

# Axial coordination regulation of MOF-based single-atom Ni catalysts by halogen atoms for enhanced CO<sub>2</sub> electroreduction

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## ABSTRACT

Single-atom catalysts (SACs), with the utmost atom utilization, have attracted extensive interests for various catalytic applications. The coordination environment of SACs has been recognized to play a vital role in catalysis while their precise regulation at atomic level remains an immense challenge. Herein, a post metal halide modification (PMHM) strategy has been developed to construct Ni-N<sub>4</sub> sites with axially coordinated halogen atoms, named Ni<sub>1</sub>-N-C (X) (X = Cl, Br, and I), on pre-synthetic nitrogen-doped carbon derived from metal-organic frameworks. The axial halogen atoms with distinct electronegativity can break the symmetric charge distribution of planar Ni-N<sub>4</sub> sites and regulate the electronic states of central Ni atoms in Ni<sub>1</sub>-N-C (X) (X = Cl, Br, and I). Significantly, the Ni<sub>1</sub>-N-C (Cl) catalyst, decorated with the most electronegative Cl atoms, exhibits Faradaic efficiency of CO up to 94.7% in electrocatalytic CO<sub>2</sub> reduction, outperforming Ni<sub>1</sub>-N-C (Br) and Ni<sub>1</sub>-N-C (I) catalysts. Moreover, Ni<sub>1</sub>-N-C (Cl) also presents superb performance in Zn-CO<sub>2</sub> battery with ultrahigh CO selectivity and great durability. Theoretical calculations reveal that the axially coordinated Cl atom remarkably facilitates \*COOH intermediate formation on single-atom Ni sites, thereby boosting the CO<sub>2</sub> reduction performance of Ni<sub>1</sub>-N-C (Cl). This work offers a facile strategy to tailor the axial coordination environments of SACs at atomic level and manifests the crucial role of axial coordination microenvironments in catalysis.

## KEYWORDS

metal-organic frameworks, single-atom catalysts, CO<sub>2</sub> electroreduction, axial coordination environment

## 1 Introduction

Single-atom catalysts (SACs), featured with isolated metal atoms dispersed on supports, present unprecedented activity and selectivity for various reactions and have become the frontier of heterogeneous catalysis [1–10]. Single metal atoms in SACs, with extremely high specific surface energy, are usually stabilized via coordination interaction by suitable supports [11, 12]. Thereby, the catalytic performance of SACs is significantly affected by their local coordination environments [13–15]. Taking CO<sub>2</sub> electroreduction as an example, it suffers from low efficiency due to chemical inertia of CO<sub>2</sub> and the competing hydrogen evolution, and the decrease of coordination number for single-atom Ni catalysts has been widely proved to alleviate these challenges [16–18]. On account of this, regulating the coordination environments of SACs precisely will be of great significance for catalytic performance optimization and mechanism investigation [19–22].

To be specific, single metal atoms implanted nitrogen-doped carbon (M-N-C) catalysts, one of the most important SACs, are usually identified to possess porphyrin-like planar M-N<sub>4</sub> sites dominantly [23, 24]. However, the high electron/structure symmetry of M-N<sub>4</sub> sites is unfavorable for the charge transfer during catalysis, hindering the further performance optimization

of M-N-C catalysts [25]. To break the symmetric electronic structure of single-atom metal sites, a series of coordination environment regulation strategies, such as changing coordination numbers and replacing some of the coordinated nitrogen atoms by other heteroatoms, have been developed and greatly enhanced the catalytic performance of M-N-C catalysts [17, 26–28]. Unfortunately, these regulations are all limited in the carbon plane and unavoidably change the properties of carbon support in the meantime [29, 30]. The axial coordination environment regulation, which can be performed out of carbon support plane, should be an ideal way to optimize the catalytic performance of SACs without influencing the properties of carbon support [31]. Although great progress has been achieved in the precise construction of SACs, the axial coordination environment regulation remains a grand challenge [25, 32, 33].

Metal-organic frameworks (MOFs) [34–37], as a class of crystalline porous material with tunable components and structure, exhibit great research prospects in diverse fields [38–43]. In recent years, MOFs have been regarded as one of the most auspicious candidates for the accurate construction of SACs [44, 45]. Thanks to the flexible structure and composition, MOFs also show great advantages on the coordination environment regulation of single-atom sites via one-step pyrolysis [29, 46]. However, due to the uncontrollability of

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pyrolysis process, traditional one-step pyrolysis of MOFs is still difficult to control the axial coordination of SACs accurately [33]. Inspired by post-metallization strategy for the construction of metal porphyrin complexes, the axial coordination atoms can be easily adjusted by selecting metal salts with different anions [47]. Following this synthesis route, the post insertion of single atoms on MOF-derived nitrogen-doped carbon, which can decouple the carbon support preparation and subsequent single metal atom decoration, should be an ideal way to regulate the axial coordination environments of SACs.

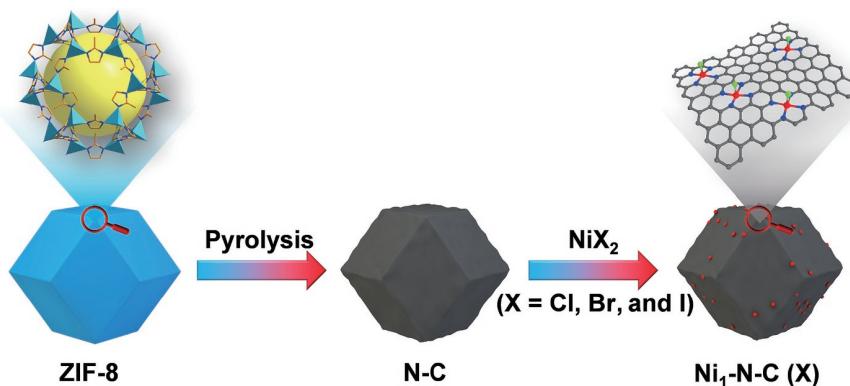
With this in mind, a facile post metal halide modification (PMHM) strategy has been developed to decorate single Ni atoms with different axial halogen atoms (Cl, Br, and I) on the pre-fabricated nitrogen-doped carbon derived from ZIF-8 (ZIF = zeolitic imidazolate framework). During the pyrolysis of ZIF-8, metallic Zn converted from  $Zn^{2+}$  can evaporate away due to the low boiling point, leaving nitrogen-doped porous carbon (N-C). The abundant nitrogen atoms of N-C can serve as anchoring sites to coordinate with  $NiX_2$  salts ( $X = Cl, Br, and I$ ), obtaining a series of single-atom Ni sites coordinated by four nitrogen atoms and an axial halogen atom (denoted as  $Ni_1\text{-}N\text{-}C(X)$ ,  $X = Cl, Br, and I$ ) (Fig. 1). The  $Ni_1\text{-}N\text{-}C(Cl)$  catalyst, with the most electronegative Cl atoms modified in the axial direction, exhibits ultrahigh Faradaic efficiency (FE) of CO up to 94.7%, much superior to  $Ni_1\text{-}N\text{-}C(Br)$  and  $Ni_1\text{-}N\text{-}C(I)$ . Theoretical simulations unveil that the single-atom Ni site with axial Cl atom can accelerate the formation of  $^*\text{COOH}$  and greatly promote the  $\text{CO}_2$  reduction process of  $Ni_1\text{-}N\text{-}C(Cl)$ .

## 2 Results and discussion

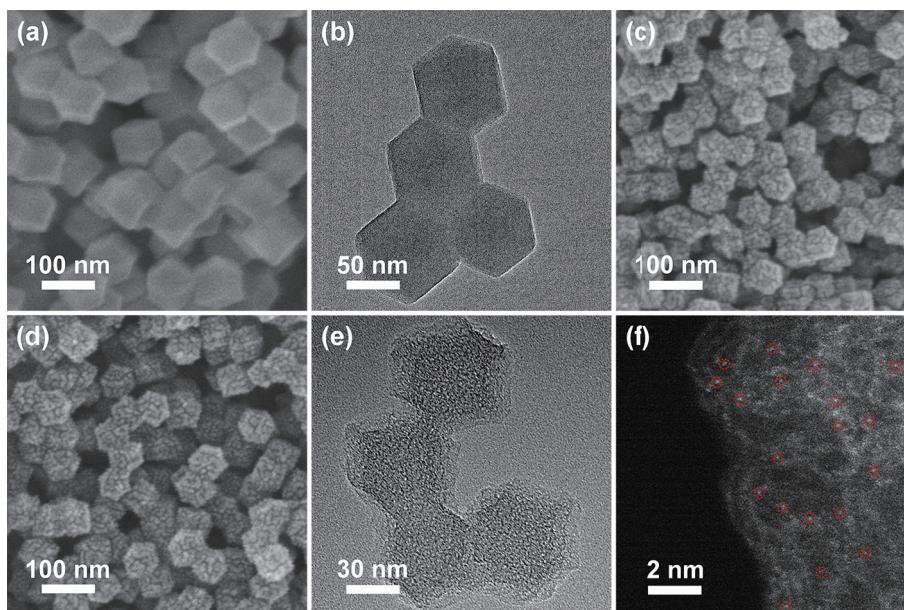
A representative Zn-MOF, ZIF-8 also named as MAF-4 [48, 49], was prepared by self-assembly of  $Zn^{2+}$  and 2-methylimidazole in methanol (Fig. S1 in the Electronic Supplementary Material (ESM)). The typical type-I  $N_2$  sorption isotherms exhibit the microporous characteristic of ZIF-8 with a large surface area of 1,331  $m^2/g$  (Fig. S2 in the ESM). As can be seen from scanning electron microscopy (SEM) and transmission electron microscopy (TEM) images, the as-synthesized ZIF-8 presents a rhombic dodecahedral morphology with an average size of 80 nm (Figs. 2(a) and 2(b)). Through the direct pyrolyzation of ZIF-8 at 900 °C, N-C with well-retained morphology as ZIF-8 can be obtained (Fig. 2(c) and Fig. S3 in the ESM). After being refluxed in the solution of nickel chloride, the N-C with abundant N coordination sites can be easily decorated by Ni atoms, forming the  $Ni_1\text{-}N\text{-}C(Cl)$  catalysts with Ni loading of 0.64 wt.% and N content of 6.25 wt.% (Table S1 in the ESM). Following this method,  $Ni_1\text{-}N\text{-}C(Br)$  and  $Ni_1\text{-}N\text{-}C(I)$ , with comparable Ni loading and N content to  $Ni_1\text{-}N\text{-}C(Cl)$ , have also been fabricated by the adoption of nickel salts with different halogen anions (Table S1 in the ESM).

Taking  $Ni_1\text{-}N\text{-}C(Cl)$  as an example, it shows inherited morphology and particle size of N-C (Figs. 2(c)–2(e) and Fig. S3 in the ESM). No obvious nanocrystals or nanoparticles can be observed in the TEM image of  $Ni_1\text{-}N\text{-}C(Cl)$ , manifesting the atomic dispersion of Ni (Fig. 2(e)). Moreover, the powder X-ray diffraction (XRD) pattern of  $Ni_1\text{-}N\text{-}C(Cl)$  only presents two broad peaks corresponding to the (002) and (101) planes of carbon and no crystalline species of Ni is detected, in accordance with TEM observation (Fig. S4 in the ESM and Fig. 2(e)). Raman spectrum of  $Ni_1\text{-}N\text{-}C(Cl)$  displays two peaks at 1,339 and 1,578  $\text{cm}^{-1}$ , which can be ascribed to the disorder (D band) and graphitic (G band) carbon, respectively (Fig. S5 in the ESM). The intensity ratio of D band to G band ( $I_D/I_G$ ) for  $Ni_1\text{-}N\text{-}C(Cl)$  is calculated to be 1.04, showing the similar graphitization degree to N-C (Fig. S5 in the ESM). Furthermore,  $Ni_1\text{-}N\text{-}C(Cl)$  exhibits a large surface area (633  $m^2/g$ ) with highly porous structure similar to that of N-C, which favors the dispersion of catalytic center atoms and would expedite mass transfer during catalysis (Fig. S6 in the ESM). Meanwhile,  $Ni_1\text{-}N\text{-}C(Br)$  and  $Ni_1\text{-}N\text{-}C(I)$  are also characterized and present similar morphology, graphitization degree, and porous structure to those of  $Ni_1\text{-}N\text{-}C(Cl)$  (Figs. S4–S7 in the ESM and Figs. 2(d) and 2(e)).

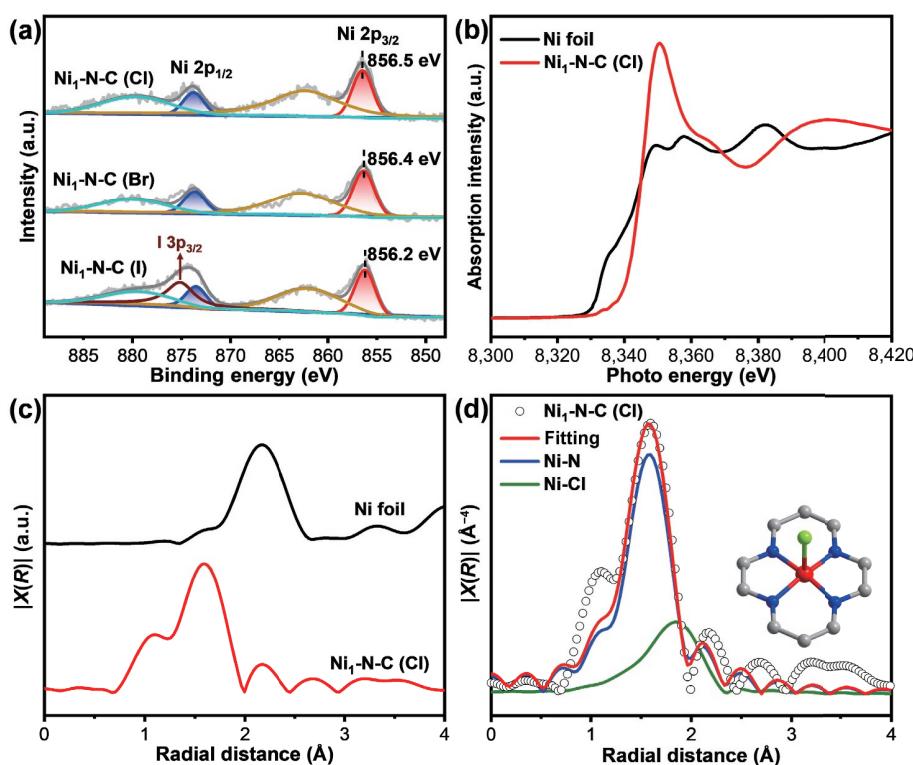
To identify the chemical compositions and element electronic states of  $Ni_1\text{-}N\text{-}C(X)$  ( $X = Cl, Br, and I$ ), X-ray photoelectron spectroscopy (XPS) is further conducted. The  $Ni 2p_{3/2}$  peaks in  $Ni_1\text{-}N\text{-}C(X)$  ( $X = Cl, Br, and I$ ) exhibit the decreased binding energy in the sequence of  $Ni_1\text{-}N\text{-}C(Cl)$ ,  $Ni_1\text{-}N\text{-}C(Br)$ , and  $Ni_1\text{-}N\text{-}C(I)$ , which can be ascribed to the decreasing electronegativity from Cl to I (Fig. 3(a)). This manifests the higher oxidation state of Ni in  $Ni_1\text{-}N\text{-}C(Cl)$  than those in  $Ni_1\text{-}N\text{-}C(Br)$  and  $Ni_1\text{-}N\text{-}C(I)$ . Additionally, it should be noted that the peak at 875.2 eV overlapping with  $Ni 2p_{1/2}$  in  $Ni_1\text{-}N\text{-}C(I)$  catalyst is the signal of  $I 3p_{3/2}$  (Fig. 3(a)) [50]. The N 1s spectra of  $Ni_1\text{-}N\text{-}C(X)$  ( $X = Cl, Br, and I$ ) can be deconvoluted into five characteristic peaks, and the existence of metal-N species, attributed to the empty orbitals of metal atoms accepting the lone pair electrons of N atoms, clearly indicates the formation of Ni-N coordination bonds (Figs. S8(a)–S8(c) and Fig. S9 in the ESM) [51]. Furthermore, the existence of halogen atoms in the three obtained catalysts is also confirmed by XPS measurements. The  $Cl 2p$  XPS spectrum of  $Ni_1\text{-}N\text{-}C(Cl)$  presents a peak at 197.7 eV, manifesting the formation of Ni-Cl coordination (Fig. S8(d) in the ESM) [52]. Moreover, the peaks at 68.0 and 68.9 eV in the  $Br 3d$  XPS spectrum of  $Ni_1\text{-}N\text{-}C(Br)$  and the peaks located at 619.2 and 630.7 eV in the  $I 3d$  spectrum of  $Ni_1\text{-}N\text{-}C(I)$  verify the existence of Ni-Br and Ni-I coordination, respectively (Figs. S8(e) and S8(f) in the ESM) [53, 54]. The aberration-corrected high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) image of  $Ni_1\text{-}N\text{-}C(Cl)$  illustrates isolated bright spots dispersed on



**Figure 1** Illustration for the construction of  $Ni_1\text{-}N\text{-}C(X)$  ( $X = Cl, Br, and I$ ) single-atom catalysts via a PMHM strategy.



**Figure 2** (a) SEM and (b) TEM images of ZIF-8. (c) SEM image of N-C. (d) SEM, (e) TEM, and (f) aberration-corrected HAADF-STEM image of  $\text{Ni}_1\text{-N-C(Cl)}$ .



**Figure 3** (a) The Ni 2p XPS spectra of  $\text{Ni}_1\text{-N-C}(X)$  ( $X = \text{Cl}, \text{Br}$ , and  $\text{I}$ ). (b) Ni K-edge XANES spectra and (c) FT-EXAFS spectra of Ni foil and  $\text{Ni}_1\text{-N-C(Cl)}$ . (d) EXAFS fitting of  $\text{Ni}_1\text{-N-C(Cl)}$  (inset: optimized coordination environment of Ni atom).

the carbon support, manifesting the atomic dispersion of Ni (Fig. 2(f)). The above results initially confirm the formation of single-atom Ni sites coordinated by nitrogen and the corresponding halogen atoms.

Furthermore, X-ray absorption spectroscopy (XAS) is employed to give more clear evidence on the coordination environments and electronic states of Ni atoms in  $\text{Ni}_1\text{-N-C}(X)$  ( $X = \text{Cl}, \text{Br}$ , and  $\text{I}$ ) catalysts. It can be observed that the near-edge absorption energy of  $\text{Ni}_1\text{-N-C(Cl)}$  shifts to higher value compared with Ni foil in the Ni K-edge X-ray absorption near-edge structure (XANES) spectrum, indicating the positive valence state of Ni in  $\text{Ni}_1\text{-N-C(Cl)}$  (Fig. 3(b)). The Fourier transform-extended X-ray absorption fine structure (FT-EXAFS) curve of  $\text{Ni}_1\text{-N-C(Cl)}$  exhibits the dominated peak centered at  $\sim 1.60 \text{ \AA}$ , which could be

attributed to the Ni-N/Cl coordination (Fig. 3(c)). In addition, the absence of Ni-Ni scattering path (around  $2.18 \text{ \AA}$ ) suggests the atomic dispersion of Ni atoms in  $\text{Ni}_1\text{-N-C(Cl)}$ , agreeing well with the results of HAADF-STEM observation (Figs. 3(c) and 2(f)). Moreover, the EXAFS curve fitting is further performed to acquire the coordination configuration of Ni atoms of  $\text{Ni}_1\text{-N-C(Cl)}$ . The best fitting result demonstrates that the Ni atom is coordinated with four in-plane N atoms and an axial Cl atom (Fig. 3(d) and Table S2 in the ESM). In addition, the XAS analysis has also been performed on  $\text{Ni}_1\text{-N-C(Br)}$  and  $\text{Ni}_1\text{-N-C(I)}$ . The Ni K-edge XANES spectra of  $\text{Ni}_1\text{-N-C(Br)}$  and  $\text{Ni}_1\text{-N-C(I)}$  suggest the positive valence states of Ni in the two samples (Figs. S10(a) and S10(b) in the ESM). Furthermore, the best EXAFS curve fitting results for  $\text{Ni}_1\text{-N-C(Br)}$  and  $\text{Ni}_1\text{-N-C(I)}$  show their similar

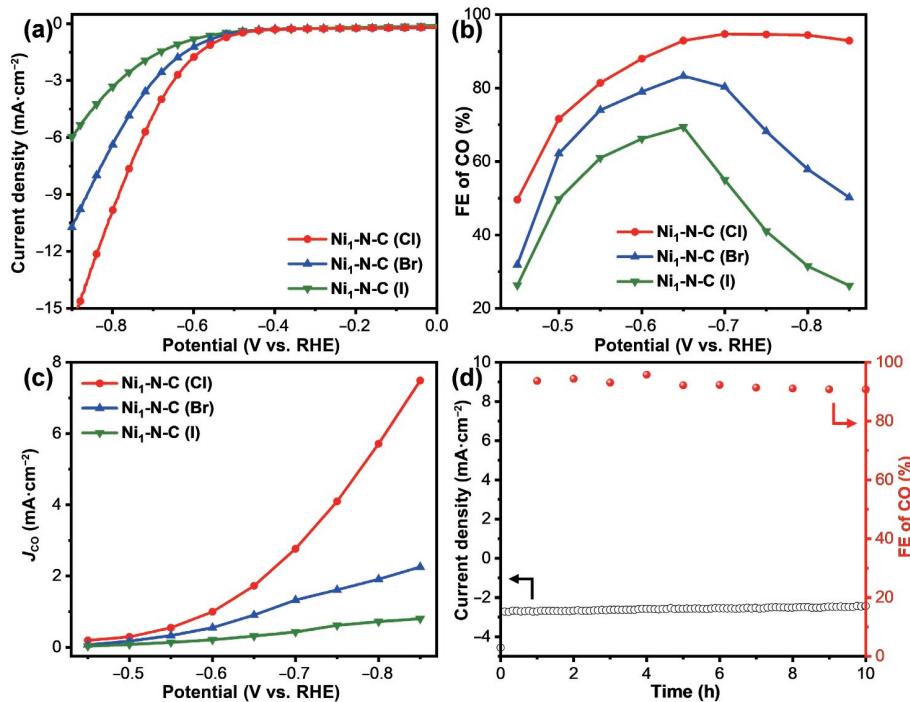
coordination configurations to  $\text{Ni}_1\text{-N-C}(\text{Cl})$ , in which Ni atom is coordinated with four in-plane N atoms and an axial Br or I atom (Fig. S11 and Table S2 in the ESM).

Inspired by the results above, electrocatalytic  $\text{CO}_2$  reduction measurements have been conducted using  $\text{Ni}_1\text{-N-C}(\text{X})$  ( $\text{X} = \text{Cl}, \text{Br}$ , and  $\text{I}$ ) as electrocatalysts. The linear sweep voltammetry (LSV) curves indicate that the current densities of  $\text{Ni}_1\text{-N-C}(\text{X})$  ( $\text{X} = \text{Cl}, \text{Br}$ , and  $\text{I}$ ) and N-C in  $\text{CO}_2$  atmosphere are all higher than those in argon, manifesting their favorable activity for  $\text{CO}_2$  reduction reaction ( $\text{CO}_2\text{RR}$ ) (Figs. S12–S15 in the ESM). In addition, the current response of  $\text{Ni}_1\text{-N-C}(\text{Cl})$  in  $\text{CO}_2$  atmosphere is much higher than those of  $\text{Ni}_1\text{-N-C}(\text{Br})$ ,  $\text{Ni}_1\text{-N-C}(\text{I})$ , and N-C (Fig. 4(a)) and Fig. S15 in the ESM). When tested at constant potentials,  $\text{Ni}_1\text{-N-C}(\text{Cl})$  shows the supreme CO FE in all potentials ranging from  $-0.45$  to  $-0.85$  V (vs. RHE) (Fig. 4(b)). Significantly,  $\text{Ni}_1\text{-N-C}(\text{Cl})$  affords a maximum CO FE of 94.7% at  $-0.7$  V, outperforming those of  $\text{Ni}_1\text{-N-C}(\text{Br})$  (83.3% at  $-0.65$  V),  $\text{Ni}_1\text{-N-C}(\text{I})$  (69.4% at  $-0.65$  V), and N-C (57.1% at  $-0.55$  V) (Fig. 4(b)) and Fig. S16 in the ESM). For all tested catalysts, CO and  $\text{H}_2$  are the main products, and no liquid products are identified from the  $^1\text{H}$  nuclear magnetic resonance ( $^1\text{H}$  NMR) results (Fig. 4(b)) and Figs. S16–S18 in the ESM). Additionally,  $\text{Ni}_1\text{-N-C}(\text{Cl})$  also exhibits a superior CO partial current density ( $J_{\text{CO}}$ ) to those of  $\text{Ni}_1\text{-N-C}(\text{Br})$  and  $\text{Ni}_1\text{-N-C}(\text{I})$ , further manifesting excellent  $\text{CO}_2\text{RR}$  performance of  $\text{Ni}_1\text{-N-C}(\text{Cl})$  (Fig. 4(c)). The Tafel slope of  $\text{Ni}_1\text{-N-C}(\text{Cl})$  is 194 mV/dec, lower than those of  $\text{Ni}_1\text{-N-C}(\text{Br})$  (209 mV/dec) and  $\text{Ni}_1\text{-N-C}(\text{I})$  (252 mV/dec), revealing the fastest kinetics of  $\text{Ni}_1\text{-N-C}(\text{Cl})$  for  $\text{CO}_2\text{RR}$  (Fig. S19 in the ESM). The double-layer capacitances ( $C_{\text{dl}}$ ) are determined to be 28.9, 28.4, and 28.0 mF/cm<sup>2</sup> for  $\text{Ni}_1\text{-N-C}(\text{Cl})$ ,  $\text{Ni}_1\text{-N-C}(\text{Br})$ , and  $\text{Ni}_1\text{-N-C}(\text{I})$ , respectively, indicating their similar electrochemical active surface area (ECSA) (Fig. S20 in the ESM). The similar ECSA as well as Ni loading of different samples further prove that the catalytic performance difference of  $\text{Ni}_1\text{-N-C}(\text{Cl})$ ,  $\text{Ni}_1\text{-N-C}(\text{Br})$ , and  $\text{Ni}_1\text{-N-C}(\text{I})$  is actually caused by the different axial halide atoms (Fig. S20 and Table S1 in the ESM). Finally, the stability measurement of best-performed  $\text{Ni}_1\text{-N-C}(\text{Cl})$  is conducted at  $-0.7$  V, and it can be seen that the FE of CO and current densities can be well retained

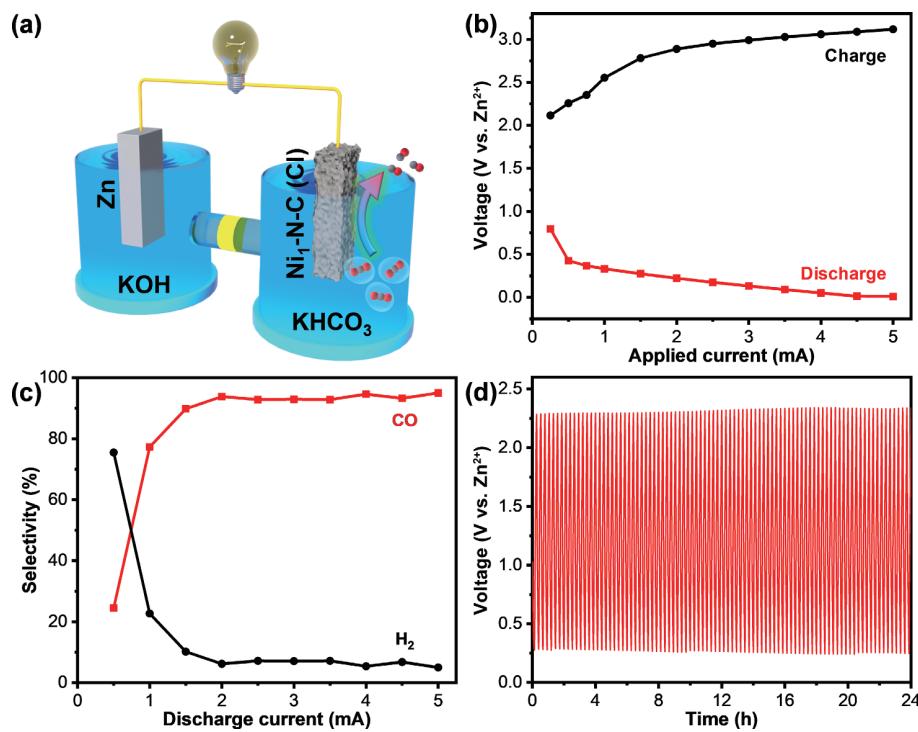
during 10 h, demonstrating the long-term stability of  $\text{Ni}_1\text{-N-C}(\text{Cl})$  for  $\text{CO}_2\text{RR}$  (Fig. 4(d)).

Given the superb  $\text{CO}_2\text{RR}$  performance of  $\text{Ni}_1\text{-N-C}(\text{Cl})$ , it is further employed as a cathode catalyst in the rechargeable Zn- $\text{CO}_2$  battery [51, 55]. The  $\text{Ni}_1\text{-N-C}(\text{Cl})$  is immersed in 0.8 M  $\text{CO}_2$ -saturated  $\text{KHCO}_3$  electrolyte. The Zn foil, serving as the anode, is put in 0.8 M KOH solution containing 0.02 M  $\text{Zn}(\text{CH}_3\text{COO})_2$  (Fig. 5(a)). During the discharge process, the cathode reaction is  $\text{CO}_2$  reduction and the Zn anode converts into  $\text{Zn}(\text{OH})_4^{2-}$  [55]. In the charge process, oxygen forms on the cathode and the anode reaction is the deposition of Zn [55]. As shown in the discharge and charge voltage profiles,  $\text{Ni}_1\text{-N-C}(\text{Cl})$  exhibits a discharge potential of 0.426 V at 0.5 mA, which still maintains at 0.175 V when 2.5 mA discharge current is applied (Fig. 5(b)). In the meantime, a charge potential of 2.257 V results from a charge current of 0.5 mA, which increases slowly after 1 mA (Fig. 5(b)). Meanwhile, the selectivity of CO reaches 93.8% at 2 mA during the discharge process and can be well retained in a wide range of discharge current, indicating the excellent  $\text{CO}_2$  reduction performance of  $\text{Ni}_1\text{-N-C}(\text{Cl})$  (Fig. 5(c)). Moreover,  $\text{Ni}_1\text{-N-C}(\text{Cl})$  displays stable voltage under consecutively discharge/charge at 2.0 mA/cm<sup>2</sup> for 24 h, suggesting the excellent stability of the Zn- $\text{CO}_2$  cell (Fig. 5(d)).

To uncover the influence of axial coordination environments on the  $\text{CO}_2\text{RR}$  performance of single Ni atoms, density functional theory (DFT) calculations are further implemented (Fig. S21 in the ESM). As shown in the charge density difference in the three samples, more electrons transfer from Ni to the axial Cl than Br and I, which is consistent with the results based on Bader charge analysis (Fig. S22 and Table S3 in the ESM). The different electron delocalization degree of Ni atoms caused by halogen atoms with different electronegativity will also affect the reaction energy barriers. To prove this, the free-energy change ( $\Delta G$ ) for  $\text{CO}_2$  reduction to CO has been calculated (Fig. S23 in the ESM). Usually, the formation of \*COOH is regarded as the rate-limiting step [56–58]. The  $\Delta G$  for the \*COOH formation on  $\text{Ni}_1\text{-N-C}(\text{Cl})$  is determined to be 1.78 eV, smaller than those on  $\text{Ni}_1\text{-N-C}(\text{Br})$  (1.86 eV) and  $\text{Ni}_1\text{-N-C}(\text{I})$  (1.90 eV), demonstrating the lowest



**Figure 4** (a) LSV curves, (b) CO FE, and (c) CO partial current densities of  $\text{Ni}_1\text{-N-C}(\text{X})$  ( $\text{X} = \text{Cl}, \text{Br}$ , and  $\text{I}$ ). (d) Stability measurement of  $\text{Ni}_1\text{-N-C}(\text{Cl})$  at  $-0.7$  V vs. RHE in  $\text{CO}_2$ -saturated 0.5 M  $\text{KHCO}_3$ .



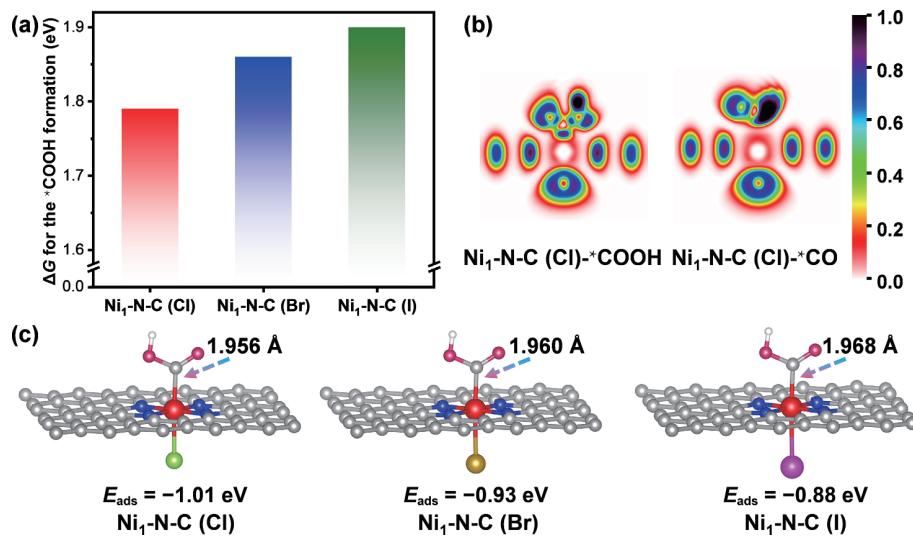
**Figure 5** (a) Schematic illustration of the Zn-CO<sub>2</sub> battery with Ni<sub>1</sub>-N-C (Cl) cathode. (b) Discharge and charge voltage profiles for Zn-CO<sub>2</sub> battery. (c) Selectivity of Ni<sub>1</sub>-N-C (Cl) for CO and H<sub>2</sub> under different discharge current densities. (d) Galvanostatic discharge-charge cycling curves at 2.0 mA/cm<sup>2</sup> for 24 h.

energy barrier of Ni<sub>1</sub>-N-C (Cl) for CO<sub>2</sub>RR (Fig. 6(a) and Fig. S24 in the ESM). Electron localization function (ELF) images exhibit the high degree of electron localization between single-atom Ni in Ni<sub>1</sub>-N-C (Cl) and \*COOH intermediate, while there is little localized electron between single-atom Ni and \*CO, suggesting the strong interaction of Ni<sub>1</sub>-N-C (Cl) with \*COOH and weak interaction with \*CO (Fig. 6(b)). Moreover, Ni<sub>1</sub>-N-C (Cl) possesses the shortest distance between absorbed \*COOH and central Ni atom, and the largest \*COOH adsorption energy, further supporting the superior performance of Ni<sub>1</sub>-N-C (Cl) for CO<sub>2</sub>RR, in accordance with the experimental results (Fig. 6(c)).

### 3 Conclusions

In summary, a straightforward and facile PMHM method is put forward to fabricate single-atom Ni catalysts coordinated with nitrogen and halogen atoms. By simply adjusting the halogen

anions of the nickel salts in the single atom decoration process, a series of single-atom Ni catalysts with different axial coordinated halogen atoms (Ni<sub>1</sub>-N-C (X), X = Cl, Br, and I) are rationally constructed. The axial coordinated halogen atoms, with different electronegativity, can significantly regulate the electronic states of the single metal atoms. When performed as an electrocatalyst for CO<sub>2</sub> reduction, the obtained Ni<sub>1</sub>-N-C (Cl) catalyst, with the single atom Ni coordinated by four N atoms and an axial Cl atom, manifests outstanding CO<sub>2</sub>RR performance with ultrahigh CO FE up to 94.7%, far surpassing Ni<sub>1</sub>-N-C (Br) and Ni<sub>1</sub>-N-C (I). Moreover, when employed as a cathode catalyst in the Zn-CO<sub>2</sub> battery, Ni<sub>1</sub>-N-C (Cl) also exhibits ultrahigh selectivity for CO and long-term stability. Theoretical calculations reveal that Ni<sub>1</sub>-N-C (Cl) with the most electronegative Cl coordination in the axial direction could reduce the formation barrier of \*COOH, thus boosting the CO<sub>2</sub>RR performance. This work presents a facile and universal strategy for the regulation of coordination environments



**Figure 6** (a) The  $\Delta G$  for the \*COOH formation in Ni<sub>1</sub>-N-C (X) (X = Cl, Br, and I) catalysts. (b) ELF images of Ni<sub>1</sub>-N-C (Cl) with \*COOH and \*CO adsorption. (c) The DFT-optimized configurations of Ni<sub>1</sub>-N-C (X) (X = Cl, Br, and I) with \*COOH adsorption and corresponding adsorption energy.

and brings great opportunities for the catalytic performance optimization of single-atom catalysts.

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**Electronic Supplementary Material:** Supplementary material (materials and instrumentation, experimental section, electrochemical measurements, the details for X-ray adsorption spectra and Zn-CO<sub>2</sub> battery, computational details, XRD results, N<sub>2</sub> sorption isotherm, Raman spectra, SEM and TEM images, XPS spectra, XAFS results, electrochemical results, <sup>1</sup>H NMR spectra, DFT calculation results, and the element contents for N and Ni) is available in the online version of this article at <https://doi.org/10.1007/s12274-022-4467-3>.

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