

Preparation and characterization of phosphate-modified mesoporous TiO₂ incorporated in a silica matrix and their photocatalytic properties in the photodegradation of Congo red

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ABSTRACT: This study describes the development of mesostructured TiO₂ photocatalysts modified with PO₄³⁻ to improve its specific surface area and reduce the recombination rate of the electron–hole pairs. The mesoporous photocatalyst was successfully incorporated into a high specific surface area silica matrix by the hydrolysis reaction of tetraethyl orthosilicate (TEOS). Pluronic 123 and phosphoric acid were used as the directing agent for the structure of the mesoporous TiO₂ and as a source of phosphorus, respectively. TiO₂, P/TiO₂, TiO₂–SiO₂ and P/TiO₂–SiO₂ materials were characterized by BET, XRD, TEM-EDS, FTIR and UV-vis DRS measurements. The photoactivity of TiO₂–SiO₂ nanocomposites containing 15 wt.% photocatalyst/silica was evaluated in the degradation reaction of anionic dyes with UV radiation. The proposed nanomaterials showed high potential for applications in the remediation of wastewater, being able to reuse in several cycles of reaction, maintaining its photoactivity and stability. The separation and recovery time of the material is reduced between cycles since no centrifugation or filtration processes are required after the photooxidation reaction.

KEYWORDS: photocatalysis; phosphated TiO₂; TiO₂–SiO₂; Congo red dye

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1 Introduction

Titanium dioxide (TiO₂) is the most studied semiconductor in photocatalytic reactions and as support of catalysts [1–4]. Its safety, ease of use, resistance to photo-induced corrosion and its low cost are some of the important properties of this material. Because the commercial TiO₂ has a low surface area and a small particle size (52 m²/g and 12 nm), the recovery of the reaction mixture is difficult, so, for the different applications expected for TiO₂, an increase of the surface area is desirable, although a complete ordered structure is not determinant, due the selectivity is not related with the shape in most of its applications [5–6]. In recent years, the synthesis of mesoporous TiO₂ under different conditions has been extensively studied [7–11]. Mesoporous TiO₂ is a semiconductor that exhibits high photoactivity, related to its large specific surface area, and its ability to activate photocatalytic reactions [12–14]. The thermal treatments at high temperatures necessary to achieve the most active phase of TiO₂ (anatase) generally induce at the recrystallization of the mesoporous structure, as well as the decrease of surface area due to the sintering of the material and growth of the crystal [15].

Among the most used methods to avoid the sintering of the mesoporous structure of TiO₂, doping with phosphorus has attracted attention in recent years [15–20]. Körösi et al. [17] prepared a series of samples of phosphate-modified TiO₂ by the sol-gel method. They found that surface phosphate delays the formation of the anatase phase, the growth of the crystallites, and the transformation of the anatase-rutile phase. Lin et al. [18] showed that phosphorous doping can effectively inhibit grain growth and increase the specific area of TiO₂ nanoparticles.

The immobilization of TiO₂ on various substrates is an important research topic for photocatalytic treatment applications of wastewater [21]. The main objective is to avoid the separation difficulties associated with the powder form of the TiO₂ catalyst, other additional advantages of the immobilized systems are: higher specific surface area, higher adsorption properties [22–23], increase of surface hydroxyl groups and reduction of charge recombination

(electron-hole) [24].

One of the most studied supports for the immobilization of TiO₂ is silicon dioxide (SiO₂), which is characterized by a large specific surface area, high thermal stability and good sedimentation capacity. Chen et al. [25] reported that the large specific area of SiO₂ potentially improves activity by increase surface available for the reagent, and thus, improving adsorption [26]. In addition, the transparency of SiO₂ allows the photons to penetrate into the mass of the photocatalyst.

In the present work, 15 wt.% of photoactive mesoporous TiO₂, with phosphate and without phosphate were dispersed in a SiO₂ matrix formed by the hydrolysis and condensation reactions of tetraethyl orthosilicate (TEOS) to prepare P/TiO₂-SiO₂ nanocomposites and TiO₂-SiO₂. Silica was selected as matrix to incorporate the photocatalyst, since it presents good light transmission. The structural, optical and chemical properties of the materials related with the photooxidation of anionic dyes in aqueous solution under ultraviolet (UV) irradiation and the reuse of materials in the photodegradation process were investigated.

2 Materials and methods

2.1 Materials

All chemicals were analytical grade and used without further purification in the experiments. Titanium(IV) isopropoxide (Merck) and tetraethyl orthosilicate (Aldrich) were used as the sources for Ti and Si respectively. Triblock copolymer (Pluronic P123, Mw = 5800, EO₂₀PO₇₀EO₂₀; Aldrich) and phosphoric acid (J.T. Baker, 85%) were used as the structure directing template and phosphate source respectively. Distilled water and ethanol (J.T. Baker) were used for the preparation of catalyst. Congo red (CR) dye (Aldrich) was used as the test molecule for activity study.

2.2 Synthesis of photocatalysts

2.2.1 Preparation of mesoporous P/TiO₂

Mesoporous P/TiO₂ materials were synthesized by the sol-gel method, using soft template mechanism with Pluronic 123 as the structure directing agent. Pluronic 123 (2g) was dissolved in absolute ethanol (30 mL), then, titanium(IV) isopropoxide (30 mL) were added to it and the mixture was stirred for 2 h. Phosphoric acid (0.3 mL) was added to the

solution to prepare TiO₂ samples with P/TiO₂ ratio of 0.2 (wt.%). The resulting suspension was stirred for 2 h, followed by the addition of deionized water (8 mL) and the stirring continued for another 24 h. The synthesis of the phosphate-free material was performed as described without the addition of phosphoric acid. The prepared samples were calcined at 400°C for 6 h at a heating rate of 1°C/min. The materials were labeled as P/TiO₂ and TiO₂.

2.2.2 Incorporation of P/TiO₂ and TiO₂ photocatalysts into silica matrix

The previously obtained photocatalysts P/TiO₂ and TiO₂ were incorporated into a silica matrix to the weight ratio $w(\text{TiO}_2)/w(\text{SiO}_2)$ of 15/85, respectively. The procedure is detailed below:

In separate containers for each photoactive material (P/TiO₂ and TiO₂), TEOS and ethanol were mixed in 1:4 molar ratio. The resulting mixture was kept under stirring for 2 h, then, the corresponding mass of each material was added, keeping constant agitation for 1 h, the amount of deionized water required to achieve a final TEOS, ethanol and water ratio of 1:4:2.5 was added, the pH of the solution was adjusted to 5 with HCl, and finally, the temperature was raised to 60°C while maintaining the system at reflux for 12 h. The prepared materials are recovered by filtration, washed with deionized water, dried at 70°C for 24 h, and finally, calcined at 400°C. The synthesized materials were labeled P/TiO₂-SiO₂ and TiO₂-SiO₂. In the nomenclature used P refers to the phosphate material, TiO₂ to the mesoporous material and SiO₂ to the silica matrix.

2.3 Characterization of photocatalysts

The X-ray diffraction (XRD) patterns were recorded on a Siemens D500 X-ray diffractometer using Cu K α radiation as the X-ray source, in the 2θ range from 10° to 70°. The low-angle XRD patterns of the materials were recorded on a Bruker D8 advanced powder X-ray diffractometer using Cu K α ($\lambda = 1.5418 \text{ \AA}$) as the radiation source in the 2θ range of 0.5°–6°. The UV-visible diffuse reflectance spectroscopy (UV-vis DRS) patterns of samples were obtained using a Varian CARY 100 UV-vis spectrophotometer over a range of 200–600 nm equipped with an integrating sphere and with BaSO₄ as the reference. The N₂ adsorption-desorption isotherms were measured at 77 K on a Micromeritics ASAP 2000 analyzer. Prior to each adsorption measurement, the samples were degassed for 2 h at 200°C. The specific surface area was calculated

by Brunauer-Emmett-Teller (BET) method, and the pore size distribution was calculated using the Barrett-Joyner-Halenda (BJH) method. The Fourier transform infrared spectroscopy (FTIR) results of the samples were recorded in the range of 4000–400 cm⁻¹ on a Perkin Elmer SPECTRUM GX FTIR spectrometer. The morphology of the materials was examined by scanning electron microscopy (SEM) using a SEM-JEOL, JSM-5600 model. The accelerating voltage of the SEM was 20 keV. Transmission electron microscopy (TEM) images were recorded using a JEOL TEM-3010 electron microscope operated at an accelerating voltage of 300 keV.

2.4 Photocatalytic activity

The photodegradation of CR dye was performed in aqueous medium in a photocatalytic reactor. In a typical procedure 200 mg of the photocatalyst was added to 100 mL of CR dye solution (50 mg/L) and the resultant mixture was stirred for 60 min in the dark to ensure adsorption-desorption equilibrium in the probe molecule. The reactor temperature was kept constant at 30°C in all determinations.

The solution was then irradiated with a UV lamp (Pen Ray 2.4 mW, $\lambda = 254 \text{ nm}$) protected with an immersed quartz tube in the solution, the solution was kept under constant stirring. At periodic intervals of 30 min, about 5 mL aliquot were taken to monitor dye degradation from the characteristic absorption band at 498 nm using a Varian Cary 100 UV-vis spectrophotometer in the range of 190–650 nm. The quantification of the total organic carbon was used to determine the extent of mineralization of CR dye using a Shimadzu TOC-V CSN analyzer.

3 Results and discussion

3.1 Characterization results

3.1.1 XRD

XRD diffractograms of TiO₂, P/TiO₂, TiO₂-SiO₂ and P/TiO₂-SiO₂ are shown in Fig. 1(a). The TiO₂ and P/TiO₂ samples present the characteristic peaks of the anatase phase (JCPDS, No. 21-1272) at 2θ angles of 25.4° (101), 37.8° (004), 48.1° (200), 54.5° (105) and 62.4° (204), and no characteristic peak of the rutile phase at 2θ angle of 27.4° (110) was observed. The material synthesized only with structure director P123, has an intense main peak of the anatase phase, which is related to the crystallinity of the

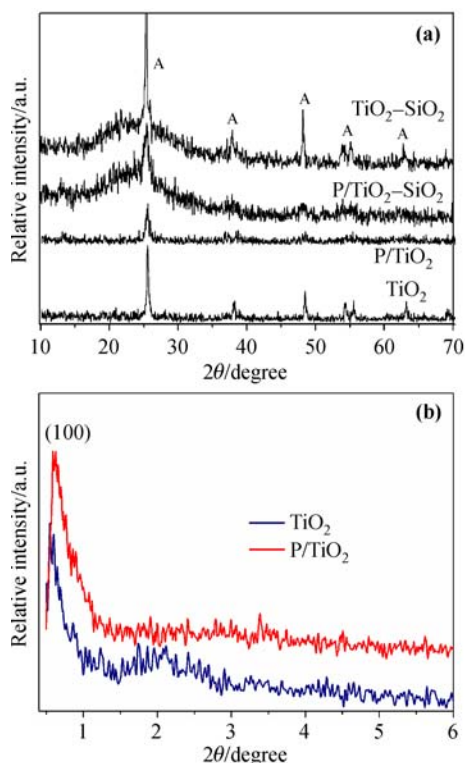


Fig. 1 (a) XRD patterns of TiO₂, P/TiO₂, P/TiO₂-SiO₂ and TiO₂-SiO₂ samples. (b) Low-angle XRD patterns of TiO₂ and P/TiO₂ samples.

material. These results agree with reported work showing that the addition of P123 structure director has a direct impact on the formation of the anatase phase and the inhibition of rutile phase formation [27]. The diffractogram of the TiO₂ material with addition of phosphoric acid, P/TiO₂ shows a decrease in the intensity of the main peak of the anatase phase at 2θ angle of 25.4° and the half-width of the diffraction peak is broadened, indicating a decrease in the crystallinity of the material and the formation of smaller TiO₂ crystals. The average crystal size was calculated using the Scherrer equation [28]. From the amplification of this peak, corresponding to the (101) plane, the calculated values were 11.4 and 9.1 nm for the materials TiO₂ and P/TiO₂, respectively. As can be observed, the crystallite size became smaller in phosphated material. This result suggests that phosphatation inhibits to a greater degree the growth of TiO₂ crystallites.

For the diffractograms of TiO₂-SiO₂ and P/TiO₂-SiO₂ silica materials, the characteristic amorphous SiO₂ signal was observed, and it consist on a broad pattern of the diffractogram in the 2θ range of 10°–35°. After inclusion of the TiO₂ and P/TiO₂ materials, the characteristic peaks of the anatase phase of TiO₂ are clearly observable, this is

attributed to the TiO₂ oxides being homogeneously distributed throughout the photocatalyst which can increase contact and the interactions with the molecules of the contaminant along the photocatalytic reaction, also suggests a good coupling of the photocatalysts in the substrate.

The low-angle XRD patterns of TiO₂ synthesized only with P123 and P123 with phosphate are presented in Fig. 1(b). Both samples show a strong diffraction peak in the low-angle region corresponding to the (100) plane, indicative of the microstructure presence. The intensity of the reflection 100 in the TiO₂ material is lower in comparison with the P/TiO₂ material, and this behavior is related to the decrease in the order of the mesopores in TiO₂ material [27]. The decrease in the mesostructure of TiO₂ compared to the P/TiO₂ material is related probably because of the absence of phosphorus in this material.

It has been reported that wormhole-like mesostructures often display a very intense low-angle XRD peak compared to the other diffraction peaks. The wormhole-like channels that are more or less regular in diameter can form three-dimensional (3D) mesoporous structures. Similar systems have been observed for disordered mesoporous TiO₂ [29].

3.1.2 Energy dispersion X-ray spectroscopy (EDX)

EDX analysis of the samples is presented in Fig. 2. The spectrum of sample P/TiO₂ presented in Fig. 2(a) shows the characteristics signals for Ti, O and the distinctive signal for P at 4.5, 0.5 and 2.1 keV, respectively.

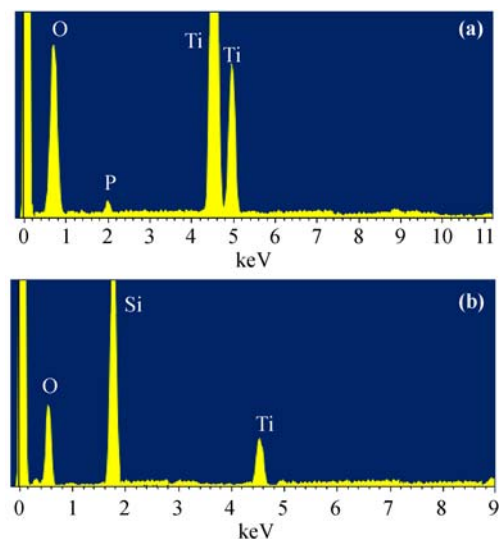


Fig. 2 EDX spectra of (a) P/TiO₂ and (b) P/TiO₂-SiO₂ samples.

EDX spectrum of the material immobilized on the SiO₂ substrate is presented in Fig. 2(b). Characteristic signals for Si at 1.7 keV, O at 0.5 keV and a lower intensity signal at 4.5 keV, attributed to Ti element can be observed.

The presence of titanium in this material suggests that nanoparticles observed in TEM images (see in Fig. 6) correspond to TiO₂ incorporated in a silica matrix; the phosphorus signal is not observable due the silicon signal.

3.1.3 N₂ adsorption–desorption isotherms (BET)

Figure 3 shows the N₂ adsorption–desorption isotherms of the TiO₂, P/TiO₂, TiO₂–SiO₂, P/TiO₂–SiO₂ materials. The isotherms of the TiO₂ and P/TiO₂ materials are type IV with a hysteresis loop type H₂ evidencing the presence of mesoporous, good connectivity between the pores and the presence of pores with narrow mouths according to the IUPAC classification [30].

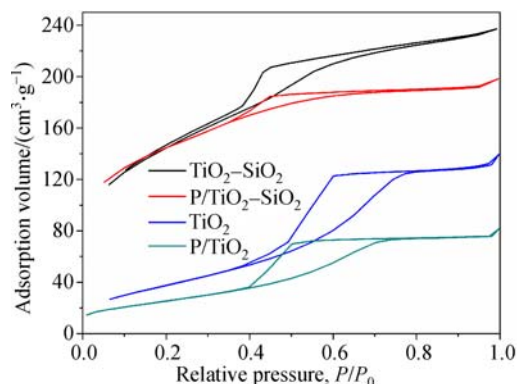


Fig. 3 N₂ adsorption–desorption isotherms of TiO₂, P/TiO₂, TiO₂–SiO₂ and P/TiO₂–SiO₂ materials.

The specific surface area, pore volume and pore diameter are presented in Table 1. The material synthesized with phosphoric acid has a higher surface area than the non-phosphorus TiO₂ material which is attributed to the fact that the incorporation of the phosphate anion was able to stabilize the mesoporous structure of the P/TiO₂ material, avoiding further sintering due to the thermal treatment. The type IV isotherms of the materials incorporated in the silicon matrix TiO₂–SiO₂ and P/TiO₂–SiO₂ indicate that

Table 1 The textural parameters of samples

| Catalyst | Specific surface area /(m ² ·g ⁻¹) | Pore volume /(cm ³ ·g ⁻¹) | Pore diameter /nm |
|--------------------------------------|--|---|----------------------|
| TiO ₂ | 112 | 0.44 | 7.7 |
| P/TiO ₂ | 141 | 0.14 | 7.1 |
| TiO ₂ –SiO ₂ | 529 | 0.34 | 4.1 |
| P/TiO ₂ –SiO ₂ | 497 | 0.29 | 3.4 |

the materials preserve the mesoporous structure of the previously synthesized TiO₂ and P/TiO₂ materials. The specific surface area of the materials was 529 and 497 m²/g for P/TiO₂–SiO₂ and TiO₂–SiO₂, respectively. These values of specific surface area suggest that the photocatalytic material was successfully dispersed in the translucent silica matrix.

3.1.4 UV-vis DRS

UV-vis DRS results of TiO₂, P/TiO₂, TiO₂–SiO₂ and P/TiO₂–SiO₂ samples are included in Fig. 4. The band gap (E_g) values of the materials were obtained with the modified Kubelka–Munk function from the absorption spectra, assuming an indirect transition for TiO₂ as has been reported in numerous articles [31–32]. This method consists in plot of the modified Kubelka–Munk function $[F(R) \cdot hv]^{1/2}$ versus hv , where $F(R) = (1-R)^2/(2R)$ is calculated from the reflectance (defined as $R = (1 - \text{Absorbance})$) and hv is the photon energy in eV [32]. The band gap value is calculated by extrapolating the linear part of the function: the intersection with the x -axis corresponds

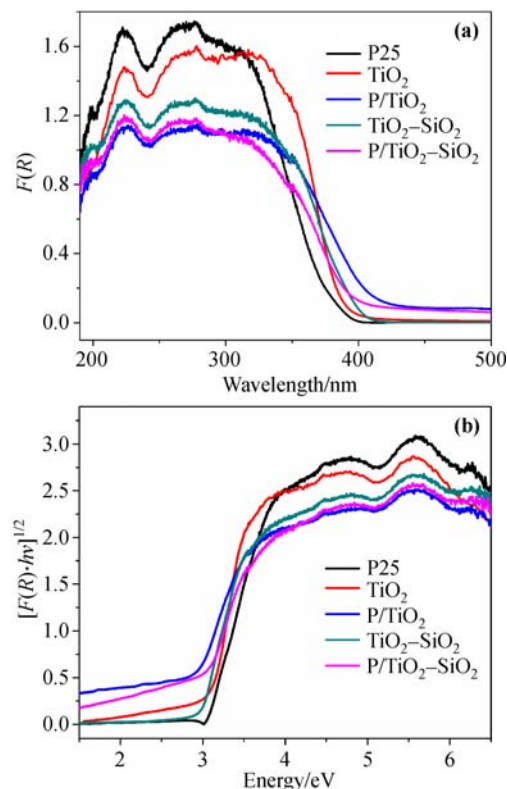


Fig. 4 (a) Diffuse reflectance spectra and (b) modified Kubelka–Munk spectra of the TiO₂, P/TiO₂, TiO₂–SiO₂ and P/TiO₂–SiO₂ materials.

to the band gap energy. All the materials show a semiconducting behavior that consists of a rise in the absorption edge.

According to Fig. 4(a) a signal corresponding to the isolated Ti at about 225 nm is observed in the spectra of the materials. The absorption peak of 200–260 nm can be attributed to the transfer of electrons from the O₂⁻ to Ti⁴⁺ ions of the TiO₂ unit with tetrahedral coordination of the photocatalysts. Yamashita et al. reported that TiO₂ with tetrahedral coordination can be chemically supported on silica matrices and has high photocatalytic activity [33].

The TiO₂ sample shows an edge with absorption around 400 nm, while the P/TiO₂ material showed a shoulder in the range 410–430 nm, reflecting a small displacement in the absorption towards the region of visible radiation, which is attributed to the phosphorus of the material P/TiO₂ [17,27]. The band gap energy values (E_g) calculated for non-included photocatalysts were 3.10 and 2.84 eV for TiO₂ and P/TiO₂, respectively (Fig. 4(b)), which are lower than 3.20 eV evaluated by the reference material Degussa P25. The values calculated for the materials synthesized with 15 wt.% TiO₂ were 3.16 and 3.04 eV for TiO₂-SiO₂ and P/TiO₂-SiO₂, respectively. This observed shift towards lower energy in the photocatalysts incorporated in silica can be indicative of the good incorporation of the material in the support.

3.1.5 FTIR measurements

Figure 5 depicts FTIR spectra of TiO₂, P/TiO₂, TiO₂-SiO₂ and P/TiO₂-SiO₂ materials. The materials TiO₂ and P/TiO₂ show a peak in the region of low energy 500–700 cm⁻¹, which is assigned to Ti–O vibrations and Ti–O–Ti bonds, typical of TiO₂. The presence of Ti ions in a tetrahedral coordination described in the UV-vis DRS

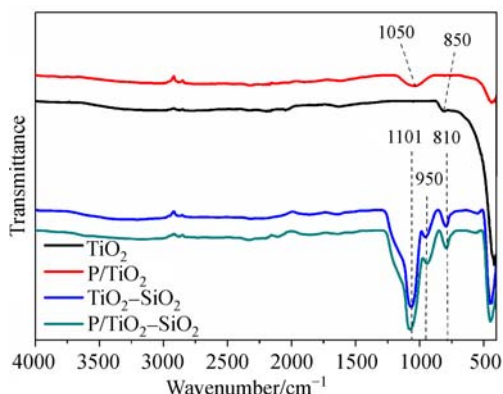


Fig. 5 FTIR results of TiO₂, P/TiO₂, TiO₂-SiO₂ and P/TiO₂-SiO₂ materials.

results are very efficient in the adsorption of water [27]. The band located at 850 cm⁻¹ in the TiO₂ material is assigned to the Ti–O–Ti stretch vibration in octahedral coordination. The signal located at 1180–910 cm⁻¹ observed in the P/TiO₂ material is related with the characteristics stretching vibrations of PO₄³⁻ groups. The band at 1050 cm⁻¹ is related to bound phosphates in bidentate form, the formation of Ti–O–P species is characteristic of a signal at 1125 cm⁻¹.

For the materials TiO₂-SiO₂ and P/TiO₂-SiO₂, the band at 1101 cm⁻¹ and the shoulder at 1193 cm⁻¹ are associated with asymmetric stretching vibrations of siloxane groups (Si–O–Si). The bands at 470 and 810 cm⁻¹ are assigned to symmetrical siloxane groups (Si–O–Si). The 950 cm⁻¹ band is associated with Si–OH groups and the presence of silanol groups [34].

The results of FTIR reveal that by addition of phosphoric acid to the TiO₂ mixture, the phosphate ions can react with the OH groups on the surface of TiO₂, resulting in the formation of bonds with the surface. The presence of the groups P–O–Ti, P=O and P–OH indicate that they may be present in the surface as mono-, bi- and tridentate phosphate units.

Additionally, several authors have reported the formation of strong bonds between the bidentate form of phosphates and the TiO₂ surface [17,35]. Connor et al. [35] reported that phosphate species bind to TiO₂ predominantly in bidentate form in neutral and acidic aqueous solution. In our case, the reaction mixture had a pH of about 6.1, therefore the interaction in a form of bidentate bond through P=O is more favorable in P/TiO₂ photocatalyst.

3.1.6 TEM

TEM images of the materials are presented in Fig. 6. Figures 6(a) and 6(b) show evidence of non-agglomerated crystalline domains of size in the range of 10–20 nm. These crystallites are surrounded by amorphous silica matrix visualized by low contrast grey slabs and spots. Figure 6(c) shows crystalline zones, which correspond to the main series of crystalline interplanar distances measured from HRTEM images (Fig. 6(d)) resulting in: $d_1 = 0.354$ nm, which conforms to the anatase TiO₂ polymorph (101) [36].

The crystallite size of the materials estimated from the TEM images agrees with the sizes calculated using the Scherrer equation in the XRD patterns. Figure 7 shows the distribution of O, Ti, and Si; the elements Ti and Si incorporated are uniformly distributed in the matrix, which

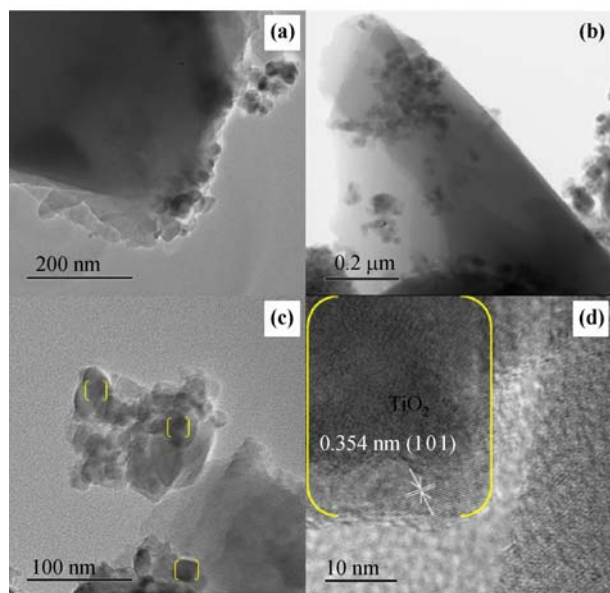


Fig. 6 (a)(b)(c) TEM images and (d) HRTEM image of the sample $\text{TiO}_2/\text{SiO}_2$.

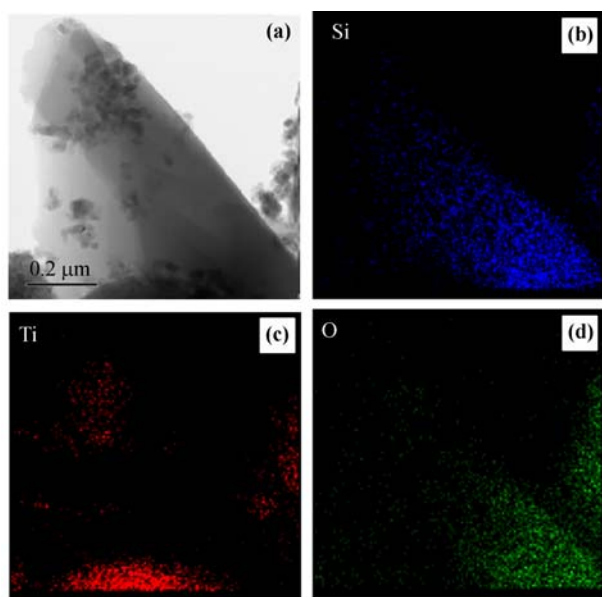


Fig. 7 (a) TEM image and (b)(c)(d) elemental mapping of the $\text{TiO}_2\text{-SiO}_2$ material.

is indicative that the synthesis method avoided the agglomeration of TiO_2 and allowed the incorporation of the photocatalyst into the silica matrix.

3.2 Photocatalytic activity

The photocatalytic activity of the samples was analyzed by the degradation of CR dye in the presence of UV light. Before assessing and comparing the activities, some

control experiments were performed. By means of experiments it is concluded that less than 5% of CR dye decomposes after 4 h in the absence of the photocatalyst or the light irradiation (photolysis) and can therefore be discarded in comparison with the degraded dye through the photocatalysis. The photocatalytic activity of the reference material Degussa P25 was measured under conditions identical to those detailed in the experimental section in order to compare the activity of our materials.

Figure 8 presents the comparative degradation of CR using P/TiO_2 , TiO_2 , $\text{P/TiO}_2\text{-SiO}_2$, $\text{TiO}_2\text{-SiO}_2$ and Degussa P25 materials. When exposed to the radiation, $\text{P/TiO}_2\text{-SiO}_2$ and $\text{TiO}_2\text{-SiO}_2$ gave complete degradation of CR within 4 h. From the graph, it was noted that P/TiO_2 , TiO_2 , $\text{P/TiO}_2\text{-SiO}_2$, $\text{TiO}_2\text{-SiO}_2$ and Degussa P25 materials were able to degrade 98%, 95%, 88%, 82% and 93% of CR dye, respectively. Moreover, among all the samples, the maximum photocatalytic activity was observed with the P/TiO_2 photocatalyst.

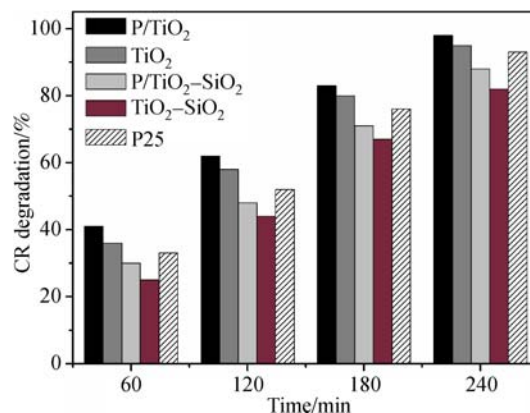


Fig. 8 Percentages of degradation of aqueous CR dye with materials.

The photocatalytic performance of $\text{TiO}_2\text{-SiO}_2$ and $\text{P/TiO}_2\text{-SiO}_2$ materials showed to be very close to that of bare commercial TiO_2 Degussa P25. It is important to note that the high activity of the photocatalysts incorporated in the silica matrix shown in Fig. 9 is achieved using a weight ratio $w(\text{TiO}_2)/w(\text{SiO}_2)$ of 15/85 in the final $\text{TiO}_2\text{-SiO}_2$ and $\text{P/TiO}_2\text{-SiO}_2$ materials. The photocatalytic activity of materials immobilized in SiO_2 are closely related to its large surface area which may offer more sites to absorb the hydroxyl groups generated during the reaction and the ease of diffusion of the reactants in the mesopores of $\text{TiO}_2\text{-SiO}_2$ and $\text{P/TiO}_2\text{-SiO}_2$ materials. Furthermore, the oxidative reaction of CR dye is believed to be initiated by $\cdot\text{OH}$ radicals [37–38], therefore the surface-adsorbed water and

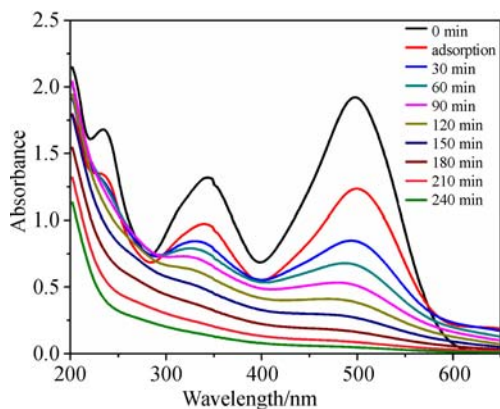


Fig. 9 UV spectra of the aqueous CR dye degradation with the P/TiO₂-SiO₂ photocatalyst.

hydroxyl groups can react with photogenerated holes on the catalyst surface and produce hydroxyl radicals, which are powerful oxidants in degrading organics [27].

Figure 9 shows UV spectra of the CR dye at different reaction times with the P/TiO₂-SiO₂ photocatalyst. The adsorption stage of dye on the surface of photocatalyst was recorded in dark conditions, with a value of 25% of the initial concentration of the solution (50 mg/L). Since the pre-adsorption of the dye on the surface of the photocatalysts is an important factor for the photodegradation of contaminants under light irradiation; a higher adsorption will improve the degradation. In addition, the specific surface of SiO₂ matrix will contribute to improve the accessibility of the dye to the surface, and the radiation will penetrate more efficiently to influence a larger number of Ti⁴⁺ metal cations, generating more electron-hole pairs (e⁻-h⁺) [25].

Figure 9 show that the intensity of the main absorption band at 484 nm due to the chromophore decreased 98% in 240 min of photoreaction in the presence of the P/TiO₂-SiO₂ photocatalyst, because of the degradation of CR dye.

In the absorption spectrum, no new band is present in the UV-vis region; this observed behavior is attributed to the photodegradation of the chromophore, responsible of the characteristic CR dye color.

Figure 10 shows the evaluation of the total organic carbon (TOC) with the catalyst P/TiO₂-SiO₂. The amount of TOC decreases to a 34% during 240 min of reaction, indicating the mineralization of the CR dye molecule to simple molecules (CO₂ and H₂O) by the action of photocatalyst.

Figure 11 shows that the photocatalytic degradation of CR dye follows the pseudo-first order kinetics represented by Eq. (1):

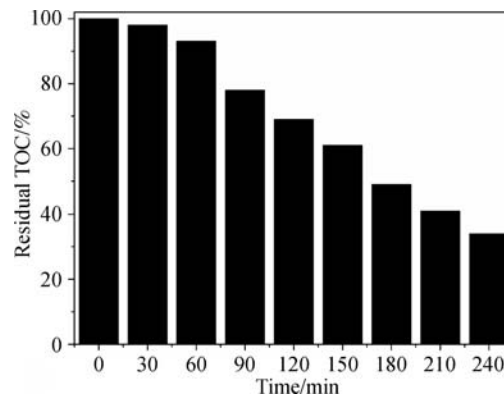


Fig. 10 The residual TOC in the photodegradation of aqueous CR dye with the P/TiO₂-SiO₂ material after 4 h.

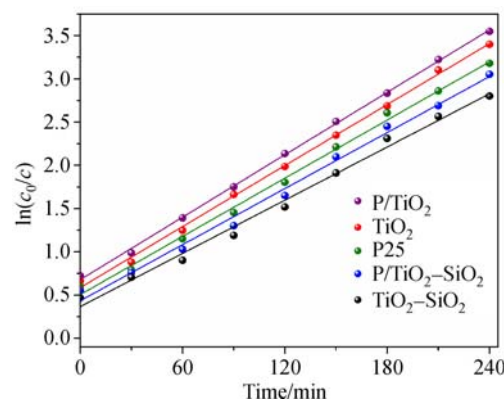


Fig. 11 Plots of the pseudo-first order kinetics of CR photodegradation.

$$\ln\left(\frac{c_0}{c}\right) = k_{\text{app}}t \quad (1)$$

where c_0 is the initial concentration of the dye, c is the concentration at time t , and k_{app} is the apparent rate constant.

A plot of $\ln(c_0/c)$ as a function of the reaction time provides the apparent rate constants for the CR dye degradation. From the slope of the curve fitting line and the intersection point equal to zero, the apparent rates constants was found to be 0.0120, 0.0116, 0.0109, 0.0104 and 0.0098 min⁻¹ for P/TiO₂, TiO₂, P25, P/TiO₂-SiO₂ and TiO₂-SiO₂, respectively.

3.3 Mechanism of the CR dye photodegradation

The absorption spectrum of the CR dye solution is characterized by a band in the visible region with a maximum absorption at 496 nm related to the azo bonds of the CR dye molecule and by two other bands in the UV region located at 235 and 338 nm (Fig. 9), attributed to the

benzene and naphthalene ring structures, respectively [39]. Throughout the course of the CR dye photooxidation reaction, the absorbance decreases and no peaks are observed at 496 nm after 240 min of reaction, almost complete degradation of the dye was achieved, which can be attributed to the breaking of the azo bonds by attack of the generated species in the photocatalyst [40]. The decrease in absorbance in the bands located at 235 and 338 nm can be ascribed to the degradation of the aromatic groups in the CR dye molecule, as well as their intermediates generated since no new peaks in the spectrum are observed. The reduction rates of the three characteristic peaks of the test molecule in the presence of the composite material that presented the best photocatalytic performance P/TiO₂-SiO₂ can be compared in terms of their pseudo-first order rate constants [40–41]. In Fig. 12, the highest decrease of the visible band, $k_{app} = 0.0141 \text{ min}^{-1}$, is attributed to the breakage of azo bonds during discoloration and photocatalytic degradation. Thus, it can be concluded that azo bonds N=N are more susceptible to degradation by hydroxyl radicals than the aromatic rings resulting in the rapid disappearance of the chromophore in the CR dye structure.

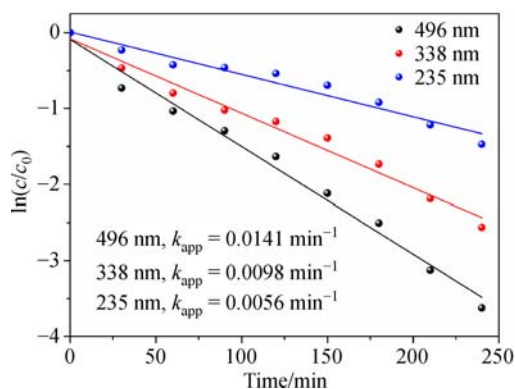


Fig. 12 Photocatalytic behavior of characteristic peaks of CR during the irradiation time with UV light.

The values of k_{app} at 235 and 338 nm are 0.0056 and 0.0098 min^{-1} , respectively, which indicate that the aromatic structures of CR dye were eliminated during the reaction, because of the TiO₂ photocatalysts incorporated into the SiO₂ matrix. This is in agreement with the results obtained in the TOC study, with a significant decrease of the organic groups.

In the case of the P/TiO₂ material, the modification of the TiO₂ structure with phosphate ions seems to be responsible for the improved CR photooxidation. On the one side, the phosphate ions on the surface of the catalyst, can reduce the

recombination rate of the photogenerated pairs (e^-h^+), due to their ability to trap holes, which in turn are capable of converting OH groups, in species highly reactive ($\cdot\text{OH}$), which react with the adsorbed organic compounds, resulting in improved photooxidation. Additionally it improves the structural properties of TiO₂ (stabilization of the anatase phase and improves surface area).

The SiO₂ support can affect both, the conservation of the gap at the interface and the photon losses due to absorption in the volume of the photocatalyst, the Ti centers of the material localized in the surface or at the surface layer which facilitates the reactions of charge transfer with the molecules of the adsorbed contaminant. The synthesis of the nanomaterial composed of P/TiO₂ particles previously synthesized and deposited in the SiO₂ matrix with a relatively low concentration (15 wt.%), allows the coverage of only a small fraction of the internal surface of the support, as a result, the particles are not agglomerated during the synthesis process, but may react with the hydroxyl groups that are located on the surface of the pores along the photocatalytic reaction. These factors can explain for the increased activity observed in TiO₂-SiO₂ and P/TiO₂-SiO₂ materials. A proposal of the before mentioned model is presented in Fig. 13.

3.4 Recycling test for the P/TiO₂-SiO₂ material

The recycling of the catalyst in photocatalytic applications is an importance topic, since it can contribute towards to the reduction of costs of the dye wastewater treatment. The experiments were run in consecutive tests once the catalyst was recovered by filtration and a fresh amount of contaminant solution was placed in the reactor. It is important to mention the absence of subsequent treatments such as washes or heat treatments to carry out the cleaning of active sites on the surface of the catalyst. Figure 14 shows the test of photocatalyst P/TiO₂-SiO₂. A degradation of 48.3 mg/L of the initial concentration of CR dye (50 mg/L) elapsed the first cycle and a degradation of 39.8 mg/L after the fifth cycle is observed. The photodegradation efficiency observed during the entire recycling test indicates that the photocatalyst P/TiO₂-SiO₂ shows stability, without aggregation and retaining its surface reactivity properties.

The recycling of the nanocomposites is derived from the separation of the SiO₂ layers, which prevent the aggregation of the active centers and the breakdown of the composite structure during the catalytic reactions. The recycling experiments results suggest the practical applic-

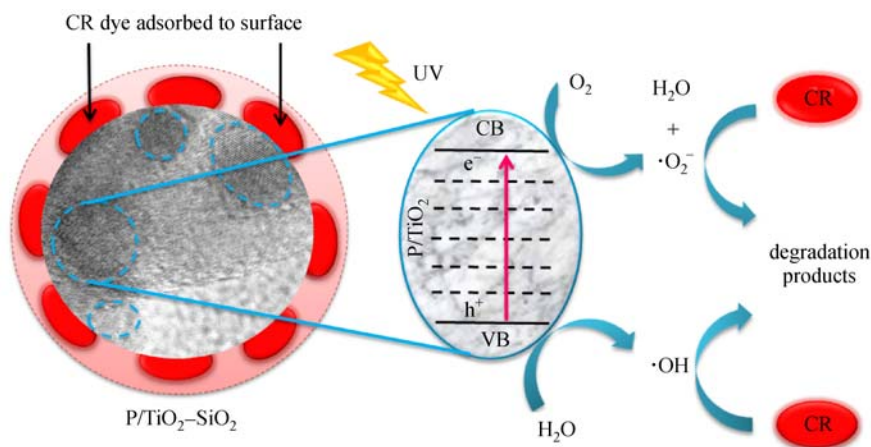


Fig. 13 Schematic illustration of the proposed mechanism for photodegradation of CR dyes by the electron-hole model.

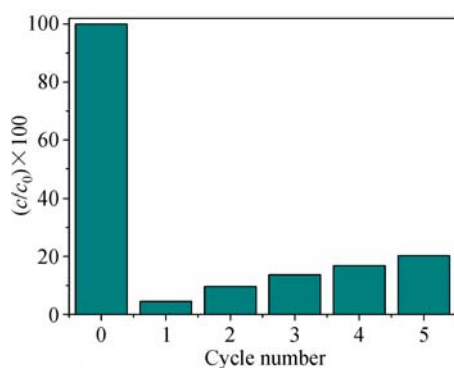


Fig. 14 Recycling and photocatalytic stability studies with the P/TiO₂-SiO₂ material for five cycles under the same experimental conditions.

ability of the TiO₂ materials incorporated in SiO₂ in environmental applications such as water remediation. Finally by stopping the stirring of the solution in the reactor, the photocatalysts P/TiO₂-SiO₂ sediment 90%–95% after 25 min of the reaction is over, while the reference material P25 continues in the mixture of reaction. This feature of the materials proposed in this work, allows the reuse more efficiently than commercial photocatalysts commonly use.

4 Conclusions

The incorporation of the structure director and the phosphoric acid slow the formation of the anatase phase, avoided the collapse of the mesoporous structure and increased the specific surface area of the photocatalysts. Furthermore phosphate modification of the TiO₂ enhances charge separation.

The TiO₂ materials incorporated in the silica matrix showed good adsorption capacity and a high degree of dispersion of the active phase.

The rate of photodegradation of CR dye using materials containing 15 wt.% TiO₂-SiO₂ was similar to that of the Degussa P25 material, which is a relevant factor in large-scale environmental remediation processes. The specific surface area and the adsorption capacity of the P/TiO₂ and TiO₂ materials included in the SiO₂ matrix contributed significantly to the increase in the rate of the CR dye photodegradation. The results of this work show that the materials included in the SiO₂ matrix presented good photoactivity, sedimentation, recovery and recycling capacity in successive cycles of reaction; factors that are of great importance for the use of large-scale photocatalytic systems.

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