterojunction system based on In<sub>2</sub>O<sub>3</sub> and Fe<sub>2</sub>O<sub>3</sub> for photocatalytic application [[4,](#page-5-3) [5\]](#page-5-4).

In order to improve the performance of acetone sensing, Zhang et al. have recently revealed simple techniques to create  $Fe<sub>2</sub>O<sub>3</sub>$ –In<sub>2</sub>O<sub>3</sub> heterostructure nanocomposites. The resulting samples showed good qualities such as a strong response and recovery capacities.

Additionally, the investigation of the Hall effect enables the determination of the concentration and mobility of charge carriers as well as, in some cases, the intrinsic energy gap and crucial data on the electrical characteristics of semiconductors [\[6](#page-5-5)].

In this work, we prepared  $In_2O_3$  thin films on glass substrate using the spray pyrolysis technique. Fe<sub>2</sub>O<sub>3</sub> was subsequently grown on the  $In_2O_3$  to form heterojunction structure. The phase structure, morphologies, Hall effect and photocatalytic performance of the resultant samples were carried out.

# **Experimental**

#### **Film preparation**

## $In_2O_3$  thin film preparation

 $In_2O_3$  thin film was prepared by spraying an aqueous solution of indium chloride  $(InCl<sub>3</sub>)$ . The concentration of indium chloride is 0.01 M. This solution was sprayed under  $N_2$  gas pressure of 0.5 bar on glass substrates, which were placed on hot plate heated at 350 °C.

# *In<sub>2</sub>O<sub>3</sub>*/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin film preparation

After preparing  $In_2O_3$  thin films,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin film was prepared from aqueous solution containing Iron(III) chloride dehydrate (FeCl<sub>3</sub>·2H<sub>2</sub>O) with concentration equal to  $10^{-2}$  M as a precursor using spray pyrolysis technique on the  $In_2O_3$ substrates at a temperature equal to 350 °C.

# **Characterization techniques**

The spectra of X-ray difraction of obtained thin flms were examined using Cu K<sub>α</sub> radiation ( $\lambda = 0.15418$  nm) and a copper-source difractometer (Analytical X Pert PROMP D), with  $2\theta$  varying from 0° to 70°. The micro-Raman system from Jobin Yvon Horibra LABRAM-HR, which is visible between 200 and 1200  $cm^{-1}$ , was used to record Raman scattering studies at room temperature. In addition, A JEOL-JSM 5400 model scanning electron microscope (SEM) was used to examine the morphology of the flms. For elemental analysis with an electron gun, an energy-dispersive X-ray (EDX) spectrometer coupled

to a thermal feld emission scanning electron microscope was employed. A Hall measuring system using the van der Pauw method was used to conduct electrical tests, resistivity, Hall mobility and carrier concentration at room temperature with a mild magnetic feld of approximately 0.554 T.

Finally, the breakdown rate of methylene blue (MB) aqueous solution under solar illumination was used to determine the photocatalytic activity of the structures based on  $In_2O_3$ and  $In_2O_3/Fe_2O_3$  thin films. MB was initially present in a concentration of 4.5 mg/L. For one hour in the dark, the MB solution was agitated at a magnetic stirrer to achieve equilibrium adsorption. In<sub>2</sub>O<sub>3</sub> and In<sub>2</sub>O<sub>3</sub>/Fe<sub>2</sub>O<sub>3</sub> thin films served as photocatalysts in the aqueous solution of MB, which was continuously stirred while exposed to sunlight. Analytical samples were taken from the solution every 1, 2, 3, 4, 5 and 6 h to determine the impact of sun light irradiation on MB solution, and a UV–Vis spectrophotometer was used to determine the amount of MB present in each sample.

The efficiency of MB degradation was calculated based on the following equation:

<span id="page-1-0"></span>Degradation efficiency(
$$
\%
$$
) =  $\left(\frac{A_0 - A}{A_0} \times 100\right) = \left(\frac{C_0 - C}{C_0} \times 100\right)$  (1)

where  $A_0$ ,  $A$ ,  $C_0$  and  $C$  are the absorbance and concentration of MB corresponding to the initial time and the variable time, respectively.

## **Results and discussion**

#### **XRD analysis**

Figure [1](#page-2-0) shows the XRD patterns obtained for the pure In<sub>2</sub>O<sub>3</sub>, pure α-Fe<sub>2</sub>O<sub>3</sub> and In<sub>2</sub>O<sub>3</sub>/α-Fe<sub>2</sub>O<sub>3</sub> heterostucture thin film. The primary diffraction peaks for the  $In_2O_3$  thin films were consistent with the conventional cubic structure represented by JCPDS card number 06-0416. These peaks demonstrate the synthesis of the pure phase  $In_2O_3$  [[7,](#page-5-6) [8](#page-5-7)]. In addition, the XRD pattern obtained for the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin films showed the presence of Rhombohedral structure according to the standard card for  $Fe<sub>2</sub>O<sub>3</sub>$  (JCPDS No. 01-1053) Fig. [1b](#page-2-0). Finally, Fig. [1](#page-2-0)c presents the X-ray pattern of  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> where the main difraction peaks correspond to the Rhombohedral phase of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (JCPDS No. 01-1053) and the cubic phase of  $In_2O_3$  (JCPDS No. 06-0416). There were no further impurity peaks or mixed oxides, proving that there was no chemical interaction between the various metal oxides [\[9](#page-5-8)].

Debye–Scherrer's equation, which is given in Eq. [2](#page-1-0), can be used to determine the average value of  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin flms crystallite size [[4\]](#page-5-3):



<span id="page-2-0"></span>**Fig. 1** X-ray diffractogram of **a**  $\text{In}_2\text{O}_3$ , **b**  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and **c** In<sub>2</sub>O<sub>3</sub>/α-Fe<sub>2</sub>O<sub>3</sub> heterostructure thin films

$$
D = \frac{K\lambda}{\beta\cos\theta} \tag{2}
$$

where *K* is Scherrer's constant equal to 0.9,  $\lambda$  is the wavelength of incident beam,  $\beta$  is the width at half height intensity or full width half maximum of the peak and  $\theta$  represents the difraction angle. The average crystallite size was 28.2 nm for In<sub>2</sub>O<sub>3</sub> and 30.61 for In<sub>2</sub>O<sub>3</sub>/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin films. The larger radius of  $In^{3+}$ , which has five layers of electrons as opposed to four layers for  $Fe^{3+}$ , can be attributed to the Debye–Scherrer Formula's rule that a narrower peak results in a bigger grain size. As a result,  $In_2O_3$  has a wider radius than Fe<sub>2</sub>O<sub>3</sub>, and the little amount of In<sup>3+</sup> that is replaced by  $Fe<sup>3+</sup>$  results in a reduction in the size of the average unit cell. Due to these factors,  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin film has higher average grains than  $In_2O_3$  [[5\]](#page-5-4).

In the same vein, the relationship shown below was used to compute the microstrain  $(\varepsilon_{str})$  that was generated in these thin flms:

$$
\varepsilon_{str} = \frac{\beta \cos \theta}{4} \tag{3}
$$

This strain, which is of order  $10^{-3}$ , is caused by a tensile stress that due to  $\text{In}^{3+}$  ions substitution by  $\text{Fe}^{3+}$  ions in the lattice. Table [1](#page-2-1) presents the estimated grain sizes *D* as well as the microstrain  $(\varepsilon_{str})$  for various thin films.

<span id="page-2-1"></span>**Table 1** The grain size and the microstrain of  $In_2O_3$ ,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and In<sub>2</sub>O<sub>3</sub>/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin films

<b>Samples</b>	$In_2O_3$	$In_2O_3/\alpha$ -Fe <sub>2</sub> O <sub>3</sub>	$\alpha$ -Fe <sub>2</sub> O <sub>3</sub>
The grain size $D$ (nm)	28.2	30.61	38.11
Strain $\varepsilon_{str}$ (10 <sup>-3</sup> )	1.23	1.13	0.91

#### **Raman measurements**

In order to investigate the crystal structure and identify the structure of thin flms, Raman spectrum was performed on the In<sub>2</sub>O<sub>3</sub>/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin films as presented in Fig. [2](#page-2-2).

The peaks at 243, 292, 404 and 604 cm<sup>-1</sup> belong to Eg modes of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, while peaks at 221 and 497 cm<sup>-1</sup> relate to A1g modes. These fndings supported the creation of pure Fe<sub>2</sub>O<sub>3</sub>, while the vibrational modes at 507 cm<sup>-1</sup> belonged to  $In_2O_3$  [[10](#page-5-9)[–12](#page-5-10)].

#### **EDX measurments**

The mapping pattern for the acquired thin flms is shown in Fig. [3](#page-3-0). The spectrum for these flms indicates that they only contained the three components O, Fe and In. No other elements were found in the flms, which were made of the material In<sub>2</sub>O<sub>3</sub>/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. In contrast to the Fe and O signals, which were evenly distributed throughout the whole structure, the In signals were primarily discovered in the core region and only rarely in the outside region. These fndings supported the distinctive quality of heterostructure nanocomposites. No other elements could be detected within the sensitivity of the apparatus indicating high purity of In<sub>2</sub>O<sub>3</sub>/α-Fe<sub>2</sub>O<sub>3</sub> thin films which is consistent with the results reported in the earlier literature [\[11](#page-5-11)[–13\]](#page-5-12).

# **Morphological properties**

The surface morphologies of  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin film were determined using scanning electronic microscopy (SEM) and presented in Fig. [4](#page-3-1). The SEM images show the development of a fairly homogeneous surface with a spherical grain that is randomly disturbed and has a various of sizes. This surface disturbance is may be caused by very tiny droplets from the spray process that evaporate above the glass substrates and condense as microcrystallites of diferent sizes.



<span id="page-2-2"></span>**Fig. 2** The Raman spectra of  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub>thin films

<span id="page-3-0"></span>**Fig. 3** EDX spectra from In<sub>2</sub>O<sub>3</sub>/α-Fe<sub>2</sub>O<sub>3</sub>thin film





**Fig. 4** SEM micrographs of  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin films

<span id="page-3-2"></span><span id="page-3-1"></span>**Table 2** Electrical resistivity  $\rho$ , the Hall mobility  $\mu$  and charge carriers concentration *n* of  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin films

Conductivity-	Resistivity $\rho$	$\mu$ (cm <sup>2</sup> /V S) Carrier con-	centration n
type	$(10^{-4} \Omega \text{ cm})$		$(10^{20} \text{ cm}^{-3})$
n	5.71	5.82	$-18.7$

# **Hall efect study**

At room temperature, the electrical characteristics of In<sub>2</sub>O<sub>3</sub>/α-Fe<sub>2</sub>O<sub>3</sub> thin films were assessed using the Hall effect. Charge carrier concentration, Hall mobility and electrical resistivity dependency are summarized in Table [2.](#page-3-2)

Hall measurements show the degenerate n-type semiconductors in the thin layer [[14\]](#page-5-13), and the majority of free charge carriers are electrons. Charge carriers concentration, resistivity and Hall mobility are equal to  $5.7 \times 10^{-4} \Omega$  cm,  $5.82 \text{ cm}^2/\text{V}$  S and  $-18.7 \times 10^{20} \text{ cm}^{-3}$ , respectively.

# **Photocatalytic test**

Indium oxide has attracted a lot of attention as a semiconductor photocatalyst because it has demonstrated considerable potential for minimizing energy and environmental issues during the past few decades. To create novel photocatalysts with even improved performance, diferent photocatalyst systems based on indium oxide heterojunctions have been developed [[15–](#page-5-14)[17\]](#page-5-15). Numerous experiments have been done to alter  $In_2O_3$  thin films to enhance their photocatalytic activity, improve their optical characteristics and extend the absorption edge to visible light. The process of coupling two semiconductors involves combining semiconductor metal oxides, such as  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> has been studied for surface and optical modifcation, and in this work, we studied the effect In<sub>2</sub>O<sub>3</sub>/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> heterojunction thin films on the degradation of MB under sun light. Wherefore, a notable way to encourage the migration of photoexcited electrons and holes and to lower its band gap is by combining metal oxides with  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> host. The photo-induced holes on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> with higher valence band potential will combine with the photo-induced electrons on  $In_2O_3$  with lower conduction band potential, while the stronger redox-capable electrons and holes remain on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and, respectively [\[18\]](#page-5-16).

In addition, the formation of  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> heterojunction can efficiently encourage the separation of photoexcited electron–hole pairs and the interfacial charge transfer; additionally, the appropriate arrangement of combined semiconductors' conduction band (CB) levels can encourage the migration of photoexcited electrons. The CB of  $In_2O_3$  ( $E_{CR}$ ) for In<sub>2</sub>O<sub>3</sub> = -0.67 V vs. NHE) is lower than that of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>  $(E_{CB}$  for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> = 0.32 V vs. NHE). Thus, when In<sub>2</sub>O<sub>3</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> are coupled together, the heterostructure that is generated is an efective junction to facilitate the separation of photo-induced electron–hole pairs [[19\]](#page-5-17).

When an active light source and a photocatalyst are present, photocatalytic activity is crucial for the breakdown and elimination of organic pollutants. It depends on an electrical phenomenon that takes place on a catalyst's surface. According to earlier investigations [[20](#page-5-18)], the potential dye degradation mechanism of In<sub>2</sub>O<sub>3</sub>/α-Fe<sub>2</sub>O<sub>3</sub> thin films can be viewed in the following ways. Under solar radiation, In<sub>2</sub>O<sub>3</sub>/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> absorbs an effective photon, which causes the production of electron–hole pairs. The oxygen radical anion  $O_2$  is created when these charge carriers are moved to the catalyst surface, where the electrons they produce are absorbed by  $O_2$  molecules, although surfacebound entities have imprisoned the holes  $H_2O$  to produce hydroxyl radicals, OH $\cdot$ . The generated radical species (O<sub>2</sub> and OH<sub>.</sub>) aid in the degradation of the organic MB dyes [[11\]](#page-5-11). The color of the MB dye solution quickly disappears when our thin layer is present. Figure [5](#page-4-0) explains this technique in more detail.

The thin flm capacity increases with dye concentration, which might be explained by a higher probability of MB molecules flling the active groups and resting on the thin flm active sites given a constant volume of MB solution and adsorbent. The progressive transformation of the aqueous solution from dark blue to light blue may be detected in various stages of the layers' adsorption of MB.

The photocatalytic activity of  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin films was assessed using the degradation of methylene blue (MB) dye under sunlight irradiation for 1–2 h Fig. [6.](#page-4-1)

Figure [7](#page-4-2) shows that when compared to  $In_2O_3$  thin films, In<sub>2</sub>O<sub>3</sub>/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> has the best photocatalytic degradation of MB. This improvement is likely brought about by the surfaces efect and the large crystallite size discovered by SEM analysis. As a result, the thin flms photocatalytic activity will be greatly infuenced by its surface properties. This study found that thin films of  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> showed good photocatalytic activity, which may have been impacted by the surface roughness and larger particle size.



<span id="page-4-1"></span>**Fig. 6** Absorption spectra of MB showing its degradation by  $In_2O_3$ and In<sub>2</sub>O<sub>3</sub>/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin film under solar radiation



<span id="page-4-0"></span>**Fig. 5** Diagram showing the mechanism of photocatalytic degradation of MB by  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin films



<span id="page-4-2"></span>**Fig. 7** Photocatalysis of methylene blue: normalized concentration of MB as a function of time for  $In_2O_3$  and  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin film under solar radiation

# **Conclusion**

On glass substrates, we effectively created  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin flms using an easy spray pyrolysis method. The rhombohedral structure of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and the cubic phase of In<sub>2</sub>O<sub>3</sub> are confrmed by X-ray difraction experiments; this conclusion was further supported by Raman spectroscopy. Scanning electronic microscopy (SEM) was used to analyze the surface morphology of  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin film, and the results show that a fairly homogeneous surface with spherical grains of various sizes that are randomly disturbed has formed. The majority of the electrons in  $In_2O_3/\alpha$ -Fe<sub>2</sub>O<sub>3</sub> are free charge carriers, according to observations of the Hall efect. Charge carriers concentration, resistivity and Hall mobility are all identical to  $5.7 \times 10^{-4} \Omega$  cm,  $5.82 \text{ cm}^2/\text{V} \text{ S}$ and  $1.87 \times 10^{21}$  cm<sup>-3</sup>, respectively. In contrast, for In<sub>2</sub>O<sub>3</sub> and In<sub>2</sub>O<sub>3</sub>/α-Fe<sub>2</sub>O<sub>3</sub> thin films, the grain size was equal to 28.2 nm and 30.61 nm, respectively. The photocatalytic activity for MB for these sprayed thin flms is then assessed. It was discovered that the photocatalytic activity and degradation mechanism of the MB dye were afected by the paired In<sub>2</sub>O<sub>3</sub>/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photoatalysts.

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**Data availability** We declare that all data are available in the manuscript fle.

## **Declarations**

**Confict of interest** We declare that this manuscript is original and has not been published elsewhere nor is it currently under consideration for publication elsewhere. We also declare that this work was funded by the Deanship of Graduate Studies and Scientifc Research at Jouf University under grant no. (DGSSR- 2023-02-02269).

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