



Simultaneously Efficient Solar Light Harvesting and Charge Transfer of Hollow Octahedral Cu₂S/CdS p–n Heterostructures for Remarkable Photocatalytic Hydrogen Generation

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Abstract

Solar-driven water splitting is a promising alternative to industrial hydrogen production. This study reports an elaborate design and synthesis of the integration of cadmium sulfide (CdS) quantum dots and cuprous sulfide (Cu₂S) nanosheets as three-dimensional (3D) hollow octahedral Cu₂S/CdS p–n heterostructured architectures by a versatile template and one-pot sulfidation strategy. 3D hierarchical hollow nanostructures can strengthen multiple reflections of solar light and provide a large specific surface area and abundant reaction sites for photocatalytic water splitting. Owing to the construction of the p–n heterostructure as an ideal catalytic model with highly matched band alignment at Cu₂S/CdS interfaces, the emerging internal electric field can facilitate the space separation and transfer of photoexcited charges between CdS and Cu₂S and also enhance charge dynamics and prolong charge lifetimes. Notably, the unique hollow Cu₂S/CdS architectures deliver a largely enhanced visible-light-driven hydrogen generation rate of 4.76 mmol/(g·h), which is nearly 8.5 and 476 times larger than that of pristine CdS and Cu₂S catalysts, respectively. This work not only paves the way for the rational design and fabrication of hollow photocatalysts but also clarifies the crucial role of unique heterostructure in photocatalysis for solar energy conversion.

Keywords Hollow octahedrons · P–n heterostructure · Charge separation · Photocatalytic water splitting

Introduction

Since the increasingly serious global environmental pollution and energy crisis greatly influence the sustainable development of society, it is urgent to explore clean and renewable solar energy conversion systems [1]. Photocatalytic water

splitting has attracted tremendous attention as a catholicon to the worldwide energy dilemma, converting solar energy resources into clean oxygen and hydrogen fuels [2]. To date, a large number of semiconductors such as metal oxides, chalcogenides, nitrides, and oxynitrides have been widely employed as photocatalytic materials [3]. Specifically, metal chalcogenides exhibit great potential in the field of solar energy conversion [4]. Among these semiconductor materials, due to various merits such as narrow bandgap and suitable band energy position, the n-type CdS is regarded as a prominent sulfide for solar energy conversion, particularly in solar hydrogen generation [3]. Meanwhile, the p-type copper sulfide is a good catalyst due to its relatively narrow bandgap and suitable band edge positions [3, 4]. However, a series of dilemmas lie in single metal sulfide-based photocatalysts, such as limited light harvesting, poor charge-carrier mobility, high bulk recombination, and large surface kinetic barrier for water-splitting reaction, resulting in unsatisfactory photoconversion efficiency and poor photocatalytic stability.

It is well known that nanostructure engineering has been regarded as a promising approach to regulate the

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light-harvesting capacity and endow active sites to optimize the charge separation and transfer, thus promoting photocatalytic performance [5–7]. Among various nanostructures, three-dimensional (3D) hierarchical materials received extensive attention owing to their unique physical and chemical advantages. Particularly, hollow architectures with a larger surface area and abundant reactive sites can enhance solar light utilization and photocatalytic performance [8–14]. For example, hierarchical $\text{Co}_9\text{S}_8@\text{ZnIn}_2\text{S}_4$ heterostructures as photocatalysts exhibited an excellent hydrogen production rate of $6250 \mu\text{mol}/(\text{g}\cdot\text{h})$ for photocatalytic water splitting [12]. Hierarchical $\text{Co}/\text{NGC}@\text{ZnIn}_2\text{S}_4$ hollow heterostructures without cocatalyst presented superior hydrogen generation activity and outstanding stability [13]. Hollow La/Rh -doped SrTiO_3 nanostructures showed a highly efficient hydrogen generation rate owing to their high light absorption ability [14]. Thus, it is a challenge to develop hollow metal chalcogenides nanostructures for photocatalytic water splitting.

Owing to the sluggish charge-carrier separation in various semiconductors, the photocatalytic performance is still far below what is expected. Aiming at these issues, diverse approaches have been exploited to address, such as heteroatoms doping [15], dye sensitization [16], defect engineering [17, 18], and heterostructure engineering [19, 20]. Among these strategies, heterojunction engineering has been considered an effective tool to suppress the photoexcited charge recombination and optimize photocatalytic performance. For instance, $\text{Co}_9\text{S}_8/\text{CdS}$ hollow nanocube photocatalyst showed enhanced light capture and photocatalytic activity compared to solid $\text{Co}_9\text{S}_8/\text{CdS}$ nanocubes [21]. $\text{NiS}/\text{Zn}_x\text{Cd}_{1-x}\text{S}$ heterojunctions were synthesized by the direct conversion of the metal–organic framework, yielding a high photocatalytic hydrogen generation rate and high stability under visible-light irradiation [22]. $\text{Co}_9\text{S}_8/\text{Cd}/\text{CdS}$ tubular heterostructures exhibited a photocatalytic hydrogen generation rate of up to $10.42 \mu\text{mol}/\text{h}$ due to the Z-scheme transition metal bridge [23]. Porous CuS/ZnS nanosheet photocatalysts prepared by hydrothermal and cation exchange reaction showed a high photocatalytic hydrogen generation rate of $4147 \mu\text{mol}/(\text{g}\cdot\text{h})$ based on the photoinduced interfacial charge transfer process [24]. Inspired by nanostructure engineering and heterostructure engineering, it is important to construct hollow heterostructures of metal chalcogenides for enhanced photocatalytic hydrogen production.

This work reports an elaborate design and synthesis of the integration of CdS quantum dots (QDs) and Cu_2S nanosheets (NSs) as 3D hollow octahedral $\text{Cu}_2\text{S}/\text{CdS}$ p–n heterostructured architectures by a versatile template and one-pot sulfidation strategy. Compared to the pristine hollow Cu_2S and CdS photocatalysts, as-synthesized hollow octahedral $\text{Cu}_2\text{S}/\text{CdS}$ p–n heterostructures exhibited an impressively high photocatalytic hydrogen generation rate of $4.76 \text{ mmol}/(\text{g}\cdot\text{h})$

and excellent stability under solar energy irradiation, which was over 8.5 and 476 times higher than that of pure CdS and hollow Cu_2S octahedrons photocatalysts, respectively. Notably, the enhanced photocatalytic hydrogen generation performance of hollow octahedral $\text{Cu}_2\text{S}/\text{CdS}$ p–n heterostructures is ascribed to the strong light-harvesting capacity, efficient charge transfer and separation, and abundant active sites through the construction of hollow architectures.

Experimental

Fabrication of $\text{Cu}_2\text{S}/\text{CdS}$ Heterostructures

Cu_2O octahedrons were synthesized by a modified approach from a previous report [25]. A certain amount of Cu_2O octahedrons was uniformly dispersed in 20 mL of deionized (DI) water under ultrasonic treatment for 10 min. $\text{Cd}(\text{CH}_3\text{COO})_2\cdot 2\text{H}_2\text{O}$ was then added to the above solution under magnetic stirring for 20 min. Then, 10 mL of Na_2S aqueous solution (0.01 M) was added to the uniform precursor solution at room temperature. The products were obtained by centrifugation, washed three times, and dried in a vacuum condition. Thus, a sequence of $\text{Cu}_2\text{S}/\text{CdS}$ - X with $X=0, 5\text{wt}\%, 10\text{wt}\%,$ and $20\text{wt}\%$ were named as Cu_2S , $\text{Cu}_2\text{S}/\text{CdS}$ -5, $\text{Cu}_2\text{S}/\text{CdS}$ -10, and $\text{Cu}_2\text{S}/\text{CdS}$ -20, respectively. In comparison, CdS nanoparticles were obtained by a similar synthesis process except for the absence of a Cu_2O precursor.

Instruments and Photocatalysts Characterization

Powder X-ray diffraction (XRD) patterns were obtained by an X-ray diffractometer equipped with graphite monochromatized high-intensity $\text{Cu K}\alpha$ radiation ($\lambda = 1.54178 \text{ \AA}$). SEM images were captured by a field-emission scanning electron microscope (FESEM, FEI Nova Nano SEM 450). TEM images were performed on a transmission electron microscope (TEM, FEI TF30) and a high-resolution TEM (HRTEM). The N_2 adsorption–desorption curves were obtained by the Micromeritics ASAP 2460 surface areas and porosities profiler. Specific surface areas were acquired via the Brunauer–Emmett–Teller (BET) approach, and X-ray photoelectron spectroscopy (XPS, ESCALAB 250) was employed to explore the elemental composition and valence states of materials. Finally, UV–Vis absorption and photoluminescence (PL) spectra were measured by a UV–Vis–NIR spectrophotometer (Shimadzu UV-3600 Plus) and a fluorescence spectrometer (Horiba, FluoroMax-4P).

Photocatalytic Tests

Photocatalytic hydrogen generation tests were performed using visible-light illumination. The reaction vessel used in

this study was a unilateral irradiation Pyrex container linked to a sealed gas circulator and exhaust system under room temperature, in which a xenon light source (300 W) was placed ~25 cm away from the system. For a regular photocatalytic water-splitting test, a certain amount (5 mg) of different photocatalysts and lactic acid (LA, 1 mL) as the hole scavenger were dispersed in 9 mL DI water. Before illumination, the reaction cell was ensured under anaerobic conditions without oxygen by a degassing process. In the photocatalytic process, the reactant system steadily proceeded under visible-light irradiation ($\lambda > 400$ nm) and ambient temperature by a circulating water system. To assess the photocatalytic stability of the photocatalyst in a long-term operation, the recycled photocatalytic hydrogen production measurements were performed under the same condition. The apparent quantum efficiency (AQE) can be determined via the same photocatalytic test procedure under visible-light irradiation. The generated hydrogen production amount was analyzed via a gas chromatograph (GC2014).

Results and Discussion

Synthesis and Characterization

Figure 1a presents the elaborate design and synthesis of the hollow octahedral $\text{Cu}_2\text{S}/\text{CdS}$ p–n heterostructure as an ideal catalytic model. First, Cu_2O octahedrons were produced as a template. Second, hollow octahedral $\text{Cu}_2\text{S}/\text{CdS}$ heterostructures assembled by Cu_2S nanosheets and CdS quantum dots were fabricated by this versatile sulfidation synthesis strategy. FESEM and TEM were employed to explore the morphology and structure evolution of as-prepared $\text{Cu}_2\text{S}/\text{CdS}$ heterostructure. As shown in Fig. 1b, Cu_2O nano-octahedrons exhibit an average size of ~400 nm. SEM images of Cu_2S and $\text{Cu}_2\text{S}/\text{CdS}$ present the nanosheet-assembled architectures (Fig. 1c and d). To confirm, EDS mappings were obtained as shown in Fig. 1e–g and Fig. S1, which indicate the existence of elements Cu, Cd, and S, demonstrating the homogeneous distribution in Cu_2S and $\text{Cu}_2\text{S}/\text{CdS}$ octahedrons. In comparison, the morphology and average size of $\text{Cu}_2\text{S}/\text{CdS}$ octahedrons are similar to those of Cu_2S (Figs. S1 and S2), exhibiting a hierarchical morphology and a size of ~500 nm for Cu_2S and $\text{Cu}_2\text{S}/\text{CdS}$ octahedrons.

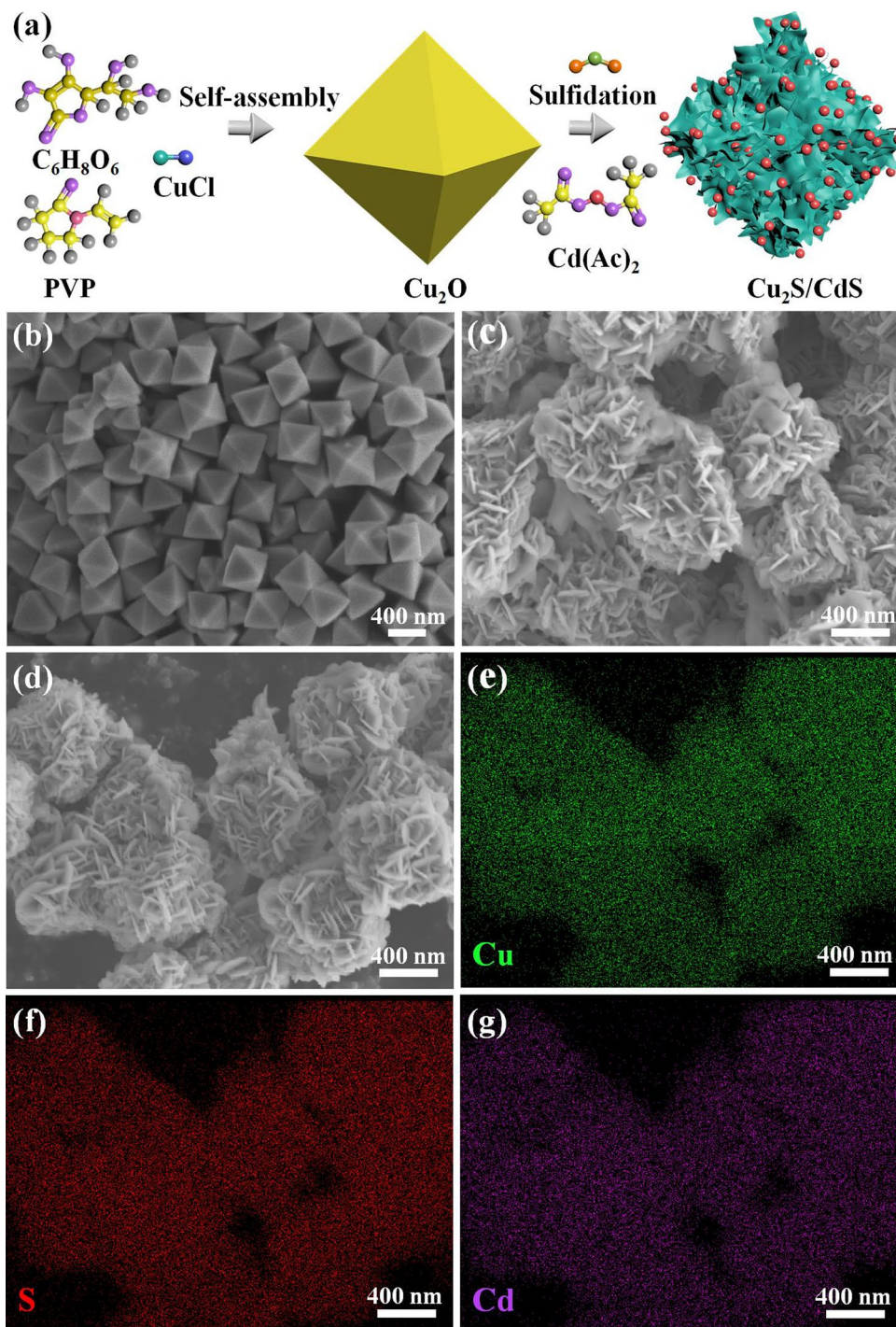
TEM images of $\text{Cu}_2\text{S}/\text{CdS}$ hybrid in Fig. 2a and b reveal an octahedral and hollow nanostructure with a homogeneous size of 500 nm, indicating that the one-pot sulfidation treatment can convert the solid into nano-octahedrons with ultrathin nanosheet-assembled hierarchical morphology (Fig. 2c). Dispersed Cu_2S hollow octahedrons were also evident in the TEM image (Fig. S2). As shown in the HRTEM image in Fig. 2d, the lattice fringe of 0.34 nm corresponds well to the (111) plane of a cubic CdS phase, revealing

the favorable crystallinity of CdS [26], while the interplanar spacing of 0.189 nm is ascribed to the (573) facet of a tetragonal Cu_2S [27]. The selected-area electron diffraction pattern in Fig. 2e shows that the diffraction rings correspond well with the (573) plane of Cu_2S and the (220) plane of CdS, demonstrating the successful preparation of the $\text{Cu}_2\text{S}/\text{CdS}$ binary composite. The corresponding elemental mapping images obtained via high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) for $\text{Cu}_2\text{S}/\text{CdS}$ present a uniform distribution of Cu, Cd, and S species (Fig. 2f–i). Thus, SEM and TEM images indicate that the hollow $\text{Cu}_2\text{S}/\text{CdS}$ octahedron heterostructures can be synthesized by this versatile strategy.

Furthermore, XRD patterns were employed to verify the phase composition of the products. The observed diffraction peaks in Fig. 3a belong to the reflections of tetragonal Cu_2S (JCPDF 02–1294), while the other diffraction peaks are indexed to the reflections of cubic CdS (JCPDS 10–0454), confirming the synthesis of $\text{Cu}_2\text{S}/\text{CdS}$ composites. XPS was also performed to check the elemental compositions and valence states. In the full range survey spectrum of $\text{Cu}_2\text{S}/\text{CdS}$ (Fig. 3b), the peaks of Cu 2p, Cd 3d, and S 2p can be examined. The two binding energies that are centered at 932.3 eV and 952.2 eV in the Cu 2p XPS spectrum (Fig. 3c) can be ascribed to Cu 2p_{3/2} and Cu 2p_{1/2}, respectively, confirming the presence of Cu⁺ in Cu_2S [28]. However, the Cd 3d XPS spectrum (Fig. 3d) reveals two obvious peaks at 405.0 and 411.7 eV, which correspond to Cd 3d_{5/2} and Cd 3d_{3/2}, proving the presence of Cd²⁺ in the binary hybrid [29]. Meanwhile, the peaks at 161.8 and 162.7 eV in the S 2p XPS spectrum (Fig. 3e) are derived from S 2p_{3/2} and S 2p_{1/2}, indicating the existence of the S²⁻ bonding to Cd²⁺ and Cu⁺ [30]. Similarly, Cu⁺ and S²⁻ species can also be detected in Cu_2S from the analysis of Cu 2p and S 2p XPS spectra in Fig. S3. Moreover, nitrogen adsorption–desorption tests were applied. As shown in Fig. 3f, all products display analogical shape plots, in which characteristic type IV curves with H3 hysteresis loops ($P/P_0 > 0.4$) are obviously detected by the Brunauer–Emmett–Teller (BET) surface area analysis, proving the mesopore feature of the products. The Barrett–Joyner–Halenda (BJH) analysis of the pore size distribution plots (Fig. S4) presents the specific surface area, pore size, and pore volume. $\text{Cu}_2\text{S}/\text{CdS}$ exhibits the largest specific surface area (129.7 m²/g) among all products, thereby affording superior photocatalytic activity due to abundant active sites [31]. The larger pore size of $\text{Cu}_2\text{S}/\text{CdS}$ than that of Cu_2S may have resulted from the filling of CdS QDs into the mesoporous $\text{Cu}_2\text{S}/\text{CdS}$ heterostructures.

To illustrate the vital function of the hollow nanostructure for solar energy capture, UV–Vis absorption spectra were recorded to explore the light response features of the products. As shown in Fig. 5a, a characteristic absorption edge in the visible-light range can be observed at ~510 nm

Fig. 1 **a** The synthesis protocol for the construction of hollow $\text{Cu}_2\text{S}/\text{CdS}$ heterostructure. SEM images of **b** Cu_2O , **c** $\text{Cu}_2\text{S}/\text{CdS}$, and **d** Cu_2S element mappings of $\text{Cu}_2\text{S}/\text{CdS}$ for **e** copper, **f** sulfur, and **g** cadmium



for CdS [32]. Notably, hollow octahedrons exhibit a stronger light absorption than other products due to the multiple reflections of incident light inside the hollow nanostructure [8–14]. The improved light capture derived from the hollow structure is expected to enhance the light response capacity of $\text{Cu}_2\text{S}/\text{CdS}$ for photocatalytic application. Meanwhile, the bandgap energies of the products can be determined using the Tauc's relation based on the absorption edges. As displayed in Fig. S5, the calculated

bandgap values of Cu_2S and CdS are 1.1 and 2.4 eV, respectively. Mott–Schottky curves (Fig. S6) were used to explore the band positions of the semiconductors [33], corresponding the bandgap structures of CdS and Cu_2S in Fig. S7. Based on the positions of conduction band (CB) and valence band (VB) for CdS and Cu_2S , the photoexcited electrons and holes can be effectively transferred and separated in $\text{Cu}_2\text{S}/\text{CdS}$ heterostructures. Thus, the construction of $\text{Cu}_2\text{S}/\text{CdS}$ heterostructures is beneficial to the charge

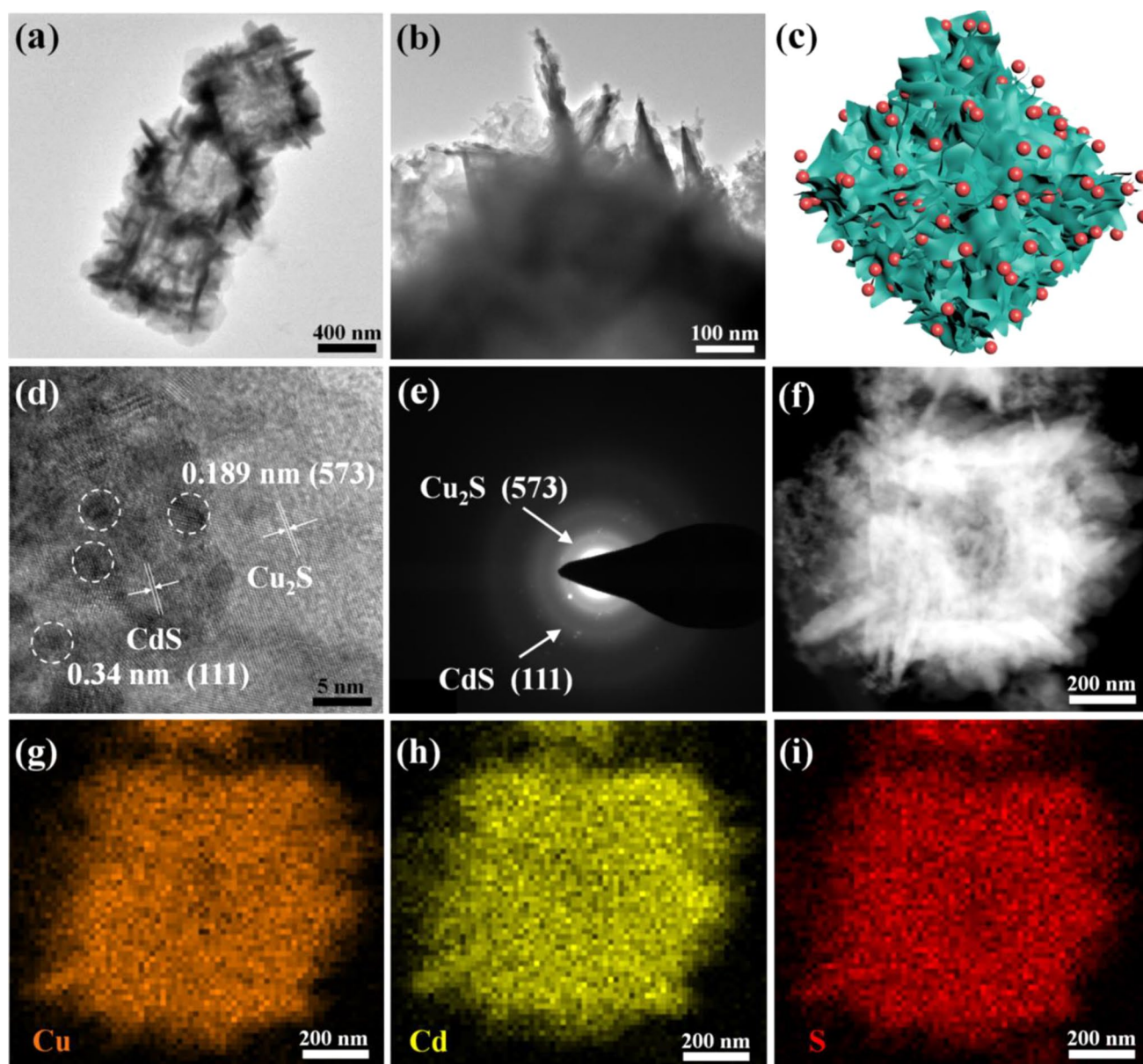


Fig. 2 **a, b** TEM images, **c** illustration of the heterostructure, **d** HRTEM image, **e** selected-area electron diffraction pattern, **f** dark-field STEM image and HAADF-STEM element mappings of $\text{Cu}_2\text{S}/\text{CdS}$ for **g** copper, **h** cadmium, and **i** sulfur

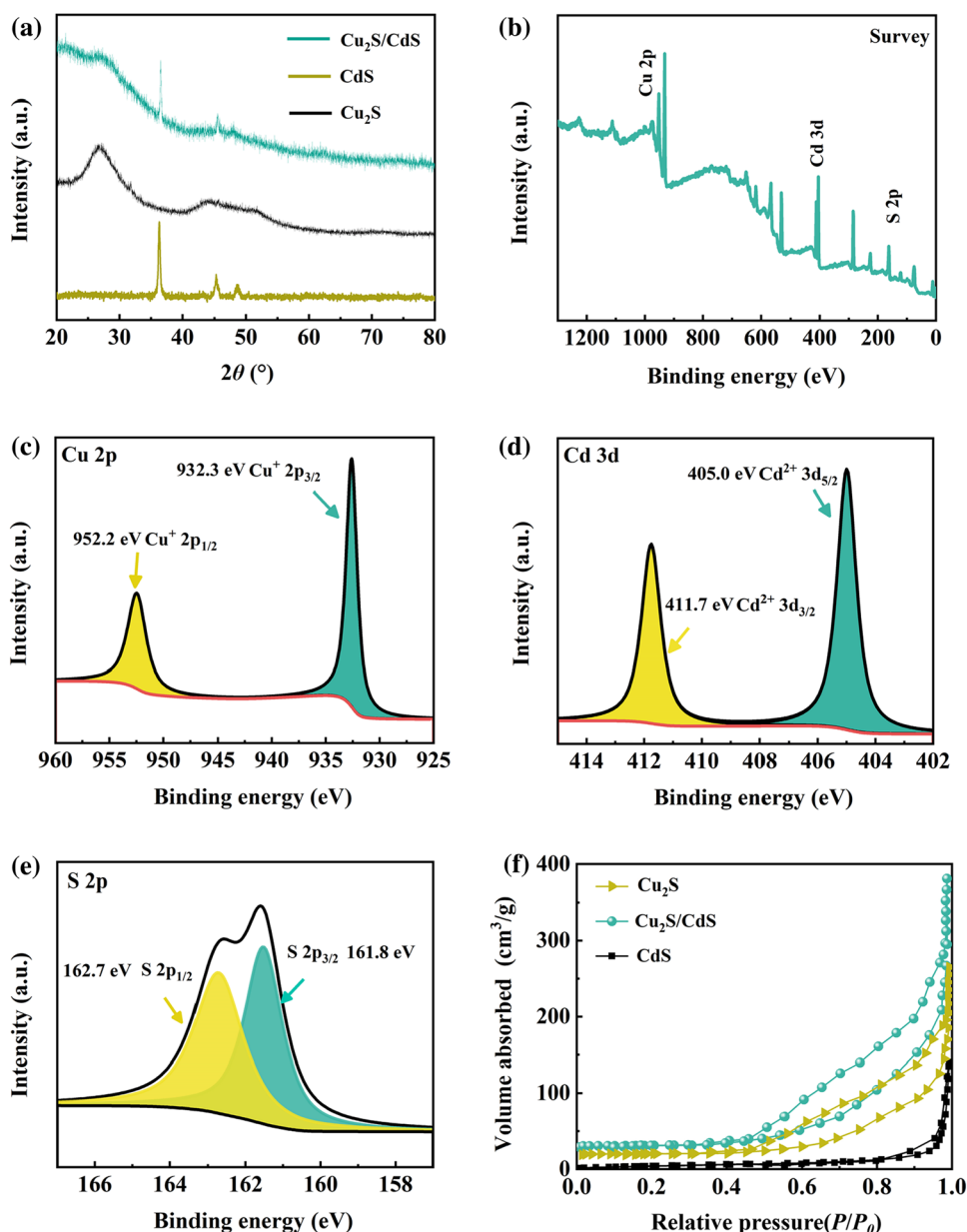
transfer and separation, which can consequently enhance the photocatalytic performance.

Photocatalytic Activity and Mechanism Analysis

The photocatalytic hydrogen production of different catalysts was conducted under visible-light irradiation ($\lambda > 400 \text{ nm}$). As depicted in Fig. 4a, both CdS and Cu_2S catalysts present low H_2 generation rates. However, all $\text{Cu}_2\text{S}/\text{CdS}$ heterostructures exhibit a significantly improved photocatalytic H_2 generation rate by loading various amounts of CdS quantum dots on the surface of Cu_2S nanosheets. With the increasing

content of CdS QDs, the photocatalytic H_2 generation rate increases for the as-prepared $\text{Cu}_2\text{S}/\text{CdS}$ heterostructures. However, a decreased H_2 generation rate was observed for the $\text{Cu}_2\text{S}/\text{CdS}$ heterostructures with high content of up to 20 wt% of CdS QDs. In comparison, the obtained hollow and octahedral $\text{Cu}_2\text{S}/\text{CdS}$ -10 photocatalyst exhibits an average hydrogen generation rate of $4.76 \text{ mmol}/(\text{g}\cdot\text{h})$, which is 8.5 and 476 times larger than that of CdS ($0.56 \text{ mmol}/(\text{g}\cdot\text{h})$) and Cu_2S ($0.01 \text{ mmol}/(\text{g}\cdot\text{h})$), respectively. This hydrogen generation rate is higher than those of many photocatalysts such as CdS/Ni-MOF, RP/CoP/ $\text{Cd}_{0.9}\text{Zn}_{0.1}\text{S}$, Au/CdS/ZnO, BP/ Cu_7S_4 , $\text{Cd}_{0.5}\text{Zn}_{0.5}\text{S}$, $\text{BiVO}_4/\text{Au}/\text{CdS}$, CdS/Ni(OH) $_2$,

Fig. 3 **a** XRD patterns of Cu_2S , CdS , and $\text{Cu}_2\text{S}/\text{CdS}$. **b** Survey spectrum, high-resolution XPS spectrum of **c** Cu 2p region, **d** Cd 3d region, and **e** S 2p region for $\text{Cu}_2\text{S}/\text{CdS}$. **f** N_2 adsorption-desorption isotherms of Cu_2S , CdS , and $\text{Cu}_2\text{S}/\text{CdS}$



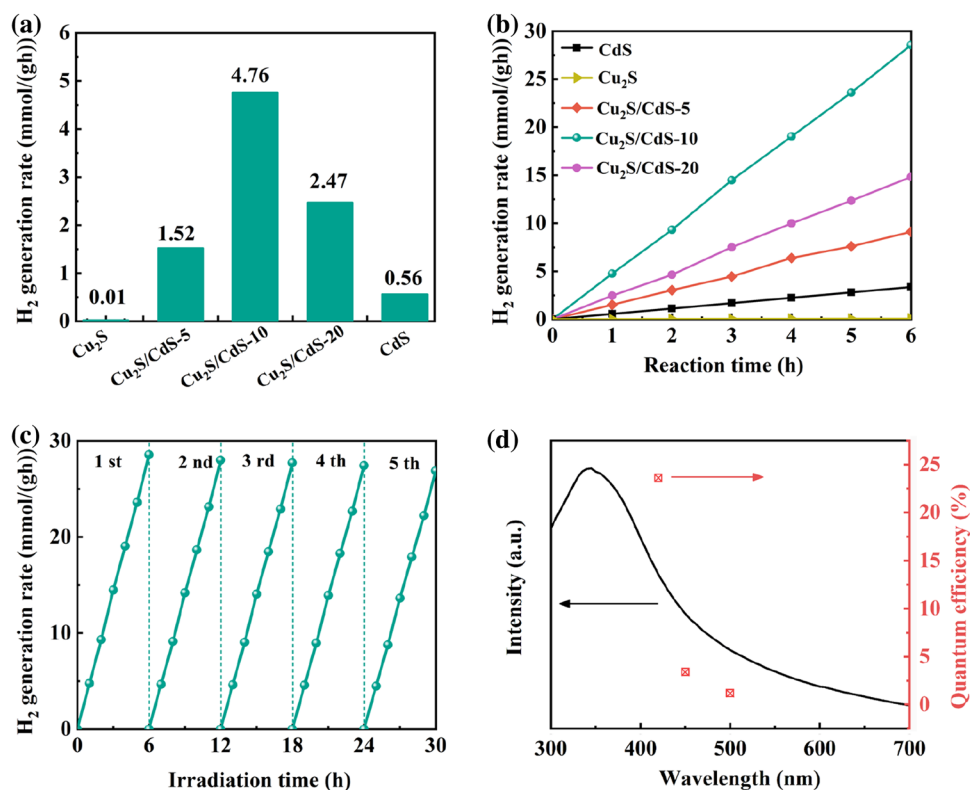
$\text{Cu}_7\text{S}_4/\text{CdS}$, and $\text{Pt-CdS}/\text{g-C}_3\text{N}_4$ (Table S1). Moreover, the relationship between photocatalytic activity and illumination time was also probed. As shown in Fig. 4b, all $\text{Cu}_2\text{S}/\text{CdS}$ heterostructures exhibit a linear increasing trend for photocatalytic hydrogen generation during 6-h light illumination.

Since photocatalytic hydrogen generation stability is an essential standard of superior photocatalysts, a stability test was performed for a continuous 30-h illumination. A constant H_2 generation rate of $\text{Cu}_2\text{S}/\text{CdS}$ was observed (Fig. 4c), suggesting its excellent stability. In contrast, the H_2 generation rate for the pristine CdS catalyst (Fig. S8) lowers by 35.6% under the same condition. The corresponding

AQE was then calculated. $\text{Cu}_2\text{S}/\text{CdS}$ shows high AQE values of ~23.6%, 3.4%, and 1.2% at 420, 450, and 500 nm (Fig. 4d), respectively. Particularly, the original hollow octahedron architecture is still maintained for the $\text{Cu}_2\text{S}/\text{CdS}$ hybrid after long-term photocatalytic reaction as evident in the SEM image (Fig. S9), indicating outstanding stability of $\text{Cu}_2\text{S}/\text{CdS}$ heterostructures.

Photoluminescence spectroscopy (PL) was conducted to verify the charge transfer behavior of the products. As displayed in Fig. 5b, $\text{Cu}_2\text{S}/\text{CdS}$ exhibits the lowest PL intensity compared to the pristine CdS and Cu_2S , indicating that the heterostructure can remarkably suppress the photoexcited

Fig. 4 **a** Apparent hydrogen generation rates and **b** photocatalytic hydrogen generation in 6 h under visible-light irradiation of $\text{Cu}_2\text{S}/\text{CdS}$ with various amounts of CdS. **c** Photocatalytic hydrogen generation stability of $\text{Cu}_2\text{S}/\text{CdS}$. **d** Wavelength dependence of apparent quantum efficiency and UV–Vis absorption spectra for $\text{Cu}_2\text{S}/\text{CdS}$

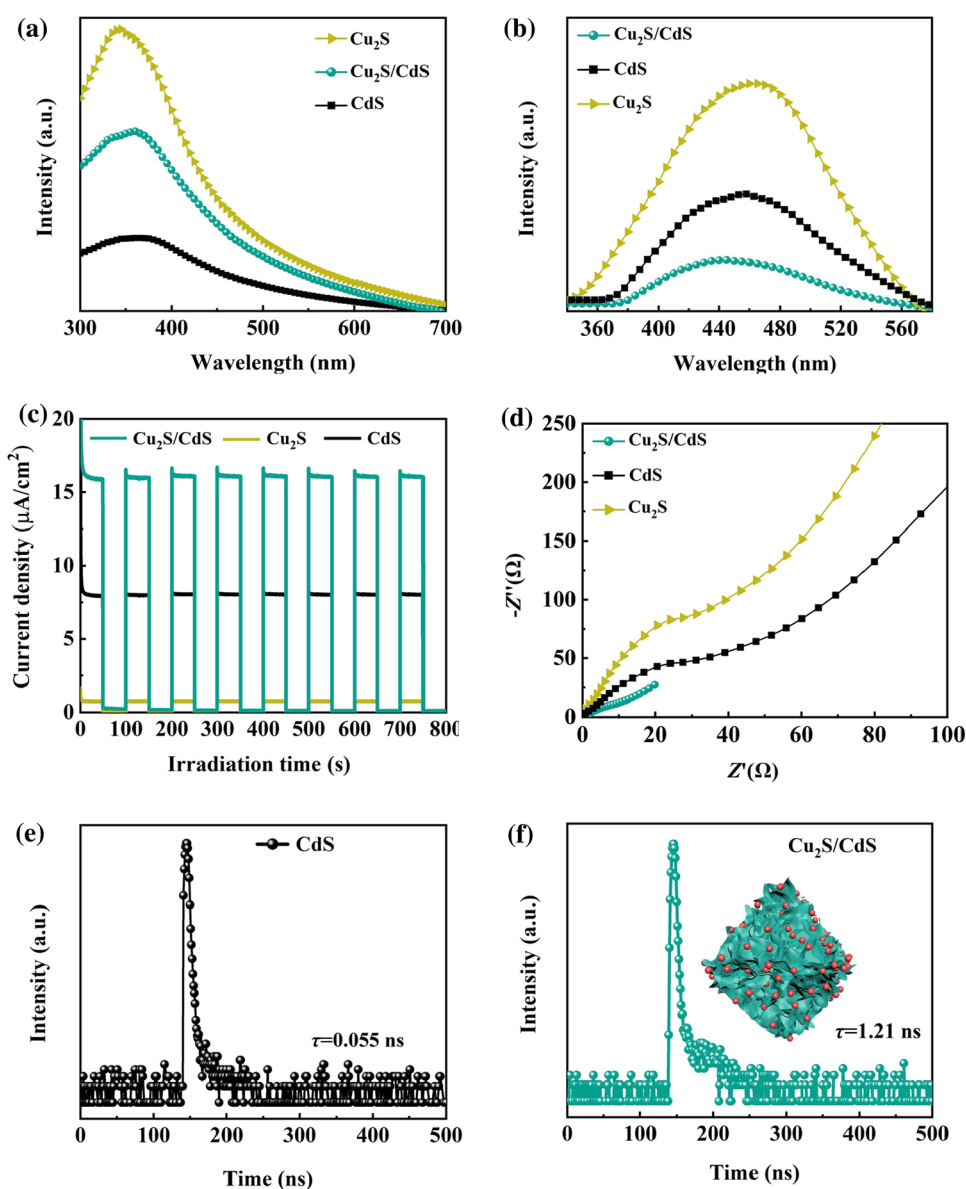


charge recombination, consequently promoting excellent photocatalytic performance [34]. Photoelectrochemical (PEC) water splitting measurements were also performed to study the photoinduced charge generation and separation of Cu_2S , $\text{Cu}_2\text{S}/\text{CdS}$, and CdS (Fig. S10). As presented in Fig. 5c, the amperometric photocurrent–time curves were conducted under chopped solar irradiation. The $\text{Cu}_2\text{S}/\text{CdS}$ heterostructure exhibits a swift and renewable photocurrent production under switching on/off illumination. Particularly, the $\text{Cu}_2\text{S}/\text{CdS}$ heterostructure photoelectrode exhibits a larger photocurrent density than those of the pristine Cu_2S and CdS . The as-synthesized $\text{Cu}_2\text{S}/\text{CdS}$ photoelectrode displays the largest photocurrent density of $16.9 \mu\text{A}/\text{cm}^2$, which is larger than those of the pristine Cu_2S and CdS . This indicates that the photoinduced electron–hole pairs can be efficiently separated under solar illumination. Furthermore, electrochemical impedance spectroscopy plots were also recorded to observe the photoinduced charge transfer kinetics on the interface of different photocatalysts [35–42]. Obviously, the small semicircle of the $\text{Cu}_2\text{S}/\text{CdS}$ heterostructure in comparison with the pristine Cu_2S and CdS is observed in Fig. 5d, revealing the small charge transfer resistance and faster interfacial charge transport in $\text{Cu}_2\text{S}/\text{CdS}$. On account of the above analysis, the introduction of CdS QDs plays an important role in the remarkable

photocatalytic hydrogen production activities. Furthermore, the charge transportation efficiencies of CdS and $\text{Cu}_2\text{S}/\text{CdS}$ were quantitatively probed via the time-resolved photoluminescence spectroscopy. From the fitted results in Fig. 5e and f, the $\text{Cu}_2\text{S}/\text{CdS}$ heterostructure exhibits a longer decay time ($\tau = 1.21 \text{ ns}$) than that of pure CdS ($\tau = 0.055 \text{ ns}$). The remarkably prolonged lifetime of the $\text{Cu}_2\text{S}/\text{CdS}$ hybrid reveals the accelerated separation of photoinduced electron/hole pairs in the $\text{Cu}_2\text{S}/\text{CdS}$ heterostructure system. Particularly, the electron–hole pairs can be directionally transferred by the internal electric field as the driving force and be accumulated on different semiconductors due to the well-matched band alignment. Thus, charge recombination is effectively restrained, consequently enhancing the charge transfer and separation of $\text{Cu}_2\text{S}/\text{CdS}$ heterostructures.

According to the aforementioned analysis, a possible mechanism for excellent photocatalytic water splitting of $\text{Cu}_2\text{S}/\text{CdS}$ heterostructures can be proposed. First, the hollow nanostructure allows multiple reflections of solar light inside the cavity, generating a large number of electron–hole pairs in the binary composite and, therefore, enhancing photocatalytic water-splitting performance. Moreover, the unique hierarchical nanostructures by the integration of CdS QDs and Cu_2S NSs can provide a larger specific surface area and porous structure, provide more active sites for the

Fig. 5 **a** UV–Vis absorption spectra, **b** photoluminescence spectra, **c** photocurrent responses under chopped light, **d** electrochemical impedance spectra under visible-light irradiation, and **e** time-resolved photoluminescence spectra of CdS and Cu₂S/CdS



photocatalytic reactions, and create a passageway for gas release in the water-splitting system. The heterostructure is then expected at the intimate interface between CdS QDs and Cu₂S NSs on account of the well-matched band edge positions of Cu₂S and CdS. The hollow Cu₂S/CdS heterostructure produced by the template and one-pot sulfidation strategy then results in the efficient transfer of photoexcited charges at the intimate interfaces. Thus, this typical Cu₂S/CdS heterostructure can not only effectively inhibit the photoexcited charge recombination, which consequently accelerates the charge transfer and separation, but also enhance the photocatalytic performance and stability of hollow Cu₂S/CdS heterostructures.

Conclusions

In summary, this work developed an elaborate design and synthesis of the integration of CdS QDs and Cu₂S nanosheets as 3D hollow octahedral Cu₂S/CdS p–n heterostructures by a versatile template and one-pot sulfidation strategy. Notably, the unique hollow Cu₂S/CdS architectures delivered a largely enhanced visible-light-driven hydrogen generation rate of 4.76 mmol/(g·h), which is nearly 8.5 and 476 times larger than that of the pristine CdS and Cu₂S catalysts, respectively. The enhanced photocatalytic hydrogen generation performance of hollow octahedral Cu₂S/CdS p–n heterostructures is ascribed to the strong light-harvesting

capacity, efficient charge transfer and separation, and abundant active sites through the construction of hollow architectures. 3D hierarchical hollow nanostructures can strengthen multiple reflections of solar light and provide a large specific surface area and abundant reaction sites for photocatalytic water splitting. Owing to the construction of the p–n heterostructure as an ideal catalytic model with a highly matched band alignment at the $\text{Cu}_2\text{S}/\text{CdS}$ interfaces, the emerging internal electric field can not only facilitate the space separation and transfer of photoexcited charges between CdS and Cu_2S , but also enhance the charge dynamics and prolong the charges' lifetimes. This work does not only provide an in-depth insight into the rational design and construction of hollow photocatalysts but also clarify the crucial role of unique heterostructures in photocatalysis for energy conversion applications.

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