



# Fabrication of triangular Cu<sub>3</sub>P nanorods on Cu nanosheets as electrocatalyst for boosted electrocatalytic water splitting

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**Abstract:** Non-precious electro catalysts with high-efficiency, cheapness and stability are of great significance to replace noble metal electro catalysts in the hydrogen evolution reaction (HER) and oxygen evolution reaction (OER). In this work, triangular Cu@CuO nanorods on Cu nanosheets were fabricated by a novel in-situ oxidation approach using Cu nanosheets as self-template and conductive nano-substrate in an aqueous solution of NaOH/H<sub>2</sub>O<sub>2</sub>, and then by low-temperature phosphorization treatments. The experimental results show that the phosphating temperature has a significant effect on the morphology, composition and number of active sites of Cu@Cu<sub>3</sub>P nanorods. The Cu@Cu<sub>3</sub>P-280 electrode exhibits a good HER catalytic activity of achieving a current density of 10 mA/cm<sup>2</sup> at 252 mV in acid electrolyte. After catalysis for 14 h, the current density can still reach 72% of the initial value. Moreover, the Cu@Cu<sub>3</sub>P-280 electrode also shows an excellent OER catalytic activity in basic electrolyte, reaching a current density of 10 mA/cm<sup>2</sup> at the overpotential value of 200 mV. After catalysis for 12 h, the current density remained more than 93% of the initial value. This work provides a theoretical basis for the directional design and preparation of sustainable, low-cost, bifunctional electrocatalytic materials.

**Key words:** Cu@Cu<sub>3</sub>P; electrocatalysis; oxygen evolution reaction; hydrogen evolution reaction

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## 1 Introduction

In recent decades, the energy donors are dominated mainly by fossil fuels, which offer some benefits, but bring about such issues as environmental pollution, greenhouse effect and resource depletion. In order to meet the ever-increasing energy demand, a great deal of efforts have been put into exploiting new clean and highly efficient energy conversion technologies via electrocatalytic reaction in recent years.

Electrocatalytical water splitting has been widely regarded as the most promising and sustainable technology for green hydrogen fuel production from aqueous solution [1 – 6]. The efficiency of electrochemical hydrogen production is determined by the activity of the electrocatalyst in water splitting reactions. Although the Pt-group metals and Ru- or Ir-based compounds are considered the most active catalyst for the hydrogen evolution reaction (HER) and oxygen evolution reaction (OER) [7 – 11], respectively, the high cost and scarcity significantly prohibit their large scale and

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practical application. Furthermore, it is urgent to develop a new type of effective HER and OER electrocatalyst with low cost and high abundance.

Transition metal phosphides are of great interest as inexpensive alternatives to precious metal catalysts owing to their appealing conductivity, chemical stability and excellent catalytic performance for HER and OER. The extensively investigated metal phosphides include Ni<sub>2</sub>P [12–14], FeP [15–16], MoP [17–18], Co<sub>2</sub>P [19–21], NiCuP [22], NiCoP [23], Cu<sub>3</sub>P [24], WP<sub>2</sub> [25–28], etc. All of them show comparable electrocatalytic performances to noble metals catalysts. Nevertheless, as an earth abundant transition metal, Cu-based catalysts have relatively fewer studies in terms of HER and OER properties.

Copper phosphide (Cu<sub>3</sub>P) is a candidate for energy conversion and storage technologies because of its favorable electronic structure, combined with low cost and thermal stability [29–30]. However, Cu<sub>3</sub>P is prone to agglomeration of active sites during the catalytic process, and part of the catalyst will be lost, thus affecting the catalytic activity. Recent reports have proved that nanocatalysts loaded or grown on the conductive substrate reveal better catalytic performance than pure catalysts. For instance, ZHOU et al [31] reported Cu<sub>3</sub>P nanowires on Cu foam via a simple two-step method. The Cu<sub>3</sub>P nanowires/copper foam (NWs/CF) exhibits enhanced activity toward OER with a small overpotential of 327 mV at the current density of 20 mA/cm<sup>2</sup>, and good durability in alkaline medium. TIAN et al [32] reported self-supported Cu<sub>3</sub>P nanowires arrays on porous copper foam. Owing to the unique structure and composition, Cu<sub>3</sub>P NW/CF maintains its HER activity for at least 25 h and the catalytic current density can approach 10 mA/cm<sup>2</sup> at an overpotential of 143 mV. CUI et al [33] reported Cu<sub>3</sub>P nanobush (NB) on copper meshes, demonstrating that the Cu<sub>3</sub>P NB/Cu exhibits a superior catalytic HER activity of achieving a current density of 10 mA/cm<sup>2</sup> at 120 mV and exhibits a long-term stability in acid solutions. Although with such success, the substrates are concentrated on bulk materials such as copper foam and nickel foam, while micro-nano conductive materials are rarely reported.

Herein, we report on our recent efforts in developing a novel self-supported triangular Cu<sub>3</sub>P

nanorods on conductive substrate of Cu nanosheets by topotactic conversion from its Cu@CuO nanorod precursor. The triangular Cu@CuO nanorod was directly grown on Cu nanosheet by a novel in-situ oxidation method, then followed by low-temperature phosphidation reaction. Benefiting from the structural and component merits, the as-prepared triangular Cu@Cu<sub>3</sub>P nanorods exhibit enhanced catalytic performance for HER and OER, respectively. The triangular Cu@Cu<sub>3</sub>P nanorods exhibit high HER activity of achieving a current density of 10 mA/cm<sup>2</sup> at 252 mV and displays a long-term stability in acid solution. This catalyst displays a high catalytic activity in basic electrolytes, reaching a current density of 10 mA/cm<sup>2</sup> at the overpotential value of 200 mV for OER. This work provides a useful strategy for developing efficient micro-nano self-supporting catalysts with high conductivity and numerous reactive sites for HER and OER, which encourages more research on other similar electrocatalysts for renewable energy. We expect that this facile synthetic strategy could be expanded to synthesize many other phosphides.

## 2 Experimental

### 2.1 Preparation of electrocatalysts

#### 2.1.1 Cu nanosheets

The well-defined Cu nanosheets were prepared according to the procedures in our previous report [34]. In brief, 1.0 g C<sub>6</sub>H<sub>12</sub>O<sub>6</sub>·H<sub>2</sub>O, 0.5 g polyvinyl pyrrolidone (PVP) and CuSO<sub>4</sub> were dissolved into 25 mL of deionized water with magnetic stirring vigorously, then the mixture was transferred into a Teflon-lined autoclave and heated at 180 °C for 3 h.

#### 2.1.2 Triangular CuO nanorods on Cu nanosheets

The triangular CuO nanorods on Cu nanosheets (Cu@CuO nanorods) were synthesized by a simple hydrothermal process. In the typical synthesis, an amount of Cu nanosheets was dispersed in 4.2 mL of NaOH (0.6 mol/L) in a Teflon-lined vessel with magnetic stirring vigorously. Then, 5 mL H<sub>2</sub>O<sub>2</sub> (30%) was injected rapidly into the mixture solution under stirring. Afterward, the vessel was sealed and then hydrothermally heated at 120 °C for 11 h. The resulting precipitate was separated and washed with deionized water, and eventually dried in vacuum at

60 °C for 12 h.

### 2.1.3 Triangular Cu<sub>3</sub>P nanorods on Cu nanosheets

The triangular Cu<sub>3</sub>P nanorods on Cu nanosheets (Cu@Cu<sub>3</sub>P nanorods) catalysts were synthesized through phosphorization procedure using the as-prepared triangular Cu@CuO nanorods as precursors. Typically, the as-prepared triangular Cu@CuO nanorods (0.1 g) and NaH<sub>2</sub>PO<sub>2</sub> (0.25 g) were placed at two different locations of the tube furnace with NaH<sub>2</sub>PO<sub>2</sub> at the upstream side. After being flushed with Ar, the center of the furnace was elevated to 280 °C at a heating rate of 5 °C/min and maintained at this temperature for 60 min. To investigate the impact of the phosphating temperature on the electrocatalytic activity, the phosphorization process was conducted at 270 °C and 300 °C, whereas the other parameters remained unchanged.

### 2.1.4 Cu/Cu<sub>3</sub>P nanosheets

For comparison, the samples directly phosphatized with Cu nanosheets were synthesized by the following steps. The as-prepared Cu nanosheets (0.1 g) and NaH<sub>2</sub>PO<sub>2</sub> (0.25 g) were placed at two different locations of the tube furnace with NaH<sub>2</sub>PO<sub>2</sub> at the upstream side. After being flushed with Ar, the center of the furnace was elevated to 280 °C at a heating rate of 5 °C/min and maintained at this temperature for 60 min.

## 2.2 Electrochemical measurements

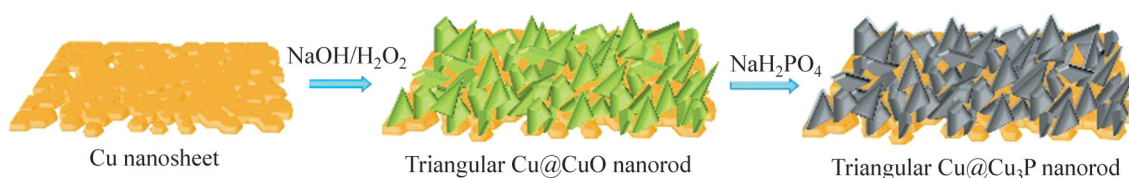
All electrochemical tests were performed at room temperature with CHI 660D electrochemical workstation. The performances of HER and OER were tested with a standard three-electrode system. The as-prepared catalysts, Ag/AgCl electrode, Pt sheet and graphite rod were chosen as working electrode, reference electrode, and counter electrode, respectively. The electrocatalytic activities of catalysts for HER and OER were measured in 0.5 mol/L H<sub>2</sub>SO<sub>4</sub> and 1 mol/L KOH solution, respectively. In all measurements, the Ag/AgCl reference electrode was calibrated with

respect to reversible hydrogen electrode (RHE). All the potentials reported were versus the RHE according to  $E_{\text{vsRHE}} = E_{\text{vsAg/AgCl}} + E_{\text{vsAg/AgCl}}^{\circ} + 0.059 \text{ pH}$ . The LSV was recorded at a scan rate of 5 mV/s and the corresponding Tafel plots were used to evaluate the catalytic kinetics. Impedance measurements were carried out with frequency ranged from 0.1 Hz to 10 kHz under the amplitude of 10 mV. The effective electrochemical surface area (ECSA) was determined from the  $C_{\text{dl}}$  of the catalyst surface. The  $C_{\text{dl}}$  was determined by measuring CVs with multiple scan rates in non-faradaic potential region.

## 3 Results and discussion

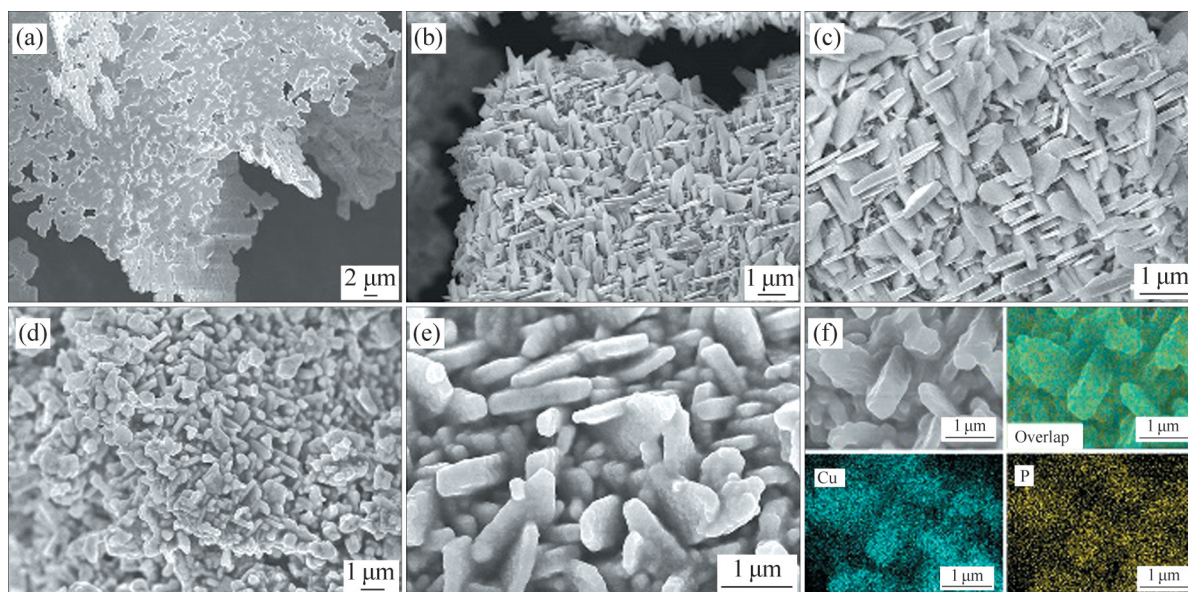
The typical procedure for synthesis of triangular Cu@Cu<sub>3</sub>P nanorod is summarized in Scheme 1. In brief, Cu nanosheets were firstly synthesized according to hydrothermal method. Then, Cu nanosheets as both self-template and nano conductive substrate were immersed in the solution of NaOH/H<sub>2</sub>O<sub>2</sub> for the formation of triangular Cu@CuO nanorod. Followed by a low-temperature phosphorization process, the triangular Cu@CuO nanorod was successfully converted to triangular Cu@Cu<sub>3</sub>P nanorod, using NaH<sub>2</sub>PO<sub>4</sub> as phosphorous source.

The SEM image shows that the Cu nanosheets display a plate-shaped morphology with smooth surface and has large area characteristic of 30 – 100 μm in diameter (Figures 1(a) and S1). The as-prepared Cu nanosheet was employed as self-template and conductive substrate in the following process to prepare triangular CuO nanorods materials. Figure 1(b) shows the SEM image of the triangular CuO nanorods grown on Cu nanosheets, in which the entire surface of the Cu nanosheet is covered uniformly by densely packed CuO nanorods (Figure S2). A close view of such nanorod (Figure 1(c)) reveals that the structure of nanorod is triangular column, and extends vertically from the Cu nanosheets with thickness of 200 – 500 nm and



**Scheme 1** Synthesis procedure of the triangular Cu@Cu<sub>3</sub>P nanorod





**Figure 1** SEM images (a–e) and EDX distribution mapping (f) in triangular  $\text{Cu@Cu}_3\text{P}$  nanorods: (a) Cu nanosheets; (b–c) Triangular  $\text{Cu@CuO}$  nanorods; (d–e) Triangular  $\text{Cu@Cu}_3\text{P}$  nanorods; (f) EDX element distribution mappings of Cu, P and overlap of elements in triangular  $\text{Cu@Cu}_3\text{P}$  nanorods

length of 1–3  $\mu\text{m}$ . After the phosphorization process through a gas-solid reaction, the morphology of triangular nanorods was preserved (Figure 1(d)). Meanwhile, the surfaces of triangular  $\text{Cu}_3\text{P}$  nanorods became rough and thick, thus improving the surface area and active sites, resulting in the enhanced catalytic activity (Figure 1(e)). The SEM and corresponding elemental mapping patterns of Cu and P for the triangular  $\text{Cu}_3\text{P}$  nanorods in Figure 1(f), clearly reveal that Cu and P were uniformly distributed on the nanorods, which further confirms the successful chemical conversion of  $\text{Cu@CuO}$  to  $\text{Cu@Cu}_3\text{P}$ .

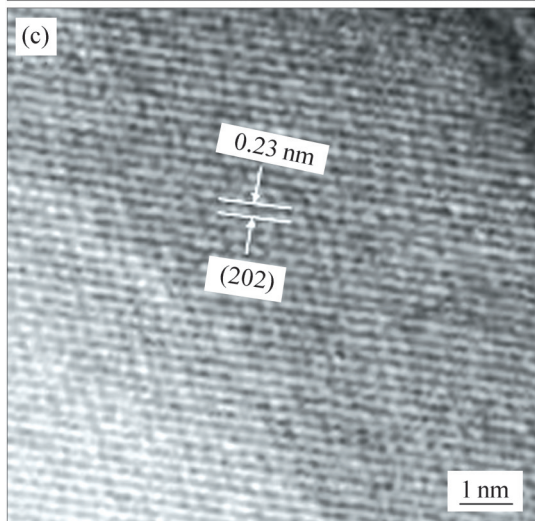
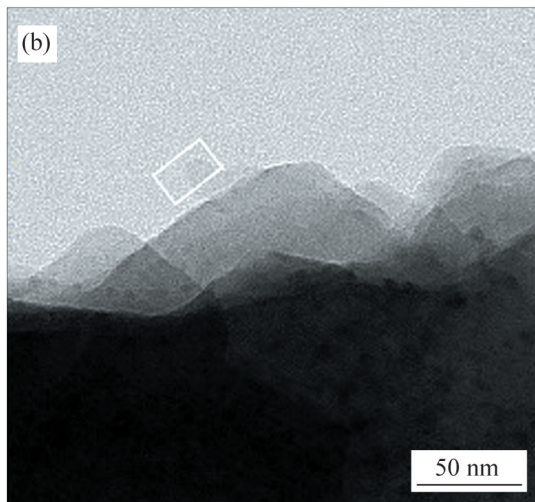
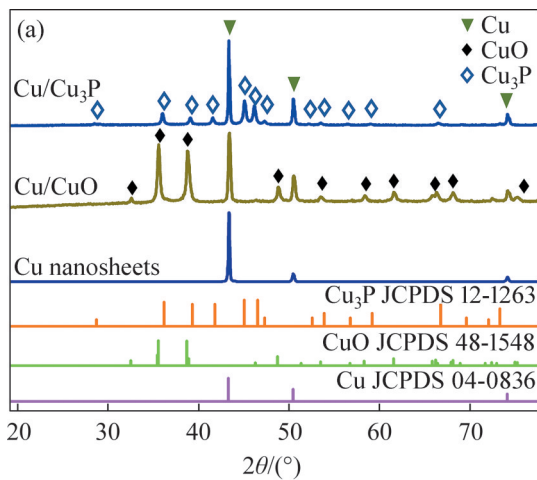
The crystalline phases were measured by XRD. Figure 2(a) shows the XRD patterns of as-synthesized Cu nanosheets,  $\text{Cu@CuO}$  nanorods and  $\text{Cu@Cu}_3\text{P}$  nanorods. It can be seen that all the samples present three strong diffraction peaks at  $43.4^\circ$ ,  $50.6^\circ$  and  $74.4^\circ$ , which can be assigned to (111), (200) and (220) crystal planes of face-centered cubic structure due to the Cu nanosheet substrate (PDF #04-0836). After the oxidation treatments, a series of new diffraction peaks can be observed, which well correspond to the crystal plane of the CuO (PDF #48-1548). After low-temperature phosphorization process, the diffraction peaks located at  $28.6^\circ$ ,  $36.2^\circ$ ,  $39.4^\circ$ ,  $41.8^\circ$ ,  $45.1^\circ$ ,  $46.5^\circ$ ,  $47.3^\circ$ ,  $52.5^\circ$ ,  $53.4^\circ$ ,  $56.4^\circ$ ,  $59^\circ$  and  $66.5^\circ$

corresponding well to the (111), (112), (202), (221), (300), (113), (212), (220), (221), (311), (222) and (223) planes of the  $\text{Cu}_3\text{P}$  (PDF #12-1263). No other XRD peak except  $\text{Cu}_3\text{P}$  crystal form was observed, demonstrating that the triangular  $\text{Cu@Cu}_3\text{P}$  nanorod structure was successfully obtained. Figure 2(b) presents the TEM image of  $\text{Cu@Cu}_3\text{P}$  nanorods, thus showing that the structure of triangular nanorod was retained after phosphating. The HRTEM image (Figure 2(c)) confirms that the  $\text{Cu}_3\text{P}$  nanorod has an interplane distance of the lattice fringe of 0.23 nm, which corresponds to the (202) facet of  $\text{Cu}_3\text{P}$ .

XPS spectrum was used to investigate the surface chemical composition and element valence states of the triangular  $\text{Cu@Cu}_3\text{P}$  nanorods. The full survey XPS spectrum in Figure 3(a) shows that the  $\text{Cu@Cu}_3\text{P}$  is composed of Cu, P, O and C elements.

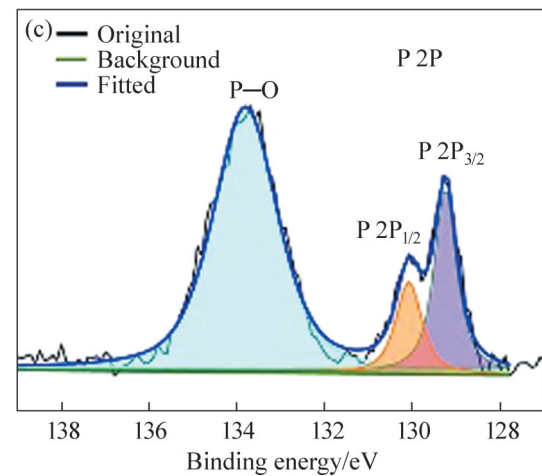
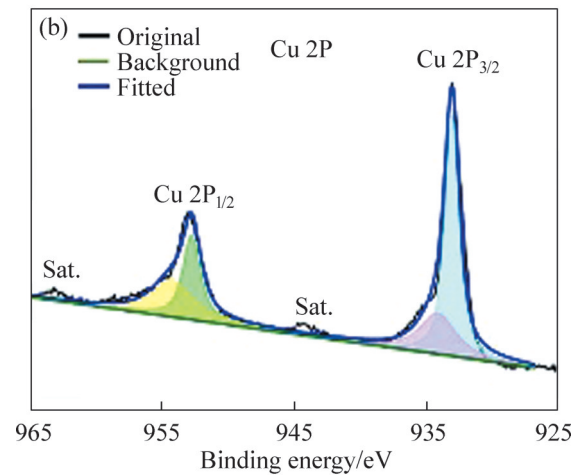
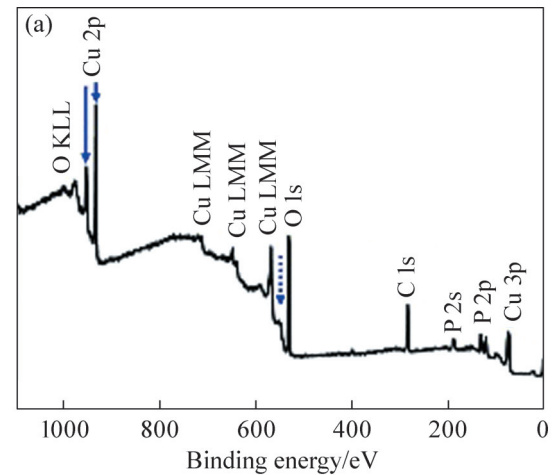
Figure 3(b) shows the high-resolution XPS spectrum of Cu 2p. Two apparent peaks located at 932.6 and 934.3 eV originating from the Cu  $2p_{3/2}$  species, and two additional peaks at 952.6 and 954.1 eV are corresponding to the Cu  $2p_{1/2}$  species. The peaks at 932.6 and 952.6 correspond to the  $\text{Cu}^{\delta+} 2p_{3/2}$  and  $\text{Cu}^{\delta+} 2p_{1/2}$  in  $\text{Cu}_3\text{P}$  and the binding energies of 934.3 and 954.1 eV correspond to the Cu 2p peaks in oxidized copper [35–39]. The satellite peaks at 944.2 and 963.0 eV also show the existence of oxidation state of copper. There is no





**Figure 2** (a) XRD patterns of Cu nanosheet, Cu@CuO nanorod, Cu@Cu<sub>3</sub>P nanorod, (b) TEM and HRTEM (c) images of triangular Cu@Cu<sub>3</sub>P nanorods

peak assigned to metal Cu, implying that Cu<sub>3</sub>P nanorods have completely covered the surface of Cu nanosheets. Figure 3(c) shows that the peaks of the P 2p are located at 129.1 and 130.0 eV,



**Figure 3** XPS survey spectrum of triangular Cu@Cu<sub>3</sub>P nanorods (a), high-resolution XPS spectra of Cu 2p (b) and P 2p (c) for triangular Cu@Cu<sub>3</sub>P nanorods

corresponding to the P 2p<sub>3/2</sub> and P 2p<sub>1/2</sub>, respectively, which are the characteristic peaks of P in Cu<sub>3</sub>P [40–41]. In addition, the peak located at 133.6 eV can be assigned to P—O bands, which is probably due to the surface oxidation of Cu<sub>3</sub>P after being exposure to air [42].

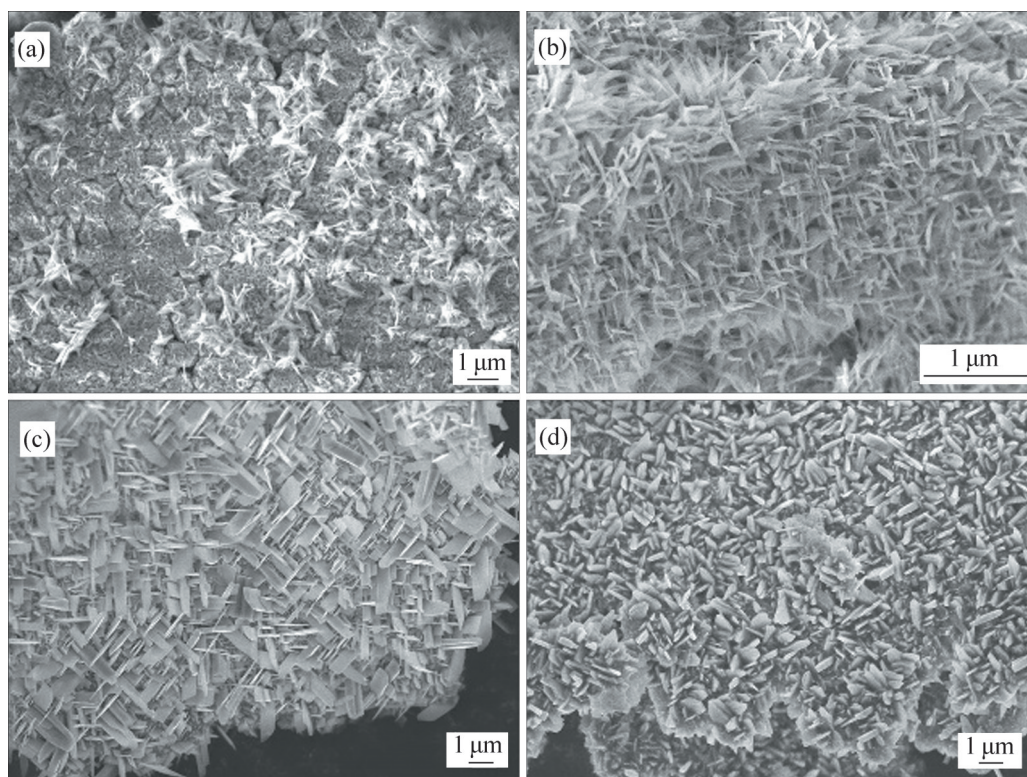
The morphology of the precursor Cu@CuO has a significant impact on the morphology of the final product. Therefore, the influence of the volume of  $\text{H}_2\text{O}_2$  on morphology of as-prepared nanostructure was investigated, as shown in Figure 4.

Without adding  $\text{H}_2\text{O}_2$  solution, the products with small amounts of flakes are grown on the surface of Cu nanosheets (Figure 4(a)), and most of the copper nanosheets are still partially exposed, indicating that the oxygen released by  $\text{H}_2\text{O}_2$  plays an important role in the growth of CuO nanorods. With volume of  $\text{H}_2\text{O}_2$  increased to 1 mL, the density of CuO grown on the surface of Cu nanosheets was also increasing (Figure 4(b)). When the volume of  $\text{H}_2\text{O}_2$  reached 3 mL, part of the thin sheet began to grow into thick columnar structures (Figure 4(c)). Until the volume of  $\text{H}_2\text{O}_2$  solution was 5 mL, triangular Cu@CuO nanorod material was obtained. However, when the volume of  $\text{H}_2\text{O}_2$  further increased to 7 mL, the density of CuO on the surface of Cu nanosheet decreased although it presented the morphology of triangular nanorods (Figure 4(d)). This may be due to the excessive addition of  $\text{H}_2\text{O}_2$  solution, which will neutralize part

of the alkalinity of NaOH, resulting in reduced pH value in the solution and affecting the growth of the product.

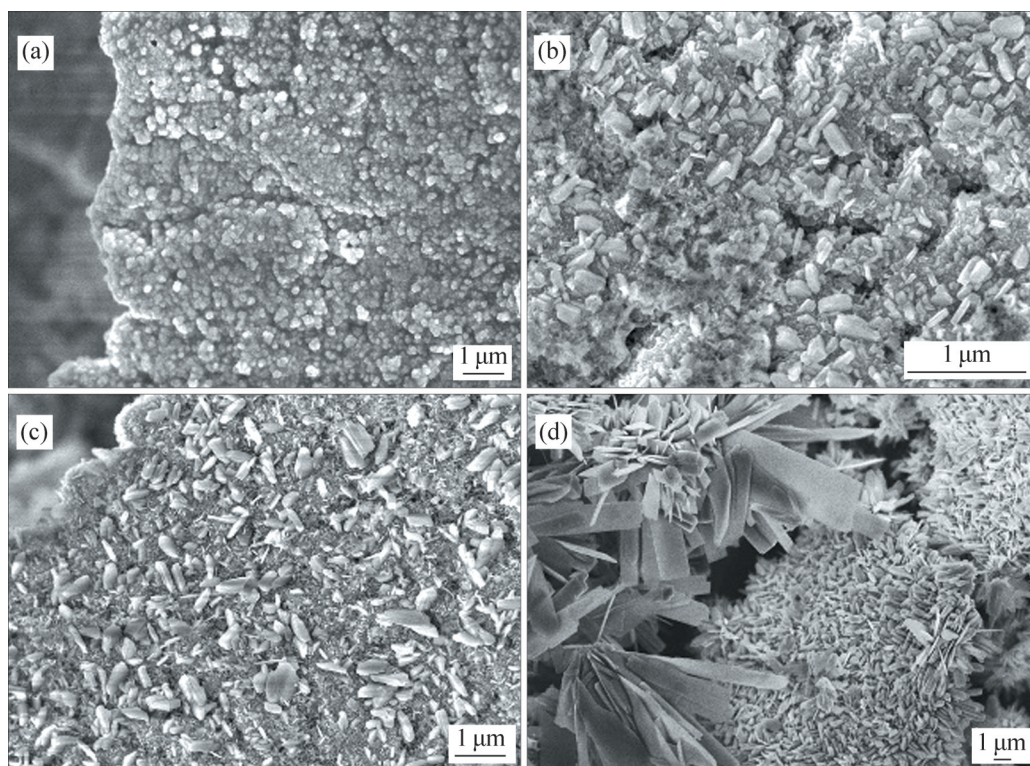
The influence of the concentration of NaOH solution on the morphology of as-prepared nanostructure was likewise studied, as shown in Figure 5. Without adding NaOH solution, a large number of nanoparticles were grown on the surface of Cu nanosheets, which means that the structure of nanorods cannot be obtained simply by adding  $\text{H}_2\text{O}_2$  solution (Figure 5(a)). When the concentration of NaOH was 0.1 mol/L, the triangular column structure began to appear, but the length of the nanorods is shorter (Figure 5(b)). With the increasing concentration of NaOH solution, the length of the nanorods also began to grow, as shown in Figure 5(c). When the concentration of NaOH was 0.8 mol/L, in addition to the formation of Cu@CuO nanorods, some multi-layered CuO nanoplates were formed in the product, which is mainly due to the rapid release of copper ions from Cu nanosheets, resulting in the generation of self-nucleating samples.

Based on the experiment results, it can be clearly seen that  $\text{H}_2\text{O}_2$  and NaOH played a vital role



**Figure 4** SEM images of Cu@CuO nanorods in solution with different  $\text{H}_2\text{O}_2$  volumes: (a) Without  $\text{H}_2\text{O}_2$ ; (b) 1 mL; (c) 3 mL; (d) 7 mL





**Figure 5** SEM images of Cu@CuO nanorods in different concentration of NaOH solution: (a) Without NaOH; (b) 0.1 mol/L; (c) 0.3 mol/L; (d) 0.8 mol/L

in the generation of precursor triangular Cu@CuO nanorods. As conductive substrate and self-template, Cu nanosheets can be dissolved at high temperature to produce  $\text{Cu}^{2+}$  ions acting as copper source. In the presence of  $\text{H}_2\text{O}_2$  solution, the gas of oxygen was produced, which acts with NaOH solution to form the triangular Cu@CuO nanorods. Without adding  $\text{H}_2\text{O}_2$  solution, the  $\text{Cu}^{2+}$  ions combine with  $\text{OH}^-$  ions to form  $\text{Cu}(\text{OH})_2$  at a slower reaction rate. In the absence of NaOH solution, Cu was oxidized by oxygen to obtain nanoparticles, but the morphology of nanorods could not be obtained.

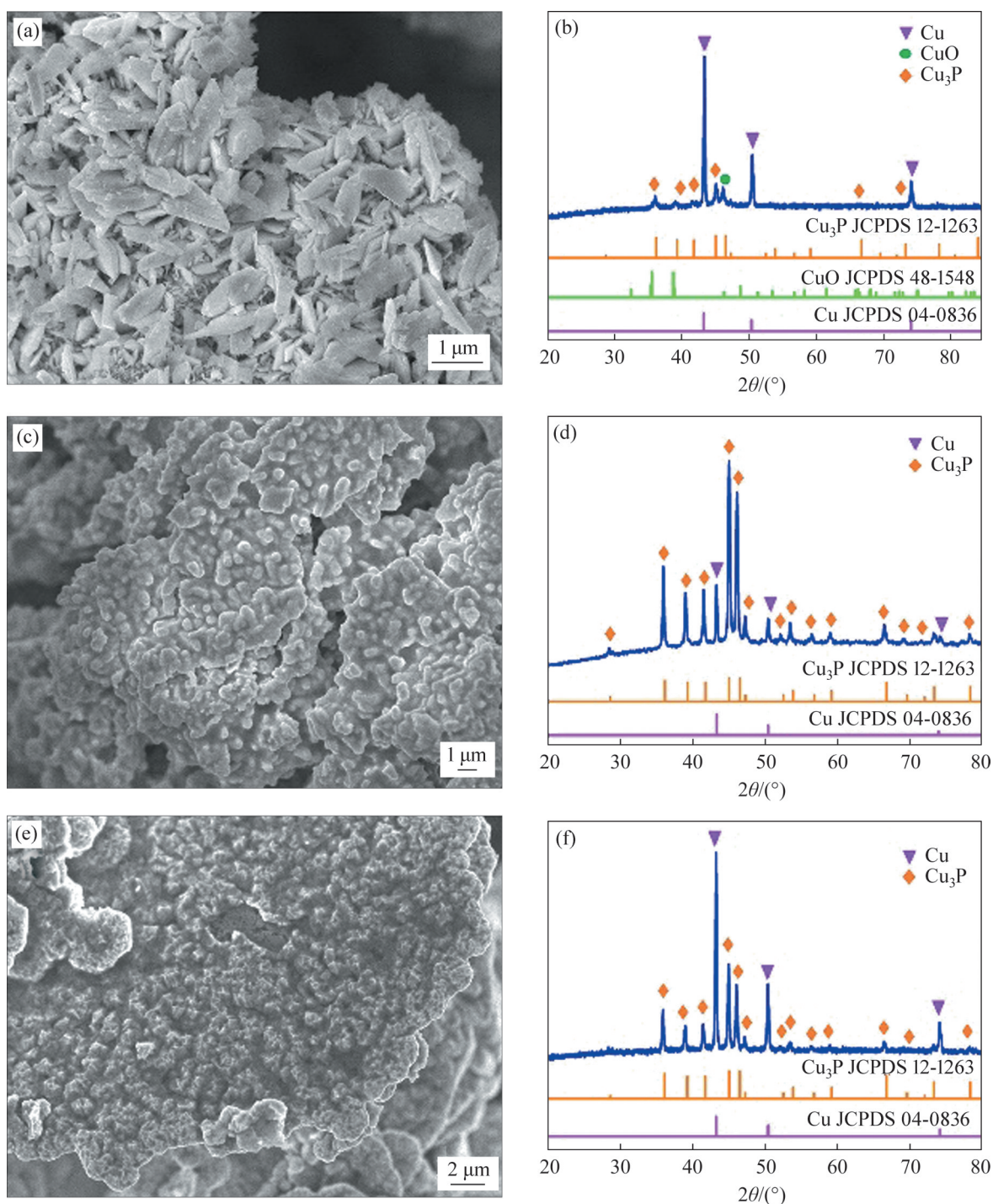
For comparison, the samples phosphated at different temperatures and the samples directly phosphated with copper nanosheets were synthesized by the same procedure. The morphologies and crystal structures of the products were characterized by SEM and XRD, as shown in Figure 6. When the phosphating temperature was 270 °C, the morphology of the sample did not change significantly compared with the CuO nanorods, and only the surface became rough (Figure 6(a)). The corresponding XRD pattern shows that the diffraction peaks of Cu, CuO and

$\text{Cu}_3\text{P}$  appeared simultaneously (Figure 6(b)), which confirmed that a small part of CuO was still not completely phosphatized at a low phosphating temperature, and the product was named Cu@CuO/Cu<sub>3</sub>P-270 nanorods. Until the phosphating temperature reached 300 °C, the surface of nanorods became rougher and all peaks of XRD pattern could be assigned to Cu and Cu<sub>3</sub>P (Figures 6(c) and (d)). The results show that the intensity of the copper peak became weaker, which proved that part of the substrates of Cu nanosheets were involved in the phosphating reaction.

As a comparison, the pure Cu nanosheets were directly phosphated at 280 °C (Cu/Cu<sub>3</sub>P nanosheet), and the morphology and crystal structure of products were characterized, as shown in Figures 6(e) and (f). The surface of copper nanosheets has changed from smooth to uneven, but no nanorod structure appeared (Figure 6(e)). The corresponding XRD pattern (Figure 6(f)) confirmed that the surface of the Cu nanosheet participated in the phosphating reaction, and part of Cu nanosheet was converted to Cu<sub>3</sub>P.

To evaluate the electrocatalytic activities of

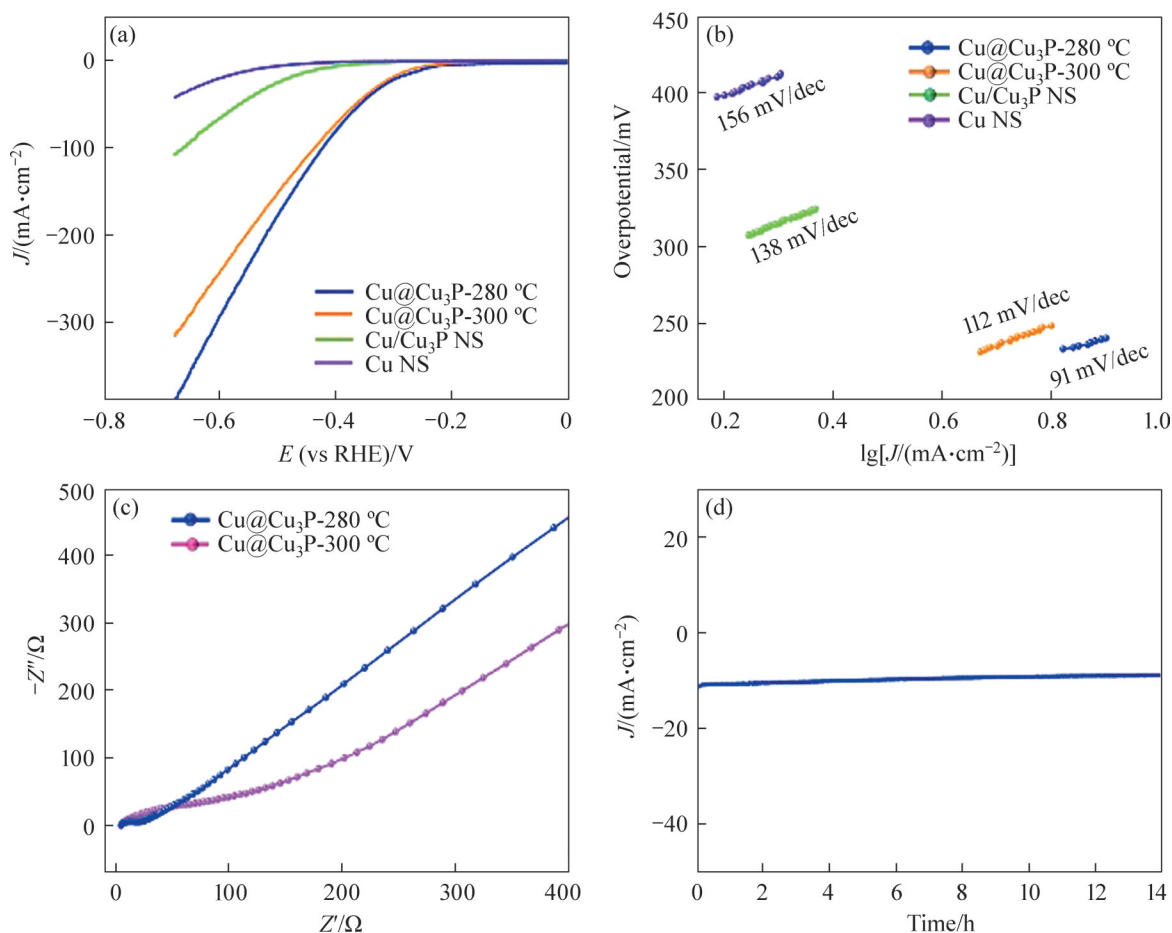




**Figure 6** SEM images (a, c, e) and XRD patterns (b, d, f) of Cu@Cu<sub>3</sub>P nanorods and nanosheets at different phosphating temperature: (a, b) Cu@Cu<sub>3</sub>P nanorods, 270 °C; (c, d) Cu@Cu<sub>3</sub>P nanorods, 300 °C; (e, f) Cu/Cu<sub>3</sub>P nanosheets, 28 °C

triangular Cu@Cu<sub>3</sub>P-280 nanorods, HER and OER were measured. The catalytic activity of triangular Cu@Cu<sub>3</sub>P-280 nanorods for HER was tested in a standard three-electrode system with 0.5 mol/L H<sub>2</sub>SO<sub>4</sub> electrolyte. For comparison, the reference samples, including Cu@Cu<sub>3</sub>P-300, direct phosphating of Cu nanosheets (Cu/Cu<sub>3</sub>P NS) and Cu nanosheet, were also measured without iR

correction in this study, as shown in Figure 7(a). The triangular Cu@Cu<sub>3</sub>P-280 nanorod displayed a good HER catalytic performance with a low overpotential of 252 mV to reach a current density of 10 mA/cm<sup>2</sup>, which is much smaller than that of Cu@Cu<sub>3</sub>P-300 (272 mV), Cu/Cu<sub>3</sub>P NS (433 mV) or Cu NS (542 mV). More importantly, an overpotential of 422 mV is required for the



**Figure 7** Electrochemical activity of the as-prepared triangular Cu@Cu<sub>3</sub>P nanorods electrode for HER in 0.5 mol/L H<sub>2</sub>SO<sub>4</sub>: (a) Polarization curves; (b) Tafel plots; (c) EIS Nyquist plots; (d) Current density curve related to time at 10 mA/cm<sup>2</sup>

triangular Cu@Cu<sub>3</sub>P-280 nanorod to reach the current density of 100 mA/cm<sup>2</sup>, which is lower than that of Cu@Cu<sub>3</sub>P-300 (439 mV) or Cu/Cu<sub>3</sub>P NS (664 mV). While, the polarization curve of Cu@CuO/Cu<sub>3</sub>P-270 in 0.5 mol/L H<sub>2</sub>SO<sub>4</sub> was also examined for comparison, as shown in Figure S3. From the results, HER performance of Cu@CuO/Cu<sub>3</sub>P-270 is worse than that of Cu@Cu<sub>3</sub>P-280 or Cu@Cu<sub>3</sub>P-300.

The Tafel slopes for all electrocatalysts were further tested to gain an insight into the HER kinetics of the triangular Cu@Cu<sub>3</sub>P nanorod (Figure 7(b)). The Tafel slope of 91 mV/dec for the triangular Cu@Cu<sub>3</sub>P-280 nanorod electrode is lower than that of Cu@Cu<sub>3</sub>P-300 (131 mV/dec), Cu/Cu<sub>3</sub>P NS (138 mV/dec) or Cu NS (156 mV/dec), implying that a more rapid HER rate can be obtained in future industrial application when using the triangular Cu@Cu<sub>3</sub>P-280 nanorod as an electro-

catalyst. Electrochemical impedance spectroscopy (EIS) measurements revealed that triangular Cu@Cu<sub>3</sub>P-280 nanorod has a much smaller impedance, further confirming that its HER kinetics is faster than Cu@Cu<sub>3</sub>P-300 nanorod (Figure 7(c)). The long-term stability of an electrode is another important issue to consider for practical application. The durability of triangular Cu@Cu<sub>3</sub>P-280 nanorod is examined at 10 mA/cm in acid solution, as shown in Figure 7(d). The time-dependent current density only exhibits a slight decrease after 14 h, which reveals the excellent stability in the electrocatalytic process of this triangular nanorod.

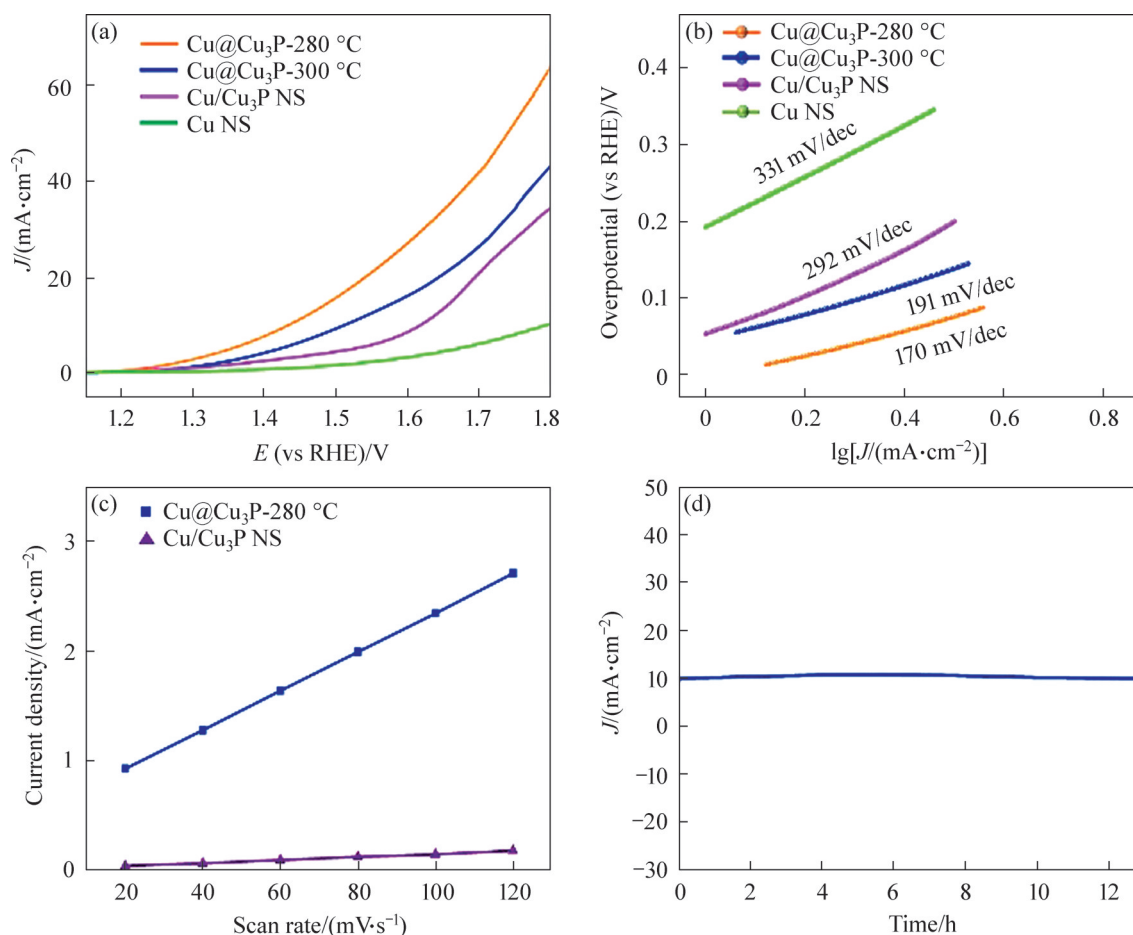
The OER activities of the triangular Cu@Cu<sub>3</sub>P-280 nanorod were also evaluated by using a typical three-electrode system in 1 mol/L KOH solution without iR drop compensation. All the potentials are relative to the reversible hydrogen electrode (RHE). Cu@Cu<sub>3</sub>P-300, Cu NS and Cu/Cu<sub>3</sub>P NS were

investigated under the same condition. Polarization curves of the samples obtained through the LSV measurements are presented in Figure 8(a). To reach a current density of  $10 \text{ mA/cm}^2$ , triangular  $\text{Cu@Cu}_3\text{P}$ -280 nanorod needs an overpotential of 200 mV, which is smaller than that of  $\text{Cu@Cu}_3\text{P}$ -300 nanorod (281 mV),  $\text{Cu/Cu}_3\text{P}$  NS (386 mV) or  $\text{Cu}$  NS (561 mV). At higher current density of  $30 \text{ mA/cm}^2$ , triangular  $\text{Cu@Cu}_3\text{P}$ -280 nanorod exhibited an overpotential of 390 mV, followed by  $\text{Cu@Cu}_3\text{P}$ -300 nanorod ( $\eta_{30}=492 \text{ mV}$ ) and  $\text{Cu/Cu}_3\text{P}$  NS ( $\eta_{30}=530 \text{ mV}$ ). The corresponding Tafel slopes are shown in Figure 8(b), the Tafel slope of triangular  $\text{Cu@Cu}_3\text{P}$ -280 nanorod is  $170 \text{ mV/dec}$ , which is lower than that of  $\text{Cu@Cu}_3\text{P}$ -300 nanorod ( $191 \text{ mV/dec}$ ),  $\text{Cu/Cu}_3\text{P}$  NS ( $292 \text{ mV/dec}$ ) or  $\text{Cu}$  NS ( $331 \text{ mV/dec}$ ), implying the most favorable catalytic kinetics toward OER. Additionally, the ECSA was evaluated through testing double-layer capacitance ( $C_{dl}$ ). The  $C_{dl}$  values of triangular  $\text{Cu@Cu}_3\text{P}$ -280 nanorod and  $\text{Cu/Cu}_3\text{P}$  NS are  $17.8$  and  $1.38 \text{ mF cm}^{-2}$ ,

respectively (Figure 8(c)), which are calculated by plotting the positive and negative current density differences at a given potential against the CV scan rates. The triangular  $\text{Cu@Cu}_3\text{P}$ -280 nanorod possesses a higher  $C_{dl}$  than  $\text{Cu/Cu}_3\text{P}$  NS since the triangular nanorod structure leads to the high exposure of the surface. This indicates that the formation of nanorod arrays on the nanosheet surface can expose more active sites and promote the reaction kinetics. Moreover, the long-term stability was tested by a constant potential method for 13.5 h, where no significant change in the current was observed (Figure 8(d)). Detailed comparison of triangular  $\text{Cu@Cu}_3\text{P}$ -280 nanorods with other reported  $\text{Cu}_3\text{P}$  electrocatalysts is shown in Table S1, indicating its superior or comparable performance [43–45].

## 4 Conclusions

In summary, we have successfully prepared

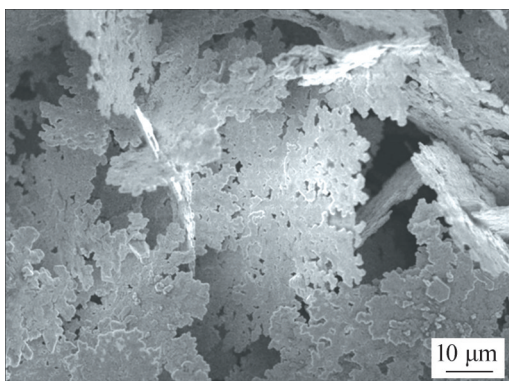


**Figure 8** Electrocatalytic activity of the as-prepared triangular  $\text{Cu@Cu}_3\text{P}$  nanorods electrode for OER in 1 mol/L KOH: (a) Polarization curves; (b) Tafel plots; (c) Capacitive current 1.07 V (vs RHE) against the scan rates; (d) Current density curve related to time at  $10 \text{ mA/cm}^2$

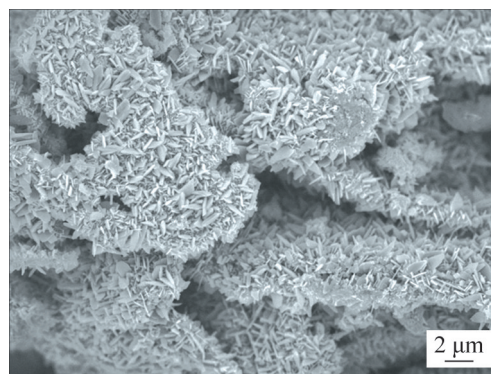


triangular Cu@Cu<sub>3</sub>P nanorod via a facile three-step synthetic strategy, including a hydrothermal synthesis of Cu nanosheet and triangular Cu@CuO nanorod precursor, and a following phosphidation with NaH<sub>2</sub>PO<sub>2</sub> in Ar atmosphere. The amount of H<sub>2</sub>O<sub>2</sub> and the concentration of NaOH solution have a great influence on the morphology of the product precursor. The catalyst exhibits a high catalytic activity for overall water splitting as a bifunctional catalyst. The enhanced catalytic performance is attributed to the structure and composition of the catalyst. The triangular nanorod provides more active sites owing to the high specific surface area. The triangular Cu@Cu<sub>3</sub>P nanorods show good electrical conductivity, benefiting from the substrate of Cu nanosheet which can effectively reduce the resistance of the catalytic system. This triangular Cu@Cu<sub>3</sub>P nanorod with enhanced HER and OER electrocatalytic performance and long-term stability might be a promising replacement to the noble metal electrocatalysts in water splitting and fuel cells.

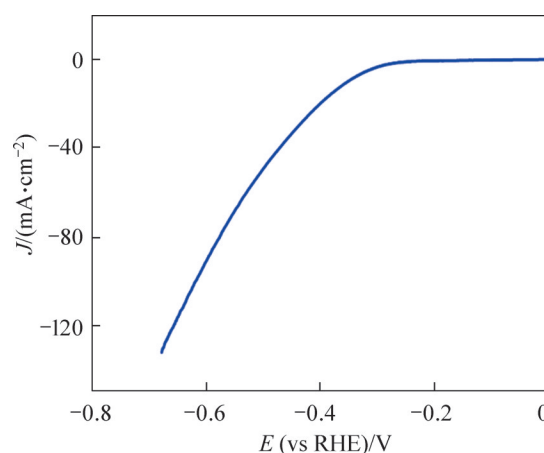
## Appendix



**Figure S1** SEM image of Cu nanosheet



**Figure S2** SEM image of triangular Cu@CuO nanorod



**Figure S3** Polarization curve for Cu@CuO/Cu<sub>3</sub>P-270 nanorods in 0.5 mol/L H<sub>2</sub>SO<sub>4</sub> solution

## Contributors

DANG Rui provided the concept and edited the draft of manuscript. DANG Rui and XU Xiu-feng validated the proposed method with practical experiments. XIE Meng-meng accomplished writing-review and editing. All authors have read and agreed to the published version of the manuscript.

**Table S1** Comparison of HER and OER catalytic performance of triangular Cu@Cu<sub>3</sub>P nanorods and other reported Cu<sub>3</sub>P electrocatalysts in acidic and basic conditions

Materials	Electrolyte	Overpotential, $\eta_{\text{onset}}$ /mV	Overpotential, $\eta_{10}$ /mV	Long-time stability	Ref.
Cu <sub>3</sub> P nanocubes	HER in 0.5 mol/L H <sub>2</sub> SO <sub>4</sub>	145	300	1000 cycles	[43]
Cu <sub>3</sub> P NB/Cu	OER in 1 mol/L KOH	320	380	—	[33]
Cu <sub>3</sub> P NB/Cu	HER in 0.5 mol/L H <sub>2</sub> SO <sub>4</sub>	-44	117	9 h ( $\eta_{20}$ )	[33]
Cu <sub>3</sub> P-450	OER in 1 mol/L KOH	—	290	19 h (1.55 V (vs RHE))	[44]
Cu <sub>3</sub> P@C-120	HER in 0.5 mol/L H <sub>2</sub> SO <sub>4</sub>	51	124	20 h ( $\eta_{10}$ )	[45]
Cu <sub>3</sub> P@C-120	OER in 1 mol/L KOH	—	300	20 h ( $\eta_{10}$ )	[45]
Cu@Cu <sub>3</sub> P NR	HER in 0.5 mol/L H <sub>2</sub> SO <sub>4</sub>	—	252	14 h ( $\eta_{10}$ )	This work
Cu@Cu <sub>3</sub> P NR	OER in 1 mol/L KOH	—	200	13.5 h ( $\eta_{10}$ )	This work

Note:  $\eta_{10}$  and  $\eta_{20}$  are overpotentials under conditions of 20 and 10 mA/cm<sup>2</sup>, respectively.

## Conflict of interest

DANG Rui, XU Xiu-feng and XIE Meng-meng declare that they have no conflict of interest.

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(Edited by FANG Jing-hua)



## 中文导读

### 铜纳米片表面磷化铜三角纳米柱的构筑及其电催化分解水研究

**摘要:** 开发能够取代贵金属的高效、廉价、稳定的非贵金属催化剂, 对于推动电催化水裂解析氢、析氧技术的发展具有重要意义。本文以铜纳米片为模板和导电基底,  $\text{H}_2\text{O}_2$  为氧源,  $\text{NaOH}$  为 pH 调节剂, 通过原位氧化的方法在 Cu 纳米片表面构筑三角柱状结构的  $\text{Cu@CuO}$ ; 再对  $\text{Cu@CuO}$  进行低温磷化, 获得三角纳米柱状  $\text{Cu@Cu}_3\text{P}$ 。实验结果表明, 磷化温度在调节  $\text{Cu@Cu}_3\text{P}$  纳米柱阵列的形貌、组成及活性位点数量方面具有显著效果。当磷化温度为  $280\text{ }^\circ\text{C}$  时, 可获得形貌均一、活性高的纳米三角柱状阵列结构材料( $\text{Cu@Cu}_3\text{P-280}$ )。  $\text{Cu@Cu}_3\text{P-280}$  电极在  $0.5\text{ mol/L H}_2\text{SO}_4$  电解质中表现出良好的析氢催化活性, 在电流密度为  $10\text{ mA/cm}^2$  时, 其过电位为  $252\text{ mV}$ , 催化  $14\text{ h}$  后, 电流密度仍可达初始值的  $72\%$ 。  $\text{Cu@Cu}_3\text{P-280}$  电极在  $1\text{ mol/L KOH}$  电解质中也展现出优异的催化析氧活性, 当电流密度为  $10\text{ mA/cm}^2$  时, 其过电位仅为  $200\text{ mV}$ , 催化  $12\text{ h}$  后, 电流密度保持初始值的  $93\%$ 。此研究为可持续、低价、双功能电催化材料的定向设计与制备提供理论基础。

**关键词:**  $\text{Cu@Cu}_3\text{P}$ ; 电催化; 析氢反应; 析氧反应