

# Mechanoactive Scaffold Induces Tendon Remodeling and Expression of Fibrocartilage Markers

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**Abstract** Biological fixation of soft tissue-based grafts for anterior cruciate ligament (ACL) reconstruction poses a major clinical challenge. The ACL integrates with subchondral bone through a fibrocartilage enthesis, which serves to minimize stress concentrations and enables load transfer between two distinct tissue types. Functional integration thus requires the reestablishment of this fibrocartilage interface on reconstructed ACL grafts. We designed and characterized a novel mechanoactive scaffold based on a composite of poly- $\alpha$ -hydroxyester nanofibers and sintered microspheres; we then used the scaffold to test the hypothesis that scaffold-induced compression of tendon grafts would result in matrix remodeling and the expression of fibrocartilage interface-related markers. Histology coupled with confocal microscopy and biochemical assays were used to evaluate the effects of scaffold-induced compression on tendon matrix collagen distribution,

cellularity, proteoglycan content, and gene expression over a 2-week period. Scaffold contraction resulted in over 15% compression of the patellar tendon graft and upregulated the expression of fibrocartilage-related markers such as Type II collagen, aggrecan, and transforming growth factor- $\beta$ 3 (TGF- $\beta$ 3). Additionally, proteoglycan content was higher in the compressed tendon group after 1 day. The data suggest the potential of a mechanoactive scaffold to promote the formation of an anatomic fibrocartilage enthesis on tendon-based ACL reconstruction grafts.

## Introduction

The anterior cruciate ligament (ACL) is the most frequently injured ligament of the knee [29], with over 300,000 injuries reported [20] and more than 100,000 ACL reconstruction procedures performed annually [2] in the United States. Primary ACL reconstruction has traditionally utilized autologous bone-patellar tendon-bone (BPTB) grafts, with a shift in recent years toward the semitendinosus or hamstring tendon grafts [19, 58, 64] due to patellar tendonitis and anterior knee pain related to BPTB grafts. Allografts are also used for ACL reconstruction [22, 28], with the tibialis and Achilles tendons being the most common [22, 24, 49, 50, 57, 62]. The long-term performance of ACL grafts depends on several factors, including the structural and material properties of the graft, the initial graft tension [6, 7, 15, 16, 21], the intraarticular position of the graft [37, 42], and graft fixation [33, 53]. Increased emphasis has been placed on graft fixation since postsurgical rehabilitation regimens require the immediate ability to regain the full range of motion, reestablish neuromuscular function, and bear weight [10, 54]. The BPTB graft has been the gold standard for ACL reconstruction in part

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Each author certifies that his or her institution has approved the animal protocol for this investigation and that all investigations were conducted in conformity with ethical principles of research.

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due to its ability to integrate with subchondral bone via the bony ends. Moreover, it possesses intact insertion sites or entheses that can serve as functional transitions between soft tissue and bone. In contrast, the autologous hamstring tendon graft and tendon allografts must be fixed mechanically within the femoral bone tunnel via a transfemoral pin, interference screw, or endobutton, while an interference screw with a washer or staple is used to fix the graft within the tibial tunnel. Although the physiological range of motion may be possible via mechanical fixation, graft-to-bone integration is not achieved as the native insertion site is lost during surgery, with nonmineralized soft tissue found within the bone tunnels [9, 33, 54]. Thus graft fixation at the tibial and femoral tunnels, instead of the isolated strength of the graft, represents the weakest point during the early postoperative healing period [33, 53, 55]. Despite improvement in fixation with interference screws, the clinical outcomes of ACL reconstructions with hamstring tendon grafts have continued to be afflicted with greater laxity and failure rates compared to BPTB reconstructions [1, 5, 8, 11, 34, 43, 56, 66]. Therefore, one significant limitation of tendon-based grafts is their inability to fully integrate with bone through an anatomic entheses. Without an anatomical interface, the graft-bone junction has poor mechanical stability [33, 53, 55], which often results in graft failure [17, 25, 33, 53]. Consequently, biological fixation that promotes the regeneration of the native interface on soft tissue-based ACL grafts will be critical for expediting graft healing and achieving long-term functionality.

The ACL inserts into subchondral bone through a fibrocartilage interface, which serves to minimize the formation of stress concentrations and facilitates load transfer between two distinct tissue types [3, 44, 46, 61, 67, 68]. While the mechanism governing interface formation is not well understood, metaplasia of tendon or ligament has been reported to play a role [18]. Using a rodent model, Nawata et al. [47] reported that during development, ACL insertion fibrochondrocytes are derived from ligament fibroblasts. Mechanical stimulation is reportedly important for fibroblast differentiation into fibrochondrocytes and subsequent fibrocartilage formation [4]. Vogel et al. [32, 41, 48, 52, 63] conducted extensive studies on the effects of compressive loading on fibrocartilage formation in flexor tendons. It was reported that compression induced metaplasia of tendinous matrix and initiated transformation into fibrocartilage. For example, while gene expression for aggrecan was absent in the wrap-around region of fetal and neonatal bovine deep flexor tendons, this proteoglycan was strongly expressed in mature animals, suggesting postnatal remodeling of fibrocartilage with physiological loading [48]. In addition, anterior translocation of the rabbit flexor digitorum profundus tendon in order to remove

compressive loading led to a decrease in the size of the fibrocartilage region, breakdown of the collagen fiber network, and lower matrix glycosaminoglycan content [41]. Moreover, *in vitro* dynamic compressive loading resulted in increased expression of aggrecan, biglycan, and versican after 72 hours [52]. Therefore, these reports collectively suggest compressive loading is necessary for inducing matrix remodeling and fibrocartilage formation on tendon grafts.

To address the challenge of achieving biological fixation of soft tissue-based ACL reconstruction grafts, our goal was to develop functional methods to regenerate an anatomic fibrocartilage transition on these grafts. To this end, the potential of two types of mechanoactive scaffold systems that can directly apply compressive mechanical loading to soft tissue grafts were evaluated. Specifically, the first scaffold system consisted of an aligned poly(lactico-glycolic acid) (PLGA) nanofiber mesh [36], while the second system combined the aligned mesh with a degradable graft collar [38, 60] fabricated from PLGA.

Our first objective was to characterize the contractile properties of the nanofiber mesh as well as the biphasic mesh + collar scaffold complex; our second objective was to evaluate the effect of scaffold-induced compression on fibrocartilage development on a tendon graft, focusing on matrix remodeling and the development of fibrocartilage-related markers. We hypothesized that with the inherent contraction of the nanofiber meshes [70], compressive mechanical loading would be applied to tendon grafts by the mechanoactive scaffold, resulting in matrix remodeling and leading to fibrocartilage formation.

## Materials and Methods

To determine the effects of scaffold design on the resultant contraction exerted by each scaffold system, we measured percentage change in length ( $n = 5$ ) and width ( $n = 5$ ) of the nanofiber mesh as well as scaffold diameter ( $n = 3$ ) as a function of scaffold design (mesh versus mesh + collar). To evaluate the effect of scaffold-induced compression on fibrocartilage development on a tendon graft we determined cell number ( $n = 5$ ) and graft glycosaminoglycan content ( $n = 5$ ); tendon matrix collagen fiber distribution ( $n = 2$ ) was evaluated by histology and polarized light microscopy. Gene expression for fibrocartilage-related markers such as Type II collagen ( $n = 2$ ), aggrecan ( $n = 2$ ), and transforming growth factor- $\beta 3$  ( $n = 2$ ) were determined using reverse-transcription polymerase chain reaction (RT-PCR). All above parameters were evaluated as a function of study variables such as scaffold-induced compression (with versus without compression) and culturing time (Day 1 versus Day 14).

We obtained bovine patellar tendons from tibiofemoral joints (1–7 days old, Green Village Packing, Green Village, NJ). Briefly, the joints were first cleaned in an antimicrobial bath. Under antiseptic conditions, midline longitudinal incisions were made through the subcutaneous fascia to expose the patellar tendon. The paratenon was removed, and the patellar tendon dissected from the underlying fat pad. Sharp incisions were made through the patellar tendon at the patellar and tibial ends in order to remove the insertions from the graft.

For scaffold fabrication, the aligned nanofiber meshes (Fig. 1A) were formed by electrospinning [12]. A viscous polymer solution consisting of 35% poly(D,L-lactico-glycolic acid) or PLGA 85:15 (I.V. = 0.70 dL/g, Lakeshore Biomaterials, Birmingham, AL), 55% *N,N*-dimethylformamide (Sigma, St. Louis, MO), and 10% ethanol (Commercial Alcohol, Inc., Toronto, ON, Canada) was loaded into a syringe fitted with an 18-gauge needle (Becton Dickinson, Franklin Lakes, NJ). Aligned fibers [69] were obtained using an aluminum drum rotating at a velocity of 20 m/s. A constant flow rate of 1 mL/hr was maintained using a syringe pump (Harvard Apparatus, Holliston, MA), and an electrical potential was applied between the needle and the grounded substrate (distance = 10 cm) using a high-voltage DC power supply (Spellman, Hauppauge, NY). For scaffold characterization, fiber morphology, diameter, and alignment of the as-fabricated mesh samples were analyzed using scanning electron microscopy (SEM, JSM 5600LV, JEOL, Tokyo, Japan). Briefly, the samples were sputter-coated with gold (LVC-76, Plasma Sciences, Lorton, VA) and subsequently imaged at a voltage of 5 kV.

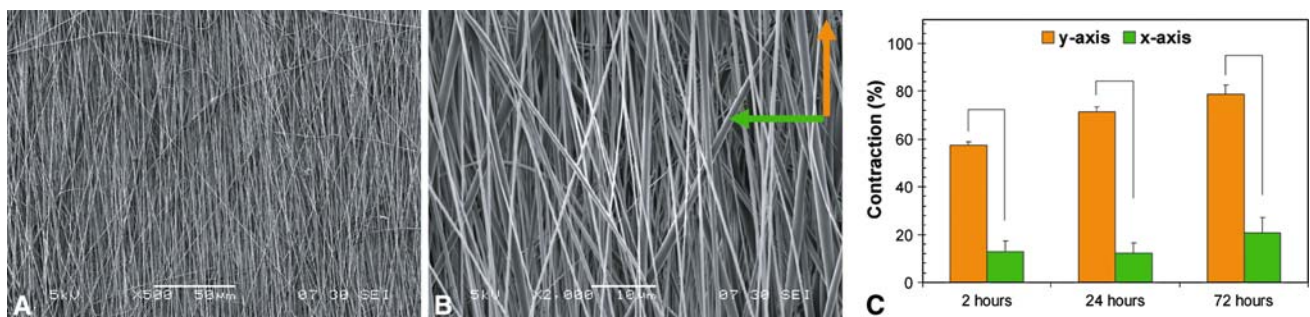
To fabricate the tendon graft collar, a scaffold based on sintered microspheres was formed following published methods [38, 60]. Specifically, the scaffold was composed of composite microspheres consisting of PLGA (85:15, I.V. = 3.42 dl/g, Purac, Lincolnshire, IL) and 45S5

bioactive glass (BG, 20  $\mu$ m; MO-SCI Corporation, Rolla, MD). The microspheres were formed by adding BG particles (20 wt%) to a solution of PLGA and dichloromethane (Acros Organics, Morris Plains, NJ). After vortexing, the suspension was poured into a 1% solution of polyvinyl alcohol (Sigma, St. Louis, MO) to form microspheres. The microspheres were collected and sintered at 70°C for 5 hours in a custom mold, in order to form cylindrical scaffolds with an outer diameter of 0.7 cm and an inner diameter of 0.3 cm.

To characterize nanofiber mesh contraction, we used digital image analysis. Briefly, the nanofiber meshes were cut into 10 mm  $\times$  10 mm squares and immersed in Dulbecco's Modification of Eagle's Medium (DMEM, Mediatech, Inc., Herndon, VA) supplemented with 10% fetal bovine serum (FBS, Atlanta Biologicals, Norcross, GA) and incubated at 37°C and 5% CO<sub>2</sub>. The meshes were imaged using stereomicroscopy at 0, 2, 24, and 72 hours. Mesh dimensions ( $n = 5$ ) were measured by image analysis (ImageJ 1.34s, NIH, Bethesda, MD), and contraction was calculated based on percent change in length both in the x-axis and along the direction of fiber alignment (y-axis).

In addition to mesh contraction, the nanofiber mesh-mediated compression of the microsphere-based graft collar was also evaluated in vitro. Briefly, strips of nanofiber mesh (15.5 cm  $\times$  1.5 cm) were wrapped around the graft collar scaffold, with the fibers aligned perpendicular to the scaffold long axis. The mesh + collar scaffold complex was then incubated in phosphate-buffered saline (PBS, Sigma) at 37°C and 5% CO<sub>2</sub>, and any change in scaffold diameter ( $n = 3$ ) due to mesh contraction was monitored over 24 hours using image analysis (ImageJ).

The potential of utilizing nanofiber mesh contraction to directly apply compression to the tendon graft was evaluated over time. Briefly, aligned electrospun meshes were cut into 10 cm  $\times$  2 cm strips, with fiber alignment oriented



**Fig. 1A–C** Characterization of nanofiber mesh contraction revealed highly oriented fibers and anisotropic contractile behavior. The as-fabricated nanofiber mesh exhibited a preferential fiber alignment as shown by scanning electron microscopy (A)  $\times$ 500 and (B)  $\times$ 2000. (C) Percent contraction of the aligned nanofiber mesh was greatest

along the direction of fiber alignment (y-axis) and contraction stabilized after 24 hours, with differences found at 2 hours ( $p = 2.102 \times 10^{-12}$ ), 24 hours ( $p = 6.719 \times 10^{-15}$ ), and 72 hours ( $p = 1.035 \times 10^{-13}$ ).

along the long axis of the mesh. The patellar tendon graft was bisected along its long axis, and one-half of the tendon was wrapped with the nanofiber mesh while the other half served as the unloaded control. The samples were cultured in DMEM supplemented with 1% nonessential amino acids, 1% antibiotics, and 0.1% antifungal (all from Mediatech), and 10% FBS (Atlanta Biologicals). At days 5 and 14, the effects of compression on tissue morphology and cellularity were characterized by histology [59]. The samples were rinsed with PBS, fixed with 10% neutral buffered formalin (Fisher Scientific, Pittsburgh, PA, and Sigma) and embedded in paraffin (Fisher Scientific). The samples were then cut into 7- $\mu$ m thick sections and stained with hematoxylin and eosin (H&E).

The potential of the mesh + collar scaffold complex to apply compression to the tendon graft was also evaluated in vitro. Specifically, the patellar tendon graft was dissected into 2 cm  $\times$  0.3 cm segments and the cylindrical scaffold was halved along its long axis. Each tendon segment was inserted between the two scaffold halves. For the experimental group, the tendon + graft collar was wrapped with the aligned nanofiber mesh (15.5 cm  $\times$  1.5 cm), while the control scaffolds were wrapped with precontracted electrospun mesh. In addition, to extend compression of the tendon graft, the experimental group was wrapped with new mesh strips on every other day during the 2-week study period. The complex of mesh + collar and tendon graft was cultured in fully supplemented media at 37°C and 5% CO<sub>2</sub>.

We determined the effects of scaffold-induced compression on tendon matrix organization (collagen distribution) at 1 and 14 days using hematoxylin and eosin stains (H&E). Collagen distribution ( $n = 2$ ) was visualized using Picosirius red, and organization of the collagen fibers was examined under polarized light microscopy [30, 51, 65]. In addition, since most of the mesh compression occurs within the first 24 hours, total cell number ( $n = 5$ ) and proteoglycan content ( $n = 5$ ) in the tendon graft were evaluated at Day 1. For the biochemical assays [26, 27, 60], both the wet and dry weights of the tendon samples were measured at Day 0 and Day 1, and the tissue was subsequently digested for 16 hours in 2% papain (Sigma) buffer at 60°C. We determined total DNA content of the digest with the PicoGreen dsDNA assay (Molecular Probes), following the manufacturer's suggested protocol. Sample fluorescence was measured by a microplate reader (Tecan, Research Triangle Park, NC), with excitation and emission wavelengths set at 485 and 535 nm, respectively. The total number of cells in the sample was calculated using the conversion factor of 8 pg DNA/cell [40].

We quantified total sulfated glycosaminoglycan (GAG) content ( $n = 5$ ) in the compressed and control tendon samples using a colorimetric 1,9-dimethylmethylene blue

(DMMB) dye-binding assay [14]. Tissue digest from the cell quantitation assay was combined with DMMB dye, and concentration of the GAG-DMMB complexes was determined using a plate reader at 540 and 595 nm and correlated to a standard prepared with chondroitin-6-sulfate.

Gene expression for fibrocartilage markers ( $n = 2$ ) such as collagen I, II, aggrecan, and transforming growth factor- $\beta$ 3 (TGF- $\beta$ 3) were determined at Day 0 and Day 1 by RT-PCR. Briefly, after removing the graft collar and nanofiber mesh, we isolated total RNA of the tendon graft using the Trizol extraction method (Invitrogen, Carlsbad, CA). The isolated RNA was reverse-transcribed into cDNA using the SuperScript III First-Strand Synthesis System (Invitrogen, Carlsbad, CA) and the cDNA product was amplified with recombinant Platinum Taq DNA polymerase (Invitrogen). Expression band intensities were measured (ImageJ) and normalized against that of the housekeeping gene GAPDH.

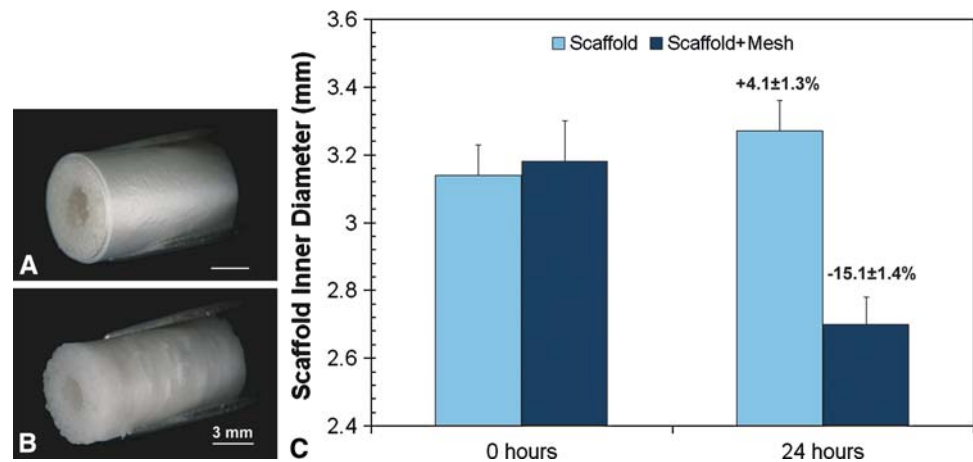
Data are presented as means  $\pm$  standard deviation, with  $n$  equal to the number of samples analyzed. Two-way analysis of variance (ANOVA) was first performed to determine the effects of compression and culturing time on cell number, GAG content, and gene expression (collagen Type II and TGF $\beta$ 3) for the two mechanoactive scaffold systems tested. Fisher's LSD post-hoc test was subsequently performed for all pair-wise comparisons, where the significance of the effects of scaffold-mediated compression on cell number and GAG content were determined between the loaded group and unloaded control, as well as a function of culturing time. For gene expression, a one-way ANOVA followed by Fisher's LSD post-hoc test were performed to determine the effect of compression on expression levels of Type II collagen, aggrecan, and TGF- $\beta$ 3. All analyses were performed using the JMP software package (SAS Institute, Cary, NC).

## Results

For the nanofiber mesh-only design, we observed a high degree of alignment with an average fiber diameter of  $0.9 \pm 0.4 \mu$ m (Fig. 1B). Moreover, anisotropic mesh contractile behavior was observed, with higher contraction along the direction of nanofiber alignment at 2 hours ( $p = 2.102 \times 10^{-12}$ ), 24 hours ( $p = 6.719 \times 10^{-15}$ ), and 72 hours ( $p = 1.035 \times 10^{-13}$ ). Specifically, the mesh contracted over 57% along the aligned fiber direction (y-axis) by 2 hours, with less than 13% reduction in the x-axis (Fig. 1C). Mesh contraction continued over time, exhibiting over 70% contraction in the y-axis and 20% in the x-axis by 24 hours and stabilizing thereafter, with no differences between the 24- and 72-hour groups. For the scaffold design with mesh wrapped around a



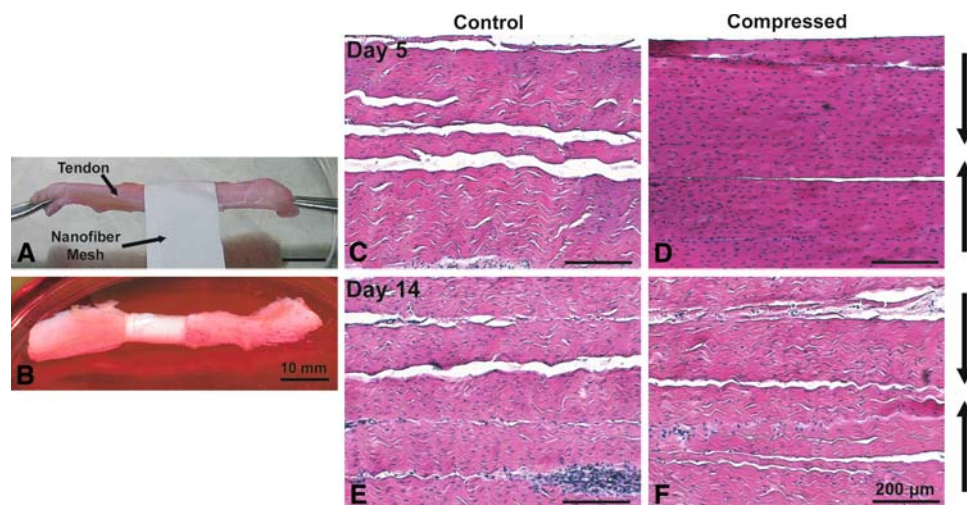
**Fig. 2A–C** Nanofiber mesh contraction resulted in compression of the graft collar scaffold. (A) The microsphere-based graft collar scaffold is wrapped with nanofiber mesh, and (B) after 24 hours of mesh contraction. (C) Scaffold inner diameter was reduced due to compression induced by the nanofiber mesh. While the scaffold-only control swelled (4%,  $p = 0.146$ ), nanofiber mesh contraction induced over 15% decrease in scaffold inner diameter after 24 hours ( $p = 3.63 \times 10^{-4}$ ).



microsphere-based graft collar (Fig. 2A), mesh contraction decreased ( $p = 3.63 \times 10^{-4}$ ) scaffold inner diameter, averaging 15% strain within 24 hours (Fig. 2B). In contrast, the control scaffold without mesh expanded and increased in inner diameter (4%,  $p = 0.146$ ) when cultured under similar conditions (Fig. 2C).

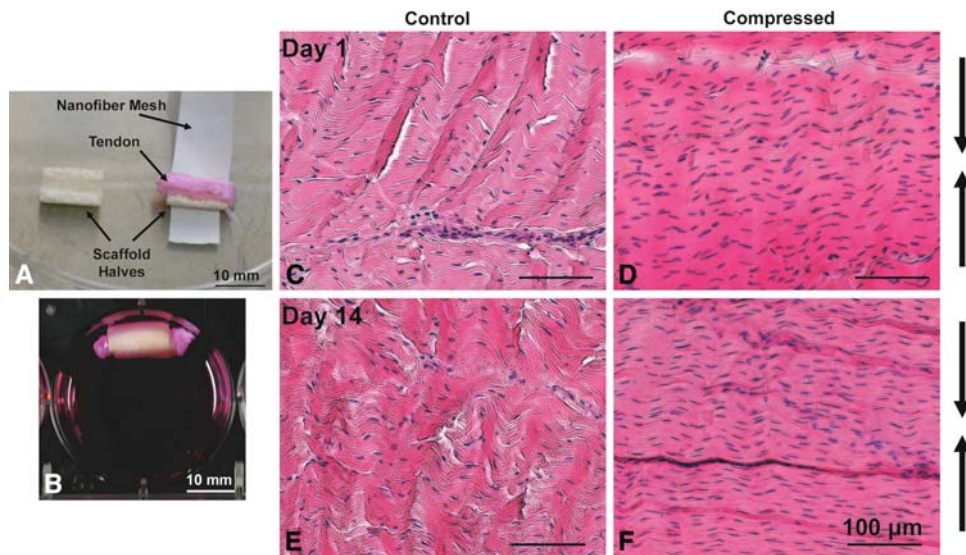
Contraction of the aligned nanofiber mesh resulted in an approximately 30% decrease in tendon graft diameter by 24 hours (Fig. 3A–B). After 5 days of explant culture, the compressed tendon exhibited less crimp structure compared to that in the control group and remodeled into a dense matrix with high cellularity (Fig. 3C–D). However, by Day 14, the crimp pattern was restored in the compressed group, with ultrastructure and cellularity

indistinguishable from the unloaded control group (Fig. 3E–F). When the nanofiber mesh + collar scaffold was used to compress the tendon graft (Fig. 4A–B), contraction of the biphasic scaffold resulted in distinct tendon-graft matrix organization from that of the unloaded control, with an apparent increase in matrix density. The characteristic crimp of the tendon observed in the control (Fig. 4C) was less evident in the experimental group (Fig. 4D). After 14 days, the control tendon group (Fig. 4E) retained its characteristic crimp, with evident disruption of the matrix ultrastructure. In contrast, in the tendon graft compressed by the mesh + collar scaffold complex, the matrix remodeling observed at 24 hours following the onset of loading was maintained over time (Fig. 4F).



**Fig. 3A–F** Tendon grafts were compressed radially with the nanofiber mesh scaffold. (A) The nanofiber mesh is wrapped around a patellar tendon sample, and (B) the samples after 24 hours of mesh contraction. Mesh contraction resulted in tendon matrix organization at day 5 (C) control and (D) loaded, with the arrows denoting the direction of compressive loading applied by the mesh. Moreover, the

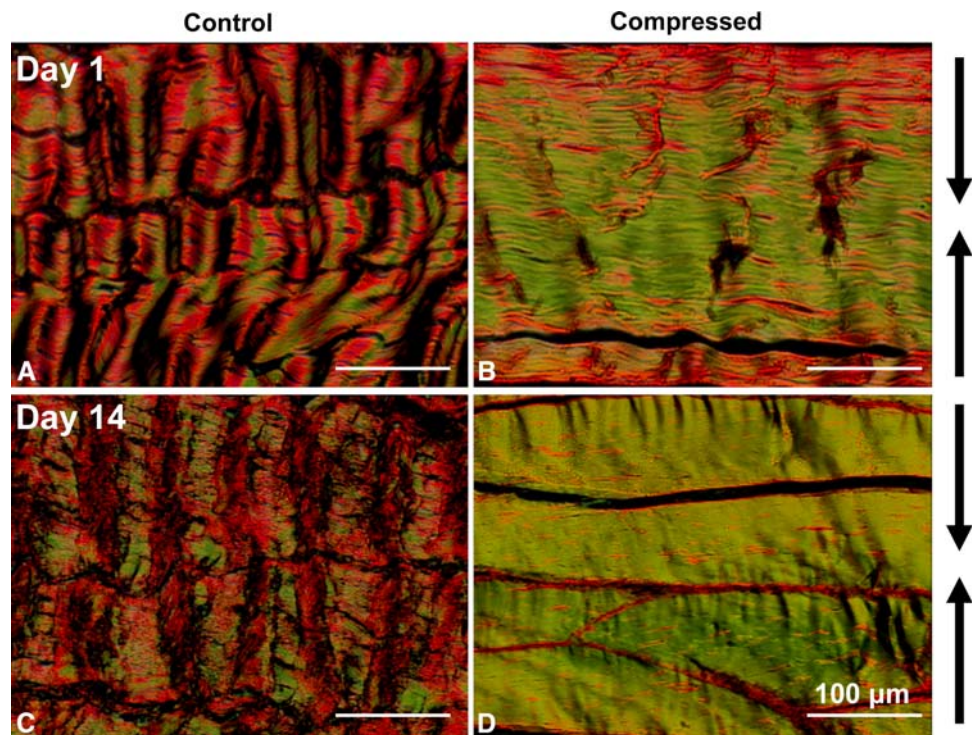
compressed tendon matrix exhibited greater cell density and was morphologically distinct from the unloaded control. After 14 days however, no difference was observed between groups (E) control and (F) loaded. (Stain, hematoxylin and eosin; original magnification,  $\times 10$ ).



**Fig. 4A–F** Tendon grafts were compressed radially with nanofiber mesh + graft collar scaffold. (A) The tendon grafts were wrapped with the graft collar scaffold and mesh, and (B) the tendon graft with mesh + scaffold complex is shown after 24 hours. Compression was found to affect matrix organization of the tendon (C) control, Day 1, (D) loaded, Day 1, (E) control, Day 14 and (F) loaded, Day 14, with arrows denoting the direction of compressive loading applied by the

scaffold. Within 24 hours of loading, the tendon matrix no longer exhibited the crimp pattern evident in the unloaded control. In addition, local cell density increased and matrix remodeling was evident, and this organization was maintained after 2 weeks of static compression (Stain, hematoxylin and eosin; original magnification,  $\times 20$ ).

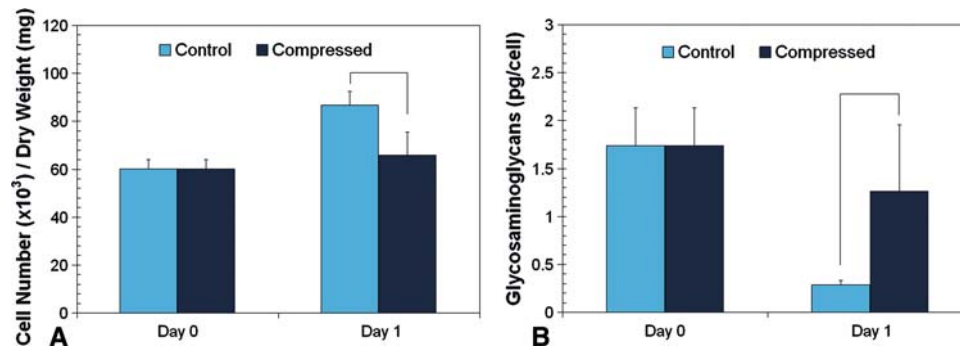
**Fig. 5A–D** Scaffold-induced compression modulated collagen organization. Collagen organization was affected by scaffold-mediated loading at (A) control, Day 1, (B) loaded, Day 1, (C) control, Day 14, and (D) loaded, Day 14. In addition, fiber diameter was smaller in the compressed group. Disruption of the collagen matrix was evident only in the control group after 14 days (Stain, picosirius red as viewed under polarized light; original magnification,  $\times 20$ ).



Compression distinctly changed matrix collagen organization. The color of collagen fibers stained with Picosirius red and viewed under polarized light is reported to correlate with fiber diameter [23, 31, 51], progressing from green, yellow, orange to red with increasing fiber diameter. While

we observed no change in fiber diameter over time in the unloaded control group (Fig. 5A), the collagen fiber diameter of the group compressed with the mesh + collar scaffold became smaller (in green) after 24 hours of loading (Fig. 5B). Moreover, disruption of the tendon collagen



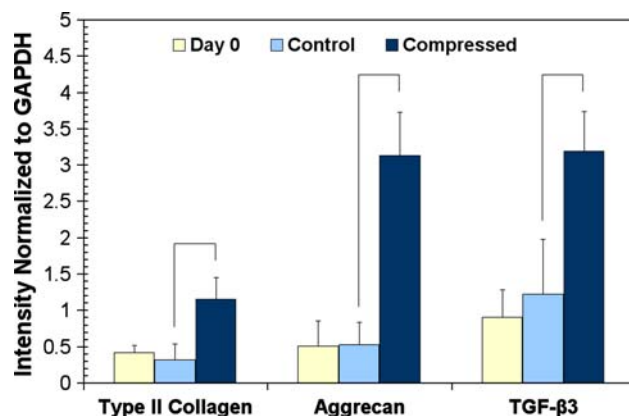


**Fig. 6A–B** Scaffold-induced compression modulated tendon cellularity and matrix composition. **(A)** Cells proliferated in the unloaded group and cell number was higher in the control compared to the

compressed group after 24 hours of loading ( $p = 0.0033$ ). **(B)** Glycosaminoglycan content in the mesh was higher in the compressed group after 24 hours of loading ( $p = 0.0342$ ).

matrix was evident in the control group by Day 14 (Fig. 5C). In general, collagen fibers remained perpendicular to the direction of loading after 24 hours, and this effect was maintained over 14 days with the mesh + collar scaffold complex (Fig. 5D). The total cell number in the tendons remained relatively constant in the compressed group, while we observed a higher ( $p = 0.0033$ ) number of cells in the control tendons by Day 1 (Fig. 6A). Matrix glycosaminoglycan (GAG) content was greater ( $p = 0.0342$ ) in the compressed tendon group after one day of loading (Fig. 6B), when compared to the unloaded control.

The mechanoactive mesh + collar scaffold also promoted the expression of interface markers. After 24 hours of compression, gene expression of fibrocartilage-related markers were upregulated in the loaded group when compared to uncompressed tendons (Fig. 7), with differences found in the expression of Type II collagen ( $p = 0.0603$ ), aggrecan ( $p = 3.11 \times 10^{-5}$ ), and TGF- $\beta 3$  ( $p = 3.66 \times 10^{-4}$ ).



**Fig. 7** Scaffold-induced compression of the tendon graft resulted in the up-regulation of fibrocartilage markers, including type II collagen ( $p = 0.0603$ ), aggrecan ( $p = 3.11 \times 10^{-5}$ ), and TGF- $\beta 3$  ( $p = 3.66 \times 10^{-4}$ ) after 24 hours. Note the increase in all three fibrocartilage interface-related markers in the tendon after scaffold-induced compression.

## Discussion

Our long-term goal is to achieve biological fixation by engineering a functional and anatomical fibrocartilage enthesis on biological and synthetic soft tissue grafts used in orthopaedic repair [39]. To this end, this study focused on the design and evaluation of two novel mechanoactive scaffold systems capable of applying mechanical loading, guided by the working hypothesis that, with the inherent contraction of PLGA nanofiber meshes, compression may be applied to tendon grafts, resulting in matrix remodeling and leading to fibrocartilage formation. The first objective of this study was to characterize the contractile properties of an aligned nanofiber mesh versus the biphasic mesh + collar scaffold complex; the second focused on the effect of scaffold-induced compression on fibrocartilage development on a tendon graft. Specifically, we used the biphasic mesh + collar scaffold to test the hypothesis that scaffold-induced compression of tendon grafts would result in matrix remodeling and the expression of fibrocartilage markers.

We evaluated only tendon grafts with viable cells in this study. Allografts, which do not contain viable cells necessary for remodeling the tendon matrix, would need to be repopulated with fibroblasts or stem cells delivered either from the scaffold or seeded in vitro before graft implantation. Mesenchymal stem cell-seeded Type I collagen sponges inserted into excised sheep patellar tendons and loaded using an ex vivo wrap-around system reportedly upregulated chondrogenic markers such as Sox9 and Fos [35]. A similar response by a devitalized tendon graft that has been repopulated with cells is anticipated following scaffold-mediated compressive loading. Moreover, the mesh-scaffold system is based on degradable poly- $\alpha$ -hydroxyester polymers, thus it is expected the mechanoactive scaffold will be replaced by newly formed tissue after a functional fibrocartilage interface has been formed on the graft. Future studies will focus on extending the

current study to longer time points, as well as evaluating the potential of coupling the mechanoactive scaffold with ACL reconstruction grafts using physiologically relevant *in vivo* models. With the exception of proteoglycans, we examined the effects of loading on Type II collagen and TGF- $\beta$ 3 with gene expression and plan to extend these analyses to protein production. Additionally, other fibrocartilage markers and regulatory factors may guide interface regeneration. Identifying these regulatory factors is critical as it is anticipated that this novel scaffold system may be optimized to control the spatial and temporal distribution of relevant growth factors, thereby exercising biochemical stimulation to direct cellular differentiation as well as promoting the transformation of the tendon matrix into fibrocartilage through mechanical loading.

We observed that the mechanoactive scaffold was able to apply compressive loading to tendon grafts. Moreover, scaffold-mediated compression promoted matrix remodeling, maintained graft glycosaminoglycan content and, interestingly, induced gene expression for fibrocartilage interface-related markers, including Type II collagen, aggrecan, and TGF- $\beta$ 3. These results demonstrate that compressive loading can be incorporated into scaffold design and used to promote fibrocartilage formation on tendon grafts.

We described two scaffold-based loading systems. The first design involved using a nanofiber mesh to directly load the tendon graft while the second design consisted of a complex of the nanofiber and microsphere-based graft collar. The alignment of the nanofiber mesh resulted in anisotropic mesh contractile behavior, effectively translating contractile force into compression, which was utilized in this study to apply compressive loading to the tendon grafts. In the first mechanoactive design, histological analysis of the grafts revealed that nanofiber-mediated compression induced extensive remodeling of the tendon ultrastructure, with the compressed graft exhibiting a denser matrix with increased local cell density. This matrix modulation effect, however, diminished over time, with the control and loaded groups nearly indistinguishable by Day 14. As mesh contraction stabilized after 24 hours, it is likely the tendon graft no longer experienced mechanical stimulation in long-term cultures. Therefore, the short-term effect of mesh-induced compressive loading on graft matrix organization and the high magnitude of compression (approximately 30%) initiated the development of the second mechanoactive scaffold system. Specifically, the nanofiber mesh was combined with a degradable microsphere-based graft collar system in order to achieve a lower compressive strain (15%). Under scaffold-induced compression, the remodeled tendon matrix with cells embedded in a dense matrix was maintained over time, with marked differences in collagen matrix organization seen between

the control and loaded groups. Proteoglycan content of the tendon matrix was also maintained in the loaded group compared to the control, further indicating that scaffold-induced compression influenced matrix maintenance and remodeling. These observations demonstrated the potential of this mesh + collar scaffold system to provide continuous mechanical stimulation and promote sustained tissue remodeling.

Scaffold-mediated compression also resulted in the upregulation of fibrocartilage markers including Type II collagen, aggrecan, and transforming growth factor- $\beta$ 3 (TGF- $\beta$ 3). The fibrocartilage entheses of tendons is largely comprised of types I and II collagen, as well as proteoglycans [4, 13, 32, 45]. Moreover, compressive loading of fibrocartilaginous regions of tendons enhances aggrecan gene expression [13, 32] and increases the synthesis of TGF- $\beta$ 1 [52] and large proteoglycans. Compression of the nonfibrocartilaginous regions of the deep flexor tendon also promotes proteoglycan synthesis [13]. Our data are in agreement with these published studies on the effects of compressive loading, and demonstrate the feasibility of implementing a degradable scaffold system for fibrocartilage interface formation on tendon grafts.

Contraction of PLGA meshes has been previously reported in the literature [70], although the phenomenon has been discredited as a shortcoming rather than promoted as an advantageous attribute of the system. Currently, the mechanism underlying mesh contraction is not known. Zong et al. [70] reported electrospun nanofiber mesh comprised of crystalline polyesters contracted less than amorphous polyester copolymers such as PLGA 75:25. It was proposed that when nanofiber meshes comprised of crystalline polymers are incubated at 37°C, the polymer glass transition temperature is approached and crystallization rapidly occurs, resulting in a lamellar structure that constrains the relaxation of the polymer chains and in turn prevents contraction [70]. The polyester copolymer utilized in this study has a high D,L-lactide content (85%) and is noncrystalline, thus the above mechanism may explain the high degree of contraction observed. Although not the focus of the current study, fiber alignment-related scaffold anisotropy may be used to modulate mesh contraction, and consequently, the magnitude and direction of compressive loading on the graft may be controlled by customizing the degree of fiber alignment. Future studies will focus on elucidating the mechanism of mesh contraction as well as exploring methods to control this process for mechanical stimulation.

We focused on the incorporation of mechanical loading into scaffold design and the current findings demonstrate the potential of using a mechanoactive scaffold system to induce fibrocartilage formation on soft tissue grafts. This is advantageous for scaffold design as the mechanoactive



scaffold can be used to apply both physiological loading as well as desired ectopic loading in vitro and in vivo. For biological fixation, the mesh-collar system is intended to be applied clinically as a degradable graft collar, and will be used to initiate and direct regeneration of an anatomical fibrocartilage interface at the insertion of tendon-based ACL reconstruction grafts. In addition to providing a 3-D environment for matrix development and growth factors for guided cell differentiation, the innovative scaffold system described here can also apply physiologic mechanical stimulation crucial for directing cellular function and tissue remodeling. For utilization with viable autografts, we envision the graft would be inserted through the collars immediately before implantation, and compression of the graft and subsequent fibrocartilage formation would occur in vivo.

We described the design and evaluation of a mechanoactive scaffold able to apply mechanical stimulation to tendon grafts. Scaffold-induced compression of the patellar tendon graft had a profound effect on matrix remodeling and led to the upregulation of fibrocartilage-related markers. These promising results demonstrate the clinical potential of the mechanoactive scaffold, as it is envisioned that the scaffold can be used to apply both biochemical and mechanical stimuli for the induction of metaplasia of the tendinous matrix, ultimately facilitating the formation of an anatomic fibrocartilage interface on these grafts. This approach offers promise as the functional transition between soft tissue and bone would be reestablished, with the potential to ensure long-term graft stability and improve clinical outcome through biological fixation.

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