

Polyoxometalate as the assembly material to self-assembled Ni(OH)₂ nanosheets with electrocatalytic performance

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ABSTRACT

Developing non-noble metal-based electrocatalyst with high catalytic activity is essential for advancing hydrogen energy technologies. This study introduces a hydrothermal method for synthesizing order Ni(OH)₂ nanosheets, with H₃O₄₀PW₁₂ (denoted as PW₁₂) loaded onto reduced graphene oxide (rGO) coated on nickel foam (referred to as PW₁₂-Ni(OH)₂/rGO). This method contrasts with the electrodeposition of Ni(OH)₂, where PW₁₂ is added to the synthetic system to direct the assembly and morphology of the Ni(OH)₂ through a hydrothermal reaction. In this work, the nickel foam acts dual roles as both the substrate and the source of nickel for the formation of Ni(OH)₂. The PW₁₂-Ni(OH)₂/rGO nanosheets, when successfully prepared and loaded onto the nickel foam (NF), exhibited superior electrocatalytic activity for the hydrogen evolution reaction (HER) in an alkaline electrolyte, achieving a current density of 10 mA·cm⁻² at an overpotential of 69 mV. Furthermore, we endeavored to expand the application of this material towards the oxygen evolution reaction (OER) by preparing PW₁₂-(Fe/Co)Ni(OH)₂/rGO through the addition of metal cations. This nanocomposite displayed outstanding electrocatalytic activity in alkaline electrolytes, with a current density of 10 mA·cm⁻² at an overpotential of 211 mV, and demonstrated excellent stability over a 50 h period in a 1 M KOH solution. The results presented in this paper offer an effective strategy for the preparation of polyoxometalate-based inorganic materials with diverse functionalities, applicable to both HER and OER.

KEYWORDS

polyoxometalate, Ni(OH)₂, assembly, hydrogen evolution reaction, oxygen evolution reaction

1 Introduction

Hydrogen energy, as a potential clean energy source, has been developed to replace fossil fuels in various fields, addressing the global energy shortage [1, 2]. With the growing demand for hydrogen energy, there is a need for electrocatalysts capable of withstanding large currents for water electrolysis. Noble-metal-based electrocatalysts, such as platinum (Pt) and iridium (Ir), have shown high electrocatalytic activity towards both hydrogen evolution reaction (HER) and oxygen evolution reaction (OER) [3–9]. However, the high cost and limited availability of these noble metals restrict their widespread industrial application. Consequently, numerous researchers are focusing on the design and synthesis of non-noble metal-based electrocatalysts with remarkable catalytic activity for HER and OER [10–14].

Polyoxometalates (POMs) constitute one of the largest families of nanocluster metal oxides, composed of cationic and polyanionic clusters, which endow them with remarkable redox properties. As the building blocks for assemblies, POMs offer high electronic diversity and exhibit superior catalytic performance in redox reactions [15–17]. Over the past two decades, polyoxometalate-based inorganic materials, in their nanoscale form, have been

extensively explored for their use in organic reaction, electrochemical reactions, and energy [18–23]. To diversify the physicochemical properties and applications of POMs, inorganic hybrid materials have been designed and prepared. For example, the encapsulation of POMs within metal-organic frameworks (MOF) is an effective strategy for catalytic oxidation reactions [24]. The Dexter-Silverton structure POM, when loaded onto fluoro-graphdiyne, has been shown to provide superior performance and stability for OER [25]. Additionally, helical microporous nanorods (HMNRs) have been synthesized through the assembly of POM clusters and cationic surfactants, combining the redox catalytic properties of POMs with the unique structural attributes of microporous materials [26].

In this work, we have reported a hydrothermal method for the synthesis of ordered Ni(OH)₂ nanosheets (NSs) with H₃O₄₀PW₁₂ (PW₁₂) loaded on reduced graphene oxide (rGO)-coated Ni foam, denoted as PW₁₂-Ni(OH)₂/rGO. Compared with the electrodeposition method for preparing Ni(OH)₂, the polyoxometalate cluster serves as a director for the assembly and morphology of Ni(OH)₂. The Ni foam functions not only as the substrate but also as the source of Ni for the formation of Ni(OH)₂, eliminating the need for additional Ni²⁺. The PW₁₂-

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$\text{Ni}(\text{OH})_2/\text{rGO}$ nanosheets exhibited superior electrocatalytic activity for HER in alkaline electrolyte. To further optimize the catalytic activity for OER, we assembled the electrode $\text{PW}_{12}\text{-Ni/Co/Fe}(\text{OH})_2/\text{rGO}$ nanosheets through the additional of Co^{3+} and Fe^{2+} . This composite demonstrated superior electrocatalytic activity towards OER. Based on the above research, we have identified that PW_{12} is a superior assembly for POM-based inorganic materials.

2 Experimental

2.1 Fabrication of rGO@Ni foam

Mercantile nickel foam (NF) with dimensions of $1\text{ cm} \times 2\text{ cm}$ was washed sequentially with an aqueous HCl solution (1 M), deionized water and ethanol solution, respectively. Graphene oxide (GO, 20 mg) was ultrasonically treated for 1 h in 20 mL deionized water. Then, an additional 10 mL of water was added, and the mixture was transferred to a 50 mL Teflon-lined stainless steel autoclave, along with the pre-processed NF. The NF was then completely soaked in the hydrothermal reaction solution at $160\text{ }^\circ\text{C}$ for 24 h. Afterward, the autoclave was allowed to cool to room temperature, the NF was washed with deionized water and ethanol, and subsequently dried under vacuum at $60\text{ }^\circ\text{C}$ overnight.

2.2 Fabrication of $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$

NH_4F (2 mmol), urea (5 mmol), and $\text{H}_3\text{O}_{40}\text{PW}_{12}$ (0.5 mmol) were mixed into 28 mL of deionized water and 5 mL of ethanol, stirred for 1 h, and then transferred into 50 mL Teflon-lined stainless steel autoclave. Subsequently, a piece of dry rGO@NF was added, and the mixture was subjected to a hydrothermal reaction at $120\text{ }^\circ\text{C}$ for 5 h. Afterward, the autoclave was allowed to cool to room temperature. The NF was then washed with deionized water and ethanol, and dried under vacuum at $60\text{ }^\circ\text{C}$ overnight.

2.3 Fabrication of $\text{PW}_{12}\text{-(Fe/Co)Ni}(\text{OH})_2/\text{rGO}$

FeCl_2 (0.25 mmol), $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ (1 mmol), NH_4F (2 mmol), urea (5 mmol), and $\text{H}_3\text{O}_{40}\text{PW}_{12}$ (0.5 mmol) were mixed into 28 mL of deionized water and 5 mL of ethanol, stirred for 1 h, and then transferred into 50 mL Teflon-lined stainless steel autoclave.

A piece of dry rGO@NF was added, and the mixture was subjected to a hydrothermal reaction at $120\text{ }^\circ\text{C}$ for 5 h. Afterward, the autoclave was allowed to cool to room temperature. The NF was then washed with deionized water and ethanol and dried under vacuum at $60\text{ }^\circ\text{C}$ overnight.

3 Results and discussion

According to the traditional method for preparing $\text{Ni}(\text{OH})_2$ -based electrode, $\text{NiCl}_2 \cdot 6\text{H}_2\text{O}$ is required as the Ni source [27–29]. Controlling the concentration of Ni^{2+} is a critical factor in preparing $\text{Ni}(\text{OH})_2$ -based electrodes. The POM-hybrid is an effective approach to enhance electrocatalytic activity. Different concentrations of PW_{12} affect the pH of the system, and the dissolution of NF varies in different pH environments [30–34]. Based on the aforementioned experiments, we have developed a novel method for preparing the POM-hybrid $\text{Ni}(\text{OH})_2$ -based electrode. The synthesis process of $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ electrode is depicted in Fig. 1(a). To increase the electroconductivity and the specific area of the Ni foam, graphene oxide was reduced on the surface of the Ni foam (rGO@Ni foam) in the first step. Subsequently, in the second step, PW_{12} not only acts as acid catalysts for the dissolution of Ni foam into a Ni source but also serves as a co-catalyst hybridized with $\text{Ni}(\text{OH})_2$ through a hydrothermal method. The acidity of PW_{12} is a key factor in dissolving Ni foam, enabling the growth of $\text{Ni}(\text{OH})_2$ nanosheets from the three-dimensional (3D) porous structure of rGO. We tested various concentrations of PW_{12} : 0.1, 0.3, 0.5, 0.6, 0.7, and 0.9 mmol, respectively. From Figs. 1(b)–1(g), we determined that a PW_{12} concentration of 0.5 mmol was optimal for preparing uniform $\text{Ni}(\text{OH})_2$ nanosheets, which could compete favorably with the traditional method using a Ni source. The scanning electron microscopy (SEM) images revealed that PW_{12} exhibits superior acidity for dissolving Ni foam, surpassing that of inorganic acids such as HCl (Fig. S12(a) in the Electronic Supplementary Material (ESM)). In summary, we have identified a new method for preparing $\text{Ni}(\text{OH})_2$ nanosheets with POM, which could prove to be a valuable POM-based inorganic material.

To identify the structure and chemical composition of the

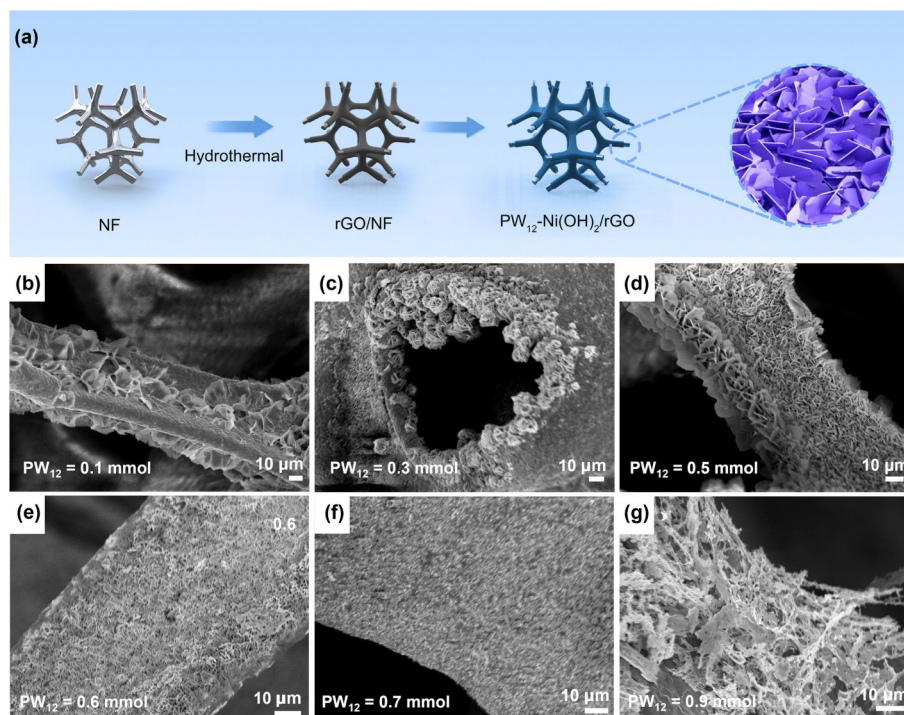


Figure 1 (a) Illustration of the preparation of the $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ electrode. (b)–(g) The morphology of $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ with the different concentration of PW_{12} (0.1, 0.3, 0.5, 0.6, 0.7, and 0.9 mmol).

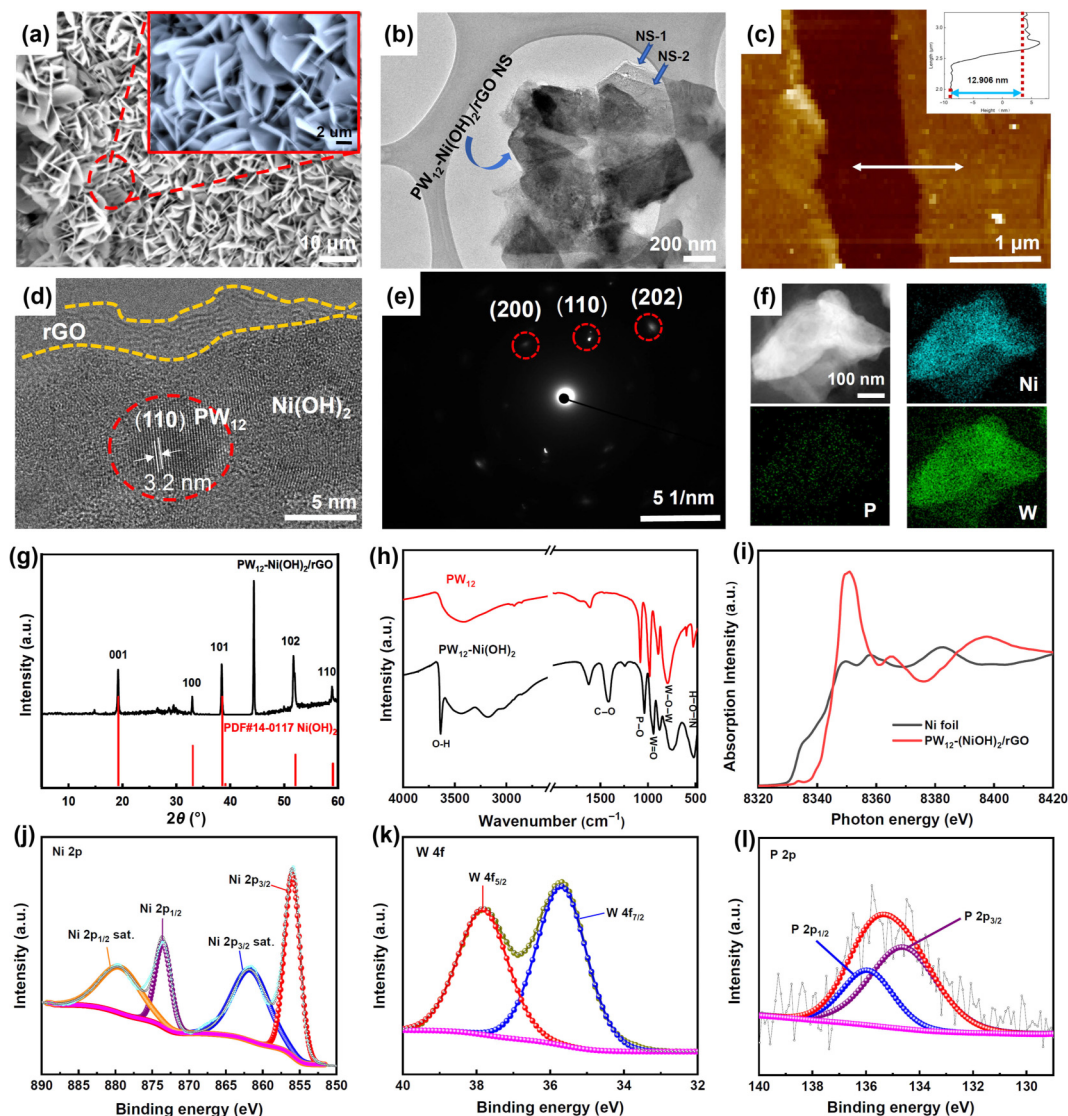


Figure 2 (a) SEM images about the morphology of $\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$ nanosheets on Ni foam. (b) TEM image of $\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$ nanosheets. (c) The thickness of $\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$ nanosheets tested by AFM. (d) Magnified HRTEM image of $\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$ nanosheets, and (e) the corresponding fast Fourier transforms. (f) Scanning TEM image of $\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$ nanosheet and the corresponding element mapping of Ni, P, W. (g) PXRD data of $\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$. (h) IR spectra of $\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$ and PW_{12} . (i) Ni K-edge XANES spectra of $\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$ and Ni foil. High-resolution XPS spectrum of $\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$ showing (j) Ni 2p, (k) W 4f, and (l) P 2p.

mentioned material, SEM architecture could provide more catalytic centers. From the transmission electron microscopy (TEM) image in Fig. 2(b), the nanosheet structure was further confirmed. For instance, the overlapping structure between nanosheet-1 (NS-1) and nanosheet-2 (NS-2) can be clearly observed, which is similar to that observed with SEM. Moreover, the thickness of the NS is a typical parameter for evaluating the hybrid capability of PW_{12} . In Fig. 2(c), an atomic force microscope (AFM) was used to measure the thickness of the NS, which was found to be 12.906 nm—sufficient to provide space for POM hybridization. Additionally, the high-resolution TEM (HRTEM) image in Fig. 2(d) shows that PW_{12} (110) has successfully hybridized with the Ni(OH)_2 nanosheets [35]; rGO is also observed, which can be attributed to the edges of the NS on the Ni foam. The corresponding selected-area electron diffraction (SAED) patterns reveal lattice fringes whose spacings are consistent with the (200), (110), and (202) planes of the Ni(OH)_2 lattice (Fig. 2(e)), thus supporting the structural assignment [36]. Furthermore, the scanning TEM and corresponding elemental mapping results (Fig. 2(f)) demonstrate the homogeneous distribution of Ni, P, and W across the entire nanosheet.

Due to the synthesis method being different from the traditional method, we should use additional characterizations to

confirm that the nanosheets are indeed Ni(OH)_2 . Firstly, the powder X-ray diffraction (PXRD) of $\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$ indicates that the crystal lattice corresponds to Ni(OH)_2 [24]. Then, infrared (IR) spectroscopy was further used to identify the structure of $\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$, the peaks at 535 and 3635 cm^{-1} could be attributed to Ni–O–H and O–H, respectively. There are three distinct peaks at 1035 cm^{-1} for P–O, 944 cm^{-1} for Mo=O, 869 cm^{-1} for Mo–O–Mo, which can be attributed to PW_{12} . The peak at 1421 cm^{-1} can be attributed to the C–O of rGO. Based on the IR data, we can confirm that PW_{12} had hybridized on Ni(OH)_2 , indicating that the POM-based inorganic material ($\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$) has been successfully prepared. We then proceeded to ascertain the structure of $\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$ using extended X-ray absorption fine structure (EXAFS). The energy absorption at the Ni K-edge X-ray absorption near edge structure (XANES, as shown in Fig. 2(i)) confirms that the structure is indeed Ni(OH)_2 [37]. Concurrently, X-ray photoelectron spectroscopy (XPS) was performed to analyze $\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$. Figure 2(j) showed the high-resolution spectrum of Ni 2p, with the main Ni 2p_{3/2} peak and its satellite peaks at 855.9 and 861.8 eV, and the Ni 2p_{1/2} main peak and its satellite peaks at 873.6 and 880.1 eV, respectively. Two signals for W^{6+} (4f_{7/2} and 4f_{5/2}) are observed at 35.9 and 38.1 eV (Fig. 2(k)) [38].

Furthermore, the P 2p high-resolution spectrum (Fig. 2(l)) can be resolved into P–O [39]. The XPS data further confirms that PW_{12} had been hybridized on $\text{Ni}(\text{OH})_2$.

According to the above data, the new method for preparing the POM-based inorganic materials $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ through the assembly of the PW_{12} cluster and $\text{Ni}(\text{OH})_2$ has been established successfully.

Then, we applied $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ as the electrocatalyst for the HER. Its electrocatalytic performance toward water oxidation was investigated in a 1.0 M KOH solution at room temperature using a standard three-electrode system setup. This setup included a working electrode $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$, a standard calomel reference electrode (SCE), and a graphite rod counter electrode. The original $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$, $\text{HCl-Ni}(\text{OH})_2/\text{rGO}$, $\text{PMo}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$, $\text{SiW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ and $\text{SiW}_{11}\text{-Ni}(\text{OH})_2/\text{rGO}$ were tested as references under the same conditions. The linear sweep voltammetry (LSV) curves of all samples were tested at a scan rate of $5 \text{ mV}\cdot\text{s}^{-1}$. As shown in Fig. 3(a), $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ exhibited an earlier onset potential of 236 mV (vs. RHE) and a higher current density than $\text{HCl-Ni}(\text{OH})_2/\text{rGO}$, $\text{PMo}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$, $\text{SiW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ and $\text{SiW}_{11}\text{-Ni}(\text{OH})_2/\text{rGO}$, suggesting its improved and remarkable activity. It requires a potential of 236 mV (vs. RHE) to reach $50 \text{ mA}\cdot\text{cm}^{-2}$, which is lower than that of $\text{HCl-Ni}(\text{OH})_2/\text{rGO}$ (292 mV vs. RHE), $\text{PMo}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ (338 mV vs. RHE), $\text{SiW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ (372 mV vs. RHE) and $\text{SiW}_{11}\text{-Ni}(\text{OH})_2/\text{rGO}$ (426 mV vs. RHE). In addition, the catalytic activity could be achieved with a low potential of 700 mV (vs. RHE) at a high current density ($358 \text{ mA}\cdot\text{cm}^{-2}$). Moreover, as shown in Fig. 3(b), the Tafel slopes of $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$, $\text{HCl-Ni}(\text{OH})_2/\text{rGO}$, $\text{PMo}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$, $\text{SiW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ and $\text{SiW}_{11}\text{-Ni}(\text{OH})_2/\text{rGO}$ are 129, 407, 281, 367 and $255 \text{ mV}\cdot\text{dec}^{-1}$, respectively, confirming the superior catalytic kinetics of $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$.

Operating under basic aqueous conditions, corrosion resistance is the most important parameter affecting the performance of the OER, including both its activity and stability. Remarkably, in this work, rGO could not only effectively protected the activity center

but also increased the electroconductivity. To evaluate the long-term stability of the electrode, we conducted chronoamperometry measurements with a -0.334 V (vs. RHE) bias at a pH of 14, the $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ electrode displayed a current density of $100 \text{ mA}\cdot\text{cm}^{-2}$, which decreased to $4 \text{ mA}\cdot\text{cm}^{-2}$ after 50 h (Fig. 3(c)). However, at the same applied potentials, the $\text{HCl-Ni}(\text{OH})_2/\text{rGO}$, $\text{PMo}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$, $\text{SiW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ and $\text{SiW}_{11}\text{-Ni}(\text{OH})_2/\text{rGO}$ electrodes exhibited a much worse stability (Fig. S29 in the ESM). This suggests that the encapsulation of rGO in $\text{Ni}(\text{OH})_2$ is an effective way to enhance the stability of the electrode. Furthermore, the stability of $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ was also assessed with post catalytic SEM and LSV analysis. The morphology can be clearly observed, being the same as that of the original electrode in Fig. S29 in the ESM. The above experimental data also indicate that the $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ electrode maintains high catalytic performance after 50 h chronoamperometry measurements for the OER in alkaline electrolytes.

The electrochemically active surface area (ECSA) of all samples was evaluated by measuring the electrochemical double-layer capacitances (C_{dl}) in 1 M KOH via cyclic voltammetry at different scan rates (Fig. S28 in the ESM). The C_{dl} of $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ is $9.78 \text{ mF}\cdot\text{cm}^{-2}$ (Fig. 3(d)), higher than that of $\text{HCl-Ni}(\text{OH})_2/\text{rGO}$ ($4.03 \text{ mF}\cdot\text{cm}^{-2}$), $\text{PMo}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ ($5.27 \text{ mF}\cdot\text{cm}^{-2}$), and $\text{SiW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ ($1.93 \text{ mF}\cdot\text{cm}^{-2}$). Next, we investigated the catalytic kinetics of the reactions at the electrode/electrolyte interface by electrochemical impedance spectroscopy (EIS). As shown in Fig. S30(a) in the ESM, $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ has the lowest solution resistance ($R_s = 1.28 \Omega$) and charge-transfer resistance ($R_{ct} = 4.72 \Omega$), indicating faster charge-transfer behavior. This demonstrates that PW_{12} provides a faster electron and mass transfer pathway, leading to an improvement in the catalytic performance. The recorded impedance data were fitted using a circuit model (Fig. S30(b) in the ESM).

Up to this point, PW_{12} -based inorganic material has exhibited the catalytic activity towards HER. For expanding its application for OER, we attempted to prepare a series of new $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ by adding of metal cationic ($\text{Co}^{2+}/\text{Fe}^{2+}$, Co^{3+} and Fe^{3+} ,

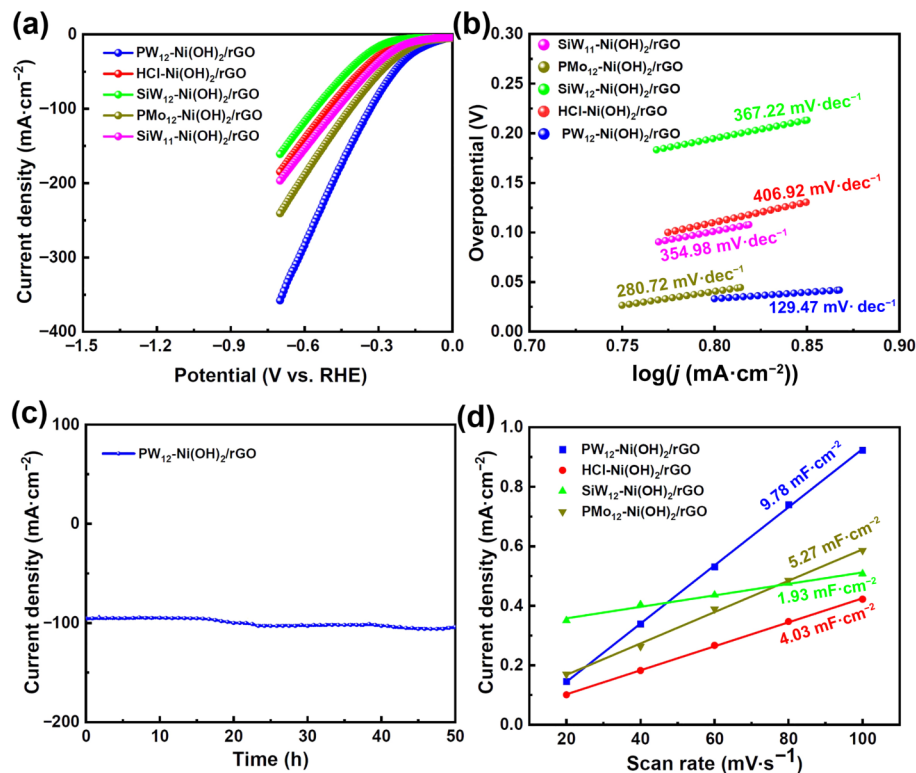


Figure 3 (a) LSV polarization curves of $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$, $\text{HCl-Ni}(\text{OH})_2/\text{rGO}$, $\text{PMo}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$, $\text{SiW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ and $\text{SiW}_{11}\text{-Ni}(\text{OH})_2/\text{rGO}$ in 1 M KOH. (b) Tafel plots of $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$, $\text{HCl-Ni}(\text{OH})_2/\text{rGO}$, $\text{PMo}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$, $\text{SiW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ and $\text{SiW}_{11}\text{-Ni}(\text{OH})_2/\text{rGO}$ in 1 M KOH. (c) Stability test results of $\text{PW}_{12}\text{-Ni}(\text{OH})_2/\text{rGO}$ at 1.56 V (vs. RHE). (d) Plots of the current density differences vs. the scan rate for various catalysts.

respectively) with more catalytic center, resulting in the corresponding electrodes $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$, $\text{PW}_{12}\text{-(Fe)Ni(OH)}_2/\text{rGO}$ and $\text{PW}_{12}\text{-(Co)Ni(OH)}_2/\text{rGO}$.

SEM and TEM images of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ revealed a hierarchical nanostructure of microscale plate, which can facilitate ion contact with electrolyte and the expose of active sites. As shown in Fig. 4(a), the $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ nanosheet deposit homogeneously and densely on the surface of NF. From TEM image of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ in Fig. 4(b), we could observe the morphology of nanosheet, which is similar to that in the SEM image in Fig. 2(a). Additional, the IR spectrum (Fig. 4(c)) was the typical characterization for confirming the structure of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ was not only similar with the structure of $\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$, but also composed of PW_{12} and Ni(OH)_2 .

For better characterization of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$, pure $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$, which was removed from the NF by sonication, was analyzed by XPS. The XPS spectra of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ exhibit Fe 2p, Co 2p, and Ni 2p compositions from Figs. 4(d)–4(f). The Fe 2p region (Fig. 4(d)) comprises four easily discernible features: the Fe $2p_{3/2}$ spectrum generally show a main peak at 711.78 eV, accompanied by a shake-up satellite at 716.04 eV, indicating the presence of Fe^{3+} cations. The Co 2p region (Fig. 4(e)) also comprises four easily discernible features: the main Co $2p_{3/2}$ peak and its satellite peaks at 780.93 and 784.08 eV, respectively, and the Co $2p_{1/2}$ main peak and its

satellite peaks at 796.72 and 803.61 eV, respectively. It can be judged that Co is mainly Co–OH, with the oxidation state of Co being Co(II). In addition, there are four peaks for the Ni 2p signal in the binding energy range of 855–884 eV (Fig. 4(f)), The energy difference between the Ni $2p_{3/2}$ and Ni $2p_{1/2}$ in the fine Ni 2p spectrum is about 17.3 eV, and the spectral peak area ratio (Ni $2p_{3/2}$:Ni $2p_{1/2}$) is about 2:1. According to the energy position of the Ni $2p_{3/2}$ peak (about 856.1 eV) and the satellite peaks (about 861.6 eV), it can be judged that Ni is mainly present as NiOOH, with the oxidation state of Ni being Ni(II). Overall, the XPS data of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ further verify its structure [30].

The $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ with its well-defined structure and chemical compositions, is applied as the electrocatalyst for OER in 1.0 M KOH. To identify the effect of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$, $\text{PW}_{12}\text{-(Fe)Ni(OH)}_2/\text{rGO}$ and $\text{PW}_{12}\text{-(Co)Ni(OH)}_2/\text{rGO}$ as reference electrodes were tested under the same conditions. The linear scanning voltammetry (LSV) curves of all the electrodes were measured at a scan rate of $5 \text{ mV}\cdot\text{s}^{-1}$. As exhibited in Fig. 5(a), $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ had the highest electrocatalytic activity, offering an overpotential of 211 mV at a current density of $10 \text{ mA}\cdot\text{cm}^{-2}$, which is smaller than that of $\text{PW}_{12}\text{-(Fe)Ni(OH)}_2/\text{rGO}$ (341 mV), $\text{PW}_{12}\text{-(Co)Ni(OH)}_2/\text{rGO}$ (255 mV). In addition, the Tafel slope is also an important factor for evaluating OER kinetics. As displayed in Fig. 5(b), the Tafel slopes of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$, $\text{PW}_{12}\text{-(Fe)Ni(OH)}_2/\text{rGO}$, and $\text{PW}_{12}\text{-(Co)Ni(OH)}_2/\text{rGO}$ are 66.53, 129.18, and $168.62 \text{ mV}\cdot\text{dec}^{-1}$,

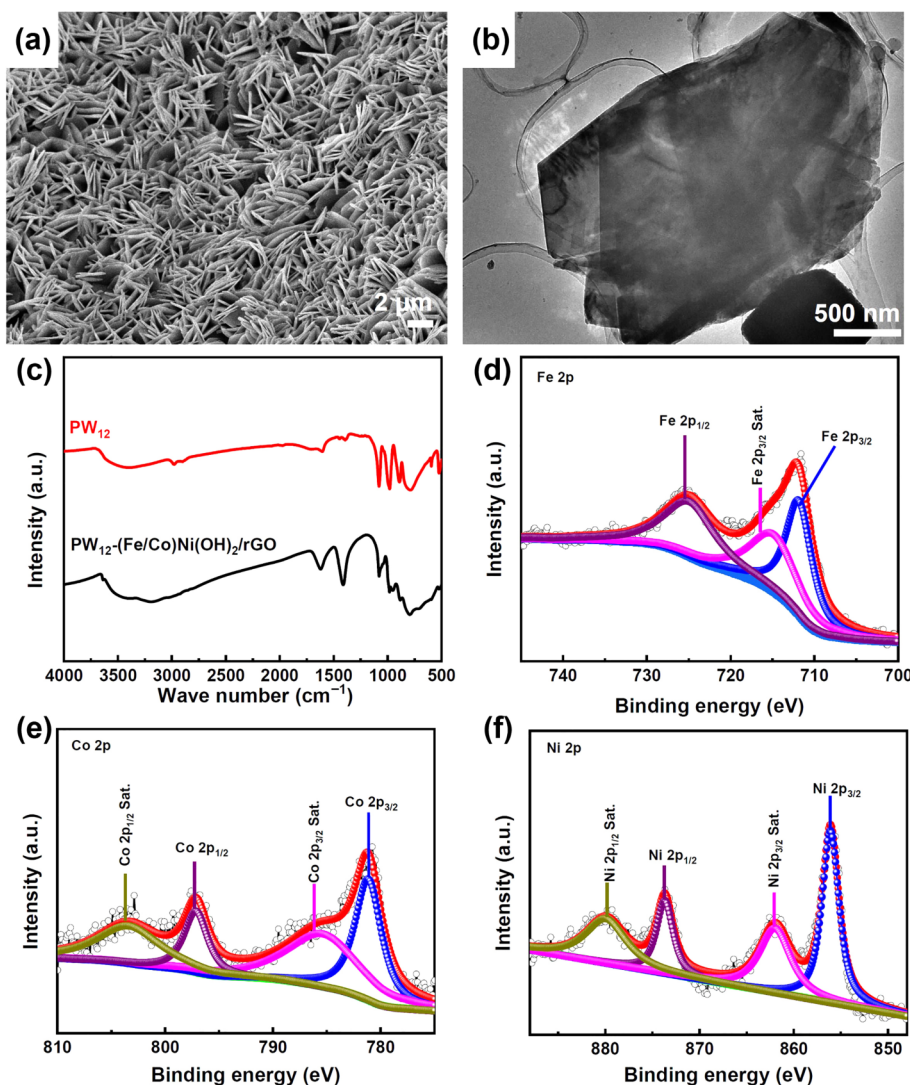


Figure 4 (a) SEM image about the morphology of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ nanosheets on Ni foam. (b) TEM image of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ nanosheets. (c) IR spectra of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ and PW_{12} . High-resolution XPS spectrum of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ showing (d) Fe 2p; (e) Co 2p, and (f) Ni 2p.

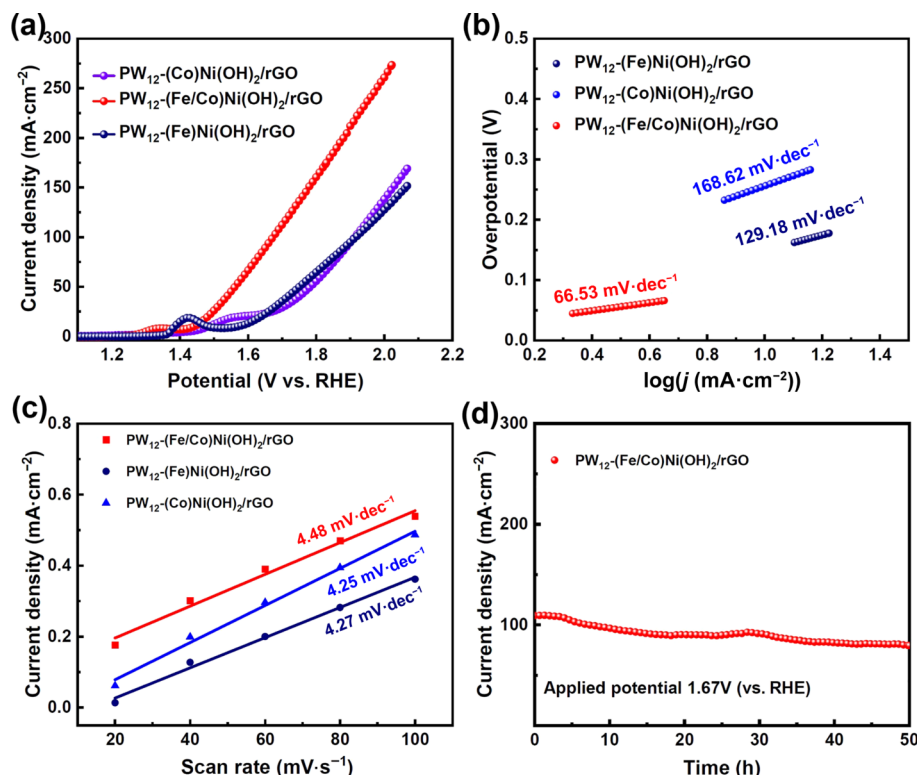


Figure 5 (a) LSV polarization curves of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$, $\text{PW}_{12}\text{-(Fe)Ni(OH)}_2/\text{rGO}$ and $\text{PW}_{12}\text{-(Co)Ni(OH)}_2/\text{rGO}$ in 1 M KOH. (b) Tafel slope of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$, $\text{PW}_{12}\text{-(Fe)Ni(OH)}_2/\text{rGO}$ and $\text{PW}_{12}\text{-(Co)Ni(OH)}_2/\text{rGO}$. (c) Plots of the current density differences vs. the scan rate for various catalysts. (d) Stability test results of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ at 1.67 V (vs. RHE).

respectively, which supports $\text{PW}_{12}\text{-(Fe)Ni(OH)}_2/\text{rGO}$ (341 mV) has excellent catalytic kinetics.

The electrochemical activity surface area (ECSA) was utilized to investigate the superior OER performance of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$. Initially, the C_{dl} is determined by cyclic voltammetry at varying scan rates (Fig. S27 in the ESM). The C_{dl} of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ is measured at 4.48 $\text{mF}\cdot\text{cm}^{-2}$, which is higher than that of $\text{PW}_{12}\text{-(Fe)Ni(OH)}_2/\text{rGO}$ at 4.27 $\text{mF}\cdot\text{cm}^{-2}$ and $\text{PW}_{12}\text{-(Co)Ni(OH)}_2/\text{rGO}$ at 4.25 $\text{mF}\cdot\text{cm}^{-2}$. Furthermore, the stability of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ is demonstrated in Fig. 5(c). Using chronoamperometry measurements with a bias of 1.67 V (vs. RHE) bias at a pH of 14, the $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ electrode exhibited a current density of 100 $\text{mA}\cdot\text{cm}^{-2}$, which gradually decreased to 20 $\text{mA}\cdot\text{cm}^{-2}$ after 50 h (Fig. 5(d)). In addition to chronoamperometry measurements, SEM and LSV analysis were conducted on the $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ electrode. The regularly nanosheets are observed to lie on the NF, indicating that the structure of $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ is structurally intact. In conclusion, $\text{PW}_{12}\text{-(Fe/Co)Ni(OH)}_2/\text{rGO}$ exhibits remarkable OER performance, where metal ions play a regulating role in the construction of composite materials.

4 Conclusions

Based on our previous research of electrocatalysts composed of polyoxometalate and non-noble metal, we have developed a method to prepare POM hybrid Ni(OH)_2 nanosheets on rGO. The nanosheet structure provides a large surface area for catalytic reactions, and the rGO offers excellent electrical conductivity. During the synthesis process, PW_{12} exhibited two key mechanisms for the electrode with electrochemical performance: (i) The activity of PW_{12} , a vital factor for dissolving Ni foam, which provides the Ni source for the preparation of Ni(OH)_2 nanosheets; (ii) PW_{12} acts as a co-catalysts of Ni(OH)_2 , enhancing the electrocatalytic activity for HER in alkaline conditions, which is superior to other POMs. PW_{12} is not only the superior assembly agent but also a

catalytic center for the $\text{PW}_{12}\text{-Ni(OH)}_2/\text{rGO}$ electrode. The interaction between the Ni(OH)_2 and the POM could lead to enhanced catalytic activity due to the synergy between Ni(OH)_2 and POM. Our findings offer new insights for the development of POM-based inorganic materials.

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