



Visible Light-Responsive N-Doped TiO₂ Photocatalysis: Synthesis, Characterizations, and Applications

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Abstract

Photocatalysis based on semiconductors has recently been receiving considerable research interest because of its extensive applications in environmental remediation and renewable energy generation. Various semiconductor-based materials that are vital to solar energy utilization have been extensively investigated, among which titanium oxide (TiO₂) has attracted considerable attention because of its exceptional physicochemical characteristics. However, the sluggish responsiveness to visible light in the solar spectrum and the inefficient separation of photoinduced electron–hole pairs hamper the practical application of TiO₂ materials. To overcome the aforementioned serious drawbacks of TiO₂, numerous strategies, such as doping with foreign atoms, particularly nitrogen (N), have been improved in the past few decades. This review aims to provide a comprehensive update and description of the recent developments of N-doped TiO₂ materials for visible light-responsive photocatalysis, such as (1) the preparation of N-doped/co-doped TiO₂ photocatalysts and (2) mechanistic studies on the reasons for visible light response. Furthermore, the most recent and significant advances in the field of solar energy applications of modified N-doped TiO₂ are summarized. The analysis indicated the critical need for further development of these types of materials for the solar-to-energy conversion, particularly for water splitting purposes.

Keywords N-doped TiO₂ · Visible light-responsive · Photocatalysis · Solar energy

Introduction

Developing advanced photocatalysts for large-scale solar energy conversion systems is an effective method to alleviate the increasing energy demand and environmental degradation issues. Since the pioneering work conducted by Fujishima and Honda in 1972 [1], who developed n-type rutile titanium oxide (TiO₂) for photoelectrochemical water splitting under light illumination, the use of semiconductor-based materials for photocatalysis has become a promising technology to convert inexhaustible solar energy into storable chemical energy. In the past few decades, numerous candidate semiconductor materials, such as TiO₂, CdS, and C₃N₄, have been developed and utilized as photocatalysts and extensively investigated. Among these numerous photocatalysts, TiO₂ has attracted considerable attention because

of its high photochemical stability, environmental friendliness, cost-effectiveness, and other excellent properties [2, 3].

In nature, TiO₂ has eight types of crystal structures, namely, anatase, rutile, brookite, TiO₂-H, TiO₂-II, TiO₂-III, TiO₂-R, and TiO₂-B [4]. Among them, only the first three types of crystal phases have been investigated, mostly for application purposes, because these titanium oxides can be naturally formed at atmospheric pressure. The structural parameters and schematic structures of the three structures are illustrated in Fig. 1 and Table 1, respectively. Figure 1 shows that, in all of the three crystalline structures of TiO₂, titanium (Ti) atoms are sixfold coordinated to oxygen (O) atoms, forming the distorted TiO₆ octahedral structure [5]. Despite having similar constituent structural units, their physical properties are different: Rutile TiO₂ is the most thermodynamically stable phase, whereas the metastable phase of anatase and brookite TiO₂ can be transformed into the rutile phase after calcination at a certain temperature. In general, both rutile and anatase TiO₂ have a tetragonal structure, whereas brookite TiO₂ has an orthorhombic structure (as shown in Table 1). These differences in the crystallite parameters of TiO₂ with different crystal phases can result

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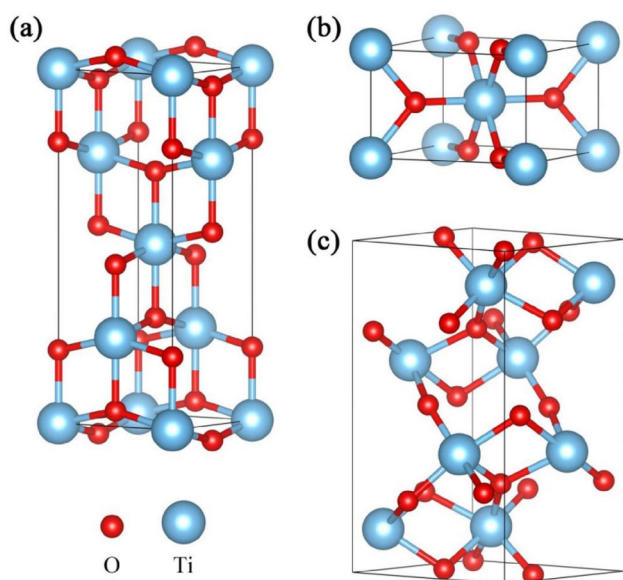


Fig. 1 Crystal configurations of **a** anatase, **b** rutile, and **c** brookite TiO_2 . The small red sphere and large blue sphere represent the O and Ti atoms, respectively

in different electronic structures and physicochemical properties of TiO_2 , which further leads to different activities in photocatalytic reactions. For instance, anatase TiO_2 shows higher water reduction activity than rutile TiO_2 , whereas rutile TiO_2 exhibits better water oxidation activity than anatase TiO_2 [6]. At the same time, compared with anatase and rutile TiO_2 , brookite TiO_2 shows better performance in several photocatalytic reactions [7]. These differences in the application of photocatalysis originate from not only the differences in physicochemical properties caused by physical structures but also the differences in electronic structures.

Density functional theory simulations revealed that the calculated bandgap for anatase, rutile, and brookite TiO_2 (as shown in Fig. 2) ranges from 1.86 to 2.38 eV [9], which are smaller than the experimental values. These underestimations can be attributed to the fact that the generalized gradient approximation function usually underestimates the bandgap of semiconductors [10]. Moreover, anatase TiO_2 can be considered an indirect bandgap material, whereas rutile and brookite TiO_2 can be considered direct bandgap materials. In the electronic structures of the three types of TiO_2 , the top of the valence band (VB) is primarily composed of O

$2p$ orbitals and a few Ti $3d$ orbitals, whereas the bottom of the conduction band (CB) is primarily composed of Ti $3d$ orbitals and a few hybrid O $2p$ orbitals. Therefore, under light irradiation, the electrons are excited from the O $2p$ orbitals in the VB to the Ti $3d$ orbitals in the CB of the TiO_2 semiconductor, leaving the holes in the VB. Subsequently, these charge carriers migrate to the surface of the catalyst and participate in the corresponding redox reactions.

However, because of its wide bandgap (approximately 3.2 eV), TiO_2 can only utilize ultraviolet (UV) light, which accounts for only approximately 5% of solar energy reaching the Earth [11], for photocatalytic reactions. To overcome the poor light response of TiO_2 photocatalyst, diverse strategies, such as introducing oxygen vacancies (OVs) [12, 13], sensitizing with dyes [14, 15], utilizing the surface plasmon resonance effect on metals [16, 17], and tuning the bandgap by heteroatom doping [18, 19], have been implemented. Among these strategies, substituting the host anion and/or cation in the TiO_2 crystal lattice with various heteroatoms has been proven to be one of the most efficient routes to facilitate the utilization of visible light of TiO_2 . Notably, ascertaining the most suitable impurity elements is the key step because foreign elements may disrupt the local chemical environment and charge balance in the TiO_2 matrix, such as the ionic radii and/or valence state deviations between the host anion and/or cation and the foreign element [20]. Thus far, cation dopants (such as transition metals [21–28], inner transition metals [29–32], and noble metals [33–35]), anion dopants (such as boron (B) [36, 37], carbon (C) [38, 39], nitrogen (N) [40, 41], fluorine (F) [42, 43], sulfur (S) [44, 45], and chlorine (Cl) [46, 47]), cation/anion co-dopants [48–51], and anion/anion co-dopants [52–54] have been utilized to adjust the optical and electrical properties of TiO_2 photocatalysts.

Extensive works on metal cation doping of TiO_2 photocatalysts have been reported. These reports show that metal cation doping can create an energy level of donor level (i.e., the level higher than the original VB maximum) or acceptor level (i.e., the level lower than the original CB minimum) in the forbidden bandgap of TiO_2 , hence decreasing the bandgap and increasing the visible light sensitivity of TiO_2 . However, doping of metal cation can decrease the carrier mobility, which is the result of the generation of strongly localized d states in the bandgap [55], and generate the electron–hole recombination centers derived from the newly inserted energy level [56], thus leading to the significantly

Table 1 Structural data for anatase, rutile, and brookite TiO_2 [8]

Structure	Crystal system	Space group	Group number	Cell parameters (Å)		
				<i>a</i>	<i>b</i>	<i>c</i>
Anatase	Tetragonal	$I4_1/amd$	141	3.758	3.758	9.514
Rutile	Tetragonal	$P4_2/mnm$	136	4.584	4.584	2.953
Brookite	Rhombohedral	$Pbca$	61	9.166	5.436	5.135

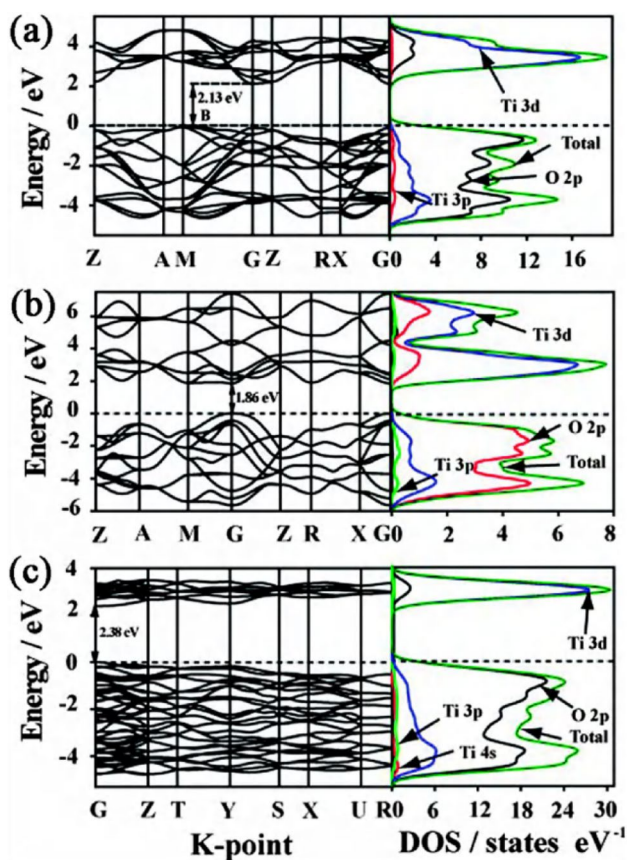


Fig. 2 Band structure and density of states of **a** anatase, **b** rutile, and **c** brookite TiO₂ (the Fermi level is set to 0). Reproduced from Ref. [9] with permission. Copyright 2014, Royal Society of Chemistry

reduced the photocatalytic performance. Moreover, the thermodynamic instability of the resulting cation-doped structure can cause photocorrosion in the photocatalysis process [57], which also hinders the practical application of TiO₂ photocatalysts.

By contrast, the use of nonmetal elements instead of metal elements to dope the TiO₂ semiconductor is a feasible pathway to develop its visible light activity. Since the breakthrough work conducted by Asahi et al. [58], who first investigated the visible light-driven catalytic performance of N-doped TiO₂ for the photodegradation of methylene blue and gaseous acetaldehyde, the nonmetal elements that act as dopants have been extensively investigated. Among these various nonmetal dopants (e.g., B, C, N, F, S, and Cl), the synthesis of N-doped TiO₂ has become the focus of considerable research because of the similar structural characteristics of O and N, such as polarizability, electronegativity, and ionic radii [56].

Thus far, an increasing number of literature reviews on the utilization of TiO₂ for solar energy utilization have been reported [59–64]. For example, Ma et al. [59] focused on the strategies to improve the generation of H₂ and the

photoreduction of CO₂ with H₂O to other energy resources on TiO₂-based catalysts. They also summarized the studies of the charge carriers properties of TiO₂ to better understand the dynamic characteristics of TiO₂ photocatalysts with different phases. Ullattil et al. [60] focused on the various synthetic routes, structures, morphological changes, and electronic structures of black TiO₂ nanomaterials and provided readers an overview of the different applications of black TiO₂ nanomaterials in the field of environmental technology. Along similar lines, Naldoni et al. [64] illustrated the basic concepts of effective design of reduced TiO₂ photocatalysts for H₂ generation. Kumaravel et al. [61] and Chen et al. [62] discussed six unique challenges that inhibit the application of TiO₂ in organic effluents. Nonetheless, these reviews ignored the important role and the recent progress of visible light-responsive TiO₂ in practical applications in indoor environments, particularly the N-doped TiO₂, which has attracted considerable attention in visible light applications. Therefore, this review critically analyzes the improvement of visible light-sensitive N-doped TiO₂ photocatalysts. Moreover, the application prospect of these photocatalysts under visible light illumination, particularly the overall photocatalytic water splitting reaction, is reviewed.

Synthesis of N-Doped TiO₂

The key to the successful utilization of N-doped TiO₂ is to rationally select the synthesis strategies to prepare the target photocatalysts. Figure 3 shows the common methods involved in the synthesis of N-doped TiO₂ material. Based on the phase of the reactants, the strategies to prepare N-doped TiO₂ can be coarsely split into the wet chemistry and solid-phase methods.

Wet Chemistry Method

The wet chemistry method is a kind of strategy to prepare the target material through a chemical reaction with the participation of liquid, including the precipitation, hydrothermal, sol–gel, and electrochemical deposition methods. The sol–gel and hydrothermal methods are usually described as soft methods for doping N and/or other dopants into the TiO₂ crystal lattice, followed by calcination of the sample at high temperature. These methods usually have the advantages of being cleanroom-free, cost-efficient, and widely applicable. For this purpose, in a typical procedure, the doped N usually originates from nitric acid (HNO₃) [65, 66], ammonium hydroxide (NH₄OH) [67], urea (CH₄N₂O) [68, 69], ammonium chloride (NH₄Cl) [70, 71], and triethylamine (TEA) [72, 73] and the Ti precursors usually include titanium trichloride (TiCl₃) [74], titanium tetrachloride (TiCl₄) [75], titanium sulfate (Ti(SO₄)₂) [76], and titanium (IV)

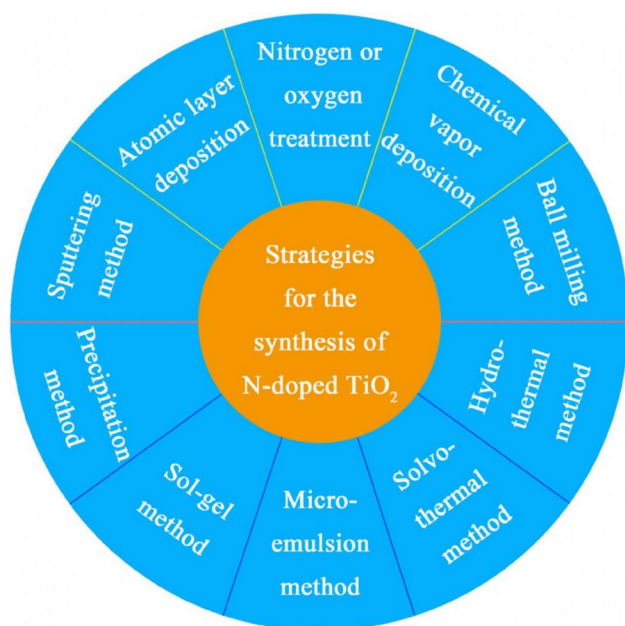


Fig. 3 Common methods for the preparation of N-doped TiO₂

isopropoxide (TTIP) [77]. For example, Zhang et al. [65] selected HNO₃ as the N source and prepared nanocrystalline N-doped TiO₂ via a one-step solvothermal route with a low temperature (180 °C). In another effort, Panda's group [67] prepared N-doped yellow TiO₂ hollow spheres through a simple solvothermal method using aqueous titanium peroxocarbonate complex solution and NH₄OH as doping N sources. Notably, the catalytic activity of the obtained N-TiO₂ hollow spheres was better than that of other reported N-TiO₂ samples with different morphologies, which may be owing to the increased surface area and microreactor environment and the decreased recombination for light scattering caused by the inner hollow structure. By adding ammonia water, Cheng et al. [78] synthesized N-doped TiO₂ through a simple solution-based strategy. Their results showed that N doping could not only effectively inhibit the phase transition of TiO₂ from anatase to brookite but also help enhance the visible light absorption. Marques et al. [68] noted that the specific surface area and photocatalytic performance of N-doped TiO₂ could be influenced by different synthesis methods and parameters, e.g., doping temperature and molar ratio of urea to TiO₂. Di Camillo et al. [70] reported that N-doped TiO₂ nanofibers could be easily deposited via the electrospinning method. Their results showed that the light absorption properties of these nanofibers could be modified even with a small amount of N, which can result in good photocatalytic performance even when illuminated with visible light. Generally, the catalysts obtained by these methods have specific sizes and morphologies and usually exhibit a higher purity and uniformity. However, the extensive

applications of the catalysts may be limited by the multistep synthesis process and inevitable use of organic solvents.

Solid-Phase Method

The solid-phase method is the thermal treatment process after mixing the N and TiO₂ sources under a given atmosphere. The solid-phase method is an economical and effective strategy for the large-scale synthesis of N-doped TiO₂. In addition to the ball milling, atomic layer deposition (ALD), and chemical vapor deposition methods (as shown in Fig. 3), the most common and efficient method used to prepare the target N-TiO₂ catalyst is to oxidize TiN powder under O₂ gas flow or nitrate TiO₂ powder under NH₃ gas flow [79]. A previous study reported that, by annealing at 200–550 °C for 1.5 h, the TiN sample could be converted into N-doped TiO₂ under an O₂ gas atmosphere. In their study, Morikawa et al. [80] determined that the calcination temperature can influence the crystal phase of the obtained sample. For instance, a homogeneous rutile phase could be obtained under the calcination temperature of 550 °C, whereas a small amount of TiN remained in the rutile phase of the TiO₂ sample under the calcination temperature of 400 °C. In another effort, Zhao's group [81] prepared N-doped TiO₂ samples with a mixed phase of anatase and rutile by the incomplete oxidation of TiN at various temperatures. Then, they investigated the photocatalytic performances of the obtained samples by photocatalytic degradation of gas-phase toluene under visible light illumination and achieved improved photocatalytic performances. Recently, Xia et al. [82] prepared highly ordered porous nanotube arrays of N-doped TiO₂ through oxidation of the preprepared porous TiN nanotube arrays in an air atmosphere. Photocatalytic investigations indicated that the obtained porous N-doped TiO₂ nanotube arrays had a visible light photocatalytic kinetic rate constant that was threefold larger than that of pure anatase TiO₂ nanotube arrays, which could be attributed to its higher visible light responsiveness and larger surface area.

As for the nitrification of the TiO₂ sample under an NH₃ gas atmosphere, the synthesis conditions have obvious effects on the morphology of the N-doped TiO₂ photocatalyst. Hashimoto's group [83] processed TiO₂ with anatase phase at different temperatures for 3 h under a mixed NH₃/Ar (67%/33%) gas flow to obtain the anatase phase of the N-doped TiO₂ material. The resulting samples with different colors signified that the change in calcination temperature and NH₃ gas flow resulted in different amounts of N in the N-doped TiO₂ material, which would result in different quantum yields because of the varying visible light absorption capabilities. In another effort, Pulgarin's group [84] used thiourea as the N source and mixed it with commercial anatase TiO₂ powder, followed by annealing at 400 °C and 500 °C, which resulted in visible light absorption of TiO₂.

Fang et al. [85] prepared N-doped TiO₂ by annealing P25 or TiO₂ xerogel powder under NH₃/Ar atmosphere at different temperatures. They determined that, compared with the N-doped TiO₂ prepared by annealing P25, the N-doped xerogel powder had a higher concentration of substituted N and exhibited a more distinct visible light absorption. Another study of N-doped TiO₂ with abundant OV's reported by Huang et al. [86], which was prepared by a facile hydrothermal method with a baking process in an NH₃ atmosphere, showed that more OV's could be generated by doping N into TiO₂, consequently ameliorating their catalytic behavior, particularly that of the OV's derived from visible light illumination.

Meanwhile, different N-containing molecules, such as urea and hexamethylenetetramine, were utilized for the doping process using a solid-phase (mechanochemical) method [68, 87]. In this process, the mixture containing TiO₂ and urea/hexamethylenetetramine was ground using a planetary ball mill and calcined at 400 °C to remove the residual organic substances. Notably, the obtained N-doped powder mainly consists of rutile TiO₂, indicating that high mechanical energy accelerates the phase transformation of anatase into rutile.

Furthermore, the sputtering process, ALD, and pulsed laser deposition are the most advanced techniques for the fabrication of crystalline N-doped TiO₂ thin films. For example, via the ALD operation, Vasu et al. [88] grew p-type epitaxial N-doped anatase TiO₂ (001) thin films on *c*-axis Al₂O₃ substrate, employing TiCl₄, double-distilled water, and NH₃ as Ti, O, and N sources, respectively. Their results showed the enhanced hole concentration and mobility of the obtained samples.

In each case, the presence of N within the lattice of TiO₂ has been shown to have an obvious effect on photocatalysis. As exhibited in Fig. 4 (Curve a), the pure anatase TiO₂ shows a single sharp edge, whereas the N-doped TiO₂ (Curve b in Fig. 4) shows two absorption edges: One is the main edge that originated from the oxide at approximately 390 nm, which does not change significantly compared with that of the undoped sample, and the other is the weak shoulder absorption at 400–550 nm wavelength. This noticeable shift of absorption to the visible light region corresponds to the yellow characteristic observed in the obtained N-doped samples [89]. Furthermore, with the increase in N content in the N-doped TiO₂, the shoulder absorption initially shifts to the lower energy region, indicating that the N-doped samples are sensitive to visible light. Although the shoulder absorption expands the light absorption of the obtained TiO₂ semiconductor to a certain extent, it cannot extend its intrinsic band absorption to the visible light range, which will result in the low visible light absorption efficiency of the prepared N-doped TiO₂. More importantly, this N substitution in TiO₂ usually leads to the generation of OV's, which results in a

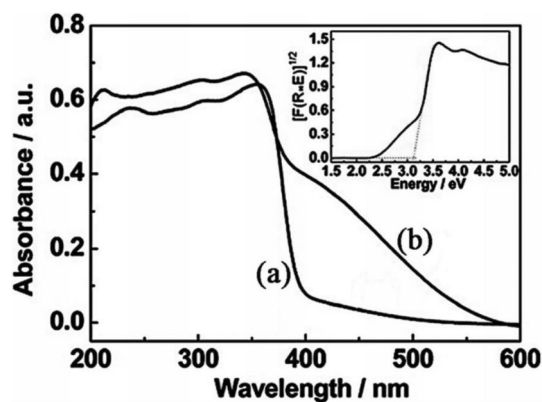


Fig. 4 UV-Vis diffuse reflectance (a pure anatase TiO₂, b N-doped TiO₂). Reproduced from Ref. [89] with permission. Copyright 2009, American Chemical Society

decrease in photocatalytic activity [90]. Hence, finding a simple and superior method for the scalable preparation of N-doped TiO₂ material with high visible light absorption efficiency is still a challenge.

Co-incorporation of N and other nonmetal anions has been considered a new potential pathway to construct the energy band structure of TiO₂. The remarkable synergetic effects in N,F-co-doped TiO₂ have been reported by some interesting works. For instance, Domen's group [91] utilized SiO₂ powder as the O source in their investigation of N,F co-doping into anatase TiO₂ by nitriding an (NH₄)₂TiF₆ precursor. Despite their results showing good absorption in the visible region ($\lambda < 570$ nm), which drives water oxidation for the N-doped catalyst, a shortcoming remains, i.e., under the preparation conditions, the N, O, and F contents of the product were uncontrollable. Afterward, Maeda et al. [92] prepared high-crystallinity TiN_xO_yF_z material with an anatase crystal structure via nitridation of NH₄TiOF₃ in an NH₃ atmosphere, which can be used as a visible light-responsive photocatalyst for water oxidation at wavelengths of up to 540 nm. In another investigation, Zong et al. [93] reported that the visible light-responsive anatase TiO₂ platelets dominated by (001) facets were prepared by simple nitridation of the TiOF₂ precursor. The synthesized TiOFN materials exhibited excellent visible light-driven activity for water oxidation. Although the aforementioned studies achieved the absorption of visible light for N-doped TiO₂, it was only used in water oxidation, and the performance of water reduction for H₂ evolution of the catalyst was not investigated. Recently, Maeda and colleagues [94] fabricated a novel N,F-co-doped rutile TiO₂ by nitridation of the rutile TiO₂ and (NH₄)₂TiF₆ mixture and evaluated its photocatalytic oxidation activity of water. Their evaluation indicated that the N,F-co-doped samples exhibited water oxidation activity even in the presence of reversible electron acceptors under the action of the RuO₂ cocatalyst. Moreover, in

the presence of a shuttle redox mediator, the mixture of Ru/SrTiO₃:Rh and N,F-co-doped TiO₂ samples was used to obtain stoichiometric H₂ and O₂ without significant active degradation under visible light illumination.

In addition to the F dopant, B was selected as the codopant of N to optimize the band structure of the TiO₂ semiconductor. Liu et al. [95] reported the first example and demonstrated the substantial effect of B,N co-doping on improving the visible light response and charge carriers separation of TiO₂. Their results also indicated that, during the N doping process, the presence of the B dopant stabilized the structure relative to undoped TiO₂. Recently, in the study conducted by Hong et al. [96], a red anatase TiO₂ with a spatially homogeneous distribution of B and N dopants, which was sensitive to the sharp-edged strong visible light absorption spectrum over 680 nm, was obtained. The synthesized red TiO₂ sample can drive photocatalytic water oxidation for O₂ evolution under the illumination of visible light exceeding 550 nm and induce photocatalytic water reduction for H₂ evolution under visible light illumination. They concluded that the adsorption of HNCO and related derivatives on the surface of TiO₂ released by the thermal hydrolysis of urea significantly promoted the diffusion of the N dopant into the microspheres, thereby inducing uniform N doping with the help of the B dopant, resulting in a strong band-to-band absorption in the entire visible light range. Furthermore, several N-doped TiO₂ nanomaterials modified with other nonmetals (such as C, P, and S) were investigated for their visible light catalytic activities [97–99]. The results showed that, compared with undoped TiO₂ or mono-N-doped TiO₂, the modified N-doped TiO₂ samples exhibited enhanced photocatalytic activities.

Mechanisms for Visible Light Response

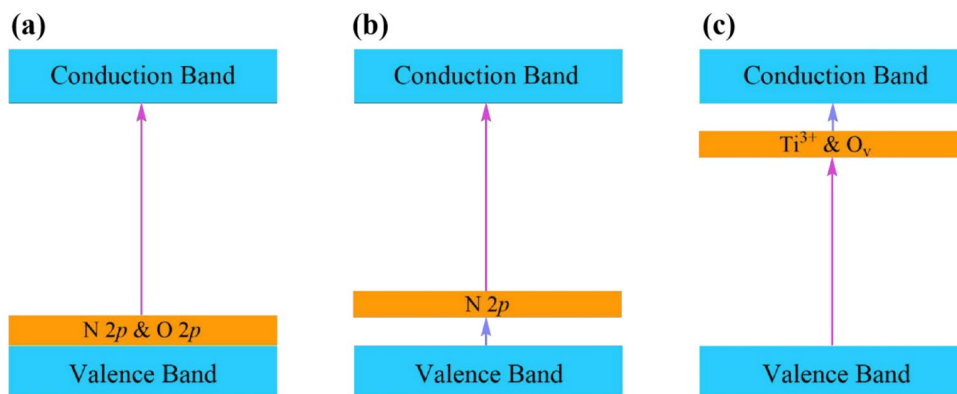
Generally, electron–hole pairs generated via light illumination are the basic components for photocatalytic reactions. Since the pioneering work conducted by Asahi et al. [58]

to synthesize N-doped TiO₂ material by sputter deposition, tremendous efforts have been exerted to investigate the potential mechanisms for the photocatalytic reactions of N-doped TiO₂ particles under visible light illumination and explore the electronic, optical, and structural characteristics of N-doped TiO₂ samples.

Even though the visible light absorption and enhanced photocatalytic activity of N-doped TiO₂ catalysts have been generally recognized, in theory, there are still open debates about the chemical properties, location of the photoactive N species involved, and origin of the visible light response. In general, the main proposed mechanisms for the improved visible light absorption of N-doped TiO₂ can be divided into three areas, which are discussed in the subsequent paragraphs.

Based on the first principle calculations, in 2001, Asahi et al. [58] investigated the contribution of N doping to the narrowed bandgap of TiO₂ and compared three N-doped TiO₂ systems, namely, substitutional N doping, interstitial N doping, and both types of doping into anatase TiO₂. Their results indicated that the N 2*p* state could hybridize with the O 2*p* state (Fig. 5a), resulting in the narrowed bandgap and enhanced visible light-driven photoactivity. However, since 2003, this conclusion was challenged by other opinions, which tend to support that N doping will not result in the reduction of the TiO₂ semiconductor bandgap based on the experimental and theoretical studies but introduce impurity energy levels or induce the formation of OV_s/Ti³⁺ defects. Specifically, Irie et al. [83] evaluated the oxidization capability of the TiO_{2-x}N_x powder by the decomposition of gaseous 2-propanol under the same illumination conditions (i.e., the absorbed photon number of visible or UV light was equal to each other). They determined that, regardless of the *x*, the quantum yield from visible light illumination was lower than that from UV light illumination. Moreover, the quantum yield decreased with the increase in the N concentration when irradiated with UV and visible light. Therefore, they concluded that the formation of the isolated and localized midgap state above the VB by N doping (as

Fig. 5 Three proposed schemes illustrating the possible origin of visible light absorption in N-doped TiO₂: **a** N 2*p* hybridization with O 2*p*, **b** introduction of the localized state above the valence band, and **c** introduction of the localized state below the conduction band. Reproduced from Ref. [100] with permission. Copyright 2010, Royal Society of Chemistry



shown in Fig. 5b) was the cause of the visible light-responsive N-doped TiO₂ materials.

Another point is that the OVs caused by N doping contribute to the absorption and photocatalytic performances in the visible light wavelength range [101], as shown in Fig. 5c. One explanation is that NH₃ can decompose into N₂ and reducing gas H₂ at approximately 550 °C [102]. As a result, the decomposed H₂ can result in the formation of reduced TiO₂ when annealing TiO₂ powder in an NH₃ flow. For instance, Zhao et al. [103] prepared N-doped TiO₂ photocatalysts through a simple solvothermal strategy, followed by a solid-state chemical reduction process. Based on the analysis of the results of the control experiments, they proposed that the reason for the narrowed N-TiO_{2-x} bandgap is the synergetic effect of the defect energy states (including Ti³⁺ and OVs) and N 2p midgap state. Another generally accepted explanation is that the OVs are derived from the charge imbalance (N³⁻ vs. O²⁻) after N doping [90, 104]. Meanwhile, the formation of Ti³⁺ is accompanied by the generation of OVs. This opinion has been verified by Serpone [105], who believe that co-doping N and other anions into the TiO₂ lattice will increase the number of OVs.

Although the aforementioned ideas can explain the reasons for the visible light absorption of N-doped TiO₂ material, there is still no consensus on whether the bandgap can be narrowed and its working mechanism. Therefore, this issue is still open for discussion and in-depth research needs to be conducted.

Characterizations of N-Doped TiO₂ Photocatalysts

The photocatalysis process has been well known to involve three crucial stages, namely, light absorption, charge carriers separation, and surface reaction [106]. Apart from the energy states, which are highly related to the light absorption stage, the investigation of the behavior of photoinduced charge carriers is essential to determine photocatalytic performance. The photoinduced charge carriers are fleetly trapped at the surface states of the photocatalyst particle within sub-picoseconds or a few picoseconds after excitation [107]. Although the lifetime is transitory, the measurement of the transfer kinetic of these charge carriers can provide important information for researchers and reveal the main process in the photocatalytic reaction. Thus far, time-resolved spectroscopy has been proven to be a powerful technique to analyze the charge carriers dynamics of photocatalysts, particularly the photocatalytic processes.

Time-resolved microwave conductivity (TRMC) is one of the powerful technologies employed to investigate the charge carriers dynamics. The obtained TRMC signal decays for TiO₂-based material reflect the change in conductivity

caused by the variations of the number of generated electrons. Consequently, the recombination and capture processes of electrons in TiO₂ samples can be determined based on the variations of these signals. Through this technology, Katoh et al. [108] investigated the charge carriers separation and recombination processes in the N-doped TiO₂ photocatalyst. Their results showed that the electronic behavior of N-doped TiO₂ excited by UV light (Ti 3d ← O 2p transition) was similar to that excited by visible light (450 nm, Ti 3d ← N 2p transition). However, the trapping rate increased (i.e., the charge carriers separation efficiency of N-doped TiO₂ excited by visible light excitation was one-third of that of pure TiO₂ excited by UV light) in the N-doped sample relative to the undoped sample, which meant that N doping will seriously affect the concentration of OVs.

Yamanaka and Morikawa [109] used femtosecond time-resolved diffuse reflectance (TDR) spectroscopy to elucidate the kinetics of charge carriers separation and trapping in N-doped TiO₂ powder under weak excitation conditions. As shown in Fig. 6a, the TDR spectra of N-doped TiO₂ revealed that the electrons and holes captured on the surface were produced fleetly after excitation, which was similar to that of TiO₂ under 360 nm light excitation (Ti 3d ← O 2p transition). The additional OVs lead to deep trapping (the time constant is approximately 300 ps); thus, the number of trapped electrons on the surface was lower than that of pure TiO₂. Under 450 nm excitation, the TDR spectra of N-doped TiO₂ showed the signals derived from the Ti 3d ← N 2p transition. Furthermore, compared with the 360 nm excitation, the 450 nm excitation exhibited two differences in time evolution, i.e., it significantly decreased immediately after excitation and the electron depth was trapped in 1 ps.

In another study, Chen et al. [110] investigated the decay kinetics of photogenerated electrons that result in recombination or carrier reactions in the synthesized N-doped TiO₂ samples using time-resolved infrared (IR) spectroscopy. The observed transient absorbance spectrum at 1910 cm⁻¹ shown in Fig. 7a revealed that the MTiO₂/Ala-DDA samples exhibited the strongest initial absorption, indicating the existence of a large number of charge carriers. The pure anatase MTiO₂/DDA exhibited weaker absorption intensity, indicating that the formation of OVs on tetrahedral Ti⁴⁺ sites created a shallow donor state below the CB of the MTiO₂/Ala-DDA samples. As shown in Fig. 7b, the normalized absorbance decay revealed that the obtained MTiO₂/Ala-DDA exhibited a slower decay rate than the two other MTiO₂ samples, indicating that the separation of electrons and holes was effectively realized. This may be benefited from the shallow donor state produced by the OVs on tetrahedral Ti⁴⁺ sites can more effectively promote charge carriers separation and electron capture in MTiO₂/Ala-DDA.

Time-resolved absorption (TA) spectroscopy is another powerful and delightful technique to investigate the nature

of photogenerated charge carriers. Tang et al. [111] explored the nature of charge carriers in N-doped films. They explained that the reason for the poor photocatalytic O_2 production from water oxidation of nanocrystalline N-doped TiO_2 film under visible light illumination was mainly the rapid recombination of charge carriers at doping-induced states rather than the deep trapping of photogenerated holes in the bulk material (as shown in Fig. 8).

Moreover, they concluded that a long lifetime of no less than 0.4 s for the photogenerated holes was essential to the evolution of O_2 on the nanocrystalline N-doped TiO_2 . Therefore, to achieve an improved O_2 evolution efficiency for this doped material, the key strategy is to enhance the charge carriers separation with the aid of cocatalysts, such as Pt cocatalyst.

Fig. 6 Schematic diagram of the spatial and energetic distribution of electrons and holes in N-doped TiO_2 powder after excitation under **a** UV light (360 nm) and **b** visible light (450 nm) irradiation. Reproduced from Ref. [109] with permission. Copyright 2010, ACS Publications

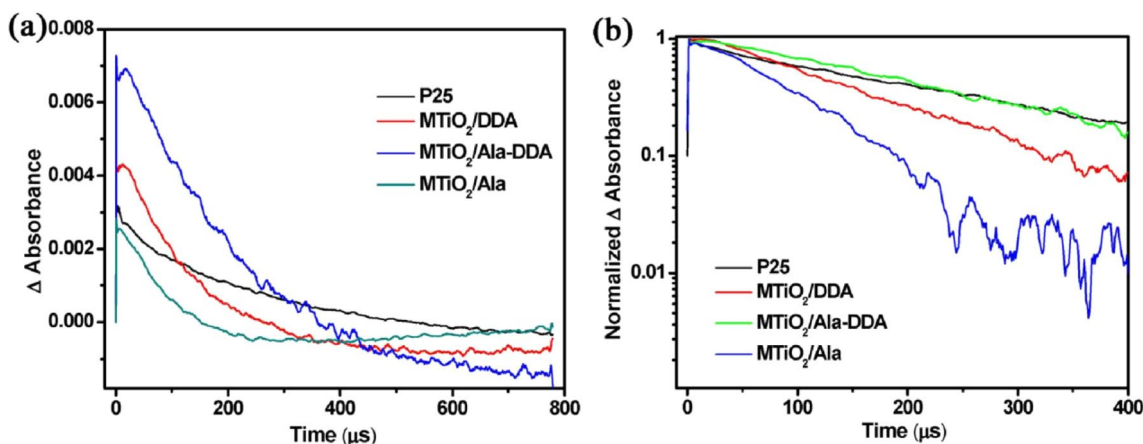
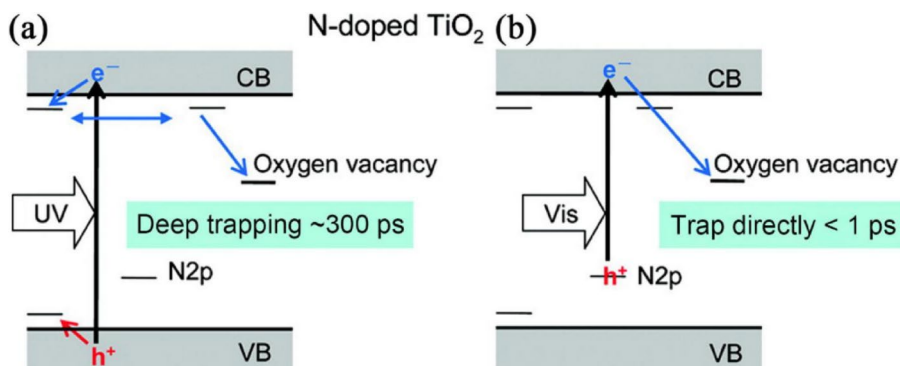
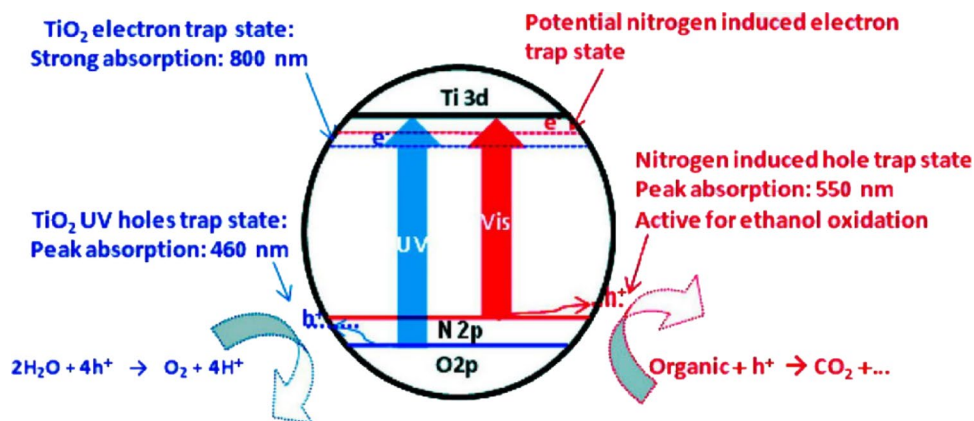


Fig. 7 **a** Temporal profiles of transient infrared absorption in the atmosphere and **b** normalized decay curves at 1910 cm^{-1} of the four catalysts (M: mesoporous, DDA: dodecylamine, Ala: L-alanine acids). Reproduced from Ref. [110] with permission. Copyright 2019

Fig. 8 Band structure of the N-doped TiO_2 sample. Reproduced from Ref. [111] with permission. Copyright 2011, ACS Publications



Main Applications

The aim of engineering functional N-doped TiO₂ samples with well-defined morphologies and crystal phases is to optimize their electronic properties, thereby achieving better performance for photocatalytic applications. In general, a semiconductor-based photocatalytic process usually includes three independent and indispensable steps (Fig. 9) [112], namely, absorption of photons and generation of electron–hole pairs, separation and migration of charge carriers from the matrix to the surface, and surface reactions (i.e., reduction and oxidation reactions). Thus far, tremendous efforts (Table 2) have been exerted to enable the practical applications of N-doped TiO₂ in photooxidation reactions (e.g., photodegradation, photocatalytic disinfection, and O₂ generation from water splitting) or photoreduction reactions (e.g., CO₂ reduction and H₂ generation from water splitting).

Degradation

N-doped TiO₂ material is an effective photocatalytic system for treating some harmful compounds in polluted water/air, and it has been extensively investigated for water treatment and air purification. In the early 2000s, Asahi et al. [58] reported that, under visible light illumination (wavelength < 500 nm), N-doped TiO₂ exhibited better photocatalytic performance for the degradation of methylene blue and gaseous acetaldehyde than undoped TiO₂. Salarian et al. [113] synthesized N-doped TiO₂ via a facile hydrothermal route and investigated the photodegradation performance of diazinon. Their results showed that the degradation and mineralization efficiencies of diazinon could reach 85% and 63%, respectively. Wang et al. [123] showed that N-doped

TiO₂ nanoparticles prepared by a post-thermal treatment exhibited enhanced visible light photocatalytic activity for the degradation of methylene blue. Marques et al. [68] used urea as the N source and prepared N-doped TiO₂ nanoparticles with a large specific surface area using the modified sol–gel method. They determined that the doping calcination temperature and molar ratio of urea to TiO₂ have a considerable influence on the N content in N-doped TiO₂ powders. The results of their photocatalytic degradation experiments on methylene blue dye indicated that, under both UV-A light (365 nm) and visible light (400–700 nm) illumination, N-doped TiO₂ exhibited higher photocatalytic efficiency than the pristine TiO₂. Furthermore, the molar ratio of N to TiO₂ used for doping strongly affects the photocatalytic performance of the obtained N-doped TiO₂.

Moreover, the photocatalysis studies on N-doped TiO₂ with different exposed facets revealed different activities. Shi et al. [125] reported that N–TiO₂ plates with dominant facets [001] had superior photocatalytic activities for the degradation of methylene blue under visible light irradiation ($\lambda > 420$ nm) and concluded that the improved performance was attributed to the smaller particle size (approximately 25 nm) and larger specific surface area in comparison with micrometer-sized N-doped TiO₂ materials. Similarly, the [001]-dominated TiO₂ nanosheets co-doped with N and La reported by Wang's group [124] exhibited higher visible light photocatalytic activity for the decomposition of RhB than pure TiO₂, N–TiO₂, La–TiO₂, and N,La–TiO₂ nanoparticles (Fig. 10). This may be because the N, La–TiO₂ nanosheets exhibited the strengthened absorption of visible light than other samples. Moreover, the synergistic effect between N and La, i.e., N doping narrowed the bandgap of TiO₂, whereas La doping improved the separation efficiency of photoelectrons and holes, was attributed to the enhanced photodegradation performance.

Sterilization

Photocatalytic disinfection is becoming a promising process for water treatment because it is cost-effective and environmentally safe. Makropoulou et al. [114] used various N precursors (such as urea, TEA, and NH₃) to synthesize N-doped TiO₂ photocatalysts and employed these photocatalysts to inactivate bacteria in an aqueous substrate under simulated sunlight illumination. As shown in Fig. 11, under simulated sunlight illumination, the Gram-negative bacteria *Escherichia coli* and *Pseudomonas aeruginosa* dispersed in solution were successfully inactivated, and the highest reduction rate of 6log10 was reached within 60 min of treatment. In a recent report, Horovitz et al. [115] investigated the efficiency of N-doped TiO₂-coated Al₂O₃ photocatalytic membrane reactors to remove MS2 bacteriophage with different

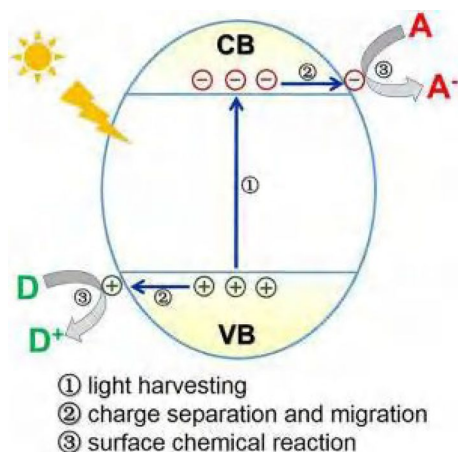


Fig. 9 Schematic illustration of a typical photocatalytic process based on semiconductors. Reproduced from Ref. [112] with permission. Copyright 2019, Elsevier

Table 2 Typical photocatalytic reactions of N-doped TiO₂-based materials

Photocatalyst	Preparation method	UV–Vis absorption	Light source ^a	Application	References
TiO _{2-x} N _x films	Sputtering and calcination	Shoulder absorption	UV light (with a peak at 351 nm) and visible light (with a peak intensity at 436 nm) irradiation	Degradation of methylene blue and acetaldehyde	[58]
N-doped TiO ₂	Hydrothermal method	Shoulder absorption	350 W Xe lamp (380 nm < λ < 900 nm)	Degradation of diazinon	[113]
N-doped TiO ₂ nanoparticles	Hydrothermal treatment and calcination	Shoulder absorption	LED light (365 nm) and 150 W Xe lamp (400 nm < λ < 800 nm)	Degradation of methylene blue	[68]
N-doped TiO ₂	Hydrolysis and calcination	Shoulder absorption	Natural sunlight	Inactivation of <i>Escherichia coli</i> , <i>Pseudomonas aeruginosa</i> , and <i>Bacillus cereus</i>	[114]
N-doped TiO ₂ thin film	Sol–gel deposition and calcination	–	300 W solar simulator with a quartz cover placed approximately 6.5 mm above the membrane	Inactivation of MS2	[115]
N-doped TiO ₂	Hydrolysis and calcination	Shoulder absorption	Xe lamp (315 nm < λ < 600 nm)	CO ₂ reduction	[116]
N-doped anatase TiO ₂ microsheets	Hydrothermal method	Shoulder absorption	350 W Xe lamp, UV–Vis and visible light (≥ 400 nm)	CO ₂ reduction	[117]
N-doped TiO ₂ nanorod arrays	Hydrothermal method	Shoulder absorption	Two 300 W visible light sources (UV < 5%)	CO ₂ reduction	[118]
TiO _{2-x} N _x	Sol–gel method and calcination	–	Visible light (≥ 400 nm)	O ₂ evolution from the oxidation of water	[119]
N-doped TiO ₂ nanorods	Sol–gel method and calcination	Shoulder absorption	Visible light (≥ 400 nm)	O ₂ evolution from the oxidation of water	[120]
N,F-doped rutile TiO ₂	Calcination	Band-to-band absorption	300 W Xe lamp (420 nm < λ < 800 nm), irradiated through a water filter	Z-scheme water splitting for O ₂ /H ₂ evolution ^b	[94]
B,N-doped TiO ₂	Hydrothermal method and calcination	Band-to-band absorption	300 W Xe lamp (≥ 420 nm)	Water splitting for O ₂ /H ₂ evolution	[96]
C,N-doped TiO ₂	Aqueous precipitation reaction and calcination	Band-to-band absorption	300 W Xe lamp (≥ 420 nm)	O ₂ evolution from the oxidation of water	[97]
N-doped TiO ₂	Hydrolysis and calcination	Shoulder absorption	300 W Xe lamp (≥ 400 nm)	H ₂ evolution from the reduction of water	[121]
N-doped TiO ₂ nanobelts	Hydrothermal method and calcination	Shoulder absorption	300 W Xe lamp (400 nm < λ < 800 nm)	H ₂ evolution from the reduction of water	[122]

^aXe lamp: xenon lamp.

^bZ-scheme water splitting was conducted with Ru/SrTiO₃:Rh (25 mg) and RuO₂/TiO₂:N,F (50 mg) catalysts in an aqueous tris(2,2'-bipyridyl)cobalt(II) sulfate ([Co(bpy)₃]²⁺, 0.5 mmol/L) solution (120 mL)

water qualities. Their results showed that the hybrid reactors exhibited high efficiency in inactivating the MS2 virus.

Organic Transformation

Photocatalytic organic transformation as a green, sustainable, and environmentally friendly technique has received increasing attention, and many organic materials with

special functions were synthesized via selective oxidation or reduction with the aid of photocatalysts in recent years. For instance, Japa et al. [126] investigated the selective transformation of benzyl alcohol and benzylamine into benzaldehyde and benzylamine under visible light illumination using N-doped TiO₂ as catalysts, which were prepared by thermal hydrolysis of TiOSO₄ using NH₄OH as both precipitating agent and N source. As shown in Fig. 12, the

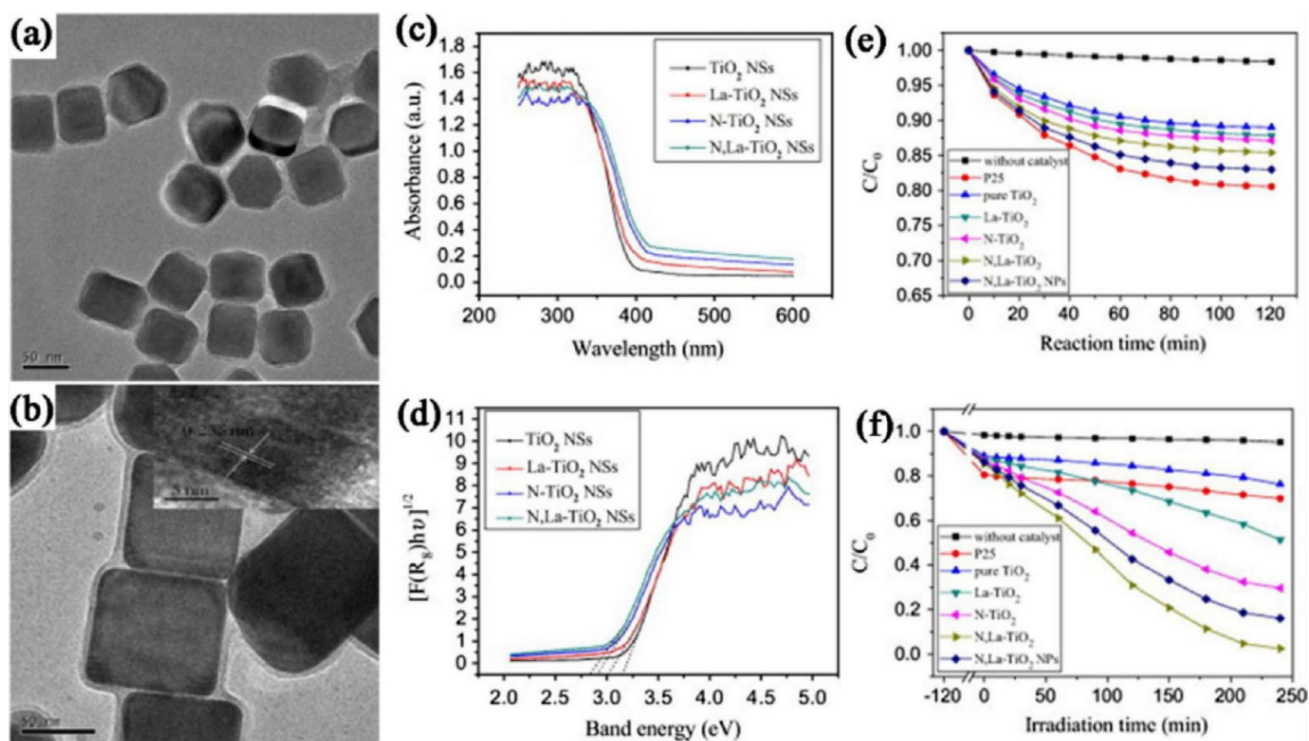


Fig. 10 **a, b** TEM of N,La-TiO₂ nanosheets; **c** UV-Vis diffuse reflectance spectrum and **d** plots of $[F(R_{\infty})hv]^{1/2}$ vs. photon energy of the samples of TiO₂, La-TiO₂, N-TiO₂, and N,La-TiO₂ nanosheets; **e**

adsorption of RhB (10 mg/L) in the dark; and **f** degradation of RhB with different photocatalysts under visible light irradiation. Reproduced from Ref. [124] with permission. Copyright 2015, Elsevier

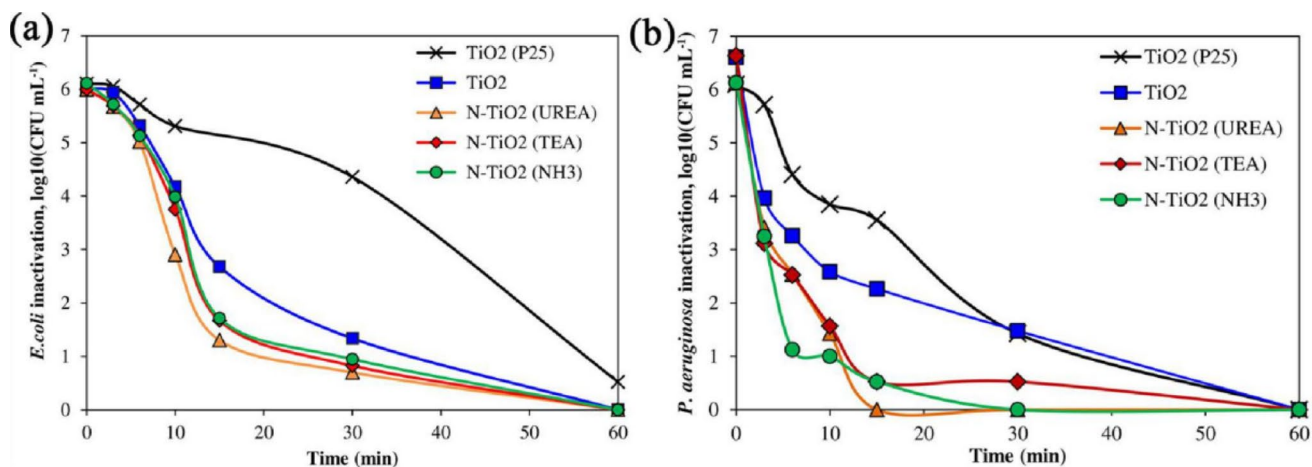


Fig. 11 Inactivation of **a** *Escherichia coli* and **b** *Pseudomonas aeruginosa* under simulated sunlight illumination in the presence of N-doped TiO₂ photocatalysts (fabricated using different N precursor

sors), pristine TiO₂, and P25 TiO₂. Reproduced from Ref. [114] with permission. Copyright 2018, Wiley-VCH

optimized N-doped TiO₂ (T_400) exhibited superior photocatalytic activity for the selective oxidation of benzyl alcohol and benzylamine to benzaldehyde and N-benzylidenebenzylamine products, respectively, with > 85% conversion and > 95% selectivity, regardless of the substituents of the benzylamine molecule. The control sample (T_400_NaOH),

prepared using NaOH (instead of NH₄OH) as precipitating agent, was also analyzed, and the results revealed the beneficial nature of N doping for selective oxidation.

Singha et al. [67] synthesized N-doped yellow TiO₂ hollow spheres using aqueous titanium peroxocarbonate complex solution as a precursor and NH₄OH and used them to

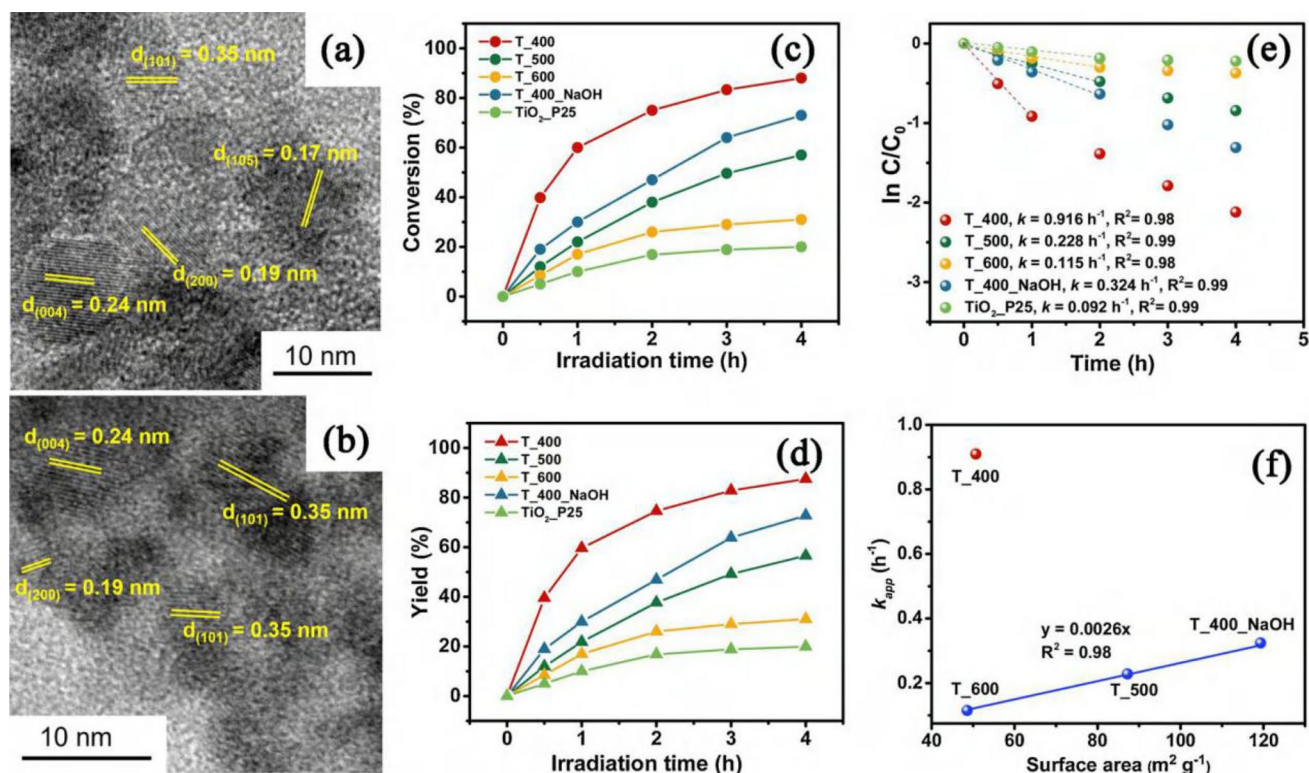


Fig. 12 HRTEM of **a** T₄₀₀ and **b** T₄₀₀_NaOH; **c** photocatalytic conversion of benzyl alcohol; **d** evolution of the benzaldehyde product; **e** corresponding initial pseudo-first-order kinetics of different catalysts at an optical power of approximately 75 mW/cm²; and

f relationship between apparent pseudo-first-order rate constant and catalyst surface area. Reproduced from Ref. [126] with permission. Copyright 2021, Elsevier

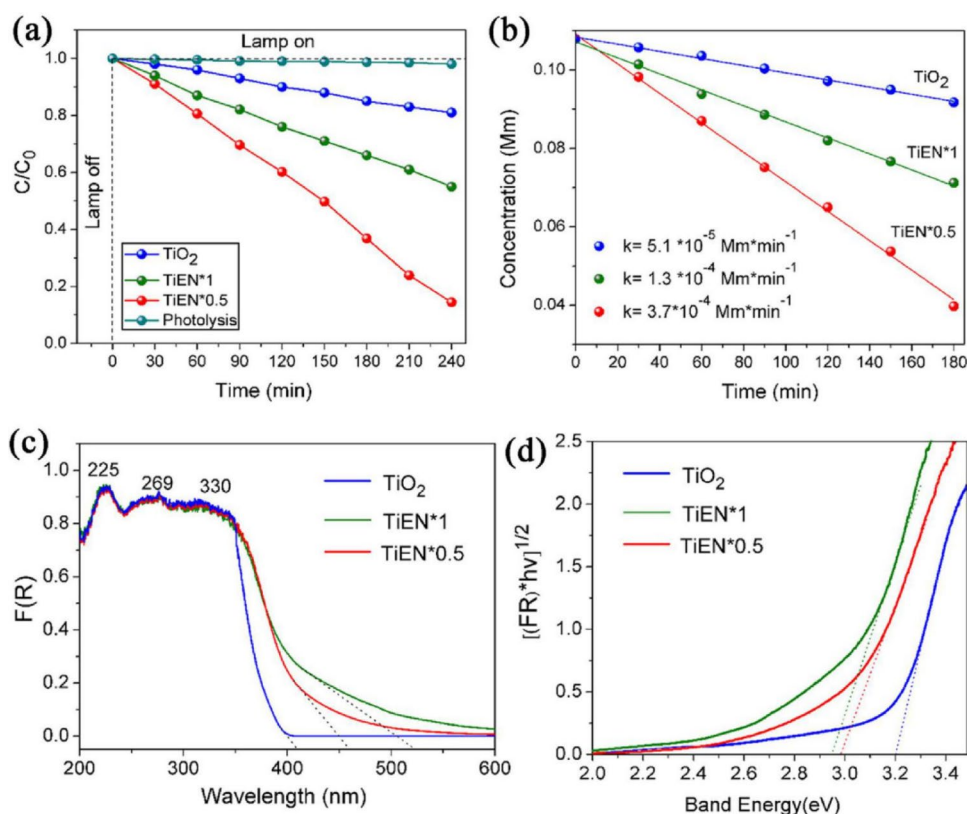
promote the synthesis of active esters of N-hydroxyimide and alcohol through simultaneous selective oxidation of alcohol to aldehyde, followed by cross-dehydrogenative coupling via visible light illumination under ambient conditions. Xu et al. [127] fabricated N-doped anatase/rutile mixed-phase TiO₂ photocatalysts with carbonaceous species by pyrolyzing MIL-125 (Ti) and P25 composites as the precursor and subsequently doping N into the pyrolysis product using urea as the dopant. In this study, the photocatalysts showed excellent performances of photocatalytic cyclohexane oxidation under visible light illumination. The yield of KA oil (cyclohexanol and cyclohexanone) reached 112.44 μmol after 5-h illumination of the photocatalyst with the optimal N-doped anatase/rutile ratio, which was 5.66 and 39.87 times that of N-doped anatase TiO₂ and N-doped P25, respectively. In the meantime, the selectivity toward cyclohexanone could reach 100%. Cipagauta-Díaz's group [128] tested the photocatalytic activity of N-doped TiO₂ materials, which were synthesized by a low-temperature microwave-assisted sol-gel method, in the photoreduction of 4-nitrophenol (4-NP) to 4-aminophenol under a visible light source. Although the enhanced photoreduction of 4-NP for N-doped TiO₂ (TiEN*0.5 and TiEN*1) in comparison with

bare TiO₂ was obtained (Fig. 13), the utilization of visible light has a low efficiency.

CO₂ Reduction

The photocatalytic reduction of CO₂ to hydrocarbons has considerable potential to provide alternative fuel to the world and solve the CO₂ emission problem. Since the first report by Inoue et al. [129], tremendous efforts have been exerted to enhance the efficiency of TiO₂-based catalysts for photocatalytic CO₂ reduction. For instance, Bjelajac et al. [116] synthesized N-doped TiO₂ nanoparticles and investigated their photocatalytic CO₂ reduction performance. The results shown in Fig. 14 indicated that the N-doped sample exhibited superior production rates of H₂, CH₄, and CO. By contrast, the P25 catalyst did not affect the productivity of CH₄. One reason for this phenomenon is that the bandgap of the obtained N-doped TiO₂ is smaller; thus, the photon absorption is higher than that of P25. The other reason is the optimal grain sizes of the synthetic samples, as the particles of the P25 sample are generally larger. The larger the grain size is, the lower the charge carriers separation, thus leading to a higher electron consumption rate.

Fig. 13 **a** C/C_0 plot as a function of time for 4-nitrophenol (4-NP) photoreduction; **b** zero-order kinetics plot of 4-NP photoreduction; **c** diffuse reflectance spectroscopy UV–Vis spectra; and **d** Kubelka–Munk modified spectra of the materials TiO₂, TiEN*0.5, and TiEN*1. Reproduced from Ref. [128] with permission. Copyright 2020, Wiley–VCH



To improve the separation efficiency of photoinduced charge carriers, Akple et al. [117] prepared visible light-responsive N-doped anatase TiO₂ microsheets, which have a heterojunction structure with 65% (101) and 35% (001) exposed faces, through a hydrothermal route. Under visible light irradiation, the obtained N-doped TiO₂ exhibited better photocatalytic performance for CO₂ reduction than pure TiN and P25 because of the synergistic effect of surface heterojunctions to improve electron–hole separation and N doping, which is beneficial to visible light absorption. Apart from the pretreatments and surface heterojunctions, the selection of N sources also has an effect on the photocatalytic activity of the N-doped TiO₂ catalyst. Zhang et al. [118] demonstrated that the N–N group in the N source (N₂H₄ or NH₃) plays a decisive role in the selectivity of the main product of photocatalytic reduction on the surface of the obtained N-doped TiO₂ catalysts. They determined that, under visible light illumination, the main products of N–TiO₂ doped with N₂H₄ and NH₃ were CH₄ and CO, respectively. Therefore, they concluded that the presence of N–N groups on the surface of N₂H₄-doped N–TiO₂ provided a reducing environment, which caused CH₄ to become the main reduction product.

Water Splitting

From the perspective of converting solar energy into fuel energy, photocatalytic water splitting has attracted considerable attention. In terms of N-doped TiO₂, scientists have exerted tremendous efforts to investigate the mechanisms and properties under visible light illumination. Yun et al. [119] evaluated the effects of the amount of N in N-doped TiO₂ nanoparticles on O₂ generation via visible light-driven water oxidation and claimed that a small amount of N dopant (up to 3 at%) added to the samples can accelerate the production of O₂ instead of the formation of OH• species. However, larger amounts of N dopant doped into TiO_{2-x}N_x can reduce the rates of both O₂ production and OH• formation, which can be attributed to the low energy level of the VB and the short lifetime of the photoinduced charge carriers. Lee et al. [120] described the improvement of O₂ evolution from water oxidation on the N-doped TiO₂ nanospheres and nanorods prepared by calcination in NH₃ flow. The measurement of photocatalytic activities showed that the two N-doped samples exhibited enhanced O₂ evolution performance than the undoped samples. Recently, Liu's group [96] developed a new nitrogenation method to dope uniform B and N into

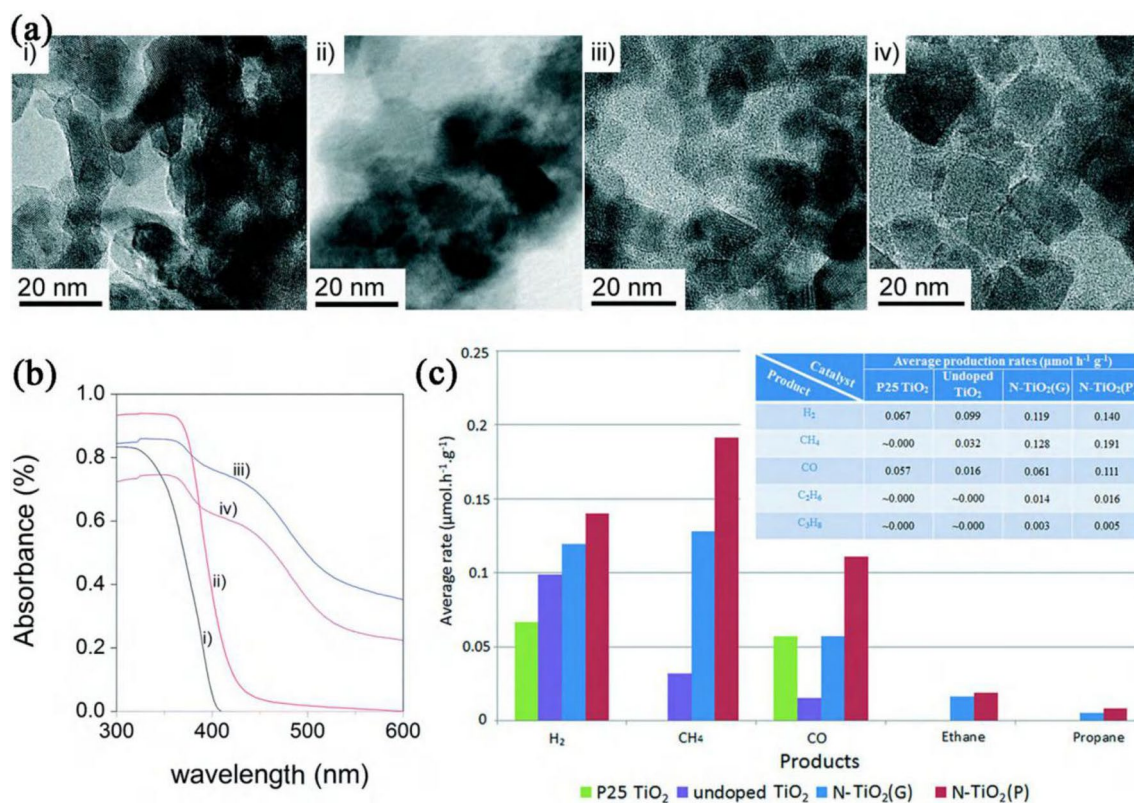


Fig. 14 **a** TEM micrographs and **b** absorption spectra of the investigated samples: (i) P25 TiO₂, (ii) undoped TiO₂, (iii) N-TiO₂(G), and (iv) N-TiO₂(P). **c** Average yield rates of each product for all catalysts (“G” denotes that the gel was directly annealed in an NH₃ atmosphere

in a tube furnace, whereas “P” denotes that the gel was pre-annealed in air at 500 °C for 3 h and annealed in an NH₃ atmosphere). Reproduced from Ref. [116] with permission. Copyright 2020, Royal Society of Chemistry

TiO₂ microspheres with exposed facets [001], as shown in Fig. 15a–e. The obtained red TiO₂ did not show an increase in defects (such as Ti³⁺), which usually exist in doped TiO₂. This decrement may be the result of charge compensation between interstitial B³⁺ and substituted N³⁻, which can hinder the formation of new OV_s in the uniform B,N-co-doped TiO₂. After loading the RuO₂ cocatalyst, the red TiO₂ exhibited a higher photocatalytic O₂ evolution activity and stability than the pristine TiO₂ (Fig. 15f, g). In their work, Wu et al. [97] reported that a C, N substitute was uniformly doped into anatase TiO₂ decahedral plates. Because of the uniform C,N co-doping to narrow the bandgap, the obtained TiO_{2-x}(CN)_y sample had an ideal spectrum of strong band-to-band visible light absorption, which had the capability to induce water oxidation to generate O₂ with CoO_x that used as cocatalyst under visible light illumination.

In addition to photocatalytic O₂ evolution, N-doped TiO₂ materials are often used as photocatalysts for H₂ generation from water reduction. Shangguan and co-workers [121] prepared N-doped TiO₂ catalysts with large specific surface areas by calcining a mixture of TiO₂ and urea and investigated their photocatalytic H₂ generation activity. They confirmed the existence of N on TiO₂ as chemisorbed N₂

molecules and substitutional N on TiO₂ via XPS measurements and summarized that the crystalline phase of TiO₂ and the substitutional N were the main factors improving the photocatalytic performance of water reduction for H₂ generation. Sun et al. [122] prepared anatase N-doped TiO₂ nanobelts with a surface heterojunction of co-exposed facets (101) and (001) and measured their photocatalytic water splitting performance. Based on the research findings, they proposed that, during the catalytic process, the surface heterojunction not only promoted the disintegration of water molecules but also enhanced the space separation of charge carriers to effectively inhibit the recombination of the charge carriers (Fig. 16), thus achieving water reduction for H₂ production under visible light stimulation. The B,N-co-doped TiO₂ prepared by Liu et al. [97] also had the photocatalytic capability to convert water into H₂ when loaded with 0.5 wt% Pt cocatalyst and methanol utilized as a sacrificial agent under visible light illumination.

Although the investigation of N,F-co-doped anatase TiO₂ has been reported, only a few reports are available for the rutile form. In recent work, Maeda et al. [94] focused on N,F-co-doped rutile TiO₂ material (R-TiO₂:N,F), which was synthesized via a nitridation process at 500 °C using

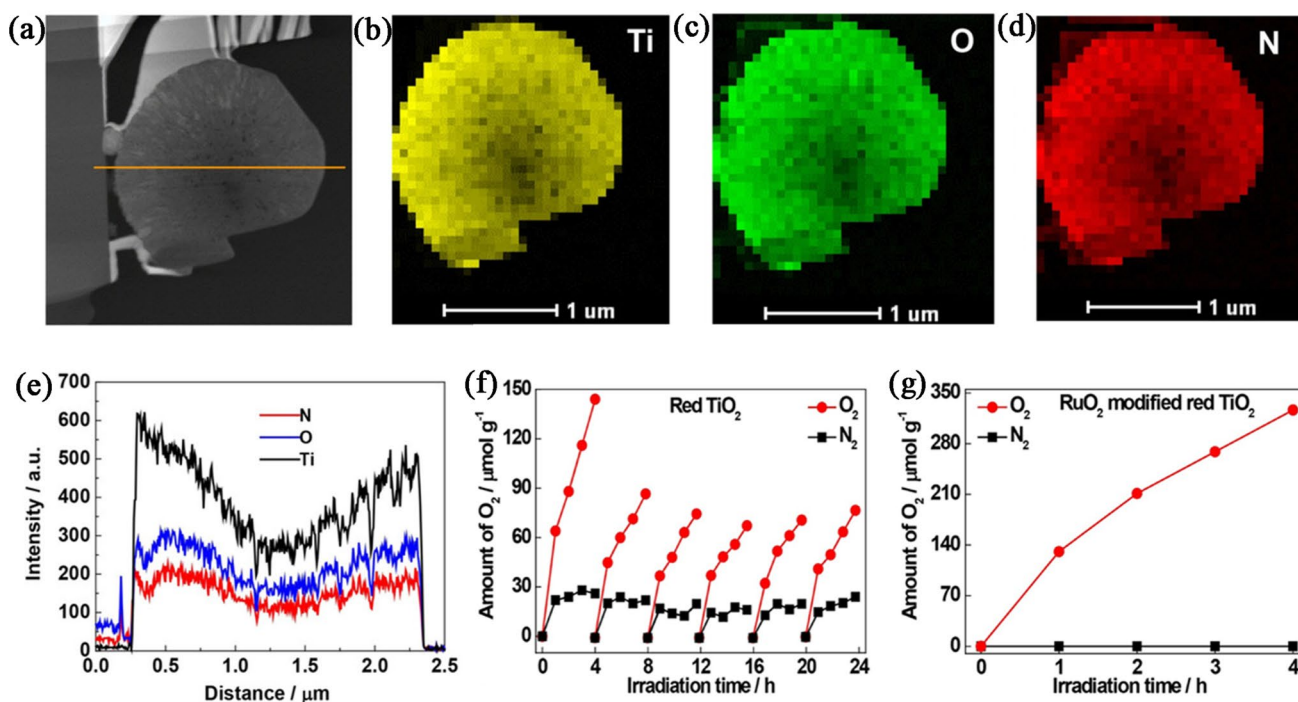
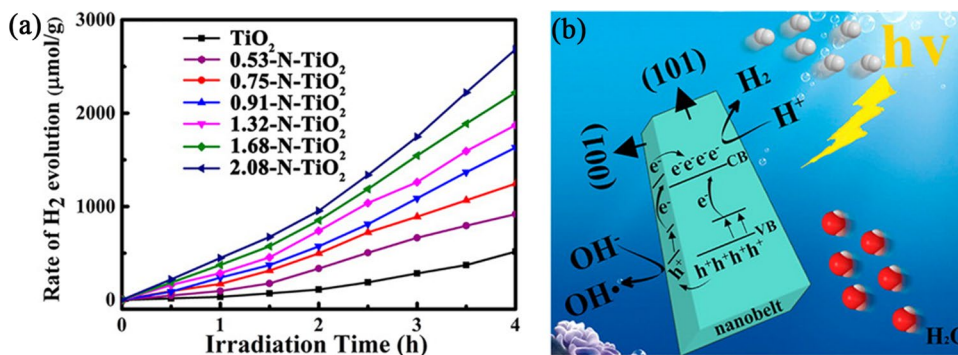


Fig. 15 **a** Scanning TEM high-angle annular dark field image and **b–d** EDS elemental mapping images of Ti–K, O–K, and N–K recorded from the sheet shown in **a**. **e** Line-scan profiles of Ti–K, O–K, and N–K along the orange line in **a**. **f** Cycling tests of the photocatalytic production of O₂ from an AgNO₃ aqueous solution through B,

N-doped TiO₂ microspheres as a function of the exposure time to visible light. **g** Photocatalytic O₂ evolution from an AgNO₃ aqueous solution through B, N-doped TiO₂ modified with 1 wt% RuO₂ cocatalyst as a function of the exposure time to visible light. Reproduced from Ref. [96] with permission. Copyright 2019, Wiley–VCH

Fig. 16 **a** H₂ production rates and **b** diagram of the charge shift process in N-doped TiO₂ nanobelts. Reproduced from Ref. [122] with permission. Copyright 2016, American Chemical Society



rutile TiO₂ and (NH₄)₂TiF₆ as raw materials, as O₂ generation photocatalyst for visible light-driven Z-scheme water splitting (Fig. 17a–d). Notably, the prepared samples showed clear absorption in the visible region, and the more pronounced visible light absorption in the present R-TiO₂:N,F samples was attributed to the increased N content. This material exhibited water oxidation activity with the help of the RuO₂ cocatalyst even in the presence of a reversible electron acceptor, such as IO₃⁻ or Fe³⁺. However, water reduction activity was not observed because of the high Ti³⁺ concentration, which was evidenced by the absorption at longer wavelengths ($\lambda > 600$ nm) [130].

By contrast, as shown in Fig. 17g, the combination of RuO₂-loaded R-TiO₂:N,F and Ru/SrTiO₃:Rh photocatalysts could be used as a visible light-driven Z-scheme water splitting system with [Co(bpy)₃]^{3+/2+} acts as a shuttle redox mediator and exhibited a stoichiometric photocatalytic overall water splitting activity with long-term stability.

Although the solar energy conversion efficiencies of these materials are still low, the methodologies proposed in these studies should contribute to the improvement of the photocatalytic performance of N-doped TiO₂-based visible light-responsive materials.

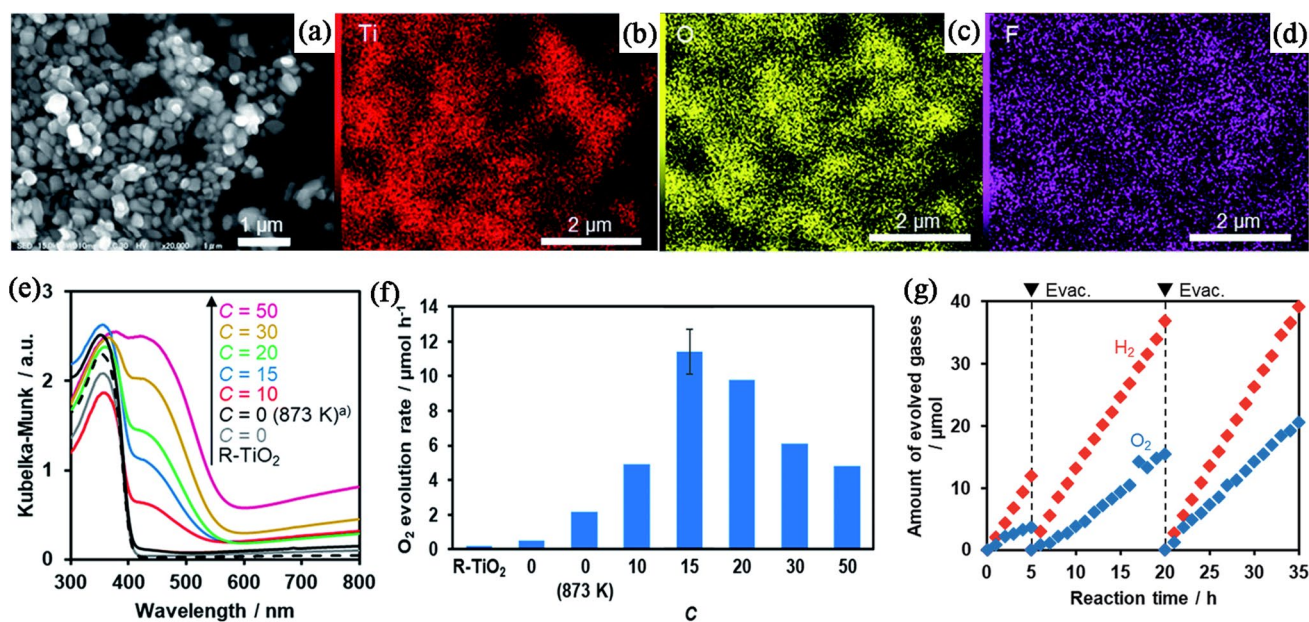


Fig. 17 **a** SEM image, **b–d** EDX elemental mapping data, and **e** UV–Vis diffuse reflectance spectra of R-TiO₂:N,F. **f** Photocatalytic activity of R-TiO₂:N,F for O₂ evolution under visible light irradiation in AgNO₃ aqueous solution. Reaction conditions: catalyst, 50 mg; La₂O₃, 200 mg; 10 mmol/LL AgNO₃ aqueous solution, 140 mL; light source, 300 W Xe lamp fitted with a 420-nm cutoff filter. **g** Time

course of H₂ and O₂ evolution from mixtures of RuO₂/R-TiO₂:N,F (50 mg) and Ru/SrTiO₃:Rh (25 mg) dispersed in an aqueous solution (120 mL) containing tris(2,20-bipyridyl)cobalt(II) sulfate (0.5 mmol/LL) under visible light irradiation ($\lambda > 420$ nm). Reproduced from Ref. [94] with permission. Copyright 2018, Royal Society of Chemistry

Summary and Outlook

This review attempts to provide a comprehensive update and summary focusing on the synthesis, characterizations, and practical applications of visible light-responsive N-doped TiO₂ photocatalyst. N-doped TiO₂ not only effectively reduces the bandgap but also reduces the recombination efficiency of photoinduced charge carriers. Although the detailed source of photocatalysis of this material remains a debate, its basic principles and mechanisms have guided the search for other novel similar photocatalysis materials. Generally, the photocatalytic reaction rates of N-doped TiO₂ are still low. The findings can be attributed to the fact that the visible light absorption of N-doped samples is still poor, which cannot achieve the ideal band-to-band absorption. Moreover, with the increase in N content, the introduced OV_s can act as recombination centers, thus resulting in a decreased photocatalytic reaction rate. Despite tremendous efforts that have been exerted in the area of enhancing the visible light absorption of TiO₂-based materials, the following investigations need to be conducted in the future:

- (1) To explore the synergistic effect of various synthesis strategies, as no single solution can make materials suitable for different applications;
- (2) How co-doping of various N precursors and other non-metals affects the band structure and photocatalytic performance of TiO₂ remains to be clarified;
- (3) Although most of these N-doped TiO₂ samples can individually produce H₂ or O₂ from H₂O splitting in the presence of appropriate sacrificial agents under visible light illumination, which indicates that overall water splitting is possible in principle, it has not yet been realized. Further effort is required to realize the utilization of TiO₂-based catalysts in photocatalytic overall water splitting activity under visible light irradiation;
- (4) The use of cocatalysts in photocatalytic water splitting systems can not only provide more active sites for redox reactions on the surface of the semiconductors but also accelerate the extraction and transfer of generated electrons for surface reactions. Recently, single-atom catalysts (SACs) with 100% atom utilization have shown superior performances in heterogeneous catalysis fields, such as photocatalytic reactions. In a sense, SACs could be employed as cocatalysts for N-doped TiO₂ materials to enhance their photocatalytic activities because of their fascinating strengths in enhancing light harvesting, charge transfer dynamics, and surface reactions of a photocatalytic system;
- (5) Conjugated polymers (CPs) have been used to produce solar energy conversion materials in photovoltaics because of their outstanding light-harvesting and low-

cost processing. However, their photocatalytic activity has only recently been highlighted with the preparation of robust, metal-free, and visible or near-IR light-active photocatalysts. In the future, hybrid photocatalysts based on N-doped TiO₂ semiconductors and CPs can be fabricated and used as novel promising photoactive materials for pollutant degradation, energy conversion through water splitting, and/or CO₂ reduction using photocatalytic processes.

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Declarations

Conflict of interest The authors declare that there is no conflict of interest.

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