

Observation of interlayer excitons in trilayer type-II transition metal dichalcogenide heterostructures

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ABSTRACT

Vertically stacked transition metal dichalcogenide (TMD) heterostructures provide an opportunity to explore optoelectronic properties within the two-dimensional limit. In such structures, spatially indirect interlayer excitons (IXs) can be generated in adjacent layers because of strong Coulomb interactions. However, due to the complexity of the multilayered heterostructure (HS), the capture and study of the IXs in trilayer type-II HSs have so far remained elusive. Here, we present the observation of the IXs in trilayer type-II staggered band alignment of MoS₂/MoSe₂/WSe₂ van der Waals (vdW) HSs by photoluminescence (PL) spectroscopy. The central energy of IX is 1.33 eV, and the energy difference between the extracted double peaks is 23 meV. We confirmed the origin of IX through PL properties and calculations by the density functional theory, we also studied the dependence of the IX emission peak on laser power and temperature. Furthermore, the polarization-resolved PL spectra of HS were also investigated, and the maximum polarizability of the emission peak of WSe₂ reached 11.40% at 6 K. Our findings offer opportunities for the study of new physical properties of excitons in TMD HSs and therefore are valuable for exploring the potential applications of TMDs in optoelectronic devices.

KEYWORDS

transition metal dichalcogenides, interlayer exciton, type-II band alignment, trilayer heterostructure

1 Introduction

Two-dimensional (2D) transition metal dichalcogenides (TMDs) have attracted widespread attention because of their unique electronic and optical properties that have wide application perspectives in the fields of electronics and optoelectronics [1–7]. The vertical stacking of two different monolayered TMDs can form a van der Waals (vdW) heterostructure (HS), creating interesting new characteristics that are different from that of the monolayer structures [8–10]. For the vdW HS, the band arrangement of the various monolayer TMDs determines the properties and applications of the HS [11–15]. In the bilayer HS with a type-II band alignment, carriers are transferred from the high-energy layer to the low-energy layer due to the energy difference between the two TMD materials, and the formation of interlayer excitons (IXs) could be the driving force for light-excited electrons and holes to separate into different layers [16]. The IXs formed by the space-separated carriers due to the Coulomb effect have a significantly longer lifetime than their intralayer counterparts [1, 17]. Consequently, TMD HSs have wide-ranging application perspectives in long-life exciton devices and exciton quantum gas research [18–20].

Recently, many studies are focused on the HSs composed of two different monolayer TMDs [21–24]. Nevertheless, our interest is to design an embedded trilayer vertical heterostructure with

type-II band alignment, and explore whether different layers can cause indirect–direct bandgap transitions. Compared to bilayer type-II HSs, the photoluminescence (PL) quantum efficiency and tunable spectral resonance of the trilayer HSs are considerably improved [13]. In addition, the IXs have a lower energy limit and a longer lifetime expectancy [25]. The trilayer vdW HSs of TMDs may have tremendous application potential such as photodetectors, light-emitting diodes, lasers, and photovoltaics [26–28]. However, the feature of IXs is only observed in the bilayer type-II heterostructures.

In this study, we were successful in fabricating a trilayer MoS₂/MoSe₂/WSe₂ HS encapsulated with hexagonal boron nitride (hBN) by dry transfer technique [29, 30]. By analyzing the PL spectrum, the IXs of the trilayer type-II MoS₂/MoSe₂/WSe₂ HS were observed. According to the energy range of the IXs, and the energy difference between the two peaks of IXs extracted by Lorentzian fitting is 23 meV. Therefore, the observed IXs were attributed to the MoSe₂/WSe₂ bilayer. The power dependence measurement indicated that the splitting of the IX was caused by the conduction band spin-splitting of MoSe₂. Furthermore, we also studied the circular polarization of the HS, and the maximum polarization rate of the WSe₂ emission peak reached 11.40% at 6 K. The results of the observation provide new insights for exploring new exciton physics and quantum emitters in TMDs materials.

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2 Results and discussion

Figure 1(a) is an optical micrograph of the trilayer HS sample, with dashed outlines in different colors indicating regions of different materials. Figure 1(b) displays the schematic diagram of the HS, where hBN is placed on the top and bottom of the MoS₂/MoSe₂/WSe₂ HS. The HS is encapsulated with hBN to reduce scattering from the substrate and protect monolayer materials. The band arrangement diagram of the embedded MoS₂/MoSe₂/WSe₂ HS with type II is shown in Fig. 1(c). Carriers will transfer from high energy levels to low energy levels due to the energy difference between the monolayer materials. It can be seen from the schematic that the holes in the valence band of the adjacent layer are transferred to the upper layer, while the electrons in the conduction band are transferred to the lower layer. The middle layer acts as an intermediate medium for carrier transfer. The embedded type-II band alignment promotes carrier separation in type-II band alignment. The Raman spectrum of each layer constituting the HS is shown in Fig. 1(d). For monolayer WSe₂ [31], the out-of-plane A_{1g} and in-plane E_{2g}¹ Raman modes were not degenerate under 532 nm excitation at room temperature, resulting in a split of 11–12 cm⁻¹. For monolayer MoSe₂, the observation result of the out-of-plane Raman mode signal is 240.5 cm⁻¹, which is consistent with that of previous reports [32]. For monolayer MoS₂ [33], the frequency difference between the A_{1g} and E_{2g}¹ modes is less than 20 cm⁻¹. It can be observed that the Raman spectrum of the HS area contains all the vibration modes of the three materials, indicating that the three materials were well coupled. PL spectra were used to characterize MoS₂, MoSe₂, and WSe₂ monolayers, as well as the HS. Figure 1(e) presents the PL spectra of different monolayer and the HS regions excited by a 532 nm (2.33 eV) laser at room temperature. The central energies of the emission peaks of MoS₂, MoSe₂, and WSe₂ monolayers at room temperature are 1.83, 1.57,

and 1.65 eV, respectively, which are consistent with previous reports [34–36]. Characteristic peaks for different monolayers were observed in the PL spectra of the HS region. Figure 1(f) shows the PL spectra of these monolayers and the HS region when the sample was cooled to 6 K. It can be observed that the PL spectra of the MoS₂, MoSe₂, and WSe₂ monolayers have strong peaks at 1.91, 1.61, and 1.67 eV, respectively, while the PL spectrum of the HS region has a peak at 1.33 eV, which is closely related to the peak of the IXs [1, 23].

To investigate the optical characteristics of the HS, we studied the PL spectra of MoS₂, MoSe₂, and WSe₂ monolayers, as well as the HS as a function of temperature, ranging from 6 to 300 K. It can be seen from Fig. 2 that as the temperature decreased, the types of the emission peaks of the TMD materials became more abundant, caused by suppressing the thermal disturbance of the electrons. Furthermore, as the temperature cooled down, the peak positions of all emission peaks were blue-shifted, which was consistent with the Varshni equation describing the change of the bandgap of traditional semiconductors with temperature [37]. The formula is as follows (Eq. (1))

$$E_g(T) = E_0 - \alpha T^2 / (T + \beta) \quad (1)$$

where α and β are the fitting parameters of the Varshni equation, while $E_g(T)$ and E_0 represent the bandgaps at the current temperature and the absolute zero, respectively.

Figure 2(a) depicts the PL spectrum of the monolayer MoS₂ as a function of temperature. As the temperature went down, one peak became two peaks, and the peak positions of all emission peaks were blue-shifted. The types of these two excitons can be identified by their PL spectra [34]. The central energies of the two emission peaks at 6 K are 1.799 and 1.907 eV, respectively, which are expressed as bound exciton (X_L) and neutral exciton (X_0). Interestingly, the measurement showed that the peak of bound

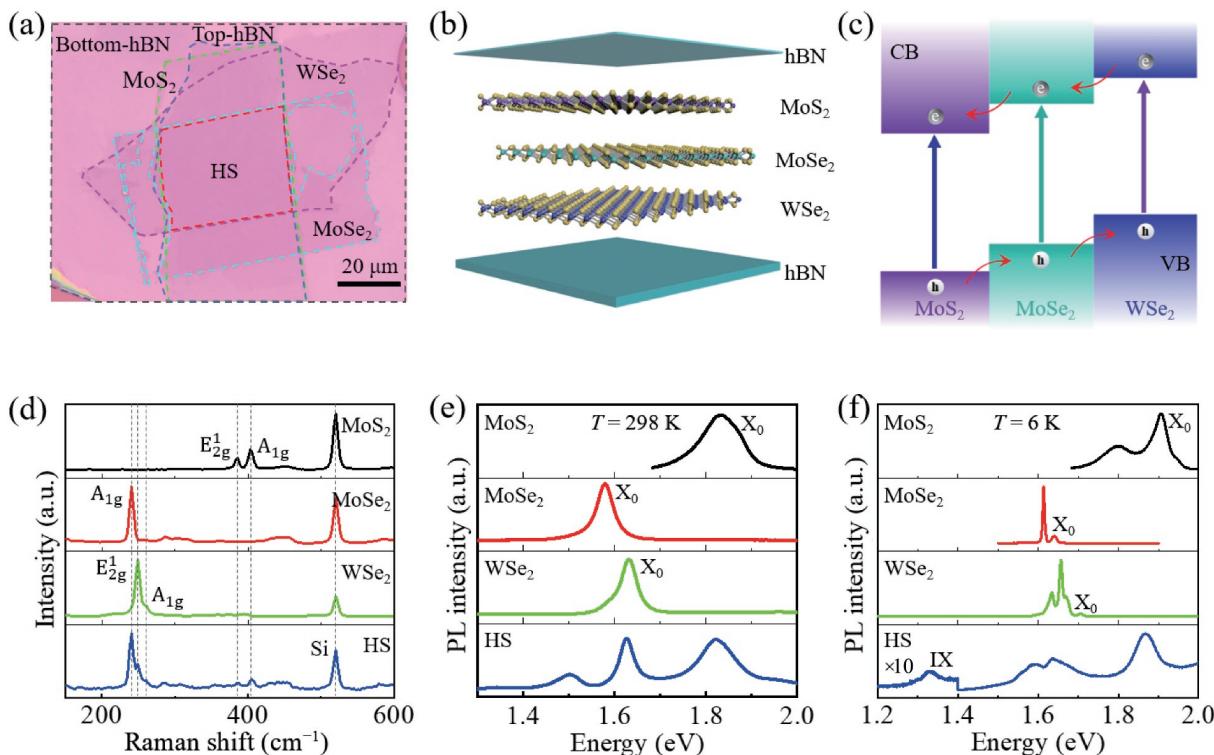


Figure 1 Design and characterization of the MoS₂/MoSe₂/WSe₂ HS. (a) Optical micrograph of MoS₂/MoSe₂/WSe₂ HS encapsulated with hBN. The dashed lines indicate different materials in different areas. (b) Schematic diagram of the hBN/MoS₂/MoSe₂/WSe₂/hBN HS. (c) Schematic diagram of the energy band of the HS. The arrows of different colors indicate the formation of different direct excitons, and the elliptical dashed box indicates the formation of IXs. The red arrows indicate the transfer process of carriers in different layers. (d) Raman spectra of single materials and HS at room temperature. (e) and (f) represent the PL spectra of MoS₂, MoSe₂, WSe₂, and HS under excitation at 532 nm at room temperature (298 K) and 6 K, respectively.

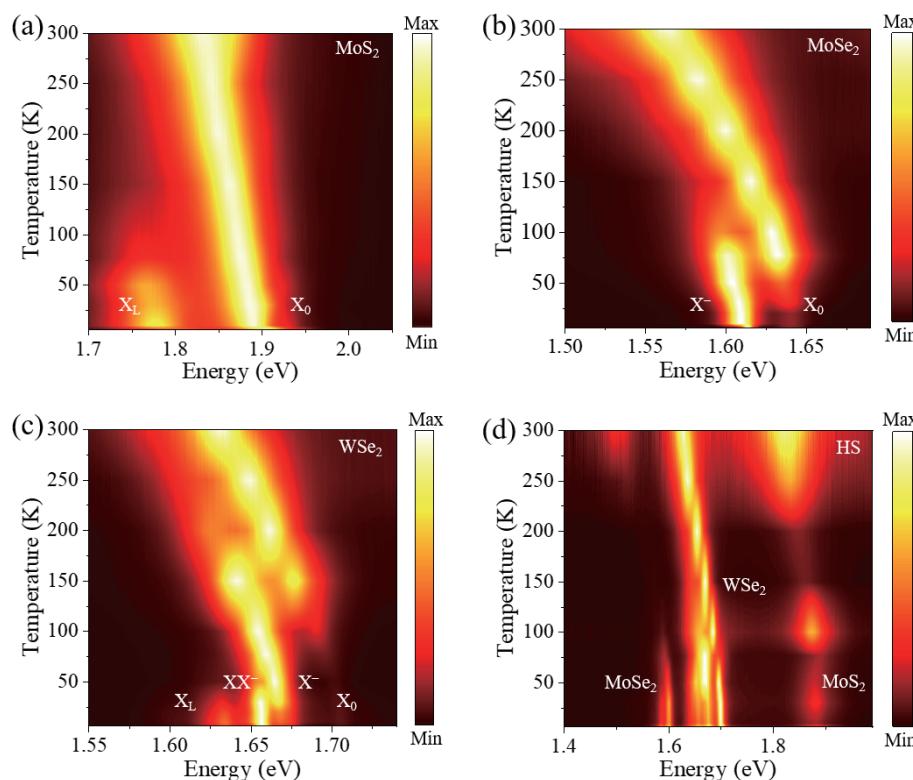


Figure 2 Temperature-dependent PL maps. The temperature dependence PL of (a) monolayer MoS₂, (b) monolayer MoSe₂, (c) monolayer WSe₂, and (d) HS, varying from 6 to 300 K. The exciton type corresponding to each PL peak is marked on the corresponding heat map. (d) The emission peaks of various materials in the HS are also marked with the corresponding materials. As the temperature decreases, the emission peak appears blue-shifted, which is consistent with the Varshni equation of traditional semiconductors.

exciton appeared at an energy below 1.85 eV, suggesting that there might be a large number of defects distributed in the monolayer MoS₂, which acted as the trap sites for excitons. The trapped electrons are combined with holes to form defect-bound excitons. Therefore, multiple defect energy levels appeared in the energy bandgap to produce a broad X_L peak. Figure 2(b) shows the variation of the PL spectrum of the monolayer MoSe₂ in relation to temperature. It should be noted that as the temperature cooled down, the primary peak split into two peaks, and the emission peaks are blue-shifted. At 6 K, the position energy difference of the two emission peaks is 27 meV, which is similar to previous reports [21]. These two emission peaks are represented as X₀ and charged exciton (X⁻) of monolayer MoSe₂, respectively. In addition, when the temperature was below 77 K, it could be seen that the X⁻ peak gradually dominated, while the X₀ peak became weaker, which was caused by the limitation of electronic thermal fluctuations [38]. Similarly, Fig. 2(c) shows the PL spectrum of the monolayer WSe₂ versus temperature. It could be observed that as the temperature decreased, the types of excitons of the monolayer WSe₂ became very abundant [39, 40]. At 6 K, the energies of the low to high emission peaks are depicted as X_L, charged biexciton (XX⁻), X⁻, and X₀ [16]. Figure 2(d) illustrates the variation of the PL spectra of the MoS₂/MoSe₂/WSe₂ HS with temperature. The PL spectra of the three materials can be clearly distinguished from the PL spectrum of the HS, with the emission peaks displaying a blue shift as the temperature decreased.

To explore the properties of the IXs, we carried out detailed research and discussions in Fig. 3. As shown in Fig. 3(a), as the temperature decreases, the peak intensity of the IX increases due to the decrease in the thermal disturbance and scattering of the electrons, and the peak position appears blue shifted [37]. When the temperature is below 77 K, two peaks of IX emission can be slightly observed, which can be caused by IX splitting due to the spin-orbit coupling effect [1]. It is worth noting that the energy of

the IX emission peak ranges from 1.26 to 1.38 eV at 6 K. According to the characteristics of the IX emission peak, it can be related to the IX emission peak of the MoSe₂/WSe₂ bilayer [41, 42]. Figure 3(b) shows the IX peak in the HS region at 6 K excited by a 532 nm laser. The observed IX peak was deconvoluted into two peaks, IX₁ and IX₂, by Lorentzian function fitting. Importantly, it can be observed that the energies of the IX₁ and IX₂ peaks are 1.327 and 1.350 eV, respectively, having the energy difference of 23 meV that is entirely consistent with the first-principle calculations on the splitting of the MoSe₂ conduction band [43]. It was theoretically predicted that the conduction band splitting of MoS₂ is only 3 eV and the binding energy of the negative interlayer trion is 28 eV, so the observed interlayer excitons cannot come from the MoS₂/MoSe₂ bilayer [21, 43]. Furthermore, the low IX energy of the MoS₂/WSe₂ bilayer is beyond the detection range of our device (1.1–2.2 eV) [44]. Therefore, the observed interlayer excitons originate from the MoSe₂/WSe₂ bilayer. Consequently, the splitting of the IX is most likely to originate from the splitting of the MoSe₂ conduction band.

Figure 3(c) shows the heat map of the IXs at 6 K as a function of the laser power. The data was derived from Fig. 3(d), making the evolution of IXs with power more intuitive. It can be observed that as the power increased, the peaks of the IXs also became stronger. At the same time, Fig. 3(d) displays the normalized PL intensity of IXs at 6 K as a function of the laser power, varying from 0.1 to 5 mW. Under the power of 0.1 and 5 mW, the PL spectra of IX₁ and IX₂ were obtained by Lorentzian function fitting, represented by the blue and red lines, respectively. The arrowed directions of the blue and red dotted lines were the evolution of IX₁ and IX₂ with the excitation power. Interestingly, as the power increased, the IX₁ peak gradually weakened, whereas the IX₂ peak became stronger gradually. The observed dipole phenomenon is in line with the spin splitting of the monolayer MoSe₂ conduction

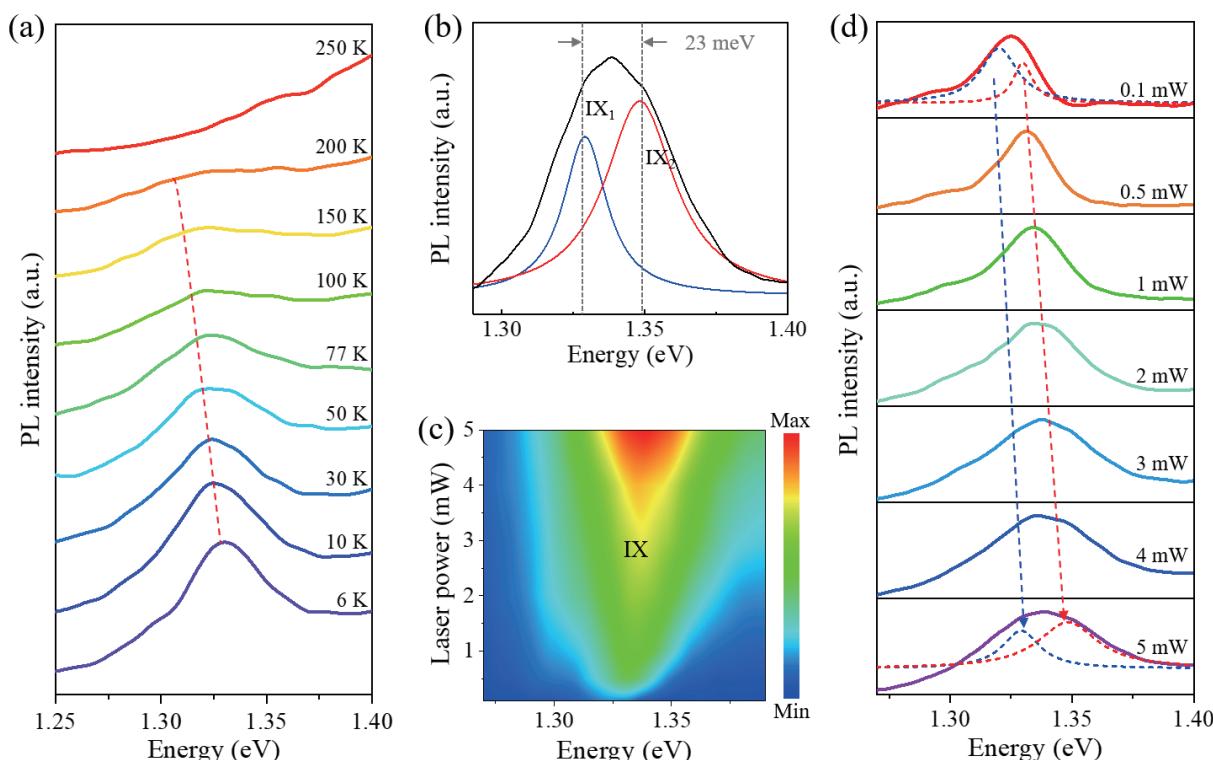


Figure 3 PL features of the HS. (a) PL spectra of IXs in the HS at varying temperatures. (b) The solid black line represents the measured data at 6 K, and the blue and red solid lines are the fitting peaks of IX₁ and IX₂, respectively. (c) The heat map of IXs as a function of power with the same data from (d). (d) The PL spectra of IXs at different excitation power range from 0.1 to 5 mW. At the power of 0.1 and 5 mW, the PL spectra of IX₁ and IX₂ obtained by Lorentzian fitting were shown by blue and red lines, respectively. The blue and red dotted arrows are the evolution of IX₁ and IX₂ under different powers, respectively.

band. At low power, photo-excited electrons first filled the lowest energy configuration within the MoSe₂ spin splitting band, so that the IX₁ peak with lower energy dominated first. In addition, as the power increased, the photo-excited electrons were excited to higher energy levels, and the electrons were located on the high spin splitting band of MoSe₂, so the high excitation power was dominated by the IX₂ peak, which was completely consistent with the phenomenon we observed. It should be noted that as the excitation power increased, the IXs appeared to be blue shifted, which is a signature of the repulsive interaction between IXs under the dipole arrangement [1]. By combining the above analysis, we attribute the observed IX splitting to the spin-splitting of the monolayer MoSe₂ conduction band.

To further support the origin of IXs, we calculated the band structure of the HS by density functional theory. Figure 4 shows the energy band structure diagrams of four different stacking modes of MoS₂/MoSe₂/WSe₂ HS, and the four different stacking modes are represented by AA, AB, BA, and BB, respectively. Three different types of interlayer excitons IX_a, IX_b, and IX_c can be formed in trilayer type-II MoS₂/MoSe₂/WSe₂ HS, which are derived from MoSe₂/WSe₂, MoS₂/MoSe₂, and the longer-distance MoS₂/WSe₂ bilayers, respectively. As shown in Fig. 4(a), it can be obtained that the energies of the IXs formed among MoSe₂/WSe₂, MoS₂/MoSe₂, and the longer distance MoS₂/WSe₂ bilayer structure are 1.26, 0.61, and 0.31 eV, respectively. The calculated bandgap of the MoSe₂/WSe₂ bilayer at the K point is 1.26 eV, which is almost consistent with the observed IX peak of 1.33 eV in our experimental results, further indicating that the observed IXs originate from MoSe₂/WSe₂ bilayer. Although the theoretical calculation results cannot be in perfect agreement with the experimental phenomena, they provide insights into the origin of IXs. It can be seen that the band structures in Figs. 4(a)–4(d) are basically the same for different stacking modes, indicating that different stacking modes have little effect on the energy of IXs.

To better understand the circular polarization of the HS, the polarization-resolved PL spectrum of the MoS₂/MoSe₂/WSe₂ HS at 6 K was studied. As shown in Figs. 5(a)–5(c), the circular polarization of excitons in the HS region could be clearly observed. When the excitation light was a negative helicity (σ), the PL spectra of σ and positive helicity (σ^+) were analyzed. The degree of PL polarization can be quantified by the helicity parameter [23, 45], and the negative helical polarizability (P_{σ^-}) is defined as Eq. (2)

$$P_{\sigma^-} = [I(\sigma^-) - I(\sigma^+)]/[I(\sigma^-) + I(\sigma^+)] \quad (2)$$

where $I(\sigma)$ and $I(\sigma^+)$ represent the emission intensity of negative and positive helicity [23], respectively. Similarly, when the excitation light was σ^+ , we also analyzed the PL spectra of σ^+ and σ . The positive helical polarizability was denoted by P_{σ^+} , similar to the definition of the P_{σ^-} .

Using the σ light excitation, the polarization-resolved PL spectrum of the HS at 6 K and its corresponding polarizability are demonstrated in Fig. 5(a). It is obvious that the emission intensity of σ light was stronger than that of the σ^+ light. The emission peaks of MoSe₂, WSe₂, and MoS₂ in the HS PL spectrum had the largest polarizabilities at 1.59, 1.68, and 1.88 eV, respectively, and the corresponding polarizabilities were 2.64%, 10.91%, and 3.47%, respectively. Meanwhile, Fig. 5(b) displays the resolved polarized PL spectrum of the HS and its corresponding polarizability at 6 K under the σ^+ light excitation. In contrast to the situation where the σ light was excited, the emission intensity of the σ light was superior to that of the σ^+ light. The polarizabilities (P_{σ^+}) at the energies of 1.59, 1.68 and 1.88 eV were 2.37%, 11.40% and 3.91%, respectively. The difference between σ^+ and σ emission intensities was due to the valley optical selection rule [46–48]. When excited by the σ light, the photogenerated carriers were generated in the $-K$ valley, and the electron-hole pairs were recombined to emit σ

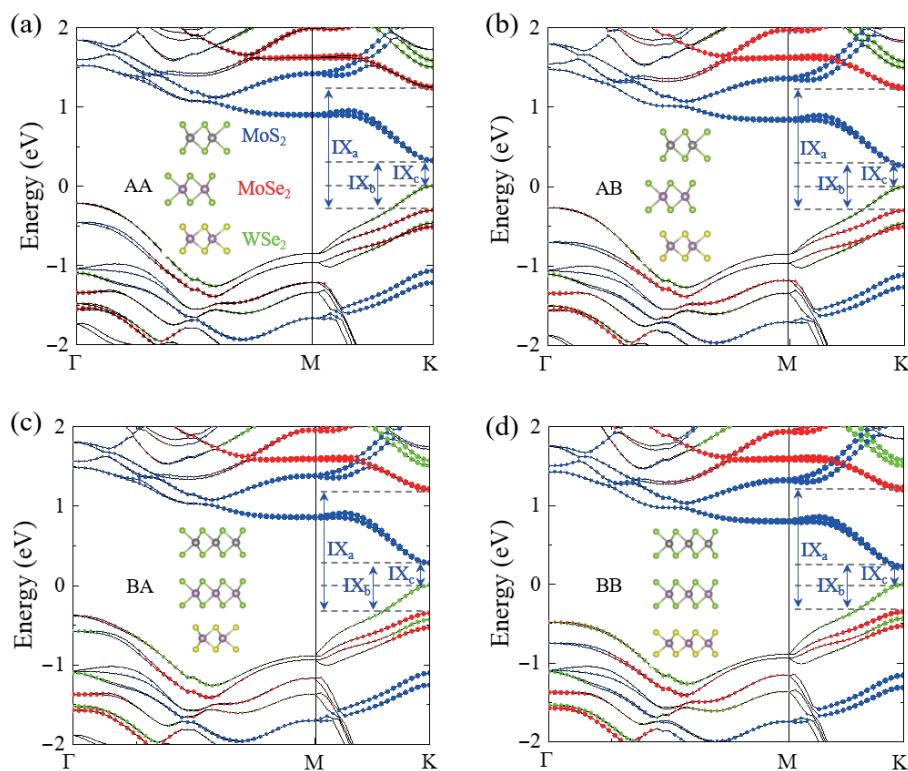


Figure 4 Density functional theory calculations of the HS. (a)–(d) Calculated the energy band structure of four different stacking patterns. AA, AB, BA, and BB represent four different stacking patterns. The atomic stacking structure of the corresponding band structure is shown in the inset. The blue, red, and green dots represent the band structures of MoS₂, MoSe₂, and WSe₂ monolayers, respectively.

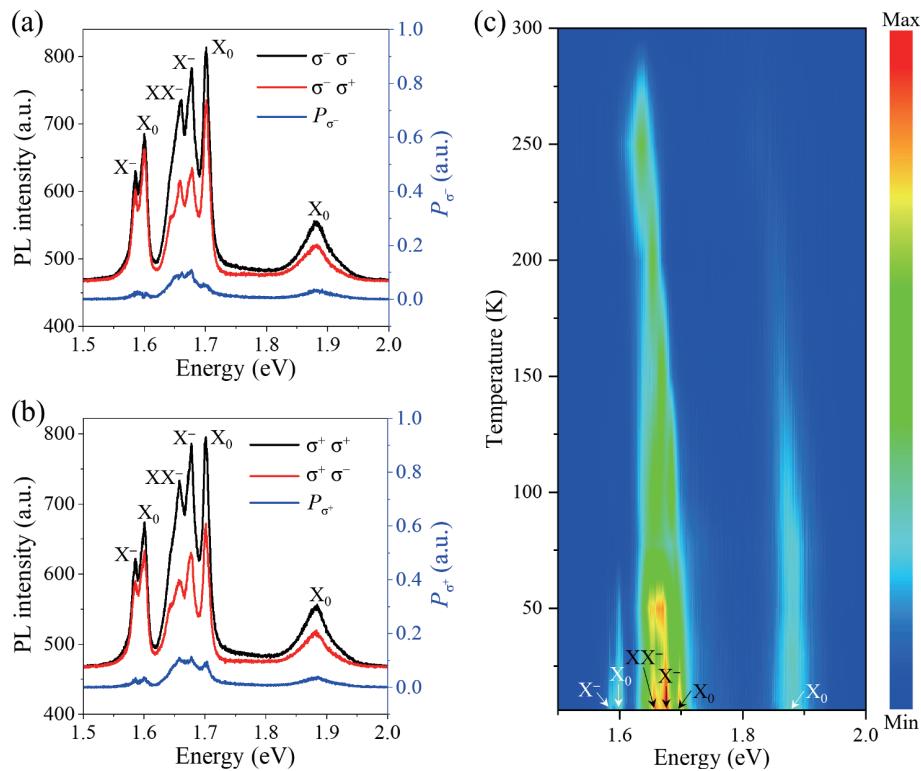


Figure 5 Valley polarization of the HS. (a) and (b) The polarization-resolved PL spectra of MoS₂/MoSe₂/WSe₂ HS were excited by σ and σ^* light at 6 K, respectively. The left y -axis of (a) and (b) represents the intensity of the polarization-resolved PL spectrum of the HS, and the right y -axis represents the corresponding polarizability. (c) Heat map of the polarizability of the HS as a function of temperature, ranging from 6 to 300 K.

light, while there were no carriers generated in the +K valley. However, there was considerable σ^- light emission at the +K valley due to intervalley scattering. Conversely, only the carriers at +K valley were excited when the excitation light was the σ^+ light. Figure 5(c) displays the polarization of the HS with respect to temperature, varying from 6 to 300 K. With the decrease of

temperature, the polarizabilities of MoSe₂, WSe₂, and MoS₂ emission peaks in the HS exhibited a blue shift and gradually increased due to the decrease of the thermal disturbance of the electrons. When the temperature was 6 K, the maximum polarization rate of the WSe₂ emission peak in the HS reached 11.40%.

3 Conclusions

In summary, we have designed and demonstrated the trilayer MoS₂/MoSe₂/WSe₂ van der Waals HS with a Type-II alignment. The IXs in the trilayer HS were successfully observed by measuring the PL spectra. The origin of IXs is revealed by analyzing the PL characteristics in the trilayer HS as well as density functional theory calculations. According to the energy range of the IXs, we attribute the generation of the IXs to the MoSe₂/WSe₂ bilayer, and the varying power further proved that the splitting of the IXs originated from the conduction band spin-splitting of MoSe₂. Furthermore, the maximum polarizability of the WSe₂ emission peak in the HS reached 11.40% at 6 K. Our research work may provide additional insights into the exploration of new exciton physics and quantum emitters in TMD materials.

4 Experimental section

We have obtained MoS₂, MoSe₂, WSe₂ monolayers, and hBN sheets from bulk (HQ graphene) crystals by mechanical exfoliation. HS samples were prepared on SiO₂/Si substrates by dry transfer technique, and the TMD monolayers were stacked at random lattice matching angles. Then, the prepared HS samples were annealed at 300 °C for 5 h to reduce impurities in the samples for better contact between different TMD materials. The model number of the cryogenic refrigeration system was C04-005-044 from Cryo Industries of America. All-optical data was acquired with a WITec Alpha 300R system, using a 50× objective lens and a laser spot of nearly 1 μm. The excitation wavelength of the laser in the experiment was 532 nm (2.33 eV). Density functional theory calculations were performed using the projector augmented-wave method implemented in the VASP code.

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