

# A facile process for fabrication of environmentally safe superhydrophobic surfaces

Tanu Mittal, Sangeeta Tiwari, S. K. Tiwari

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Abstract Superhydrophobic coatings have opened new vistas in the field of self-cleaning surfaces by having improved performance, robustness and preservation of cleaning agents including water. Presently, fluoropolymers are extensively explored and used for this purpose. However, the major drawback accompanying fluoropolymers is formation of environmentally persistent and toxic compounds, viz. PFOA (perfluorooctanoic acid) and PFOS (perfluorooctanesulfonic acid) on their degradation raising concern on their use. In the present study, an effort has been made to develop a facile process by using an environmentally safe material. A simple dip coating technique is reported to fabricate superhydrophobic surface.  $SiO<sub>2</sub>$ nanoparticles of size 25–30 nm, prepared by sol–gel route, were dip-coated on glass substrate in multilayers followed by coating of a surfactant to achieve homogeneity of  $SiO<sub>2</sub>$  layers and enhanced binding with subsequent layer of NUVA-2114 [a commercially used<br>fluorotelomer with chemical composition fluorotelomer with chemical composition  $C_2F_6(CF_2CF_2)_nX$ . It was observed that SiO<sub>2</sub> nanoparticles and sodium dodecyl sulfate (SDS) surfactant exhibit a synergistic effect on interface stability at optimized SDS concentration. A water contact angle of  $152^{\circ}$  is achieved. The established method is simple, scalable, environmentally safe and cost-effective.

Keywords Nanosilica, Superhydrophobic coatings, Self-cleaning, Dip coating, Fluoropolymers, NUVA-2114, PFOA, PFOS

T. Mittal, S. Tiwari  $(\boxtimes)$ Amity Institute of Applied Sciences (AIAS), Amity University, Noida, Uttar Pradesh, India e-mail: drtiwarisangeeta@gmail.com; stiwari2@amity.edu

S. K. Tiwari

Council of Scientific and Industrial Research (CSIR), New Delhi, India

### Introduction

Superhydrophobic surfaces with a water contact angle (WCA) higher than  $150^\circ$  have lately fascinated researchers due to their prospective application in various fields such as antireflection,<sup>[1](#page-5-0)</sup> self-cleaning,<sup>[2](#page-5-0)</sup> antisticking, $3$  lenses and windows, $4$  antifogging, $5$  anticorrosion,  $6-8$  $6-8$  $6-8$  oil–water separation,  $9-11$  $9-11$  $9-11$  and selective transportation of microdroplets.<sup>[12–](#page-5-0)[14](#page-6-0)</sup> Several methods have been reported to attain superhydrophobic sur-faces including plasma method,<sup>[15](#page-6-0)</sup> template method,<sup>[16](#page-6-0)</sup> chemical vapor deposition,<sup>[17](#page-6-0)</sup> sol–gel processing,  $18-20$  $18-20$  $18-20$ wet chemical reaction, $21-23$  lithography, electrospinning, $24,25$  and solution immersion. Among these, layerby-layer (LbL) deposition is a simple, accessible, broadly applicable and reliable method to obtain superhydrophobic surfaces. On the basis of the previous studies, the conventional approach of fabricating superhydrophobic surfaces involves two steps: creation of a hierarchical rough surface to prepare essential small surface topographies  $(\leq 100 \text{ nm})$  and further treatment of the surface in order to reduce the surface energy. $26-28$  The antisticking and anticontamination properties obtained by self-cleaning surfaces are of great significance for applications including antifouling surfaces, automobile parts, stain resistance textiles, dust-free coatings for glasses in buildings and PV solar cells. Superhydrophobicity can be achieved by either roughening the surface or lowering of the surface energy or both. Surface energy and surface roughness are the dominant factors influencing the wettability. By lowering the surface energy, the hydrophobicity of surfaces is enhanced.  $CF_3$  having the lowest surface energy  $(6.7 \text{ mJ/m}^2)$  gives a water contact angle of around 120°.<sup>[29](#page-6-0)</sup> However, to achieve higher hydrophobicity, appropriate surface roughness is desirable. Surfaces with water contact angles of more than 150° were developed by introducing appropriate roughness on materials having low surface energies. Nanostruc-

tured surfaces are preferably prepared by self-assembly, chemical or electrochemical deposition, or sol–gel process followed by further surface modification using low-surface-energy materials. The most important material used in such type of coatings is fluoropolymers. Since the fluoropolymers are generally synthesized from monomers with carbon chain lengths of eight or greater, these materials produce perfluorooctanoic acid (PFOA) and perfluorooctanesulfonic  $(PPOS)$  acid on degradation.<sup>[30](#page-6-0)</sup> PFOA and PFOS are proven biopersistent $31$  and therefore a major cause of concern. In recent years, efforts are being made to develop alternatives to longer perfluoro chain length alcohols which degrade to perfluorooctanoic acid (PFOA) and perfluorooctanesulphonic (PFOS) acid and higher homologue perfluoro acids. Fluorotelomers (C-6 or lower products) such as NUVA-2114 are shortchain molecules and do not produce PFOA on decomposition. However, they exhibit similar characteristics of larger-chain fluoropolymers. Therefore, these are environmentally acceptable alternatives which could be promising candidates for superhydrophobic coating applications.

# Experimental

### **Materials**

The chemicals used in the study are tetraethylorthosilicate (TEOS; Merck 99%), aqueous ammonia (NH4OH), (Merck 25%), hydrochloric acid, (Merck 37%), and ethanol (Merck 99%). NUVA-2114, used in the study, was obtained from M/s. Clariant, New Delhi. Sodium dodecyl sulfate (SDS), HPLC grade, was employed as surfactant.

# Preparation of  $SiO<sub>2</sub>$  NPs by sol–gel route

 $SiO<sub>2</sub>$  NPs were prepared by the hydrolysis of tetraethylorthosilicate (TEOS). Aqueous ammonia was added to a solution containing 74 ml ethanol and 10 ml of deionized water which was stirred for an hour, and pH was maintained at 8–9 by using NH<sub>4</sub>OH. 5 ml of TEOS was added to this solution with stirring. The temperature was maintained between  $65$  and  $70^{\circ}$ C, and stirring was continued for 3 h for complete hydrolysis and condensation. Formation of silica NPs takes place by condensation of the partially hydrolyzed species.

### Deposition of  $SiO<sub>2</sub>$  NPs on glass substrate

For fabrication of hydrophobic coatings on glass substrate, strips of glass of size 10 mm  $\times$  20 mm  $\times$  2 mm were used. The glass substrate was cleaned by acetone for 30 min followed by DI water in an ultrasonic cleaner. The glass substrates were etched

by immersing in 1:1 aqueous HCl for 1 h, washed with deionized water and dried. Glass substrate was then immersed in the above prepared nanosilica sol for 5 min and withdrawn at the rate of 3 mm  $s^{-1}$ . The samples were partially dried in ambient conditions. Second layer of nanosilica was subsequently coated by a similar process followed by drying.

### Deposition of SDS and NUVA 2114 layers over  $SiO<sub>2</sub> NPs$

A layer of SDS (sodium dodecyl sulfate) was coated over  $SiO<sub>2</sub>$  NPs layer to improve the adhesion of final layer of NUVA coating. 2% (w/v) solution of SDS in ethanol was used, and the layer was deposited by dip coating in a similar fashion as described above. A layer of NUVA was finally deposited by preparing a solution of NUVA in water (1:1 v/v). The samples were left in the solution for 20 min. The coated samples were then washed with ethanol and dried in ambient conditions followed by heating at  $100^{\circ}$ C for 30 min with heating rate of  $5^{\circ}$ C.

### Characterization

Silica NPs obtained by centrifugation were characterized by an X-ray diffractometer (Rigaku Mini Flex II) using Cu K $\alpha$  radiation;  $\lambda = 0.154$  nm, at a scan rate of  $8^{\circ}$  min<sup>-1</sup>. The particle size distribution was obtained using dynamic light scattering (DLS) instrument, Model Zetasizer Nano-S90. The morphology of silica NPs was studied using scanning electron microscope (FE-SEM; Hitachi S-4500). The particle size and crystallinity of silica NPs were studied using transmission electron microscopy (TEM; Model Jeol-2010), and the orientation of  $SiO<sub>2</sub>$  NPs was studied using highresolution transmission electron microscopy (HR-TEM). Fourier transform infrared spectroscope, (FT-IR; Perkin Elmer PE-1600) was used to study the interaction of subsequent layers of coatings with each other. Water contact angles of the coatings were measured using optical contact angle goniometer (model SL150) with a drop shape analyzer at ambient temperature. Surface roughness of coatings was measured by atomic force microscopy (AFM) (VECCO DICP2) with SPM software.

# Results and discussion

### Particle size and morphology of  $SiO<sub>2</sub>$  NPs

The hierarchical structure of a surface contributes significantly to its hydrophobic character. The size of nanoparticles plays an important role in formation of nanohierarchical structure of  $SiO<sub>2</sub>$  particles. Figure [1c](#page-2-0) indicates the uniform and spherical nature of  $SiO<sub>2</sub>$  NPs.

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Fig. 1: (a) TEM micrographs of SiO<sub>2</sub> NPs and the inset showing DLS pattern, (b) HR-TEM micrographs of SiO<sub>2</sub> NPs, (c) SEM images of  $SiO<sub>2</sub>$  NPs, (d) SAED pattern of  $SiO<sub>2</sub>$  NPs

The particles appear as spherical nanostructures with well-defined boundaries. The size of the particles as observed in the SEM image is 25–30 nm. The TEM images of  $SiO<sub>2</sub>$  NPs and inset showing particle size distribution, obtained using dynamic light scattering (DLS) technique, can be seen in Fig. 1a. The particle size of 25–30 nm was observed by TEM which is predictably less than the particle size range obtained by DLS (50–53 nm). This can be attributed to the fact that DLS gives hydrodynamic radius which is always higher than the actual particle size.<sup>[32](#page-6-0)</sup> Figure 1b shows the high-resolution transmission electron microscopy (HR-TEM) micrographs of  $SiO<sub>2</sub>$  NPs that indicate lack of orientation in  $SiO<sub>2</sub>$  NPs (highlighted in micrographs). Selected area electron diffraction (SAED) pattern for  $SiO<sub>2</sub>$  NPs can be seen in Fig. 1d which confirms the amorphous nature of  $SiO<sub>2</sub>$  NPs. It is inferred that the particle size obtained by SEM is in accordance with the results obtained by DLS and TEM.

#### $SiO<sub>2</sub>$  NPs deposition on glass substrate

SEM analysis was carried out to study the deposition pattern of  $SiO<sub>2</sub>$  NPs layer on glass after single- and double-layer coating. Figures [2](#page-3-0)a and [2](#page-3-0)b show SEM images of single-layer coating of  $SiO<sub>2</sub>$  NPs on glass substrate at two different magnifications. Double-layer coating of  $SiO<sub>2</sub>$  NPs at different magnifications is shown in Figs. [2](#page-3-0)c and [2](#page-3-0)d.

In case of single-layer coating, the particles are uniformly distributed. In double-layered coating, a more uniform layer completely covering the substrate is visible. The deposition of  $SiO<sub>2</sub>$  NPs is uniform and the particles are closely packed, thereby providing the essential nanoroughness to the coating.

### Role of SDS in stabilization of interfaces between various layers

Figure [3](#page-3-0) gives a schematic diagram for the interface interaction between various layers which accounts for its stabilization. When the layer of SDS is applied on  $SiO<sub>2</sub>$  NPs, the SDS molecules adhere on nanoparticle surface, rendering it amphiphilic and thereby enhancing the stability of the coatings and its adhesion to the subsequent layers. The adsorption of nanoparticles is usually irreversible, and its adsorption energy enhances the stability of interfaces.<sup>[33](#page-6-0)</sup> At specific SDS concentration, a more uniform layer of  $SiO<sub>2</sub>$  is adsorbed on the surface. The SDS molecules attach to the layer of  $SiO<sub>2</sub>$  NPs through their hydrophilic end. It renders the opposite surface hydrophobic thus amenable for adhesion of the subsequent layer of NUVA 2114. To obtain best results, the concentration of SDS was taken slightly above its critical micelle concentration (CMC). At this concentration, the adsorbed surfactant molecules help in pulling particles together to the air/ water interface.<sup>[34](#page-6-0)</sup>

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Fig. 2: (a) SEM micrograph of single layer of silica nanoparticles, (b) single layer of silica nanoparticles at higher magnification, (c) SEM micrograph of double layer of silica nanoparticles, (d) double layer of nanoparticles at higher magnification; the insets show the water contact angle of single and double layer, respectively



Fig. 3: Schematic diagram for interface interactions between various layers of the coating

To further understand the interface interactions, IR spectra of  $SiO<sub>2</sub>$  NPs (Fig. [4](#page-4-0)a), SDS surfactant (Fig. [4b](#page-4-0)) and NUVA 2114 (Fig. [4](#page-4-0)c) and nanoparticles was recorded after deposition of each layer. Figure [4d](#page-4-0) shows IR spectra for SDS-coated  $SiO<sub>2</sub>$  NPs, and Fig. [4e](#page-4-0) shows IR spectra for NUVA coated over SiO2 NPs/SDS layer. A comparison of IR spectra for

all the above samples is depicted in Fig. [4.](#page-4-0)<br>Characteristic peaks of SDS  $(1030 \text{ cm}^{-1})$ . Characteristic peaks of  $SDS$   $(1030 \text{ cm}^{-1},$ 819 cm<sup>-1</sup>)<sup>[35](#page-6-0)</sup> and SiO<sub>2</sub> (1086 cm<sup>-1</sup>, 800 cm<sup>-1</sup>)<sup>[36](#page-6-0)</sup> are

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Fig. 4: (a) FTIR spectra of  $SiO<sub>2</sub>$  NPs, (b) SDS surfactant, (c) NUVA 2114, (d) SDS over two subsequent layers of SiO<sub>2</sub> NPs, (e) final coat of NUVA

observed in IR of SDS-coated SNPs. This indicates the ionic nature of the interaction thus ruling out any chemical reaction between the two layers. In NUVAcoated SDS/SNP layer, low-intensity peaks of SDS  $(819 \text{ cm}^{-1})$  can be seen while characteristic peaks of  $\text{NUVA}$  (1706 cm<sup>-1</sup>, 1450 cm<sup>-1</sup>) appear prominent in IR spectra. The IR studies support ionic interaction between the layers. The peaks of  $SiO<sub>2</sub>$  are not clearly visible in NUVA-coated sample which appears to be suppressed by the two subsequent layers of SDS and NUVA. From these results, it is evident that ionic interaction between various layers is responsible for the stability of coatings.

#### Contact angle and AFM analysis

Surface roughness is acknowledged to be utmost important for imparting superhydrophobicity. Surface roughness and contact angle of the coatings with single and double layers of silica were determined. All measurements were made in static contact angle mode using Laplace–Young calculation method. CA measurements were conducted at five different places for layers of  $SiO<sub>2</sub>$  NPs to establish the uniformity of the coating and consistency of the CA values. The standard deviation of WCA was found to be  $\pm 2$  and  $\pm 1$ ,



Fig. 5: (a) 3D AFM topographical images silica single layer, (b) silica double layer

respectively, for recorded values of double layer and single layer of silica coatings. Figure 5 shows singlelayer coating of SNPs with a root mean square (RMS) roughness value of 2.41 nm and a contact angle of  $130^{\circ} \pm 1^{\circ}$ . With double layer of SNPs, the RMS value is increased to 3.07 nm with a corresponding increase in the contact angle to  $152^{\circ} \pm 2$ , thus converting the surface superhydrophobic. The nanoroughness plays an important role in imparting superhydrophobicity. Jianmen Chen et al.<sup>[37](#page-6-0)</sup> reported a CA of  $163^\circ$  with a micrometer roughness. In the present work, the nanoroughness obtained is appropriate for achieving superhydrophobicity.<sup>[38,39](#page-6-0)</sup> The higher hydrophobicity reported by Chen is due to the use of C-8 fluorocompound which in general has better hydrophobicity. The superhydrophobicity obtained in this work is because of the combination of both C-6 fluorocompound and nanoroughness achieved through nanosilica multilayers. Thus, surface chemistry and roughness of coatings together are seen to have a significant impact on the apparent contact angles which leads to superhydrophobicity. $40,41$  $40,41$  $40,41$ 

#### Optical characterization

Transmission spectra were recorded on UV–Vis spectrophotometer to evaluate the transparency of the coatings. The coatings of single layer (A) and double layer (B) show 90% transmission in the wavelength range of 300–800 nm. The images of water droplets on the prepared coatings are shown in Fig. [6b](#page-5-0). As can be seen, the letters beneath the superhydrophobic glass coatings are visible clearly. Therefore, the prepared coatings retain a significant transparency.

<span id="page-5-0"></span>

Fig. 6: (a) Transmission spectra of coatings: single-layer SNPs/SDS/NUVA (Coating A), double-layer SNPs/SDS/NUVA (Coating B), (b) images of coated and uncoated surfaces

#### **Conclusions**

In the present study, a facile way of obtaining superhydrophobic surfaces is developed. The method involves fabrication of multilayer coating on glass substrate by adopting layer-by-layer (LBL) deposition approach using dip coating technique. Nanoroughness is obtained by deposition of nano- $SiO<sub>2</sub>$  in bilayers. NUVA-2114, an environmentally safe alternative to fluorosilanes, has been used to lower the surface energy. A layer of SDS is used between the layers of  $SiO<sub>2</sub>$  and NUVA to facilitate better adhesion at the interface, thereby stabilizing the multilayered coating. Use of a short-chain telomere as an ecofriendly alternative to longer perfluoro chain length alcohols in obtaining superhydrophobic coating has been successfully demonstrated. By combining bilayers of  $SiO<sub>2</sub>$ NPs and NUVA-2114, we are able to produce superhydrophobic coatings, which exhibit similar characteristics as those of larger-chain fluoropolymers. The imparted stability is explained by ionic interactions and van der Waals forces of attraction between the various layers of the coating. A reasonable optical transparency of 90% is achieved. Environmental acceptability, ready availability of all components and ease of fabrication are the highlights of the developed process for obtaining these superhydrophobic coatings.

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