

Electrocatalytic water splitting: Mechanism and electrocatalyst design

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Received: 17 December 2022 / Revised: 7 January 2023 / Accepted: 12 January 2023

ABSTRACT

Hydrogen energy, a new type of clean and efficient energy, has assumed precedence in decarbonizing and building a sustainable carbon-neutral economy. Recently, hydrogen production from water splitting has seen considerable advancements owing to its advantages such as zero carbon emissions, safety, and high product purity. To overcome the large energy barrier and high cost of water splitting, numerous efficient electrocatalysts have been designed and reported. However, various difficulties in promoting the industrialization of electrocatalytic water splitting remain. Further, as high-performance electrocatalysts that satisfy industrial requirements are urgently needed, a better understanding of water-splitting systems is required. In this paper, the latest progress in water electrolysis is reviewed, and experimental evidence from *in situ/operando* spectroscopic surveys and computational analyses is summarized to present a mechanistic understanding of hydrogen and oxygen evolution reactions. Furthermore, some promising strategies, including alloying, morphological engineering, interface construction, defect engineering, and strain engineering for designing and synthesizing electrocatalysts are highlighted. We believe that this review will provide a knowledge-guided design in fundamental science and further inspire technical engineering developments for constructing efficient electrocatalysts for water splitting.

KEYWORDS

water splitting, mechanism, *operando* spectroscopy, design strategy

1 Introduction

Critical challenges arising from environmental issues and the increasing consumption of fossil fuels have necessitated the development of high-efficiency, green, and sustainable energy technologies [1]. Renewable energy resources, including solar, wind, and geothermal energies, are alternatives to traditional fossil fuels [2]. However, the spatial distribution of these energy resources has severely limited their large-scale applications. Hydrogen energy is considered as an ideal substitute owing to its widespread sources, high calorific values of combustion, and zero carbon emissions. Currently, approximately 95% of the world's hydrogen is produced by the conversion of natural gas and fossil fuels. Additionally, the production process is energy-intensive, and the produced hydrogen is prone to contamination [3]. Conversely, electrocatalytic water splitting is a simple, safe, and high-purity process for hydrogen production and has attracted considerable attention in recent years [4].

Water splitting comprises a cathodic hydrogen evolution reaction (HER) and an anodic oxygen evolution reaction (OER). In recent years, abundant studies have focused on exploring the mechanism of water splitting on electrocatalyst surfaces, especially OER. The mechanism of OER can be mainly divided into the adsorption evolution mechanism (AEM), oxidation path mechanism (OPM), and lattice oxygen-mediated mechanism (LOM). Compared to the HER, which is a two-electron process, the OER is a four-electron process involving multiple sub-

reactions, which leads to a high energy barrier and low energy conversion efficiency for water electrolysis [5]. Over recent decades, significant efforts have been devoted toward optimizing water-splitting systems and developing efficient HER/OER electrocatalysts. Various materials with high activities and selectivities, including noble metals, oxides/hydroxides, sulfides, and phosphates [6–8], have been developed. Among them, noble metal-based electrocatalysts (e.g., ruthenium, platinum, and iridium) and the corresponding oxides are excellent electrocatalysts for the HER and OER [9, 10], respectively. However, their limited reserves and high price have significantly limited their widespread application. To reduce the amount of noble metals while maintaining the excellent activity of these electrocatalysts, some effective strategies, such as reducing the particle size or nanocrystallization, alloying, and constructing heterogeneous interfaces, have been successfully developed, and the composition and electronic structures have been finely regulated such that a sufficient number of active sites are exposed for water splitting. Consequently, these materials exhibit immense potential to replace commercial electrocatalysts under experimental conditions [11–13]. However, when subjected to industrial conditions, these as-designed electrocatalysts have to withstand high currents and long running times, which requires the materials to simultaneously possess excellent activities and good corrosion resistances. Therefore, an atomic-level in-depth study of the water-splitting process has become an important topic

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in recent decades [14–16]. Using advanced *in situ/operando* characterization techniques, including Raman/infrared spectroscopy and X-ray absorption fine structure analyses combined with theoretical predictions, researchers have realized the real-time tracking and monitoring of intermediates at a molecular level during water splitting [17–20]. Understanding the microscopic processes occurring on the electrocatalyst surfaces during water splitting is conducive toward establishing the intrinsic structure–activity relationship and providing important guidelines to design and develop highly efficient HER/OER electrocatalysts.

In this paper, the development of water splitting technology, HER and OER mechanisms, and design strategies for efficient electrocatalysts reported recently are introduced, as illustrated in Fig. 1. Experimental evidence from spectroscopic investigations combined with theoretical studies is summarized and analyzed in detail to present a mechanistic understanding. More importantly, some promising strategies developed to construct active sites and regulate electronic properties, such as the alloying and engineering of morphologies, interfaces, and defects, are discussed, and the corresponding structure–property relationships are established to enhance the catalytic efficiency of water splitting. Finally, this review provides a detailed summary of the problems and challenges regarding the future development of water-splitting technologies, with the aim of providing valuable insights into fundamental study and technical engineering for the optimal design and modification of highly efficient HER/OER electrocatalysts.

2 Reaction mechanism of water splitting

Water splitting is a simple and rapid process that can be performed at room temperature ($2\text{H}_2\text{O} + \text{energy} \rightarrow 2\text{H}_2 + \text{O}_2$). Specifically, positive and negative electrodes are inserted into an aqueous solution and connected to a power supply, and a certain voltage is applied to overcome the energy barrier required for the catalytic splitting of water molecules. Generally, hydrogen ions in an aqueous solution undergo a reduction reaction at the cathode to form H_2 , whereas hydroxide ions undergo an oxidation reaction at the anode to form O_2 . During this process, the change in the free energy (ΔG) of water electrolysis is $237.2 \text{ kJ}\cdot\text{mol}^{-1}$ at $25 \text{ }^\circ\text{C}$ and 1 atm with a theoretical voltage of 1.23 V to initiate water splitting. However, owing to the poor kinetics of the

electrochemical reaction, a high overpotential is required to compensate the potential loss. Therefore, the water-splitting process often uses highly active electrocatalysts to facilitate electrochemical reactions on electrode surfaces to reduce energy consumption. Successful commercial applications of water electrolyzers, which are regarded as the next-generation renewable energy devices for the conversion of intermittent electricity generated from sustainable wind, solar, and tidal power into fuels [21, 22], are of great significance for addressing environmental issues and the energy crisis. Based on the different membrane materials in electrolytic cells, water splitting can selectively occur via proton exchange membrane, basic, or solid oxide electrolysis. In general, the HER and OER during energy conversion/storage play pivotal roles in improving the overall efficiency of water electrolysis. Therefore, knowledge-guided principles are needed to design optimal electrocatalysts with satisfactory activity, selectivity, and electrochemical stability for a particular reaction.

2.1 HER mechanism

The HER is a two-electron transfer process that occurs at the cathode surface to produce a hydrogen molecule. During this process, the electrical energy is converted into chemical energy. Although the mechanism of the HER is slightly different under acidic and alkaline conditions, both involve the adsorption and desorption of some typical reaction intermediates. As shown in Figs. 2(a) and 2(b), a proton is first adsorbed on the catalyst surface to form an adsorbed hydrogen intermediate ($\text{M}\text{-H}_{\text{ads}}$, where M represents the active site), which is accompanied by the deduction of one electron according to the Volmer reaction. The second step can be divided into the Tafel and Heyrovsky reactions, depending on the coverage of H_{ads} . These two reaction routes can be approximately identified using the Tafel slopes, which are calculated from the experimental plots of the overpotential as a function of $\log|\text{current density}|$. Theoretically, the Tafel reaction refers to the direct coupling of two adsorbed hydrogens on an electrocatalyst surface to generate a hydrogen molecule, whereas the Heyrovsky reaction involves the combination of one adsorbed hydrogen on the electrocatalyst surface with another hydrogen ion in the electrolyte under acidic conditions, or another water molecule under alkaline conditions, to form a hydrogen molecule. Because the adsorbed hydrogen intermediate exists in both, the Volmer–Heyrovsky and Volmer–Tafel pathways, the free energy of hydrogen adsorption (ΔG_{Hads}) can serve as an important indicator reflecting the HER activity of electrocatalysts. According to the Sabatier principle, the ΔG_{Hads} of an ideal electrocatalyst should be close to zero. When ΔG_{Hads} is below zero, the bonding strength is too low to effectively activate the hydrogen adsorption reaction, resulting in a limited HER activity. In addition, ΔG_{Hads} values above zero are not ideal because the strong binding suggests difficult desorption of the $\text{M}\text{-H}_{\text{ads}}$ intermediates, which correspondingly limits the total catalytic efficiency. Figure 2(c) shows a volcanic diagram of the HER activity and theoretical ΔG_{Hads} for different metal active sites [23]. Noble metals, such as platinum, have a ΔG_{Hads} close to zero and show high HER activity. Furthermore, ΔG_{Hads} can be adjusted close to zero by modifying the geometry and electronic structures of the active site, thus improving the hydrogen evolution activity. For example, to explore the relationship between the HER activity and thermodynamic ΔG_{Hads} , Lee et al. synthesized a series of silicon nanowire-supported metal electrocatalysts (metal/SiNWs) with modified metal–silicon interfaces [24]. Density functional theory (DFT) calculations showed that the constructed silicon–osmium interface performed best and exhibited an optimal ΔG_{Hads} of only -0.03 eV , far lower than those of monomeric osmium (-0.26 eV) and the noble metal platinum (-0.09 eV). Further tests also

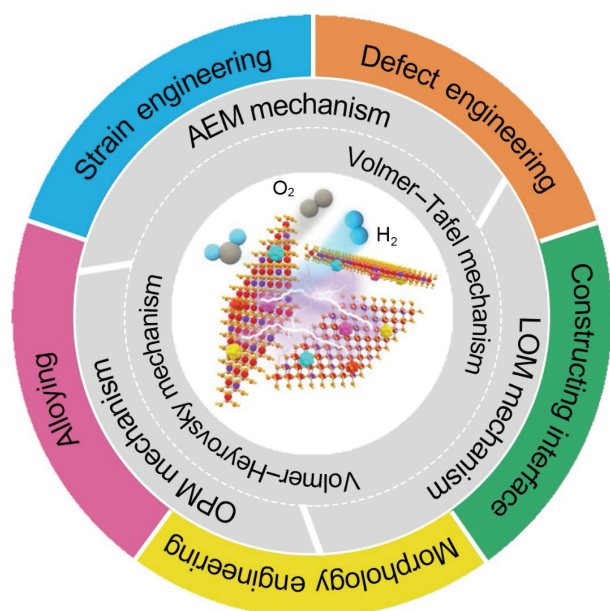


Figure 1 Mechanism of water splitting and electrocatalyst design strategies.

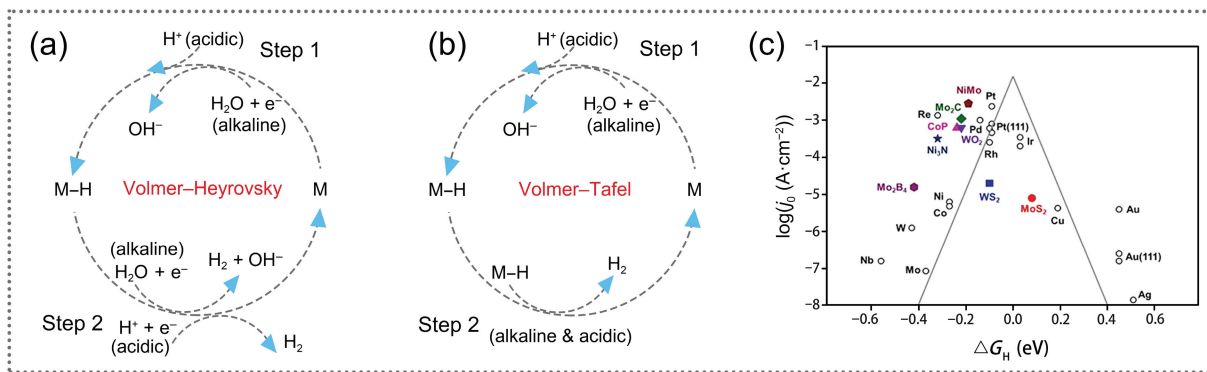


Figure 2 Reported HER mechanisms: (a) Volmer–Heyrovsky and (b) Volmer–Tafel pathways. (c). The HER Volcano plot for various transition metals against the ΔG_{Hads} . Reproduced with permission from Ref. [23], © Elsevier Ltd. 2019.

confirmed that Os/SiNWs exhibited an HER activity superior to that of commercial Pt/C.

2.2 OER mechanism

Compared with the HER, the OER follows a multistep proton-coupled electron transfer process with various elementary reactions. Therefore, the OER is a relatively slow kinetic process, which is a key factor restricting the total efficiency of water splitting. Under standard conditions, the thermodynamic equilibrium potential of the OER is 1.23 V; however, in most cases, an additional potential, i.e., the overpotential, is needed because of some adverse kinetic factors in actual water electrolysis. The overpotential can be reduced by introducing suitable electrocatalysts. Thus, understanding the OER mechanism can be conducive for theoretically designing more efficient electrocatalysts to reduce the overpotential.

According to recent explorations, three main OER mechanisms, including AEM, OPM, and LOM, have been proposed. The specific reaction process is illustrated in Fig. 3. In the AEM pathway (Fig. 3(a)), the OER is a four-step electron transfer process involving several reaction intermediates (such as OH_{ads} , O_{ads} , OOH_{ads} , and $\text{O}_{2\text{ads}}$). Specifically, hydroxides (in alkaline electrolytes) or water molecules (in acidic electrolytes) are first adsorbed onto the active sites of electrocatalysts to form M-OH_{ads} . The pre-adsorbed M-OH_{ads} is then converted to M-O_{ads} , which further converts to $\text{M-OOH}_{\text{ads}}$ species. Finally, O_2 can be generated by losing a proton and an electron from $\text{M-OOH}_{\text{ads}}$ in an acidic environment or by a proton-coupled electron transfer of M-OOH^* under the attack of OH^- in an alkaline environment. For the OER in alkaline and acidic electrolytes, the change in the free energy of the entire reaction is 4.92 eV, and the step with a high Gibbs free energy is considered as the rate-determining step (RDS) of the OER. Referring to other reported studies, the adsorption free energy difference between OOH_{ads} and OH_{ads} intermediates ($\Delta G_{\text{OOHads}} = \Delta G_{\text{OHads}} + 3.2 \pm 0.2$ eV) remains constant, and the scaling relationship indicates that the two intermediate steps (step 2: OH_{ads} deprotonation; step 3: OOH_{ads} formation) are the RDSs. The overpotential can be expressed as: $\eta_{\text{OER}} = \max[(\Delta G_{\text{Oads}} - \Delta G_{\text{OHads}}), 3.2 \text{ eV} - (\Delta G_{\text{Oads}} - \Delta G_{\text{OHads}})]/e$. According to the Sabatier principle, the oxygen binding strength acts as a key indicator and reflects the volcano-like activity in the η_{OER} vs. $(\Delta G_{\text{Oads}} - \Delta G_{\text{OHads}})$ plot, as shown in Fig. 3(c) [25]. A minimum overpotential of approximately 0.37 V can be predicted for the OER at a current density of 10 mA·cm⁻². Breaking the energy relationship between OH_{ads} and OOH_{ads} is the key to obtaining a lower overpotential and improving activities. Based on this, many effective strategies, such as doping heteroatoms, introducing vacancies, strain engineering, and interface construction [26, 27], have been successfully developed. For example, Sun et al. synthesized Ni(OH)₂ nanoparticles (NPs) using

a reflux method; Ir NPs were anchored to the Ni(OH)₂ NPs (Ir/Ni(OH)₂) with rich interfaces aided by polyol reduction [28]. The results showed that the overpotential of Ir/Ni(OH)₂ at a current density of 100 mA·cm⁻² was 270 mV, which was lower than those of Ni(OH)₂ (437 mV) and Ir NPs (343 mV). Moreover, an overpotential of 224 mV was required to deliver a current density of 10 mA·cm⁻², which was significantly lower than the minimum theoretical overpotential of the OER under the AEM pathway, as discussed above. Further investigation showed that the synergistic effect at the interface between the locally reconstructed IrO_x and NiOOH sites breaks the scaling relationship between OH_{ads} and OOH_{ads} , resulting in a remarkable enhancement in the OER kinetics. Furthermore, strong electronic interactions between the Ir species and Ni(OH)₂ and the Ir–O–Ni bridge bonds stabilized the metastable Ir(V) species and accelerated the formation of O–O bonds during oxygen evolution, further increasing the OER activity.

Recently, the LOM originating from lattice oxygen redox chemistry has served as an alternative route to rationalize the reconstruction of active species. Herein, oxygen ligands are electrochemically activated and released from the lattice matrix of the electrocatalysts, resulting in a high intrinsic activity. To date, several LOM-based OER pathways with different active centers have been proposed. Specifically, for a single active center, the activated lattice oxygen on the electrocatalyst surface can directly serve as the reactive site (known as the oxygen-vacancy-site mechanism (OVSM), shown in Fig. 3(d)) and receive OH^- through nucleophilic attack to form $\text{M-O}_{\text{lat}}\text{-OH}_{\text{ads}}$ species. Subsequently, the produced O_2 is desorbed from the electrocatalyst surface, and thus, an oxygen vacancy is generated and subsequently refilled by OH^- . Alternatively, surface reconstruction of the electrocatalysts also allows direct coupling of the activated lattice oxygen and *O intermediates, in a process known as the single-metal-site mechanism (SMSM), which is energetically conducive to the formation of high-valence metal cations (Fig. 3(e)). Moreover, a dual-metal-site mechanism (DMSM) has been proposed, wherein adjacent activated lattice oxygen atoms directly couple to form an $\text{M-O}_{\text{lat}}\text{-O}_{\text{lat}}\text{-M}$ motif, and the produced O_2 can be released from the $\text{O}_{\text{lat}}\text{-O}_{\text{lat}}$ moiety which acts as peroxy-like species (Fig. 3(f)). The DMSM was reported in metal (oxy)hydroxides because of their structural flexibility. Generally, electrocatalysts following the LOM can easily overcome the thermodynamic relationship limitations in the case of AEMs and provide a higher oxygen evolution activity. However, the participation of activated lattice oxygen can also induce the formation of oxygen vacancies on the electrocatalyst surface, and the resulting unsaturated metal sites tend to dissolve in the electrolyte, thus reducing the stability of the electrocatalyst [29, 30]. In other words, the high activity of electrocatalysts through the LOM is achieved at the expense of the stability, which is not

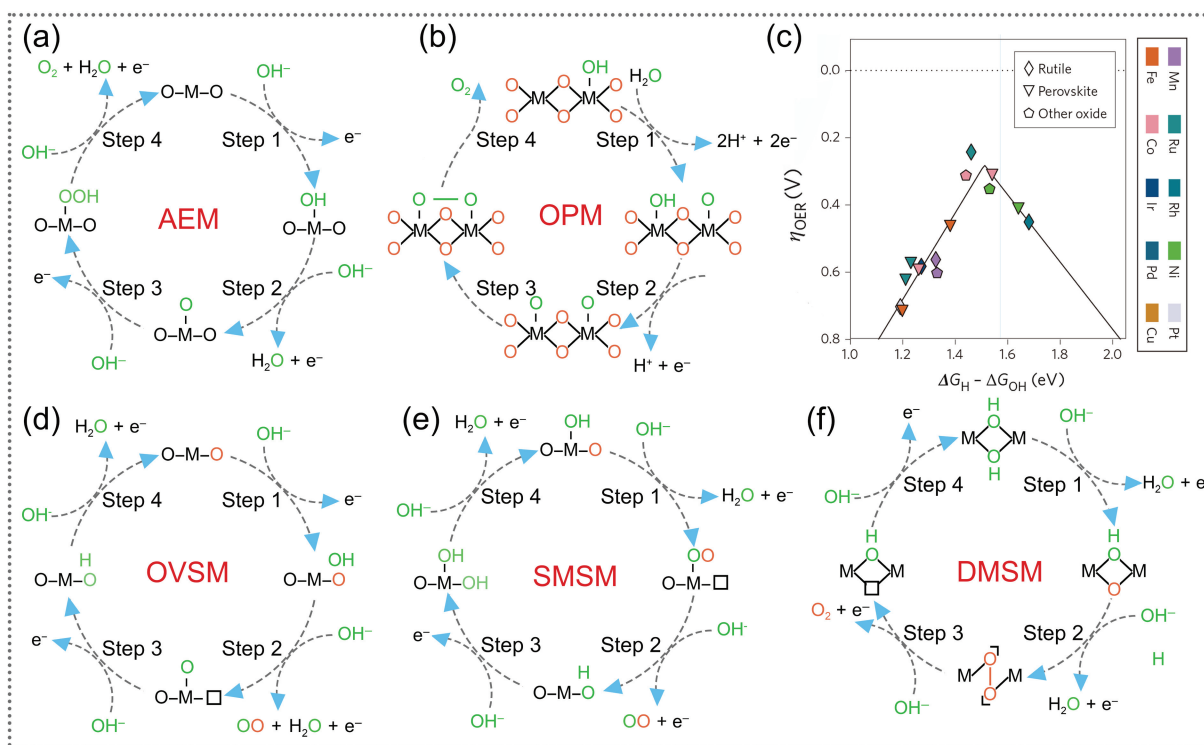


Figure 3 Reported OER mechanisms: (a) AEM, (b) OPM, and ((d)–(f)) LOM. (c) The volcano plot for various TMOs against the oxygen binding strength ($\Delta G_{\text{O}_{\text{ads}}} - \Delta G_{\text{OH}_{\text{ads}}}$). Reproduced with permission from Ref. [25], © Nature Publishing Group, a division of Macmillan Publishers Limited, 2016.

beneficial for the development of electrocatalysts, particularly in acidic media. Ideally, heterogeneous electrocatalysts that operate through the OPM can effectively avoid this problem. Unlike the AEM and LOM, the OPM is more likely to enable the design of efficient OER electrocatalysts, where direct $\text{O}_{\text{ads}}-\text{O}_{\text{ads}}$ radical coupling can be achieved without oxygen vacancies or additional reaction intermediates (e.g., OOH_{ads}) and only O_{ads} and OH_{ads} act as intermediates during the OER, as shown in Fig. 3(b). Specifically, two OH_{ads} groups are generated on two adjacent metal active centers via proton transfer from H_2O molecules in acidic media or the direct adsorption of OH^- from alkaline media. The two neighboring OH_{ads} are converted into two O_{ads} through two subsequent proton-coupled electron transfer steps, and $\text{O}_{2\text{ads}}$ can be formed through a direct $\text{O}_{\text{ads}}-\text{O}_{\text{ads}}$ radical coupling. Finally, $\text{O}_{2\text{ads}}$ is desorbed from the surface. Under ideal conditions, polyphase electrocatalysts based on the OPM exhibit excellent activity and good stability. However, there are many stringent requirements for designing the geometry of a metal site that satisfies the OPM. Generally, symmetric dual-metal sites with appropriate atomic distances can accelerate the coupling of $\text{O}_{\text{ads}}-\text{O}_{\text{ads}}$ radicals with a low energy barrier. Hu et al. described a unique iron-nickel oxide electrocatalyst, whose turnover frequencies (TOFs) under alkaline conditions were significantly higher than those of commercial NiFeO_x electrocatalysts [31]. Theoretical calculations showed that compared with single-phase $\gamma\text{-NiOOH}$ and $\gamma\text{-FeOOH}$, the iron-nickel oxide electrocatalyst had unique Fe and Ni dual active sites, where oxygen molecules could be directly formed through the nucleophilic attack of OH^- , thus greatly reducing the overpotential of the OER. Additionally, similar OER electrocatalysts in acidic media have been reported. Lee et al. reported an $\alpha\text{-MnO}_2$ nanofiber-supported Ru composite ($12\text{Ru}/\text{MnO}_2$) that met the OPM design rules [32]. Small and regularly arranged chains of Ru atoms were generated, with Ru atoms replacing the surface Mn atoms by a cation exchange method (Fig. 4(a)). The results indicated that the Ru nanoarray in $12\text{Ru}/\text{MnO}_2$ was composed of symmetrical Ru sites and the interatomic Ru–Ru distance (2.9 Å) was shorter than that in RuO_2

(3.1 Å), which is conducive to the $\text{O}_{\text{ads}}-\text{O}_{\text{ads}}$ radical coupling. The *operando* synchrotron fourier transform infrared (FTIR) spectrum of $12\text{Ru}/\text{MnO}_2$ shows that some enhancing absorption peaks assigned to O–O and metal–O–O bonds can be observed as the voltage increases, suggesting the forming oxygen bridges between metal sites in the OPM-type OER. In addition, the ^{32}O and ^{36}O signals were also detected by *operando* differential electrochemical mass spectrometry (DEMS) in the H_2^{18}O and H_2^{16}O electrolytes using $12\text{Ru}/\text{MnO}_2$ as the electrocatalysts for OER, respectively, which further indicated that the OER in $12\text{Ru}/\text{MnO}_2$ typically follows the OPM pathway. As a result, $12\text{Ru}/\text{MnO}_2$ with a Ru loading of 11.6 wt.% required a low overpotential of only 161 mV to achieve a current density of 10 $\text{mA}\cdot\text{cm}^{-2}$. Ru/MnO_2 also exhibited excellent long-term durability for more than 200 h and is one of the best acid-stable OER electrocatalysts reported to date.

The OER mechanism of an electrocatalyst is closely related to the intrinsic characteristics of active sites with a particular electronic structure. Generally, electrocatalytic oxygen evolution for metal oxides with good crystallinity and few defects tends to occur either through the AEM pathway with the generation of key OOH_{ads} intermediates at a single metal site or through the OPM pathway with $\text{O}_{\text{ads}}-\text{O}_{\text{ads}}$ directly coupling at two adjacent metal sites. However, the LOM pathway tends to occur in amorphous metal oxides with abundant oxygen vacancies and perovskites with high metal–oxygen ligand covalency. For real applications of electrocatalysts, the determination of the OER pathway has great importance for decoupling the intrinsic activities of the electrocatalysts and optimizing the active sites toward better electrocatalytic performance. Various *in situ* characterization techniques, including *in situ* differential electrochemical mass spectrometry, FTIR spectroscopy, and Raman spectroscopy, and theoretical calculations have been used to capture and analyze the reaction intermediates or origin of the generated gas to further determine the oxygen evolution pathway [33–35]. Through a well-designed experiment in which Ir single atoms were selectively loaded onto the surface or into the lattice of CoOOH to obtain two kinds of structures ($\text{Ir}_1/\text{CoOOH}_{\text{surf}}$ and $\text{Ir}_1/\text{CoOOH}_{\text{lat}}$,

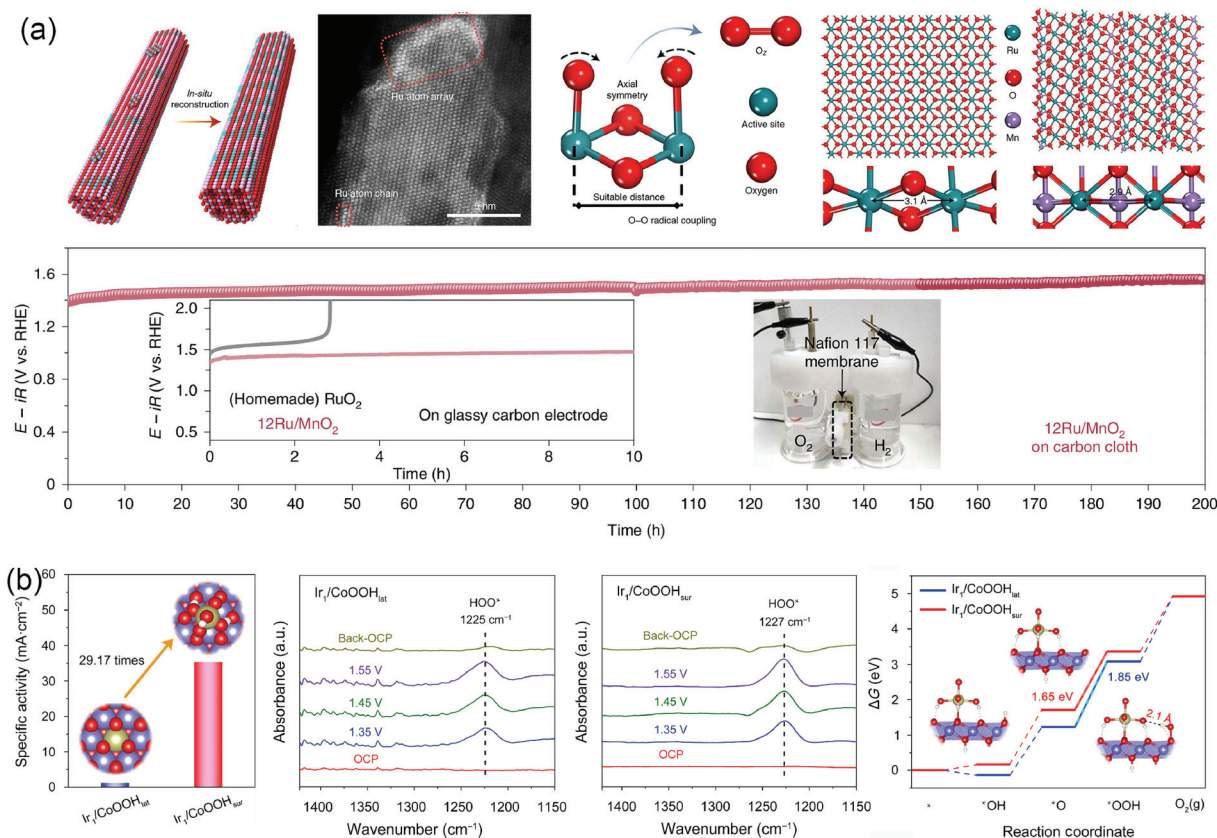


Figure 4 (a) Ru atom array on α -MnO₂ with enhanced performance for acidic water oxidation, and their structure and performance. Reproduced with permission from Ref. [32], © Lin, C. et al. 2021. (b) Iridium single atoms at different locations of CoOOH and their influences on oxygen evolution reaction: *in situ* FTIR spectroscopy and theoretical calculations. Reproduced with permission from Ref. [36], © American Chemical Society 2022.

respectively) with different OER activities [36], Zeng et al. thoroughly studied the influence of the location of the Ir active site on the OER mechanism using *in situ* FTIR spectroscopy combined with theoretical studies. As shown in Fig. 4(b), when the applied potential is gradually increased from the open-circuit potential to 1.35 V, notable peaks at approximately 1227 and 1225 cm⁻¹, representing O–O vibrations, are detected for Ir₁/CoOOH_{sur} and Ir₁/CoOOH_{lat}, respectively. Additionally, the peaks disappear after the potential returns to the open-circuit potential, indicating that these peaks can be attributed to the emergence of OOH_{ads} intermediates during oxygen evolution. This validates that the oxygen evolution pathway of Ir₁/CoOOH_{sur} and Ir₁/CoOOH_{lat} follows a four-step AEM reaction, excluding the lattice oxygen-mediated O_{lat}–O_{lat} coupling pathway. Furthermore, the possible O_{ads}–O_{ads} coupling via the OPM was explored by DFT analysis at these two interfaces, and the resultant high theoretical overpotential suggested that the AEM pathway was more thermodynamically favored than the OPM. Finally, free energy diagrams of the AEM pathway for Ir₁/CoOOH_{sur} and Ir₁/CoOOH_{lat} were calculated for comparison. The energy barrier of the RDS (O_{ads} to OOH_{ads} transition) decreased from 2.09 (CoOOH) to 1.85 (Ir₁/CoOOH_{lat}) and to 1.65 eV (Ir₁/CoOOH_{sur}), indicating that the Ir atoms distributed on the surface of CoOOH could more effectively regulate the adsorption behaviors of intermediates than those in the lattice, which aptly explains the origin of the high OER activity for Ir₁/CoOOH_{sur}.

3 Strategies for optimizing electrocatalyst

Given the high energy barriers in water splitting, especially for the OER, numerous highly efficient electrocatalysts have been developed to accelerate water splitting and reduce energy consumption. Among them, noble metals (platinum, iridium, and

ruthenium) and their oxides (iridium dioxide and ruthenium dioxide) are considered to be the best electrocatalysts for the HER and OER [37–39], respectively. However, various problems need to be resolved prior to practically applying these electrocatalysts. For example, pure noble metal-related electrocatalysts are prone to inactivation because of the inevitable agglomeration of active components during the reactions and the noble metal oxides tend to dissolve into the electrolyte because of excessive oxidation during oxygen evolution, especially in acidic media, thus reducing their electrocatalytic activities and stability [40, 41]. To resolve the aforementioned challenges faced by noble metal electrocatalysts and simultaneously enhance the activity of other non-noble metal electrocatalysts, some effective strategies have been proposed and successfully employed in the design and synthesis of electrocatalysts. Few representative electrocatalysts and their electrocatalytic performance for water splitting are listed in Tables S1 and S2 in the Electronic Supplementary Material (ESM).

3.1 Alloying

An alloy is typically composed of two or more metals. Compared with single metals, different metal atoms within an alloy usually exhibit strong interactions or synergies, which often result in better electrocatalytic performance [42–44]. Yao et al. synthesized a PtRu alloy with Pt and Ru dual sites using a facile laser irradiation strategy in a liquid [45], as shown in Fig. 5(a). The *in situ* X-ray absorption fine structure results show that the alloying degree of PtRu increases during electrocatalytic hydrogen evolution and that the alloying of Pt and Ru remarkably regulates the hydrogen adsorption energy at the Pt site. Benefiting from these, the adsorption of hydroxyl on the Ru site is significantly improved, and the energy barrier for water dissociation is correspondingly reduced. Therefore, PtRu alloy NPs show excellent HER activities and stability in all pH ranges. Alloying

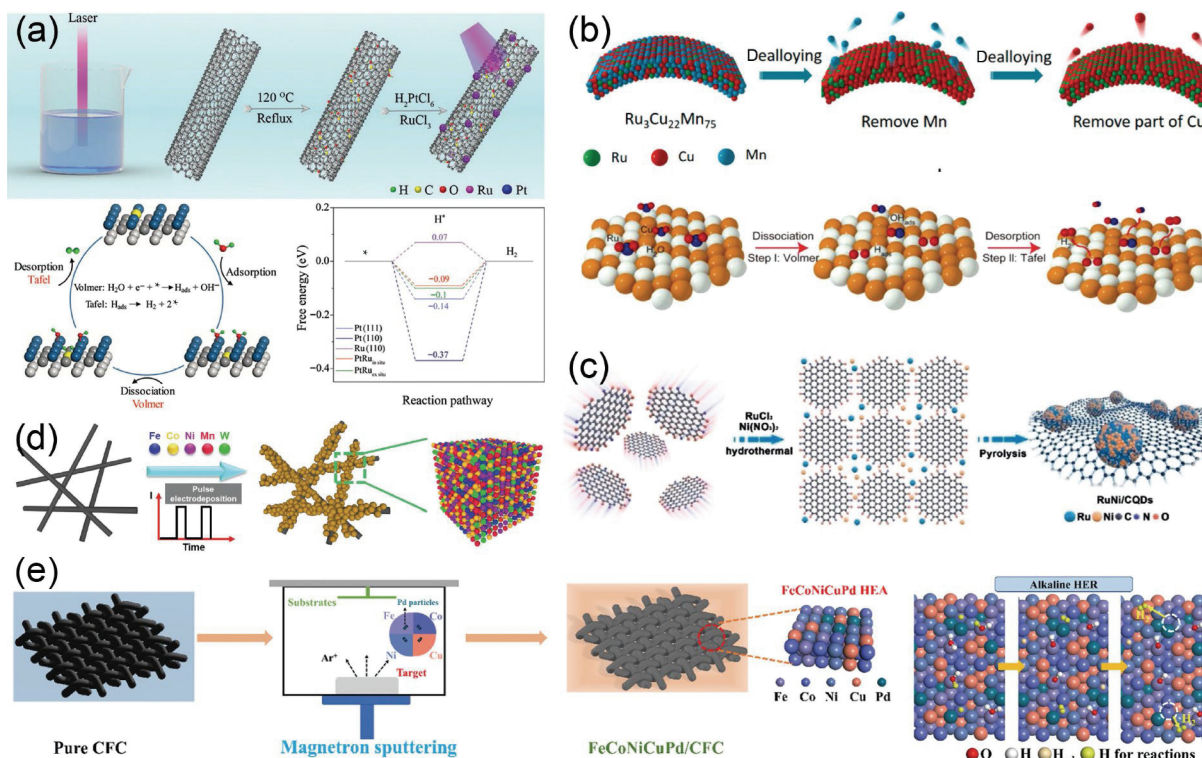


Figure 5 (a) The schematic illustration of the manufacture of the PtRu NPs on CNTs electrocatalyst and their alkaline HER mechanism. Reproduced with permission from Ref. [45], © The Royal Society of Chemistry 2022. (b) The schematic illustration of the preparation of the porous CuRu alloys and their alkaline HER mechanism. Reproduced with permission from Ref. [48], © American Chemical Society 2020. (c) The schematic illustration of the synthesis of the RuNi/CQDs electrocatalyst. Reproduced with permission from Ref. [51], © Wiley-VCH Verlag GmbH & Co. KGaA, Weinheim 2020. (d) Schematic illustration for fabrication of FeCoNiMnW. Reproduced with permission from Ref. [54], © Elsevier B.V. 2022. (e) Schematic illustration of key FeCoNiCuPd thin film fabrication steps and the alkaline HER process. Reproduced with permission from Ref. [55], © Elsevier B.V. 2022.

between non-noble and noble metals is also a good approach to enhance the activity of non-noble metal electrocatalysts and simultaneously reduce the cost of noble metals [46, 47]. For example, copper is a poor electrocatalyst for the HER because of its low hydrogen-bonding energy. In contrast, ruthenium is the most widely used electrocatalyst for water splitting in alkaline media, but the interactions between Ru and H atoms are too strong, which will make it hard to desorb hydrogen. To overcome this, Tan et al. reported a three-dimensional porous CuRu alloy using a dealloying method [48], as shown in Fig. 5(b). The results showed that introducing Ru atoms into the Cu matrix can optimize the electronic interaction of Ru–H and accelerate the subsequent adsorption/desorption of intermediates during water splitting, which can significantly improve the HER efficiency in neutral and alkaline electrolytes. Pan et al. synthesized IrCo nanoalloys (IrCo NPs) with a low Ir content of 6.7 wt.% by directly calcinating binary metal-organic framework materials [49]. When tested in 0.5 M H_2SO_4 , the IrCo NPs showed an outstanding catalytic activity for the HER and OER with only 23.9 and 270.0 mV overpotentials, respectively, to achieve a current density of $10 \text{ mA}\cdot\text{cm}^{-2}$. Alessandri et al. synthesized a novel AuFe nanoalloy by laser ablation synthesis [50]. Owing to the highly loaded Fe atom (up to 11 at.%) in the Au lattice, the AuFe nanoalloy exhibited a high OER activity with an increase of up to 20 times the current density than the counterpart of pure metal NPs when measured in an alkaline aqueous solution. Lu et al. obtained a series of Ru-based bimetallic ultrafine nanocrystals (Ru–M, where M represents Ni, Mn, or Cu) using carbon quantum dots (CQDs) as the substrate to uniformly disperse metal ions and effectively induce the nucleation and growth of nanocrystals during pyrolysis (Fig. 5(c)) [51]. All these nanocrystals exhibit excellent HER activities much higher than that of the corresponding pure Ru nanocrystals at all pH levels.

Specifically, the overpotentials of the as-designed RuNi/CQDs in 1.0 M KOH, 0.5 M H_2SO_4 , and 1.0 M PBS were 13, 58, and 18 mV at a current density of $10 \text{ mA}\cdot\text{cm}^{-2}$, respectively, when the loading of Ru was $5.93 \text{ mg}_{\text{Ru}}\cdot\text{cm}^{-2}$. Theoretical studies demonstrated that the Ni atoms in the RuNi alloy can moderately modify the electronic environment of the adjacent Ru site to obtain an optimized hydrogen binding energy, thus improving the HER activity.

In recent years, high-entropy alloys (HEAs), a new type of nanoalloy material, have attracted wide interests in electrocatalysis because of their flexible compositions and high tolerance, especially in harsh testing environments. The selection of multiple components in HEAs can maximize the interatomic synergies and effectively optimize the electronic structure of alloys, thus decreasing the energy barrier of water electrolysis with enhanced activities [52, 53]. Lu et al. developed a pulsed electrodeposition method for the *in situ* growth of a high-entropy FeCoNiMnW alloy on the surface of carbon paper (Fig. 5(d)) [54]. Because of the unique advantage of atomic-scale synergies among multiple components in HEAs, FeCoNiMnW exhibited an ultra-low overpotential of 15 mV for the HER and an overpotential of 512 mV for the OER in 0.5 M H_2SO_4 at $10 \text{ mA}\cdot\text{cm}^{-2}$. Furthermore, the assembled FeCoNiMnW//FeCoNiMnW couple can achieve the current density of $10 \text{ mA}\cdot\text{cm}^{-2}$ at a cell voltage of 1.76 V for overall water splitting and good operation stability for up to 6 d. Fang et al. prepared thin films of a high-entropy FeCoNiCuPd alloy with good crystallization by magnetron sputtering, as shown in Fig. 5(e) [55]. The FeCoNiCuPd film exhibited excellent activity for both the HER and OER. The electrolyzer fabricated using the FeCoNiCuPd films had a low cell voltage of 1.52 V to deliver the current density of $10 \text{ mA}\cdot\text{cm}^{-2}$ in 1.0 M KOH and maintained notable durability for more than 100 h at a high current of $800 \text{ mA}\cdot\text{cm}^{-2}$, achieving efficient and stable alkaline water splitting.

The rich active sites and high-entropy cocktail effects in the FeCoNiCuPd film are conducive to accelerating the alkaline HER process. *In situ* structural characterization and electrochemical tests showed that the HEA surface was reconstructed during the OER process, and the formed (FeCoNi)OOH species further lowered the energy barrier of the RDS ($O_{ads} \rightarrow OOH_{ads}$), thereby enhancing the OER performance.

3.2 Morphology engineering

The microstructure of a material has a significant influence on its physicochemical properties. Generally, the construction of highly open structures is helpful for enhancing the electrocatalytic performance [56]. A porous structure can not only promote the penetration and diffusion of the electrolyte into the electrocatalysts, thus ensuring that the reactants can adequately contact the active sites inside the bulk, but also provide more paths for charge and mass transport (including gas diffusion) during water splitting [57, 58]. In the last decade, material synthesis technologies have developed rapidly, and many effective strategies have been reported for regulating the microstructures of electrocatalysts. Among the various synthesis methods, there are two categories: directly regulating the active components and introducing foreign carriers with large surface areas and porous structures. The former is usually based on template-assisted methods, nanoarray construction, and surface etching methods during the synthesis process [59–61]. Qiao et al. directly grew $CoHPO_4$ nanoarrays on a highly conductive nickel foam through a low-temperature hydrothermal reaction, which further served as a sacrificial template in the subsequent potentiostat treatment for fabricating porous Co_3O_4 microtubule arrays (denoted as Co_3O_4 -MTA), as shown in Fig. 6(a) [62]. The resultant unique structures enhanced mass and electron mobilities during hydrogen and

oxygen evolution, achieving a high activity and stability. Furthermore, the as-constructed Co_3O_4 -MTA materials were applied for overall water splitting as bifunctional electrodes and exhibited excellent performance comparable to that of commercial Pt/C and IrO_2/C electrodes. Yu et al. developed an ultrafast electrodeposition method to prepare a series of transition metal foams with three-dimensional porous structures for efficient water electrolysis (Fig. 6(b)) [63]. Benefiting from the *in situ* hydrogen bubble template generated during electrodeposition, the as-prepared metal foams had a unique three-dimensional channel for electrolyte diffusion and gas release and also exhibited high conductivity for facilitating fast interfacial charge transfer. Consequently, porous NiCo and NiFe foams can achieve good stability and outstanding activities for the HER and OER with only 86 and 206 mV, respectively, to deliver the current density of 10 $mA \cdot cm^{-2}$ in 1.0 M KOH. In addition, the assembled NiCo foam (-)/NiFe foam (+) electrolyzer only required 1.52 and 1.64 V to achieve the current density of 10 and 100 $mA \cdot cm^{-2}$, respectively, significantly surpassing the Pt/C(-)//RuO₂(+)-based electrolyzer. Moreover, introducing appropriate carriers can also effectively regulate the microstructure or particle size of the electrode material to expose more reactive sites. Simultaneously, the rich porosity, good hydrophilicity, and large specific surface area of carriers can increase the accessibility of metal sites and improve the intrinsic activity of electrocatalysts [64, 65]. Lu et al. synthesized hydrophilic nitrogen-doped carbon quantum dots (NGQDs) using biomass ginkgo leaves as a source (Fig. 6(c)) [66]. Owing to their large surface areas and strong coordination with metal ions, NGQDs can effectively induce nucleation of Ru nanocrystals and further restrict their thermal agglomeration during pyrolysis. Ru nanocrystals (Ru@NGQDs) with an average size of 3.28 nm were thus obtained under optimal conditions. The

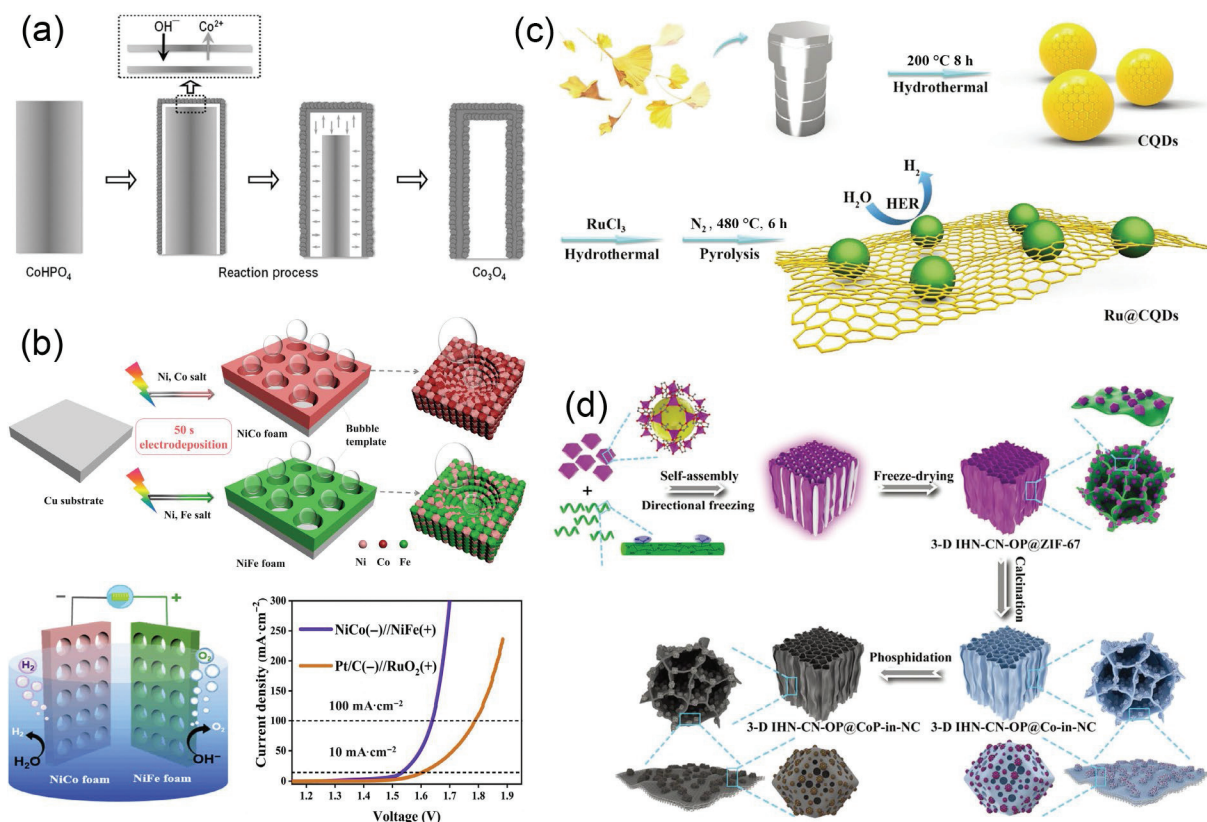


Figure 6 (a) Schematic illustration of the formation of Co_3O_4 -MTA. Reproduced with permission from Ref. [62], © Wiley-VCH Verlag GmbH & Co. KGaA, Weinheim 2017. (b) Schematic diagram of the synthetic route of NiCo and NiFe foams and overall water splitting for two electrolyzers (NiCo foam and NiFe foam function as cathode and anode separately) in 1.0 M KOH. Reproduced with permission from Ref. [63], © Elsevier B.V. 2021. (c) The schematic illustration of the synthesis of the Ru@NCQDs electrocatalyst. Reproduced with permission from Ref. [66], © WILEY-VCH Verlag GmbH & Co. KGaA, Weinheim 2018. (d) The schematic illustration of the synthesis process of 3D IHN-CN-OP@CoP-in-NC. Reproduced with permission from Ref. [67], © Wiley-VCH Verlag GmbH 2021.

small size of Ru nanocrystals combined with the good hydrophilicity of NGQDs endowed the as-designed Ru@NGQDs with excellent electrocatalytic performance for the HER in 1.0 M KOH: The onset overpotential was 0 mV and an overpotential of only 10 mV was required to deliver the current density of 10 mA·cm⁻² with good durability. In addition, Li et al. synthesized a three-dimensional carbon nanofiber-supported zeolite imidazole framework (ZIF-67) aerogel and converted it to three-dimensional CoP NP-embedded hierarchical N-doped carbon nanosheets (3D IHN-CN-OP@CoP-in-NC) via calcination and phosphatization (Fig. 6(d)) [67]. Electrochemical tests showed that 3D IHN-CN-OP@CoP-in-NC exhibited an outstanding HER activity at high current densities in both acidic and alkaline solutions, and the current remained stable after 2000 CV cycles or even after 20 h of chronoamperometric measurements, surpassing that of the commercial Pt/C electrocatalysts. The study showed that the high catalytic activity and fast reaction kinetics of 3D IHN-CN-OP@CoP-in-NC under high current densities were because of the three-dimensional interconnected carbon nanostructures in the bulk, which are conducive to fast electron transport, electrolyte penetration, and rapid release of hydrogen molecules within the electrocatalyst. Moreover, the highly dispersed CoP NPs embedded in the three-dimensional framework provided abundant electrochemically accessible active sites for the HER.

3.3 Interface engineering

Recently, rational design and development of abundant heterogeneous electrocatalysts have received extensive attention owing to their unique interfacial structures, where redistribution of electrons occurred, forming new chemical bonds and generating abundant active sites [68, 69]. Various types of electronic interactions at the interface have been proposed, such as the Mott–Schottky effect, electronic metal-support interactions (EMSI), and support-stabilizing effect, among which, the Mott–Schottky effect is the most common. Owing to the thermal equilibrium and generation of two oppositely charged regions, the electronic density at the interface can be selectively changed, and thus the adsorption of reaction intermediates can be modified, ultimately accelerating the reaction kinetics of water splitting [70, 71]. Regulating d-orbital electrons of the active center based on the EMSI effect between the metal sites and carriers has been demonstrated to be effective for the further improvement of both the activity and durability of electrocatalysts [72]. Wang et al. constructed Pt-RuO₂@KB heterostructures with strong metal-carrier interactions [73]. Pt acted as a stable electron donor at the interfaces, and the electron transfer from the Pt to RuO₂ further gave rise to a stable charge redistribution around the interfaces, thus significantly reducing the energy barriers of water splitting. Moreover, owing to the strong electronic effects on stabilizing the Ru–O bond, oxygen vacancies in the bulk were significantly relieved because of the improved formation energy, which effectively inhibited the dissolution of active metal atoms into the acidic medium. Hence, Pt-RuO₂@KB exhibited an ultrahigh intrinsic activity and outstanding stability for the OER in 0.1 M HClO₄ electrolytes. Additionally, the reasonable construction of two-phase interfaces with two different active sites but complementary functions is an effective strategy to obtain high electrocatalytic performance based on intimate synergistic effects [74]. Wang et al. doped noble metal Pt into a Ni₄Mo alloy (denoted as Pt-Ni₄Mo) using a solvent-free microwave reduction method [75]. Electrochemical tests showed that the carbon nanotube (CNT)-supported Pt-Ni₄Mo exhibited excellent HER activity in alkaline electrolytes. Furthermore, *in situ* infrared absorption spectroscopy tests and DFT analysis showed that strong synergistic effects existed between the Pt and Ni₄Mo

interfaces during hydrogen evolution and the Ni₄Mo site primarily dissociated water to produce H_{ads}, which was then transferred to the Pt active site, accelerating the H₂ desorption process. Shao et al. constructed a (Ni, Fe)₂@MoS₂ heterostructure via hydrothermal synthesis using NiFe-LDH as the precursor. The (Ni, Fe)₂@MoS₂ showed excellent activity, requiring low overpotentials of only 130 and 270 mV for the HER and OER, respectively, to obtain the current density of 10 mA·cm⁻² and good durability for water splitting in an alkaline solution [76]. Moreover, *in situ* Raman spectroscopy further demonstrated that the as-designed interface is advantageous for the formation of stable S–H_{ads} and OH_{ads} intermediates to promote overall water splitting.

An in-depth understanding of the interfacial structure is essential for the optimal design and development of high-performance electrocatalysts. According to the structural features of materials, these interfaces mainly exist in supported, core-shell, and heterogeneous structures [59, 77, 78]. Peng et al. synthesized mesoporous Fe₂O₃/CuO heterostructures fixed on nickel foam using a template-free method [79], as shown in Fig. 7(a). The as-designed Fe₂O₃/CuO electrocatalysts with rich interfaces and Fe–O–Cu bridge bonds promoted the adsorption/desorption of oxygen-containing intermediates by enhancing the oxygen binding energy, which endows them with excellent performance and long-term durability in both the HER and OER. Additionally, single-atom materials are a new type of supported electrocatalyst owing to their nearly 100% atomic utilization and abundant atomic-scale interfaces with intrinsically different coordination structures [80–82]. Wang et al. reported Ni single atoms loaded on porous carbon nanofibers as HER electrocatalysts, where the chemical coordination structure of single-atom Ni was regulated by doping different heteroatoms, such as P, N, B, and S [83]. Studies have shown that introducing heteroatoms can polarize the charge distribution around Ni sites and adjust the electronic states toward the Fermi level, eventually leading to the optimal adsorption/desorption behavior for H/O-containing intermediates during water splitting. Based on the experimental and theoretical calculation results, the Ni atoms coordinated with one P and three N atoms were optimal to achieve outstanding electrocatalytic activity for water splitting. Furthermore, interfaces consisting of different crystalline phases have also been widely reported, providing new insights into the design of heterogeneous electrocatalytic materials [84, 85]. Fei et al. developed a core-shell-configured composite composed of an amorphous/crystalline phase NiFe alloy coated with ultrathin graphene layers (denoted as a/c-NiFe-G) using a microwave thermal-shock method [86], as shown in Fig. 7(b). The amorphous/crystal heterostructure generated abundant active sites, while graphene on the outside acted as a protective layer and electron transport pathway during water electrolysis, both of which resulted in a significantly improved OER performance. Therefore, the overpotential of a/c-NiFe-G at the current density of 10 mA·cm⁻² was 250 mV in 1.0 M KOH, which was much lower than that of the corresponding graphene-layer-encapsulated crystalline phase NiFe alloys. Further, characterization methods such as X-ray absorption fine structure analysis combined with DFT calculations, also indicated that the abundant phase boundaries can effectively adjust electronic structures in the a/c-NiFe-G, leading to suitable adsorption energies for reaction intermediates, thereby enhancing the OER performance.

3.4 Defect engineering

Defects are common or inevitable in as-prepared electrocatalytic materials [87, 88], and various types of defects have been reported, including point (e.g., vacancy defects, heteroatomic substitution,

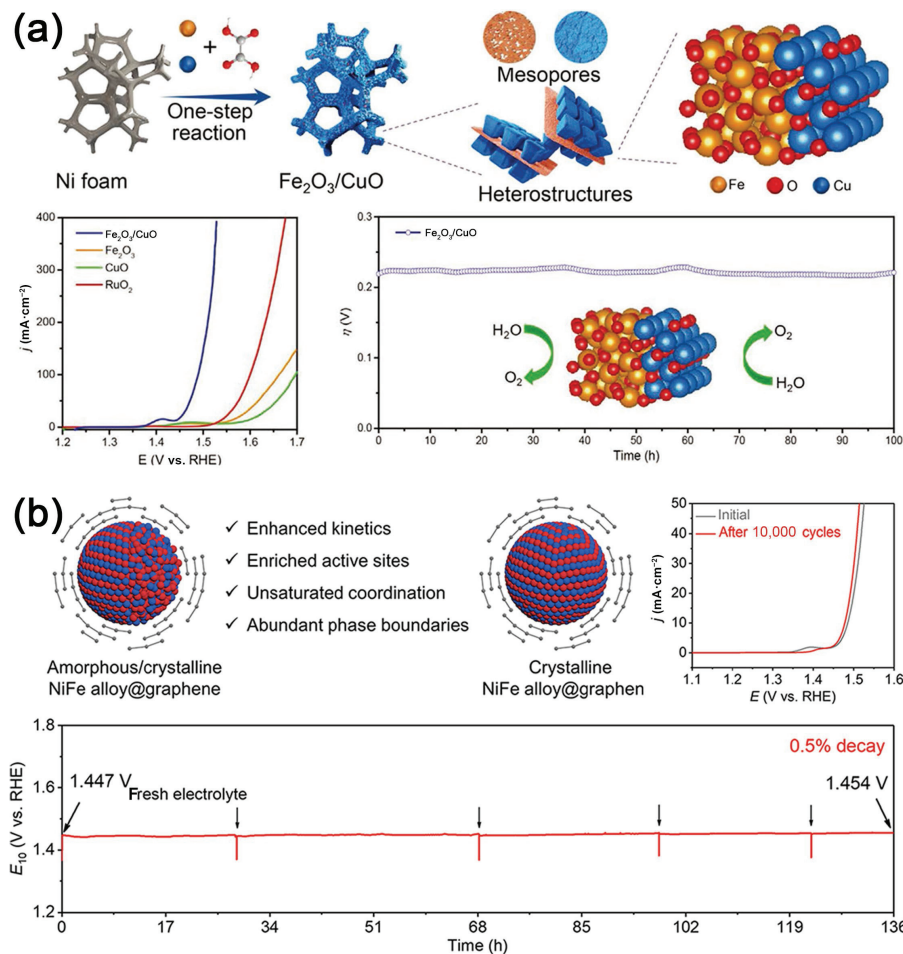


Figure 7 (a). Scheme illustrating the one-step growth of $\text{Fe}_2\text{O}_3/\text{CuO}$ heterostructures on Ni foam and water splitting performance in 1.0 M KOH. Reproduced with permission from Ref. [79], © Wiley-VCH GmbH 2022. (b). The structural and compositional characterization of a/c-NiFe-G and c-NiFe-G and electrochemical OER performance in 1.0 M KOH. Reproduced with permission from Ref. [86], © American Chemical Society 2021.

and interstitial atoms), linear (e.g., edge dislocations), two-dimensional (e.g., crystal boundaries), and bulk defects (e.g., pores and foreign impurities) [89–92]. During electrocatalytic water splitting, the defects act as reactive centers as well as nucleation sites for constructing more active components. For example, the defect sites on the surface of substrates can capture and anchor metal species during material synthesis, thus achieving uniform dispersion and stabilization of metal-isolated sites in electrocatalysts [93]. To obtain high-density accessible active sites on the electrocatalyst surface, Liu et al. applied cobalt-doped ZIF-8-derived porous carbon (denoted as Co_1NC) for adsorbing a platinum precursor (chloroplatinic acid) and subjected it to a mild electrochemical reduction condition to prepare nitrogen-doped porous carbon materials with atomically dispersed Pt and Co (denoted as $\text{Pt}_1/\text{Co}_1\text{NC}$) [94]. As shown in Fig. 8(a), the abundant defects on the surface of Co_1NC play an important role in anchoring and dispersing Pt species toward forming atomically distributed Pt–N/C sites. The results showed that the $\text{Pt}_1/\text{Co}_1\text{NC}$ exhibited excellent HER activity with a low overpotential of 4.15 mV to reach the current density of $10 \text{ mA}\cdot\text{cm}^{-2}$ with good stability for 5000 CV cycles. Xie et al. demonstrated that the introduction of defects into MoS_2 nanosheets (denoted as DR- MoS_2) activated the basal planes and generated abundant additional active sites for the HER. DR- MoS_2 had large surface areas exposing abundant defects, which can also provide abundant reaction sites for the adsorption and anchoring of Pt atoms, leading to uniformly dispersed Pt nanocrystals on DR- MoS_2 (denoted as DR- MoS_2 -Pt) [95]. Benefiting from the strong synergy between the Pt nanocrystals and DR- MoS_2 , the DR- MoS_2 -Pt

electrocatalyst showed excellent HER performance, better than the Pt/C electrocatalyst under both acidic and alkaline conditions. Wang et al. adopted electrochemical exfoliation to prepare double-transition-metal MXene nanosheets with abundant surface Mo vacancies (denoted as $\text{Mo}_2\text{TiC}_2\text{T}_x\text{-V}_{\text{Mo}}$) that were utilized to anchor single Pt atoms (designated as $\text{Mo}_2\text{TiC}_2\text{T}_x\text{-Pt}_{\text{SA}}$), as shown in Fig. 8(b) [96]. During the electrochemical exfoliation process, where $\text{Mo}_2\text{TiC}_2\text{T}_x$ deposited on carbon paper acted as a working electrode and Pt foil was adopted as a counter electrode, the Mo vacancies were refilled by Pt single atoms, and the Pt–C covalent bonds were reconstructed on the MXene surface. The as-synthesized $\text{Mo}_2\text{TiC}_2\text{T}_x\text{-Pt}_{\text{SA}}$ electrocatalyst exhibited excellent kinetics for the HER with outstanding overpotentials of 30 and 77 mV for 10 and $100 \text{ mA}\cdot\text{cm}^{-2}$, respectively, and remarkable stability of up to 10,000 cycles or 100 h of testing. Furthermore, the introduction of defects can also greatly regulate the local electronic structure to generate strong synergistic effects and form extra active sites [97–99]. To date, various strategies for defect engineering have been used for the surface modification of electrocatalysts, such as high-temperature reduction, plasma treatment, surface etching, and chemical reduction [100–102]. Because perturbations of the electron/charge distribution at a nanoscale are more pronounced than those at a macroscale, significant impacts of defects at the nanoscale on the physicochemical properties of electrocatalysts can be expected. Dai et al. treated Co_3O_4 nanosheets using an argon-plasma etching strategy; the obtained Co_3O_4 nanosheets possessed a large surface area, where abundant oxygen vacancies and Co^{2+} were generated [103] (Fig. 8(c)). The oxygen vacancies on the surface enhanced

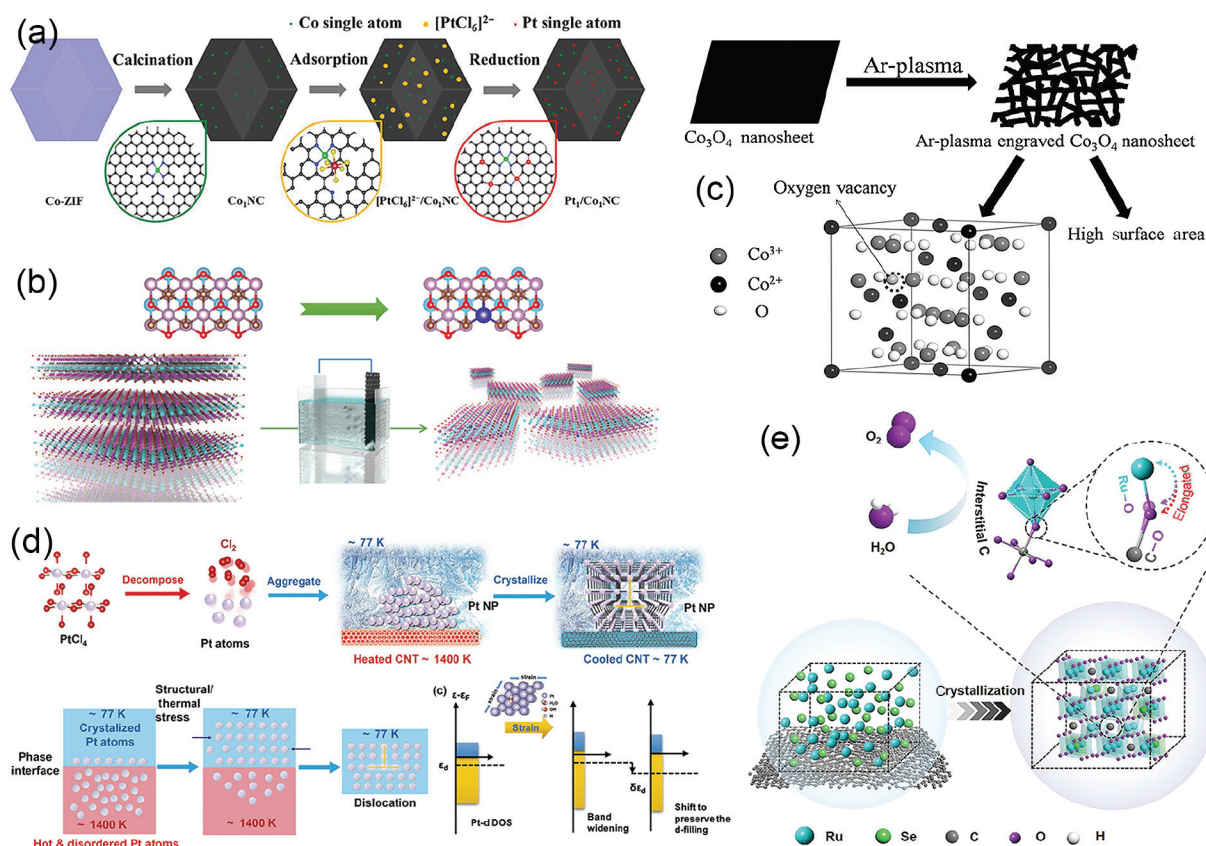


Figure 8 (a) The scheme illustrates the preparation of Pt₁/Co₃NC. Reproduced with permission from Ref. [94], © Elsevier B.V. 2021. (b) The schematic of the electrochemical exfoliation process of MXene with immobilized single Pt atoms. Reproduced with permission from Ref. [96], © Zhang, J. Q. et al. 2018. (c) The illustration of the preparation of the Ar-plasma-engraved Co₃O₄. Reproduced with permission from Ref. [103], © WILEY-VCH Verlag GmbH & Co. KGaA, Weinheim 2016. (d) Schematic diagram of the preparation of dislocation-rich Pt nanoparticles by environmental high-temperature thermal-shock. Reproduced with permission from Ref. [104], © Wiley-VCH GmbH 2021. (e) The formation process of C-RuO₂-RuSe. Reproduced with permission from Ref. [108], © Elsevier Inc. 2022.

the electronic conductivity of Co₃O₄, and highly exposed surface of Co₃O₄ provided more active sites for the OER. Compared with the pristine Co₃O₄, the etched Co₃O₄ showed a higher specific activity and current density at the same voltage. Chen et al. proposed a simple high-temperature thermal-shock strategy, followed by rapid cooling using liquid nitrogen, to introduce abundant dislocations into Pt NPs (denoted as Dr-Pt) [104], as shown in Fig. 8(d). The results showed that these dislocations induced a strong strain effect, which effectively modulated the electronic structure of Dr-Pt and optimized its catalytic HER activity. Dr-Pt possessed a low overpotential of 25 mV at 10 mA·cm⁻² in a 1.0 M KOH electrolyte with good electrocatalytic stability for 20 h of testing. Recently, heteroatom doping has been regarded as a simple and efficient method to introduce defects and enhance the intrinsic activity of electrocatalysts [105–107]. Huang et al. demonstrated that the existence of interstitial carbon in the lattice of ruthenium dioxide (C-RuO₂-RuSe) can enhance its stability and activity for the OER in acidic media. The C-RuO₂-RuSe electrocatalyst was successfully obtained by pyrolyzing a carbon-supported RuSe₂ precursor, in which a large amount of free space was generated for rapid penetration and fixation of carbon atoms in the RuO₂ lattice gaps [108], as shown in Fig. 8(e). Compared with commercial RuO₂, the synthesized C-RuO₂-RuSe showed enhanced OER activity and excellent stability in 0.5 M H₂SO₄. Both experimental and theoretical results showed that the activation of Ru sites by interstitial C can optimize the reaction energy barrier toward enhanced activity and the interstitial C can also weaken the Ru–O bond covalency to effectively inhibit Ru dissolution, thus significantly improving its stability. Lu et al.

reported CD-loaded molybdenum phosphide composites (MoP/NCDS) with different N-doping amounts [109]. In 1.0 M KOH, the MoP/NCDS delivered the current density of 10 mA·cm⁻² under an overpotential of only 70 mV. DFT calculations further revealed that the incorporation of N atoms can function as promoters to reduce the ΔG_{Hads} of the MoP/NCDS; specifically, ΔG_{Hads} was effectively decreased by increasing the N-doping content. By optimizing the N-doping content, the electronic arrangement and charge density around the MoP active sites can be modified to promote the adsorption of HER intermediates.

3.5 Strain engineering

Strain engineering provides a new strategy for optimizing the geometric configuration and electronic structures of electrocatalysts and improving their HER/OER performance [110, 111]. Based on the relative direction of external forces applied to the stress region, the elastic strain can be classified as tensile, compressive, and shear strain. For electrocatalysts, the d-band center of the metal site is important because the electronic interactions originating from the orbital hybridization between the adsorbate and metal center are the key factors influencing the adsorption/desorption behavior. Elastic lattice strain can be used to adjust the Fermi level of the metal site, where small changes in the environment can significantly change the d-band states of the metal and its interactions with the adsorbate states [112, 113]. Hong et al. successfully introduced in-plane strain into noble metal nanosheets by constructing rich amorphous crystalline phase boundaries [114]. As shown in Fig. 9(a), for the

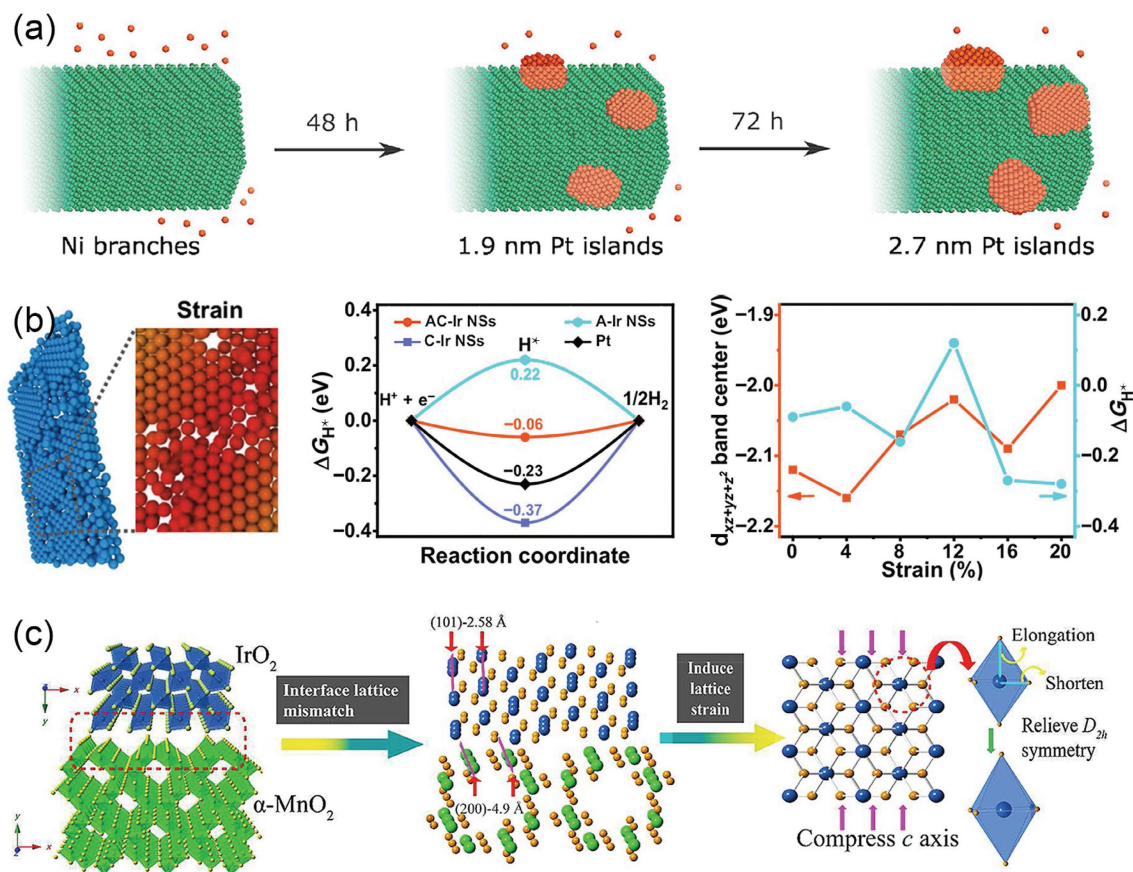


Figure 9 (a) The in-plane strain engineering through constructing amorphous-crystalline phase boundaries and the calculated ΔG_{Hads} and d-band centers on AC-Ir NSs. Reproduced with permission from Ref. [114], © Wu, G. et al. 2022, Ref. [121], © American Chemical Society 2019. (b) The scheme illustration growth of Pt atoms directly on a Ni branch. (c) The diagram process of $\alpha\text{-MnO}_2$ substrate inducing the lattice strain of IrO_2 . Reproduced with permission from Ref. [122], © American Chemical Society 2017.

as-synthesized ultrathin amorphous-crystalline Ir nanosheets (AC-Ir NSs), approximately 4% of the compressive strain was observed owing to the difference in the lattice constants in the local amorphous/crystalline boundary. DFT calculations confirmed that tensile strain can effectively adjust the d-band states of AC-Ir NSs, thereby optimizing the hydrogen adsorption energy and improving the HER performance. The TOF of the AC-Ir NSs was 4.5 times higher than that of the Pt/C electrocatalyst in the reference. Lu et al. reported a cobalt-ruthenium nanoalloy supported by CQDs (CoRu/CQDs) as an electrocatalyst for hydrogen evolution [115]. The introduction of a small amount of Ru (1.33 wt.%) into the cobalt crystal produced tensile strain in the CoRu alloy NPs and further accelerated the electron transport dynamics during the HER. Simultaneously, the introduction of Ru greatly regulated the electronic structure of Co, and thus, the electroactivity and adsorption strength of H_{ads} intermediates reached a perfect harmony, delivering efficient HER catalytic activities. Transition metal sulfides, such as molybdenum disulfide, can also be modified by introducing sulfur vacancies, thus inducing strain on the inert surface to enhance its electrocatalytic hydrogen evolution performance [116, 117]. Therefore, constructing compressive strain can effectively improve the activity of metal-based electrocatalysts by optimizing the adsorption capacity of the catalytic sites for the reaction intermediates. To date, many mechanisms based on lattice mismatch, morphology modulation, and substrate control [118–120] have been reported to introduce lattice strains. Lattice mismatch usually occurs at crystal interfaces with different lattice parameters and its effects on the structures of electrocatalysts have been widely explored. This mismatch can lead to strain at the edges of crystal frameworks, heterostructures, or defect sites, thus

optimizing the adsorption strength of reaction intermediates and, eventually, improving electrocatalytic activities. Tilley et al. reported Pt islands grew directly on Ni NPs via slow reduction synthesis [121], as shown in Fig. 9(b), in which the strain content could be controlled by adjusting the size of the Pt islands. The study demonstrated that strain can affect the H-adsorption/desorption ability, and the strained Ni-branched Pt-island NPs had weaker hydrogen binding energy than that of commercial Pt, which led to faster H_2 desorption and thus, increased TOF during electrocatalysis. Yang et al. reported the construction of lattice strains by growing IrO_2 NPs directly onto the selectively exposed (200) crystal face of $\alpha\text{-MnO}_2$ nanorods via continuous hydrothermal synthesis, as shown in Fig. 9(c) [122]. Interestingly, the OER activity of the prepared hybrid increased when the mass of IrO_2 decreased. Experimental results combined with theoretical calculations demonstrated that lattice mismatches at the underlying interface induced lattice strain in IrO_2 . More importantly, the trend of lattice strain variation of IrO_2 NPs was consistent with their electrocatalytic activities, thus verifying the key role of the lattice strain in determining the OER activity.

4 Summary and outlook

In this paper, recent developments in this field are reviewed in detail, focusing on the mechanisms of water splitting, including hydrogen and oxygen evolution. The development of electrocatalysts is summarized, and some representative strategies for the design and synthesis of electrocatalysts, such as alloying, morphological engineering, interface construction, defect engineering, and strain engineering, are described. Although great progress has been made in developing efficient water-splitting

electrocatalysts and numerous studies have been reported in recent years, many difficulties remain to be urgently resolved to promote the industrial applications of electrolytic water splitting.

Firstly, the stability of electrocatalysts, especially at high current densities and under rigorous electrolyte environments, is of great significance for the practical application of water-splitting systems. Long-term and stable operation under a high current density is a substantial requirement for electrocatalysts, which requires high structural strength and excellent corrosion resistance. However, most of the stability evaluations of electrocatalysts for water splitting were performed at current densities of less than 1000 mA·cm⁻² and exhibited poor durability, for example, less than one week or a much shorter time, which is not beneficial for the design and development of high-performance electrocatalysts. Additionally, the traditional electrode preparation process of fixing powders on the electrode surface using a binder may not be suitable, because rapid bubble desorption under high currents causes the peeling off of these active components and reduces the hydrogen evolution performance. Therefore, the development of binder-free self-supporting electrocatalysts is necessary.

Secondly, reducing the consumption of electrical energy and manufacturing cost of electrocatalysts is conducive toward reducing the cost of hydrogen production. The reduction in energy consumption in water splitting can be achieved by reducing the overall voltage of electrolyzers under the same current density. Owing to the slow kinetics of the OER, it is important to develop efficient oxygen evolution electrocatalysts and particularly, bifunctional electrocatalysts. However, oxidation reactions at low potentials (such as urea oxidation) have been widely reported to replace the anodic OER, reducing the overall voltage of water splitting. Therefore, substantial advancements are needed for the rational design and preparation of efficient redox media to facilitate the development of water splitting technology.

Finally, elucidating the structure–activity relationship can provide in-depth insights into the design and development of powerful HER/OER electrocatalysts aimed at the commercial application of water electrolyzers. Thus far, *in situ* characterization techniques, such as *in situ* FTIR spectroscopy, Raman spectroscopy, and DEMS, have been applied to reveal the electrocatalytic mechanisms of water splitting. Nevertheless, during actual water electrolysis, the active sites of the electrocatalysts are often reconstructed by the continual adsorption/desorption steps of the key intermediates and the dynamic changes in the interface between the electrocatalyst and the electrolyte, which makes it difficult to track a specific reaction species on the electrocatalyst surface and identify the real active sites. Hence, spectroscopic techniques with atomic resolutions and sensitivity remain in demand. *Operando* X-ray absorption spectroscopy is regarded as an effective tool for identifying the local coordination and electronic structures of the electrocatalysts during water splitting. Theoretical studies that use simplified models to analyze complex reaction processes may not be able to simulate real catalyst structures and chemical reactions. Therefore, a new calculation method involving more accurate models at a (semi-)quantitative level is required to effectively guide the design of electrocatalysts.

Acknowledgements

This work was partly supported by the National Natural Science Foundation of China (Nos. 52202050, 52122308, 21905253, and 51973200), the China Postdoctoral Science Foundation (No. 2022TQ0286), and the Natural Science Foundation of Henan Province (No. 202300410372).

Electronic Supplementary Material: Supplementary material (some representative electrocatalysts and their electrocatalytic performance for water splitting, and Tabs. S1 and S2) is available in the online version of this article at <https://doi.org/10.1007/s12274-023-5502-8>.

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