

# Synergetic effects of gold-doped copper nanowires with low Au content for enhanced electrocatalytic CO<sub>2</sub> reduction to multicarbon products

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## ABSTRACT

As efficient catalysts of electrochemical CO<sub>2</sub> reduction reaction (CO<sub>2</sub>RR) towards multicarbon (C<sub>2+</sub>) products, Cu-based catalysts have faced the challenges of increasing the reactive activity and selectivity. Herein, we decorated the surface of Cu nanowires (Cu NWs) with a small amount of Au nanoparticles (Au NPs) by the homo-nucleation method. When the Au to Cu mass ratio is as little as 0.7 to 99.3, the gold-doped copper nanowires (Cu-Au NWs) could effectively improve the selectivity and activity of CO<sub>2</sub>RR to C<sub>2+</sub> resultants, with the Faradaic efficiency (FE) from 39.7% (Cu NWs) to 65.3%, and the partial current density from 7.0 (Cu NWs) to 12.1 mA/cm<sup>2</sup> under −1.25 V vs. reversible hydrogen electrode (RHE). The enhanced electrocatalytic performance could be attributed to the following three synergetic factors. The addition of Au nanoparticles caused a rougher surface of the catalyst, which allowed for more active sites exposed. Besides, Au sites generated \*CO intermediates spilling over into Cu sites with the calculated efficiency of 87.2%, which are necessary for multicarbon production. Meanwhile, the interphase electron transferred from Cu to Au induced the electron-deficient Cu, which favored the adsorption of \*CO to further generate multicarbon productions. Our results uncovered the morphology, tandem, and electronic effect between Cu NWs and Au NPs facilitated the activity and selectivity of CO<sub>2</sub>RR to multicarbons.

## KEYWORDS

electrochemical CO<sub>2</sub> reduction, multicarbon production, bimetallic catalyst, Cu nanowires, Au nanoparticles

## 1 Introduction

Excessive carbon dioxide accumulation from the consumption of conventional fossil fuels has caused a series of problems like global warming and ocean acidification [1]. The electrochemical CO<sub>2</sub> reduction reaction (CO<sub>2</sub>RR) serves as a promising approach to disrupt the status quo toward the carbon-neutral future [2–4], which could convert CO<sub>2</sub> into value-added fuels such as ethylene [5, 6], ethanol [7, 8], and n-propanol [9, 10] with the advantage of high energy density and broad applications. Although CO<sub>2</sub>RR to multicarbon (C<sub>2+</sub>) products attracts enormous attention, it has been confronted with sluggish thermodynamics and kinetics. On the one hand, linear CO<sub>2</sub> molecules are chemical inert by virtue of the significantly high bonding energy of C=O (750 kJ/mol), making CO<sub>2</sub> hard to be converted into other chemicals containing C–C (336 kJ/mol), C–O (327 kJ/mol), and C–H (411 kJ/mol) bonds. On the other hand, the generation of multicarbons involves multiple electron–proton transfer and C–C coupling, which have to overcome the high energy barrier and high overpotentials [11]. Except for the own difficulty of CO<sub>2</sub>RR, hydrogen evolution reaction (HER) as the competitive reaction occurs unavoidably due to the low solubility for CO<sub>2</sub>, causing the

mass transfer limitation in liquid electrolyte. Hence, highly active electrocatalysts are urgently needed to address the bottleneck of the CO<sub>2</sub>RR to multicarbons.

Among a wide variety of metals, Cu-based catalysts stand out with the uniqueness of both the negative adsorption energy for \*CO and positive adsorption energy for \*H [12], recognized as the most effective catalysts for CO<sub>2</sub>RR to multicarbons. However, Cu electrocatalysts are plagued by problems of insufficient activity and selectivity for CO<sub>2</sub>RR for the reason of chemically inert CO<sub>2</sub> reactants and the similar equilibrium potentials for reducing to different products. To date, several strategies have been engineered to address these issues. For example, one of the efficacious approaches is based on the morphological effect [13–16], since the crystal defects and facets are known to directly influence the CO<sub>2</sub>RR. Gao et al. confirmed that cubooctahedron-shaped Cu favored the production of multicarbons compared with the various shaped Cu catalysts [17]. The enhancement was attributed to Cu (100)/Cu (111) interfaces, which provided a favorable local electronic structure to encourage the rate-limiting C–C coupling step. Among those, Cu nanowires (Cu NWs) feature a high aspect ratio and rich grain boundaries with five-fold twinned structures,

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which are confirmed to facilitate electron transport and offer abundant active sites for CO<sub>2</sub>RR [18, 19].

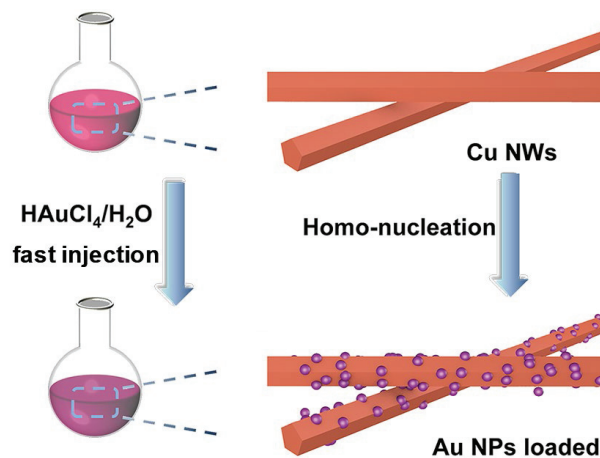
Another remarkable strategy is introducing a secondary metal [20], such as Au [21, 22], Ag [23, 24], and Pd [25], which can boost the surface coverage of \*CO, the necessary intermediate in all pathways of producing multicarbons [26]. Buonsanti et al. developed Ag-Cu nanodimers with an enhanced selectivity of ethylene via the tandem and electronic effects [27]. Compared with Ag, Au presents superior chemical stability and selectively for CO<sub>2</sub> reduction to CO [28, 29] and a larger work function [30], so Au has the potential as a second metal for Cu-based catalysts. Meanwhile, Sargent et al. confirmed that introducing Au in Cu catalysts could suppress HER efficiently relative to Cu [31]. However, several CuAu electrocatalysts were too easy to desorb \*CO which tended to form methane or carbon monoxide instead of multicarbon products [32, 33]. For example, Wang et al. reported an elaborate Au-Cu catalyst composed of an alloyed AuCu shell and a Cu core exhibited a high CO Faradaic efficiency (FE) approaching 94% with 17% Au content [34]. Furthermore, in consideration of the rareness and high expense of Au, it is imperative to investigate how to obtain more multicarbon products by the optimal synergetic effects between Cu and Au at a considerable Au utilization efficiency.

In this context, we reported an advanced CO<sub>2</sub>RR catalyst by doping a small number of gold nanoparticles (Au NPs) onto the surface of Cu NWs via homo-nucleation of the Au(0) species at different Au/Cu ratios. Specifically, Cu<sub>99.3</sub>Au<sub>0.7</sub> nanowires (Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs) exhibited even considerably higher Faradaic efficiency of multicarbon products approaching 65.3% than pure Cu NWs (39.7%) but at a low Au content of 0.7%. Concomitantly, the partial current density increased greatly from 7.0 to 12.1 mA/cm<sup>2</sup>. The augmentation could be owing to the surface roughness, the tandem effect, and the electronic effect. Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs enabled an ~ 87.2% efficient spillover of \*CO intermediates from Au sites, which were further reduced by the electron-deficient Cu sites generated from the charge transfer from Cu to Au. On contrary, limited active Cu sites on Cu NWs favored such competitive productions as hydrogen and formate with lower energy barriers. This work brings a deeper understanding of both the structures and interactions between the various metals for the CO<sub>2</sub>RR catalyst to facilitate breaking through the existing barrier of selectivity and activity.

## 2 Results and discussion

Cu-Au NWs were synthesized by the deposition of Au NPs on Cu NWs (Fig. 1) [35]. Firstly, the Cu NWs were prepared according to the reported literature with small modifications [36]. Subsequently, the HAuCl<sub>4</sub> solution was rapidly introduced into the Cu NWs solution with the presence of ascorbic acid (AA). The gold precursor was instantaneously reduced, leading to a rapidly increased concentration of Au(0) species around Cu NWs, followed by further initiated nucleation and crystal growth on their own. The catalysts of different Cu to Au mass ratios were obtained by changing the amounts of HAuCl<sub>4</sub> solution, denoted as Cu<sub>99.8</sub>Au<sub>0.2</sub>, Cu<sub>99.3</sub>Au<sub>0.7</sub>, and Cu<sub>96.7</sub>Au<sub>3.3</sub> NWs.

The morphology and size of Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs as the representative were characterized by scanning electron microscopy (SEM) and transmission electron microscopy (TEM) (Fig. 1). The Cu NWs possessed a linear morphology with a flat and clean surface (Fig. S1 in the Electronic Supplementary Material (ESM)). A band-like contrast on the NWs was induced by bending or twisting [37]. After surface modification of Au NPs, the nanowires showed a uniformly linear morphology consistent with Cu NWs (Fig. 2(a)). As shown in Figs. 2(b) and 2(c), the Au nanoparticles

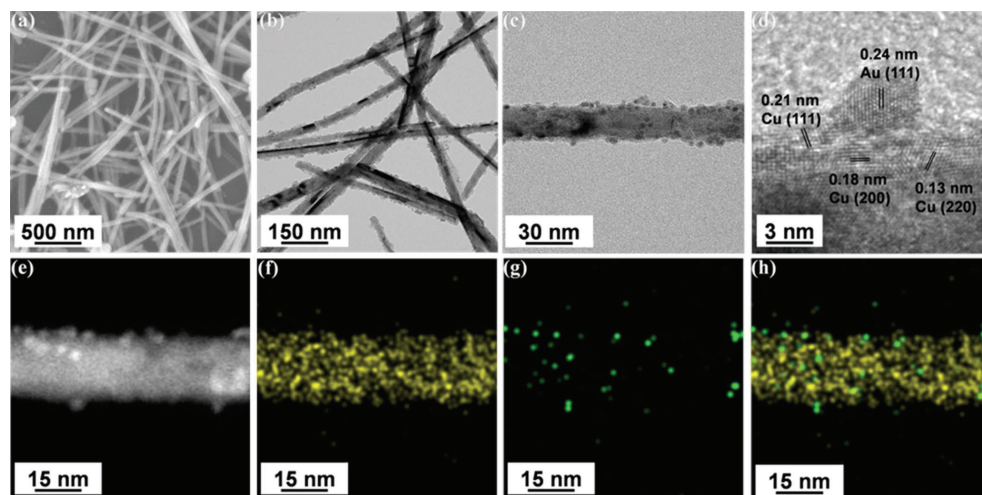


**Figure 1** Schematic illustration of the preparation of Cu-Au NWs by decorating Au NPs on the surfaces via the homo-nucleation of Au(0) species.

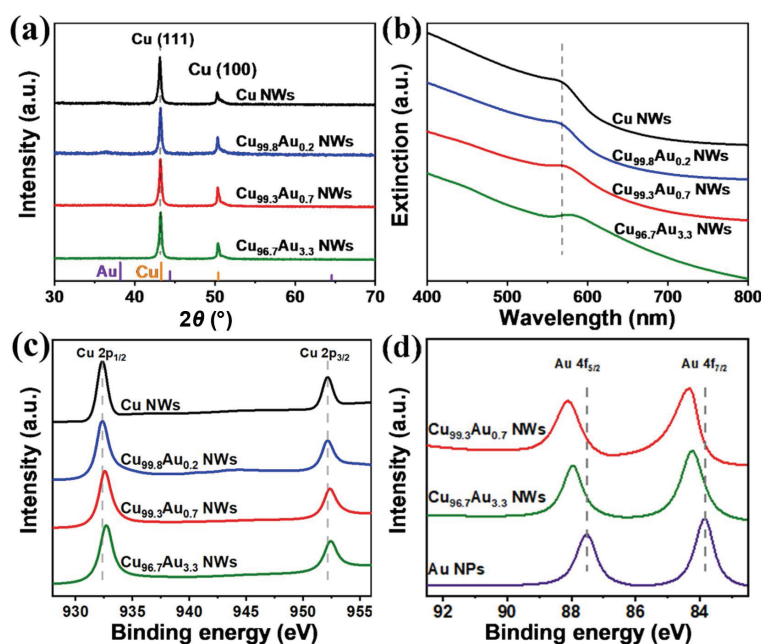
were loaded on the surface of Cu NWs with an average size of  $5.9 \pm 0.6$  nm. High-resolution transmission electron microscopy (HRTEM) images of both Cu-Au and Cu NWs showed the exposed lattice spacings of 0.21, 0.18, and 0.13 nm, respectively, which can be assigned to the (200), (111), and (220) plane of Cu. Cu-Au NWs distinguished the lattice spacing of 0.24 nm for the bulging particles, which can be assigned to Au (111) plane (Fig. 1(d)). The elemental mapping using energy dispersive X-ray spectroscopy (EDS) indicated the spatial distributions of Cu and Au (Figs. 2(f)–2(g)). According to the mapping data, the Cu signals were distributed in the core, while the small spheres composed of Au were evenly distributed across, confirming the formation of Au NPs loaded on the surface of Cu NWs. Furthermore, as shown in Figs. S3–S5 in the ESM, the more content of Au, the more Au NPs distributed on the Cu NW surface.

Several characterizations were executed to gain insight into the composition and electronic structures of Cu and Cu-Au NWs. X-ray diffraction (XRD) data confirmed the Cu crystal structures of all samples (Fig. 3(a)). There were two peaks at  $2\theta = 43.3^\circ$  and  $50.4^\circ$ , corresponding to the diffractions in the (111) and (200) of the face-centered cubic (fcc) Cu (JCPDS 04-0836), which suggested the fcc structure typical of Cu in the Cu and Cu-Au NWs. It is noteworthy that there was no Bragg angle shift for the Cu diffraction signal, indicating a phase-separated bimetallic nature of the Cu-Au NWs without the CuAu alloy. In the ultraviolet–visible (UV–vis) extinction spectra, primarily localized surface plasmon resonance (LSPR) peaks were also associated with Cu (Fig. 3(b)). As the Au content increased, the peak exhibited a red shift from 568 to 580 nm, indicating the charge transfer between Cu NWs and Au dopants. Due to the relatively small amounts of Au, no obvious Au-related peaks were found in both the XRD patterns and the UV–vis extinction figures while the loading of Au could be confirmed by X-ray photoelectron spectroscopy (XPS) and inductively coupled plasma optical emission spectroscopy (ICP-OES). By integration of corresponding XPS peaks, the surface of Cu<sub>99.3</sub>Au<sub>0.7</sub> contained 96.1% of Cu and 3.9% of Au, while the bulk was measured to be 99.3% of Cu and 0.7% of Au obtained by ICP-OES (Table S1 in the ESM). This difference also occurred in the other two samples, as confirmed that Au dopants were mainly distributed on the sample surface.

Furthermore, the electronic interaction between Cu and Au was also verified in the high-resolution XPS spectra (Figs. 3(c) and 3(d)). The peak of Cu 2p<sub>3/2</sub> shifted from 932.2 (Cu NWs) to 932.5 eV (Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs) and further to 932.7 eV for Cu<sub>96.7</sub>Au<sub>3.3</sub> NWs corresponding to the binding energy of Cu 2p<sub>1/2</sub> from 952.1



**Figure 2** (a) SEM, (b) low-, (c) high-magnification TEM, (d) HRTEM, (e) high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) images, and (f)–(h) the corresponding EDS mapping images of Cu-Au NWs. In the EDS elemental mapping, yellow color represents Cu, and green color represents Au.



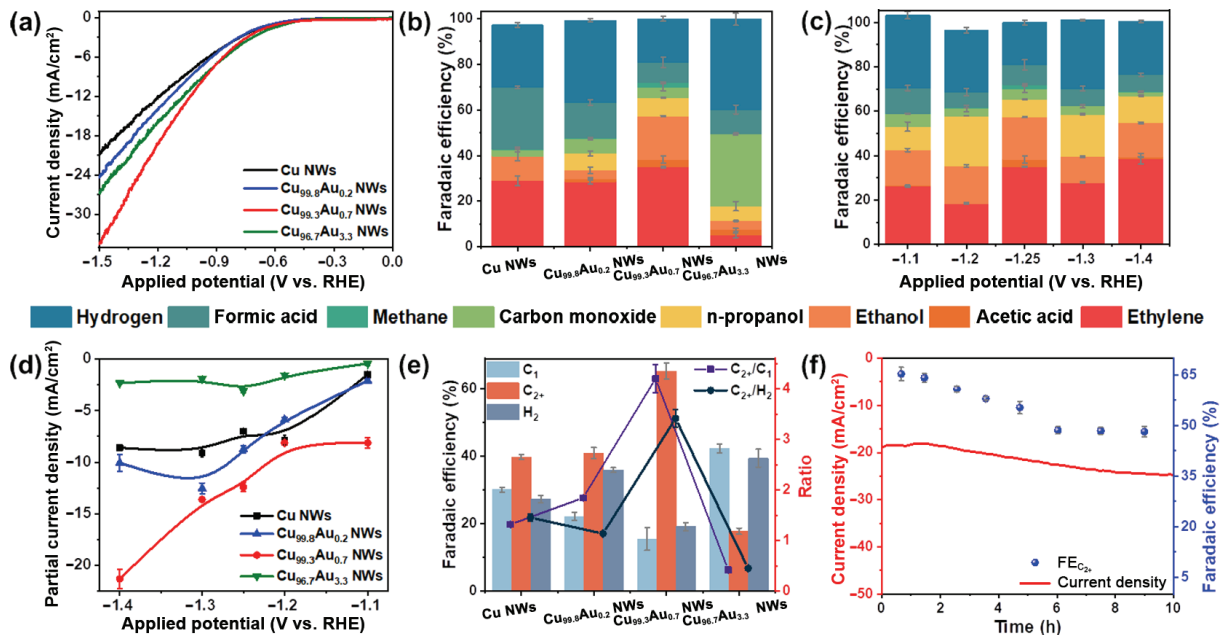
**Figure 3** (a) XRD patterns and (b) UV-vis extinction spectra of the as-prepared samples of Cu and Cu-Au NWs, respectively. High-resolution XPS spectra of (c) Cu 2p from the Cu-Au and Cu NWs and (d) Au 4f from the Cu-Au NWs and Au NPs.

to 952.5 eV. To get sight into the electronic state of gold, similar spherical Au NPs were obtained with an average size of 6.6 nm (Fig. S6 in the ESM). For the Au counterpart, the shift of the binding energy of Au 4f<sub>7/2</sub> presented the opposite trend from 84.3 (Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs) to 83.8 eV (Au NPs). The Au atomic fraction of the Cu<sub>99.8</sub>Au<sub>0.2</sub> sample could not achieve the lowest detection limit of XPS test, so the discussion of Au 4f peak shift was conducted on the other samples. These changes of binding energy positions could be explained by the interphase electron transfer from Cu to Au atoms. As the electron donor, the energy binding of Cu increased, while that of Au as the electron acceptor decreased.

Inspired by the above characterizations, the linear morphology and the electron transfer of Cu-Au NWs provided the potential for the CO<sub>2</sub>RR to multicarbon products. The electrocatalytic CO<sub>2</sub>RR performances of Cu-Au NWs were tested in an H-type three-electrode cell in CO<sub>2</sub>-saturated mixed 0.1 M KHCO<sub>3</sub> and 0.1 M KCl (Fig. 4) along with Cu NWs and Au NPs as comparisons. The CO<sub>2</sub>RR product distribution was obtained using gas chromatographic (GC) analysis for gas products and nuclear magnetic resonance (NMR) analysis for liquid products via the

internal standard method. The linear sweep voltammetry (LSV) curves of all samples were conducted at a sweeping rate of 10 mV/s (Fig. 4(a)). As noted by the total current densities at potentials in the range of 0 to 1.5 V vs. reversible hydrogen electrode (RHE), the Cu-Au nanowires showed a clearly rapid increment, wherein Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs obtained the largest current density. This observation suggested that the moderate amount of Au NPs doped could contribute to the enhanced CO<sub>2</sub>RR activity of the nanowires.

The selectivity and partial current density for each CO<sub>2</sub>RR product obtained were then measured and compared. As the product distribution shown in Fig. 4(b), Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs exhibited excellent selectivity for the multicarbon products. For Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs, FEs of total C<sub>2+</sub> products were boosted from 39.7% for pristine Cu NWs to 65.3%, while the C<sub>1</sub> and H<sub>2</sub> productions were inhibited from 27.3% and 30.1% to less than 15% and 19% at -1.25 V vs. RHE. The major C<sub>2+</sub> products were ethylene (35.0%) and ethanol (19.1%) with small amounts of n-propanol (8.0%) and acetate (3.3%). When the applied potentials extended to the range of -1.1 and -1.4 V vs. RHE, the FEs of the multicarbon



**Figure 4** (a) LSV curves with the Cu and Cu-Au NWs, respectively. (b) The selectivity of CO<sub>2</sub>RR at -1.25 V vs. RHE for different catalysts. (c) Total product FEs of Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs at different potentials. (d) Partial current density for the C<sub>2+</sub> products over the different catalysts in 0.1 M KHCO<sub>3</sub> + 0.1 M KCl. The current densities were normalized to the geometric area. (e) FEs of H<sub>2</sub>, C<sub>1</sub>, and C<sub>2+</sub> for the Cu and Cu-Au NWs at -1.25 V vs. RHE. (f) Electrochemical stability test of Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs under -1.25 V vs. RHE.

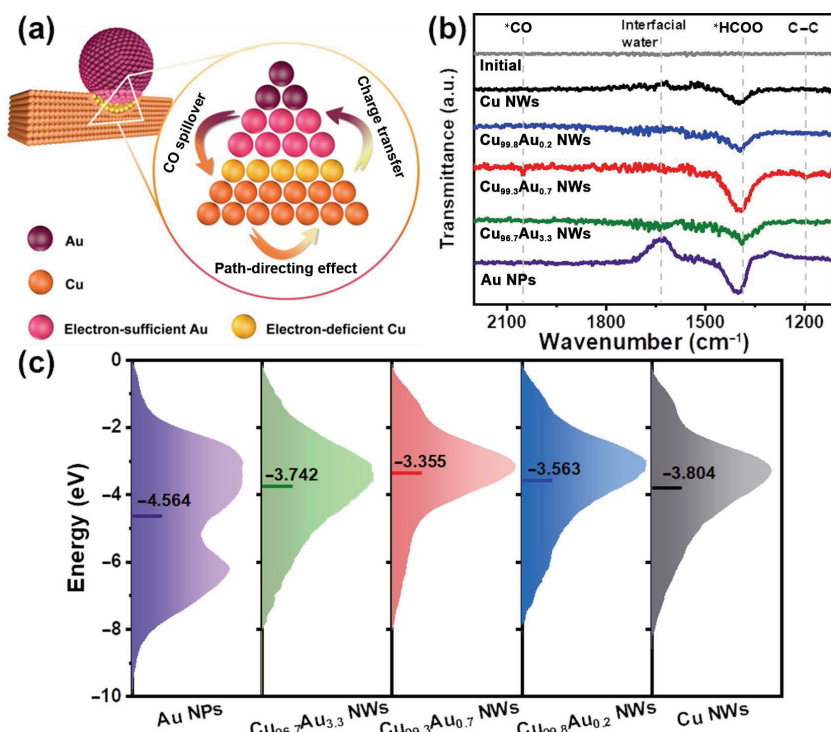
products for Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs could be maintained at over 53% (Fig. 4(c)). Notably, the high selectivity toward n-propanol reached 22.5% at -1.20 V vs. RHE. To explore the effect of Au content on CO<sub>2</sub>RR performance, we also prepared Cu<sub>99.5</sub>Au<sub>0.5</sub> and Cu<sub>98.3</sub>Au<sub>1.7</sub> NWs for comparison (Fig. S9 in the ESM). For Cu<sub>99.8</sub>Au<sub>0.2</sub> NWs, the highest FEs of C<sub>2+</sub> products were below 55.0% and the H<sub>2</sub> FEs were maintained at ~ 34% at -1.3 V vs. RHE. The selectivity of Cu<sub>99.5</sub>Au<sub>0.5</sub> NWs was maintained at 55% within a wide potential range. When the Au mass ratio was 0.7%, the selectivity reached the highest value with the 65.3 % FE of C<sub>2+</sub> products. When the Au content further increased, Cu<sub>96.7</sub>Au<sub>1.7</sub> and Cu<sub>98.3</sub>Au<sub>3.3</sub> NWs exhibited poor selectivity of C<sub>2+</sub> productions as the FE<sub>C<sub>2+</sub></sub> was below 30% at all applied potentials. Carbon monoxide, formic acid, and hydrogen are the main products of Cu<sub>96.7</sub>Au<sub>1.7</sub> and Cu<sub>98.3</sub>Au<sub>3.3</sub> NWs. Correspondingly, Au NPs favored the selectivity of carbon monoxide at low potential. As the potential became more negative, the FE<sub>CO</sub> decreased while the FE<sub>H<sub>2</sub></sub> increased with a small amount of formic acid production at -1.4 V. Detailed FE data for the catalysts are displayed in Fig. S10 and Table S2 in the ESM. Accordingly, the multicarbon partial current density of Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs was up to 12.40 mA/cm<sup>2</sup> at -1.25 V vs. RHE and further increased at the more negative potential (Fig. 4(d)). In contrast, the partial current densities for Cu and Cu<sub>96.3</sub>Au<sub>3.3</sub> NWs were less than 10 mA/cm<sup>2</sup> at the same reduction condition and the largest partial current density was 12.7 mA/cm<sup>2</sup> at -1.3 V for Cu<sub>96.3</sub>Au<sub>3.3</sub> NWs. Thus, Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs exhibited the highest multicarbon production rate. To evaluate the selectivity towards C<sub>2+</sub> versus C<sub>1</sub> products in CO<sub>2</sub>RR, we compared the ratio of FE<sub>C<sub>2+</sub></sub> to FE<sub>C<sub>1</sub></sub> on different catalysts at -1.25 V vs. RHE (Fig. 4(e)). Relative to the Cu NWs, the ratio of FE<sub>C<sub>2+</sub></sub> to FE<sub>C<sub>1</sub></sub> for Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs was increased from 1.3 to 4.2, representing a 3.3 times improvement, further indicating the doping of Au NPs notably suppressed the C<sub>1</sub> production as well as promoting the generation of multicarbon. However, when more Au content was added, i.e., Cu<sub>96.3</sub>Au<sub>3.3</sub>, the decline of the FE<sub>C<sub>2+</sub></sub> and the FE<sub>CO<sub>2</sub>RR</sub> corresponded to a steep downward of the FE<sub>C<sub>2+</sub></sub>/FE<sub>H<sub>2</sub></sub> and FE<sub>C<sub>2+</sub></sub>/FE<sub>C<sub>1</sub></sub> values, which was ascribed to Au occupying the active sites of Cu on the surface. Au atoms promoted the desorption of intermediate \*CO and in turn inhibited C-C

coupling. As a result, the C<sub>2+</sub> FEs exhibited a volcano-shaped dependence on Au nanoparticle capacity with the Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs possessing the highest C<sub>2+</sub> selectivity.

With the above superior activity and selectivity, Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs could be regarded as an efficient catalyst for CO<sub>2</sub>RR to generate multicarbons. Except for the enhanced activity and selectivity, the high stability for Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs was also confirmed under the CO<sub>2</sub>RR electrolysis of -1.25 V vs. RHE for 10 h. After 10 h, the FE<sub>C<sub>2+</sub></sub> maintained ~ 50%, corresponding to ~ 75% retention. The TEM and HRTEM images confirmed the preservation of the one-dimensional (1D) morphology of Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs (Fig. S10 in the ESM). The XPS spectra presented similar spectra before and after the CO<sub>2</sub>RR test (Fig. S11 in the ESM). Hence, the morphology and the electronic structures of the post-reaction Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs have been stably preserved.

To gain an ideal CuM (M represents an additional metal) linear catalyst, it is important to get insight into the enhanced selectivity and activity of Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs. A schematic description showing the proposed mechanism behind the electrocatalytic enhancement of the Cu-Au NWs is plotted in Fig. 5(a). Firstly, the electrochemical surface area (ECSA) for various nanowires was analyzed by the double-layer capacitance test after the Au nanoparticles modification (Fig. S12 in the ESM). The largest ECSA was obtained on the Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs with 1.6 times larger than that of Cu NWs, in conjunction with the highest current density [38]. Although Cu<sub>99.3</sub>Au<sub>0.7</sub> and Cu<sub>96.3</sub>Au<sub>3.3</sub> NWs had similar ECSA, their catalytic performances for the reduction favor of multicarbon and carbon monoxide were not nearly identical, suggesting the higher surface area was not the only reason for the augmented CO<sub>2</sub>RR.

Secondly, as the tandem effect was also widely applied in bimetallic catalytic systems [39–41], we researched the roles of Cu and Au in the process of CO<sub>2</sub>RR. The tandem process of CO<sub>2</sub>RR to multicarbon products involved the CO<sub>2</sub>RR to \*CO, the \*CO overspill into the neighbor metal, and the coupling of two \*CO. Our result verified that Au was an excellent CO<sub>2</sub>RR to CO catalyst (Fig. S6(d) in the ESM). In our system, the CO production rates on Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs were much higher than Cu NWs, increasing with more negative potential while the production rate of gaseous CO was maintained at ~ 2 μmol/s (Fig. S13 in the ESM).



**Figure 5** (a) Schematic representation of the proposed mechanism of  $C_{2+}$  promotion in the Cu-Au NWs, which couples the path-directing effect of the high aspect ratio nanowire morphology, the tandem catalysis induced by Au and Cu as separated phases, and the electronic effect caused by the charge transfer at the interface. (b) *In situ* electrochemical ATR-FTIR spectra of Cu NWs, Cu-Au NWs, and Au NPs at  $-1.25$  V vs. RHE in the  $CO_2$ -saturated mixture of  $0.1$  M KCl and  $0.1$  M  $KHCO_3$  at the fifth minute. (c) Surface valence band photoemission spectra of Cu NWs, Cu-Au NWs, and Au NPs.

Furthermore, normalized by the ECSA, the CO spillover efficiency at  $-1.25$  V vs. RHE was 87.2%, indicating that  $\sim 87\%$  of the CO was produced on Au sites [42]. Furthermore, *in situ* electrochemical attenuated total reflection Fourier transformed infrared (ATR-FTIR) spectroscopy was employed for the reaction intermediate (Fig. 5(b)). All three samples exhibited the infrared peak at around  $1400\text{ cm}^{-1}$  which was associated with  $*HCOO$  [43, 44]. Compared with those of the Cu NWs and Au NPs, the red shift of  $10\text{ cm}^{-1}$  was observed for the  $*HCOO$  band on the Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs, suggesting the enhanced adsorption strength of  $*HCOO$ . Compared with Cu NWs and Au NPs, two new peaks were observed at  $1195$  and  $2050\text{ cm}^{-1}$  on the Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs, which were related with the C–C vibration band and  $*CO$ . These peaks disclosed that the Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs showed a strong interaction with the  $*CO$  and following C–C coupling. To further evaluate the interaction between the intermediates and catalyst surface, surface valence band photoemission spectra of Cu NWs, Cu-Au NWs, and Au NPs were collected (Fig. 5(c)). Among them, Au NPs had the lowest lying d-band center, suggesting the weakest binding ability with CO. This result could elucidate that Au NPs exhibited a high selectivity of CO, which would be an essential prerequisite for the tandem effect. By contrary, Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs had the highest lying d-band center, indicating the enhanced adsorption strength of  $*CO$  intermediates [45, 46], which was consistent with the result of *in situ* ATR-FTIR.

Lastly, as demonstrated by XPS and UV–vis spectra, it can be inferred that the formation of electron-deficient Cu sites generated from the interphase electron transfer from Cu to Au atoms would favor the adsorption of  $*CO$  intermediates. Above all, the Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs exhibited the remarkable performance compared with the other materials reported in the previous literature (Table S3 in the ESM). The  $CO_2$ RR improvement of Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs could be attributed to the enhanced adsorption of the intermediates and the promoted C–C coupling, favoring the multicarbon productions.

### 3 Conclusions

In summary, built on the superior structural condition of nanowires, we have developed Cu-Au NWs loaded Au NPs on the Cu nanowire surface through Au(0) homo-nucleation strategy. Among those, Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs led to the optimal selectivity and activity of  $C_{2+}$  products. The introduction of Au nanoparticles on the Cu NW could bring more active sites, the CO spillover, and the electron transfer between Cu and Au. The *in situ* ATR-FTIR results together with surface valence band photoemission spectra demonstrated that Cu<sub>99.3</sub>Au<sub>0.7</sub> NWs had the strongest interaction between the catalyst surface and the  $*CO$  intermediate and the following step of  $*CO$  dimerization for the multicarbon products. Our work suggested an effective Cu-based catalyst for  $CO_2$  electrochemical reduction and highlighted the critical role of intermetallic interactions in the design of bimetallic catalysts.

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**Electronic Supplementary Material:** Supplementary material (TEM and HRTEM images; SEM images; EDS mapping spectra; XPS measurements; LSV curves; and Faradaic efficiencies of products for  $CO_2$ RR) is available in the online version of this article at <https://doi.org/10.1007/s12274-023-5430-z>.

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