



Direct fabrication of highly porous graphene/TiO₂ composite nanofibers by electrospinning for photocatalytic application

HE Yuan(何缘), LIU Yun-guo(刘云国)

College of Environmental Science and Engineering, Key Laboratory of Environmental Biology and Pollution, Hunan University, Changsha 410072, China

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Abstract: We reported the fabrication of highly porous graphene/TiO₂ composite nanofibers in the form of a nonwoven mat by electrospinning followed by calcination in air at 450 °C. The graphene can uniformly disperse in highly porous TiO₂ nanofibers. The highly porous graphene/TiO₂ composite nanofibers exhibited excellent catalytic activities. The new method for producing graphene/TiO₂ composite nanofibers is versatile and can be extended to fabricate various types of metal oxide and graphene nanocomposites.

Key words: porous structure; graphene; titanium dioxide; photocatalytic application

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1 Introduction

Nanomaterials are expected to play an increasingly important role in industrial manufacture due to their fascinating electronic and optoelectronic properties compared to their bulk counterparts [1–3]. Graphene, a perfect two-dimensional material, has become a new star on the horizon of nanomaterials since it was found in 2004 due to the extraordinary electronic properties, such as ballistic transport over 0.4 mm in length, quantum hall effect at room temperature, high electron mobility, and single-molecule field-effect sensitivity [4–8]. Graphene and its composites can be widely applied in many fields including photocatalysis, liquid crystal displays, lithium ion batteries, solar cells, wound healing [9], and so on [10–15]. So, fabrication of graphene composites is very interesting and important, especially for simple

method to obtain graphene composites.

TiO₂ is regarded as one of the most important metal-oxide-semiconductors and has been found its applications in a wide range of fields, such as solar cells, photocatalysis, lithium-ion batteries, H₂ energy production, and antimicrobial activities [16–19]. However, the photoefficiency of TiO₂ is substantially limited by its large band gap energy (3.0–3.2 eV). It is very important for patenting environment remediation to developing visible-light-driven photocatalysts [20, 21]. The TiO₂ and graphene composites are currently being considered a potential material to develop visible-light-driven photocatalysts and improve the photoefficiency of TiO₂ [22]. So far, TiO₂/graphene composites (TGCs) have been fabricated by many kinds of methods, including molecular grafting, solvothermal method, and so on [23, 24]. However, most of the graphene/TiO₂ composites have been fabricated with complicated processes, and most of TiO₂ is

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Corresponding author: LIU Yun-guo, PhD, Professor; Tel: +86–13974881035; E-mail: liuyunguo@hnu.edu.cn

of nanoparticles but not 1D nanostructures, which is more promising in the photocatalysis applications due to excellent mobility of charge carriers, scattering more light at the red part of the solar spectrum, and the existence of straight pores can enhance the accessibility of electrodes to the hole transporting materials and hence enhance charge collection and transport compared to nanoparticles in photovoltaic and photocatalysis applications [25, 26]. Thus it is desirable to develop a simple method to integrate graphene into one-dimensional (1D) nanostructured TiO₂ for enhanced photovoltaic and photocatalytic applications.

As a straightforward method to produce 1D nanostructures ranging from the microscale to the nanoscale, electrospinning has attracted extensive attentions in the past two decades [27–30]. RAMAKRISHNA et al [30] have reported direct fabrication of TiO₂ nanorods-graphene composite nanomaterials by electrospinning. The composite showed excellent photovoltaic and photocatalytic properties [31]. However, as far as we known, 1D graphene/TiO₂ composite with highly porous structure has not been prepared.

In this paper, the highly porous graphene/TiO₂ nanofibers were directly prepared by electrospinning. The composites displayed excellent photocatalytic performance for the degradation of Rhodamine B (RhB).

2 Experimental

The morphology of the graphene was investigated by scanning electron microscope (SEM, Hitachi S-4800) with an accelerating voltage of 5 kV and transmission electron microscope (TEM, FEI Tecnai F30) with an accelerating voltage of 300 kV. Figures 1(a)–(c) (SEM images) and Figure 1(d) (TEM image) present a typical 2D structure of graphene. Raman spectroscopy is used as an efficient way to identify graphene. Figure 1(e) shows Raman spectrum of our graphene sample by employing a LabRAM HR800 (HORIBA Jobin Yvon, France) confocal Raman spectrometer with an excitation laser wavelength of 600 nm. The major Raman features of graphene are so called G band (~1580 cm⁻¹) and 2D band (~2670 cm⁻¹). The G band originates from in-plane vibration of sp²

carbon atoms and is a doubly degenerated (TO and LO) phonon mode (E_{2g} symmetry) at the Brillouin zone center. The 2D band originates from a two-phonon double resonance Raman process. Apparently, HRTEM can directly determine the number of graphene layers. The inset of Figure 1(e) HRTEM demonstrates few layers of graphene (about four layers).

2.1 Materials preparation

All the chemicals were of analytical grade and were used without further purification. The synthesis procedure of the graphene/TiO₂ composite nanofibers is presented in Figure 2. In the first step, 0.60 g tetrabutyl titanate was dissolved in 0.79 g ethanol and 1.05 g ethanoic acid (Tianjin Chemical Corp., China). This mixed solution was stirred for 20 min. Second, 0.2 g polyvinyl acetate (PVAC, Sigma Aldrich) was dissolved in N,N-dimethylformamide and stirred for 60 min. Then the two prepared solutions were mixed together and stirred for 1 h to obtain a viscous gel of PVAC/titanium acetate composite solution. At last, 0.2 g mineral oil (Tianjin Chemical Corp., China) and 4 mg graphene were added to the above solution and stirred at room temperature for 48 h to obtain a stable and homogeneous emulsion. The graphene was prepared by micromechanical cleavage in which highly oriented pyrolytic graphite was peeled by using a scotch tape [32, 33]. In a typical electrospinning process in our work, inner diameter of the spinneret was 0.6 mm. Grounded aluminum strips (1 cm in width) with parallel gaps about 1 cm were used as the collectors. The needle was connected to a high-voltage supply capable of generating DC voltage up to 60 kV. The voltage for electrospinning was kept at 18 kV. A distance of 15 cm was maintained between the tip of the spinneret and the collector. During the electrospinning process, the environmental temperature was maintained at 60 °C. After electrospinning, the fibers were heated from room temperature to 450 °C at a heating rate of 2 °C/min, and then held at 450 °C for 2 h in air.

2.2 Photocatalytic test

Absorption spectrum was tested by UV-Vis spectrum of powders using a TU-1901 spectrophotometer. The inset in Figure 2 shows a

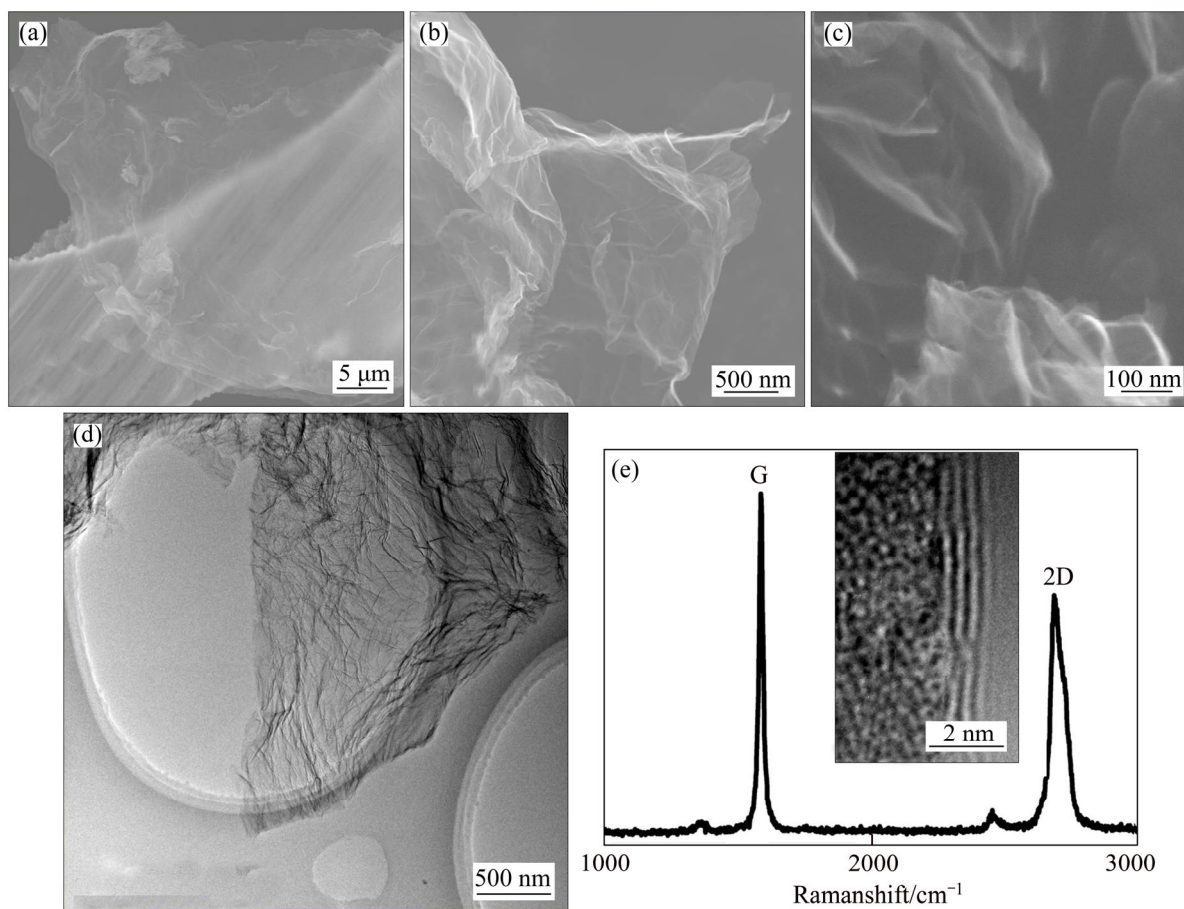


Figure 1 Scanning electron micrographs (a, b, c), transmission electron micrograph (d) of graphene and Raman spectrum of few layers of graphene (e) (the inset is high-resolution TEM (HRTEM) of graphene)

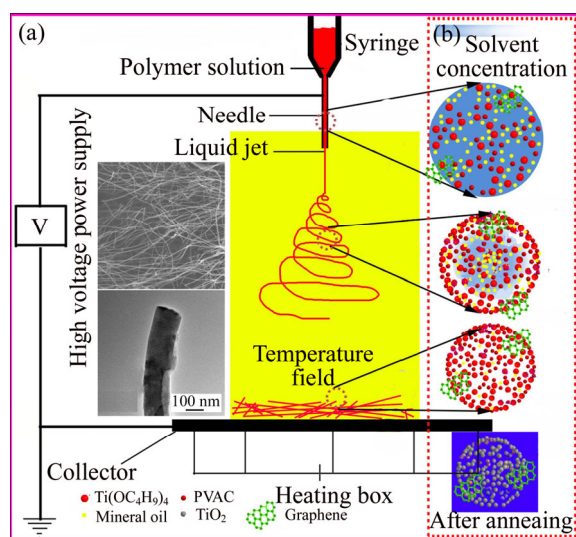


Figure 2 Schematic diagram of electrospinning fabrication of graphene and highly porous TiO₂ composite nanotubes (a) and proposed mechanism of formation of graphene and highly porous TiO₂ composite nanotubes (b)

possible mechanism of the formation of the

graphene/TiO₂ composite nanofibers. During the electrospinning process (before the electrospun nanofibers were evenly distributed on the collector (Figure 2(a)), oil, ethanol and ethanoic acid began to evaporate once the fibers were spun out from the nozzle. The region of solvent evaporation near the surface of the fiber enriched in PVAC and tetrabutyl titanate (Figure 2(b)). The remaining solvent and oil evaporated from the fibers. Graphene-PVAC-tetrabutyl titanate nanofibers formed on the collector, and some oil remained in graphene-PVAC and tetrabutyl titanate. Because there was some oil in graphene-PVAC and tetrabutyl titanate, highly porous nanofibers can be achieved after annealing process [34].

3 Results and discussion

Figure 3(a) presents PVAC-tetrabutyl titanate nanofibers membrane containing graphene by electrospinning for 60 min. The inset in Figure 3(a)

is optical micrograph and clearly shows that well-aligned nanofibers are obtained by electrospinning over large area. Here, in order to get aligned nanofibers, a stainless steel U-shaped frame with a distance of 8 mm between two branches was used to transfer the aligned nanofibers by vertically moving through the gap. Our group demonstrated that the color of as-spun PVAC–tetrabutyl titanate nanofibers was usually white [9], which is an indirect proof of the homodisperse of graphene in PVAC–tetrabutyl titanate nanofibers. Figure 3(b) shows SEM image of the graphene/TiO₂ composite nanofibers, and the inset is the enlarged view of the nanofibers. Many pores with an average diameter of 15–20 nm are evenly distributed on the surface of the nanofibers, which can be further determined by TEM image. As shown in of Figure 3(c), the lattice spacing is measured to be 0.23 nm comparable to 0.25 nm for graphite. Additionally, the inset in

Figure 3(a) displays a lattice spacing of 0.35 nm, corresponding to graphene (002). The lattice fringe of the area marked in Figure 3(d) is 0.35 nm, corresponding to (101) planes of anatase TiO₂. The selected area electron diffraction pattern of the nanofibers is shown in Figure 3(e) and demonstrates the composition of the graphene and TiO₂.

The photoactivities of as-synthesized samples were evaluated for photodecomposition of RhB under visible light irradiation. The RhB concentration (%) after various intervals of time could be estimated by using the following equation:

$$\text{RhB concentration} = \frac{A_t}{A_0} \times 100\%$$

where A_0 is the initial RhB absorbance at 554 nm (RhB shows a major absorption band at 554 nm) and A_t is the absorbance after various intervals of

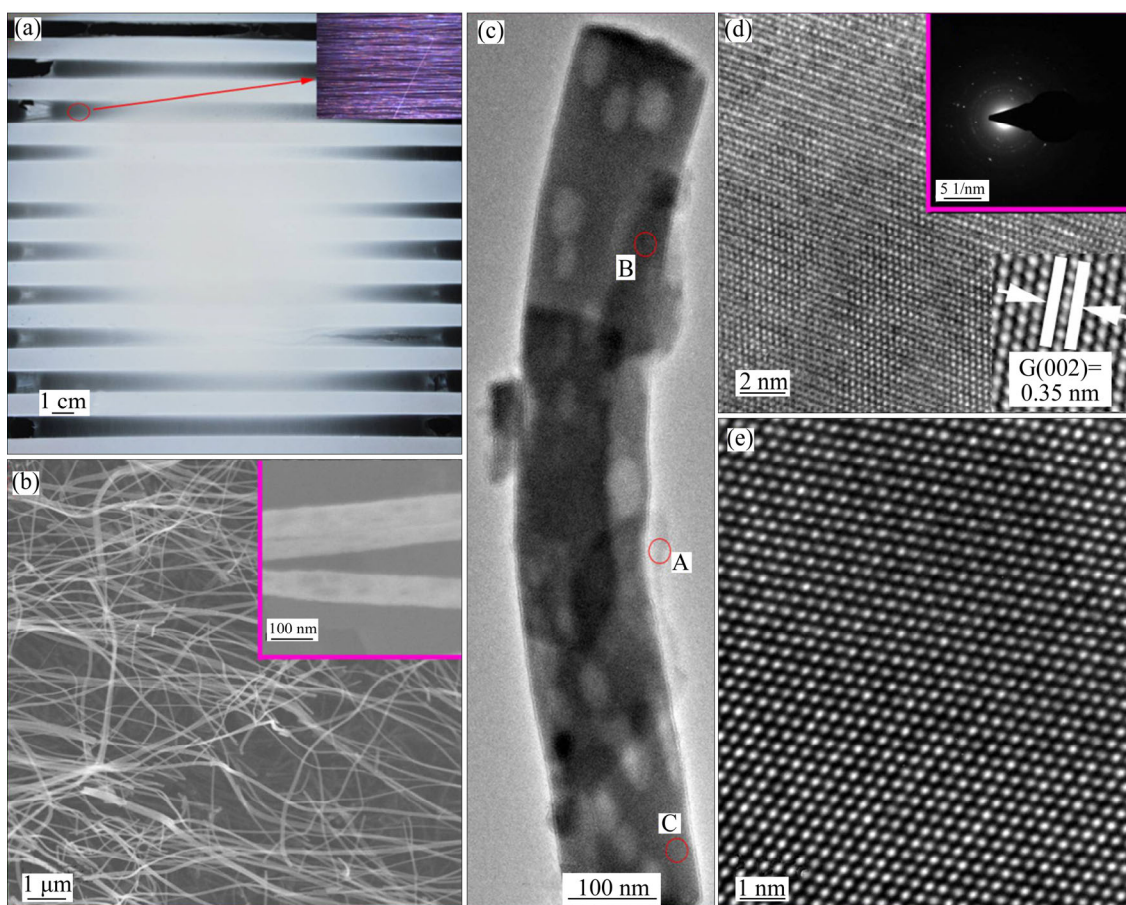


Figure 3 Photograph of PVAC/titanium acetate containing graphene nanosheets (a) (The inset is optical micrograph of the aligned nanofibers suspend over the U-shaped frame), scanning electron microscopy image of graphene and highly porous TiO₂ composite nanotubes (b) (The inset is enlarged image of the nanofibers), transmission electron microscopy image of graphene and highly porous TiO₂ composite nanotubes (c), High-magnification TEM (HRTEM) of region (d) (the insert show graphene interlayer spacing), and selected area electron diffraction pattern (e) of HRTEM lattice image taken from graphene and highly porous TiO₂ composite nanotubes marked by C

time t . To eliminate the effect of heating during irradiation, the beaker was immersed in an ice–water mixture. To demonstrate the excellent photocatalytic performance of highly porous graphene/TiO₂ composite-nanofibers, TiO₂ nanofibers were synthesized for comparisons. The evolution of the corresponding UV–Vis spectrum with UV exposure time is presented in Figure 4(a). It can be observed that the graphene/TiO₂ nanofibers composite presents more prominent photocatalytic activity. After 120 min irradiation, almost 100 % of the RhB molecules were decomposed by the porous graphene/TiO₂ nanofibers composite, while about 25% RhB molecules were intacted by the TiO₂ nanofibers as the photocatalyst. The ratio of RhB molecules self-decomposed is also shown for comparison in Figure 4(a). Moreover, the graphene/TiO₂ nanofibers composite exhibits very stable photocatalytic activity. There is no obvious decrease in the removal rate of RhB even after 10 cycles as shown in Figure 4(b). This phenomenon may be attributed to the photocatalytic process that the catalyst change is physical change, while the structure of the catalyst is not influenced.

Absorption range is very important in the photocatalysis [35], especially for the visible light photo degradation of contaminants. There is an obvious red shift of 20–30 nm in the absorption edge of highly porous graphene/TiO₂ nanofibers composite compared to pure TiO₂ nanofibers. This red shift should be attributed to the follow reasons: 1) TiO₂ and graphene can form Ti–O–C bond, similar to the case of carbon doped TiO₂ composites [36]; 2) Owing to the coupling of a small band gap or band gap-less semiconductor material of graphene with TiO₂, the visible light absorption capability of highly porous graphene/TiO₂ nanofibers composite is greatly enhanced as compared with TiO₂ nanofibers (Graphene was found to absorb a significant fraction of incident white light) [37, 38]. Therefore, a more efficient utilization of the solar spectrum could be achieved due to the extended photo responding range of 420–430 nm.

As also shown in Figure 5, there is an obvious red shift ranging from 10 to 25 nm in the absorption edge of the highly porous graphene/TiO₂ nanofibers composite, which may be attributed to

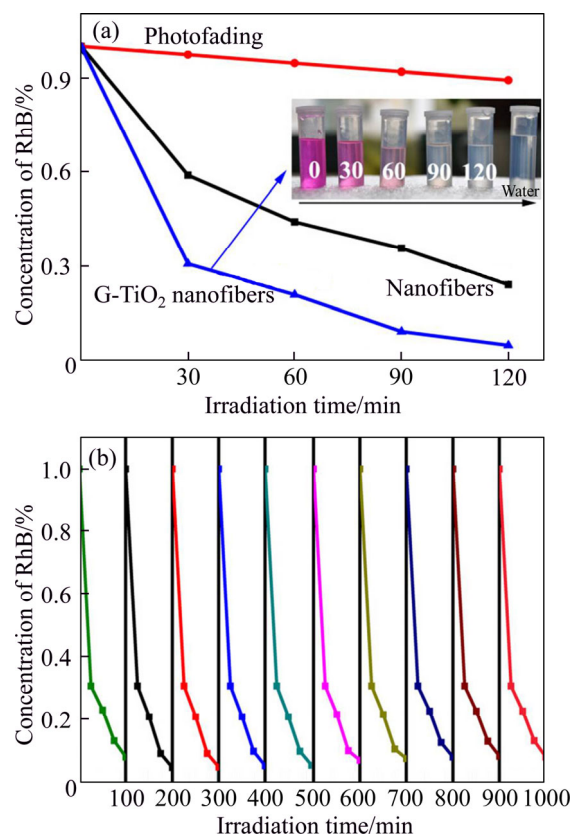


Figure 4 Photograph of Rhodamine B carried out under visible light irradiation for different time (a) and cycling tests of photocatalytic activity of graphene composite highly porous/TiO₂ nanofibers (b) (Catalyst 60 mg; 10 mg/L RhB 50 mL; 250 W mercury lamp; Luminous range 380–800 nm)

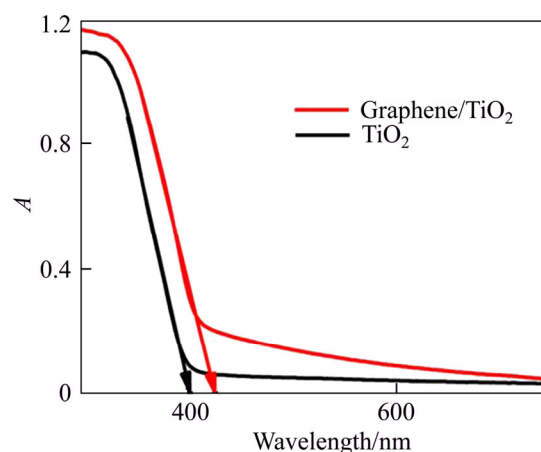


Figure 5 Absorption spectrum of graphene/highly porous TiO₂ nanofibers composite and pure TiO₂ nanofibers

the graphene with zero band gap introducing to TiO₂ nanofibers compared to pure TiO₂ nanofibers. The photoactivity of the highly porous graphene/TiO₂ nanofibers catalyst is expected to

facilitate its use in practical environmental remediation.

The photocatalyst stability has further been confirmed by XRD (before and after cycle test) in Figure 6(a). The XRD peaks unchanged after cycle test compared with XRD before cycle test, indicating that the photocatalyst is stable [39]. The photocatalytic degradation mechanism of RhB is proposed in Figure 6(b). Under UV-visible light irradiation, TiO_2 is excited. Some electrons can be quickly injected into the graphene from the CB of TiO_2 . The separated electrons can directly react with the dissolved oxygen molecules for yielding superoxide radical anions ($\text{O}_2^{\bullet-}$), then the superoxide radical anions experience protonation to generate hydroperoxy (HO_2^{\bullet}), and the HO_2^{\bullet} can further react with H_2O to produce $\bullet\text{OH}$ [40, 41]. Meanwhile, the h^+ can directly react with RhB and also react with H_2O to produce $\bullet\text{OH}$. Finally, RhB can be decomposed by the generated radicals and holes.

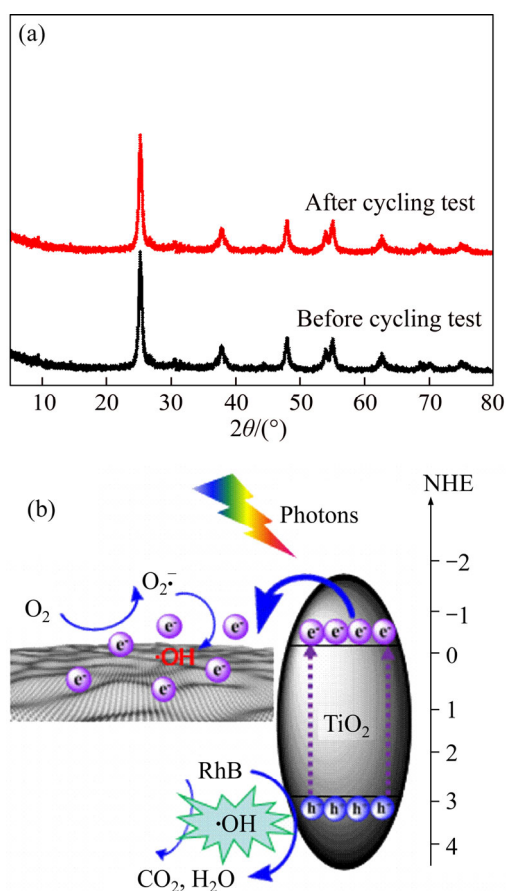


Figure 6 XRD patterns of graphene/ TiO_2 composite before and after cycle test (a) and diagram of photocatalytic reaction of RhB on graphene/ TiO_2 composite (b)

4 Conclusions

Graphene/ TiO_2 nanofibers composite was directly fabricated by electrospinning. The graphene can uniformly disperse in the highly porous TiO_2 nanofibers. The highly porous graphene/ TiO_2 nanofibers composite displayed excellent photocatalytic performance for the degradation of RhB. Such highly porous graphene/ TiO_2 nanofibers composite may bring new insights into the design of highly efficient photo catalysts for organic compounds and may have potential technological applications. This new method is anticipated to open a new possibility in the investigation of graphene/ TiO_2 composites and other metal oxide/graphene composite and promote their application in lithium ion batteries, hydrogen preparation, and so on.

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中文导读

静电纺丝制备高度多孔石墨烯/TiO₂ 复合材料的纳米纤维及光催化性能研究

摘要: 本文报道了一种利用静电纺丝制备高度多孔的石墨烯/TiO₂ 复合材料纳米纤维的方法。该方法通过静电纺丝将石墨烯与有机钛源复合, 经 450 °C 焙烧后得到分布均匀、具有高度多孔的石墨烯/TiO₂ 复合材料纳米纤维。该复合材料表现出优异的光催化性能。制备石墨烯/TiO₂ 复合材料纳米纤维的新方法用途广泛, 可用于制备各种金属氧化物和石墨烯纳米复合材料。

关键词: 多孔结构; 石墨烯; 二氧化钛; 光催化应用