# Electrical conductivity of  $Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>)$  cermet inert anode for aluminum electrolysis

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**Abstract:** Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermets containing mass fractions of Cu of  $5\%$ ,  $10\%$ ,  $15\%$  and  $20\%$  were prepared, and their electrical conductivities were measured at different temperatures. The effects of temperature and content of metal Cu on the electrical conductivity were investigated especially. The results indicate that the metallic phase Cu distributes evenly in  $10NiO-NiFe<sub>2</sub>O<sub>4</sub>$ ceramic matrix. The mechanism of electrical conductivity of  $Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>)$  cermets obeys the rule of electrical mechanism of semiconductor, the electrical conductivity for cermet containing 5% Cu increases from 2.70 to 20.41 S/cm with temperature increasing from 200 to 900 ℃. The change trend of electrical conductivity with temperature is similar with each other and it increases with increasing temperature and content of metal Cu. At 960 ℃, the electrical conductivity of cermet increases from 2.88 to 82.65 S/cm with the content of metal Cu increasing from 0 to 20%.

Key words: NiFe<sub>2</sub>O<sub>4</sub>; NiO; cermet; inert anode; aluminum electrolysis; electrical conductivity;

# **1 Introduction**

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The application of inert anode will eliminate anode carbon consumption and reduce environmental problems such as the emission of carbon dioxide, perfluorocarbon and evolve  $oxygen^{[1-2]}$ , which makes it commercially attractive. As an inert anode material, the anodic ohmic voltage drop caused by its resistance should be comparable with that of current carbon anode used in aluminum electrolysis, or the purpose for decreasing cell voltage and saving energy cannot be reached. So the inert anode materials need not only good corrosion resistance to the molten salts and high stability with respect to oxidizing gases such as oxygen, but also good electrical conductivity[3−4].

Recently, the studies of materials used as inert anode were mainly concentrated on the alloy<sup>[5]</sup> and the cermet<sup>[6]</sup>. Nickel ferrite spinel was found to have very low solubility at high temperature in the molten cryolite and was regarded as one of the most promising matrix of cermet inert anodes with respect to the corrosion resistance to melts<sup>[6−8]</sup>. But generally, the ceramic oxides have very low electrical conductivity even at high temperature<sup>[9−11]</sup>. To improve the electrical conductivity, metal was added and dispersed in ceramic oxide matrix. The intention is to make the inert anode material possess the desirable properties of metallic material as well as those of the ceramic.

LAI et al $^{[12]}$  studied the effect of content of NiO on the corrosion resistance and electrical conductivity of  $NiFe<sub>2</sub>O<sub>4</sub>$  based cermets, and the content of NiO in ceramic phase was determined to be 10%. Though  $NiFe<sub>2</sub>O<sub>4</sub>$ -based cermet anode containing 17% Cu was chosen and tested by Alcoa, the reason for the determination of the content of metal Cu was not explained<sup>[13]</sup>. In this study, 10NiO-NiFe<sub>2</sub>O<sub>4</sub> based cermet inert anodes with Cu content of 0, 5%, 10%, 15% and 20%(mass fraction) were prepared by cold pressingsintering based on previous works $^{[14]}$ , and their electrical conductivities were measured at different temperatures.

#### **2 Experimental**

#### **2.1 Preparation of ceramic and cermets**

The raw materials, Cu powder, NiO and  $Fe<sub>2</sub>O<sub>3</sub>$  were all of reagent grade. 10Ni-NiFe<sub>2</sub>O<sub>4</sub> based cermet samples were prepared by conventional cold isostatic pressing-sintering process<sup>[14]</sup>. A proper amount of NiO and  $Fe<sub>2</sub>O<sub>3</sub>$  were mixed using a ball mill and then calcined to form the  $10NiO-NiFe<sub>2</sub>O<sub>4</sub>$  ceramic powder. The calcined ceramic powder was then mixed with Cu powder (mass fraction of Cu is 0, 5%, 10%, 15% and 20%, respectively) by ball milling in the medium containing organic dispersant and adhesive to avoid the metal oxide. Finally, the mixed ceramic-metal powder was dried and cold isostatically pressed into some cylindrical blocks ( $d20$  mm $\times$ 40 mm) at 200 MPa, and was sintered at certain temperature in the range from

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1 150 to 1 350 ℃ for 2 h under atmosphere of efficaciously controlled oxygen partial pressure to get the desired cermet samples. High density, low porosity and uniform dispersion of metal phase among ceramic phase were desirable.

#### **2.2 Measurement methods**

Phase compositions of  $10$ NiO-NiFe<sub>2</sub>O<sub>4</sub> ceramic and cermets were identified by X-ray diffraction analysis using Rigaku3014 X-ray diffractometer with Cu K*<sup>α</sup>* radiation. The microstructure was analyzed with JSM-6360LV scanning electron microscope.

High-temperature electrical conductivity of the ceramic and cermet samples were determined with a direct current four-end electrodes measuring apparatus<sup>[15]</sup>. In this study, test was carried out in atmosphere of efficaciously controlled oxygen partial pressure to avoid the oxidization of metal. The electrical conductivity *σ* was determined by the following equation:

$$
\sigma = \frac{l}{\pi r^2} \times \frac{I}{U} \tag{1}
$$

where *r* is the radius of cylinder sample, *l* and *U* are the interval and voltage drop between two electrodes, respectively, *I* is the current through the sample.

## **3 Results and discussion**

## **3.1 Phase analysis and microstructure of Cu/(10NiO-**NiFe<sub>2</sub>O<sub>4</sub>) cermet

The X-ray diffraction patterns of  $10NiO-NiFe<sub>2</sub>O<sub>4</sub>$ ceramic and  $5\%$ Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermet are shown in Fig.1. From Fig.1, it can be seen that there are no impurity diffract peaks for the ceramic matrix and cermet, which shows that the inert anode material with the desired phase composition can be obtained by controlling the partial oxygen pressure.



**Fig.1** XRD patterns of 10NiO-NiFe<sub>2</sub>O<sub>4</sub> ceramic and  $Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>)$  cermet (a)  $10NiO-NiFe<sub>2</sub>O<sub>4</sub>$ ; (b)  $5\%Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>)$ 

Fig.2 shows the microstructures of Cu/(10NiO- $NiFe<sub>2</sub>O<sub>4</sub>$ ) cermets with Cu content of 10% and 20%(mass fraction). From Fig.2, it can be seen that the metallic phase Cu is distributed evenly in 10NiO-NiFe<sub>2</sub>O<sub>4</sub> ceramic matrix. However, there are conglomerations for the cermet with 20%Cu, and the grain diameter of metallic phase Cu is larger than that with 10%Cu.



**Fig.2** Microstructures of  $Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>)$  cermets (a)10%Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>); (b)20%Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>)

## **3.2 Effect of Cu content on electrical conductivity of Cu/(10NiO-NiFe2O4) cermets**

The electrical conductivities of  $10$ NiO-NiFe<sub>2</sub>O<sub>4</sub> ceramic and  $Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>)$  cermets were measured at different temperatures, and the results are listed in Table 1.

As illustrated in Table 1, although the content of metal Cu added to  $10$ NiO-NiFe<sub>2</sub>O<sub>4</sub> ceramic is only 5%. the electrical conductivity at the same temperature is much higher than that of  $10$ NiO-NiFe<sub>2</sub>O<sub>4</sub> ceramic. For example, the electrical conductivities of  $10NiO-NiFe<sub>2</sub>O<sub>4</sub>$ ceramic and  $5\%$ Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermet at 900 °C are 2.40 and 20.41 S/cm, respectively, there exists difference of about 8 times between them. And the difference becomes more obvious at low temperature than at high temperature. From Table 1, the electrical conductivity of  $5\%$ Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermet is 2.70 S/cm at 200 ℃, which is 100 times as high as that of 10NiO-NiFe<sub>2</sub>O<sub>4</sub> ceramic.

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Mass fraction of $Cu$ %	Relative $density\%$	<b>Table 1</b> Electrical conductivities of TOINIO-INIFE <sub>2</sub> O <sub>4</sub> certainic and Cu/HOINIO-INIFE <sub>2</sub> O <sub>4</sub> ) certifiels Electrical conductivity/ $(S \cdot cm^{-1})$								
		$200$ °C	300 $^{\circ}$ C	400 °C	500 °C	600 °C	$700 \text{ °C}$	800 °C	900 °C	960 °C
$\overline{0}$	93.2	0.03	0.15	0.62	0.84	1.40	1.80	2.10	2.40	2.88
5	97.1	2.70	4.56	6.10	8.54	11.74	14.23	17.39	20.41	21.34
10	95.1	4.35	7.66	10.66	13.75	20.70	24.31	26.73	29.16	33.45
15	92.4	10.65	15.74	20.36	26.10	32.46	37.25	40.19	43.36	47.90
20	92.9	21.16	28.83	47.43	55.33	65.32	69.46	77.31	83.00	82.65
Carbon		172.70	178.30	154.50	190.10	187.90	194.20	208.80	211.40	216.40

**Table 1** Electrical conductivities of 10NiO-NiFe  $\Omega$  ceramic and  $Cu/(10Ni\Omega NiFe\Omega)$  cermets

The data listed in Table 1 also show that the electrical conductivity of  $Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>)$  cermet increases with increasing content Cu at the same temperature. But there is no liner correlation between the electrical conductivity and Cu content. For example, at 960 ℃, the electrical conductivity increases from 2.88 to 21.34 S/cm when 5%Cu is added to the 10NiO-NiFe<sub>2</sub>O<sub>4</sub> ceramic, the ascend value is one level of magnitude. And when Cu content is added up to 20%, the electrical conductivity of cermet reaches 82.65 S/cm. Though their electrical conductivities are low compared with that of the current carbon anode used in aluminum electrolysis, which is 216.40 S/cm at 960 ℃, the anodic voltage drop caused by its resistance can be decreased by optimizing the configuration of the cermet inert anode. Thus, the voltage drop should be comparable with that of current carbon anode. So, it is acceptable as inert anode material in aluminum electrolysis.

From Fig.2,  $Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>)$  cermets prepared include metallic phase Cu and oxide ceramic phase  $10NiO-NiFe<sub>2</sub>O<sub>4</sub>$ . So, the cermets may conduct electricity by semiconductors and metallic conductivity. WANG et al<sup>[9]</sup> found that the electrical conductivity of cermet( $\sigma$ ), with metal content less than 30%, obeys the following equation:

$$
\sigma = \sigma_0 / (1 - x_1) \tag{2}
$$

where  $\sigma_0$  is the electrical conductivity of ceramic,  $x_1$  is the fraction of the effective transfer distance of electric charges of metal phase in the cermet and it is related to: 1) the content of the metal; 2) the dispersion of the metal phase among ceramic phases. Eqn.(2) shows that  $\sigma$  is only affected by  $\sigma_0$  and  $x_1$ , not by the electrical conductivity of metal phase. When the content of the metal added in the cermet increases,  $x_1$  increases, therefore the electrical conductivity of Cu/(10NiO- $NiFe<sub>2</sub>O<sub>4</sub>$ ) cermet increases.

### **3.3 Effect of temperature on electrical conductivity of**  Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermets

According to the theory of energy band of the solid $[16]$ , the electrical conductivity of semiconductor can be given by

 $\sigma = n_{\rm e} e u$  (3)

where *e* is the electronic charge, *u* is the mobility of the electron, which does not change obviously with temperature,  $n_c$  is the charge carrier density, which is specified as

$$
n_{\rm c} = N_0 \exp[-E/(2kT)] \tag{4}
$$

where  $E$  is the energy gap that is the difference between the conduction band and the valence band in oxide crystal,  $N_0$  is the density of energy state,  $k$  is the Boltzmann constant, *T* is absolute temperature.

Therefore,

$$
\sigma = N_0 e u \exp[-E/(2kT)] = \sigma_0 \exp[-E/(2kT)]
$$
 (5)

where  $\sigma_0$  is the electrical conductivity at infinitely high temperatures.

Fig.3 shows the relation between electrical conductivity and temperature, which indicates that the electrical conductivity of  $Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>)$  cermet changes with the temperature, plotted as  $\ln \sigma$  vs  $1/T$ . It is obvious that  $\ln \sigma$  is linear with  $1/T$ , and this is fit to this theory. And the curves also show that there is the Curie temperature, where a change in the slope of the straight line appears. There are similar slopes of straight lines for  $Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>)$  cermets.



**Fig.3** Relation between  $\ln \sigma$  and  $1/T$  of different samples  $1-10$ NiO-NiFe<sub>2</sub>O<sub>4</sub>;  $2-5\%$ Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>);  $3-10\%$ Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>);  $4-15\%$ Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>);

 $5 - 20\%$ Cu $(10$ NiO-NiFe<sub>2</sub>O<sub>4</sub>)

In fact, for 10NiO-NiFe<sub>2</sub>O<sub>4</sub> ceramic, the presence of Ni ions on octahedral-(B) sites, Fe ions on tetrahedral-(A) sites and B-sites favor the ion exchange interactions expressed as

$$
Ni^{2+} + Fe^{3+} = Ni^{3+} + Fe^{2+}
$$
 (6)

Thus the conductive mechanism for the N-type semiconductor is predominant due to the hopping of electrons from  $Fe^{2+}$  to  $Fe^{3+}$  ions, while that for the P-type semiconductor is due to the holes transfer from  $Ni<sup>3+</sup>$  to  $Ni<sup>2+</sup>$  ions. It seems that the present sample has both types of charge carriers, which participate in the conduction process, and the conductivity is the sum of conductivity for both types. When temperature increases, the charge carrier density increases and therefore the electrical conductivity is improved.

## **4 Conclusions**

1) The metallic phase Cu distributes evenly in  $10NiO-NiFe<sub>2</sub>O<sub>4</sub>$  ceramic matrix. But conglomeration phenomenon appears when Cu content increases up to 20%, the grain diameter of metallic phase Cu becomes larger.

2) The mechanism of electrical conductivity of  $10NiO-NiFe<sub>2</sub>O<sub>4</sub>$  ceramic and  $Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>)$ cermets obeys the rule of electrical mechanism of semiconductor. Their electrical conductivities mainly depend on temperature, electrical conductivity of ceramic matrix, content and dispersion of the metal phase Cu among ceramic matrix. The change trend of the electrical conductivity with temperature is similar with each other and increases with increasing temperature and content of metal Cu. The electrical conductivity of  $5\%$ Cu/(10NiO-NiFe<sub>2</sub>O<sub>4</sub>) cermet increases from 2.70 to 20.41 S/cm when temperature increases from 200 to 900 ℃. At 960 ℃, the electrical conductivity of the sample increases from 2.88 to 82.65 S/cm when the content of metal Cu increases from 0 to 20%.

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