

# Embedding CoS<sub>2</sub> nanoparticles in N-doped carbon nanotube hollow frameworks for enhanced lithium storage properties

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## ABSTRACT

The construction of metal sulfides-carbon nanocomposites with a hollow structure is highly attractive for various energy storage and conversion technologies. Herein, we report a facile two-step method for preparing a nanocomposite with CoS<sub>2</sub> nanoparticles in N-doped carbon nanotube hollow frameworks (NCNTFs). Starting from zeolitic imidazolate framework-67 (ZIF-67) particles, *in situ* reduced metallic cobalt nanocrystals expedite the formation of the hierarchical hollow frameworks from staggered carbon nanotubes via a carbonization process. After a follow-up sulfidation reaction with sulfur powder, the embedded cobalt crystals are transformed into CoS<sub>2</sub> nanoparticles. Benefitting from the robust hollow frameworks made of N-doped carbon nanotubes and highly active CoS<sub>2</sub> ultrafine nanoparticles, this advanced nanocomposite shows greatly enhanced lithium storage properties when evaluated as an electrode for lithium-ion batteries. Impressively, the resultant CoS<sub>2</sub>/NCNTF material delivers a high specific capacity of ~937 mAh·g<sup>-1</sup> at a current density of 1.0 A·g<sup>-1</sup> with a cycle life longer than 160 cycles.

## 1 Introduction

Lithium-ion batteries (LIBs) are promising electrical energy-storage devices for addressing the imminent shortage of fossil fuels and growing environmental concerns [1–3]. LIBs based on the traditional carbonaceous electrode materials are approaching their theoretical power/energy density limit [4, 5]. Therefore, the elaborate exploration and rational design of new electrode materials are crucial for realizing new

generations of high-performance LIBs [6–8]. Among numerous candidates, metal sulfides have received growing attention as advanced anodes owing to their high theoretical capacity arising from their rich stoichiometric compositions and better electrical conductivity compared with their corresponding metal oxide counterparts [9–14]. However, their practical applications are hampered by pronounced volume expansion and contraction during the repetitive charging/discharging processes and sluggish ion/

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electron-transport kinetics, which result in a poor cycle life and unsatisfactory rate performance [15–17].

Hollow-structured micro/nanomaterials have proven their superiority as smart architectures for LIBs by providing extra space to alleviate the dramatic (de)lithiation strain and maintain the integrity of electrodes [18–22]. The utilization of hollow structures in combination with conductive carbon materials is promising for ameliorating the conductivity of metal sulfides and buffering volumetric changes during electrochemical reactions [23–27]. For example, Yu et al. reported the synthesis of a unique hybrid hollow structure by assembling ultrathin MoS<sub>2</sub> nanosheets on N-doped carbon shells, which exhibited a high capacity, long cycle life, and superior rate performance for LIBs [23]. Zou et al. prepared hollow polyhedral composites of porous carbon/short carbon nanotube-stabilized CoS nanoparticles through a one-step simultaneous pyrolysis and sulfidation treatment of zeolitic imidazolate framework-67 (ZIF-67) precursors [24]. Owing to the strong coupling effect of the unique hybrids, the novel hollow structures exhibited superior electrochemical properties as an anode material for LIBs. Notwithstanding these advances, many carbon materials in previously reported nanocomposites are microporous, with a poor graphitic degree, which is not beneficial for ion and electron transport. In comparison, a hierarchical hollow framework involving interconnected highly conductive carbon nanotube networks might be an ideal host for loading metal sulfides, not only offering abundant active sites and a porous structure for fast charge transport and mass transfer but also helping to buffer the volume expansion/contraction upon cycling [28].

Herein, we report a two-step metal-organic-framework-engaged method for preparing CoS<sub>2</sub> nanoparticles in hollow carbonaceous N-doped carbon nanotube frameworks (NCNTFs). The hierarchical NCNTFs are converted from the thermal decomposition of ZIF-67 crystals in the presence of hydrogen as a conductive matrix. Metallic cobalt nanoparticles are uniformly embedded within the three-dimensional carbon nanotube frameworks and transformed into CoS<sub>2</sub> ultrafine nanocrystals after a sulfidation reaction with sulfur powder at an elevated temperature. Encouragingly, the as-obtained novel CoS<sub>2</sub>/NCNTF

nanocomposite exhibits a remarkably enhanced lithium storage performance as an anode material for LIBs.

## 2 Experimental

### 2.1 Synthesis of ZIF-67 polyhedrons

All the chemicals were directly used as received. ZIF-67 polyhedrons were synthesized using a method that we previously reported [29]. In a typical procedure, 1 mmol of Co(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O and 4 mmol of 2-methylimidazole were dissolved in 25 mL of methanol. Then, the solution of 2-methylimidazole was quickly poured into a solution of Co(NO<sub>3</sub>)<sub>2</sub> under vigorous stirring. The mixed solution was aged for one day at the ambient temperature. Finally, the purple precipitate was washed with ethanol three times by centrifugation and dried for future use.

### 2.2 Synthesis of Co/NCNTFs

The ZIF-67 particles were heated to 350 °C and maintained for 1.5 h under an Ar/H<sub>2</sub> flow (95%/5% in volume ratio). Then, the temperature was further increased to 700 °C and maintained for 3.5 h. The ramp rate was fixed at 2 °C·min<sup>-1</sup> for these two processes.

### 2.3 Synthesis of CoS<sub>2</sub>/NCNTFs

The obtained Co/NCNTF materials were mixed with sulfur powder at a mass ratio of 1:2. Then, the mixture was heated to 300 °C at a ramp rate of 10 °C·min<sup>-1</sup> and maintained for 2 h in a tube furnace under a N<sub>2</sub> flow. Finally, the furnace was cooled to room temperature naturally.

### 2.4 Materials characterization

The morphologies and structures were characterized by field-emission scanning electron microscopy (FESEM; JEOL, JEM-6701F, 5 kV) and transmission electron microscopy (TEM; JEOL, JEM-2010, 200 kV). Powder X-ray diffraction (XRD) patterns were recorded using a Bruker diffractometer with Cu K $\alpha$  radiation (D2 phaser, Cu K $\alpha$ ,  $\lambda = 1.5406 \text{ \AA}$ ) to examine the crystallographic structure. The composition of the samples was analyzed using an energy-dispersive X-ray spectroscope attached to the FESEM instrument. The

$N_2$  sorption isotherms were examined at the liquid-nitrogen temperature. The Raman spectra of the dried samples were collected using a Renishaw System 1000 micro-Raman spectroscope. Thermogravimetric analysis (TGA, Perkin Pyris Diamond) was performed under an air flow ( $200 \text{ mL}\cdot\text{min}^{-1}$ ) with a ramp rate of  $10 \text{ }^\circ\text{C}\cdot\text{min}^{-1}$ .

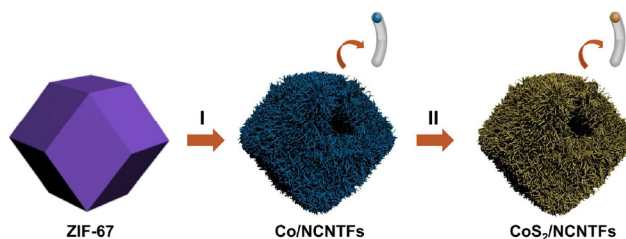
## 2.5 Electrochemical measurements

The working electrode was composed of the active material, carbon black (Ketjenblack), and a polymer binder (polyvinylidene fluoride) at a weight ratio of 70:20:10. This mixture was pressed onto Cu foil and dried at  $70 \text{ }^\circ\text{C}$  for 24 h. The mass loading of electroactive materials was  $\sim 1.0 \text{ mg}\cdot\text{cm}^{-2}$ . Pure lithium foil was used as both the counter electrode and the reference electrode. The electrolyte was composed of  $\text{LiPF}_6$  (1.0 M) in a 50:50 (w/w) mixture of ethylene carbonate and diethyl carbonate. Cell assembly was performed in an Ar-filled glovebox with concentrations of moisture and oxygen below 1.0 ppm. Galvanostatic charge/discharge tests were performed using a NEWARE battery tester with a cut-off voltage window of 0.05–3.0 V versus  $\text{Li}/\text{Li}^+$ .

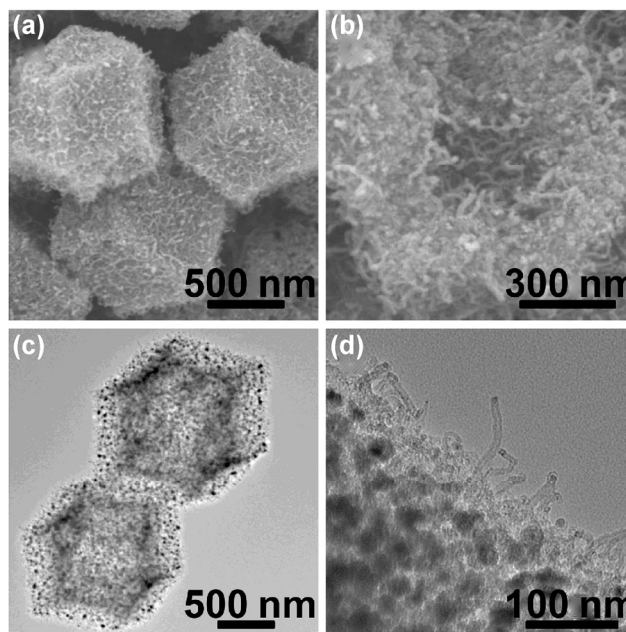
## 3 Results and discussion

Figure 1 schematically illustrates the synthesis of the hierarchical  $\text{CoS}_2/\text{NCNTFs}$ . Well-defined ZIF-67 polyhedrons prepared via a precipitation method are selected as the starting material. Next, a facile thermal treatment in the  $\text{Ar}/\text{H}_2$  atmosphere transforms the solid ZIF-67 particle into a hybrid structure of  $\text{Co}/\text{NCNTFs}$ . During the heating process in the strongly reducing gas, *in situ* generated elemental Co nanoparticles are highly active, catalyzing the growth of CNTs using carbon sources derived from organic ligands in ZIF-67 [30–32]. After a follow-up sulfidation reaction with sulfur, the confined Co particles are converted into ultrafine  $\text{CoS}_2$  nanocrystals embedded within the hierarchical carbon nanotube framework.

After the annealing treatment in  $\text{Ar}/\text{H}_2$ , the morphology and microstructure of the  $\text{Co}/\text{NCNTF}$  products are examined by FESEM. As shown in Fig. 2(a), the as-obtained composite particles are quite uniform with a rhombic dodecahedron shape, which is inherited



**Figure 1** Schematic of the formation process of  $\text{CoS}_2/\text{NCNTFs}$ .



**Figure 2** FESEM images of (a)  $\text{Co}/\text{NCNTFs}$  and (b) a broken particle. TEM images of (a)  $\text{Co}/\text{NCNTFs}$  and (d) the hierarchical shell.

from the ZIF-67 precursor (Fig. S1 in the Electronic Supplementary Material (ESM)). In contrast to its solid precursor with a smooth surface, the  $\text{Co}/\text{NCNTF}$  hybrid is rough on surface, which is entwined by numerous interlaced CNTs. Moreover, the internal void space of the framework with a shell thickness of  $\sim 300 \text{ nm}$  is clearly revealed by the broken  $\text{Co}/\text{NCNTF}$  particle shown in Fig. 2(b). The hollow interior and detailed geometrical structure of the as-synthesized hierarchical structures are further elucidated by TEM observation in Fig. 2(c). Figure 2(d) shows a partially enlarged view of the composite shell, confirming the existence of Co nanoparticles embedded at the tips of CNTs. The corresponding XRD pattern (Fig. S2(a) in the ESM) corroborates the complete transformation from ZIF-67 to  $\text{Co}/\text{NCNTFs}$ . Compared with metallic Co, which has distinct diffraction peaks, the signals of

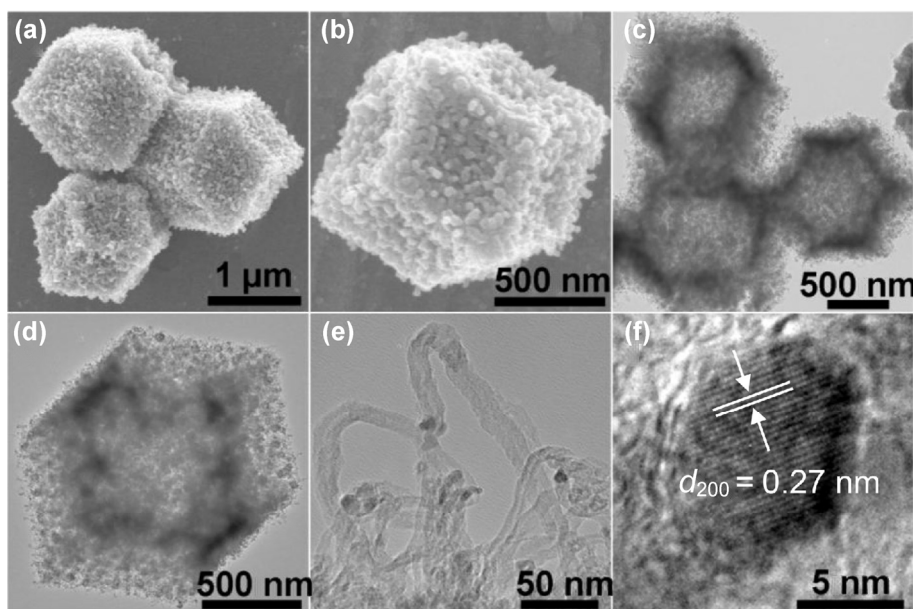


the carbon nanotubes are relatively weak. The energy-dispersive X-ray spectroscopy (EDX) spectrum (Fig. S2(b) in the ESM) confirms the large amount of Co in the nanocomposite. Additionally, the EDX results indicate the existence of nitrogen species, which are beneficial for modifying the chemical and electronic properties of the carbon host [13, 23, 27]. The Raman spectrum of the Co/NCNTF product provides information about the characteristic G and D bands of the carbon material (Fig. S3 in the ESM), which are related to the  $sp^2$ -carbon and disordered or defect carbon due to the microstructure rearrangement and the incorporation of N atoms during the transformation [31, 33]. As indicated by the Raman spectrum, the peak intensity of the G band is slightly higher than that of the D band, indicating a relatively high graphenic order.

After the sulfidation reaction with sulfur, the Co particles within the Co/NCNTF composite are successfully converted into  $CoS_2$ . All the visible Bragg peaks in the XRD spectrum (Fig. S4(a) in the ESM) for the final product are perfectly indexed to  $CoS_2$  (JCPDS card No.: 03-0772). The EDX spectrum confirms the existence of a large amount of sulfur species with a Co/S atomic ratio of  $\sim 0.49$  (Fig. S4(b) in the ESM). The FESEM image in Fig. 3(a) elucidates the well-retained overall polyhedral morphology of the materials, indicating their robust structural stability. A magnified

view of an individual particle (Fig. 3(b)) demonstrates that the exterior CNTs can survive repetitive annealing without apparent agglomeration. TEM examinations (Figs. 3(c) and 3(d)) confirm that the sulfurized samples retain the structural features of the Co/NCNTFs well. Moreover, highly uniform hollow structures are observed, and the inner cavities are unambiguously revealed by the sharp contrast between the hierarchical  $CoS_2$ /NCNTF shell and the hollow interior. A closer TEM view (Fig. 3(e)) shows that these CNTs have a length of several hundred nanometers with an ultrafine  $CoS_2$  particle enclosed at the tip. Consistent with the XRD analysis, the high-resolution TEM (HRTEM) image in Fig. 3(f) verifies that these  $CoS_2$  crystals are crystalline with an inter-planar distance of  $\sim 0.27$  nm corresponding to the (200) plane of  $CoS_2$ . TGA is employed to determine the amount of  $CoS_2$  in the  $CoS_2$ /NCNTF product. According to the TGA data (Fig. S5 in the ESM), the  $CoS_2$  content in the sample is estimated to be 46 wt.% [31]. Owing to its hollow structure and highly porous shell, the hierarchical  $CoS_2$ /NCNTF composite sample exhibits a large specific Brunauer–Emmett–Teller surface area of  $110.4$   $m^2 \cdot g^{-1}$  (Fig. S6 in the ESM). The pore-size distribution curve indicates that the majority of these pores range from 2 to 10 nm in size.

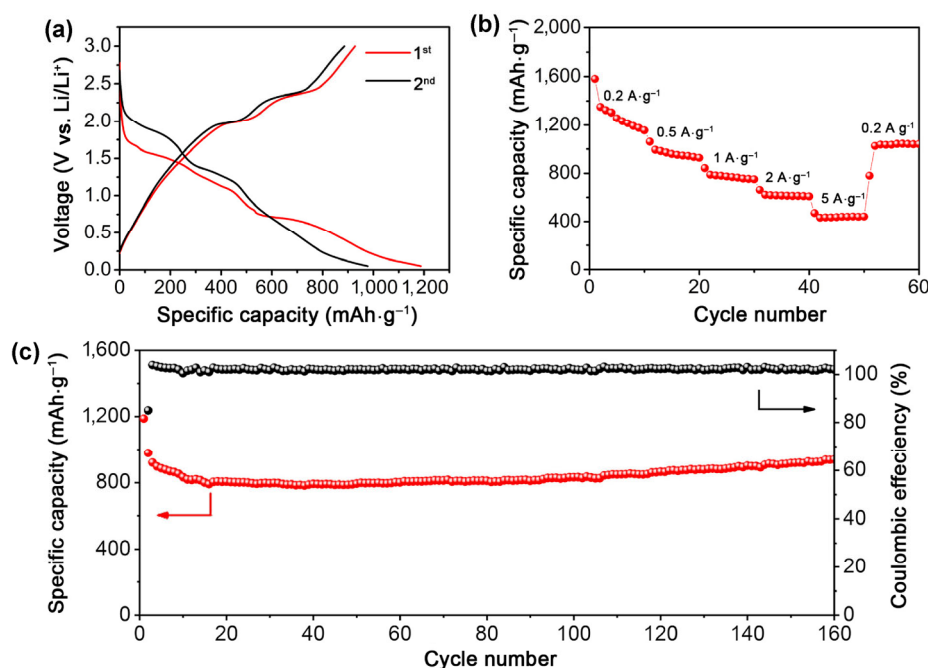
Next, we perform an electrochemical evaluation



**Figure 3** (a) and (b) FESEM, (c)–(e) TEM, and (f) HRTEM images of  $CoS_2$ /NCNTFs.

of the  $\text{CoS}_2/\text{NCNTFs}$  as an anode material for LIBs. Figure 4(a) presents the charge–discharge voltage profiles for the first two cycles at a constant current density of  $1.0 \text{ A}\cdot\text{g}^{-1}$ . The first discharge and charge specific capacities are approximately  $1,187$  and  $927 \text{ mAh}\cdot\text{g}^{-1}$ , respectively, leading to an initial Coulombic efficiency (CE) of  $\sim 78\%$ . The irreversible capacity could be due to the formation of the solid–electrolyte interface film and the decomposition of the electrolyte, which is a common drawback of nanostructured anodes. In the first discharge curve, three discernable plateaus are observed. The voltage plateaus in the ranges of  $1.8\text{--}1.6$  and  $1.5\text{--}1.2 \text{ V}$  are related to the insertion of a small amount of  $\text{Li}^+$  ions and the displacement reaction of  $\text{Li}_2\text{S}$  and  $\text{Co}$ , respectively [13, 22]. During the second cycle, small voltage shifts for these two peaks towards higher potentials are observed, which are mainly attributed to the structural variation of the electrode materials. The representative cyclic-voltammetry (Fig. S7 in the ESM) curves of the composite electrode agree well with the plateaus of the galvanostatic-discharge curves, confirming the multi-step electrochemical processes of the  $\text{CoS}_2$ -based electrode. Figure 4(b) shows the cycling stability of

the  $\text{CoS}_2/\text{NCNTF}$  electrode tested by galvanostatic measurements at different current rates. The reversible discharge capacities of the as-prepared electrode are approximately  $1,191$ ,  $952$ ,  $779$ , and  $615 \text{ mAh}\cdot\text{g}^{-1}$  at current densities of  $0.2$ ,  $0.5$ ,  $1.0$ , and  $2.0 \text{ A}\cdot\text{g}^{-1}$ , respectively. Even at a relatively high current density of  $5.0 \text{ A}\cdot\text{g}^{-1}$ , the electrode delivers a specific capacity as high as  $439 \text{ mAh}\cdot\text{g}^{-1}$ , demonstrating its excellent high-rate capability. Importantly, the capacity recovers to  $1,040 \text{ mAh}\cdot\text{g}^{-1}$  when the current density returns to  $0.2 \text{ A}\cdot\text{g}^{-1}$ . Notably, the  $\text{CoS}_2/\text{NCNTF}$  electrode also exhibits remarkable cycling stability at a constant charge–discharge current density of  $1.0 \text{ A}\cdot\text{g}^{-1}$ . As shown in Fig. 4(c), the discharge capacity is quickly stabilized from the second cycle onwards and maintained at a high value of  $\sim 937 \text{ mAh}\cdot\text{g}^{-1}$  after 160 cycles. The CE for the  $\text{CoS}_2/\text{NCNTF}$  composite electrode is close to  $100\%$  after the first cycle. Overall, this performance is superior or comparable to that of many other cobalt sulfides-based anodes (Table S1 in the ESM) [13, 18, 22, 24, 34–36]. The remarkable performance of  $\text{CoS}_2/\text{NCNTFs}$  is attributed to their unique structural and compositional features. Specifically, the construction of ultrafine  $\text{CoS}_2$  nanoparticles embedded in the carbon



**Figure 4** Electrochemical evaluation of the  $\text{CoS}_2/\text{NCNTF}$  electrode. (a) Galvanostatic charge–discharge voltage profiles at  $1.0 \text{ A}\cdot\text{g}^{-1}$ . (b) Rate-capability test at various current densities; (c) cycling performance at  $1.0 \text{ A}\cdot\text{g}^{-1}$  and the corresponding CE. All measurements are conducted in the voltage range of  $0.05\text{--}3.0 \text{ V}$  versus  $\text{Li}/\text{Li}^+$ .

frameworks not only enables a short diffusion distance for the fast diffusion of Li<sup>+</sup> ions but also provides sufficient contact between the active material and the electrolyte for the rapid charge-transfer reaction. Additionally, the hollow architectures composed of interconnected CNTs provide sufficient space for effectively withstanding large volume variations upon cycling, resulting in good structural integrity.

## 4 Conclusions

We report the rational design and synthesis of CoS<sub>2</sub> nanoparticles embedded in NCNTFs via the direct pyrolysis of metal-organic frameworks and a subsequent sulfidation treatment. The resultant interconnected carbonaceous frameworks provide high conductivity and structural stability for well-dispersed CoS<sub>2</sub> ultrafine nanoparticles. Owing to their prominent structural and compositional advantages, the as-prepared hierarchically structured CoS<sub>2</sub>-carbon nanocomposite manifests enhanced lithium storage properties with high specific capacities at various current densities and an excellent cycling stability as an anode material for lithium-ion batteries.

**Electronic Supplementary Material:** Supplementary material (FESEM images, TEM images, XRD patterns, TGA curve, Raman spectroscopy measurement, N<sub>2</sub> adsorption–desorption isotherms, and additional electrochemical measurements) is available in the online version of this article at <http://dx.doi.org/10.1007/s12274-016-1394-1>

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