

Synthesis of fluorinated polycarbonate-based polyurethane acrylate for UV-curable coatings

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Abstract Fluorinated polycarbonate-based UV-curable polyurethane acrylate (F-PCUA) was synthesized by incorporating 1H, 1H, 2H, 2H-Perfluoro-1-octanol to the end of polycarbonate-based PUA chains. The structure of F-PCUA was determined by 1 H-NMR, ¹⁹F-NMR, and FTIR analyses. The physical, surface, and thermal properties of F-PCUA were also examined. The F-PCUA was used as a hydrophobic additive in PUA coatings, and the water and oil wettability of the UV-cured film was investigated by contact angle measurements. The results showed that the coating system had great hydrophobicity. Furthermore, X-ray photoelectron spectroscopy research confirmed that a hydrophobic fluorine-enriched surface was obtained in the coating system. Moreover, the mechanical and chemical properties of the hydrophobic coatings did not show deterioration with the introduction of elemental F.

Keywords Polycarbonate, UV curing, Fluorinate, Hydrophobic

Introduction

Polyurethanes (PU) have been widely used in many applications such as coatings, adhesives, and thermoplastic elastomers due to their excellent mechanical and physical properties and film-forming ability.^{[1](#page-7-0)} Their films show an excellent elasticity and toughness. The chemical reaction underlying PU synthesis is polyaddition between the hydroxyl group (–OH) of polyol and the isocyanate group $(-NCO)^{2-5}$ $(-NCO)^{2-5}$ $(-NCO)^{2-5}$ $(-NCO)^{2-5}$ $(-NCO)^{2-5}$ PU is generally composed of a polyol (soft segment), a diisocyanate (hard segment), and a low-molecular-weight diol component (chain extender, hard segment). Polycarbonate diol (PCDL) polyols have been recently commercialized and are claimed to provide good resistance to hydrolysis, heat aging, oil resistance, weathering, and fungi.^{[6](#page-7-0)} Synthesis of PCDL-based polyurethane acrylate for UV-curable coatings was meaningful.

Recently, F-modified hydrophobic polymers were extensively studied⁷⁻¹⁰ for incorporation into UVcurable coatings such as antifrosting coatings, 11 selfcleaning coatings, 12 and antifingerprint coatings.^{[13](#page-7-0)} Interactions between molecular chains are reduced when elemental F is introduced into a polymer owing to the low surface energy of F. A film surface enriched with fluorinated species imparts excellent hydrophobic and oleophobic properties on the coating. Miao et al. synthesized a series of fluorinated hyperbranched polyester acrylates and used them as additives for UV-curable coatings, which exhibited increased contact angles ranging from 80° to 120° .^{[14](#page-7-0)}

Because of its better performance relative to that of other diols, polycarbonate diol (PCDL) is used in materials such as antistatic materials.¹⁵ biomedical materials such as antistatic materials, $\frac{1}{2}$ materials, 16 16 16 shape memory composites, $^{17-20}$ and elec-tromagnetic interference shielding materials.^{[21](#page-7-0)} Compared to other polyurethane acrylates (PUAs), PCDLbased PUA (PCUA) exhibits stronger mechanical properties and higher heat resistance, abrasion resistance, oxidation resistance, hydrolysis resistance, and chemical resistance. $22-24$ $22-24$ $22-24$ Hyeon-Deuk Hwang's group has performed many studies on PCDL-based waterborne PUAs (WPCUAs), for example, UV-curable low-surface energy fluorinated polycarbonate-based polyurethane and waterborne UV-curable polycarbonate-based polyurethane (meth)acrylate incorporated with polydimethylsiloxane.²

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In this work, a series of fluorinated PCDL-based PUAs (F-PCUAs) were synthesized using PCDL as the soft segment with different molecular weights (500, 1000, and 2000 g/mol). The F-PCUAs were then used as hydrophobic additives in PUA, yielding hydrophobic UV-curable coatings. FTIR, NMR, TGA, and tensile strength tests were carried out to characterize the properties of the F-PCUAs. In addition, contact angle measurements, XPS, and coating performance tests were performed to examine the UV-cured coatings.

Experimental

Materials

Polycarbonate diols (PCDLs, with molecular weights of 500, 1000, and 2000 g/mol) were supplied by Asahi Kasei Co., Ltd. (Tokyo, Japan). 1H, 1H, 2H, 2H-Perfluoro-1-octanol (PFEL) was supplied by Harbin Xeogia Fluorine-silicon Co., Ltd. (Harbin, China). Isophorone diisocyanate (IPDI) was purchased from Hui Li Chemical Co., Ltd. (Wuxi, China). Pentaerythritol triacrylate (PET3A) was supplied by Eternal Chemical Co., Ltd. (Taiwan, China). Toluene, acetone, N-butyl acetate, sodium hydroxide (NaOH), sulfuric acid (H_2SO_4) , ethanol, oleic acid, and di-butyl hydroxyl toluene (BHT) were all purchased from Sinopharm Chemical Reagent Co., Ltd. (Shanghai, China). 1- Hydroxycyclohexyl-phenyl ketone (Irgacure 184) was provided by Sartomer Chemical Co., Ltd. (Guangzhou, China).

Synthesis of fluorinated PCDL-based PUA (F-CPUA)

The synthesis route of F-PCUA is shown in Fig. 1. A flame-dried 500-mL three-necked round-bottom flask equipped with a magnetic stirrer and a thermometer was charged with 0.2 mol of IPDI. Then, 0.1 mol of PCDL with an appropriate amount of butyl acetate and DBTDL (300 ppm) was added to IPDI dropwise over a period of 2.5 h at 40° C and maintained for 6 h, yielding polyurethane prepolymer. PFEL was then added dropwise to the urethane prepolymer in a mole ratio of NCO:OH = 1:0.5. The mixture was heated to 70°C and maintained for 6 h. Then PET3A with BHT (0.15 wt) was added dropwise to the mixture. Completion of the reaction was confirmed by FTIR with the disappearance of the NCO absorption peak. F-500- PCUA, F-1000-PCUA, and F-2000-PCUA, corresponding to PCDL molecular weights of 500, 1000, and 2000 g/mol, respectively, were obtained.

Preparation of F-CPUA films

The optimized composition of F-PCUA was mixed with the appropriate amounts of photoinitiator and nbutyl acetate in a beaker. The mixture was poured into glass molds, and bubbles were removed by magnetic stirring. After 2 days, F-PCUA films measuring approximately 1 mm in thickness took shape in an oven heated to 40°C. The F-PCUA films were exposed to UV radiation with an energy density of 1500 mJ/cm^2 under a UV lamp for curing. The tensile strength and thermal properties of the films were examined.

Fig. 1: Synthesis route of the fluorinated UV curing PCUA

Preparation of hydrophobic PUA Coatings

To examine the hydrophobicity of F-PCUA as a hydrophobic additive, nonfluorinated PCUA-2000 was used as the matrix resin. PCUA-2000 was synthesized in a manner similar to that used to synthesize F-2000- PCUA, except that both ends of the polymer were capped by PET3A. F-2000-PCUA with different mass fractions was used as the hydrophobic additive and mixed with PCUA-2000 and the appropriate amounts of photoinitiator and n-butyl acetate in a beaker. The mixture was coated on PC plates with a film roller measuring approximately $120 \mu m$ in thickness and then prebaked at 50° C for 20 min in an oven. Lastly, the coatings were exposed to 1500 mJ/cm^2 of UV radiation for curing. A series of coatings with different fluorine contents (0, 0.15, 0.31, 0.46, 0.62, 0.77, and 1.54%) were denoted as PCUA, F0.15-PCUA, F0.31-PCUA, F0.46- PCUA, F0.62-PCUA, F0.77-PCUA, and F1.54-PCUA, respectively. The F-PCUA contents and F content ratios of the coatings are shown in Table 1.

Characterization

The PCDL and oligomers were characterized by Fourier transform infrared spectroscopy (FTIR) and nuclear magnetic resonance spectroscopy $(^1H\text{-}NMR$ and $^{19}F\text{-}$ NMR). The FTIR spectra of these resins were recorded with an FTLA2000-104 (ABB BOMEN, Canada) using a KBr cell. 1 H-NMR and 19 F-NMR spectra were obtained by a Bruker Spectrospin MSL-300 MHz using deuterated chloroform $(CDCl₃)$ as a solvent and tetramethylsilane (TMS) as an internal standard.

The tensile strength and elongation at break of the films were investigated using an LRX single-column universal testing machine. The samples were cut into dumbbell shape and submitted to tensile stress–strain measurements. The cross-head speed was set to 10 mm min⁻¹.

The thermal degradation behavior of the F-PCUAs was investigated by thermal gravity analysis (TGA/ 1100SF, METTLER). In each case, a 10 mg sample was examined under a N_2 flow rate of 5×10^{-5} m³ min⁻¹ at a heating rate of 10° C min⁻¹ from room temperature to $600\degree$ C.

Table 1: F-PCUA content and F content of the hydrophobic coatings

The surface-free energy of the UV-cured films was evaluated from the static contact angles using a contact angle analyzer (OCA40, DATA physics) at 25° C. The surface-free energy was calculated from the contact angles based on equations (1) and (2) :^{[28](#page-8-0)}

$$
\gamma_{\rm s} = \gamma_{\rm s}^d + \gamma_{\rm s}^p \tag{1}
$$

$$
(1 + \cos \theta) \gamma_1 = 4 \gamma_1^p \gamma_s^p / (\gamma_1^p + \gamma_1^p) + 4 \gamma_1^d \gamma_s^d / (\gamma_1^d + \gamma_1^d). \tag{2}
$$

X-ray photoelectron spectroscopy (XPS) was carried out to analyze the surface composition of the fluorinated UV-cured films. The experiments were carried out on an RBD upgraded PHI-5000C ESCA system (Perkin Elmer) with Mg K α radiation (hv = 1253.6 eV) or Al K α radiation (hv = 1486.6 eV). Briefly, the X-ray anode was operated at 250 W and a high voltage of 14.0 kV with a detection angle of 54. The pass energy was fixed to 23.5, 46.95, or 93.90 eV to ensure sufficient resolution and sensitivity. The base pressure of the analyzer chamber was approximately 5×10^{-8} Pa. To compare the F content at different depths of the coatings, a coating sample was chosen to run at different take-off angles of 30° and 90° .

To evaluate the oil resistance of the samples, coated panels were immersed in oleic acid for 24 h at room temperature. To examine their chemical resistance, samples were immersed in 0.5 mol L^{-1} H₂SO₄, 1 mol L^{-1} NaOH, and 50% ethanol.

Results and discussion

FTIR

The FTIR spectra obtained for PCDL, PCDL-IPDI, PCDL-IPDI-F, and F-PCUA are shown in Fig. [2](#page-3-0). It is known that N-H stretching occurs at 3380 cm^{-1} , C–N stretching and δ N–H bending occur at 1526 cm⁻¹, C– H stretching occurs from 2780 to 3045 cm⁻¹, and C=O stretching occurs from 1643 to 1865 cm^{-1} . Compared to PCDL-IPDI and PCDL-IPDI-F, the peak at 2265 cm-¹ disappeared, whereas a new peak at 1635 cm⁻¹ appeared due to the double bond of the acrylate group, suggesting that the hydroxyl group on the PCDL segment and PET3A had been consumed due to its reaction with –NCO on IPDI. In addition, the absorption peak at 1243 cm^{-1} assigned to C–F stretching vibration can be observed in the F-PCUA curve, indicating that F-PCUA was successfully end-capped with PET3A and perfluoroalkyl groups.

H -NMR and ^{19}F -NMR

A typical ¹H-NMR survey spectrum of F-PCUA is shown in Fig. [3](#page-3-0). The peaks at 6.80 and 7.31 ppm are assigned to the hydrogen atom of secondary amine,

Fig. 2: FTIR spectra of PCDL, PCDL-IPDI, PCDL-IPDI-F, and F-PCUA

Fig. 3: ¹H-NMR spectrum of F-PCUA

which was connected to the carbonate ester group of PCDL and NCO groups of IPDI. The split peaks at 2.3–2.6 ppm are characteristic peaks of the hydrogen atoms of methylene connected to perfluoroalkyl

groups, which reveal the successful synthesis of PCDL-IPDI-F. Two characteristic peaks at 5.70– 6.40 ppm are attributed to the hydrogen atom of double bonds, which indicates that the methacrylate

group was successfully connected to the end of the molecular chain. In addition, peaks between 0.78 and 1.80 ppm are assigned to the methyl and methylene groups in the molecular structure.

The 19F-NMR spectrum obtained for F-PCUA is shown in Fig. 4. There are four peaks in Fig. 4

Fig. 4: 19F-NMR spectrum of F-PCUA

Fig. 5: Strain curves of F-PCUA's UV-cured films

corresponding to the chemical shift of F atoms in four different positions. The ratio of the peak areas is 2:2:3:6, which coincides with the number of F atoms that occur at different positions. FTIR, ¹H NMR, and ¹⁹F-NMR analyses confirm that the F-PCUAs were successfully synthesized as designed.

Tensile strength

Tensile strength tests are carried out to study the basic mechanical properties of polymeric materials, including Young's modulus, tensile strength, elongation at break, and toughness. Figure 5 shows the strain curves of F-1000-PET3A and F-2000-PET3A, and the tensile strength, elongation at break, and Young's modulus of the cured films are listed in Table 2. The effects of the molecular weight of PCDL on the tensile behavior of the synthesized F-PCUA resins were also demonstrated. With increasing molecular weight of PCDL, the maximum tensile strength and initial modulus decreased $(F-1000-PET3A > F-2000-PET3A)$; however, the elongation at break increased (F-1000- PET3A < F-2000-PET3A). The maximum stresses of F-1000-PET3A and F-2000-PET3A were 14.15 and 4.76 MPa, respectively. However, in terms of elongation, the opposite trend to that of the stress appeared. The elongation of the coatings increased sharply with the increasing molecular weight of PCDL. In fact, the F-500-PET3A film was too hard and fragile to stretch. This behavior can be explained by considering the fact that a network with a high crosslink density exhibits higher resistance to an external extension force, leading to higher tensile strength and lower elongation at break. Moreover, as the molecular weight of PCDL increased, the mass fraction of the hard segment decreased and the mass fraction of the soft segment increased, which also caused the film to exhibit better ductility and flexibility.

Thermogravimetric analysis (TGA)

The thermal stability of the UV-cured films was measured by TGA at temperatures ranging from 25 to 600° 600° C as shown in Fig. 6. The first and second stages were due to the degradation of the hard and soft segments of polyurethane linkages.^{[29,30](#page-8-0)} The maximum thermal degradation of the hard and soft segments of polyurethane occurs at $235-245$ and $355-375$ °C, respec-

Table 2: Elongation at break, tensile strength, and Young's modulus of F-PCUAs with different molecular weights of **PCDL**

Sample	Elongation at break (%)	Tensile strength (MPa)	Young's modulus (MPa)
F-500-PET3A	$\overline{}$	$\overline{}$	-
F-1000-PET3A	21.74	14.15	112.87
F-2000-PET3A	101.71	4.76	8.18

Fig. 6: TGA curves of PCDLs with different molecular weights

tively. The third peak is associated with degradation of the fluorine connected to the end of the F-PCUA chain at approximately 430–450 $^{\circ}$ C.²⁵ The molecular weight of the F-PCUA was increased with the molecular weight of PCDL, resulting in a decrease in the density of double bonds. In addition, the content of the hard segment decreased at the same time. Therefore, the thermal stability of the fluorine-modified PUAs was decreased with the increased molecular weight of PCDL.

Contact angle

The contact angles of the UV-cured films with different concentrations of F-PCUA are shown in Fig. 7. It is clear that the water and oil wettability of the cured films depend on the concentration of F-PCUA and that the contact angle increased with the addition of F-PCUA. Without adding F-PCUA, the water contact angle was 78° . By adding F-PCUA, the contact angle increased to above 100° , even though the F content was very low, exhibiting great hydrophobicity. We attribute these results to the presence of hydrophobic tail-like fluorine-substituted chains on the surfaces of the PCUAs.^{[26](#page-7-0)} However, no appreciable increase in hydrophobicity was observed at F contents above 0.6%. Similar trends were observed for the oil test.

Surface-free energy

The calculated surface tensions (y_s) of the UV-cured films are listed in Table 3. With the incorporation of only 0.15 wt% F, the surface-free energy decreased from 44.0 to 26.3 mN m^{-1} . Only a very small amount of F-PCUA affected the surface-free energy significantly. These results demonstrate that the $CF₂$ groups

Fig. 7: Contact angle values of hydrophobic coatings

Table 3: Surface tension of UV-cured films with different concentrations of F and the surface tension parameters of water and 1-bromonaphthalene

Fig. 8: XPS curves of UV-cured films with different F contents. (a) PCUA, (b) F0.15-PCUA, (c) F0.31-PCUA, and (d) F0.77-PCUA

migrated to the outermost surface and provided UVcured films with a hydrophobic fluorine-enriched surface. 31

X-ray photoelectron analysis (XPS)

Figure [8](#page-5-0) shows the XPS spectra of the UV-cured films according to the F-PCUA content. Figure 9 shows the XPS spectra obtained at different take-off angles for the same sample (F1.54-PCUA). Signals of fluorine, oxygen, nitrogen, and carbon can easily be observed from the survey scan in the region corresponding to 0–

Fig. 9: XPS survey spectra of samples at different take-off angles

Table 4: Fluorine contents on the surfaces of UV-cured films

		F0.15-PCUA F0.31-PCUA F0.77-PCUA	
F_{bulk} (wt%) F_{surface} (wt%) S_{F}	0.15% 6.98% 46.53	0.31% 10.27% 33.13	0.77% 15.20% 19.74

800 eV, with the bonding energies of 694, 536, 401, and 290 eV, respectively. Figure 9 shows that the fluorine content at a take-off angle of 30° was higher than that at 90. This result was observed because the detecting depth of the film at 30 $^{\circ}$ was smaller than that at 90 $^{\circ}$, closer to the surface. This finding proves that the coating surface was enriched with fluorine. Table 5 also supports this conclusion.

The surface enrichment factor (S_F) was defined as the ratio of the fluorine content measured experimentally at the top surface of a dried film $(F_{surface})$ to the estimated atomic % fluorine in the film solids, assuming a uniform and random distribution (F_{bulk}) , as indicated in equation $(3)^{32}$:

$$
S_{\rm F} = F_{\rm surface}/F_{\rm bulk}.\tag{3}
$$

Table 4 lists the fluorine contents of the experimental formulation and at the interface of the coating films, which indicates the enrichment factor of the fluorine element. The fluorine surface enrichment ranged from 19- to 46-fold with respect to the bulk composition. All of the samples containing F-PCUAs showed a higher fluorine atomic content at the surface than that estimated in the bulk. Therefore, the fluorine chain can easily tailor the surface owing to the thermodynamic driving force of the PFEL segments associated with their very low surface energy. 3

Coating performance

To evaluate the adhesion of the UV-cured coatings on PC substrates and the samples' pencil hardness, coated PC panels were tested by conventional methods. The samples were then immersed in 0.5 mol L^{-1} sulfuric acid, 1 mol L^{-1} sodium hydroxide, 50% ethanol, and oleic acid for 24 h to evaluate the chemical resistance of the coatings. The results are shown in Table 5. The results indicate that the addition of F-PCUA did not adversely affect the coating performance. The coatings still exhibited good adhesion on PC and good chemical resistance.

Table 5: Performance of hydrophobic coatings with F-PCUA as a hydrophobic additive

^a The coating films did not pale, wrinkle, bubble, or fall off the PC panels

Conclusion

Fluorinated PCDL-based PUAs (F-PCUAs) were successfully synthesized by incorporating perfluoroalkyl to the end of PCUA chains. The contact angle of the UV-cured coatings could be increased to 110° even with a very low content of F-PCUA, indicating good hydrophobicity. XPS demonstrated the enrichment of elemental fluorine on the coating surfaces. The hydrophobic coatings exhibited good chemical resistance and adhesion on PC panels.

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